

Cite this: *Nanoscale Adv.*, 2020, 2, 1333

# Few-layer hexagonal bismuth telluride ( $\text{Bi}_2\text{Te}_3$ ) nanoplates with high-performance UV-Vis photodetection<sup>†</sup>

Ye Zhang,<sup>†a</sup> Qi You,<sup>†a</sup> Weichun Huang,<sup>†b</sup> Lanping Hu,<sup>b</sup> Jianfeng Ju,<sup>b</sup> Yanqi Ge<sup>\*a</sup> and Han Zhang<sup>ID \*a</sup>

It is widely known that the excellent intrinsic electronic and optoelectronic advantages of bismuthene and tellurene make them attractive for applications in transistors and logic and optoelectronic devices. However, their poor optoelectronic performances, such as photocurrent density and photoresponsivity, under ambient conditions severely hinder their practical application. To satisfy the demand of high-performance optoelectronic devices and topological insulators, bismuth telluride nanoplates ( $\text{Bi}_2\text{Te}_3$  NPs) with different sizes, successfully synthesized by a solvothermal approach have been, for the first time, employed to fabricate a working electrode for photoelectrochemical (PEC)-type photodetection. It is demonstrated that the as-prepared  $\text{Bi}_2\text{Te}_3$  NP-based photodetectors exhibit remarkably improved photocurrent density, enhanced photoresponsivity, and faster response time and recovery time in the UV-Vis region, compared to bismuthene and tellurene-based photodetectors. Additionally, the PEC stability measurements show that  $\text{Bi}_2\text{Te}_3$  NPs have a comparable long-term stability for on/off switching behaviour for the bismuthene and tellurene-based photodetectors. Therefore, it is anticipated that the present work can provide fundamental acknowledgement of the optoelectronic performance of a PEC-type  $\text{Bi}_2\text{Te}_3$  NP-based photodetector, shedding light on new designs of high-performance topological insulator-based optoelectronic devices.

Received 6th January 2020  
Accepted 9th February 2020

DOI: 10.1039/d0na00006j

rsc.li/nanoscale-advances

## Introduction

The individual elements tellurium (Te) and bismuth (Bi) have been widely applied to the fabrication and applications of photodetectors over the past decade due to their simple composition, excellent nonlinear photonic performance,<sup>1–4</sup> and intriguing thermoelectric<sup>5–7</sup> and photoelectric properties.<sup>8–10</sup> Similar to the famous black phosphorus (BP), both Te and Bi have a layer-dependent energy band gap ( $E_g$ )<sup>1,3,8</sup> that can be easily tuned (Te: 0.35–1.0 eV;<sup>1</sup> Bi: 0–0.55 eV (ref. 3)) when the thickness of Te or Bi decreases from bulk to monolayer. Recently, Ye *et al.* reported that ultrathin 2D nonlayered Te nanosheets synthesized by a substrate-free solution process, displayed high on/off ratios ( $10^6$ ), remarkable field-effect mobility ( $700 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ ) and comparable air-stable performance.<sup>8</sup> Besides, the ultrathin Te nanosheets fabricated by

a simple liquid phase exfoliation (LPE) method, showed excellent photoresponse behaviors from the UV to the visible region in association with strong time and cycle stability for the on/off switching behaviors.<sup>10</sup> In 2017, Zhang *et al.* reported that ultrasmall Bi quantum dots fabricated by a LPE approach, exhibited good photoresponse performance from the UV to visible region as well as long-term photoresponse stability.<sup>11</sup> Nevertheless, the poor photoresponse performances of Te or Bi nanomaterials, especially photoelectrochemical (PEC) photocurrent density and photoresponsivity, still severely limit their device development for practical applications. Therefore, it is still a challenge to explore a new method to improve their photoresponse performance under ambient conditions.

