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Homogeneous nanodiscs of native membranes formed by stilbene–maleic-acid copolymers†

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Methylstilbene-*alt*-maleic acid copolymers spontaneously convert biological membranes into bilayer discs with ~20 nm diameters. This readily functionalizable class of copolymers has the compositional homogeneity, hydrophobicity, dynamics, and charge that may help to achieve optimal structural resolution, membrane dissolution, stability, and broad utility.

Introduction

Developing native nanodiscs¹ that solubilize intact membrane–protein assemblies (memteins) by using hydrolyzed styrene–maleic anhydride (SMA) copolymers has led researchers to synthesize various polymers and copolymers to enhance applications of this technology.² Resolving structures of memteins will inform how integral membrane proteins function in native lipid bilayer systems. A big challenge remains! The highly disperse memteins limit resolution in any structural determination technique. Without uniformity of memteins, visualizing, at atomic resolution, the polymers, lipids, and proteins cannot occur.

Conventional radical alternating copolymerization of styrene (S) and maleic anhydride (MA) has a long history.³ As in all radical copolymerizations, the resulting copolymers are polydisperse in molecular weight and vary in composition. SMA copolymers may be predominately alternating; however, as S can add to itself, some random distributions of S–S dyads and S–S–S triads exist in the copolymer chain. In contrast, MA does not homopolymerize; consequently, no MA–MA dyads exist in the chain. A continuous stirred tank reactor can

control the composition of the copolymers, but not the polydispersity of chains.³

Controlled radical copolymerizations with RAFT to produce SMA copolymer show promise in improving compositional homogeneity.^{4,5} In one study, varying [total monomer]/[RAFT agent] at high ratios of 43–400 : 1 and [S]/[MA] ratios of 2–20 : 1 produces copolymers with controlled compositional gradients and low polydispersity.⁴ In extracting a protein from HEK293F cells, the hydrolyzed copolymer produced with [total monomer]/[RAFT agent] of 43 : 1 and [S]/[MA] of 3 : 1 forms nanodiscs with lower polydispersity than those formed by SMA2000, a commercial SMA copolymer with [S]/[MA] of 2 : 1 and molar mass range of 3–7.5 kDa. In the other study, at a [total monomer]/[RAFT agent] ratio of 250 : 1 and [S]/[MA] ratios of 1.5–4 : 1, SMA copolymers with S-rich blocks form when [MA] reaches zero.⁵ When mixed with lipid vesicles, these hydrolyzed SMA-*block*-S copolymers produce nanodiscs of different sizes depending on the [S]/[MA] ratio.

Biomedical applications usually require specific, chemically well-defined compositions to achieve a desired result. To date, hydrolyzed SMA copolymers, the most widely used reagents to form native nanodiscs, still have (a) compositional irregularities, (b) batch-to-batch variabilities, and (c) challenges with aggregation in aqueous solutions. A (co)polymer must overcome these deficiencies to solve the critical challenge of forming stable and homogenous nanodiscs. Such protein–lipid–polymer nanodiscs would enable downstream functional and structural characterization with methods including cryo-electron microscopy, which has successfully determined memtein structures, although lipid headgroups and ligands remain unresolved.^{6–8}

Here we report initial studies with hydrolyzed methyl-substituted stilbene–maleic anhydride (STMA) copolymers (Scheme 1a) for extracting proteins from membranes by forming nanodiscs. Stilbene (ST), a 1,2-diphenylethylene monomer, differs from S, 1-phenylethylene monomer. In contrast to S, ST does NOT homopolymerize; neither does MA. As both monomers are not homopolymerizing, *strictly* alternating

