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# Harnessing photo/electro-catalytic activity *via* nano-junctions in ternary nanocomposites for clean energy

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Though solar energy availability is predicted for centuries, the diurnal and asymmetrical nature of the sun across the globe presents significant challenges in terms of harvesting sunlight. Photo/electro-catalysis, currently believed to be the bottleneck, promises a potential solution to these challenges along with a green and sustainable environment. This review aims to provide the current highlights on the application of inorganic-semiconductor-based ternary nanocomposites for H<sub>2</sub> production and pollutant removal. Various engineering strategies employing integration of 2D materials, 1D nanorods, and/or 0D nanoparticles with inorganic semiconductors to create multiple nano-junctions have been developed for the excellent photocatalytic activity. Following a succinct description of the latest progress in photocatalysts, a discussion on the importance of ternary electrocatalysts in the field of next-generation supercapacitors has been included. Finally, the authors' perspectives are considered briefly, including future developments and critical technical challenges in the ever-growing field of photo/electro-catalysis.

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## 1. Introduction

The unprecedented industrial growth, uncontrolled environmental pollution, and the current energy scenario pose a significant challenge in realizing a carbon-neutral society. The key to this challenge is to expand sustainable and environmentally benign energy systems to avert the impending climate change. Thus, tangible steps are required to have a greener era, wherein one is expected to rely on a renewable and clean energy source that leaves limited and regulated environmental damage. Solar-light-driven energy conversion, environmental purification, and storage technologies *via* photo- and electrocatalysts have drawn tremendous attention. Solar energy is an inexhaustible, clean, and green resource, along with the fact that the sunlight reaching the Earth's surface on an hourly basis surpasses the annual global energy consumption.<sup>1,2</sup>

Since the demonstration of water splitting *via* TiO<sub>2</sub> by employing ultraviolet (UV) excitation light by Fujishima and Honda in 1972,<sup>3</sup> photocatalytic technology incorporating inexhaustible solar energy-driven water splitting and degradation of pollutants has been contemplated as one of the most promising approaches to solve the global energy and environmental crisis. Solar water splitting involving hydrogen production *via*

sunlight is indisputably one of the most imperative approaches to produce clean energy. The produced hydrogen can release energy *via* direct combustion or in a hydrogen fuel cell, with the only by-product being water, whereas degradation of pollutants tackles another major global issue as a significant amount of pollutants are discharged into lakes, rivers, and groundwater due to rapid industrialization. It is known that around 1.8 billion people consume contaminated water. Thus, these harmful compounds must be turned into harmless compounds, which can be realized by exploiting photocatalysts to absorb light and produce active radicals that degrade the pollutants *via* redox processes.

For an active photocatalyst, various parameters such as excitation wavelength, suitable bandgaps, band positions with adequate driving potential, and charge carrier separation play a significant role. Since photocatalytic H<sub>2</sub> production and pollutant degradation hinge on the sun and photocatalyst, not consumed during the process, this is possibly a low-cost and environmentally friendly approach for energy production and pollution abatement. However, though solar energy availability is predicted for centuries, the diurnal and asymmetrical nature of this source across the globe presents significant challenges in terms of harvesting, storage, and utilization. Furthermore, the large bandgap of a commonly used active semiconductor such as TiO<sub>2</sub> (~3.2 eV for anatase) requires an excitation wavelength that falls in the UV region. The UV spectral regime consists of less than ~5% of the total sunlight (solar light consists of ~5% UV, ~45% visible, and ~52% infrared) incident at the

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Earth's surface. In contrast, the photocorrosion of the visible light active CdS ( $\sim 2.42$  eV) catalyst is considered as the Achilles heel of CdS. So far, the research has suggested that a solar-driven energy conversion efficiency of  $\sim 10\%$  under optimal operational conditions could allow the photocatalytic technology to be economically viable for hydrogen production and environmental enrichment.<sup>4,5</sup> However, the current water-splitting technologies using particulate homogeneous photocatalysts on the laboratory scale are only  $\sim 1\%$  efficient.<sup>6</sup> This necessitates the idea of developing highly stable and efficient photocatalytic systems covering a broad range of sunlight to achieve the required performance.

Another efficient and practical way to harness sunlight is to convert it into electricity using photovoltaics.<sup>7</sup> However, the sun's diurnal nature brings out the importance of energy storage systems in electronics, automotive, aerospace, and stationary markets to counterbalance its variability.<sup>8,9</sup> Furthermore, with the persistent demands in miniaturization of electronic devices due to the addition of portable technology into our daily life, there are serious struggles to manufacture small but powerful devices. Among various possible energy storage systems, supercapacitors have enormous potential owing to their superior power density, fast charge/discharge rate, and long cycle life compared to conventional capacitors and batteries. The most widely used electrode materials are carbon, conducting polymers, and transition metal oxides for active supercapacitors. However, these materials do have their drawbacks. For example, graphite has low specific capacitances, and conducting polymers have poor cyclability. Among various available transition metal oxide materials,  $\text{Co}_3\text{O}_4$ <sup>10</sup> and  $\text{MnO}_2$ <sup>11</sup> have attracted much attention as a pseudocapacitive electrode material because of their high theoretical capacitance of  $3560 \text{ F g}^{-1}$  and  $1233 \text{ F g}^{-1}$ , respectively, along with their excellent redox activity, cost-effectiveness, and environmentally friendly nature. However, the performances are often compromised in terms of actual capacity and cycling stability, which originate from the limited redox kinetics for ion diffusion, electron transfer rates, and poor structural stability.<sup>12</sup> The novel material design needs to be implemented to transcend these limitations to achieve unique properties such as an abundant specific surface area and high electrical conductivity to simultaneously improve both the capacitive performance and the stability of the electrode material.

With the research being pioneered in the "conventional catalysts", particularly homogeneous and binary catalysts, the development of "unconventional ternary catalysts" has been stimulated. The idea of harnessing an active system ensemble effect with two different components in one-pot is conceptualized as a "ternary system". The application of the ternary system has been noticed in the field of biosciences.<sup>13</sup> The Nobel Prize (Chemistry, 2004) winning discovery of "ubiquitin-mediated protein degradation" by Aaron Ciechanover, Avram Hershko, and Irwin Rose exhibits one of the most pertinent examples of nature's ternary enzyme system.<sup>14</sup> An attempt to master the "nature-like ternary system", the availability of highly sophisticated growth techniques and atomic-nano scale

characterization tools have inspired scientists to develop highly efficient catalytic systems. Ternary photo/electrocatalysts involve active catalysts with electron transfer mediators such as 2D materials, 1D nanorods (NRs), and 0D nanoparticles (NPs). These heterostructures combine materials with distinctly different physical and chemical properties to yield a unique hybrid nano-system with multiple nano-junctions giving rise to tunable and enhanced properties that may not be attainable otherwise. Although the concept of using a ternary catalyst is sprouting, it has demonstrated its invaluable implications in terms of efficiency and stability. A substantial number of apposite reports utilizing various ternary photo/electrocatalysts have drastically increased in a decade; however, these reports remain scattered in the literature.

Furthermore, we would like to emphasize here that although several review articles have been published on catalysts along with some recent reviews on photocatalysts<sup>15,16</sup> and electrocatalysts;<sup>17</sup> however, none is related to inorganic semiconductor-based ternary systems. A recent extensive review on the photocatalytic activity of heterogeneous nanocomposites has also been reported; however, it primarily focused on the one-step reaction in binary nanocomposites, which produced  $\text{H}_2$ .<sup>18</sup> Another review article discussing the application of the ternary system for the degradation of pollutants has been reported; however, it only involves tungsten element-based ternary systems.<sup>19</sup> This review article, on the other hand, focuses on the photo/electro-catalytic properties of semiconductor-based ternary nanocomposites, which have not been the topic of review so far.

We aim to propose the current advancements in ternary catalysts in a decade, their current status, and possible future directions. The article is divided into two sections, "ternary photocatalyst" and "ternary electrocatalyst", highlighting the importance of multiple nano-junctions in these integrated systems comprising various components of distinct functionalities. In the end, we present our perspective on the current research situation, the challenge in this field, and where this field is headed.

## 2. Ternary photocatalyst

Based on the excellent and different properties of various nanostructures due to the confinement in different dimensions and surface to volume ratio, an active photocatalyst can be well integrated with (i) 2D materials that show excellent mobility of charge carriers and a high surface area, (ii) 1D NRs that show high electrical conductivity and directional flow of charge carriers due to the confinement in the radial direction and large aspect ratio, and (iii) 0D NPs with varying size to tune the excitation wavelength.

### 2.1. Solar water splitting

In the last decade, a substantial amount of work has been carried out on the integration of 2D materials,<sup>20–30</sup> 1D NRs,<sup>31–34</sup> 0D NPs,<sup>35–43</sup> and Ag NP-graphene blends,<sup>44–46</sup> with



active catalysts, wherein these materials act as electron collectors/transporters to enhance the overall hydrogen evolution rate (HER). This review will instead focus on selected case studies that relate to the physicochemical aspects of different photocatalysts. Furthermore, sacrificial reagents such as electron donors or hole scavengers have been employed to enhance the overall activity. It is to be noted that the unit of the HER will be  $\mu\text{mol h}^{-1} \text{g}^{-1}$  throughout the article unless noted otherwise.

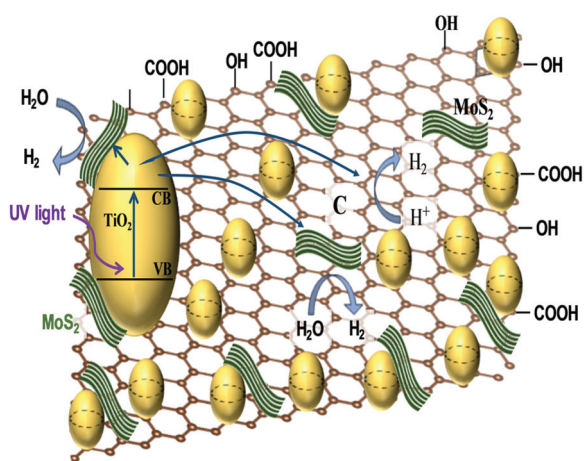
**2.1.1. 2D materials.** Yu and Jaroniec *et al.*<sup>20</sup> demonstrated the photocatalytic activity of  $\text{TiO}_2/\text{MoS}_2/\text{graphene}$  ternary nanocomposites (Fig. 1) synthesized using a two-step hydrothermal process. The layered  $\text{MoS}_2/\text{graphene}$  serves as a novel support for the uniform decoration of anatase  $\text{TiO}_2$  NPs (size  $\sim 7\text{--}10\text{ nm}$ ), as evidenced by high-resolution transmission electron microscopy imaging (HRTEM). The photocatalytic activity was measured in an aqueous solution comprising 25% ethanol hole scavenger under UV irradiation. In a control experiment, authors synthesized different wt% of  $\text{TiO}_2$ ,  $\text{MoS}_2$ , and graphene to boost the  $\text{H}_2$  yield. The highest HER of  $2066\ \mu\text{mol h}^{-1} \text{g}^{-1}$  with an apparent quantum efficiency (AQE) of 9.7%@365 nm was achieved for the 99.5 wt%  $\text{TiO}_2/0.5\text{ wt}\%$   $\text{MoS}_2$ -graphene cocatalyst with 5.0 wt% graphene. A higher amount of cocatalyst ( $>5.0\text{ wt}\%$ ) led to a reduction in the overall activity, which is mainly attributed to the significant increment in the opacity due to the presence of black  $\text{MoS}_2$ -graphene. The observed trend of the HER of  $\text{TiO}_2/\text{MoS}_2/\text{graphene}$  ternary (2066)  $>$   $\text{TiO}_2/\text{MoS}_2$  binary (460)  $>$   $\text{TiO}_2/\text{graphene}$  binary (371)  $>$  pristine  $\text{TiO}_2$  (53) is backed with the highest photocurrent density in the  $\text{TiO}_2/\text{MoS}_2/\text{graphene}$  nanocomposite, suggesting efficient charge transportation due to the formation of nano-junctions. No significant loss in the activity was observed after four recycles, which indicates the ternary photocatalyst's high stability.

In another report, Ye *et al.*<sup>21</sup> have reported the 3D porous hierarchical architecture (Fig. 2a) of CdS NPs (size  $\sim 10\text{ nm}$ ) decorated on  $\text{MoS}_2/\text{reduced graphene oxide (rGO)}$  synthesized

by a solution-chemistry method, as evidenced by scanning electron microscopy (SEM) and HRTEM (Fig. 2b & c). For different contents of the  $\text{MoS}_2/\text{rGO}$  (MG) cocatalyst, 1 : 2 molar ratio of the M : G exhibited the highest activity, which is attributed to the uniform dispersion of  $\text{MoS}_2$  on rGO.<sup>47</sup> Among various wt% of the MG cocatalyst, CdS/2.0 wt% MG annealed at 573 K showed the highest HER of  $\sim 6100$  (AQE: 20.6%) and  $\sim 9000$  (AQE: 28.1%), in  $\text{Na}_2\text{S}\text{--}\text{Na}_2\text{SO}_3$  and lactic acid solution (Fig. 2d–f), respectively, @420 nm. Furthermore, the observed trend of the HER of CdS/MG (9000)  $>$  CdS/M (6400)  $>$  CdS/G (200)  $>$  CdS (150)  $>$  MG (0) suggests highest activity of the ternary nanocomposite. The authors believed that a lower annealing temperature gives low crystallization, whereas too high temperature (673 K) leads to the agglomeration of CdS NPs and non-uniform dispersion of MG on the surface of CdS, thus, not facilitating the charge transfer. The observed remarkable HER in the lactic acid solution is attributed to the abundance of  $\text{H}^+$  ions facilitating higher production. No significant loss in the activity was noted after five recycles. The observed higher activity of ternary nanocomposites is attributed to the nanosized MG with more active edges, which shorten the electron transfer path, as evidenced by the transient photocurrent measurements (Fig. 2g). Thus, the formation of a ternary nanocomposite would enhance the catalytic performance of CdS and make it more stable.

