# **RSC Advances**



# **PAPER**

View Article Online
View Journal | View Issue



Cite this: RSC Adv., 2020, 10, 7600

# Relationship between the structure and catalytic performance of MoS<sub>2</sub> with different surfactant-assisted syntheses in the hydrodesulfurization reaction of 4,6-DMDBT

Suya Chu, D Weixia Zhou, Chenyang Zhang, Yi Zheng, Yang Liu and Yongjun Liu\*

Surfactants are important factors in the hydrothermal synthesis of MoS<sub>2</sub> with different morphologies. Herein, we report the synthesis of MoS<sub>2</sub> via the hydrothermal method combined with a single-source precursor with the assistance of different surfactants (CTAB/SDS/SDBS). The synthesis mechanisms of MoS<sub>2</sub> with different morphologies and their effects on 4,6-DMDBT in hydrodesulfurization (HDS) have been systematically studied. MoS<sub>2</sub>-CTAB was prepared by the adsorption of molybdate radicals, nucleation and formation. MoS<sub>2</sub>-SDS and MoS<sub>2</sub>-SDBS were synthesized via four steps, namely, adsorption, insertion, exfoliation and assembly, and the relationship between the morphology-structure-performance of MoS<sub>2</sub> in the hydrodesulfurization of 4,6-DMDBT was investigated. It was established that the desulfurization rate of MoS<sub>2</sub>, HYD ratio and selectivity of the MoS<sub>2</sub> increased in the order: MoS<sub>2</sub>-SDBS > MoS<sub>2</sub>-CTAB > MoS<sub>2</sub>-SDS, which exhibited a positive correlation with the average number of layers and dispersion, and a negative correlation with the average slab length and the ratio of the Mo edge/corner sites of MoS<sub>2</sub>. Among all the MoS<sub>2</sub>, MoS<sub>2</sub>-SDBS exhibited the best HDS performance.

Received 15th October 2019 Accepted 24th December 2019

DOI: 10.1039/c9ra08407j

rsc.li/rsc-advances

## Introduction

Energy is an important driving force for the development of human society and diesel is widely used as a primary energy source in the world at present. However, significant harm is done to the environment on burning diesel, which contains a large number of sulfur compounds.¹ Thus, controlling the sulfur content of diesel is needed to reduce air pollution and also satisfy the increasingly strict regulations.²,³ The removal of 4,6-dimethyldibenzothiophene (4,6-DMDBT), which is the most difficult sulfur compound of diesel oil, is highly desirable. To date, hydrodesulfurization (HDS) remains one of the principal techniques for achieving this goal.⁴-6 Molybdenum disulphide (MoS₂) is considered to be an excellent catalyst for the HDS reaction because of its active sites including edge and corner sites exposed by its layered structure.²

Different preparation methods have different effects on the morphology and structure of molybdenum disulphide, which further affect its catalytic activity. The methods of preparation of  $MoS_2$  have been quite developed, *e.g.*, the hydrothermal method, chemical vapor deposition (CVD), precipitation method and precursor thermolysis.<sup>8-12</sup> The morphology and crystallinity of  $MoS_2$  are substantially affected by different

Institution of Chemical Process and Intrinsic Safety, School of Chemical Engineering and Technology, Huaqiao University, Fujian, 361021, P. R. China. E-mail: yongjunliu@hqu.edu.cn

preparation methods. For example, Li *et al.*<sup>9</sup> synthesized hollow  $MoS_2$  microspheres composed of nanoflakes by the hydrothermal method. Lee *et al.*<sup>10</sup> used  $MoO_3$  and S as reducing agents to prepare single-layer  $MoS_2$  films by CVD. Hu *et al.*<sup>11</sup> synthesized  $MoS_x$  (x = 3–20) in an ethanol–water solution by the precipitation method, and it was calcined in hydrogen at different temperatures to obtain  $MoS_2$  with different morphologies. Wang *et al.*<sup>12</sup> synthesized  $MoS_2$  and  $MoO_3$  hierarchical nanostructures using a single-source molecular precursor.

Single-source molecular precursors refer to a single-source molecule containing all the elements required in order to form the final product, which means fewer irrelevant elements in reaction systems; 13 the use of such precursors combined with the hydrothermal method could gain products in a more moderate way. For example, Wang et al.12 prepared molybdenum sulfide nanoclusters using Mo((C<sub>2</sub>H<sub>5</sub>)<sub>2</sub>NCS<sub>2</sub>)<sub>2</sub>O as the single-source molecule by the hydrothermal method. Judging by current research, this combination method is relatively uncommon in the preparation of MoS2. Moreover, there is a lack of micro-level understanding of the HDS reaction of MoS<sub>2</sub> and related materials, and how to establish the morphologystructure-activity relationship between MoS2 and HDS for 4,6-DMDBT. In other words, the numbers of layers and active sites of MoS2 are estimated by characterization, and the catalytic performance, being directly and systematically related to HDS performance, is seldom studied. Therefore, we have proposed a strategy that combines the reaction product of silver

diethyldithiocarbamate (DDTC) and ammonium paramolybdate ( $(NH_4)_6Mo_7O_{24} \cdot 4H_2O$ ) as the single-source precursor with the hydrothermal method to prepare different morphologies of  $MoS_2$ , assisted by different surfactants. The products were characterized by XRD, SEM, FETEM, BET and Raman spectroscopy. All the different morphologies of  $MoS_2$  as catalysts showed different HDS activities and desulphurization rates with 4,6-DMDBT for varied micro-structures of  $MoS_2$ .

# Experimental

#### Materials

 $(NH_4)_6Mo_7O_24\cdot 4H_2O$ ,  $HO-NH_2\cdot HCl$ , diethyldithiocarbamic acid sodium salt (DDTC),  $NH_4OH$ , SDS, CTAB, SDBS, KI (AR) were purchased from Sinopharm Group Chemical Reagents Co. Ltd. 4, 6-DMDBT was obtained from Shanghai Vauquet, and decalin was obtained from Macklin. The above reagents were used without further purification.

