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## A surface precleaning strategy intensifies the interface coupling of the $\text{Bi}_2\text{O}_3/\text{TiO}_2$ heterostructure for enhanced photoelectrochemical detection properties<sup>†</sup>

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The interfacial coupling effect plays a crucial role in tailoring the photoelectrochemical performance of heterostructured photocatalysts. However, it is an urgent need but challenging to intensify the interfacial coupling effect of heterojunctions. Herein, we proposed the surface precleaning of n-type  $\text{TiO}_2$  via a facile low-temperature hydrogenation to facilitate strong coupling with p-type  $\text{Bi}_2\text{O}_3$  ( $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$ ) and thus disruptively accelerate the electron transfer and electron–hole pair separation. The comparative studies of uncleared  $\text{Bi}_2\text{O}_3/\text{TiO}_2$  and  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  heterostructures by X-ray photoelectron spectroscopy revealed an unneglected valence change and thus a highly strong coupling effect between  $\text{Bi}_2\text{O}_3$  and the cleaned  $\text{TiO}_2$  originating from a possible weakening role of the nitrogen species adsorbed on the surface of pristine  $\text{TiO}_2$  for the interfacial coupling effect. When integrated into a photoelectrochemical sensor,  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  presented both significantly high detection photocurrent response and selectivity for organics in a buffer solution. The current response of the as-built  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  was two-fold higher than that of uncleared  $\text{Bi}_2\text{O}_3/\text{TiO}_2$  and was even almost five-fold that of bulk  $\text{TiO}_2$ . We believe that this work will provide new perspectives and insights into the construction of efficient heterojunctions for impressive applications in photoelectrochemical detection.

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## Introduction

Water and environmental issues are listed among the top ten problems facing humanity for the next 50 years, where wastewater pollution affects the water and environment in many ways; in particular, organic matter pollution in water has caused widespread concern.<sup>1–3</sup> It is essential to detect the pollution of organic compounds in water to improve the water quality and unburden the environment load. The traditional determination method of potassium dichromate oxidation has been widely used for a relatively long period.<sup>4,5</sup> However, some limitations still exist in this approach, including a long reflux time, low detection sensitivity, and secondary pollution.

Hence, a couple of investigations have been done to improve the detection of the organic matter pollutants in water. The past decade has witnessed an explosive interest in constructing photoelectrochemical sensors based on photocatalysts due to their excellent intrinsic properties, such as non-toxicity, high oxidation capacity, and low cost.<sup>6–8</sup> Impressively, Zhou and colleagues reported a new type of organic detector based on the photoelectrocatalytic activity of  $\text{TiO}_2$  nanotube arrays ( $\text{TiO}_2$  NTAs), which endow no any toxic oxidants.<sup>9,10</sup> The development of a novel photoelectrochemical sensor based on  $\text{TiO}_2$  NTAs via constructing the relation between the photocurrent and organic concentration in water has become a research object in recent years.<sup>11–16</sup> However, due to the wide band gap of  $\text{TiO}_2$  and fast recombination of photogenerated carriers, the efficiency of photoelectrochemical detection is still extremely limited.

Effective photogenerated charge pairs are generally believed to be the key factor affecting the performances of semiconductor-based photocatalysts, which however will be counteracted if heterostructure engineering is not taken into account. The construction of a heterojunction by rationally integrating the appropriate components and selecting a suitable band gap has been well proven as an effective method to both suppress the recombination of photogenerated electron–hole pairs and

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accelerate the transport of the photocarriers of  $\text{TiO}_2$ .<sup>17–19</sup> For example,  $\text{Bi}_2\text{O}_3$ -modified  $\text{TiO}_2$  NTAs have been successfully constructed to realize the regulation of the photoelectrochemical reaction process of  $\text{TiO}_2$  NTAs when applied as a photoelectrochemical sensor.<sup>16,20</sup> However, the large charge transport resistance in the photoelectrochemical detection process decreases the efficiency of the photoelectrochemical reaction on the  $\text{Bi}_2\text{O}_3/\text{TiO}_2$  heterostructure. Therefore, the insufficient coupling function in the above-mentioned heterojunction urgently needs to be improved for further enhancing the photoelectrochemical detection efficiency.