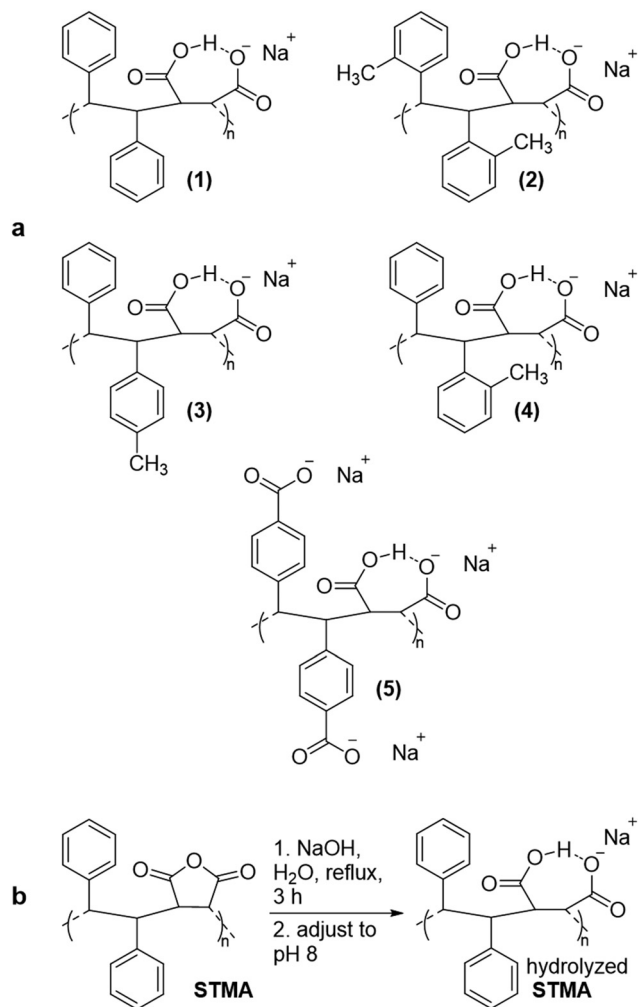
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Scheme 1 (a) Chemical structures of several hydrolyzed STMA (1), unsubstituted; (2), 2,2'-dimethyl-; (3), 4-methyl-; (4) 2-methyl; (5) 4,4'-dicarboxylato. (b) Hydrolysis conditions for activating all STMA copolymers.

copolymers form. These ST-MA copolymers meet one of the challenges—a precisely defined composition—that SMA copolymers do not.

The strictly alternating copolymers of MA with ST and with many substituted STs have similar persistence lengths (~ 3 nm).^{9,10} Both SMA and STMA copolymers have semi-rigid backbones; STMA copolymers have 30–50% greater persistence lengths than those (~ 2.2 nm) of SMA analogues. This increase results from the chain stiffening caused by steric crowding in the backbone of STMA.^{10,11} The chemical composition of STMA copolymers has $-(\text{CHPh-CHPh-MA})-$ repeating units; SMA, $-(\text{CH}_2\text{-CHPh-MA})-$. Replacing $-\text{CH}_2-$ with $-\text{CHPh-}$ increases the rigidity of the polymer.

Comparing SMA to STMA, a second phenyl group increases the hydrophobicity of the repeating unit. The most effective hydrolyzed SMA copolymers for forming nanodiscs have an $[\text{S}]/[\text{MA}]$ ratio of 2 : 1, presumably containing $(\text{CH}_2\text{-CHPh-CH}_2\text{-CHPh-MA})$ repeating units. This repeating unit has con-

siderably more hydrophobicity than that of a 1 : 1 SMA copolymer and nearly identical to that of a methyl-substituted STMA copolymer. However, the additional S unit in 2 : 1 SMA has conformational freedom, which reduces the rigidity of the copolymer.

Both hydrolyzed SMA and STMA show a large difference in the pK_a of the two carboxyl groups— ~ 4.4 and ~ 9.0 .¹² This difference indicates strong stabilization of the monoanion. At low pH, SMA and STMA exist in compact form; upon reaching a pH of ~ 6 , they form loose coils with statistical segment lengths of 5.9 and 6.7 nm, respectively. The repeating unit for the monoanion resembles that of succinate, which forms a strong intramolecular hydrogen bond.¹³ This bond restricts rotation of the backbone and contributes to the rigidity of the copolymers in the loosely coiled state.