#### Preparation of MoS<sub>2</sub>

 $MoS_2$  was prepared by hydrothermal synthesis with the assistance surfactants (CTAB/SDS/SDBS). In a typical procedure, the experiment consists of two steps. First, the precursor was synthesized by the following steps. (NH<sub>4</sub>)<sub>6</sub>Mo<sub>7</sub>O<sub>24</sub>·4H<sub>2</sub>O and DDTC, in a molar ratio of Mo: S = 1:4, were dissolved in 100 mL deionized water, stirred with a magnetic stirrer for 5 h then allowed to react for 12 h as the molybdenum source and sulfur source, respectively. The products were washed several times with deionized water and anhydrous ethanol and were then transferred and dried in a vacuum dryer at 60 °C for 5 h.

Second,  $MoS_2$  was prepared as follows. The precursor (1.275 g) and hydroxylamine hydrochloride (2.085 g) were dissolved in 60 mL of deionized water, and then SDS, CTAB and SDBS, three different kinds of surfactants, were respectively added to the hydrothermal system under vigorous stirring. The mixed solutions were then transferred to 100 mL hydrothermal reactors and heated at 220 °C for 24 h. After cooling to room temperature, black powders were obtained, which were washed several times with deionized water and ethanol. The final catalysts were obtained after drying under vacuum at 60 °C for 12 h and were denoted as " $MoS_2$ -X", where X indicates the different surfactants.

#### Characterization

The crystalline structures of the samples were detected by a SmartLab X-ray powder diffractometer (XRD) with Cu K $\alpha$  radiation operated at 40 kV and 30 mA; the diffraction angle (2 $\theta$ ) ranged from 5° to 80°, and the scanning rate was 10° min<sup>-1</sup>. XRD spectra were analyzed using MDI Jade 6.

Before the FT-IR analysis, the KBr was dried in an oven at 120 °C for 12 h. Small amounts of the dry samples were mixed with KBr and pressed into pellets, and were measured on an FT-IR spectrometer (NICOLET iS 50).

The surface morphologies of the samples were recorded by S-4800 scanning electron microscopy (SEM). Secondary electron imaging was conducted and the surface morphologies and

element distributions were obtained by observing the samples at low voltage.

High-resolution transmission electron microscopy (HRTEM) characterization was performed on a JEM-2100 transmission electron microscope. Before characterization, tiny samples were dispersed in absolute ethanol by ultrasound, and a small number of liquid droplets were put on the copper mesh to be tested after volatilization. We acquired 25 representative pictures, including 400 slabs and 150 independent layered areas obtained from different regions of each catalyst. The average stacking number  $(\Delta N)$  and average slab length  $(\Delta L)$  were respectively calculated using the following equations<sup>14</sup>

$$\Delta N = \frac{\sum_{i=1}^{n} x_i N_i}{\sum_{i=1}^{n} x_i}$$

$$\Delta L = \frac{\sum_{i=1}^{n} x_i L_i}{\sum_{i=1}^{n} x_i}$$

where i is the total number of  $MoS_2$  slabs.  $N_i$  and  $L_i$  represent the stacking layer number and length of  $MoS_2$ , respectively, and  $x_i$  is the number of  $MoS_2$  slabs with  $N_i$  layers of length  $L_i$ .

The Mo edge/corner ratio of the  $MoS_2$  slab  $((f_e/f_c)_{Mo})$  was gained by the following equation:<sup>15</sup>

$$\left(\frac{f_{\rm e}}{f_{\rm c}}\right)_{\rm Mo} = \frac{\rm Mo_{\rm e}}{\rm Mo_{\rm c}} = \frac{5\Delta L}{3.2} - 1.5$$

where  $Mo_e$  is the number of edge sites and  $Mo_c$  is the number of corner sites.

 $\rm N_2$  adsorption–desorption (BET) isotherms were obtained on a QUADRASORB evo-2 automatic microporous physical adsorption instrument. The principle is that the surface of the sample is physically adsorbed with nitrogen at a low temperature to calculate the specific surface area and pore size distribution. The specific surface area of the sample was calculated by the BET (Brunauer–Emmett–Teller) formula. The pore size distribution was modeled by BJH (Barrett–Joyner–Halenda) analysis. The samples were vacuum degassed at 250  $^{\circ}{\rm C}$  for 3 h before testing, and subsequently transferred to the analysis station for adsorption–desorption at -196  $^{\circ}{\rm C}$ .

Raman spectra were obtained on a confocal inVia micro Raman spectrometer produced by Renishaw, UK. Monochrome light was scattered on the sample and the structural information of the samples was gained after calculating the wavelength of the scattered light. Generally speaking, the wavelength was mainly located in the range of molecular vibration, rotation and electronic level transition in the molecular system.

#### HDS of 4,6-DMDBT on the prepared MoS<sub>2</sub>

A mixture of 4,6-DMDBT and decalin (0.4 wt% S) was used as the simulated diesel oil to test the catalytic activity of  $\text{MoS}_2$ ; the test was performed in a microform high-pressure autoclave (200 mL). First, 70 mL of simulated diesel oil and 0.6 g catalyst were

poured into the autoclave. The reactor was installed according to the operation and the gas tightness of the device was good before reaction, then hydrogen was injected three times into the autoclave to replace the air in the autoclave until it was discharged. The reaction conditions were as follows:  $\rm H_2$ , 3 MPa, 300 °C, 500 rpm, 5 h. After the reaction was finished, the power was turned off, and the gas discharged from the reactor was fed into the saturated NaOH solution when the temperature dropped to room temperature.

Analysis of the total sulfur content of the solution after the HDS reaction was tested on a WK-2D microcoulomb comprehensive instrument, Jiangfen, from Jiangsu. The hydrocarbon composition was determined using a Pano A90 gas chromatograph equipped with an HP-5 (30 m  $\times$  0.32 mm  $\times$  0.25  $\mu m)$  and FID detector for quantitative analysis; the qualitative analysis was carried out using a Trace 1310-ISQ LT GC-MS equipped with an HP-5.