Herein, we demonstrated the surface precleaning of n-type  $\text{TiO}_2$  *via* a facile low-temperature hydrogenation to facilitate strong coupling with p-type  $\text{Bi}_2\text{O}_3$  ( $\text{Bi}_2\text{O}_3/c\text{-TiO}_2$ ) and thus disruptively accelerate the electron transfer and electron–hole pair separation. Specifically, the adsorbed nitrogen species on the surface of  $\text{TiO}_2$  were found to weaken the coupling effect in the  $\text{Bi}_2\text{O}_3/\text{TiO}_2$  heterojunction system. The strong coupling effect in  $\text{Bi}_2\text{O}_3/c\text{-TiO}_2$  was demonstrated by obvious changes in the valence states of the Ti and Bi elements. Consequently, compared to the bulk  $\text{Bi}_2\text{O}_3/\text{TiO}_2$  heterostructure,  $\text{Bi}_2\text{O}_3/c\text{-TiO}_2$  as a photoelectrochemical sensor exhibited fast charge transfer and highly selected electrochemical surface reactions, resulting in both a high detection photocurrent response and selectivity for organic targets in a buffer solution. The complicated coupling mechanism was also systematically investigated and discussed.

## Experimental

### Chemicals and sample preparation

Ti foils with a purity of 99.7% were purchased from Cuibolin (Beijing). Ethylene glycol, ammonium fluoride, disodium hydrogen phosphate ( $\text{Na}_2\text{HPO}_4$ ), sodium dihydrogen phosphate ( $\text{NaH}_2\text{PO}_4$ ), and absolute ethanol were purchased from Sinopharm Chemical Reagent.

Before anodization, the Ti foil was washed in acetone, DI water, and ethanol for 20 minutes each.  $\text{TiO}_2$  NTAs were fabricated *via* two-step anodization. First, in a self-made double-electrode cell, the Ti foil with a diameter of 2.5 cm was anodized at a voltage of 60 V for 2 h. Specifically, in glycol solution containing 0.15 M  $\text{NH}_4\text{F}$  and 5 vol%  $\text{H}_2\text{O}_2$ , a graphite plate was used as the counter electrode and the Ti foil was used as the working electrode. Then, the obtained  $\text{TiO}_2$  layer was cleared by ultrasonication in DI water for 10 min. The second anodization was performed at the same voltage of 60 V with the anodization time of 6 h. Finally, the products were ultrasonically cleaned in ethylene glycol to remove the broken nanotubes covered on the top surface of  $\text{TiO}_2$  NTAs and then dried in an oven at 60 °C. The crystallization of the as-obtained amorphous  $\text{TiO}_2$  was achieved by annealing at 500 °C for 2 h in air.

The two-step anodization of  $\text{TiO}_2$  NTAs and the following preparation process are illustrated in Fig. 1. Precleaned  $\text{TiO}_2$  NTAs ( $c\text{-TiO}_2$  NTAs) were obtained by a low-temperature hydrogen thermal annealing method, in which  $\text{TiO}_2$  was placed in a tube furnace and heated to 300 °C in a hydrogen environment at a heating rate of 1 °C min<sup>−1</sup> for 4 h.  $\text{Bi}_2\text{O}_3$  was loaded on  $c\text{-TiO}_2$  NTAs by an ultrasonication-assisted successive ionic layer

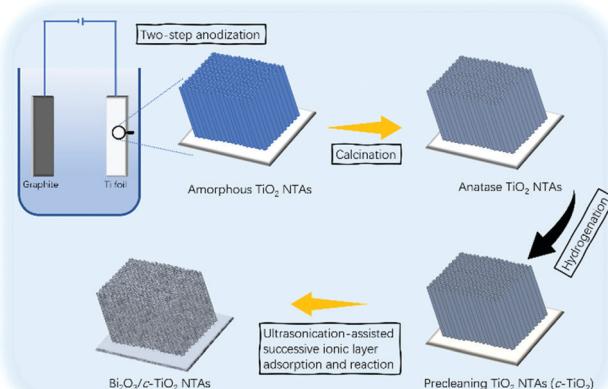


Fig. 1 Schematic illustration for the preparation of  $\text{Bi}_2\text{O}_3/c\text{-TiO}_2$  NTAs.

adsorption and reaction strategy. First, the treated  $\text{TiO}_2$  was soaked in a 10 mM  $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$  ethylene glycol solution and a 0.1 M NaOH ethanol solution for 10 minutes, respectively, using an ultrasonic generator. The product was then thoroughly rinsed in ethanol to remove any residual solution between each soaking step. This operation was repeated for 3 cycles. Then, the products were dried in an oven and annealed at 500 °C for 2 h in a tube furnace at a heating rate of 5 °C min<sup>−1</sup>.