However, bismuth telluride ( $\text{Bi}_2\text{Te}_3$ ), basically known as a compound of the post-transition metal element Bi and the non-metal element Te, also exhibits a thickness-dependent  $E_g$  (from 0.16 eV to 1.36 eV),<sup>12,13</sup> and a high structural stability.<sup>14,15</sup> Versatile strategies have already been employed to synthesize  $\text{Bi}_2\text{Te}_3$  nanomaterials, including template synthesis,<sup>16</sup> evaporation,<sup>17</sup> electrochemical deposition,<sup>18</sup> chemical solution process,<sup>19,20</sup> solvothermal approaches<sup>21,22</sup> and microwave-assisted methods.<sup>23</sup>  $\text{Bi}_2\text{Te}_3$  as one of the common topological insulators, features an unconventional phase of quantum matter possessing an insulating bulk state as well as a metallic surface state.<sup>17</sup> Such metallic surface states were experimentally

<sup>a</sup>SZU-NUS Collaborative Innovation Centre for Optoelectronic Science & Technology, International Collaborative Laboratory of 2D Materials for Optoelectronics Science and Technology of Ministry of Education, Institute of Microscale Optoelectronics, Shenzhen University, Shenzhen 518060, China. E-mail: hzhang@szu.edu.cn

<sup>b</sup>Nantong Key Lab of Intelligent and New Energy Materials, College of Chemistry and Chemical Engineering, Nantong University, Nantong 226019, Jiangsu, P. R. China

<sup>†</sup> Electronic supplementary information (ESI) available. See DOI: 10.1039/d0na00006j

<sup>‡</sup> Equal first-author contributions.



evidenced to be protected by time-reversal symmetry and demonstrated to be robust against non-magnetic perturbation.<sup>24,25</sup> In addition, topological insulator  $\text{Bi}_2\text{Te}_3$  exhibits an excellent surface mobility<sup>26</sup> and good optoelectronic performance.<sup>27</sup> This, combined with the relatively narrow  $E_g$  of  $\text{Bi}_2\text{Te}_3$  and low cost and facile synthesis of  $\text{Bi}_2\text{Te}_3$  nanomaterials, has drawn great interest in photodetection,<sup>28–30</sup> field effect transistors,<sup>26,31</sup> spintronics,<sup>32,33</sup> thermoelectrics,<sup>22,34</sup> and lasers.<sup>35–37</sup> These advantages of  $\text{Bi}_2\text{Te}_3$  merit it to be qualified for the practical application in high-performance optoelectronic devices.

In this work,  $\text{Bi}_2\text{Te}_3$  nanoplates (NPs) with a rhombohedral phase in the space group  $D_{3d}^5(R\bar{3}m)$ , have been successfully synthesized by a solvothermal approach. To determine the size-dependent PEC performances of  $\text{Bi}_2\text{Te}_3$  NPs, different sizes of  $\text{Bi}_2\text{Te}_3$  NPs were readily obtained by simply tuning the reaction time. The as-synthesized  $\text{Bi}_2\text{Te}_3$  NPs were, for the first time, developed as working materials to fabricate a PEC-type photodetector in various electrolytes. The PEC results demonstrate that the  $\text{Bi}_2\text{Te}_3$  NP-based photodetectors exhibit not only a largely improved photocurrent density and photoresponsivity, but also a comparable photoresponse stability compared to that of Bi or Te nanomaterial-based photodetectors. It is anticipated that this work can provide fundamental guidance for constructing high-performance  $\text{Bi}_2\text{Te}_3$  NP-based photodetectors, paving the way to new designs of topological insulator-based optoelectronic devices with excellent properties.

## Methods

### Materials

Poly(vinyl pyrrolidone) (PVP, K30), poly(vinylidene fluoride) (PVDF,  $M_w = 534\,000\text{ g mol}^{-1}$ ),  $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$ , dimethyl formamide (DMF, 99.9%), acetone, and ethanol were purchased from Sigma-Aldrich. Sodium tellurite ( $\text{Na}_2\text{TeO}_3$ , 99.9%) and indium tin oxide (ITO) were purchased from Aladdin Co., Inc. All chemical reagents were used without further purification. Double-distilled deionized (DI) water was used for synthesis.

### Synthesis of $\text{Bi}_2\text{Te}_3$ hexagonal NPs

In a typical procedure, 20 mmol  $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$  and 40 mmol  $\text{Na}_2\text{TeO}_3$  were first dissolved in 30 mL DI water. Then 0.03 mmol PVP was added into the solution and it was kept stirring for 30 min to form a homogeneous mixture. The mixture was transferred into a 50 mL Teflon-lined autoclave and placed in an oven at 180 °C. After a predetermined reaction time (2 h or 12 h), the reaction was stopped by quenching the system to room temperature. The  $\text{Bi}_2\text{Te}_3$  hexagonal NPs were obtained by centrifugation at 4000 rpm for 20 min and washed with deionized water, ethanol and acetone, each. The product was finally dried in a vacuum oven at 80 °C overnight for the next use.

