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First principles study of ferroelectric hexagonal compounds $R\text{InO}_3$ ($R = \text{Dy}, \text{Er},$ and Ho): electronic structure, optical and dielectric properties

 Yifei Yu,^a Na Lin,^b *^{ab} Haoyuan Wang,^a Ran Xu,^a Hao Ren ^c and Xian Zhao*^a

The 4f-shell electrons of rare-earth ion R have a certain influence on both the electrical and optical properties of $R\text{InO}_3$. We have studied the electronic structures, optical and dielectric properties of hexagonal compounds $R\text{InO}_3$ ($R = \text{Dy}, \text{Ho},$ and Er) by performing first principles calculations. After optimization, the lattice parameters are in great agreement with the experiment with an error within 1%. Band structure calculations reveal decreasing band gaps with the increase of atomic number, yielding the highest conductivity for ErInO_3 , which has the smallest band gap in the present study. We have calculated the density of states of $R\text{InO}_3$ and characterized the displacement of In and O. The dielectric properties and ionic contribution have been calculated and a detailed comparison has been made on the dielectric function including the static dielectric tensor. The absorption coefficient, the reflectivity, the refractive index, the conductivity and the electron energy-loss spectrum of hexagonal DyInO_3 , HoInO_3 and DyInO_3 have been calculated and analyzed.

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1 Introduction

A group of materials with hexagonal ABO_3 -type structure have been extensively invoked for their geometric multiferroic characteristics and potential applications in capacitors, FRAM storage, computer memory and other multifunctional devices.^{1–4} Research on these materials mainly pays attention to the correlation between ferromagnetism and ferroelectricity, however, rarely does a study exist on their optical and dielectric properties which are also important to explore. It is worth noting that there are some ABO_3 perovskites reflecting superior dielectric properties^{5,6} which have potential applications in manufacturing dielectric devices. People can control the dielectric and optical properties of the perovskite structures by selecting appropriate A and B cations due to the fact that physical properties of ABO_3 perovskite depend on the mass and ionic radius of the A and B cations.

Among the ABO_3 perovskite structures, RMnO_3 ($R =$ rare-earth element) has been widely studied due to its excellent ferroelectric and dielectric properties,^{7–13} and hexagonal $R\text{InO}_3$ with the space group of $P6_3cm$ is isomorphic to the hexagonal RMnO_3 .¹⁴ If we replace Mn^{3+} ions with non-transition metal In^{3+} ions, the dielectric and optical properties of the $R\text{InO}_3$ perovskite system

are only contributed by the 4f-shell electrons of the rare-earth ions R^{3+} ,^{15–21} which makes it much more straightforward to study the optical and dielectric properties of $R\text{InO}_3$ than in RMnO_3 . Studies have shown that the spontaneous polarization of hexagonal $R\text{InO}_3$ increases with the increase of the atomic number of rare-earth elements.²² In order to explore the dielectric and optical properties as a function of the atomic number of rare-earth elements, in present study we have chosen Dy, Ho and Er with increasing atomic number. Moreover, the hexagonal DyInO_3 and HoInO_3 are stable at normal pressure and temperature but transform to 6% denser pseudohexagonal orthorhombic phases at 20 kbar, 1000 °C,²³ while the properties of ErInO_3 have not yet been confirmed. In this work, we have performed first principles calculations on the electronic structure, optical and dielectric properties of hexagonal $R\text{InO}_3$ ($R = \text{Dy}, \text{Ho},$ and Er) and further discussed their potentials in the preparation of high quality multifunctional devices.

2 Computational details

All calculations in present paper are based on density functional theory (DFT) with Perdew–Burke–Ernzerhof (PBE) of the generalized gradient approximation (GGA)²⁴ functional for the exchange–correlation function, implemented in Vienna *Ab initio* Simulation Package (VASP)^{25,26} by solving the Kohn–Sham equation and corrected the results by the GW method of quasi-particle approximation. The specific Kohn–Sham equation is as follows,

$$\left\{ -\frac{1}{2}\nabla^2 + \left[V(\vec{r}) + \mu_{xc}(n) \right] \right\} \phi_i(\vec{r}) = \varepsilon_i \phi_i(\vec{r}) \quad (1)$$

^aState Key Laboratory of Crystal Materials, Shandong University, 250100 Jinan, Shandong, PR China. E-mail: linnakth@gmail.com; xianzhao@sdu.edu.cn

^bDepartment of Chemistry, University of Calgary, T2N 1N4 Calgary, Alberta, Canada

^cState Key Laboratory of Heavy Oil Processing, Center for Bioengineering and Biotechnology, China University of Petroleum (East China), 266580 Qingdao, PR China



where $V(\vec{r}) + \mu_{xc}(n) = V_{\text{eff}}(r)$ is the effective potential and ε_i is the eigenvalue of the Kohn–Sham equation. We employ the projector augmented-wave (PAW) method to describe the interaction of ions and electrons. The cut-off energy in the plane-wave expansion has been set to 450 eV, which had been accurately proved to ensure the well-converged total energy.²⁷ In order to obtain the well-polymerized geometrical configurations, we fully relaxed the hexagonal structures with a $5 \times 5 \times 2$ k -point mesh grid and $8 \times 8 \times 3$ for the subsequent calculations of properties. Convergences of energy and force have been respectively set to 0.01 meV per cell and $-0.02 \text{ eV \AA}^{-1}$. We have also employed the density functional perturbation theory (DFPT) with a linear response method implemented in the Cambridge Serial Total Energy (CASTEP)²⁸ code in the calculations for optical and dielectric properties. The calculated lattice parameters of RInO_3 ($\text{R} = \text{Dy, Ho, Er}$) has been determined to be $a = b = 6.357 \text{ \AA}$, $c = 12.404 \text{ \AA}$, which are in great agreement with the experimental value of $a = b = 6.300 \text{ \AA}$ and $c = 12.300 \text{ \AA}$,²⁹ with an error within 1%, indicating the accuracy of present calculations.