### Characterization

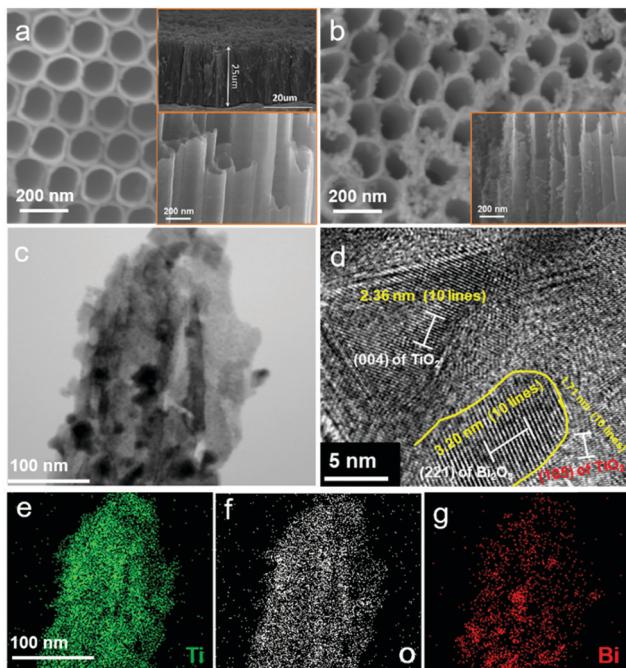
The phase structures of the products were analyzed by using X-ray diffraction (XRD) with  $\text{Cu-K}\alpha$  radiation (D/MAX2500V). The morphologies were characterized by using field emission scanning electron microscopy (FE-SEM, SU8020) and high-resolution transmission electron microscopy (HR-TEM, JEOL JEM-2100F). The elemental composition of the samples was investigated by X-ray photoelectron spectroscopy (XPS, CALAB250) using Al  $\text{K}\alpha$  monochromatized radiation. Photoluminescence (PL) spectra were obtained on an F-4500 fluorescence spectrophotometer. UV-vis optical absorption was recorded by using a Hitachi UV-3600 spectrophotometer (Japan), and  $\text{BaSO}_4$  was used as a reference.

### Photoelectrochemical measurements

The photoelectrochemical performances were evaluated in a self-made circulatory system using a CHI660D electrochemical workstation.<sup>16</sup> The obtained nanotube arrays were used as the working electrode. A 3 M KCl saturated Ag/AgCl electrode and a Pt electrode were used as the reference electrode and counter electrode, respectively. The supporting electrolyte was 0.05 M phosphate buffer solution having a pH of 7 by adjusting the ratio of  $\text{Na}_2\text{HPO}_4$  to  $\text{NaH}_2\text{PO}_4$ . A UV LED was used as the light source with a spot diameter of 10 mm, fixed wavelength at 365 nm, and adjustable optical power from 0 to 1200 mW cm<sup>−2</sup> (an optical power of 4% was applied in this study).

## Results and discussion

To explore the difference in the morphologies of these samples, Fig. 2a–c and Fig. S1 (ESI†) show the SEM images of  $\text{TiO}_2$ ,  $c\text{-TiO}_2$ ,



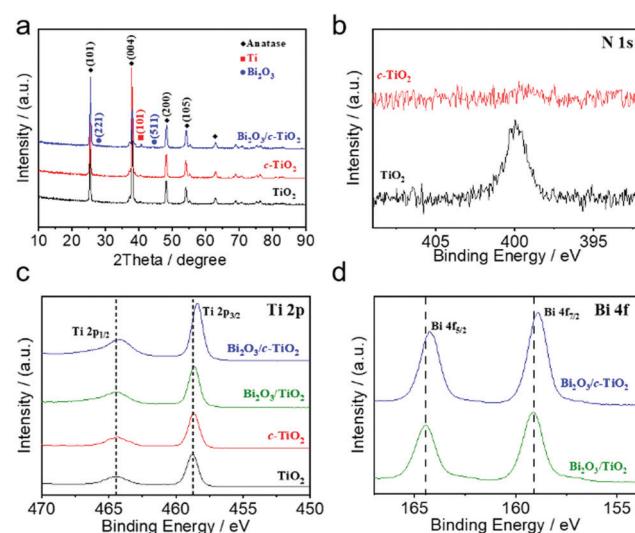
**Fig. 2** Morphology and structural characterization. (a and b) SEM images of  $\text{TiO}_2$  and  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  NTAs. The corresponding insets show cross-sectional images. (c-g) TEM images at different magnifications and the corresponding EDS element mapping images of Ti, Bi, and O for  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  NTAs.