Increased backbone rigidity of hydrolyzed dicarboxylated STMA, (5) compared to hydrolyzed carboxylated SMA may enhance anti-infectivity against HIV.¹⁴ Further docking studies of HIV surface proteins to the repeating unit of carboxylated STMA suggest a strong affinity of (5) to these proteins.¹⁵ The water solubility of (5) obviates using it to form nanodiscs. The hydrophobicity of methylated STMA copolymers, (2)–(4), should be conducive to membrane insertion and formation of nanodiscs.

Results and discussion

Methylated STMA copolymers were synthesized by free-radical polymerization of the substituted stilbene monomers and maleic anhydride reactants,¹⁶ purified by reprecipitation and then hydrolyzed (Scheme 1b). Hydrolyzed SMA2000 copolymer, which is well known to form native nanodiscs,¹ served as a control. FT-IR and NMR spectra (Fig. S1†) confirmed the conversion of maleic anhydride groups into maleic acids to form the water-soluble product. All hydrolyzed STMA copolymers were soluble at pH 8.0 in 10 mM Tris and 100 mM NaCl.

The sizes of the copolymers range between 4–6 kDa, except (5) (40 kDa) (Table 1). The polydispersity index (PDI) for the copolymers range from 1.19 to 1.54; the narrow PDIs likely arise from precipitation–reprecipitation workups of the copolymers. The PDI of commercial SMA is usually 2.0 or more.

To test whether hydrolyzed STMA copolymers spontaneously form nanodiscs from lipid bilayers, we examined whether they solubilized vesicles composed of dimyristoylphosphatidylcholine (DMPC). The polymer concentrations used equaled that of SMA2000 (1–2% w/v). Only (3) and (4) efficien-

Table 1 Properties of STMA copolymers (2)–(5) synthesized and tested for forming lipid–protein nanodiscs. As (1) is insoluble in THF, number-average molecular weight (M_n) and PDI values were not measured

Polymer number, name, abbreviation	M_n (kDa)	PDI
(1) Poly(stilbene- <i>alt</i> -maleic anhydride), STMA	Nd	Nd
(2) 2,2'-Dimethyl-STMA, 2,2'-STMA	5.1	1.52
(3) 4-Methyl-STMA, 4-STMA	5.8	1.54
(4) 2-Methyl-STMA, 2-STMA	4.4	1.19
(5) 4,4'-Dicarboxylato-STMA, 4,4'-STMA	40	1.34



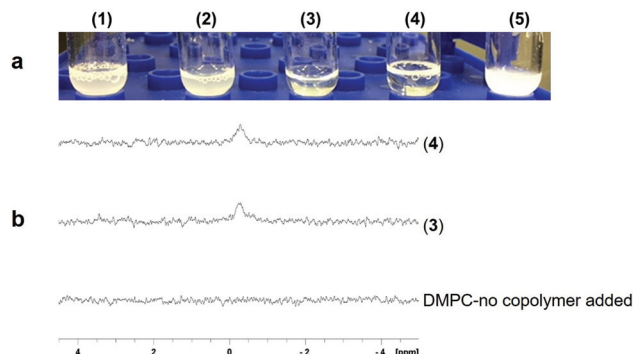


Fig. 1 (a) Solubilizing lipid vesicles in hydrolyzed STMA copolymers. Only (3) and (4) clarified the lipid suspension. (b) The broad ^{31}P NMR signal of DMPC (5 mM) resolved upon adding (3) and (4) at 2% w/v concentrations, indicating the formation of rapidly tumbling nanodiscs from larger vesicles.

tly dispersed multilayer vesicles of DMPC (Fig. 1a). In contrast, (1), (2) and (5) were relatively inactive, which was likely due to nonoptimal balances of hydrophobicity, negative charge, high molecular weight¹⁷ and dynamics needed to insert into and disrupt membranes. Given the weak activity of these three, only (3) and (4) were assayed further.

^{31}P NMR spectroscopy validated membrane-solubilizing activities of (3) and (4). Large vesicles composed of DMPC phospholipid exhibit ^{31}P NMR signals that are broadened beyond detection; adding (3) or (4) clarified liposome suspensions and readily resolved ^{31}P NMR signals (Fig. 1b). These results indicated formation of water-soluble nanodiscs and compared favorably to solubilization by the widely used SMA2000 copolymer.