Comparison of the HER reported by Ye *et al.*<sup>21</sup> and Li and Xu *et al.*<sup>23</sup> suggests that CdS NPs decorated on the MG ternary nanocomposite show a remarkable  $\text{H}_2$  evolution rate of  $\sim 9000$  (ref. 21) and  $\sim 19\,000$  (ref. 23) in lactic acid solution @420 nm. Furthermore, Xu *et al.*<sup>23</sup> have reported the highest HER of  $\sim 23\,200$  with an AQE of 65.8% for CdS NR-NPs decorated onto MG. Although both the reports have used the same components, however, different results have been observed. The observed enhanced efficiency by Li and Xu *et al.*<sup>23</sup> may be attributed to the smaller sized nanocomposite with more specific area leading to more active sites and the combination of CdS NP-NR morphology. Further comparison with Xiang *et al.*<sup>27</sup> suggests that the observed lower activity of CdS NRs/ $\text{WS}_2/\text{G}$  ( $\sim 1842$ ) as compared to CdS NRs/ $\text{MoS}_2/\text{G}$  ( $\sim 4000$ )<sup>23</sup> may be attributed to the use of the  $\text{Na}_2\text{S}\text{--}\text{Na}_2\text{SO}_3$  sacrificial agent instead of lactic acid, as both  $\text{WS}_2$  and  $\text{MoS}_2$  have shown similar cocatalytic activity.<sup>48</sup>

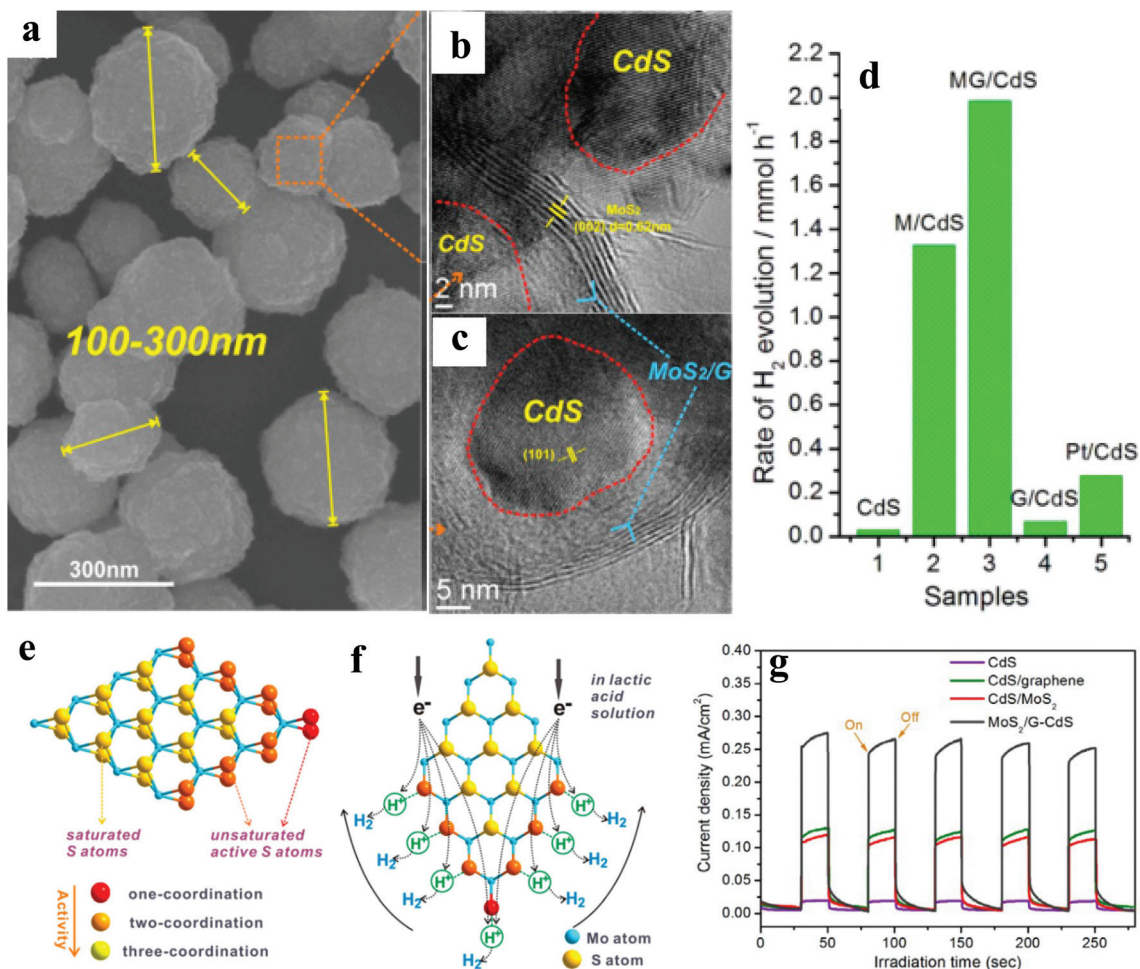
Qiao *et al.*<sup>22</sup> integrated  $\text{Zn}_{0.5}\text{Cd}_{0.5}\text{S}$  (ZCS) NPs within NiS NPs and rGO to form the ternary nanocomposite. The authors synthesized a series of NiS/ZCS/rGO nanocomposites containing different wt% and mol% of rGO and NiS by a co-precipitation-hydrothermal treatment. TEM and HRTEM imaging show NiS NPs ( $\sim 5\text{--}10\text{ nm}$ ) decorated onto ZCS NPs ( $\sim 50\text{ nm}$ ), which were uniformly dispersed on rGO nanosheets. The HER of the as-prepared samples was measured in  $\text{Na}_2\text{S}\text{--}\text{Na}_2\text{SO}_3$  solution. The results suggest that 0.25 wt% of rGO and 3 mol% of NiS show the highest activity of  $7514\ \mu\text{mol h}^{-1} \text{g}^{-1}$  with an AQE of 31.1%@420 nm. The authors proposed that the higher concentration of NiS, leading to the high loading amount of NiS NPs, may shield the incident light, thus, suppressing the photocatalytic activity. The detected HER from



**Fig. 1** Schematic illustration of photocatalytic activity of the  $\text{TiO}_2/\text{MoS}_2/\text{graphene}$  nanocomposite.







**Fig. 2** (a) SEM image of CdS/MG nanocomposites after annealing at 573 K for 2 h under an Ar atmosphere, (b) and (c) HRTEM images of the CdS/MG nanocomposite. (d) Photocatalytic H<sub>2</sub> production activities of CdS, CdS/M, optimized CdS/MG, CdS/G, and Pt/CdS (0.5 wt%) in lactic acid solution, (e) schematic illustration of the microstructure of MoS<sub>2</sub>, and (f) its cocatalytic mechanism of H<sub>2</sub> generation. (g) Transient photocurrent responses of pristine CdS and its binary/ternary nanocomposite under simulated irradiation. Copyright 2014, American Chemical Society. Reprinted from ref. 21 with permission.

the optimized NiS/ZCS/rGO (7514) exceeds that of the bare ZCS (2228), ZCS/rGO binary (4118), and the ZCS/NiS binary (4790) nanocomposite and is in corroboration with the observed highest photocurrent intensity. Ternary nanocomposite's higher activity has been attributed to the formation of a p-n junction between p-type NiS NPs and n-type ZCS NPs, which is crucial for the charge transfer, as evidenced *via* Mott-Schottky plots. The authors have not reported stability studies.

In another report, Min and Xu *et al.*<sup>26</sup> integrated ZCS NRs with MoS<sub>2</sub>/rGO, wherein ZCS NRs (diameter ~ 100 nm) were synthesized by a solvothermal method followed by a dual reduction reaction to produce the ternary nanocomposite. TEM and HRTEM imaging showed the MoS<sub>2</sub> location at interfacial contact formed between the ZCS and rGO. The ternary nanocomposites were synthesized with varying wt% MoS<sub>2</sub> and rGO and found that 1.5 wt% MoS<sub>2</sub> and 3.0 wt% rGO showed the best photocatalytic activity measured in a lactic acid solution. The observed HER from the ternary ZCS NRs@MG

(77 000) is much higher than that of pristine ZCS NRs (933) and is in corroboration with the observed negligible photoluminescence (PL) intensity and higher photocurrent response suggesting hampered charge recombination and efficient charge transfer in ZCS@MG, respectively. The nanocomposite's stability was noted for 25 h, and the decrease in the HER after 25 h was ascribed to the consumption of lactic acid in the solution.

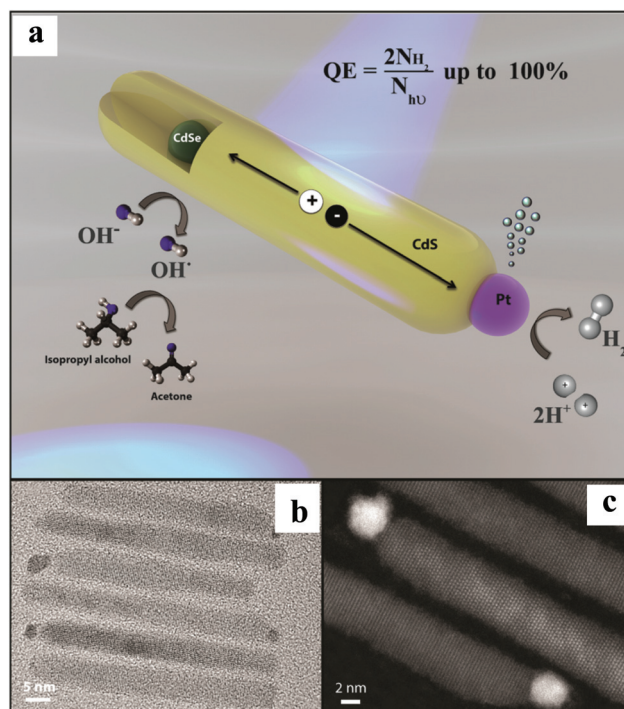
The NiS/ZCS/rGO nanocomposite gave a HER yield of ~7514  $\mu\text{mol h}^{-1} \text{g}^{-1}$  with an AQE of 31.1%,<sup>22</sup> whereas ZCS NRs@MG gave 77 000  $\mu\text{mol h}^{-1} \text{g}^{-1}$ . Although HER yield is higher for ZCS NRs@MG;<sup>26</sup> however, the authors have not mentioned the AQE, making the comparison difficult. The ZCS NRs@MG nanocomposite was found to be stable for 25 h,<sup>26</sup> whereas the stability of the NiS/ZCS/rGO nanocomposite is not reported. Interestingly, in the graphene-based ternary nanocomposite, the highest AQE of 65.8% has been observed by employing CdS NR-NPs decorated onto MG.<sup>23</sup> Furthermore,



whether it is TiO<sub>2</sub>, CdS or ZCS, the photocatalytic mechanism of a semiconductor integrated with rGO and MoS<sub>2</sub> can be summarized as the semiconductor-based catalyst generating electron-hole pairs. The photogenerated electrons in the conduction band (CB) of the semiconductors will be transferred to the edge of MoS<sub>2</sub> due to the formation of intimate contact at the nano-junction followed by a reduction of the adsorbed H<sup>+</sup> to H<sub>2</sub> under the cocatalytic activity of unsaturated active S atoms (Fig. 2e & f). Electrons on the inactive MoS<sub>2</sub> basal planes can be transferred to the edge of MoS<sub>2</sub> through the rGO sheets followed by H<sub>2</sub> evolution. This mechanism contrasts with that reported for NiS/ZCS/rGO,<sup>22</sup> wherein the photogenerated electrons of ZCS are injected into the rGO and the lower graphene/graphene redox potential to reduce H<sup>+</sup> to H<sub>2</sub>. The left-behind photogenerated holes in the valence band (VB) of ZCS will be transferred to NiS due to the inner electric field created in the p-n junction.

**2.1.2. 1D NRs.** Lu and Cheng *et al.*<sup>31</sup> coated metal Cd NRs (diameter ~ 10 nm) with a ZnO-CdS shell (~15 nm) by using the solution chemistry method, wherein the CdS NR shell is directly grown on a Cd core and ZnO NPs are embedded in the CdS shell as islands. The rodlike ZnO-CdS@Cd heterostructure was formed by sulfurization of the Cd-ZnO heterostructures in a Na<sub>2</sub>S aqueous solution with sulfurization time varying from 0.5 to 24 h. For the sulfurization time of 24 h, the inductively coupled plasma showed the composition ratio of  $n_{\text{Cd}} : n_{\text{CdS}} : n_{\text{ZnO}} = 83.81 : 10.55 : 5.64$ . The photocatalytic activity was measured in an aqueous solution of Na<sub>2</sub>S-Na<sub>2</sub>SO<sub>3</sub> using an Xe lamp. The observed HER of ~6000 μmol h<sup>-1</sup> g<sup>-1</sup> is much higher than that of bare CdS (120). Based on the Z-scheme charge-carrier transport, the authors have interpreted that in the ZnO-CdS@Cd ternary nanocomposite, effective and rapid recombination of the photoexcited electrons from the CB of ZnO with the holes from the VB of CdS is a key step to prolong the lifetime of the photoexcited electrons in the CB of CdS and holes in the VB of ZnO. The presence of a metal Cd core acting as an effective channel is certainly beneficial for such recombination. The photocatalyst was found to be stable for four recycles. The authors have not mentioned the AQE.

In another report, Amirav *et al.*<sup>32</sup> have reported the Pt tipped CdSe@CdS ternary nanocomposite with a remarkable AQE of 100%. CdSe NPs were prepared by the solution chemistry method followed by the growth of CdS NRs, including the injection of CdSe NPs into these NRs. The photo-deposition of Pt NPs was carried out to realize the final nanocomposite. TEM and high angle annular dark field micrographs show CdSe NPs embedded asymmetrically within a CdS NR (diameter ~ 2.3 nm) and a single Pt NP placed at the end of the NR (Fig. 3). The authors have examined the photocatalytic activity of Pt tipped CdSe@CdS with a hydroxyl anion/radical redox couple operating as a shuttle to relay the holes. A typical sample containing ~6 × 10<sup>14</sup>, Pt tipped 50 nm long CdS NRs embedded with CdSe NPs, isopropyl alcohol (electron donor), and KOH gave rise to a HER of 142 000 μmol h<sup>-1</sup> g<sup>-1</sup> corresponding to an AQE of 100% at pH of 15@455 nm excitation. Furthermore, the AQE > 90% was observed for numerous



**Fig. 3** (a) Illustration of the NP photocatalyst depicting the photocatalytic cycle. (b) TEM and (c) high-resolution HAADF micrographs of a few representative Pt tipped CdSe@CdS rods. Copyright 2016, American Chemical Society. Reprinted from ref. 32 with permission.

samples with varying CdSe NP size (2.3–2.7 nm) and CdS NR length (45–65 nm). The synthesized nanocomposite showed astounding stability of 44 h. The observation of an exponential decrease in the activity with the decrease in the pH (15 to 8) is supported by the hypothesis that high pH is required to shift the VB of CdS for the oxidation of hydroxyl ions,<sup>49</sup> which enable a crossover between the VB maximum and the redox potential for hydroxyl anions and radicals. As a result, the hydroxyl anion-radical redox couple can be employed as a shuttle to efficiently relay the hole from CdSe NPs to the scavenger. The authors have proposed that this architecture facilitates localization of holes, which are three-dimensionally confined to the CdSe. At the same time, delocalized electrons are transferred to the Pt tip *via* CdS NRs for the reduction of H<sup>+</sup> to H<sub>2</sub>. This design enabled efficient long-lasting charge carriers' separation, thus, extending their availability for redox reactions. It is noteworthy that the system had a single Pt reduction catalyst. The authors interpreted that the observed decreased AQE of 58.5% for two Pt NPs can be correlated with the system's ability to mediate hole transfer on the CdS NR surface, a process that might be highly sensitive to the ligand coverage. Based on a previous report,<sup>50</sup> the authors have proposed that for a multi-electron reaction system, the final photocatalyst should only include a single cocatalytic site per each segment of the semiconductor capable of light excitation.