and  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  NTAs. A large area of highly ordered nanotube arrays can be observed in the SEM image for the as-synthesized  $\text{TiO}_2$  (Fig. S1a, ESI<sup>†</sup>). The  $\text{c-TiO}_2$  showed similar morphology to that of the as-received  $\text{TiO}_2$  NTAs with high density and a well-ordered and uniform tubular structure (Fig. 2a and Fig. S1b, ESI<sup>†</sup>), indicating that there was no obvious change in the surface morphology after the hydrogen thermal treatment. Moreover, with the ultrasonication-assisted method, ultra-small-sized  $\text{Bi}_2\text{O}_3$  was successfully deposited on  $\text{c-TiO}_2$  NTAs (Fig. 2b). In addition, the controlled introduction of  $\text{Bi}_2\text{O}_3$  could be simply achieved by regulating the cycle of the deposition process (see SEM images in Fig. S1c, ESI<sup>†</sup> and Fig. 2b). The TEM image of  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  NTAs is shown in Fig. 2c, where  $\text{Bi}_2\text{O}_3$  with an ultra-small size can be observed to be evenly deposited on  $\text{TiO}_2$  NTAs. As observed in the high-resolution TEM image (Fig. 2d), the typical lattice spacings of 0.236 nm, 0.171 nm, and 0.320 nm correspond to the (111) and (105) planes of the  $\text{TiO}_2$  anatase phase (JCPDS file No. 21-1272) and the (221) plane of  $\text{Bi}_2\text{O}_3$  (JCPDS file No. 29-0236).<sup>21-23</sup> The corresponding EDS element mapping (Fig. 2e-g) demonstrates that the as-prepared  $\text{Bi}_2\text{O}_3$  is highly dispersed on the  $\text{c-TiO}_2$  nanotubes. Specifically, except for the Ti and O elements, the as-built  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  sample has an atomic Bi percentage of about 5.89% (Fig. S1d, ESI<sup>†</sup>).

The crystal structures and phases of all the studied samples were identified by XRD patterns and Raman spectra. As displayed in Fig. 3a,  $\text{c-TiO}_2$  NTAs show similar diffraction peaks to those of  $\text{TiO}_2$  NTAs. The diffraction peaks at  $25.36^\circ$  and  $37.91^\circ$  for  $\text{TiO}_2$  and  $\text{c-TiO}_2$  NTAs can be well indexed to the (101) and (004) planes of the  $\text{TiO}_2$  anatase phase (JCPDS file No. 21-1272),

respectively, indicating that the hydrogenation treatment did not change the phase and crystal properties of  $\text{TiO}_2$  NTAs. The  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  NTA sample showed some new peaks at the  $2\theta$  values of  $27.46^\circ$  and  $44.96^\circ$ , corresponding to the (310) and (431) planes of  $\text{Bi}_2\text{O}_3$  (JCPDS file No. 29-0236).<sup>23,24</sup> Furthermore, in the Raman spectra (Fig. S3a, ESI<sup>†</sup>), no differences can be found between the peaks for  $\text{TiO}_2$  and  $\text{c-TiO}_2$ , and these characteristic peaks at 141, 192, 392, 512, and 634  $\text{cm}^{-1}$  belong to anatase  $\text{TiO}_2$ . However, after  $\text{Bi}_2\text{O}_3$  deposition, an obvious peak of  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  NTAs appeared at  $309 \text{ cm}^{-1}$  due to the existence of Bi-O bonds with various bond lengths.<sup>23</sup> Both the XRD and Raman results demonstrated that  $\text{Bi}_2\text{O}_3$  was successfully deposited on  $\text{c-TiO}_2$  NTAs *via* such a facile ultrasonication-assisted method.