3 Results and discussion

3.1 Electronic structure of hexagonal DyInO_3 , HoInO_3 and ErInO_3

The optimized geometrical configurations for hexagonal RInO_3 is shown in Fig. 1(a), containing 6 rare-earth (R) atoms, 18 oxygen (O) atoms and 6 indium (In) atoms. A $2 \times 2 \times 1$ super-cell of RInO_3 has been made for a better understanding, as shown in Fig. 1(b). We can see that the structure of RInO_3 along c -axis direction is formed by the alternating bonding of In and O atoms with hexagonal structure, and the bond length of In–O is 2.11 Å. Therefore, hexagonal RInO_3 has been proved to be quite stable which lays the foundation for further study on its optical and dielectric properties.

The calculated band structure of hexagonal RInO_3 ($\text{R} = \text{Dy, Ho, Er}$) is shown in Fig. 2. We position the Fermi level corresponding to the top of the valence band at zero energy and marked with dotted line. From Fig. 2 we can see that hexagonal

RInO_3 are indirect band gap crystals with band-gap values respectively to be 1.90 eV, 1.85 eV and 1.80 eV for DyInO_3 , HoInO_3 and ErInO_3 , which reveals decreasing band gaps with the increase of atomic number. The smaller the band gap, the easier it is for the electron to be excited by the valence band to the conduction band, and the higher the conductivity. Among three crystals in present study, DyInO_3 which has the largest band gap will give the smallest conductivity, whereas ErInO_3 which has the smallest band gap yields highest conductivity.

Fig. 3 shows the total density of states (TDOS) and the partial density of states (PDOS) of unit-cell hexagonal DyInO_3 , HoInO_3 and ErInO_3 . Due to the similarity between these three crystals and the properties of the lanthanides compound, it's not difficult to observe from Fig. 3(a), (c) and (e) that the curves of total density of states of DyInO_3 , HoInO_3 and ErInO_3 are roughly similar and there only exist minor different density peak in the energy range from -27 to -20 eV. In addition to observing the overall state density distribution, we also plot the distribution of

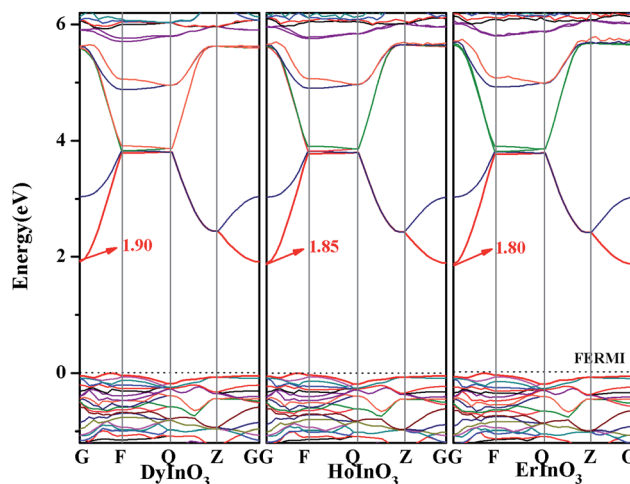


Fig. 2 Band structure of hexagonal DyInO_3 , HoInO_3 and ErInO_3 along the high symmetry direction. The Fermi level is positioned at zero energy and marked with dotted line.

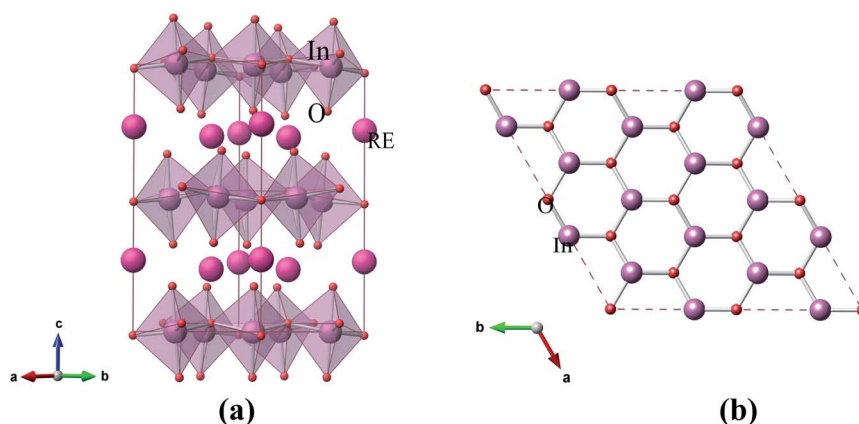


Fig. 1 (a) Optimized structure of hexagonal RInO_3 ($\text{R} = \text{Dy, Er, Ho}$). (b) Quadrilateral arrangement of a $2 \times 2 \times 1$ super-cell of optimized hexagonal RInO_3 along c -axis.