We expected STMA copolymers to behave in solution similarly to SMA because both have similar pK_a values,^{12,13} similar charge distributions, and pH-dependent solubility profiles. SMA2000 was most water-soluble between pH 7.0 and 8.0. The STMA solubilities were tested. Surprisingly, (3) and (4) were soluble from pH 5 to 10 (Fig. 2).

These solubilities provide STMA copolymers a broader utility across the biological range of pH under which proteins operate in, for example, acidophiles, lysosomes and endosomes. The increased pH solubility of STMA likely results in part from the

- lack of polystyrene elements, which can seed aggregation,¹⁸
- strict alternation, which results in compact hydrophilic and hydrophobic groups, and
- more restricted backbone dynamics.

These advantages impart these hydrolyzed STMA copolymers with inherent practical advantages over hydrolyzed SMA copolymers.

Given the sensitivity of hydrolyzed SMA to divalent cations because of the maleic acid subunits, we expected both (3) and (4) to show similar sensitivities to levels of $[\text{Ca}^{2+}]$ that exceed 5 mM. Both (3) and (4) formed visible precipitates (Fig. S2†).¹⁹

This does limit using hydrolyzed STMA copolymers in functional assays that rely on high polycation levels over physiologi-

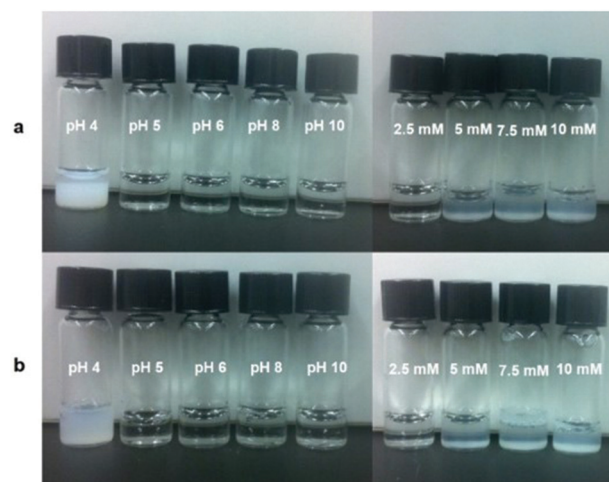


Fig. 2 The pH sensitivity (left panel) and divalent cation (*i.e.*, Ca^{2+} , pH 8.0) tolerance (right panel) (a) (3) and (b) (4).

cal levels. However, it does provide a method to coax membrane proteins from STMA-based nanodiscs and into liposomes or detergent phases.²⁰

We investigated (3) and (4) for their abilities to solubilize *E. coli* outer membranes, which are tightly packed with proteins and glycolipids. The β -barrel protein PagP was expressed into the outer membrane to test whether (3) and (4) extract it in a physiologically relevant lipid bilayer complex. Copolymers (3) and (4) were added individually to raw outer membrane samples at a copolymer concentration of 0.5% w/v, which is two-fold lower than the level generally recommended for SMA2000 activity, as high polymer concentrations can induce large, undesirable aggregates. The yield of PagP monomers and dimers^{21,22} solubilized directly from the outer membrane into the supernatant was comparable to the levels of PagP extracted by SMA2000 at 1 w/v% concentration (Fig. 3b).

These data suggest that (3) and (4) function like SMA2000 by releasing protein-lipid complexes directly from biological membranes into native nanodiscs. Moreover, (3) and (4) appear to stabilize protein dimers, implying a gentler interaction with multimeric complexes. Low concentrations of (3) and (4) (0.5% w/v) can still solubilize PagP from native membrane with no effect on its SEC elution volume as compared with PagP solubilized in SMA2000 polymer (Fig. S3†). However, the considerable amount of PagP in flow-through (FT) fractions suggests that higher concentration of (3) and (4) is required to achieve optimal solubilization.