The desulfurization ratio was calculated by the following equations:

$$D_{\rm s} = \frac{X_0 - X_{\rm s}}{X_0}$$

where  $X_s$  and  $X_0$  represent the total sulfur contents before and after the reaction, respectively.

The ratio of the product content of the HYD pathway to that of the DDS pathway was expressed by calculating the product selectivity S. The formula is as follows:

$$S = \frac{S_{\mathrm{HYD}}}{S_{\mathrm{DDS}}} = \frac{S_{\mathrm{THDMBCH}} + S_{\mathrm{DMBCH}} + S_{\mathrm{MCHT}}}{S_{\mathrm{DMBP}}}$$

where  $S_{\text{THDMBGH}}$ ,  $S_{\text{DMBCH}}$ ,  $S_{\text{MCHT}}$ ,  $S_{\text{DMBP}}$  represent the contents of THDMBCH, DMBCH, MCHT, DMBP in solution after HDS reaction.

#### Results and discussion

#### XRD results

The preparation of the precursors was the first step in the preparation of MoS<sub>2</sub>. To determine the composition of the

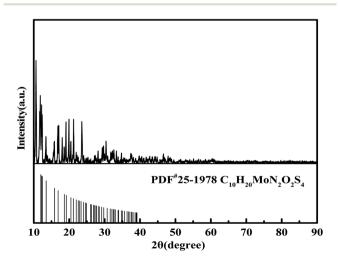


Fig. 1 The XRD spectra of the precursor.

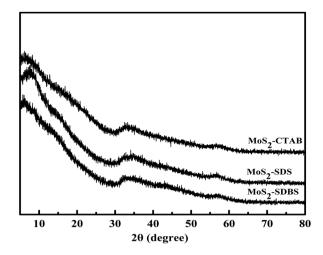


Fig. 2 XRD patterns of MoS<sub>2</sub>-CTAB, MoS<sub>2</sub>-SDBS, MoS<sub>2</sub>-SDS.

precursors, XRD characterization was conducted as shown in Fig. 1. Upon comparison with the standard card PDF 25-1978#, it was concluded that the precursor was MoO<sub>2</sub> (DDTC)<sub>2</sub>, the complex of ammonium paramolybdate and DDTC. The formation mechanism of the precursor is as follows:

$$20H^{+} + Mo_{7}O_{24}^{6-} + 14DDTC^{-} \rightarrow MoO_{2}(DDTC)_{2} + 10H_{2}O$$

In the hydrothermal reaction,  $MoO_2$  (DDTC)<sub>2</sub> decomposed into  $MoO_4^{2-}$ , then the  $O^{2-}$  was replaced by  $S^{2-}$  and  $MoS_2$  was obtained.<sup>16</sup>

The XRD patterns of the three types of  $MoS_2$ -X, with  $MoO_2$  (DDTC)<sub>2</sub> as the precursor, are shown in Fig. 2. The peaks of the three types of  $MoS_2$  are similar, all exhibiting weak crystals of  $MoS_2$  synthesized according to different surfactant-assisted hydrothermal methods. The  $MoS_2$  peaks with  $2\theta$  at  $34^\circ$  and  $57^\circ$  were detected but not obvious in the three  $MoS_2$ , and were attributed to the (100) and (110) planes of 2H-MoS<sub>2</sub> (JCPDS card no.75–1539), respectively, which indicated that the three  $MoS_2$ -

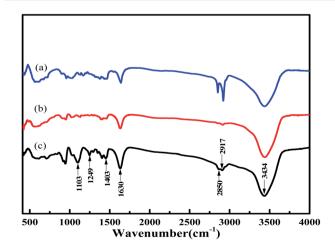


Fig. 3 FT-IR spectra of the samples. (a)  $MoS_2$ -CTAB; (b)  $MoS_2$ -SDS; (c)  $MoS_2$ -SDBS.

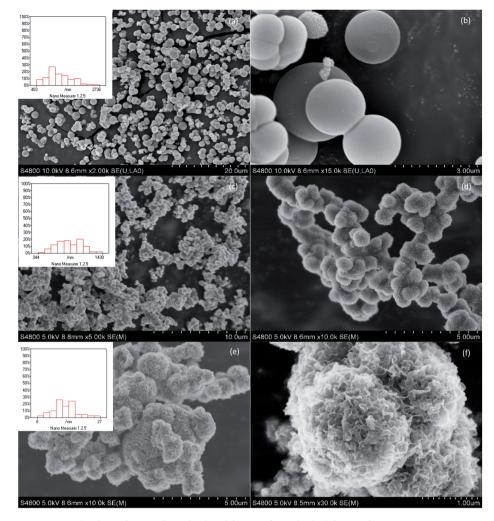


Fig. 4 SEM micrographs of the MoS<sub>2</sub>-CTAB (a and b), MoS<sub>2</sub>-SDBS (c and d), MoS<sub>2</sub>-SDS (e and f).

X had the same arrangement of atoms in basal planes of 2H-MoS<sub>2</sub>.<sup>17</sup> Moreover, the (002) peaks represented the accumulation of the S-Mo-S layers along the c-axis; since only weak van der Waals forces connect the MoS2 layers, it is very easy to insert heteroatoms between the layers. 17,18 A weak peak at 9° was observed instead of the (002) peak at 14° for MoS<sub>2</sub>-CTAB, MoS<sub>2</sub>-SDS and MoS<sub>2</sub>-SDBS as compared to the pristine 2H-MoS<sub>2</sub>. This indicated that the (002) peaks moved forward, which was

attributed to the increase in the interlayer spacing caused by the addition of surfactants, 19,20 demonstrating that there were few stacking layers of MoS2. On the whole, the effects of the three surfactants on the crystalline form and intensities of MoS2-X in the hydrothermal reaction were slight, and all the MoS2 exhibited poor crystallinity, consistent with the previous results of the synthesis of MoS<sub>2</sub> using the hydrothermal method.<sup>21</sup>

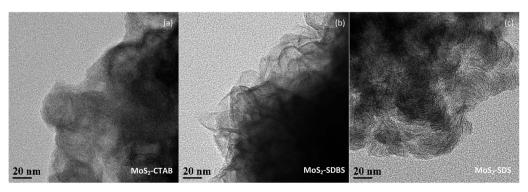


Fig. 5 FETEM images of the MoS<sub>2</sub>-CTAB (a), MoS<sub>2</sub>-SDBS (b), MoS<sub>2</sub>-SDS (c).