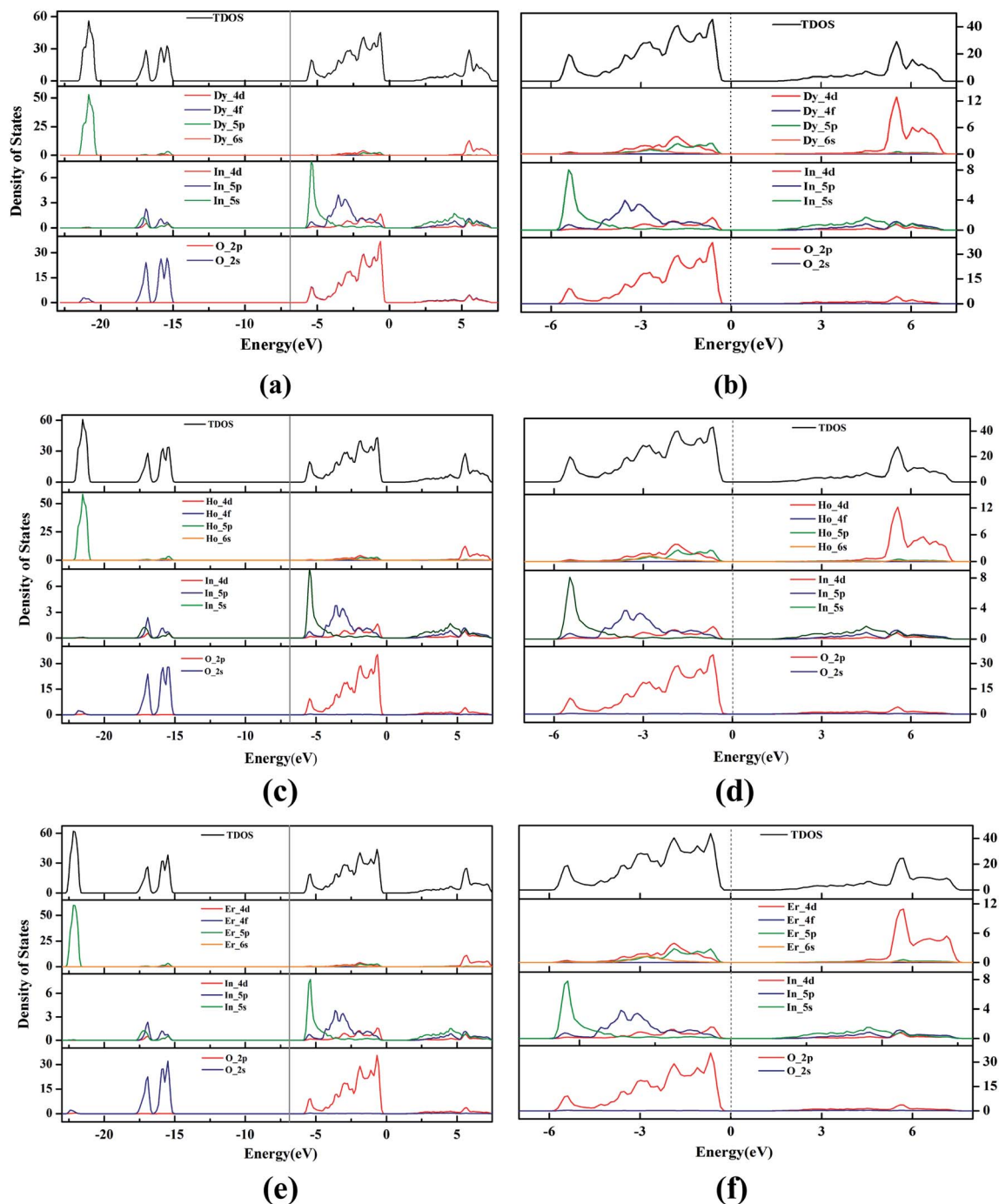


Fig. 3 The total and partial density of states of hexagonal DyInO_3 (a), HoInO_3 (c) and ErInO_3 (e) in the energy range from -27 to 7 eV and the enlarged map of DyInO_3 (b), HoInO_3 (d) and ErInO_3 (f) in energy range from -7 to 7 eV.

density of states near the valence band maximum (VBM) and the conduction band minimum (CBM), that is, in the energy range from -7 to 7 eV, as collected in Fig. 3(b), (d) and (f). From Fig. 3(b), we can see in the energy range from -7 to 0 eV the VBM of DyInO_3 mainly contributed by O 2p states, together with a part of contribution from In 4d states. Since the electronic contribution of Dy to partial density of states is less than 0.5 eV, therefore, we do not take the contribution of Dy into account. In the energy range from 0 to 7 eV, the CBM of DyInO_3 is mainly

contributed by In 4d states, together with a small contribution from O 2p states. The density of In is much larger than the density of O in the contribution to the VBM, which indicates that a large part of the electrons of O transfer from the valence band to the conduction band and hybridize with Ti 3d electrons. For the case where the contribution of different atomic orbitals to the conduction band and the valence band of RInO_3 , we can see similar situations in Fig. 3(d) and (f). The VBM and the CBM of these three structures are mainly contributed by O



and In, while the contribution of the lanthanides to the total density of states is not significant. This indicates that the band structure properties of hexagonal DyInO₃, HoInO₃ and ErInO₃ are mainly provided by the strong bonding of In and O whereas lanthanides show limited effects.