### Characterization

The morphologies and dimensions of  $\text{Bi}_2\text{Te}_3$  NPs were determined by both scanning electron microscopy (SEM, Hitachi-

SU8010) and transmission electron microscopy (TEM, FEI Tecnai G2 F30). High-resolution TEM (HRTEM) was performed to determine the atomic arrangements of the as-synthesized  $\text{Bi}_2\text{Te}_3$  NPs. Energy-dispersive X-ray spectroscopy (EDS) was carried out using an FEI Tecnai G2 F30 TEM equipped with an Oxford EDAX EDS system. Atomic force microscopy (AFM, Bruker, with 512 pixels per line) was performed after depositing a drop of dispersion onto a silicon substrate. The X-ray diffraction (XRD) analysis was performed on an X'Pert-Pro MPD diffractometer with a Cu K- $\alpha$  radiation source at room temperature. Ultraviolet-visible (UV-Vis) absorption spectroscopy was performed with a spectral range of 200–1500 nm by using a UV-Vis absorbance spectrometer (Cary 60, Agilent) at room temperature.

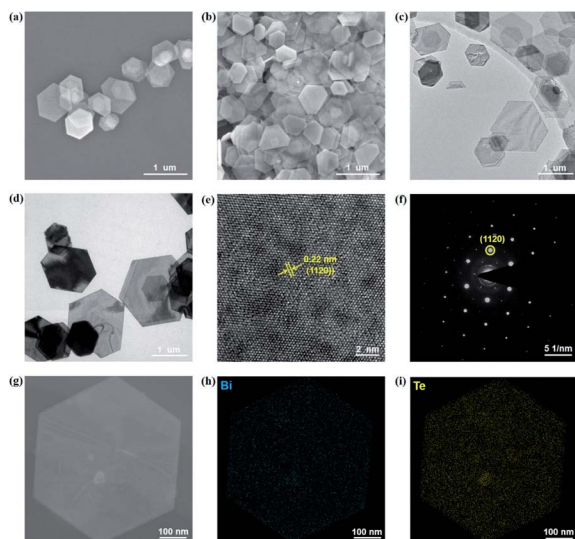
### Photoresponse activity

The PEC measurement system in Scheme S1† was used to characterize the photoresponse behaviour of  $\text{Bi}_2\text{Te}_3$  NPs. A standard three-electrode system, that is, a working electrode (for example,  $\text{Bi}_2\text{Te}_3$  NPs deposited on ITO-coated glass, photoanode), a counter electrode (platinum wire, photocathode), and a reference electrode (Ag/AgCl electrode), was assembled in various aqueous electrolytes, including KOH (0.1 M, 0.5 M, and 1.0 M), KCl (0.5 M), and HCl (0.5 M). To ensure good adhesion between ITO-coated glass and the sample, the as-synthesized samples were first re-dispersed in a  $0.2\text{ mg mL}^{-1}$  PVDF/DMF solution, and then the dispersion was deposited onto ITO-coated glass and dried under vacuum at 80 °C overnight. Electrochemical impedance spectra (EIS) were obtained in the frequency range from 1 to  $10^5$  Hz with an amplitude of 0.005 V. Amperometric current–time ( $I-t$ ) curves were recorded at bias voltages of 0 V, 0.3 V, and 0.6 V with increasing power densities at a sampling interval of 5 s. Simulated light (300–800 nm) and lasers with different  $\lambda$  ( $\lambda = 365\text{ nm}$ , 400 nm, 475 nm, 550 nm, 600 nm and 700 nm) were employed to irradiate the  $\text{Bi}_2\text{Te}_3$  NP-based photodetectors. Light power densities ( $P_i$ ) of these irradiations with labels of Dark, I, II, III, IV, and VI levels gradually increased (Table S1†). As a control experiment, a piece of naked ITO-coated glass was also irradiated by using a SL under the same conditions.