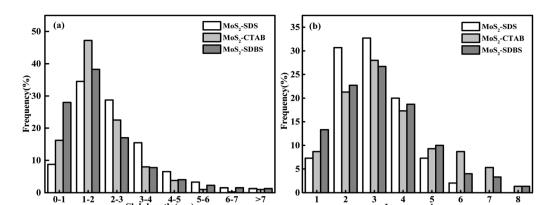


Fig. 6 Distributions of the slab length (a) and the layer number (b) of MoS<sub>2</sub>-CTAB, MoS<sub>2</sub>-SDBS, MoS<sub>2</sub>-SDS.

#### FT-IR analysis

**RSC Advances** 

The FT-IR spectra of MoS<sub>2</sub>-CTAB (a), MoS<sub>2</sub>-SDS (b), MoS<sub>2</sub>-SDBS (c) are given in Fig. 3. The peaks at 3434 cm<sup>-1</sup> and 1630 cm<sup>-1</sup> for the three samples were assigned to the N-H stretching vibrations and C=S stretching vibrations of the amide bond from the precursor, respectively.22 The stronger intensity of MoS2-SDBS was because of the effect of the C=C bond from the benzene ring in SDBS. The peaks at 2917 cm<sup>-1</sup> and 2850 cm<sup>-1</sup> were assigned to the stretching vibration modes of C-H from the surfactants (CTAB, SDS, SDBS).23 The C-H stretching vibration intensity was stronger in MoS2-CTAB than in the others, which was attributed to CTAB having the most C-H groups. The characteristic peaks of -SO<sub>2</sub> from MoS<sub>2</sub>-SDBS appeared at about 1103 cm<sup>-1</sup> and 1249 cm<sup>-1</sup>. In Fig. 3(c), the peak at 945 cm<sup>-1</sup> was strengthened, which indicated that there was a strong coordination effect between MoS2 and SDBS rather than simple adsorption and modification.24

#### **SEM results**

The SEM micrographs of the synthesized  $MoS_2$  with three surfactants are provided in Fig. 4. The introduction of three surfactants in the synthesis process gave rise to remarkable morphological changes in the  $MoS_2$ . Some irregularly shaped particles with sizes ranging from 300 to 3000 nm were detected.  $MoS_2$ -CTAB exhibited smooth, ball-like particles with sizes ranging from 400 to 2800 nm and larger; the particles were evenly dispersed and less agglomerated. This was attributed to the larger steric hindrance caused by the hydrophilic functional group ((CH<sub>3</sub>)<sub>3</sub>N<sup>-</sup>) of CTAB, which led to the larger size of the micelles formed by CTAB. In addition, the selective adsorption of CTA<sup>+</sup> on the specific crystal surface of the nuclei of  $MoS_2$ 

Table 1 HRTEM statistics of MoS<sub>2</sub>-CTAB, MoS<sub>2</sub>-SDS and MoS<sub>2</sub>-SDBS

Catalysts	$\Delta N$	$\Delta L$	D	$f_{ m e}/f_{ m c}$
MoS <sub>2</sub> -CTAB	3.31	1.94	3.22	1.53
$MoS_2$ -SDS	2.95	2.35	3.16	2.45
MoS <sub>2</sub> -SDBS	3.62	1.89	3.26	1.45

limited the growth of the nuclei of  $MoS_2$  and hindered its reunion in the hydrothermal process, thus increasing its dispersion. The size of the  $MoS_2$  formed by CTAB as the template was fairly large;<sup>25</sup> its average size was 1.2 um and the range of mean sizes was 700–900 nm.  $MoS_2$ -SDBS exhibited rough, ball-like particles that were greatly aggregated; there were hardly any single ball-like structures. The average particle size was 800 nm in the range from 300 nm to 1.5  $\mu$ m.  $MoS_2$ -SDS exhibited a flower-like morphology with many stacked nanosheets, and the average thickness of the nanosheets was 15 nm, which indicated substantial aggregation.

#### **FETEM results**

HRTEM studies were carried out in order to acquire more information regarding the distribution and structure of MoS<sub>2</sub> as catalysts. The HRTEM images in Fig. 5 further show the specific outlines of the morphology of MoS2 prepared by heating at 220 °C for 24 h and adding CTAB, SDS and SDBS, respectively. It was found that the fringes of the three samples were arranged in a relatively disorderly manner, which may have been caused by cracks on the crystal plane. Those disorders led to the disorder of the interlayer arrangement, which resulted in the poor crystallinity of MoS<sub>2</sub>, entirely consistent with the XRD results. Typical MoS<sub>2</sub> fringes with layer stacking spacing of 0.63 nm were visible on all the samples, in accordance with the characteristic crystalline MoS<sub>2</sub> basal plane (002).<sup>26,27</sup> The dislocations and distortions on the surface between different fringes indicated that the structure of MoS2 was defective and the existence of defects means the exposure of more active sites of MoS2.

**Table 2** Specific surface area, pore volume and pore diameter of MoS<sub>2</sub>-CTAB, MoS<sub>2</sub>-SDS and MoS<sub>2</sub>-SDBS

Samples	${ m MoS}_2 ext{-CTAB}$	$MoS_2$ -SDS	MoS <sub>2</sub> -SDBS
Specific surface area (m <sup>2</sup> g <sup>-1</sup> )	28	33	59
Pore volume (cm <sup>3</sup> g <sup>-1</sup> )	0.13	0.11	0.17
Pore diameter (nm)	3.82	3.82	3.82

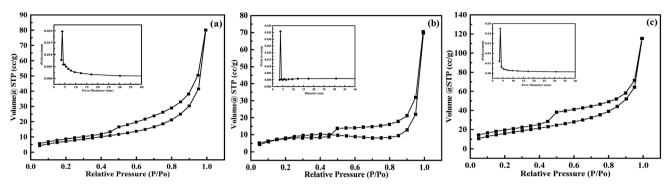


Fig. 7 Adsorption-desorption isotherms and pore volume distribution curves of MoS<sub>2</sub>-CTAB (a), MoS<sub>2</sub>-SDBS (b), MoS<sub>2</sub>-SDS (c).