3.2 Dielectric properties of hexagonal DyInO₃, HoInO₃ and ErInO₃

We calculate the optical properties of hexagonal DyInO₃, HoInO₃ and ErInO₃ by calculating the static dielectric function $\varepsilon(\omega)$,

$$\varepsilon(\omega) = \varepsilon_1(\omega) + i\varepsilon_2(\omega) \quad (2)$$

where $\varepsilon_1(\omega)$ is the real part and $\varepsilon_2(\omega)$ is the imaginary part of the static dielectric function. The real part $\varepsilon_1(\omega)$ is characterized by the spontaneous polarization of the dielectric under an applied electric field, while the imaginary part $\varepsilon_2(\omega)$ is characterized by the loss of the dielectric and we can further calculate the optical properties of RInO₃ through imaginary parts.

Fig. 4(a) shows the real part curves of the static dielectric function of hexagonal DyInO₃, HoInO₃ and ErInO₃ and we can see that the static dielectric constants of these three crystals are all 4.53. The real part curves all reach the maximum when the photon energy reaches 2.24 eV. As shown in Fig. 4(b), the corresponding imaginary part curves of the dielectric function of DyInO₃, HoInO₃ and ErInO₃ show an increasing pattern in the energy range from 0 to 10.4 eV and decrease from 10.4 eV to 24.3 eV. All three systems under study give similar imaginary part curves, only with mutual shifts on peaks located in the energy range from 20 eV to 40 eV. It can be seen from Fig. 4(b) that the imaginary part function at 10.1 eV is much stronger than other dielectric function peaks, which corresponds to the transition of electrons from the O 2s valence band to the 4d conduction band of these three rare-earth elements. The peak at high energy (at around 28.0 eV) should be originated from the transition of electrons from the O 2p valence band to the 4d conduction band of rare-earth elements. The imaginary part function of DyInO₃ at 10.1 eV is slightly stronger than that of the other two materials. As the energy increases, the peaks show

a blue-shift from DyInO₃ to HoInO₃ and to ErInO₃, which is consistent with the trend of the reflectivity curves of these three materials in Fig. 6(a) shown below.

3.3 Optical properties of hexagonal DyInO₃, HoInO₃ and ErInO₃

We calculated the real part $\varepsilon_1(\omega)$ and the imaginary part $\varepsilon_2(\omega)$ respectively of dielectric function of hexagonal RInO₃ in Perdew–Burke–Ernzerhof method by VASP and calculate the absorption spectrum coefficient $\alpha(\omega)$,^{30,31}

$$\alpha(\omega) = \sqrt{2}\omega \left[\sqrt{\varepsilon_1^2(\omega) + \varepsilon_2^2(\omega)} - \varepsilon_1(\omega) \right]^{\frac{1}{2}} \quad (3)$$

Fig. 5(a) shows the absorption spectra of unit-cell hexagonal DyInO₃, HoInO₃ and ErInO₃ along the directions of XX, YY and ZZ respectively in the energy range from 0 to 60 eV. We can see from Fig. 5(a) that the absorption spectra of all three crystals are all very similar in the energy range from 0 to 25 eV. In the energy range from 25 to 30 eV, the absorption intensity of ErInO₃ at the absorption peak along the XX direction is significantly larger than that of HoInO₃, and the absorption intensity of DyInO₃ is the smallest among these three compounds, while their absorption intensity along the ZZ direction is equivalent. The absorption spectrum of DyInO₃ shows a distinct absorption peak at an energy of around 43 eV along all directions of XX, YY and ZZ, which is absent in the absorption spectra of both ErInO₃ and HoInO₃. The total absorption averaged over all three directions has been displayed in Fig. 5(b), from which we can see all three crystals show an absorption peak located at around 11 eV. Visible differences start to appear in the second peak region at around 30 eV. The three materials show very slight differences in peak intensities. A blue-shift of the peak can be observed with the decrease of atomic number of R. In addition, only DyInO₃ has a distinct absorption peak at the energy of around 43 eV.