In a recent report, Burda *et al.*<sup>34</sup> have reported the HER in the MoS<sub>2</sub>-stratified CdS-Cu<sub>2-x</sub>S ternary nanocomposite (Fig. 4)





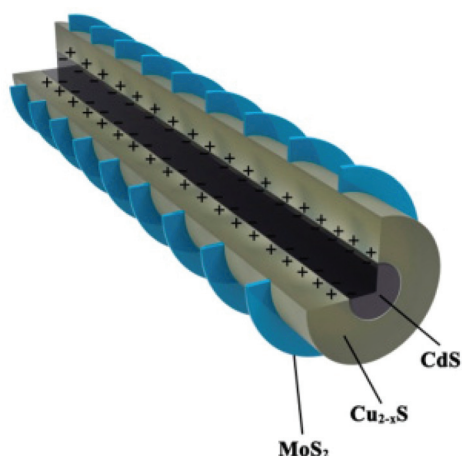


Fig. 4 Schematic illustration of the crystal structure of heterointerfaces. Copyright 2020, American Chemical Society. Reprinted from ref. 34 with permission.

with an optimal 3.42 wt% of  $\text{MoS}_2$ . The fascinating epitaxial vertical growth of 2D  $\text{MoS}_2$  on 1D  $\text{CdS-Cu}_{2-x}\text{S}$  NRs (diameter  $\sim 51$  nm, length  $\sim 987$  nm) was carried out using sulfur powder dissolved in the oleyl amine precursor. The photocatalytic activity of the stratified screw-shaped  $\text{MoS}_2@1\text{D CdS-Cu}_{2-x}\text{S}$  ternary nanocomposite was measured in an aqueous solution of  $\text{Na}_2\text{S-Na}_2\text{SO}_3$ . The detected highest HER with a  $\text{MoS}_2$ -stratified  $\text{CdS-Cu}_{2-x}\text{S}$  ternary system (14 184) as compared to bare  $\text{CdS}$  (146), and the  $\text{CdS-Cu}_{2-x}\text{S}$  binary system (1121) is in corroboration with the observed higher transient photocurrent response and weaker PL emission. However, the corresponding AQE is  $\sim 1\%$  at 400 nm. The authors have proposed that the photogenerated electrons of  $\text{Cu}_{2-x}\text{S}$  can transfer to the CB of  $\text{CdS}$  and  $\text{MoS}_2$ ; however, they mainly accumulate in the  $\text{MoS}_2$  due to the stronger driving force between  $\text{Cu}_{2-x}\text{S}$  and  $\text{MoS}_2$ . The electrons could migrate to the surface of  $\text{MoS}_2$  for the reduction of  $\text{H}^+$  ions. Furthermore, the mobile  $\text{Cu}^{\text{I}}$  ions of  $\text{Cu}_{2-x}\text{S}$  can intercalate within the  $\text{MoS}_2$  basal plane due to nano-junction formation, thus making a  $\text{Cu}^{\text{I}}@ \text{MoS}_2$  basal plane catalytically active. The stratified ternary nanocomposite was found to be stable during the cycling runs.

It is interesting to note that the Pt tipped  $\text{CdSe@CdS}$  ternary nanocomposite with a remarkable AQE of 100% has been reported.<sup>32</sup> However, this ternary nanocomposite involves Pt, an expensive component, making it less plausible for the large-scale applications. Furthermore,  $\text{CdS}$  is used as an active photocatalyst in the two potential reports,<sup>31,32</sup> whereas the role of  $\text{CdS}$  in the overall photocatalytic activity is not very clear in the  $\text{MoS}_2$ -stratified  $\text{CdS-Cu}_{2-x}\text{S}$  ternary nanocomposite.<sup>34</sup> For both graphene and 1D nanorod-based ternary nanocomposites,  $\text{CdS}$ -based nanocomposites have been shown to have the highest AQE.

**2.1.3. 0D NPs.** Xue *et al.*<sup>35</sup> have reported visible light photocatalytic activity of the  $\text{Au@TiO}_2\text{-CdS}$  ternary nanocomposite grown using the hydrothermal method (Fig. 5). The final sample is denoted as  $\text{Au@TiO}_2\text{-xCdS}$ , wherein  $x$  varying

from 0.1 to 0.4 refers to the molar ratio of  $\text{CdS}$  to  $\text{TiO}_2$ . SEM and TEM measurements show that  $\text{CdS}$  NPs of size  $\sim 10\text{-}20$  nm are in intimate contact on the outer surfaces of  $\text{TiO}_2$  NPs, which form a shell to encapsulate Au NPs (diameter  $\sim 40$  nm). In a typical run,  $\text{Au@TiO}_2\text{-xCdS}$  photocatalysts were suspended in an aqueous solution of  $\text{Na}_2\text{S-Na}_2\text{SO}_3$ . The optimal  $\text{Au@TiO}_2\text{-0.2CdS}$  ternary nanocomposite exhibited the highest HER (1970) as compared to bare  $\text{CdS}$  NPs (125),  $\text{TiO}_2\text{-0.2CdS}$  binary (140), and  $\text{Au@TiO}_2$  binary (0.5). The observed remarkable enhancement in the photocatalytic activity has been attributed to the ternary nanocomposite's unique design for an efficient electron transfer from  $\text{CdS} \rightarrow \text{TiO}_2 \rightarrow \text{Au}$ . Upon light excitation, the photogenerated electrons in the CB of  $\text{CdS}$  transport to the Au core *via*  $\text{TiO}_2$  to induce the proton reduction for  $\text{H}_2$  evolution, whereas remaining photogenerated holes on  $\text{CdS}$  could be quenched by the sulfide ions in the solution.

In another report, Liu *et al.*<sup>36</sup> prepared a  $\text{CdS/Au/ZnO}$  ternary nanocomposite, wherein a hydrothermal process was used to grow wurtzite  $\text{ZnO}$  ( $\sim 5$   $\mu\text{m}$ ) followed by photo-deposition of Au NPs (10–20 nm) and subsequent deposition of hexagonal  $\text{CdS}$  NPs ( $\sim 50$  nm) *via* chemical bath deposition. Interestingly, SEM imaging showed the final morphology of  $\text{ZnO}$  as nanosheets, which is attributed to the agglomeration of flower-like shaped crystals. The SAED pattern showed the single-crystalline nature of the nanosheet. The photocatalytic activity was measured in an aqueous solution of  $\text{Na}_2\text{S-Na}_2\text{SO}_3$ . Compared with the HER of  $\text{CdS/ZnO}$  (134),  $\text{CdS/Au/ZnO}$  (608) showed remarkably improved photocatalytic activity under UV irradiation. The authors performed a comparative test in order to prove that the Au NPs strengthen the interfacial charge-carrier transfer between  $\text{CdS}$  (holes) and  $\text{ZnO}$  (electrons) *via* the Z-scheme by studying the photocatalytic activity of  $\text{Au/CdS/ZnO}$  (HER: 197). The photogenerated electrons and holes remaining in the CB  $\text{CdS}$  and VB  $\text{ZnO}$ , respectively, can be involved effectively in photocatalytic reactions for hydrogen evolution.

Zhang *et al.*<sup>39</sup> have synthesized the  $\text{g-C}_3\text{N}_4/\text{Pt/TiO}_2$  ternary nanocomposite (Fig. 6) through a facile two-step synthetic methodology, involving photo-deposition of Pt NPs on the surface of anatase  $\text{TiO}_2$  NRs (diameter  $\sim 30\text{-}80$  nm and lengths of 2–20  $\mu\text{m}$ ) and subsequent growth of  $\text{g-C}_3\text{N}_4$  (CN) layers, as evidenced by electron microscopy imaging. The  $\text{TiO}_2$  NRs loaded with Pt NPs are uniformly wrapped with the porous fluffy CN layers. The photocatalytic performance was evaluated in an aqueous solution of 17 vol% triethanolamine sacrificial reagent. Under visible light excitation,  $\text{TiO}_2$  and  $\text{Pt/TiO}_2$  did not show any  $\text{H}_2$  as expected. The observed higher HER of  $\sim 893$  (CN/Pt/ $\text{TiO}_2$ ) compared to 715 (CN/ $\text{TiO}_2$ ) is further supported with the observed higher photocurrent response and longer fluorescence lifetime in the CN/Pt/ $\text{TiO}_2$  ternary nanocomposite (2.60 ns) compared to CN/ $\text{TiO}_2$  binary (2.50 ns), Pt/ $\text{TiO}_2$  binary (0.38 ns), and bare  $\text{TiO}_2$  (0.36 ns). Furthermore, the electrochemical impedance spectra (EIS) curve radius of all electrodes showed the smallest radius for CN/Pt/ $\text{TiO}_2$ , further suggesting the more efficient separation of electron-hole pairs and faster interfacial



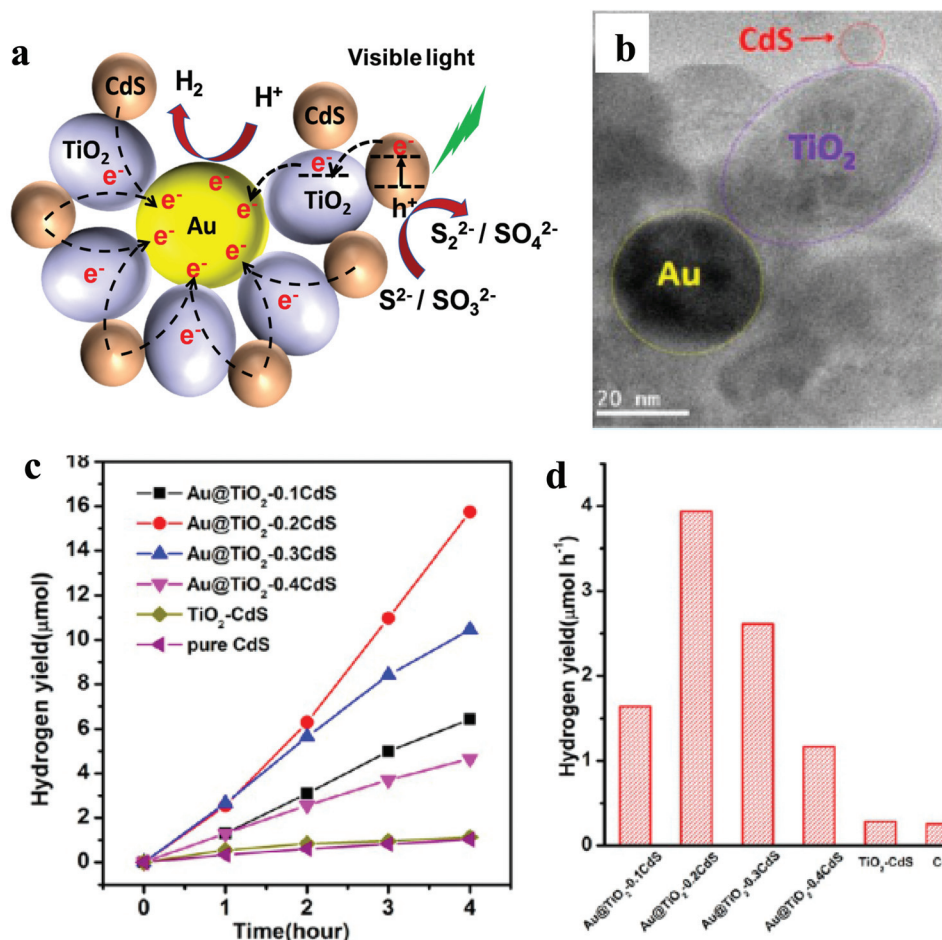


Fig. 5 (a) Schematic illustration of photocatalytic activity of the Au@TiO<sub>2</sub>-CdS nanocomposite, (b) TEM, (c) plots of the photocatalytic HER versus visible light irradiation time and (d) HER for pristine CdS, binary TiO<sub>2</sub>-CdS and various ternary nanocomposites. Fig. 4b–d: Copyright 2013, American Chemical Society. Reprinted from ref. 35 with permission.

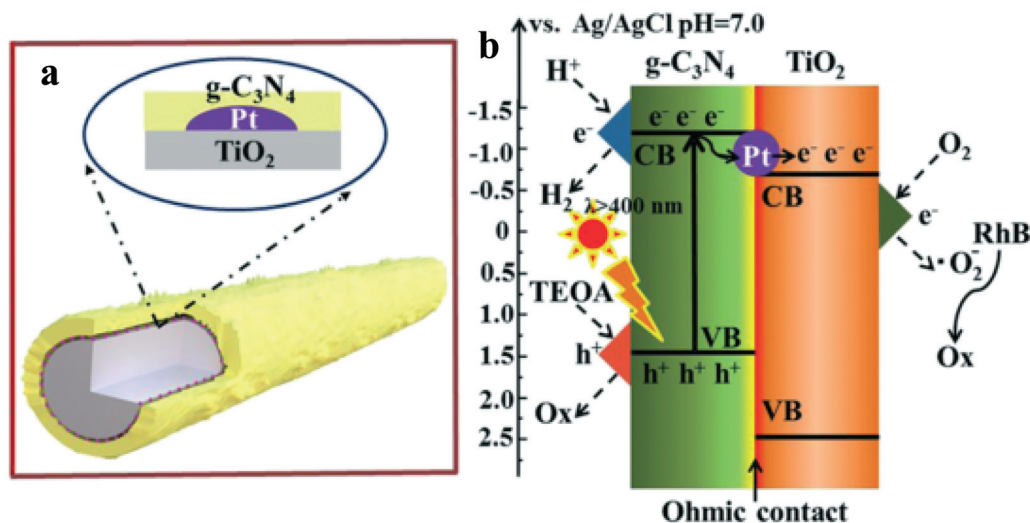


Fig. 6 Schematic structure (a) and the photogenerated electron–hole transfer pathways (b) inside the CN/Pt/TiO<sub>2</sub> composite under visible light irradiation. Copyright 2019, Royal Society of Chemistry. Reprinted from ref. 39 with permission.



charge transfer. It is further shown that CN/Pt/TiO<sub>2</sub> can simultaneously produce H<sub>2</sub> by water splitting and degrade rhodamine B (RhB) in the same reaction system. The authors have described that under visible light irradiation, some of the photogenerated electrons in CN tend to transfer to the CB of TiO<sub>2</sub> via Pt NPs, whereas some of them will reduce H<sup>+</sup> at the CN CB. At the same time, triethanolamine will consume h<sup>+</sup> in the CN VB, while O<sub>2</sub> is reduced to O<sub>2</sub><sup>-</sup> at the TiO<sub>2</sub> CB, which will further degrade RhB.

It is important to note here that none of the reports mentioned AQE, making it difficult to compare these three systems. However, based on HER yield, Au@TiO<sub>2</sub>-CdS has shown the highest activity (1970 μmol h<sup>-1</sup> g<sup>-1</sup>). Besides, these reports again suggest that the ternary nanocomposite involving CdS shows the highest activity, which is similar to that observed for graphene and 1D-NR-based ternary nanocomposites.

**2.1.4. Ag NP-graphene blend.** Cho *et al.*<sup>46</sup> have reported Ag-TiO<sub>2</sub>-graphene ternary nanocomposites, wherein GO was prepared by the Hummers' method followed by ultrasonication with P25 and AgNO<sub>3</sub> to synthesize the final compound of A<sub>y</sub>TG (x). Herein, y represents the concentration of AgNO<sub>3</sub> solution (1, 5, 10, 15, and 20 mM), and x is related to the weight ratio of TiO<sub>2</sub>:GO (TG), which is 5:1 and 10:1, denoted as 5 and 10, respectively. SEM and TEM imaging suggest smaller Ag NPs decorated on spherical and oval TiO<sub>2</sub> NPs (diameter ~ several to tens of nm), which are randomly distributed over GO's surface. Among various concentrations of AgNO<sub>3</sub>, the highest absorption was obtained for A<sub>15</sub>TG(5) and A<sub>10</sub>TG(10) nanocomposites with the smallest bandgap of ~1.63 eV and 1.82 eV, respectively, as evidenced by UV-vis spectroscopy, thus indicating a required optimum concentration of Ag for the nanocomposite. The HER was measured in 20% methanol aqueous solution. The observed HER of 705.6 and 427.2 μmol g<sup>-1</sup> h<sup>-1</sup> with an AQE of 26.2% and 15.8% for A<sub>15</sub>TG(5) and A<sub>10</sub>TG(10), respectively, suggests that A<sub>15</sub>TG(5) is the optimal nanocomposite. The authors have not discussed the probable mechanism for H<sub>2</sub> production, and they have not mentioned the photocatalyst's stability. It is to be noted that the authors have discussed various trapping experiments to explain the degradation mechanism of pollutants in the water, which in turn increases the H<sub>2</sub> production activity.