The distribution of the slab length and the layer numbers of the MoS<sub>2</sub>-X are presented in Fig. 6. The values of the average slab length, layer number and  $(f_e/f_c)_{MO}$  are shown in Table 1. By changing the type of surfactant, the synthesized MoS2 showed differences in slab length and layer number. The number of layers in MoS2-X was between 1 and 8, and the slab length was mostly within 7 nm. For all the MoS<sub>2</sub> samples, the slab lengths were mostly between 1 and 2 nm and the number of layers was mostly 3. MoS2-SDS, MoS2-CTAB, and MoS2-SDBS accounted for 34.5%, 47.25%, and 38.25% of the total, respectively. With this as the demarcation point, the slab length first increased and then decreased. The percentage of MoS<sub>2</sub> slabs with less than 5 layers exceeded 87.5% and the percentage of MoS<sub>2</sub> slabs shorter than 4.0 nm exceeded 75%. As listed in Table 1, the average slab lengths of MoS<sub>2</sub>-SDS, MoS<sub>2</sub>-CTAB, and MoS<sub>2</sub>-SDBS were 2.35, 1.94, and 1.89 nm, the average numbers of layers were 2.95, 3.31, 3.62 nm, the dispersions were 3.16, 3.22, 3.26, and the values of  $(f_e/f_c)_{Mo}$  were 2.45, 1.53, 1.45, respectively. There were three layers for MoS2-CTAB, and MoS2-SDBS, and MoS2-SDS had greater than three layers. The results, combined with the analysis results from Raman spectroscopy, are in good agreement with the FETEM results, suggesting that the distribution of MoS<sub>2</sub> slabs was finely adjusted by adding different surfactants.

#### Raman results

Raman spectroscopy is an effective method for characterizing the interface quality of 2H-MoS2. As layered materials with incomplete crystalline structures, the interface quality was affected by strain, defects and dislocation. The Raman spectra of the MoS<sub>2</sub> series are displayed in Fig. 8. All the MoS<sub>2</sub> samples exhibited typical bands in the  $E_{2g}^1$  (378–382 cm<sup>-1</sup>) and  $A_{1g}$  (404– 408 cm<sup>-1</sup>) ranges; both the  $A_{1g}$  peak and the  $E_{2g}^1$  peak had blueshifts, and the bands were attributed to the interlayer displacement of Mo and S atoms and outside vibrations of the Mo-S bond along the c-axis. Comparing the three morphologies of MoS<sub>2</sub>, the interaction intensity between the adjacent layers of MoS<sub>2</sub>-CTAB was the weakest. The low-angle displacement of the vibration peak position of E<sub>2g</sub> was caused by interlayer dislocations or intra-layer defects of Mo and S atoms in crystalline MoS2-CTAB and MoS2-SDBS, which could lead to greater exposure of edge active sites<sup>28,29.</sup> The distance between  $A_{1g}$  and  $E_{2g}^{1}$  (d  $= \nu(A_{1g}) - \nu(E_{2g}^1)$  increased with the increasing layer number of  $MoS_2$ .<sup>30</sup>  $MoS_2$ -CTAB displayed two peaks at 376 cm  $^{-1}$  ( $E_{2g}^1$ ) and  $403 \text{ cm}^{-1}$  (A<sub>1g</sub>), and d = 27 cm. MoS<sub>2</sub>-SDS displayed two peaks at 381 cm<sup>-1</sup> ( $E_{2g}^{1}$ ) and 406 cm<sup>-1</sup> ( $A_{1g}$ ) (d = 25 cm). MoS<sub>2</sub>-SDBS displayed two peaks at 378.5 cm<sup>-1</sup> ( $E_{2g}^1$ ) and 403 cm<sup>-1</sup> ( $A_{1g}$ ) (d=24.5 cm). Raman spectroscopy demonstrated that the average

#### **BET results**

The specific surface area of MoS<sub>2</sub> affects its catalytic performance to some extent. It was found that the specific surface areas of all the MoS2 synthesized by adding different surfactants were in the order of  $MoS_2$ -SDBS >  $MoS_2$ -SDS >  $MoS_2$ -CTAB. From Table 2, the pore size followed the sequence of MoS<sub>2</sub>-CTAB > MoS<sub>2</sub>-SDBS > MoS<sub>2</sub>-SDS, and pore volume was of the order  $MoS_2$ -SDBS >  $MoS_2$ -CTAB >  $MoS_2$ -SDS.  $MoS_2$ -SDBS had the largest specific surface area and pore volume of 59.47 m<sup>2</sup> g<sup>-1</sup> and 11 cm<sup>3</sup> g<sup>-1</sup>, respectively. The N<sub>2</sub> adsorption-desorption and pore size distribution of MoS<sub>2</sub> with different morphologies are illustrated in Fig. 7. The adsorption and desorption isotherms were classified as all delayed rings of type H3, which are associated with slit-shaped pores formed by the accumulation of plate-like particles and the existence of mesoporous structures, but there was no saturated adsorption platform. At 3-5 nm, this is consistent with the results of SEM characterization.

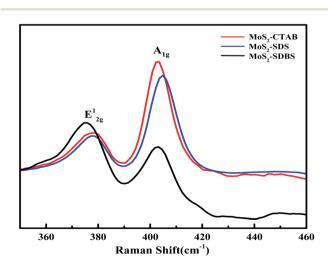


Fig. 8 Raman spectra of MoS<sub>2</sub>-CTAB, MoS<sub>2</sub>-SDS, MoS<sub>2</sub>-SDBS.