We calculate the reflectivity $R(\omega)$ and refractive index $n(\omega)$ of all three crystals under study according to the imaginary part of the dielectric function,

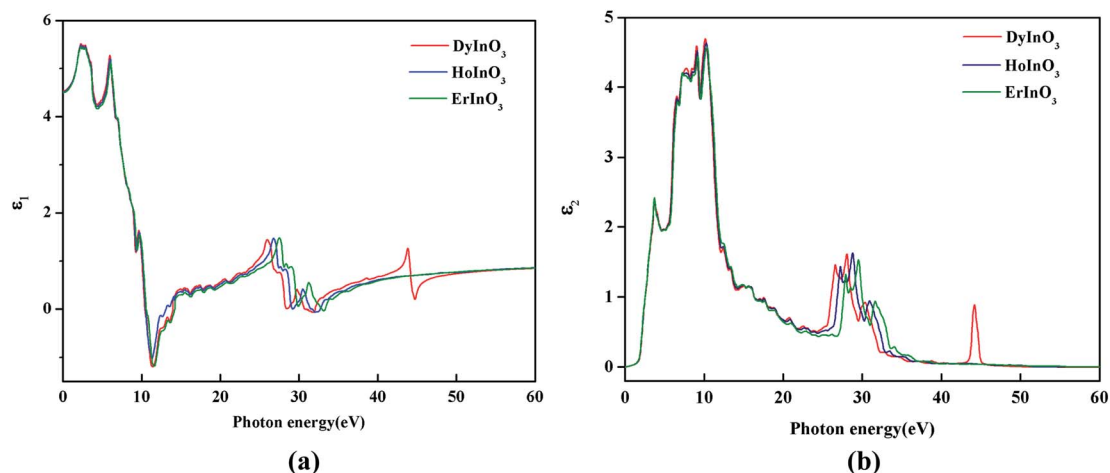


Fig. 4 The real part (a) and the imaginary part (b) of the static dielectric function of hexagonal DyInO₃, HoInO₃ and ErInO₃.



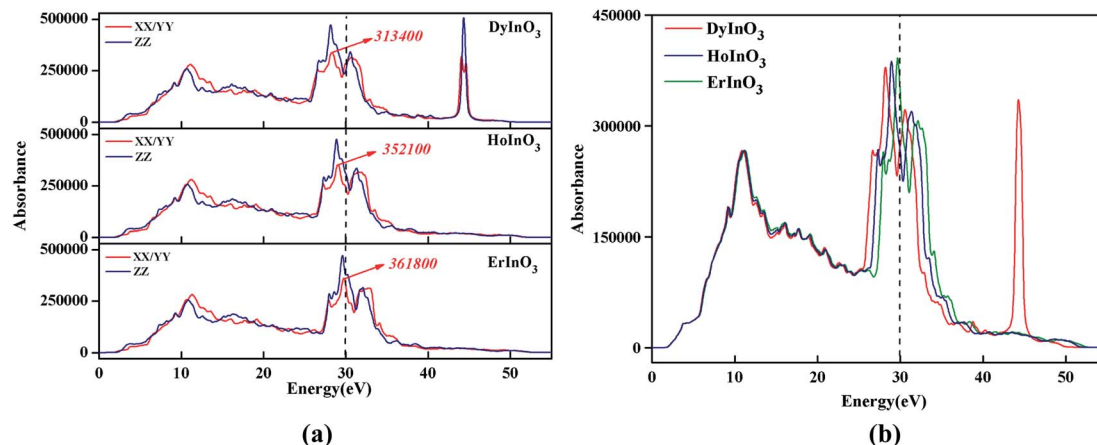


Fig. 5 Absorption spectra of hexagonal DyInO₃, HoInO₃ and ErInO₃ along different directions respectively (a) and their comparison of total absorption spectra (b) in energy range from 0 to 60 eV.

$$R(\omega) = \left| \frac{\sqrt{\varepsilon_1(\omega) + i\varepsilon_2(\omega)} - 1}{\sqrt{\varepsilon_1(\omega) + i\varepsilon_2(\omega)} + 1} \right|^2 \quad (4)$$

$$n(\omega) = \frac{1}{\sqrt{2}} \left\{ [\varepsilon_1^2(\omega) + \varepsilon_2^2(\omega)]^{\frac{1}{2}} + \varepsilon_1(\omega) \right\}^{\frac{1}{2}} \quad (5)$$

Fig. 6(a) shows the reflectivity of hexagonal DyInO₃, HoInO₃ and ErInO₃, from which we can see that the strongest peak locates at around 10 eV. A similar trend as the absorption spectra can be found for the second peak which shows a shift with the increase of atomic number of R. Again, only DyInO₃ has a reflectance peak at the energy of around 45 eV, the same as observed in the absorption spectra. In the energy range from 0 to 6 eV and 12 to 32 eV, the reflectivity curve of hexagonal RInO₃ is lower than 0.2, showing that these three structures are transparent for photons in these energy ranges.

Fig. 6(b) shows the refractive index of unit-cell hexagonal DyInO₃, HoInO₃ and ErInO₃. The refractive peak in the energy range from 0 to 10 eV arise from the transition from the O 2p to

In 5s states, while a relatively small refractive peak at an energy of around 26 eV corresponds to the transition from the In 5p states to the conduction band. We can see from the figure that the static refractive index of DyInO₃, HoInO₃ and ErInO₃ are calculated to be 2.130, 2.127 and 2.124 respectively, which are in good consistency with the trend in the dielectric constant curve as shown in Fig. 4(a). The high refractive index of DyInO₃, HoInO₃ and ErInO₃ appearing in the low energy range reflect a high band gap,^{32,33} which is consistent with the band structure calculations.