## Results and discussion

The  $\text{Bi}_2\text{Te}_3$  NPs with a well-defined hexagonal shape were synthesized by the solvothermal method. In order to investigate the influence of the size of  $\text{Bi}_2\text{Te}_3$  NPs on photoresponse performances, two kinds of  $\text{Bi}_2\text{Te}_3$  NPs with different sizes have been facilely synthesized by tuning the reaction time. For convenience, the  $\text{Bi}_2\text{Te}_3$  NPs reacted after 2 h and 12 h are abbreviated as  $\text{Bi}_2\text{Te}_3$  NPs-1 and  $\text{Bi}_2\text{Te}_3$  NPs-2, respectively. Fig. 1a and b give the SEM images of the as-prepared  $\text{Bi}_2\text{Te}_3$  NPs-1 and  $\text{Bi}_2\text{Te}_3$  NPs-2, both of which exhibit a well-defined hexagonal shape, while the average lateral dimensions of  $\text{Bi}_2\text{Te}_3$  NPs-1 and  $\text{Bi}_2\text{Te}_3$  NPs-2 are  $620 \pm 150\text{ nm}$  and  $730 \pm 210\text{ nm}$ , respectively. The TEM characterization (Fig. 1c and d) reveals that the as-prepared  $\text{Bi}_2\text{Te}_3$  NPs also represent a well-defined

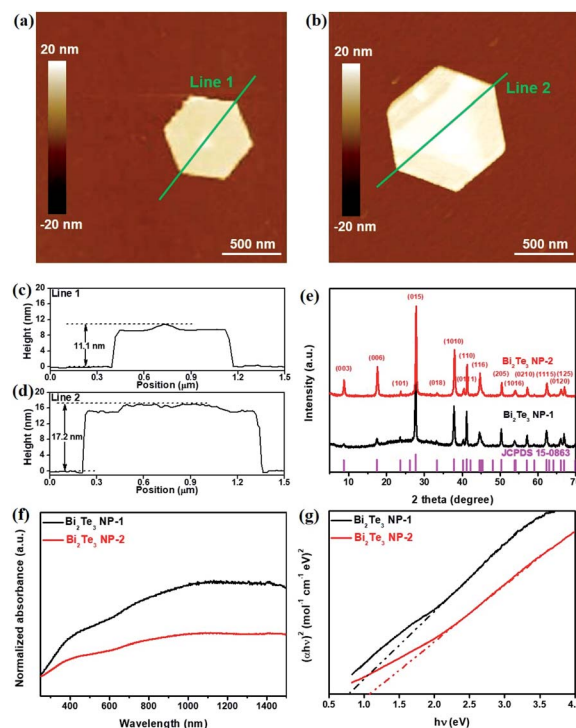




**Fig. 1** Structural characterization of the as-prepared  $\text{Bi}_2\text{Te}_3$  NPs. SEM images of (a)  $\text{Bi}_2\text{Te}_3$  NPs-1 and (b)  $\text{Bi}_2\text{Te}_3$  NPs-2. TEM images of (c)  $\text{Bi}_2\text{Te}_3$  NPs-1 and (d)  $\text{Bi}_2\text{Te}_3$  NPs-2. (e) HRTEM image of  $\text{Bi}_2\text{Te}_3$  NPs-1. (f) SAED pattern of  $\text{Bi}_2\text{Te}_3$  NPs-1. (g) TEM image of one single hexagonal  $\text{Bi}_2\text{Te}_3$  NP and results of its EDS elemental mapping of (h) Bi and (i) Te shown in (g).

hexagonal shape. Besides,  $\text{Bi}_2\text{Te}_3$  NPs-2 have darker hexagons (Fig. 1d), compared to  $\text{Bi}_2\text{Te}_3$  NPs-1 (Fig. 1c), indicating that the longer the reaction time the thicker the  $\text{Bi}_2\text{Te}_3$  NPs are, which is in good agreement with the “oriented attachment” mechanism of  $\text{Bi}_2\text{Te}_3$  NPs.<sup>38,39</sup> The HRTEM image (Fig. 1e) shows a clear lattice spacing of 0.22 nm, consistent with the (1120) plane of layered  $\text{Bi}_2\text{Te}_3$ .<sup>27</sup> Sharp diffraction spots are observed in the selected area electron diffraction (SAED) pattern (Fig. 1f), and the EDS line scan analysis (Fig. 1g) reveals compositional variation in a single  $\text{Bi}_2\text{Te}_3$  NP, suggesting that Bi and Te are evenly distributed (Fig. 1h and j).