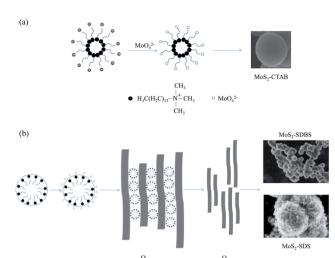


Fig. 9 Schematic diagram of formation mechanism of MoS<sub>2</sub>-CTAB (a), MoS<sub>2</sub>-SDBS and MoS<sub>2</sub>-SDS (b).

•  $H_3C(H_2C)_{11}O - \sum_{i=0}^{|I|} = O \quad H_3C(H_2C)_{11} - \sum_{i=0}^{|I|} - C(H_2C)_{11} - C(H_$ 

layers of MoS<sub>2</sub>-SDS and MoS<sub>2</sub>-SDBS were three layers, in agreement with the HRTEM results.

#### The formation mechanism of MoS<sub>2</sub>

According to the above characterization and results,  $MoO_2(-DDTC)_2$  was acidic (pH < 4) in aqueous solution and existed in the form of  $MoO_4^{2-}$  and  $DDTC^-$ ; the  $O^{2-}$  in  $MoO_4^{2-}$  was easily replaced by  $S^{2-}$  because of the weak Mo–O bond in  $MoO_4^{2-}$ .  $MoO_4^{2-}$  aggregated to form polymolybdate, and hydroxylamine hydrochloride was used as the reducing agent to reduce Mo(v) to Mo(v). During the hydrothermal reaction,  $DDTC^-$  reacted with polymolybdate to form polytetrathiomolybdate, and the nucleus of  $MoS_2$  was finally formed by oxidation and reduction. The addition of surfactants resulted in different morphologies and structures of  $MoS_2$ , and the surfactants also acted as templates in the hydrothermal processes. The insertion of surfactant into the layered structure of  $MoS_2$  led to the formation of nanosheets. These nanosheets aggregated on SDS/SDBS-formed micelles and grew into flower-like morphologies. When

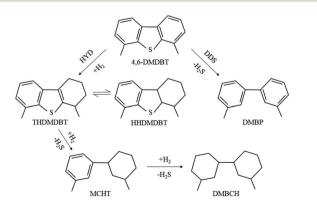


Fig. 10 Schematic diagram of the hydrodesulfurization path of 4,6-DMDBT.

**Table 3** Comparison of the HDS desulphurization, HYD ratio and HDS selectivity of MoS<sub>2</sub>-CTAB MoS<sub>2</sub>-SDS and MoS<sub>2</sub>-SDBS

Catalysts	Desulphurization (%)	HYD ratio (%)	Path selectivity
MoS <sub>2</sub> -CTAB	58.5	37.1	1.6
$MoS_2$ -SDS	44.6	26.3	1.1
MoS <sub>2</sub> -SDBS	95.9	90.1	15.5

CTAB/SDS/SDBS was added to the reaction, the reaction mechanism was as given in Fig. 9(a) and (b). From Fig. 9(a), with CTAB as the cationic surfactant, MoO<sub>4</sub><sup>2-</sup> was adsorbed on it as a template to produce a smooth spherical morphology. When SDS and SDBS were added to the reaction system as anionic surfactants, the formation mechanism of MoS2 was similar. Specifically, the nanosheets were easier to grow along the (002) plane because this plane is the thermodynamically stable plane of MoS<sub>2</sub> nanosheets. Fig. 9(b) shows the mechanism with SDS/ SDBS as surfactants. During the reaction, SDBS and SDS were dissolved in water to form micelles; then, MoO<sub>4</sub><sup>2-</sup> was adsorbed onto the surface of the micelles. A large number of nanosheets were formed at the same time, and then those nanosheets were exfoliated and assembled into spheres under the action of the surfactants. In addition, the head-to-base ratio of SDBS is larger and more easily agglomerated to form micelles, as compared to SDS, which may explain why the morphology of MoS2-SDS is flower-like and MoS2-SDBS is rough-ball-like.

#### Catalytic activity

The hydrodesulfurization of 4,6-DMDBT can be divided into the hydrodesulfurization path (HYD) and the direct desulfurization path (DDS). HDS products of 4,6-DMDBT mostly contain THDMDBT, MCHT, DMBCH and DMBP. THDMDBT, MCHT and DMBCH are HYD products and DMBP are DDS products. The hydrogenation intermediate THDMDBT in the HYD pathway breaks the C–S bond to form MCHT, which is further hydrogenated to form DMBCH. In the DDS pathway, the C–S

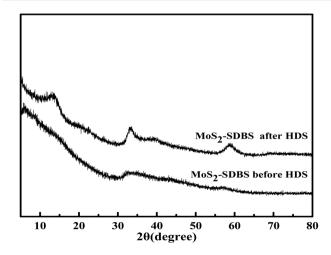


Fig. 11  $\,$  XRD patterns of  $\,$ MoS $_2$ -SDBS before and after HDS.

Paper RSC Advances

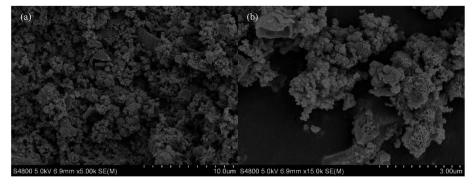


Fig. 12 SEM micrographs of MoS<sub>2</sub>-SDBS (a and b) after HDS.