In short, CdS combined with cocatalysts has shown the best HER yield and high stability. Observed higher yield could be due to the intrinsic high visible light absorption of pristine CdS (bandgap ~ 2.4 eV) and synergistic effect of the ternary nanocomposite. Simultaneously, the photocorrosion of the visible light active pristine CdS catalyst (~2.4 eV) is considered the Achilles heel of pristine CdS, whereas the stability has been found to be increased in ternary nanocomposites. This suggests that the formed ternary nanocomposites have enhanced photocatalytic efficiency and increased overall stability.

## 2.2. Degradation of pollutants (3500)

A substantial amount of work has been published on integrating 2D materials,<sup>51-67</sup> 0D NPs,<sup>68-83</sup> and Ag NP-graphene

blends<sup>84-89</sup> with an active catalyst for pollutant removal. This review will focus on selected case studies that relate to the degradation of various pollutants such as alcohol, methylene blue (MB), methyl orange (MO), Congo red (CR), and RhB.

**2.2.1. 2D materials.** Xu *et al.*<sup>51</sup> have studied a TiO<sub>2</sub>/CdS/G ternary nanocomposite, wherein GO was prepared by Hummers' methods followed by the solvothermal growth of CdS on GO and hydrothermal growth of TiO<sub>2</sub> NPs. Based on the authors' previous work on the CdS/G binary nanocomposite,<sup>90</sup> 5 wt% of G was chosen, which gave the best output for the oxidation of benzyl alcohol to benzaldehyde. Furthermore, among various wt% of TiO<sub>2</sub>, 10 wt% was chosen for this study. TEM and corresponding elemental mapping displayed that CdS and TiO<sub>2</sub> NPs cover the GO nanosheets uniformly. Given its favorable structure, the visible light photocatalytic activity was evaluated in an oxygen saturated benzotri-fluoride solution of 0.1 mmol alcohol and 8 mg of catalyst. The % oxidation of benzyl alcohol (~100), *p*-methyl benzyl alcohol (~85), *p*-methoxyl benzyl alcohol (~90), *p*-nitro benzyl alcohol (~60), *p*-fluoro benzyl alcohol (~95), *p*-chloro benzyl alcohol (~85), cinnamyl alcohol (~75), 3-methyl-2-buten-1-ol (~60), and 2-buten-1-ol (~60) was observed after 5 h, which is ~2 and 2.3 times higher than that of CdS-5% G, and CdS-10% TiO<sub>2</sub>, respectively. Observed higher activity in the ternary system is corroborated with the observed significant enhancement in the transient photocurrent response, negligible PL emission, and smaller arc in Nyquist impedance, suggesting an efficient suppression of charge recombination and transfer of charge carriers. The authors have proposed that under visible-light irradiation, the photogenerated electrons in CdS transfer to GO and TiO<sub>2</sub> due to their intimate interfacial contact and react with the adsorbed O<sub>2</sub> to give superoxide radicals. As displayed in electron spin resonance (ESR) spectra, the superoxide radicals are detectable and stable in the reaction system. Simultaneously, the hole perching in the VB of CdS can oxidize the alcohols to form alcohol radicals; then, the alcohol radicals are oxidized by oxygen or superoxide radicals to give the targeted product. The authors have additionally reported that the ternary hybrid possesses stable durability of photoactivity for three recycles.

Chen *et al.*<sup>53</sup> have studied 2D porous g-C<sub>3</sub>N<sub>4</sub>/nitrogen-doped GO/MoS<sub>2</sub> (CN/NrGO/MoS<sub>2</sub>) synthesized by the hydrothermal method. Detailed HRTEM (Fig. 7a & b) imaging shows a distinguished and coherent interface among the CNNS, NrGO, and MoS<sub>2</sub>, indicating the formation of nano-junctions, resulting in better separation of photoinduced charge carriers and more efficient electron transfer within the nanocomposite structure. The photocatalytic oxidation of MB and the reduction of Cr(VI) were simultaneously carried out at a pH value of 3, containing 30 mg of catalyst mixed with aqueous 10 mg L<sup>-1</sup> MB solution and 10 mg L<sup>-1</sup> K<sub>2</sub>Cr<sub>2</sub>O<sub>7</sub>. The kinetic rate constant (*k*, min<sup>-1</sup>) for the oxidation of MB (0.0338) and the reduction of Cr(VI: 0.0157) is, respectively, enhanced by 141 and 265% when the binary photocatalyst of CN/N-doped partially rGO was replaced with the CN/NrGO/MoS<sub>2</sub> (Fig. 7c & d). The authors have attributed the enhanced photocatalytic





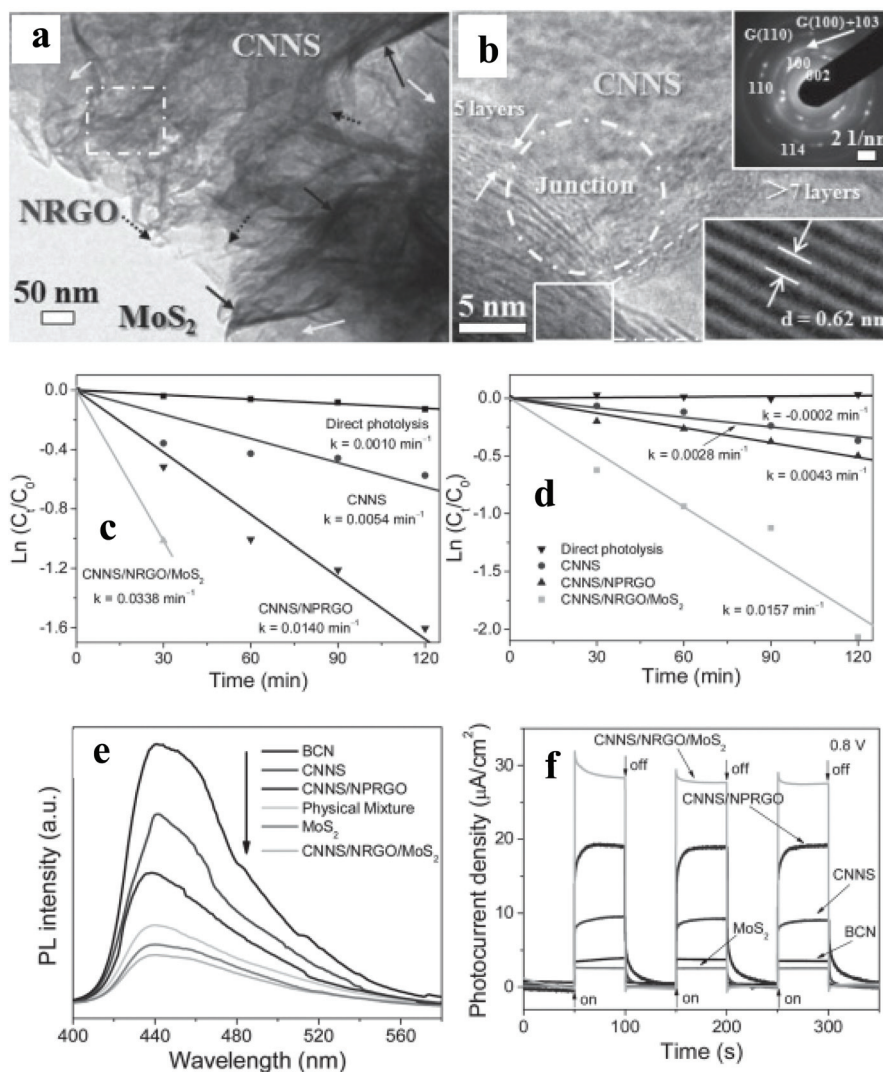


Fig. 7 (a and b) TEM images, SAED pattern, and HRTEM images of CNNS/NRGO/MoS<sub>2</sub> nanojunctions, (c and d)  $\ln(C_t/C_0)$  of MB and Cr(VI) versus time for different photoactive materials under simulated sunlight irradiation, (e) PL spectra and (f) transient photocurrent density versus Na<sub>2</sub>SO<sub>4</sub> electrolyte. Copyright 2013, Wiley Online Library. Reprinted from ref. 53 with permission.

activity to the formation of nano-junctions and efficient light absorption. The observed significant enhancement is further supported with drastic quenching of PL emission (Fig. 7e) in the ternary nanocomposite, implying an enhanced separation of the photogenerated carrier. The photocurrent density of CNNS/NrGO/MoS<sub>2</sub> reaches 27.76  $\mu\text{A cm}^{-2}$ , which is 8.14, 3.06, 1.46, and 10.93-fold higher than those of bulk CN (3.41  $\mu\text{A cm}^{-2}$ ), CNNS, CNNS/N-doped partially rGO, and MoS<sub>2</sub>, respectively (Fig. 7f). Moreover, no significant photocurrent changes were observed within 1400 s of illumination, indicating that the nano-junctions are stable. The authors have proposed that photogenerated electrons in the CN CB can be transferred easily to MoS<sub>2</sub> sheets *via* NrGO, where photoreduction reactions occur. Simultaneously, NrGO interlayers also provide pathways for photogenerated holes at the VB of MoS<sub>2</sub> to reach the VB of CN for photooxidation reactions.

In another report, Lu and Liang *et al.*<sup>54</sup> have studied MB degradation by using a CN/BiOI/GO composite with 10 mg L<sup>-1</sup> MB solution and 10 mg catalyst@300 W Xe lamp. For the preparation of g-C<sub>3</sub>N<sub>4</sub>/BiOI/GO, hydrothermally prepared g-C<sub>3</sub>N<sub>4</sub>/BiOI and GO were ultrasonicated, followed by drying at 60 °C for 6 h. TEM and HRTEM imaging demonstrated the deposition of g-C<sub>3</sub>N<sub>4</sub>/BiOI on the surface of GO. The kinetic rate constant for the photodegradation of MB was found to be  $\sim 0.2334 \text{ min}^{-1}$  for g-C<sub>3</sub>N<sub>4</sub>/BiOI/GO, which is  $\sim 6$  times higher than that reported by Chen *et al.*<sup>53</sup> ( $0.0338 \text{ min}^{-1}$  @150 W Xe lamp). The authors have interpreted that the key issue for photocatalyst performance enhancement is electron-hole separation rather than adsorption capacity. The studied photocatalyst was found to be stable for five recycles.

In another report by Chen *et al.*,<sup>59</sup> the Bi<sub>2</sub>WO<sub>6</sub>/MoS<sub>2</sub>/rGO (BMG) ternary nanocomposite synthesized *via* a two-step



hydrothermal process has been reported for the photoreduction of Cr(vi). TEM and HRTEM imaging showed anchoring of MoS<sub>2</sub> nanosheets (~100 nm) and orthorhombic B NPs (~2–15 nm) on rGO uniformly. The photocatalytic activity was determined at  $\lambda = 540$  nm by decomposing K<sub>2</sub>Cr<sub>2</sub>O<sub>7</sub> (10 mg L<sup>-1</sup>) aqueous solution containing 30 mg of the photocatalyst in 0.02 mL lactic acid solution. The BMG ternary nanocomposite with 7.5 wt% MG showed the highest activity with  $k \sim 0.05513$  min<sup>-1</sup>, which is in corroboration with the observed higher photocurrent response and negligible PL emission in the same. The photocatalyst was found to be stable for four recycles. This observed rate constant is much higher than that observed by Chen *et al.*<sup>53</sup> (0.0157 min<sup>-1</sup>), which could be due to the use of lactic acid, monochromatic excitation source, and different active catalysts leading to a difference in the interaction at the nano-junctions.

Tian *et al.*<sup>62</sup> have reported an effective strategy to synthesize a noble metal-free Bi<sub>2</sub>WO<sub>6</sub>@MoS<sub>2</sub>/rGO ternary nanocomposite using a two-step hydrothermal method with a visible light harvester to degrade RhB. The as-prepared Bi<sub>2</sub>WO<sub>6</sub> (BWO) showed a clew-like microsphere structure with an average diameter of ~4  $\mu$ m. Among various ternary nanocomposites prepared, BWO/MG with 1.0% MG, containing 5% G showed the best performance. The catalyst's photocatalytic activity was measured in an aqueous solution of RhB (10 mg L<sup>-1</sup>) and 20 mg of photocatalyst@300 W Xe arc lamp. The observed kinetic rate constant of ~0.01157 min<sup>-1</sup> for Bi<sub>2</sub>WO<sub>6</sub>@MoS<sub>2</sub>/rGO is much higher than ~0.00118 min<sup>-1</sup> (BWO), suggesting that the ternary heterojunction showed the highest photodegradation rate. The authors have proposed that the photo-excited electrons on the CB of MoS<sub>2</sub> can transfer to the CB of BWO *via* rGO, which can react with O<sub>2</sub> to produce O<sub>2</sub><sup>•-</sup>. These oxidative species play a vital role in the oxidative degradation of organics.