To further analyze the surfacial/interfacial chemical valence and coupling effect of the samples, the XPS survey spectra were performed. Two main peaks of O 1s and Ti 2p could be observed due to the existence of O and Ti, respectively, for all the samples (see the overall survey patterns in Fig. S3b, ESI<sup>†</sup>). First, the high-resolution spectra of N 1s for  $\text{TiO}_2$  and  $\text{c-TiO}_2$  were recorded to compare the surface functional groups. In the region of N 1s (Fig. 3b), there is almost no N 1s XPS peak in the measured spectrum of  $\text{c-TiO}_2$ , while it is evidently observed for pristine  $\text{TiO}_2$ . According to previous reports, the peak at  $\sim 400.2 \text{ eV}$  can be assigned to various adsorbed nitrogen-containing species.<sup>25,26</sup> Therefore,  $\text{TiO}_2$  with clean surfaces was obtained by a low-temperature hydrogenation treatment. The comparison of the Ti 2p spectra of  $\text{TiO}_2$  and  $\text{c-TiO}_2$  NTAs (Fig. 3c) suggested no obvious binding energy shift. Moreover, unlike the unclean  $\text{Bi}_2\text{O}_3/\text{TiO}_2$  sample,  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  NTAs presented an obvious shift in the Ti 2p peaks compared with  $\text{TiO}_2$  NTAs, suggesting a strong interaction between the Ti atoms and the adjacent atoms.<sup>27-29</sup> A highly strong coupling between  $\text{Bi}_2\text{O}_3$  and  $\text{c-TiO}_2$  was thus established, which holds great potential for



**Fig. 3** Phase and surface termination species analysis of  $\text{TiO}_2$ ,  $\text{c-TiO}_2$ ,  $\text{Bi}_2\text{O}_3/\text{TiO}_2$ , and  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  NTAs. (a) XRD patterns. (b-d) High-resolution XPS spectra of N 1s, Ti 2p, and Bi 4f.

accelerating the electron transfer and electron-hole pair separation. Therefore, we can also conclude that the adsorbed radicals on the surface of semiconductors can weaken the coupling function between  $\text{Bi}_2\text{O}_3$  and  $\text{TiO}_2$ , leading to an unexpected resistance enhancement in heterostructure engineering. Additionally, the high-resolution XPS spectra of Bi 4f in  $\text{Bi}_2\text{O}_3/\text{TiO}_2$  and  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  are shown in Fig. 3d. Two peaks belonging to Bi 4f<sub>7/2</sub> and Bi 4f<sub>5/2</sub> can be observed. Obviously, the binding energy for  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  NTAs presented a negative shift of  $\sim 0.2$  eV when compared to that for  $\text{Bi}_2\text{O}_3/\text{TiO}_2$  NTAs, further demonstrating the strong coupling effect in the as-synthesized  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  NTAs. In order to further confirm the stability of the strong coupling, the high-resolution Ti 2p and Bi 4f XPS spectra of  $\text{Bi}_2\text{O}_3/\text{TiO}_2$  and  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  after detection tests for 10 cycles were measured (see Fig. S5, ESI<sup>†</sup>). Clearly, the Ti 2p and Bi 4f XPS spectra of  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  still presented a visible shift compared with that of  $\text{Bi}_2\text{O}_3/\text{TiO}_2$ , which confirmed the excellent stability of the strong coupling interaction.

When applied in the PEC detection of organics in aqueous solutions, the strength of the current response and selectivity between the base solution and organics are crucial for the construction of a high-performance PEC sensor. Thereby, curves for the background photocurrent (decomposition from a buffer solution) and relative current response to organics (decomposition from a 0.1 mM glucose target) were obtained by the amperometric method using a self-made flow-injection device at an applied potential of 0.2 V in a buffer solution (Fig. 4a and b).<sup>16</sup> On the one hand, by comparing the background photocurrents among the four samples, both  $\text{TiO}_2$  and  $\text{c-TiO}_2$  exhibited higher background photocurrents than  $\text{Bi}_2\text{O}_3/\text{TiO}_2$  and  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$ , showing that the introduction of  $\text{Bi}_2\text{O}_3$  suppressed the photolysis of water during the detection processes. Specifically, the background photocurrents of  $\text{TiO}_2$ ,  $\text{c-TiO}_2$ ,  $\text{Bi}_2\text{O}_3/\text{TiO}_2$ , and  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  NTAs were measured to be 136.82  $\mu\text{A}$ , 261.63  $\mu\text{A}$ , 57.83  $\mu\text{A}$ , and 32.68  $\mu\text{A}$ , respectively (Fig. 4b). A limited background photocurrent is beneficial to obtain efficient selectivity. On the other hand, it could be observed that the current responses to glucose of the corresponding four samples were 6.73  $\mu\text{A}$ , 12.84  $\mu\text{A}$ , 14.52  $\mu\text{A}$ , and 29.12  $\mu\text{A}$ , which suggested that both a low background current and enhanced current response were achieved for the  $\text{Bi}_2\text{O}_3/\text{TiO}_2$  and  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  samples. Moreover, by comparing the performances of the heterostructure with and without precleaning, the enhanced coupling effect in  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  was seen to play a great role in enhancing the current response. Impressively, the as-prepared  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  exhibited a response two-fold higher than that of blank  $\text{Bi}_2\text{O}_3/\text{TiO}_2$  and even 5 times that of bulk  $\text{TiO}_2$ .