The nanodiscs containing His-tagged PagP were purified in a single step over a standard metal affinity resin, indicating compatibility with common affinity tags. The stain-free SDS-PAGE gel displays a prominent band for (4) after elution (lane 6) (Fig. 3c) (right) with apparent molecular weights of 18 and 36 kDa. The bands correspond to the monomeric unfolded and dimeric states, respectively, of PagP following heat denaturation.²³ These states are less conspicuous in the purification when using (3) (Fig. 3c, lane 5) (left). The yield of



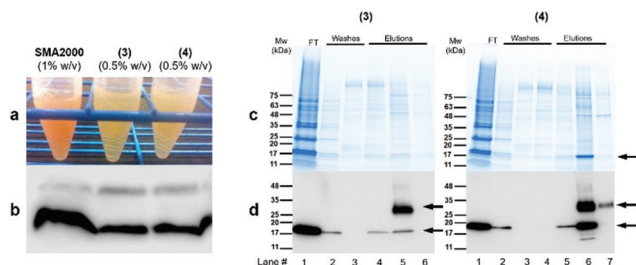


Fig. 3 (a) Solubilizing the total membrane of *E. coli* cultures expressing His₆-tagged PagP protein by copolymers (3) and (4). (b) Solubilization of PagP by (3), (4), and SMA2000 in total lysate was detected and compared on western blot. (c) The Ni-NTA affinity purification stages of PagP-STMA nanodiscs from the outer membrane of *E. coli* include flow-through (with 10 mM imidazole, lane 1), washing (20 mM imidazole, lanes 2, 3) and elution (250 mM imidazole, lanes 4, 5, 6) fractions, which were analyzed on stain-free gels with monomer and dimer bands confirmed by (d) western blotting. All samples were heated in buffer before running on SDS-PAGE. The band at ~30 kDa (upper arrows) represents dimeric PagP that is not fully unfolded after heating; hence showing different electrophoretic mobility than both fully folded dimeric PagP (~40 kDa) and fully unfolded monomeric PagP (~20 kDa, lower arrows).

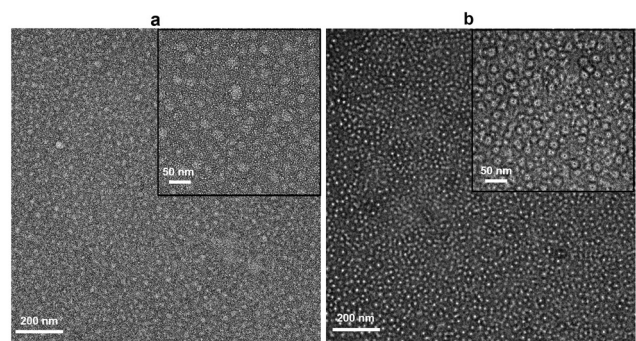


Fig. 4 Negative stain EM images of native nanodiscs formed from outer membranes containing PagP by using (a) (3) and (b) (4).

PagP monomer and dimer appears to depend on which hydrolyzed STMA copolymer is used, with (4) generating a higher overall yield of purified protein. In both cases, PagP nanodiscs elute at the same retention volume (~15 mL) by size exclusion chromatography (Fig. S3†), implying the similarity of sizes of nanodiscs.

Negative-stain TEM imaging of SEC fractions shows nanodiscs with diameters of ~20 nm (Fig. 4b), with those formed by (4) being most homogeneous. These nanodiscs appear to be more regular and double the size of the PagP-containing nanodiscs formed by SMA2000 copolymer.¹ Copolymer (4) appears to provide significantly more carrying capacity for larger memteins.