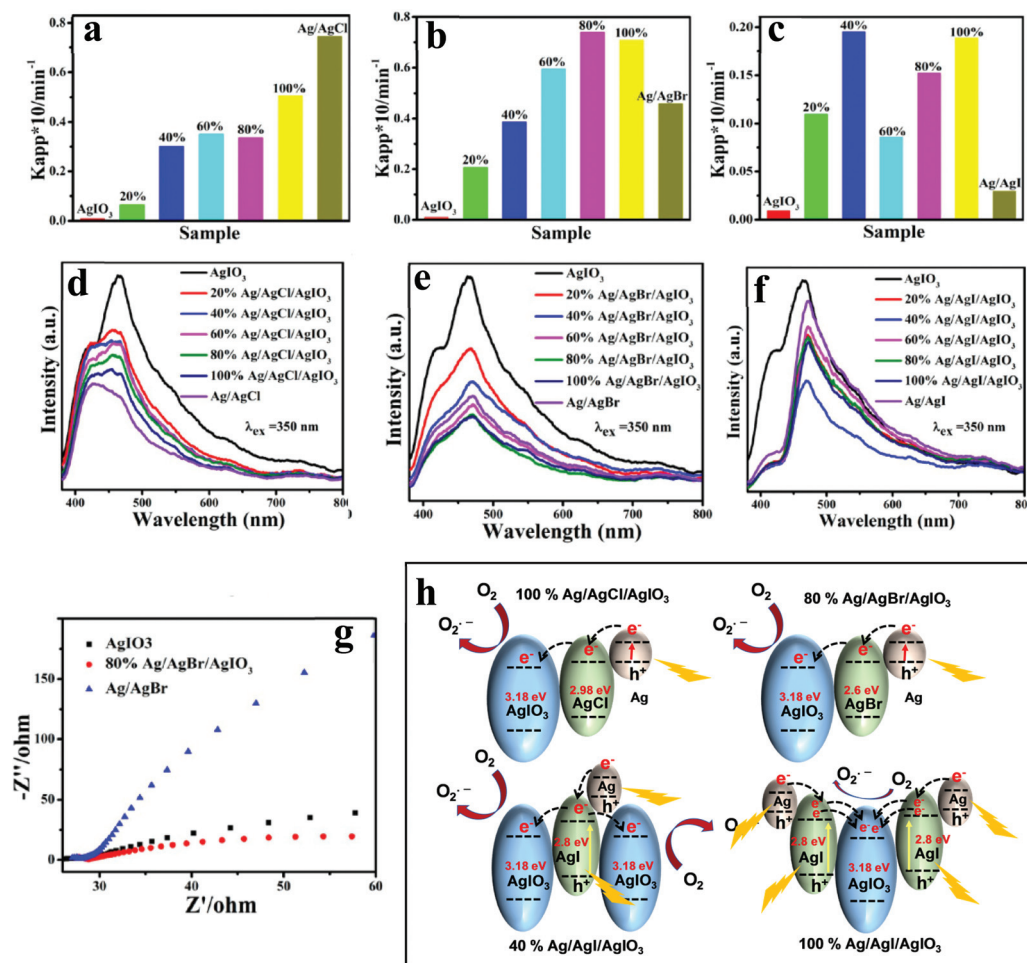
Lee *et al.*<sup>67</sup> have reported that CN hybridized with CdS NPs and rGO using a facile chemical method for the photodegradation of RhB and CR dyes under visible light and UV irradiation. HRTEM imaging suggests decorating hexagonal CdS NPs (~40 nm) over the entire CN sheets, which are further dispersed onto rGO. The photocatalytic activity toward photodegradation of a 4 mg L<sup>-1</sup> solution of RhB and a 20 mg L<sup>-1</sup> solution of CR was measured with 100 mg photocatalyst. For RhB, the rate constant was found in the order of 0.02192 (CN/CdS/rGO) > 0.01049 (CN/rGO) > 0.00931 (rGO/CdS) > 0.00643 (CN/CdS) > 0.00177 (CN), whereas, for CR, the rate constant was found in the order of 0.02437 (CN/CdS/rGO) > 0.01480 (CN/rGO) > 0.00556 (rGO/CdS) > 0.00406 (CN/CdS) > 0.00191 (CN). The observed higher activity of CN/CdS/rGO is further corroborated with the decreased PL emission and the larger surface area of the CN/CdS/rGO (70.42 m<sup>2</sup> g<sup>-1</sup>), resulting in rapid adsorption of dye onto the surface of the photocatalyst leading to effective degradation of organic pollutants. Post photodegradation, the total organic carbon concentration was measured to confirm the dye molecules' complete degradation. Due to the large electron mobility of the rGO, the authors suggested that captured photoelectrons effectively transported from

g-C<sub>3</sub>N<sub>4</sub> to CdS and then to rGO. The captured electrons were utilized for the formation of O<sub>2</sub><sup>•-</sup> and OH<sup>•</sup> radicals, which were further used for degradation.

**2.2.2. 0D NPs.** Hu and Huang *et al.*<sup>75</sup> fabricated three series of ternary hierarchical architecture of Ag/AgX (X = Cl, Br, I)/AgIO<sub>3</sub> by a facile *in situ* ion-exchange route with molar ratios of KX/AgIO<sub>3</sub> in the range of 20–100%. The AgIO<sub>3</sub> single-crystal possesses a spindly shaped morphology and a smooth surface with a diameter of ~5  $\mu$ m. The authors have shown that the amount of Ag NPs and AgX assembled on the surface of AgIO<sub>3</sub> gradually increases with the increment in the molar ratio of XI<sup>-</sup>/IO<sub>3</sub><sup>-</sup>. The photocatalytic activity of Ag/AgX/AgIO<sub>3</sub> was tested by the photodegradation of MO (2  $\times$  10<sup>-5</sup> mol L<sup>-1</sup>) with an Xe lamp using 20 mg of the photocatalyst. For X = Cl, the Ag/AgCl binary nanocomposite showed the highest decomposition rate (0.074 min<sup>-1</sup>) and degraded 97% of MO within 40 min (Fig. 8a). For X = Br, the 80% Ag/AgBr/AgIO<sub>3</sub> ternary nanocomposite showed the highest photocatalytic activity, which could degrade 98% of MO with an apparent rate constant of 0.074 min<sup>-1</sup> (Fig. 8b). For X = I, the rate constant reaches the maximum of 0.019 min<sup>-1</sup> at 40% and 100% Ag/AgI/AgIO<sub>3</sub> (Fig. 8c). The observed results are in corroboration with negligible PL emission and a smaller arc in EIS Nyquist plots (Fig. 8d–g). A probable mechanism has been proposed to explain these intriguing observations (Fig. 8h). For Cl, the authors have interpreted that AgCl has a more negative CB level, which can endow electrons with stronger reducing capacity. Thus, the Ag/AgCl binary photocatalyst exhibits better photocatalytic activity compared to the ternary nanocomposite. For Br, Ag NP and AgBr can produce electrons and holes under visible light illumination, transferring to the CB of AgIO<sub>3</sub>. The relative CB position of AgIO<sub>3</sub> decreases the oxidizing ability of O<sub>2</sub><sup>•-</sup>. Consequently, the holes play a more important role in the photodegradation of MO. For I, 40% Ag/AgI/AgIO<sub>3</sub> has more AgIO<sub>3</sub> in components, which permits more O<sub>2</sub><sup>•-</sup> to participate in the photocatalytic process, whereas for 100% Ag/AgI/AgIO<sub>3</sub>, the contents of Ag and AgI covered on the AgIO<sub>3</sub> increase. This would allow more h<sup>+</sup> to take part in the degradation of the pollutant directly. This interpretation is further supported with active species trapping experiments carried out using benzoquinone (BQ), disodium ethylenediaminetetraacetate (EDTA-2Na), and isopropanol (IPA) to quench O<sub>2</sub><sup>•-</sup>, h<sup>+</sup>, and OH<sup>•</sup> radicals, respectively. For 80% Ag/AgBr/AgIO<sub>3</sub>, it was found that the addition of 1 mM IPA had only slight effect on the photocatalytic degradation of MO, demonstrating that OH<sup>•</sup> has little influence in the photocatalytic process. When BQ and EDTA-2Na were used, MO degradation was largely suppressed, suggesting that O<sub>2</sub><sup>•-</sup> and h<sup>+</sup> should be the main reactive species for 80% Ag/AgBr/AgIO<sub>3</sub>. Regarding 40% Ag/AgI/AgIO<sub>3</sub>, the addition of EDTA-2Na and BQ greatly affected MO degradation. However, O<sub>2</sub><sup>•-</sup> plays a more dominant role than h<sup>+</sup>, which is in contrast to 100% Ag/AgI/AgIO<sub>3</sub>, wherein, more h<sup>+</sup> participates in the MO photodegradation. The studied photocatalyst was found to be stable for five recycles.

Niu and Zeng *et al.*<sup>76</sup> have studied the Ag/AgCl/SrTiO<sub>3</sub> ternary nanocomposite grown using the precipitation reaction





**Fig. 8** Apparent rate constants for the photodegradation of MO over AgI/O<sub>3</sub>, Ag/AgCl, and Ag/AgCl/AgI/O<sub>3</sub> composites (a); AgI/O<sub>3</sub>, Ag/AgBr, and Ag/AgBr/AgI/O<sub>3</sub> (b); AgI/O<sub>3</sub>, Ag/AgI, and Ag/AgI/AgI/O<sub>3</sub> (c) under the irradiation of visible light, corresponding (d–f) PL spectra under an excitation of 350 nm, and (g) EIS Nyquist plot of AgI/O<sub>3</sub>, 80% Ag/AgBr/AgI/O<sub>3</sub>, and Ag/AgBr. (h) Schematic illustration of different charge-transfer mechanisms of 100% Ag/AgCl/AgI/O<sub>3</sub>, 80% Ag/AgBr/AgI/O<sub>3</sub>, 40% Ag/AgI/AgI/O<sub>3</sub>, and 100% Ag/AgI/AgI/O<sub>3</sub>. Fig. 6a–g: Copyright 2016, American Chemical Society. Reprinted from ref. 75 with permission.

between AgNO<sub>3</sub> and NaCl at ambient temperature, wherein Ag/AgCl NPs were immobilized on the surface of SrTiO<sub>3</sub> nanocubes. RhB, MO, MB, phenol, and bisphenol A (BPA) mixed with 50 mg of photocatalyst were utilized as target contaminants (10 mg L<sup>-1</sup>) to assess the photocatalytic activities under visible light excitation. The optimized ternary nanocomposite with 21.6% mass ratio of Ag to SrTiO<sub>3</sub> could degrade ~96% RhB (30 min), 93% MO (40 min) and 96% MB (70 min). The degradation efficiency for phenol and BPA could reach 70% and 83% within 4 h, respectively. Based on these results, the authors suggested that the optimal Ag/AgCl/SrTiO<sub>3</sub> was conducive to preventing the charge carrier recombination, as evidenced by PL emission quenching. The authors have proposed that the photogenerated electrons in Ag NPs could transfer to the CB of SrTiO<sub>3</sub> and AgCl. Since the CB potentials of SrTiO<sub>3</sub> (−0.86 eV) and AgCl (−0.06 eV) are much more negative than the potential of O<sub>2</sub>/O<sub>2</sub><sup>•−</sup> (0.046 eV), the electrons in the CB would reduce O<sub>2</sub> to O<sub>2</sub><sup>•−</sup>, which would further oxidize the

organic contaminants. Simultaneously, the photogenerated holes can migrate to the AgCl surface to oxidize Cl<sup>−</sup> to Cl<sup>•</sup>, which can oxidize organic contaminants and get reduced to Cl<sup>−</sup> again. No significant changes in the photocatalytic degradation of RhB repeated five times suggest good stability.

Choi *et al.*<sup>81</sup> have studied noble metal-free CdS/TiO<sub>2</sub>/WO<sub>3</sub> for photodegradation of polyoxometalate (POM: PMo<sub>12</sub>O<sub>40</sub><sup>3−</sup>). P25 TiO<sub>2</sub>–WO<sub>3</sub> was synthesized by the sol–gel method followed by hydrolysis of CdS and it was found that 1 : 1 : 1 molar ratio of CdS : TiO<sub>2</sub> : WO<sub>3</sub> gave the best output. HRTEM imaging showed uniform decoration of CdS onto TiO<sub>2</sub>, which is uniformly dispersed on WO<sub>3</sub>. The visible-light photocatalytic activity was measured with 100 mg L<sup>-1</sup> POM in the presence of methanol (electron donor) and *t*-butyl alcohol at pH of ~2.5 to support the reduction of POM is indeed due to CB electrons. The observed activity increased in the order of TiO<sub>2</sub>–WO<sub>3</sub> < CdS < CdS–WO<sub>3</sub> ≈ CdS–TiO<sub>2</sub> < CdS–TiO<sub>2</sub>–WO<sub>3</sub>. The authors have suggested that photogenerated electrons in CdS efficien-





tly transfer to  $\text{WO}_3$  via  $\text{TiO}_2$  because of the cascaded positioning of the CBs of three semiconductors, which is consistent with the observed significant photocurrent generation efficiency of the ternary nanocomposite. However, the authors have also mentioned that the ternary nanocomposite was not very stable, and its photocatalytic activity was gradually reduced with repeated uses. The authors have attributed the issue to the instability of  $\text{CdS}$ ; this is in contrast to other reports,<sup>32</sup> which established enhanced stability of  $\text{CdS}$  in the ternary nanocomposite.

**2.2.3. Ag NP-graphene blend.** Chen and Liu *et al.*<sup>84</sup> have synthesized  $\text{Ag}/\text{AgBr}/\text{GO}$  by utilizing a surfactant-assisted assembly protocol, where an oil/water (oil is the guest and water is the host) and water/oil (water is the guest and oil is the host) microemulsion is used as the synthesis medium (Fig. 9). For the oil/water microemulsion, SEM imaging showed  $\text{Ag}/\text{AgBr}$  (average diameter  $\sim 200$  nm), whose surface is distinctly encapsulated with silk-like GO nanosheets. The photocatalytic activity was measured by mixing 7 mg of photocatalysts in MO's aqueous solution ( $20 \text{ mg L}^{-1}$ ). It was found that  $\text{Ag}/\text{AgBr}$  and  $\text{Ag}/\text{AgBr}/\text{GO}$  could degrade 45% and 75% within 4 min, respectively, for the water/oil synthesis medium, whereas, for the oil/water synthesis medium,  $\text{Ag}/\text{AgBr}$  and  $\text{Ag}/\text{AgBr}/\text{GO}$  could degrade 62 and 94% of MO within 4 min. The observed distinct enhancement of the catalytic activity upon the hybridization of GO is attributed to the smaller size of  $\text{Ag}/\text{AgBr}/\text{GO}$ , and the excellent adsorptive capacity of  $\text{Ag}/\text{AgBr}/\text{GO}$ .  $\text{Ag}/\text{AgBr}/\text{GO}$ 's observed higher efficiency has been attributed to the reinforced charge transfer and the suppressed recombination of electron-hole pairs. This is further evidenced by Raman spectroscopy showing a red shift in the G-band (from  $1604$  to  $1588 \text{ cm}^{-1}$ ) of GO, solidly confirming that the charge transfer could be essentially facilitated upon hybridization, leading to the enhanced photocatalytic activity of  $\text{Ag}/\text{AgBr}/\text{GO}$ . Furthermore, the higher photocatalytic performance of  $\text{Ag}/\text{AgBr}$  and  $\text{Ag}/\text{AgBr}/\text{GO}$  manufactured *via* the oil/water medium is attributed to the relatively higher surface content of metallic Ag.

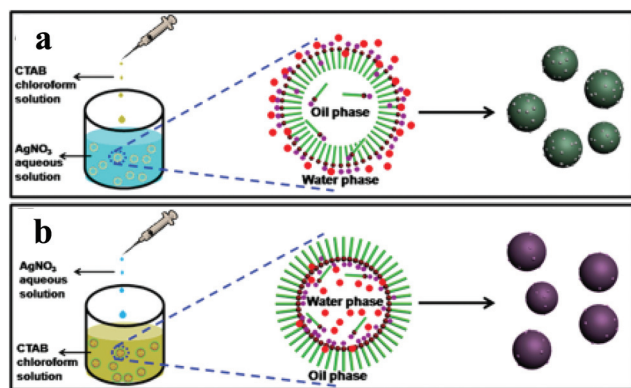


Fig. 9 Schematic explanation for the one-pot surfactant-assisted synthesis of  $\text{Ag}/\text{AgBr}$ -based nanostructures *via* an oil/water (a) and a water/oil (b) microemulsion system. Copyright 2012, American Chemical Society. Reprinted from ref. 84 with permission.

As an application, Chen *et al.*<sup>88</sup> have reported a  $\text{Ag}/\text{TiO}_2/\text{rGO}$  based reusable and highly sensitive surface-enhanced Raman scattering (SERS) substrate by implying the photocatalytic ability of  $\text{TiO}_2$  for the photodegradation of the adsorbed 4-aminothiophenol (ATP) molecules. The presence of Ag NPs onto  $\text{TiO}_2$  NPs decorated uniformly on wrinkled rGO was evidenced by TEM. A remarkable reduction in the detection limit of 4-ATP using ternary  $\text{Ag}/\text{TiO}_2/\text{rGO}$  ( $10^{-14} \text{ M}$ ) as compared to the binary  $\text{Ag}/\text{rGO}$  ( $10^{-10} \text{ M}$ )<sup>91</sup> substrate was observed. The substrate cleaning was carried out *via* the photodegradation under UV illumination (30 min), and all characteristic peaks of 4-ATP were found to disappear. In the five cycles examination, the SERS spectrum of 4-ATP using the recycled substrate had almost the same intensity. This demonstrated that the presence of the  $\text{TiO}_2$  interlayer could restrain the enhancement effect of Ag nanoparticles on the Raman signals of rGO and could act as the photocatalyst for the surface cleaning to make the substrate reusable.