To further determine the photoelectrochemical detection performance including the sensitivity, current detection noise, linear range and detection limit of the as-prepared samples, the photoelectrochemical performances of the as-assembled four samples were obtained *via* the step-by-step infusion of a glucose target (Fig. 4c). For convenient comparison, the current-time curves were moved to the same starting point. Obviously, the current increment of  $\text{c-TiO}_2$ ,  $\text{Bi}_2\text{O}_3/\text{TiO}_2$ , and  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  went

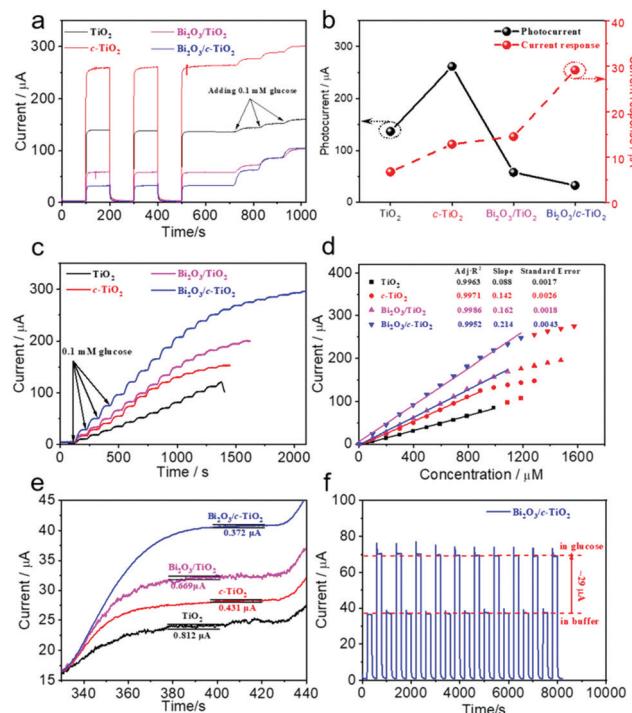


Fig. 4 Photoelectrochemical properties of  $\text{TiO}_2$ ,  $\text{Bi}_2\text{O}_3/\text{TiO}_2$ ,  $\text{c-TiO}_2$ , and  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  NTAs. (a and b) Photocurrent and current response curves. (c and d) Current-time curves and plots of current increment vs. concentration. (e) Detection current noise. (f) Stability test.

faster than that for bulk  $\text{TiO}_2$  with the same injection, indicating a higher response current to glucose, especially for  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  NTAs. Furthermore, plots of the current increment *vs.* detection concentration range were obtained by calculating from Fig. 4c, where the detection sensitivity and range could be obtained (Fig. 4d). The as-built  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  NTAs presented both highly improved sensitivity ( $0.214 \mu\text{A} \mu\text{M}^{-1}$ ) and range ( $1185.5 \mu\text{M}$ ) performances, which were much better than those of others. To vividly understand the influence of the background photocurrent on the photoelectrochemical detection of organics, the current noises of these samples with the first injection of glucose in the current-time curves are presented in Fig. 4e. The current noises of  $\text{TiO}_2$ ,  $\text{c-TiO}_2$ ,  $\text{Bi}_2\text{O}_3/\text{TiO}_2$ , and  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  were  $0.812 \mu\text{A}$ ,  $0.431 \mu\text{A}$ ,  $0.669 \mu\text{A}$ , and  $0.372 \mu\text{A}$ , respectively. In addition, the detection limit (dl) was obtained by the sensitivity and current noise data ( $dl = 3\sigma/m$ , where  $\sigma$  is the background current noise, and  $m$  is the slope of the linear part of the calibration curve).<sup>12,30,31</sup> The dl values of the corresponding samples were  $27.68 \mu\text{M}$ ,  $8.73 \mu\text{M}$ ,  $12.39 \mu\text{M}$ , and  $5.21 \mu\text{M}$ . The detailed determination performance parameters are listed in Table S1 (ESI<sup>†</sup>) for better comparison. The constructed  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  NTAs also presented superior stability (Fig. 4f). By comparison, we can conclude that the photoelectrochemical detector based on  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  NTAs presents superior detecting performances with the sensitivity of  $0.214 \mu\text{A} \mu\text{M}^{-1}$ , detection limit of  $5.21 \mu\text{M}$ , and linear range from 0 to  $1185.8 \mu\text{M}$ , thus holding a greater potential than others.