We calculate the electron energy loss function $L(\omega)$ and molar extinction coefficient $k(\omega)$ of hexagonal DyInO₃, HoInO₃ and ErInO₃ according to the imaginary part of the dielectric function,

$$L(\omega) = \frac{\varepsilon_2(\omega)}{\varepsilon_1^2(\omega) + i\varepsilon_2^2(\omega)} \quad (6)$$

$$k(\omega) = \frac{1}{\sqrt{2}} \left\{ [\varepsilon_1^2(\omega) + \varepsilon_2^2(\omega)]^{\frac{1}{2}} - \varepsilon_1(\omega) \right\}^{\frac{1}{2}} \quad (7)$$

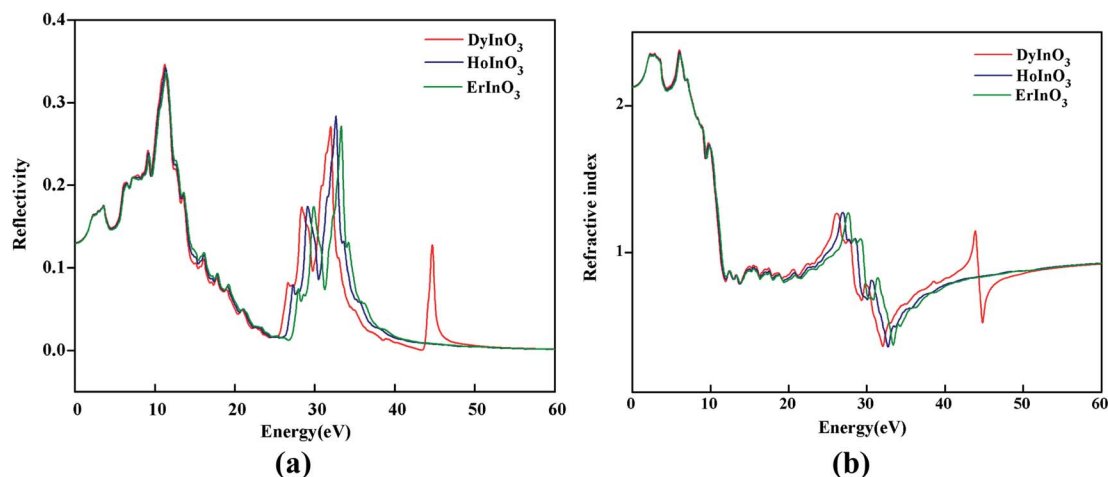


Fig. 6 The reflectivity (a) and refractive index (b) of hexagonal DyInO₃, HoInO₃ and ErInO₃.



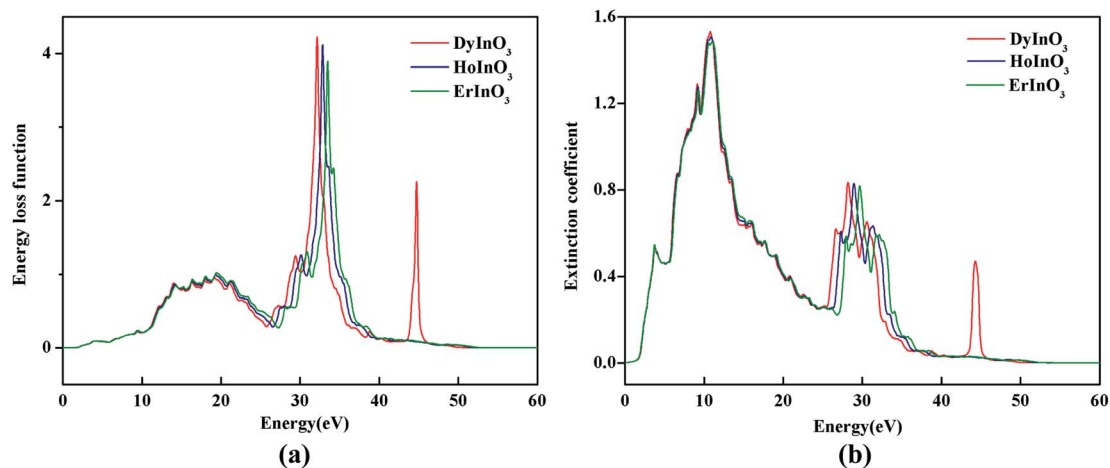


Fig. 7 The electron energy loss spectrum (a) and molar extinction coefficient (b) of hexagonal DyInO₃, HoInO₃ and ErInO₃.

Fig. 7(a) shows the electron energy loss spectrum of hexagonal DyInO₃, HoInO₃ and ErInO₃. The electron energy loss spectrum is the inelastic scattering of the incident electron beam in DyInO₃, HoInO₃ and ErInO₃, and explores the electronic structure of them. We can see from Fig. 7(a) that there are 5 weak peaks in the electron energy loss spectrum of DyInO₃, HoInO₃ and ErInO₃ in the energy range from 14 to 21 eV. At an energy of around 33 eV, they both have a narrow and sharp peak and the ordering of the energy loss function is DyInO₃ > HoInO₃ > ErInO₃. The peaks of DyInO₃, HoInO₃ and ErInO₃ occur at the energy of 32.2 eV, 33.0 eV and 33.5 eV respectively, which are related to the decrease of refractive index of DyInO₃, HoInO₃ and ErInO₃ as shown in Fig. 6(b). It indicates that the unit-cell of hexagonal DyInO₃, HoInO₃ and ErInO₃ can be used as effective optical storage materials at the energy of around 33 eV.