Degradation of various pollutants such as MB,  $\text{Cr(IV)}$ , RhB, MO, and alcohol by employing ternary nanocomposites has shown better performance than their binary counterparts. Furthermore, a comparison of the different reports suggests that the highest kinetic rate constant for the photodegradation of MB was found to be  $\sim 0.2334 \text{ min}^{-1}$  by employing the g-CN/BiOI/GO ternary nanocomposite.<sup>54</sup> The observed highest degradation rate constant for  $\text{Cr(IV)}$  is  $\sim 0.05513 \text{ min}^{-1}$  using  $\text{Bi}_2\text{WO}_6/\text{MoS}_2/\text{rGO}$ .<sup>59</sup> Lee *et al.*<sup>67</sup> have reported an observed rate constant of  $\sim 0.02192$  for the degradation of RhB by employing CN/ $\text{CdS}/\text{rGO}$ . These studies suggest that in the case of degradation, GO plays an important role compared to 0D NPs, whereas not a single report of the 1D NR-based nanocomposite has been reported. Chen *et al.*,<sup>88</sup> on the other hand, have observed 94% degradation of MO within 4 by employing  $\text{Ag}/\text{AgBr}/\text{GO}$ . The observed distinct enhancement of the catalytic activity upon the hybridization of GO is attributed to the smaller size of  $\text{Ag}/\text{AgBr}/\text{GO}$  and the excellent adsorptive capacity of  $\text{Ag}/\text{AgBr}/\text{GO}$ . This shows that the synergistic effect of graphene and 0D NPs can be combined to have high efficiency.

In summary, the lower overpotential for a redox reaction, hampered charge recombination and efficient charge transfer are the three crucial factors to enhance the photocatalytic activity. The superior activity of ternary nanocomposites can be attributed to the lower over-potential and synergistic effect of multiple nano-junctions. The formation of multiple nano-junctions leads to intimate contact between various nanocomposite components to facilitate efficient charge transfer among various components and thus enhanced separation of the photogenerated carriers. Although these are possible in binary nanocomposites, however,  $\text{H}_2$  generation or pollutant removal cannot be maximized because of either a low electron transfer rate or high overpotential, whereas, in the bare catalyst, fast recombination of charge carriers hampers the photocatalytic activity. Furthermore, the nano-size effect giving rise to more active sites for the adsorption of  $\text{H}^+$  ion or pollutants enhances the activity. The formation of ternary nanocomposites has additionally led to the increased stability of the active catalyst.



### 2.3. The photocatalytic mechanism in ternary nanocomposites

Several types of developed hetero-photocatalysts with different mechanisms,<sup>92</sup> such as type-II, S-scheme, and Z-scheme, are discussed here, in brief, to bring out the similarity and difference in the photocatalytic mechanism of ternary and binary nanocomposites. For example, in the type-II binary photocatalyst, interparticle electron transfer in one direction and interparticle hole transfer in the opposite direction when two semiconductors are coupled together give rise to spatial separation of charge carriers. Although spatial separation of charge carrier type-II photocatalysts can hamper the charge recombination; however, the type-II system has its shortfalls. The improved charge-separation efficiency occurs at the cost of reduced redox ability, an unfavorable condition for the photocatalytic reaction. This is because photogenerated electrons pile up in the CB of semiconductors with weak reduction potential. In contrast, photogenerated holes gather in the VB of semiconductors with weak oxidation potential, which may weaken the overall driving force. The S-scheme binary heterojunction,<sup>93</sup> on the other hand, is composed of reduction and oxidation photocatalysts with staggered band structures, wherein the significant photogenerated electrons and holes are reserved in the CB and VB of reduction and oxidation photocatalysts, respectively. In contrast, the inefficient photogenerated charge carriers are recombined. This whole process, thus, introduces a strong redox potential and stronger driving force for the photocatalytic reaction. In comparison, the Z-scheme involves the generation of electron-hole pairs in semiconductor (1), followed by the injection of these electrons into the coupled semiconductor (2) for hydrogen evolution, whereas VB holes of the semiconductor (1) are usually involved in a water oxidation reaction.<sup>94</sup>

It is to be noted that this clear distinction of various mechanisms is made for binary nanocomposites, wherein both the components used are semiconductors and usually take part in the generation of charge carriers (type-II, S-scheme). Whereas when it comes to ternary nanocomposites, three different components are involved with mainly one active component (*i.e.*, semiconductor to generate charge carriers upon light irradiation) and two supportive (cocatalyst) components. Due to (uni- or bi-) directional charge flow from the active catalyst towards cocatalyst/s and *vice versa*, it is tricky to recognize the specific type of mechanism in these ternary nanocomposites, a mechanistic fact which is in contrast to that of binary nanocomposites. Although a few of the reported studies on ternary nanocomposites have titled the mechanism type (such as the Z-scheme<sup>36</sup>), the underlying mechanism in most of the reported potential systems has been conversed without any specific type<sup>21,23,32</sup> due to the complexity of charge transfer among the aforesaid three components.

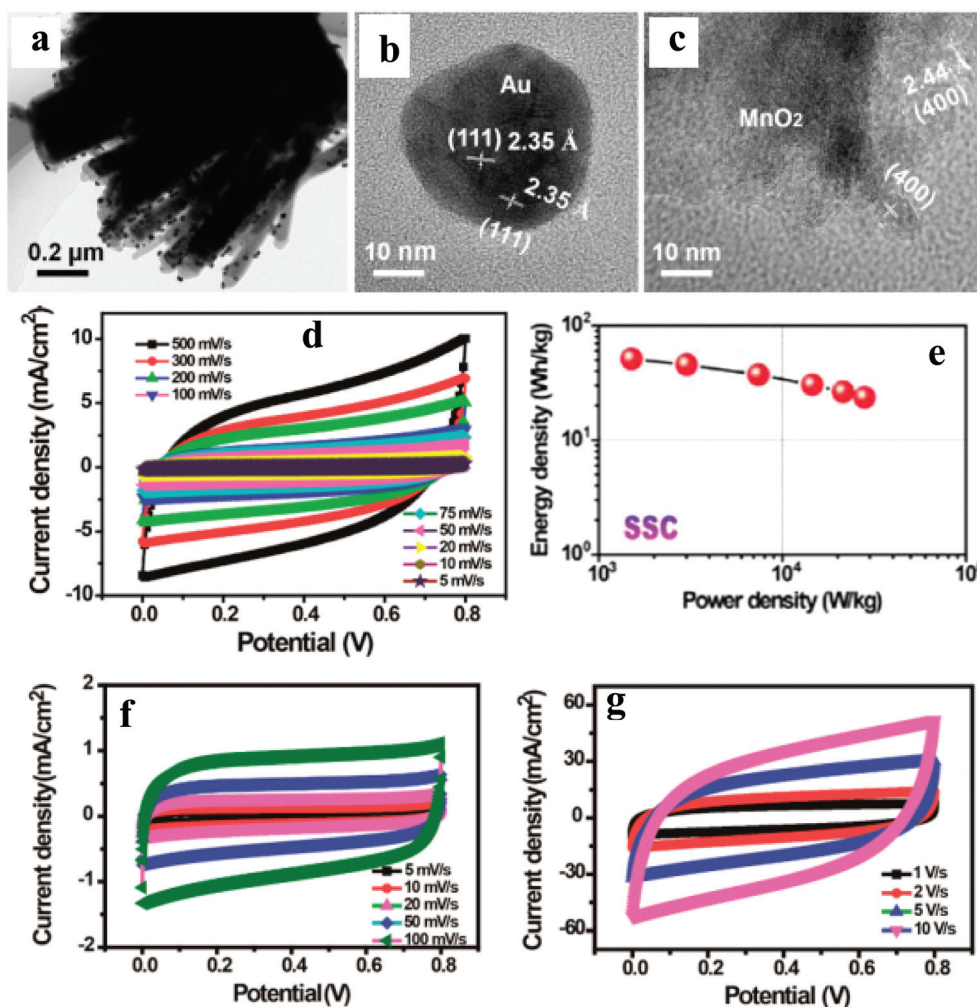
### 2.4. Ternary electrocatalyst

A few reports have been published on the integration of noble metal NPs and carbon nanotube/graphene with active electrocatalysts for their application in supercapacitors.<sup>95–102</sup>

Wang *et al.*<sup>99</sup> have reported an N-doped carbon nanotube (NCT)/Au NP/MnO<sub>2</sub>-based thin film supercapacitor with ultrafast charge/discharge. TEM imaging shows the as-prepared MnO<sub>2</sub> thin films with distributed Au NPs covering the surface of NCT (Fig. 10a–c). A three-electrode system is used to evaluate the electrochemical performances of NCTs/Au/MnO<sub>2</sub> and NCT/MnO<sub>2</sub> electrodes. The achieved gravimetric capacitance of ~1106 and ~740 F g<sup>−1</sup> at a current density of 5 A g<sup>−1</sup> with a capacity retention of 103% and 75% over 5000 cycles with NCT/Au/MnO<sub>2</sub> and NCT/MnO<sub>2</sub> electrodes, respectively, shows better performance of the ternary nanocomposite. The observed enhanced performance is attributed to three-dimensional scaffolds of NCT, providing a larger surface area and shorter ion diffusion. Furthermore, Au NPs distributed on MnO<sub>2</sub> thin layers can provide abundant Au/MnO<sub>2</sub> interfaces for rapid electron transport, thus, enhancing the conductivity of MnO<sub>2</sub> without sacrificing its inherent electrochemical performance. To further explore the electrochemical properties of the NCTs/Au/MnO<sub>2</sub> in practical application, the authors have studied a solid-state flexible symmetrical supercapacitor (SSC: NCTs/Au/MnO<sub>2</sub>//NCTs/Au/MnO<sub>2</sub>) and asymmetrical supercapacitor (ASC: NCTs/Au/MnO<sub>2</sub>//Fe<sub>2</sub>O<sub>3</sub> nanorods) prototype device. The observed symmetrical discharge-charge curves have been attributed to the good capacitive behavior of the SSC and ASC devices (Fig. 10d–g). Furthermore, the SSC device exhibited a large energy density of 51 W h kg<sup>−1</sup> (Fig. 10e), whereas the ASC device exhibited ultrafast charge/discharge (10 V s<sup>−1</sup>). The superior cycling performance for SSC (93%) and ASC (97%) devices over 5000 cycles has been noted. The SSC and ASC device's excellent rate performance was attributed to the lower resistance and faster charge transport, which is supported by nearly no semicircle in the high-frequency region in the Nyquist plots, indicating an ultrasmall transfer resistance ensuring fast charge transfer between the electrolyte and electrodes.

In another report, Li and Liu *et al.*<sup>100</sup> have synthesized a 3D macroporous structure of Ag NPs decorated on MnO<sub>2</sub>/sulfonated graphene (SG) by a hydrothermal process. SEM and TEM imaging demonstrated the hollow SG nanobeads with a well-ordered 3D porous structure. The average diameter of the hollow beads is ~300 nm. SEM imaging further showed that MnO<sub>2</sub> NPs with Ag NPs (~8 nm) were deposited on SG's surface. A three-electrode system has evaluated the electrochemical performances of Ag-MnO<sub>2</sub>/SG, MnO<sub>2</sub>/SG, and MnO<sub>2</sub> electrodes. The cyclic voltammetry (CV) curves at a 5 mV s<sup>−1</sup> scan rate show that the 3D Ag-MnO<sub>2</sub>/SG electrode possesses enhanced charge storage ability compared to bare MnO<sub>2</sub> and the MnO<sub>2</sub>/SG binary nanocomposite. A retained quasi-rectangular shaped CV curve of the Ag-MnO<sub>2</sub>/SG electrode at a high scan rate of 100 mV s<sup>−1</sup> indicates fast charge propagation capability and facile ion transport. The maximum specific capacitance for Ag-MnO<sub>2</sub>/SG could reach 537.0 F g<sup>−1</sup>, which is higher than that of the MnO<sub>2</sub>/SG (508.8 F g<sup>−1</sup>) and MnO<sub>2</sub> (172.26 F g<sup>−1</sup>) at a current density of 0.5 A g<sup>−1</sup>. Furthermore, the observed capacity retention of 99.72% for the Ag-MnO<sub>2</sub>/SG electrode after 5000 cycles is much higher than that of bare





**Fig. 10** (a) Low-magnification TEM image of NCTs/ANPDM nanocomposite, (b and c) TEM images of Au NPs covered by the MnO<sub>2</sub> and MnO<sub>2</sub> besides the Au NPs on the surface of the NCTs, respectively. (d) CV curves of the assembled solid-state SSC device collected at different scan rates, (e) Ragone plots of the solid-state SSC device, and (f and g) CV curves of the assembled solid-state ASC device collected at different scan rates. Copyright 2016, American Chemical Society. Reprinted from ref. 98 with permission.

MnO<sub>2</sub> (77.36%). From the Nyquist plots, the authors have inferred that the Ag-MnO<sub>2</sub>/SG shows the minimum charge transfer resistance due to the formation of intimate contact at the interface. The superior performances of Ag-MnO<sub>2</sub>/SG are attributed to (i) the interconnected 3D porous SG networks, which can reduce the diffusion distance for the electrolyte and accumulating more electrolyte ions, improving the rate performance and capacity, and (ii) improved conductivity due to Ag NPs.

Lan and Wang *et al.*<sup>101</sup> have synthesized Ag NP embedded Co<sub>3</sub>O<sub>4</sub> nanosheets on a three-dimensional graphene (3DG) network (Ag/Co<sub>3</sub>O<sub>4</sub>/3DG) by a hydrothermal method. The lack of the D-peak in the Raman spectra has been attributed to fewer defects in 3DG, giving rise to excellent conductivity, a desirable factor for current collectors in supercapacitors. The electrochemical performance was measured in a three-electrode configuration. Observed clear redox peaks in the CV curves were attributed to the conversion between different

cobalt oxidation states. At a current density of 1 mA cm<sup>-2</sup>, the galvanostatic charge/discharge curves show that the charge and discharge times of bare Co<sub>3</sub>O<sub>4</sub> foam and the Co<sub>3</sub>O<sub>4</sub>/3DG binary nanocomposite are shorter than those of the Ag/Co<sub>3</sub>O<sub>4</sub>/3DG hybrid. The areal-specific capacitance ( $C_a$ ) was calculated to be 303.9 mF cm<sup>-2</sup> (Ag/Co<sub>3</sub>O<sub>4</sub>/3DG), 145.9 (Co<sub>3</sub>O<sub>4</sub>/3DG), and 6.7 (Co<sub>3</sub>O<sub>4</sub> foam), which is attributed to the enhanced conductivity in the ternary nanocomposite due to the formation of an ohmic contact. Furthermore, the areal capacitance retention of ~120% after 104 cycles indicates a strong cycling performance. The galvanostatic charge/discharge curves of the Ag/Co<sub>3</sub>O<sub>4</sub>/3DG hybrid are found to be highly linear and symmetrical without an obvious IR drop, indicating a rapid IV response and excellent electrochemical reversibility. The coulombic efficiency of the Ag/Co<sub>3</sub>O<sub>4</sub>/3DG hybrid is above 97% during cycles. As displayed, the Nyquist plot of the Ag/Co<sub>3</sub>O<sub>4</sub>/3DG hybrid exhibits a negligible semicircular shape in the high-frequency range, which is related to the faradaic charge transfer





resistance suggesting that the electrode possesses effective interfacial charge transfer efficiency. The authors have further the electrochemical properties of the two-electrode configuration for practical applications. The two-electrode configuration exhibited the  $C_a$  of  $53.3 \text{ mF cm}^{-2}$  at  $1 \text{ mA cm}^{-2}$ . After 5000 cycles, the areal capacitance reaches  $49.1 \text{ mF cm}^{-2}$  and remains unchanged up to 10 000 cycles.