In addition to the above-mentioned  $\text{Bi}_2\text{O}_3/\text{c-TiO}_2$  due to the strong coupling effect, which is more favorable for the desorption

of organic decomposition products, the change in photoelectrochemical properties also has an important influence on the detection performance of organics. Therefore, all the complicated photoelectrochemical processes, including optical absorption, charge separation efficiency, transfer rate, and electrochemical surface reactions, are deeply discussed in the following section.

First, optical properties are the key factors for the photoelectrochemical performance, as shown in Fig. 5a; all the samples have excellent optical absorption properties in the UV range. Compared with pristine  $\text{TiO}_2$  NTAs,  $c\text{-TiO}_2$  showed improved absorption in both the UV and visible regions, as described in many previous reports. The light absorption decreased in the UV region with the introduction of  $\text{Bi}_2\text{O}_3$ ; in particular, when taking into account the light wavelength (365 nm) used in this study, it could be found that the modification of  $\text{Bi}_2\text{O}_3$  alone did not improve the light absorption performance. This may be due to the low content of  $\text{Bi}_2\text{O}_3$  in  $\text{TiO}_2$  or  $c\text{-TiO}_2$ . In other words, the reason for the superior photoelectrochemical detection performances was not attributed to the optical properties.

Second, the recombination rate and charge transfer rate of the photogenerated electron–hole pairs were studied by using the photoluminescence (PL) method with an excitation wavelength of 315 nm and electrochemical impedance spectra (EIS) with a range from  $10^{-1}$  to  $10^5$  Hz at a voltage of 0.2 V, respectively (Fig. 5b and c).<sup>32–34</sup> On the one hand, by comparing

the PL intensity of the as-prepared samples, we found that (1) the introduction of  $\text{Bi}_2\text{O}_3$  in both the samples could promote the separation of photogenerated charge carriers and (2)  $\text{Bi}_2\text{O}_3/c\text{-TiO}_2$  exhibited strongly superior separation efficiency when compared with blank  $\text{Bi}_2\text{O}_3/\text{TiO}_2$ . On the other hand, obviously, the Nyquist plots for all four samples presented a similar shape but with different impedance arcs (Fig. 5c). In general, if the radius of the arc becomes smaller, it implies a fast charge transfer rate. The arc radius of  $c\text{-TiO}_2$  was smaller than that of bulk  $\text{TiO}_2$ , indicating better conductivity after surface precleaning, which is consistent with the result of a significant enhancement in the background photocurrent of  $c\text{-TiO}_2$  NTAs in a buffer solution. Impressively, higher charge transfer resistance of both  $\text{Bi}_2\text{O}_3/\text{TiO}_2$  and  $\text{Bi}_2\text{O}_3/c\text{-TiO}_2$  was obtained by the coupling of  $\text{Bi}_2\text{O}_3$ , indicating that the existence of  $\text{Bi}_2\text{O}_3$  could not improve the conductivity for pristine  $\text{TiO}_2$  and  $c\text{-TiO}_2$  in this case. However, with the precleaning treatment of pristine  $\text{TiO}_2$ , the charge transfer resistance was reduced extremely, suggesting enhanced coupling between the heterostructured semiconductors in the  $\text{Bi}_2\text{O}_3/c\text{-TiO}_2$  system, further resulting in highly enhanced photoelectrochemical detection properties.