Conclusions

In summary, we report that, when activated, synthetic copolymers with *strictly* alternating ST and maleic acid subunits

having minimal sequence and batch polydispersity can effectively solubilize biological membrane, including native bacterial membranes. Both (3) and (4) have less backbone flexibility than SMA copolymers. Yet both retain effective lipid-bilayer interactions, penetrate biological membranes to excise nanodiscs, and capture memteins. Hydrolyzed STMA copolymers can be utilized for detergent-free purification of native membrane proteins. As such, these copolymers add valuable diversity to the repertoire of amphipathic polymers available for native nanodisc production. Copolymers (3) and (4) offer notable advantages:

- semi-rigidity of the polymer backbone,
- compositional regularity with strictly alternating periodicity of hydrophobic and hydrophilic groups,
- enhanced solubility over a wider range of pH values, and
- most significantly, generating large and homogeneous nanodiscs suitable for holding multimeric proteins in native bilayer sections.

Future refinements of the STMA copolymers can be envisaged. Various modifications of the maleic anhydride subunit^{24–27} could improve the polycation tolerance or enable incorporating thiols or other moieties (*e.g.* affinity tags and fluorescent labels). Such functional groups^{28–31} would further enhance the power of STMA copolymers and the utility of the resulting nanodiscs for large-scale production and high throughput assays of membrane protein targets. This unique potential of these STMA copolymers for detailed biophysical and structural analysis bodes well for future studies to increase the resolution of native protein–lipid–polymer complexes to the atomic level. The above results of (3) and (4) to enhance nanodisc solution behaviour, stability, and homogeneity represent a potential breakthrough for designing copolymers to maximize structural resolution of integral membrane proteins function in native lipid bilayers.

Conflicts of interest

M.E. and M.O. declare competing interests as they are co-inventors on a related patent filing.

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References

- 1 T. J. Knowles, R. Finka, C. Smith, Y.-P. Lin, T. Dafforn and M. Overduin, *J. Am. Chem. Soc.*, 2009, **131**(22), 7484–7485.