In short, superior electro-catalytic activity in ternary nanocomposites, *i.e.*, good electric conductivity, excellent specific capacitance, power performance, and good cyclic stability, can be attributed to fast electrode kinetics for facilitating fast electron transport from the electrode/electrolyte interface and improved conductivity due to the presence of NCT/graphene and Ag NPs.

### 3. Conclusions

With increasing attention toward the carbon-neutral society, the solar energy resource dominating the fossil-based energy resource is receiving heightened attention as a potentially widespread approach to achieve sustainable and green energy production. The possible energy production with sunlight striking earth in 1 h is much more than the entire year's energy consumed. However, the sun's diurnal nature necessitates the idea of harnessing solar energy for conversion and storage cost-effectively by employing photo/electrocatalysts. However, the conventional (homogeneous) catalyst appears far-fetched, demanding special efforts to develop unconventional (heterogeneous) catalysts. Nature has provided the existential proof of heterogeneous catalysts, with the hydrogenase enzymes operating at the thermodynamic potential for production of  $\text{H}_2$  from  $\text{H}_2\text{O}$ , and with the  $\text{O}_2$  evolving complex of photosystem II producing  $\text{O}_2$  from  $\text{H}_2\text{O}$  in an energy-efficient fashion.

Photo/electrocatalysts have been a part of our society for decades with an aim to achieve clean and green sustainable energy systems. Photocatalysis draws inspiration from natural photosynthesis and promises a direct route of photo-to-chemical energy conversion/storage and environmental pollutant removal, whereas photovoltaics employs photo-to-electrical energy conversion, which is stored in batteries/supercapacitors involving electro-catalysis. What has been grouped in this article represents the latest progress in the application of ternary nanocomposites. For convenience, we have categorized the photocatalytic section into four types: 2D materials, 1D nanorods, 0D nanoparticles, and 0D nanoparticle–2D material blends. Additionally, we have discussed Ag nanoparticles–graphene blend-based electrocatalyst, although there are only a few reports for the same. The formations of novel materials with multi nano-junctions have shed light on energy storage capabilities. We believe that the underlying mechanisms discussed can be of great inspiration to study and improve a large class of catalytic reactions.

Photocatalytic activity involving solar water splitting and pollutants' degradation has been measured employing mainly

hydrothermally grown particulate ternary nanocomposites. Sacrificial agents such as  $\text{Na}_2\text{S}$ – $\text{Na}_2\text{SO}_3$  and lactic acid solution have been noticed to show a huge difference in the photocatalytic activity. Mainly,  $\text{TiO}_2$ , CdS, and ZnCdS semiconductor-based ternary nanocomposites have been explored and shown enhanced activity compared to their pristine and peers under simulated UV and visible light excitation. It is particularly interesting to note that the apparent quantum efficiency of 100% has already been achieved by employing Pt tipped  $\text{CdSe@CdS}$ . However, this report includes an expensive Pt element, which makes it impossible for large scale applications in the future. It is evident from the current research that the CdS-based ternary nanocomposite comprising graphene shows AQE values greater than 60%; thus, it holds promise as a next-generation photocatalyst in terms of scalability, stability, and selectivity. Ternary nanocomposites have shown higher stability than their pristine and peers; for example, bare CdS is well known for its shown lesser stability, whereas CdS-based ternary nanocomposites have been proved to be highly stable. The observed higher activity in the ternary systems has been referred to the nano-size effect leading to more active sites and intimate contact between individual components, which can effectively suppress recombination and facilitate charge separation and transfer. Many visible light-responsive materials have been successfully implemented as  $\text{H}_2$  evolution photocatalysts and pollutants' degradation due to the relaxation of thermodynamic limitations on photocatalyst materials in Z-scheme systems.

Ternary electrocatalysts involving noble metal nanoparticles and active carbon materials such as carbon nanotubes and graphene have been shown to give better cycling behavior and high energy density and power density. This enhancement has been attributed to three-dimensional scaffolds for binder-free electrodes, larger surface area, and shorter ion diffusion along with noble metal nanoparticle/active electrocatalyst interfaces, which provide direct and stable pathways for rapid electron transport, thus enhancing the conductivity. Thus, the electrocatalytic property can greatly be improved without sacrificing the inherent electrochemical performance. A three-electrode-based-prototype device has already been brought into the light, which has surpassed the performance of many other electrocatalysts.

Although significant improvements have been achieved by employing ternary nanocomposites, challenges remain that need to be addressed to achieve benchmark efficiency. In the current scenario, the challenges that limit its development are by far connected to low cost, and requisite activity, selectivity, and durability. The challenges nonetheless provide a fertile field for research, to test different hypotheses and material design principles. For photocatalysts, compared with particle suspension systems, immobilized particulates can be more readily scaled up. In addition, the immobilization of photocatalysts on a conducting sub-layer can retain the material's intrinsic activity and show dramatic enhancement in photocatalytic performance.<sup>6</sup> Thus, immobilization can provide a prototype device for the large-scale production of low-cost



renewable energy resources and illustrate the importance of interfacial interaction, the kinetics of charge transport, adsorption of reactants, and transport of guest species to the active sites in the practical operation of such processes. In the literature, the photocatalytic activity has been mainly carried out using simulated sunlight. However, more experiments need to be carried out with solar light. Furthermore, experiments have been mainly carried out using UV and visible excitation; however, new active materials capable of harnessing IR or near-IR light are required as it covers a significant part of sunlight.

It is essential to mention here that even after decades of research, there is no standard method to compare the different reported efficiencies of photocatalysts in the literature. The amount of H<sub>2</sub> produced can differ depending on the photocatalyst mass, reactor used, excitation source/power, and growth parameters.<sup>21,23</sup> Therefore, the same photocatalyst can produce different values when used in different labs. We have made an attempt to compare different reports by comparing the rate of hydrogen in units of  $\mu\text{mol h}^{-1} \text{g}^{-1}$  and AQE. However, we would like to mention that hydrogen production and catalyst mass may not follow a linear relation, and often AQE is not measured with a standard protocol.<sup>103</sup>

Although encouraging results are being achieved; however, the field of a next-generation flexible supercapacitor is nascent. Great efforts are needed to go forward in the electrocatalyst technologies. A few of the challenges are to attain higher energy and higher-power density to fulfill our ever-increasing energy-dependency. Furthermore, advancements in ionic and electrical conductivity are indispensable for next-generation electrode materials at commercial levels. Thinner, lighter, smarter, or even transparent supercapacitors are needed for increasingly multifunctional consumer electronics. However, all this ends at one requirement; the energy storage mechanisms must be deeply understood for controlling interfacial reactions among various components of the electrode as well as between the electrode and electrolyte. A deeper level of understanding requires state-of-the-art *in situ* microscopic and spectroscopic studies to provide direct experimental proof of various reactions.

Despite significant progress towards high efficiency (HER) and high energy/power density (supercapacitors), most materials are far from the expected performance. Furthermore, the stringent requirements of multifunctionalities and industrial compatibility run far beyond the materials state-of-the-art. The current challenge in the field is the lack of rigorous control in designing decisive morphology for optimal performance on a large scale. In a broader perspective, ternary photo/electro-catalysts are still in their developing stage. No human-made catalytic system, either homogeneous or heterogeneous, has yet been identified that shows performance even close to natural enzymatic systems. To use the solar source abundantly, it is indispensable to distinguish the functional catalyst and to develop a deep understanding of the mechanism of ternary nanocomposites. This necessitates the structure–property–function relationships in more detail, particularly *in operando*

*in situ* condition *via* recent Synchrotron advancement. The structure here can refer to the crystallographic crystal structure, morphology, and spectroscopic average and nanoscale local structure, for example, a cubic spinel crystal structure with different spectroscopic local structures such as octahedral and tetrahedral sites. Minute changes in spectroscopic signals can lead to enormous change in optical and electronic properties,<sup>104,105</sup> which can affect material performances. For example, contrary to the current perspective of P25 with remarkable photocatalytic properties, rutile TiO<sub>2</sub>-based Ni–Ag–TiO<sub>2</sub> has shown higher activity compared to the P25-based ternary system due to enormous change in optical and electronic properties.<sup>106</sup> In another example, 1 at% doped Mn in lithium titanate showed ~20% change in the electrochemical performance due to Mn<sup>2+</sup> occupation in tetrahedral (8a) sites.<sup>107</sup> Furthermore, various charge transfer pathways in such sophisticated designed systems such as mediator-induced interfacial charge transfer<sup>108,109</sup> can be probed by synchrotron radiation-based technology as it has turned out to be an effective tool with remarkably high spatial/temporal-resolution,<sup>110,111</sup> thanks to its energy tunability and high brilliance. Thus, the elusive contact mechanism between the active photocatalyst and co-catalyst can be revealed *via* simultaneous spectro-microscopic determinations empowered by synchrotron-based advanced methods for investigating the structure–property function relationships. This will bring all three characteristics to the same page to establish the underlying mechanism, which in turn will be useful to tune the material performance efficiently. A decisive role in tuning and optimizing the existing properties along with the control of their responsible key factors will bring us a step closer to the industrialization of such ternary systems for the greener environment. Overall, efforts inspired by a broad range of scientific areas such as materials science, chemistry, physics, chemical engineering, nanoscience, and nanotechnology will be highly desirable for future breakthroughs in the promotion of overall catalytic efficiency. The development of such catalysts would provide key enabling technology for a full solar energy conversion and storage system. One of the critical possible future directions is to combine experiments and theory to uncover the underlying catalytic mechanism in the ternary systems and the role of various nano-junctions. Besides, future work needs to focus on different facile synthetic approaches for constructing stable ternary materials with a highly active crystal surface and/or quantum size on a large scale. Clearly, better catalysts for the multi-electron transformations are needed along with the understanding of multiscale energy conversion processes in these ternary nanocomposites to promote fundamental research into practical applications. Notably, the era of photo/electro-catalytic reactions conducted with a prototype device is expected soon.

## Conflicts of interest

The authors declare no conflicts of interest.



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## References

- 1 J. Barber, *Chem. Soc. Rev.*, 2009, **38**, 185–196.
- 2 N. S. Lewis, *Science*, 2007, **315**, 798–801.
- 3 A. Fujishima and K. Honda, *Nature*, 1972, **238**, 37–38.
- 4 B. A. Pinaud, J. D. Benck, L. C. Seitz, A. J. Forman, Z. Chen, T. G. Deutsch, B. D. James, K. N. Baum, G. N. Baum, S. Ardo, H. Wang, E. Miller and T. F. Jaramillo, *Energy Environ. Sci.*, 2013, **6**, 1983–2002.
- 5 D. M. Fabian, S. Hu, N. Singh, F. A. Houle, T. Hisatomi, K. Domen, F. E. Osterloh and S. Ardo, *Energy Environ. Sci.*, 2015, **8**, 2825–2850.
- 6 Q. Wang, T. Hisatomi, Q. Jia, H. Tokudome, M. Zhong, C. Wang, Z. Pan, T. Takata, M. Nakabayashi, N. Shibata, Y. Li, I. D. Sharp, A. Kudo, T. Yamada and K. Domen, *Nat. Mater.*, 2016, **15**, 611–615.
- 7 P. K. Nayak, S. Mahesh, H. J. Snaith and D. Cahen, *Nat. Rev. Mater.*, 2019, **4**, 269–285.
- 8 Y. Zhou, C. H. Wang, W. Lu and L. Dai, *Adv. Mater.*, 2020, **32**, 1–24.
- 9 J. Ding, W. Hu, E. Paek and D. Mitlin, *Chem. Rev.*, 2018, **118**, 6457–6498.
- 10 J. H. Kwak, Y.-W. Lee and J. H. Bang, *Mater. Lett.*, 2013, **110**, 237–240.
- 11 R. S. Kalubarme, H. S. Jadhav and C.-J. Park, *Electrochim. Acta*, 2013, **87**, 457–465.
- 12 W. Hong, J. Wang, Z. Li and S. Yang, *J. Mater. Chem. A*, 2015, **3**, 2535–2540.
- 13 S. P. Sancheti, Urvashi, M. P. Shah and N. T. Patil, *ACS Catal.*, 2020, **10**, 3462–3489.
- 14 A. Ciechanover, A. Hershko and I. Rose, *J. Biol. Chem.*, 2006, **281**, e32–e36.
- 15 C. Xu, P. R. Anusuyadevi, C. Aymonier, R. Luque and S. Marre, *Chem. Soc. Rev.*, 2019, **48**, 3868.
- 16 J. You, Y. Guo, R. Guo and X. Liu, *Chem. Eng. J.*, 2019, **373**, 624–641.
- 17 Z. Chang, J. Xu and X. Zhang, *Adv. Eng. Mater.*, 2017, **7**, 1700875.
- 18 Z. Wang, C. Li and K. Domen, *Chem. Soc. Rev.*, 2019, **48**, 2109–2125.
- 19 J. Ke, M. A. Younis, Y. Kong, H. Zhou, J. Liu, L. Lei and Y. Hou, *Nano-Micro Lett.*, 2018, **10**, 69.
- 20 Q. Xiang, J. Yu and M. Jaroniec, *J. Am. Chem. Soc.*, 2012, **134**, 6575–6578.
- 21 K. Chang, Z. Mei, T. Wang, Q. Kang, S. Ouyang and J. Ye, *ACS Nano*, 2014, **8**, 7078–7087.
- 22 J. Zhang, L. Qi, J. Ran, J. Yu and S. Z. Qiao, *Adv. Energy Mater.*, 2014, **4**, 1301925.
- 23 M. Liu, F. Li, Z. Sun, L. Ma, L. Xu and Y. Wang, *Chem. Commun.*, 2014, **50**, 11004–11007.
- 24 W. Jiang, Y. Liu, R. Zong, Z. Li, W. Yao and Y. Zhu, *J. Mater. Chem. A*, 2015, **3**, 18406–18412.
- 25 Y. J. Yuan, J. R. Tu, Z. J. Ye, D. Q. Chen, B. Hu, Y. W. Huang, T. T. Chen, D. P. Cao, Z. T. Yu and Z. G. Zou, *Appl. Catal., B*, 2016, **188**, 13–22.
- 26 S. N. Guo, Y. L. Min, J. C. Fan and Q. J. Xu, *ACS Appl. Mater. Interfaces*, 2016, **8**, 2928–2934.
- 27 Q. Xiang, F. Cheng and D. Lang, *ChemSusChem*, 2016, **9**, 996–1002.
- 28 S. Tonda, S. Kumar, Y. Gawli, M. Bhardwaj and S. Ogale, *Int. J. Hydrogen Energy*, 2017, **42**, 5971–5984.
- 29 M. Wang, P. Ju, J. Li, Y. Zhao, X. Han and Z. Hao, *ACS Sustainable Chem. Eng.*, 2017, **5**, 7878–7886.
- 30 S. Chen, F. Li, T. Li and W. Cao, *J. Colloid Interface Sci.*, 2019, **547**, 50–59.
- 31 X. Wang, G. Liu, L. Wang, Z. G. Chen, G. Q. Lu and H. M. Cheng, *Adv. Energy Mater.*, 2012, **2**, 42–46.
- 32 P. Kalisman, Y. Nakibli and L. Amirav, *Nano Lett.*, 2016, **16**, 1776–1781.
- 33 Z. Yue, A. Liu, C. Zhang, J. Huang, M. Zhu, Y. Du and P. Yang, *Appl. Catal., B*, 2017, **201**, 202–210.
- 34 G. Liu, C. Kolodziej, R. Jin, S. Qi, Y. Lou, J. Chen, D. Jiang, Y. Zhao and C. Burda, *ACS Nano*, 2020, **14**, 5468–5479.
- 35 J. Fang, L. Xu, Z. Zhang, Y. Yuan, S. Cao, Z. Wang, L. Yin, Y. Liao and C. Xue, *ACS Appl. Mater. Interfaces*, 2013, **5**, 8088–8092.
- 36 Z. B. Yu, Y. P. Xie, G. Liu, G. Q. Lu, X. L. Ma and H. M. Cheng, *J. Mater. Chem. A*, 2013, **1**, 2773–2776.
- 37 Z. Jiang, J. Pan, B. Wang and C. Li, *Appl. Surf. Sci.*, 2018, **436**, 519–526.
- 38 H. Yu, W. Liu, X. Wang and F. Wang, *Appl. Catal., B*, 2018, **225**, 415–423.
- 39 H. Dou, Y. Qin, F. Pan, D. Long, X. Rao, G. Q. Xu and Y. Zhang, *Catal. Sci. Technol.*, 2019, **9**, 4898–4908.
- 40 J. Y. Do, R. K. Chava, Y. Il Kim, D. W. Cho and M. Kang, *Appl. Surf. Sci.*, 2019, **494**, 886–894.
- 41 K. Maeda, A. Xiong, T. Yoshinaga, T. Ikeda, N. Sakamoto, T. Hisatomi, M. Takashima, D. Lu, M. Kanehara, T. Setoyama, T. Teranishi and K. Domen, *Angew. Chem., Int. Ed.*, 2010, **49**, 4096–4099.
- 42 Y. Che, Q. Liu, B. Lu, J. Zhai, K. Wang and Z. Liu, *Sci. Rep.*, 2020, **10**, 721.
- 43 D. Gao, W. Liu, Y. Xu, P. Wang, J. Fan and H. Yu, *Appl. Catal., B*, 2020, **260**, 118190.
- 44 Y. Yang, E. Liu, H. Dai, L. Kang, H. Wu, J. Fan, X. Hu and H. Liu, *Int. J. Hydrogen Energy*, 2014, **39**, 7664–7671.
- 45 F. J. Sheu and C. P. Cho, *Int. J. Hydrogen Energy*, 2017, **42**, 17020–17029.
- 46 F. J. Sheu and C. P. Cho, *ChemistrySelect*, 2018, **3**, 354–362.
- 47 K. Chang, D. Geng, X. Li, J. Yang, Y. Tang, M. Cai, R. Li and X. Sun, *Adv. Energy Mater.*, 2013, **3**, 839–844.
- 48 W. Ho, J. C. Yu, J. Lin, J. Yu and P. Li, *Langmuir*, 2004, **20**, 5865–5869.