Molar extinction coefficient, also known as molar absorptivity, refers to the extinction coefficient at a concentration of 1 mol L⁻¹. When the unit of concentration is expressed in g L⁻¹, the molar absorptivity $k(\omega)$ is numerically equal to the product of the extinction coefficient and the molecular weight (M) of the substance,

$$k(\omega) = \alpha(\omega)M \quad (8)$$

Fig. 7(b) shows the molar extinction coefficient of unit-cell hexagonal DyInO₃, HoInO₃ and ErInO₃. We can see the extinction peaks of DyInO₃, HoInO₃ and ErInO₃ are the same as their absorption peaks at the energy of around 30 eV. However, unlike the trend of the absorption spectrum, the extinction peaks appearing at around 12 eV are larger than the extinction peak at 30 eV, which is also in good consistency with the trend of the reflectivity curves as shown in Fig. 6(a).

4 Conclusions

In this paper, we have studied the electronic structure, the optical and dielectric properties of hexagonal DyInO₃, HoInO₃ and ErInO₃ by performing first principles studies. After optimization, the lattice parameters are in great agreement with the

experimental values within an error of 1%. Our calculation has determined that the three hexagonal RInO₃ in present study are indirect band gap crystals with band-gap values decreasing with the increase of atomic number. DOS calculations indicate that the electronic properties of hexagonal DyInO₃, HoInO₃ and ErInO₃ are mainly provided by the strong bonding of In and O whereas lanthanides show limited effect. The dielectric properties and ionic contribution are calculated and a comparison has been made on the dielectric function including the static dielectric tensor. We have also calculated the absorption coefficient, the reflectivity, the refractive index, the conductivity and the electron energy-loss spectrum of hexagonal DyInO₃, HoInO₃ and ErInO₃. The present study reveals that the dependence of the optical and dielectric properties of RInO₃ on the rare-earth atomic number is not evident, which also can be a useful guideline in the manufacture of multifunctional devices based on RInO₃.

Conflicts of interest

There are no conflicts to declare.

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References

- 1 D. W. Bondurant and F. P. Gnadinger, Ferroelectrics for nonvolatile RAMs, *IEEE Spectrum*, 1989, **26**, 30–33.
- 2 C. A. Paz de Araujo, J. D. Cuchiaro, L. D. McMillan, M. C. Scott and J. F. Scott, Fatigue-free ferroelectric capacitors with platinum electrodes, *Nature*, 1995, **374**, 627–629.