- 2 M. Overduin and M. Esmaili, *Chem. Phys. Lipids*, 2019, **218**, 73–84.
- 3 B. Klumperman, *Polym. Chem.*, 2010, **1**, 558–562.
- 4 A. A. A. Smith, H. E. Autzen, T. Laursen, V. Wu, M. Yen, A. Hall, S. D. Hansen, Y. Cheng and T. Xu, *Biomacromolecules*, 2017, **18**, 3706–3713.
- 5 A. F. Craig, E. E. Clark, I. D. Sahu, R. Zhang, N. D. Frantz, M. S. Al-Abdul-Wahid, C. Dabney-Smith, D. Konkolewicz and G. A. Lorigan, *Biochim. Biophys. Acta, Biomembr.*, 2016, **1858**, 2931–2939.
- 6 C. Sun, S. Benlekbi, P. Venkatakrishnan, Y. Wang, S. Hong, J. Hosler, E. Tajkhorshid, J. L. Rubinstein and R. B. Gennis, *Nature*, 2018, **557**(7703), 123–126.
- 7 W. Qiu, Z. Fu, G. G. Xu, R. A. Grassucci, Y. Zhang, J. Frank, W. A. Hendrickson and Y. Guo, *Proc. Natl. Acad. Sci. U. S. A.*, 2018, **115**(51), 12985–12990.
- 8 I. Tascon, J. S. Sousa, R. A. Corey, D. J. Mills, D. Griwatz, N. Aumuller, V. Mikusevic, P. J. Stansfeld, J. Vonck and I. Hanelt, *Nat. Commun.*, 2020, **11**(1), 626.
- 9 A. M. Savage, X. Zhou, J. Huang and S. R. Turner, *Appl. Petrochem. Res.*, 2015, **5**(1), 27–33.
- 10 J. Huang, X. Geng, C. Peng, T. Z. Grove and S. R. Turner, *Macromol. Rapid Commun.*, 2018, **39**, 1700530.
- 11 Y. Li, M. Zhang, M. Mao, S. R. Turner, R. B. Moore, T. H. Mourey, L. A. Slater and J. R. Hauenstein, *Macromolecules*, 2012, **45**, 1595–1601.
- 12 Y. Li, A. M. Savage, X. Zhou, S. R. Turner and R. M. Davis, *J. Polym. Sci., Part B: Polym. Phys.*, 2013, **51**(21), 1565–1570.
- 13 M. S. Rudner, S. Jeremic, K. A. Petterson, D. R. Kent IV, K. A. Brown, M. D. Drake, W. A. Goddard and J. D. Roberts, *J. Phys. Chem. A*, 2005, **109**(40), 9076–9082.
- 14 A. M. Savage, Y. Li, L. E. Matolyak, G. F. Doncel, S. R. Turner and R. D. Gandour, *J. Med. Chem.*, 2014, **57**(15), 6354–6363.
- 15 L. R. Hollingsworth, A. M. Brown, D. R. Bevan and R. D. Gandour, *PLoS One*, 2018, **13**, e190658.
- 16 Y. Li and S. R. Turner, *Eur. Polym. J.*, 2010, **46**(4), 821–828.
- 17 K. A. Morrison, A. Akram, A. Mathews, Z. A. Khan, J. H. Patel, C. Zhou, D. J. Hardy, C. Moore-Kelly, R. Patel, V. Odiba, T. J. Knowles, M. U. Javed, N. P. Chmel, T. R. Dafforn and A. J. Rothnie, *Biochem. J.*, 2016, **473**(23), 4349–4360.
- 18 S. Scheidelaar, M. C. Koorengel, C. A. van Walree, J. J. Dominguez, J. M. Dörr and J. A. Killian, *Biophys. J.*, 2016, **111**, 1974–1986.
- 19 M. C. Fiori, Y. Jiang, G. A. Altenberg and H. Liang, *Sci. Rep.*, 2017, **7**(1), 1–10.
- 20 S. J. Hesketh, D. P. Klebl, A. J. Higgins, M. Thomsen, I. B. Pickles, F. Sobott, A. Sivaprasadarao, V. L. G. Postis and S. P. Muench, *Biochim. Biophys. Acta, Biomembr.*, 2020, **1862**(5), 183192.
- 21 D. J. K. Swainsbury, S. Scheidelaar, N. Foster, R. van Grondelle, J. A. Killian and M. R. Jones, *Biochim. Biophys. Acta, Biomembr.*, 2017, **1859**(10), 2133–2143.
- 22 M. M. Xue, L. S. Cheng, I. Faustino, W. L. Guo and S. J. Marrink, *Biophys. J.*, 2018, **115**(3), 494–502.
- 23 R. E. Bishop, H. S. Gibbons, T. Guina, M. S. Trent, S. I. Miller and C. R. Raetz, *EMBO J.*, 2000, **19**(19), 5071–5080.
- 24 N. Z. Hardin, T. Ravula, G. D. Mauro and A. Ramamoorthy, *Small*, 2019, e1804813.
- 25 S. Lindhoud, V. Carvalho, J. W. Pronk and M. E. Aubin-Tam, *Biomacromolecules*, 2016, **17**(4), 1516–1522.
- 26 T. Ravula, N. Z. Hardin, S. K. Ramadugu and A. Ramamoorthy, *Langmuir*, 2017, **33**(40), 10655–10662.
- 27 K. Yasuhara, J. Arakida, T. Ravula, S. K. Ramadugu, B. Sahoo, J. I. Kikuchi and A. Ramamoorthy, *J. Am. Chem. Soc.*, 2017, **139**(51), 18657–18663.
- 28 A. O. Oluwale, J. Klingler, B. Danielczak, J. O. Babalola, C. Vargas, G. Pabst and S. Keller, *Langmuir*, 2017, **33**(50), 14378–14388.
- 29 S. Hall, C. Tognoloni, J. Charlton, É. Bragginton, A. Rothnie, P. Sridhar, M. Wheatley, T. Knowles, T. Arnold, K. Edler and D. TR, *Nanoscale*, 2018, **10**(22), 10609–10619.
- 30 T. Ravula, N. Z. Hardin, S. K. Ramadugu and A. Ramamoorthy, *Langmuir*, 2017, **33**, 10655–10662.
- 31 T. Ravula, N. Z. Hardin, J. Bai, S. C. Im, L. Waskell and A. Ramamoorthy, *Chem. Commun.*, 2018, **54**, 9615–9618.