- 49 C. She, G. W. Bryant, A. Demortière, E. V. Shevchenko and M. Pelton, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2013, **87**, 1–8.
- 50 Y. Nakibli, P. Kalisman and L. Amirav, *J. Phys. Chem. Lett.*, 2015, **6**, 2265–2268.
- 51 N. Zhang, Y. Zhang, X. Pan, M. Q. Yang and Y. J. Xu, *J. Phys. Chem. C*, 2012, **116**, 18023–18031.
- 52 X. Liu, L. Pan, T. Lv and Z. Sun, *J. Colloid Interface Sci.*, 2013, **394**, 441–444.
- 53 Y. Hou, Z. Wen, S. Cui, X. Guo and J. Chen, *Adv. Mater.*, 2013, **25**, 6291–6297.
- 54 K. Dai, L. Lu, C. Liang, G. Zhu, Q. Liu, L. Geng and J. He, *Dalton Trans.*, 2015, **44**, 7903–7910.
- 55 A. Akhundi and A. Habibi-Yangjeh, *Appl. Surf. Sci.*, 2015, **358**, 261–269.
- 56 N. Raghavan, S. Thangavel and G. Venugopal, *Mater. Sci. Semicond. Process.*, 2015, **30**, 321–329.
- 57 W. K. Jo and N. Clament Sagaya Selvam, *J. Hazard. Mater.*, 2015, **299**, 462–470.
- 58 D. B. Nimbalkar, H. H. Lo, P. V. R. K. Ramacharyulu and S. C. Ke, *RSC Adv.*, 2016, **6**, 31661–31667.
- 59 C. Zhang, G. Chen, C. Li, J. Sun, C. Lv, S. Fan and W. Xing, *ACS Sustainable Chem. Eng.*, 2016, **4**, 5936–5942.
- 60 Q. Liu, Y. Guo, Z. Chen, Z. Zhang and X. Fang, *Appl. Catal., B*, 2016, **183**, 231–241.
- 61 C. Zhou, J. Yan, B. Chen, P. Li, X. Dong, F. Xi and J. Liu, *RSC Adv.*, 2016, **6**, 108955–108963.
- 62 M. Liu, X. Xue, S. Yu, X. Wang, X. Hu, H. Tian, H. Chen and W. Zheng, *Sci. Rep.*, 2017, **7**, 1–11.
- 63 L. Zhang, X. He, X. Xu, C. Liu, Y. Duan, L. Hou, Q. Zhou, C. Ma, X. Yang, R. Liu, F. Yang, L. Cui, C. Xu and Y. Li, *Appl. Catal., B*, 2017, **203**, 1–8.
- 64 M. Darvishi, M. Jamali-Paghaleh, F. Jamali-Paghaleh and J. Seyed-Yazdi, *Mater. Res. Express*, 2017, **4**, 016501.
- 65 A. Ali, M. R. D. Biswas, Y. Areerob, D. C. T. Nguyen and W. C. Oh, *J. Korean Ceram. Soc.*, 2018, **55**, 381–391.
- 66 Z. Xie, Y. Feng, F. Wang, D. Chen, Q. Zhang, Y. Zeng, W. Lv and G. Liu, *Appl. Catal., B*, 2018, **229**, 96–104.
- 67 R. C. Pawar, V. Khare and C. S. Lee, *Dalton Trans.*, 2014, **43**, 12514–12527.
- 68 L. Ye, J. Liu, C. Gong, L. Tian, T. Peng and L. Zan, *ACS Catal.*, 2012, **2**, 1677–1683.
- 69 Q. Zhu, W. S. Wang, L. Lin, G. Q. Gao, H. L. Guo, H. Du and A. W. Xu, *J. Phys. Chem. C*, 2013, **117**, 5894–5900.
- 70 Y. Bu, Z. Chen and C. Sun, *Appl. Catal., B*, 2015, **179**, 363–371.
- 71 J. Li, Y. Xie, Y. Zhong and Y. Hu, *J. Mater. Chem. A*, 2015, **3**, 5474–5481.
- 72 K. H. Leong, S. L. Liu, L. C. Sim, P. Saravanan, M. Jang and S. Ibrahim, *Appl. Surf. Sci.*, 2015, **358**, 370–376.
- 73 L. Sun, W. Wu, Q. Tian, M. Lei, J. Liu, X. Xiao, X. Zheng, F. Ren and C. Jiang, *ACS Sustainable Chem. Eng.*, 2016, **4**, 1521–1530.
- 74 S. Wu, X. Shen, G. Zhu, H. Zhou, Z. Ji, K. Chen and A. Yuan, *Appl. Catal., B*, 2016, **184**, 328–336.
- 75 C. Zeng, Y. Hu, Y. Guo, T. Zhang, F. Dong, Y. Zhang and H. Huang, *ACS Sustainable Chem. Eng.*, 2016, **4**, 3305–3315.
- 76 S. F. Yang, C. G. Niu, D. W. Huang, H. Zhang, C. Liang and G. M. Zeng, *Environ. Sci.: Nano*, 2017, **4**, 585–595.
- 77 Y. Xu, Q. Liu, C. Liu, Y. Zhai, M. Xie, L. Huang, H. Xu, H. Li and J. Jing, *J. Colloid Interface Sci.*, 2018, **512**, 555–566.
- 78 Z. Zhang, X. Li, H. Chen, G. Shao, R. Zhang and H. Lu, *Mater. Res. Express*, 2018, **5**, 015021.
- 79 Y. Zhen, J. Wang, F. Fu, W. Fu and Y. Liang, *Nanomaterials*, 2019, **9**, 1054.
- 80 X. H. Zou, S. W. Zhao, J. G. Zhang, H. L. Sun, Q. J. Pan and Y. R. Guo, *Open Chem.*, 2019, **17**, 779–787.
- 81 H. I. Kim, J. Kim, W. Kim and W. Choi, *J. Phys. Chem. C*, 2011, **115**, 9797–9805.
- 82 S. Zhang, T. Yu, H. Wen, R. Guo, J. Xu, R. Zhong, X. Liac and J. You, *RSC Adv.*, 2020, **10**, 16892.
- 83 A. I. Vaizogullar, *J. Inorg. Organomet. Polym. Mater.*, 2020, **30**, 4129–4141.
- 84 M. Zhu, P. Chen and M. Liu, *Langmuir*, 2012, **28**, 3385–3390.
- 85 L. C. Sim, K. H. Leong, S. Ibrahim and P. Saravanan, *J. Mater. Chem. A*, 2014, **2**, 5315–5322.
- 86 Y. Chen, W. Huang, D. He, Y. Situ and H. Huang, *ACS Appl. Mater. Interfaces*, 2014, **6**, 14405–14414.
- 87 J. Xue, S. Ma, Y. Zhou, Z. Zhang and M. He, *ACS Appl. Mater. Interfaces*, 2015, **7**, 9630–9637.
- 88 K. C. Hsu and D. H. Chen, *ACS Appl. Mater. Interfaces*, 2015, **7**, 27571–27579.
- 89 K. H. Leong, L. C. Sim, D. Bahnemann, M. Jang, S. Ibrahim and P. Saravanan, *APL Mater.*, 2015, **3**, 104503.
- 90 N. Zhang, Y. Zhang, X. Pan, X. Fu, S. Liu and Y. J. Xu, *J. Phys. Chem. C*, 2011, **115**, 23501–23511.
- 91 K. C. Hsu and D. H. Chen, *Nanoscale Res. Lett.*, 2014, **9**, 1–9.
- 92 Q. Xu, L. Zhang, B. Cheng, J. Fan and J. Yu, *Chem*, 2020, **6**, 1543–1559.
- 93 J. Fu, Q. Xu, J. Low, C. Jiang and J. Yu, *Appl. Catal., B*, 2019, **243**, 556–565.
- 94 Z. Qiang, X. Liu, F. Li, T. Li, M. Zhang, H. Singh, M. Huttula and W. Cao, *Chem. Eng. J.*, 2021, **403**, 126327.
- 95 J. Yang and J. Y. Ying, *Angew. Chem., Int. Ed.*, 2011, **50**, 4637–4643.
- 96 Y. Feng, H. Liu, P. Wang, F. Ye, Q. Tan and J. Yang, *Sci. Rep.*, 2014, **4**, 1–7.
- 97 J. Kim, H. Ju, A. I. Inamdar, Y. Jo, J. Han, H. Kim and H. Im, *Energy*, 2014, **70**, 473–477.
- 98 W. Lan, Y. Sun, Y. Chen, J. Wang, G. Tang, W. Dou, Q. Su and E. Xie, *RSC Adv.*, 2015, **5**, 20878–20883.
- 99 Q. Lv, S. Wang, H. Sun, J. Luo, J. Xiao, J. W. Xiao, F. Xiao and S. Wang, *Nano Lett.*, 2016, **16**, 40–47.
- 100 L. Liu, J. Zhao, W. H. Li, X. Y. Zhang, H. Z. Xu, K. Li, Y. Kai, L. S. Sun, C. Q. Li and F. Q. Liu, *RSC Adv.*, 2016, **6**, 94682–94686.



- 101 X. Wang, X. Zhang, W. Han, X. Mu, Y. Zhang, X. Zhao, Y. Chen, Z. Yang, Q. Su, E. Xie and W. Lan, *Electrochim. Acta*, 2017, **224**, 260–268.
- 102 J. Li, Y. Sun, J. Wang, J. Tian, X. Zhang, H. Yang and B. Lin, *J. Alloys Compd.*, 2018, **749**, 783–793.
- 103 H. Kisch and D. Bahnemann, *J. Phys. Chem. Lett.*, 2015, **6**, 1907–1910.
- 104 E. Rani, A. A. Ingale, A. Chaturvedi, C. Kamal, D. M. Phase, M. P. Joshi, A. Chakrabarti, A. Banerjee and L. M. Kukreja, *J. Raman Spectrosc.*, 2016, **47**, 457–467.
- 105 E. Rani, A. Ingale, D. M. Phase, A. Chaturvedi, C. Mukherjee, M. P. Joshi and L. M. Kukreja, *Appl. Surf. Sci.*, 2017, **425**, 1089–1094.
- 106 P. M. Leukkunen, E. Rani, A. A. S. Devi, H. Singh, G. King, M. Alatalo, W. Cao and M. Huttula, Synergistic effect of Ni-Ag-rutile TiO<sub>2</sub> ternary nanocomposite for efficient visible-light-driven photocatalytic activity, *RSC Adv.*, 2020, **10**, 36930.
- 107 H. Singh, M. Topsakal, K. Attenkofer, T. Wolf, M. Leskes, Y. Duan, F. Wang, J. Vinson, D. Lu and A. I. Frenkel, *Phys. Rev. Mater.*, 2018, **2**, 125403.
- 108 K. Wang, J. Li and G. Zhang, *ACS Appl. Mater. Interfaces*, 2019, **11**, 27686.
- 109 X. Shi, S. Posysaev, M. Huttula, V. Pankratov, J. Hoszoeska, J. Dousse, F. Zeeshan, Y. Niu, A. Zakharov, T. Li, O. Miroshnichenko, M. Zhang, X. Wang, Z. Huang, S. Saukko, D. L. Gonzalez, S. van Dijken, M. Atatalo and W. Cao, *Small*, 2018, **14**, 1–10.
- 110 E. Paris, T. Sugimoto, T. Wakita, A. Barinov, K. Terashima, V. Kandyba, O. Proux, J. Kajitani, R. Higashinaka, T. D. Matsuda, Y. Aoki, T. Yokoya, T. Mizokawa and N. L. Saini, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2017, **95**, 035152.
- 111 H. Singh, D. Donetsky, J. Liu, K. Attenkofer, B. Cheng, J. R. Trelewicz, I. Lubomirsky, E. Stavitski and A. I. Frenkel, *Rev. Sci. Instrum.*, 2018, **89**, 045111.