Last but not the least, the electrochemical surface reaction plays a key role in enhancing the photoelectrochemical detection performances, including the direct oxidative decomposition of organic compounds by holes ( $\text{h}^+$ ) and the indirect oxidation and decomposition of organic matter in water by  $\cdot\text{OH}$  radicals.<sup>16</sup> Thereby, we investigated the change in the photocurrents of the four samples by adding trapping agents for holes and hydroxyl radicals, respectively (Fig. 5d and e). Ammonium oxalate (AO) and isopropanol (IPA) contributed to holes and hydroxyl radicals, respectively.<sup>35,36</sup> Clearly, the main active species for pristine  $\text{TiO}_2$  and  $c\text{-TiO}_2$  were still hydroxyl radicals, indicating that precleaning could not regulate the electrochemical surface reactions of  $\text{TiO}_2$ . However, for  $\text{Bi}_2\text{O}_3$ -containing  $\text{TiO}_2$  and  $c\text{-TiO}_2$ , the holes became the main active species during the PEC processes. Considering that the introduction of  $\text{Bi}_2\text{O}_3$  could enhance the detection response, the holes were more efficient in the decomposition of organics when compared with the hydroxyl radicals. We further proved the intensity of hydroxyl radicals in samples by testing the fluorescence density of 2-hydroxyterephthalic acid when excited by light with a wavelength of 315 nm, which is the product of the reaction of terephthalic acid with hydroxyl radicals (Fig. 5f),<sup>37,38</sup> further supporting the trapping agent results. The emission intensities of  $\text{Bi}_2\text{O}_3/\text{TiO}_2$  and  $\text{Bi}_2\text{O}_3/c\text{-TiO}_2$  NTAs were lower than those of the two pristine samples, confirming the lower productivity of hydroxyl radicals. Clearly, it was demonstrated that the detection action by holes is more advantageous to the PEC detection performances in comparison with the detection by hydroxyl radicals when considering the electrochemical surface reactions.

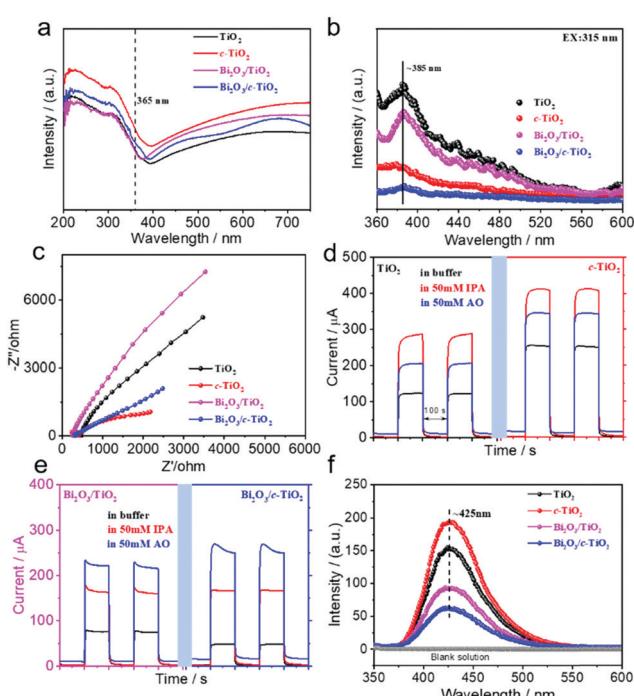


Fig. 5 Optical properties and electrochemical surface reactions of the as-prepared samples. (a) UV-Vis absorption spectra. (b) PL spectra. (c) EIS spectra. (d and e) Trapping experiment for holes and hydroxyl radicals. (f) PL spectra of terephthalic acid after being illuminated under UV light with  $\text{TiO}_2$ ,  $c\text{-TiO}_2$ ,  $\text{Bi}_2\text{O}_3/\text{TiO}_2$ , and  $\text{Bi}_2\text{O}_3/c\text{-TiO}_2$  NTAs for 30 min.

## Conclusions

In summary, we exploited an effective surface precleaning approach to enhance the performance of a photoelectrochemical (PEC) sensor for the detection of organics by intensifying the

coupling effect between  $\text{Bi}_2\text{O}_3$  and precleaned  $\text{TiO}_2$  ( $\text{Bi}_2\text{O}_3/c\text{-TiO}_2$ ). In comparison with the current response of the unclean  $\text{Bi}_2\text{O}_3/\text{TiO}_2$  heterostructure, the current response of the as-built  $\text{Bi}_2\text{O}_3/c\text{-TiO}_2$  was enhanced by two-fold, which was even five-fold that of bulk  $\text{TiO}_2$ . Impressively, we revealed the negative role of the nitrogen-containing radicals adsorbed on the surface of  $\text{TiO}_2$  in constructing the high-performance heterojunction system. Specifically, the photoelectrochemical detector based on  $\text{Bi}_2\text{O}_3/c\text{-TiO}_2$  NTAs presented superior detecting performances, *e.g.*, sensitivity, detection limit, and linear range, thus holding the greatest potential among  $\text{TiO}_2$ ,  $c\text{-TiO}_2$ , and  $\text{Bi}_2\text{O}_3/\text{TiO}_2$  NTAs originating from the accelerated photogenerated charge transfer and proper regulated electrochemical surface reactions by intensifying the interfacial coupling effect.

## Conflicts of interest

There are no conflicts to declare.

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