bond directly broke and generated the direct desulfurization product DMBP. The schematic diagram is shown in Fig. 10. The catalytic performances of MoS<sub>2</sub>-X in the HDS of simulated diesel oil (0.4 wt% S content of 4,6-DMDBT/decalin) are set out in Table 3. MoS<sub>2</sub>-SDBS exhibited the best HDS performance, with highest desulphurization, HYD ratio and selectivity, combined with FETEM analysis, the results showed that MoS2-SDBS had more active catalytic sites, which included edge sites and corner sites and this can be attributed to its shortest average slab length, average layer number and the smallest value of  $(f_e/f_c)_{Mo}$ . The BET results showed that MoS<sub>2</sub>-SDBS had the largest specific surface area among the three MoS2. A comprehensive comparison of the results of HRTEM and BET analyses showed that they were accordant. The order of the  $\Delta L$  and  $(f_e/f_c)_{Mo}$  values of MoS<sub>2</sub> with three morphologies from large to small is MoS<sub>2</sub>-SDS  $> MoS_2$ -CTAB  $> MoS_2$ -SDBS, while the order of  $\Delta N$  and dispersion are the opposite,  $MoS_2$ -SDBS >  $MoS_2$ -CTAB >  $MoS_2$ -SDS. Many studies have shown that HDS catalysts with shorter slab length, higher stacking and greater dispersion have better catalytic performances because more active sites are exposed.31-34 Therefore, the catalytic activities of the three MoS2 can be inferred as follows: MoS2-SDBS > MoS2-CTAB > MoS2-SDS. The shortest slab length, the smallest value ratio of Mo edge sites to corner sites and the greatest dispersion of MoS<sub>2</sub>-SDBS determine that it has the most active sites and the larger corner position as compared to the other two catalysts, which ultimately led to the best HDS performance. The desulfurization rate of MoS<sub>2</sub>-X was arranged in the order of MoS<sub>2</sub>-SDBS > MoS<sub>2</sub>-SDS > MoS<sub>2</sub>-CTAB. According to BET and HRTEM results, the specific surface area and pore volume of MoS2-SDBS were larger than those of MoS<sub>2</sub>-CTAB and MoS<sub>2</sub>-SDS. This may be the point of the higher desulfurization rate of MoS2-SDBS. From the analysis results, the desulfurization rate of 4,6-DMDBT was influenced greatly by the average slab length, followed by the average number of layers and finally the value of  $(f_e/f_c)_{Mo}$ . In addition, the HYD ratio and selectivity of MoS<sub>2</sub>-X were in the order of  $MoS_2$ -SDBS >  $MoS_2$ -CTAB >  $MoS_2$ -SDS. The three  $MoS_2$ as catalysts preferred to react along the HYD path. The  $(f_e/f_c)_{MO}$ value of MoS2 was negatively correlated with the degree of hydrogenation. It was reported that the corner sites on the MoS<sub>2</sub> are favorable for HYD reactions and the edge sites on the MoS<sub>2</sub> promote HDS reactions because the corner sites are more

favorable for the adsorption and dissociation of molecular hydrogen into ionic hydrogen. The values of  $(f_e/f_c)_{Mo}$  were calculated in combination with HRTEM:  $f_e/f_c$  (MoS<sub>2</sub>-SDBS) >  $(f_e/f_c)_{Mo}$  (MoS<sub>2</sub>-CTAB) >  $(f_e/f_c)_{Mo}$  (MoS<sub>2</sub>-SDS), which exhibited the same trends as the HYD ratio and selectivity of MoS<sub>2</sub>-X. These results can be reviewed as the reason the HYD ratio and selectivity of MoS<sub>2</sub>-CTAB are higher as compared to MoS<sub>2</sub>-SDS. In conclusion, the desulfurization rate of MoS<sub>2</sub>-X is positively correlated with the average number of layers and the HYD ratio and selectivity are positively correlated with the average slab length, and the ratio of the Mo edge/corner.

#### Reuse rate of MoS<sub>2</sub>-SDBS

MoS<sub>2</sub>-SDBS has the best catalytic activity of three catalysts; its morphology, structure and reuse in the HDS process have been studied. Fig. 11 compares the XRD spectra of MoS<sub>2</sub>-SDBS before and after the HDS reaction. MoS<sub>2</sub>-SDBS, after HDS reaction, exhibited higher crystallinity than it did before the HDS reaction, which may be due to the high temperature during the reaction. Fig. 12 shows SEM micrographs of MoS<sub>2</sub>-SDBS after the HDS reaction; the morphology changed from rough, spherical particles before the reaction to granular particles that just stuck together after the reaction. MoS<sub>2</sub>-SDBS after the HDS reaction was used as a catalyst for HDS to explore the reuse rate; the result showed that the desulphurization of the catalyst was 78.7% as compared with MoS<sub>2</sub>-SDBS without HDS. The desulfurization rate was reduced by 17.9%, which showed that its reuse rate had decreased.

#### Conclusions

Three types of MoS<sub>2</sub> were synthesized for use as catalysts by the combination of the precursor method and the hydrothermal method and were used in HDS to assess their catalytic activity. The results are as follows. The smooth-surfaced and ball-like, rough-surfaced and ball-like, and flower-like morphologies of MoS<sub>2</sub> were prepared with the assistance of different surfactants (CTAB/SDBS/SDS). The TEM images showed that in general, the average slab length and the ratio of edge/corner of the MoS<sub>2</sub> increased remarkably in the following order: MoS<sub>2</sub>-SDBS > MoS<sub>2</sub>-CTAB > MoS<sub>2</sub>-SDS. The average number of layers was in the order MoS<sub>2</sub>-SDBS > MoS<sub>2</sub>-SDS > MoS<sub>2</sub>-CTAB. When MoS<sub>2</sub>

was used as the catalyst in hydrodesulfurization, the desulfurization rate of the MoS<sub>2</sub> increased in the order MoS<sub>2</sub>-SDBS > MoS<sub>2</sub>-CTAB, which exhibited a positive correlation with the average number of layers and dispersion of MoS<sub>2</sub>. The HYD ratio and selectivity of the MoS<sub>2</sub> increased in the order MoS<sub>2</sub>-SDBS > MoS<sub>2</sub>-CTAB > MoS<sub>2</sub>-SDS, which exhibited a negative correlation with the average slab length and the ratio of Mo edge/corner sites of MoS<sub>2</sub>. The results showed that MoS<sub>2</sub>-SDBS exhibited the highest HDS performance among the three catalysts because it had the shortest average slab length, highest Mo ratio of edge to corner, and highest dispersion and specific surface area. The HDS performance of the reused MoS<sub>2</sub>-SDBS was lower than that of the original MoS<sub>2</sub>-SDBS.