- 3 R. Ramesh and N. A. Spaldin, Multiferroics: progress and prospects in thin films, *Nat. Mater.*, 2007, **6**, 21–29.
- 4 M. Bibes and A. Barthélémy, Towards a magnetoelectric memory, *Nat. Mater.*, 2008, **7**, 425–426.
- 5 E. Cockayne and B. P. Burton, Phonons and Static Dielectric Constant in CaTiO₃ from First Principle, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2000, **62**, 3735.
- 6 G. A. Samara, The relaxational properties of compositionally disordered ABO₃ perovskites, *J. Phys.: Condens. Matter*, 2003, **15**, R367–R411.
- 7 T. Katsufuji, M. Masaki, A. Machida, M. Moritomo, K. Kato, E. Nishibori, M. Takata, M. Sakata, K. Ohoyama, K. Kitazawa and H. Takagi, Crystal structure and magnetic properties of hexagonal RMnO₃ (R=Y, Lu and Sc) and the effect of doping, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2002, **66**, 134434.
- 8 C. J. Fennie and K. M. Rabe, Polar phonons and intrinsic dielectric response of the ferromagnetic insulating spinel CdCr₂S₄ from first principles, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2005, **72**, 1–4.
- 9 Q. Zhang, G. Tan, L. Gu, Y. Yao, C. Jin, Y. Wang, X. Duan and R. Yu, Direct Observation of Multiferroic Vortex Domains in YMnO₃, *Sci. Rep.*, 2013, **3**, 1–5.
- 10 M. G. Han, Y. Zhu, L. Wu, T. Aoki, V. Volkow, X. Wang, S. C. Chae, Y. S. Oh and S. W. Cheong, Ferroelectric Switching Dynamics of Topological Vortex Domains in a Hexagonal Manganite, *Adv. Mater.*, 2013, **25**, 2415–2421.
- 11 M. Lilienblum, T. Lottermoser, S. Manz, S. M. Selbach, A. Cano and M. Fiebig, Ferroelectricity in the multiferroic hexagonal manganites, *Nat. Phys.*, 2015, **11**, 1070–1073.
- 12 B. B. V. Aken, T. T. M. Palstra, A. Filippetti and N. A. Spaldin, The origin of ferroelectricity in magnetoelectric YMnO₃, *Nat. Mater.*, 2004, **3**, 164–170.
- 13 Z. J. Huang, Y. Cao, Y. Y. Xue and C. W. Chu, Coupling between the ferroelectric and antiferromagnetic orders in YMnO₃, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1997, **56**, 2623.
- 14 C. L. Lin, J. Liu, Y. C. Lin, X. D. Li and R. Li, Pressure-induced structural and vibrational evolution in ferroelectric RInO₃ (R=Eu, Gd, Dy), *Solid State Commun.*, 2013, **173**, 51–55.
- 15 Y. B. Li, Y. Z. Wang, W. J. Tan, W. B. Wang, J. J. Zhang, J. W. Kim, S. W. Cheong and X. T. Tao, Laser floating zone growth of improper geometric ferroelectric GdInO₃ single crystals with Z₆ topological defects, *J. Mater. Chem. C*, 2018, **6**, 7024.
- 16 S. C. Abrahams, Ferroelectricity and structure in the YMnO₃ family, *Acta Crystallogr., Sect. B: Struct. Sci.*, 2001, **57**, 485–490.
- 17 R. Shukla, V. Grover, S. K. Deshpande, D. Jain and A. K. Tyagi, Synthesis and Structural and Electrical Investigations of a Hexagonal Y1-xGdxInO₃ (0.0 ≤ x ≤ 1.0) System Obtained via Metastable C-Type Intermediates, *Inorg. Chem.*, 2013, **52**, 13179–13187.
- 18 R. Shukla, F. N. Sayed, V. Grover, S. K. Deshpande, A. Guleria and A. K. Tyagi, Quest for Lead Free Relaxors in YIn1-xFexO₃ (0.0 ≤ x ≤ 1.0) System: Role of Synthesis and Structure, *Inorg. Chem.*, 2014, **53**, 10101–10111.
- 19 V. Grover, R. Shukla, D. Jain, S. K. Deshpande, A. Arya, C. G. S. Pillai and A. K. Tyagi, Complex GdSc1-xInxO₃ Oxides: Synthesis and Structure Driven Tunable Electrical Properties, *Chem. Mater.*, 2012, **24**, 2186–2196.
- 20 B. Paul, S. Chatterjee, S. Gop, A. Roy, V. Grover, R. Shukla and A. K. Tyagi, Evolution of lattice dynamics in ferroelectric hexagonal REInO₃ (RE = Ho, Dy, Tb, Gd, Eu, Sm) perovskites, *Mater. Res. Express*, 2016, **3**, 075703.
- 21 H. Sawamoto, High-Pressure Synthesis of Perovskite Type RInO₃ (R=Eu, Gd, and Dy), *Jpn. J. Appl. Phys.*, 1973, **12**, 1432.
- 22 T. Tohei, H. Moriwake, H. Murata, A. Kuwabara, R. Hashimoto, T. Uamamoto and I. Tanaka, Geometric ferroelectricity in rare-earth compounds RGaO₃ and RInO₃, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2009, **79**, 744125.
- 23 C. W. F. T. Pistorius and G. J. Kruger, Stability and structure of noncentrosymmetric hexagonal LnInO₃ (Ln = Eu, Gd, Tb, Dy, Ho, Y), *J. Inorg. Nucl. Chem.*, 1976, **38**, 1471.
- 24 J. P. Perdew, K. Burke and M. Ernzerhof, Generalized Gradient Approximation Made Simple, *Phys. Rev. Lett.*, 1996, **77**, 3865.
- 25 G. Kresse and J. Furthmüller, Efficient iterative schemes for ab initio total-energy calculations using a plane-wave basis set, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1996, **54**, 11169.
- 26 G. Kresse and J. Furthmüller, Efficiency of ab-initio total energy calculations for metals and semiconductors using a plane-wave basis set, *Comput. Mater. Sci.*, 1996, **6**, 15.
- 27 H. J. Monkhorst and J. D. Pack, Special points for Brillouin-zone integrations, *Phys. Rev. B: Solid State*, 1976, **13**, 5188–5192.
- 28 C. W. F. T. Pistorius and G. J. Kruger, Stability and structure of noncentrosymmetric hexagonal LnInO₃ (Ln = Eu, Gd, Tb, Ho, Y), *J. Inorg. Nucl. Chem.*, 1976, **38**, 1471–1475.
- 29 S. J. Clark, M. D. Segall, C. J. Pickard, P. J. Hasnip, M. I. J. Probert, K. Refson and M. C. Payne, First principles methods using CASTEP, *Z. Kristallogr.*, 2005, **220**, 567–570.
- 30 S. Saha, T. P. Sinha and A. Mookerjee, Electronic structure, chemical bonding, and optical properties of paraelectric BaTiO₃, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2000, **62**, 8828–8834.
- 31 C. Chen, T. Sasaki, R. Li, Y. Wu, Z. Lin, Y. Mori, Z. Hu, J. Wang, G. Aka, M. Yoshimura and others, Theoretical Basis for the Development of Borate Nonlinear Optical Crystals, *Nonlinear Optical Borate Crystals, Principals and Applications*, John Wiley & Sons, 2012.
- 32 B. Luo, X. Wang, E. Tian, G. Li and L. Li, Electronic structure, optical and dielectric properties of BaTiO₃/CaTiO₃/SrTiO₃ ferroelectric superlattices from first-principles calculations, *J. Mater. Chem. C*, 2015, **3**, 8625–8633.
- 33 X. D. Zhang, M. L. Guo, W. X. Li and C. L. Liu, First-principle study of electronic and optical properties in wurtzite Zn1-xCdxCdO, *J. Appl. Phys.*, 2008, **103**, 063721.