## Conflicts of interest

There are no conflicts to declare.

# Acknowledgements

The work was supported by National Natural Science Foundation of China projects (Grant No. 21603077, 51603077) and the industry-University-Research Collaborative Innovation Project of Xiamen (Grant No. 3502z20183024).

#### Notes and references

- 1 S. Oda and K. Cui, J. Jpn. Pet. Inst., 2004, 47, 145-163.
- 2 A. Stanislaus, A. Marafi and M. S. Rana, *Catal. Today*, 2010, 153, 1–68.
- 3 Y. Wang, C. Yin, X. Zhao and C. Liu, *Catal. Commun.*, 2017, 88, 13–17.
- 4 X. Li, Y. Chai, B. Liu, H. Liu, J. Li, R. Zhao and C. Liu, *Ind. Eng. Chem. Res.*, 2014, 53, 9665–9673.
- 5 Y. Wang, C. Yin, X. Zhao and C. Liu, *Catal. Commun.*, 2017, **88**, 13–17.
- 6 W. Zhou, Q. Wei, Y. Zhou, M. Liu, S. Ding and Q. Yang, Appl. Catal., B, 2018, 238, 212–224.
- 7 M. Li, D. Wang, J. Li, Z. Pan, H. Ma, Y. Jiang and Z. Tian, RSC Adv., 2016, 6, 71534–71542.
- H. Miao, X. Hu, Q. Sun, Y. Hao, H. Wu, D. Zhang, J. Bai,
   E. Liu, J. Fan and X. Hou, *Mater. Lett.*, 2016, 166, 121–124.
- 9 Y. Li, H. Li, S. Zhou, H. Yang, F. Xie and W. Li, *Res. Chem. Intermed.*, 2018, **44**, 4353–4364.
- 10 Y. Li, X. Zhang, W. Zhang, M. Chang, C. Lin, K. Chang, Y. Yu, J. Wang, C. Chang, L. Li and T. Lin, *Adv. Mater.*, 2012, 24, 2320–2325.
- 11 K. Hu, Y. Wang, X. Hu and H. Wo, J. Mater. Sci. Nanotechnol., 2013, 23, 242–246.

- 12 T. Wang, J. Li and G. Zhao, *Powder Technol.*, 2014, 253, 347–351.
- 13 J. Theerthagiri, R. A. Senthil, B. Senthilkumar, A. Reddy Polu, J. Madhavan and M. Ashokkumar, J. Solid State Chem., 2017, 252, 43-71.
- 14 A. M. van der Kraan, V. H. J. de Beer, J. A. R. van Veen and R. A. van Santen, *J. Catal.*, 2001, **199**, 224–235.
- 15 H. Li, W. Zhang, Y. Wang, M. Shui, S. Sun, J. Bao and C. Gao, J. Energy Chem., 2019, 30, 57–62.
- 16 W. Hu, G. Han, F. Dai, Y. Liu, X. Shang, B. Dong, Y. Chai, Y. Liu and C. Liu, *Int. J. Hydrogen Energy*, 2016, 41, 294–299.
- 17 J. Xie, Z. Hao, L. Shuang, R. Wang, S. Xu, Z. Min, J. Zhou, W. L. Xiong and X. Yi, *Adv. Mater.*, 2013, **25**, 5807–5813.
- 18 J. Xie, J. Zhang, S. Li, F. Grote, X. Zhang, H. Zhang, R. Wang, Y. Lei, B. Pan and Y. Xie, *J. Am. Chem. Soc.*, 2013, **136**, 17881–17888.
- 19 F. Wang, M. Zheng, B. Zhang, C. Zhu, Q. Li, L. Ma and W. Shen, *Sci. Rep.*, 2016, **6**, 31092.
- 20 Y. Tian, J. Zhao, W. Fu, Y. Liu, Y. Zhu and Z. Wang, *Mater. Lett.*, 2005, **59**, 3452–3455.
- 21 X. Xie, L. Gao and J. Sun, Colloids Surf., A, 2007, 308, 54-59.
- 22 X. Tian, M. Fang, C. Chen, S. Yu and X. Wang, *Carbon*, 2008, **46**, 1741–1750.
- 23 Z. Wu, D. Wang and A. Sun, Mater. Lett., 2009, 63, 2591-2593.
- 24 Y. Zhang, W. Zeng and Y. Li, *Appl. Surf. Sci.*, 2018, **455**, 276–282.
- 25 E. Payen, R. Hubaut, S. Kasztelan, O. Poulet and J. Grimblot, J. Catal., 1994, 147, 123–132.
- 26 T. Fujikawa, H. Kimura, K. Kiriyama and K. Hagiwara, *Catal. Today*, 2006, **111**, 188–193.
- 27 C. Lee, H. Yan, L. E. Brus, T. F. Heinz, J. Hone and S. Ryu, *ACS Nano*, 2010, 4, 2695–2700.
- 28 H. Li, Q. Zhang, C. C. R. Yap, B. K. Tay, T. H. T. Edwin, A. Olivier and D. Baillargeat, *Adv. Funct. Mater.*, 2012, 22, 1385–1390.
- 29 J. Xu, Y. Guo, T. Huang and Y. Fan, Appl. Catal., B, 2019, 244, 385–395.
- 30 Y. Gao, W. Han and X. Long, Appl. Catal., B, 2018, 224, 330-
- 31 W. Han, P. Yuan, Y. Fan, G. Shi, H. Liu, D. Bai and X. Bao, *J. Mater. Chem.*, 2012, 22, 25340–25353.
- 32 S. V. Budukva, O. V. Klimov and A. S. Noskov, *Catal. Ind.*, 2015, 7, 214–220.
- 33 L. Woolfolk, C. Geantet, L. Massin, D. Laurenti and J. D. Reyes, *Appl. Catal.*, *B*, 2017, 201, 31–338.
- 34 M. Li, H. Li, F. Jiang, Y. Chu and H. Nie, *Catal. Today*, 2010, **149**, 35–39.