The thicknesses of the as-synthesized  $\text{Bi}_2\text{Te}_3$  NPs-1 and  $\text{Bi}_2\text{Te}_3$  NPs-2 were characterized by AFM, as shown in Fig. 2a and b, respectively. It can be clearly seen in Fig. 2c and d that with the increase in the reaction time, the measured thickness of  $\text{Bi}_2\text{Te}_3$  NPs obviously increases from 11.1 nm to 17.2 nm, which correspond to 11 and 17 layers, respectively, given that one layer is regarded as an average quintuple layer of Te–Bi–Te–Bi–Te with a thickness of 1.0 nm.<sup>17</sup> XRD patterns of the as-synthesized  $\text{Bi}_2\text{Te}_3$  NPs, as shown in Fig. 2e, can be indexed to a rhombohedral  $\text{Bi}_2\text{Te}_3$  structure (JCPDS Card Number 15-0863). UV-Vis absorption spectroscopy was employed to characterize the optical response of differently sized  $\text{Bi}_2\text{Te}_3$  NPs (Fig. 2f). Broadband absorption from 260 nm to 1500 nm, is observed for  $\text{Bi}_2\text{Te}_3$  NPs-1 and  $\text{Bi}_2\text{Te}_3$  NPs-2, which is in good agreement with previously reported results,<sup>40,41</sup> implying great potential for application in broadband optoelectronic devices. Besides, Tauc plots of  $\text{Bi}_2\text{Te}_3$  NPs-1 and  $\text{Bi}_2\text{Te}_3$  NPs-2 (Fig. 1g) were calculated based on the results in Fig. 1f, and size-dependent  $E_g$  values of 0.83 eV ( $\text{Bi}_2\text{Te}_3$  NPs-1) and 1.0 eV ( $\text{Bi}_2\text{Te}_3$  NPs-2) were obtained, close to that of the previously reported  $\text{Bi}_2\text{Te}_3$  nanoparticles,<sup>42</sup> suggesting that the  $E_g$  of  $\text{Bi}_2\text{Te}_3$  NPs



**Fig. 2** AFM images of (a)  $\text{Bi}_2\text{Te}_3$  NPs-1 and (b)  $\text{Bi}_2\text{Te}_3$  NPs-2. (c, d) Height profiles along the green lines in (a) and (b), respectively. (e) XRD patterns of  $\text{Bi}_2\text{Te}_3$  NPs-1 and  $\text{Bi}_2\text{Te}_3$  NPs-2. (f) UV-Vis spectra of the as-prepared  $\text{Bi}_2\text{Te}_3$  NPs-1 and  $\text{Bi}_2\text{Te}_3$  NPs-2 and (g) their Tauc plots for the calculations of  $E_g$ .

could be easily controlled by simply tuning the reaction conditions.

The typical photoresponse behaviour of the  $\text{Bi}_2\text{Te}_3$  NP-based photodetector was evaluated using a PEC system equipped with a standard three-electrode configuration, as shown in Scheme S1.† Fig. 3a gives the patterns of the as-fabricated  $\text{Bi}_2\text{Te}_3$  NPs cast onto ITO-coated glass, exhibiting strong on/off switching photoresponse behaviours at an applied bias voltage of 0.6 V. For clarity, the profile of the naked ITO-coated glass was added in Fig. 3a, which displays a negligible signal as compared to  $\text{Bi}_2\text{Te}_3$  NPs under the same conditions, revealing that the photoresponse signal indeed comes from the  $\text{Bi}_2\text{Te}_3$  NPs rather than ITO. In addition, it should be noted that the type of electrolytes plays an important role in the PEC performance. As is shown in Fig. 3b, c and e,  $\text{Bi}_2\text{Te}_3$  NPs-2 irradiated by using several single-wavelength lasers, exhibit an excellent on/off switching photoresponse behaviour in 0.5 M KCl (Fig. 3c) and 0.5 M KOH (Fig. 3e) while they show poor performance in 0.5 M HCl (Fig. 3b) at 0.6 V, suggesting that KCl and KOH are preferred electrolytes for  $\text{Bi}_2\text{Te}_3$  NPs in this PEC system but HCl is not. Besides, the influences of lasers with different  $\lambda$  and  $P_\lambda$  on the PEC performance were investigated (Fig. 3c–f). Six lasers with specific  $\lambda$  ( $\lambda = 365, 400, 475, 550, 650,$  and  $700$  nm) and  $P_\lambda$  were employed to demonstrate the laser wavelength-dependent PEC performance of  $\text{Bi}_2\text{Te}_3$  NPs. Similarly, the profiles of naked ITO-coated glass irradiated by using a SL are also added for comparison in Fig. 3c–f. It can be observed that when the  $\lambda$  value





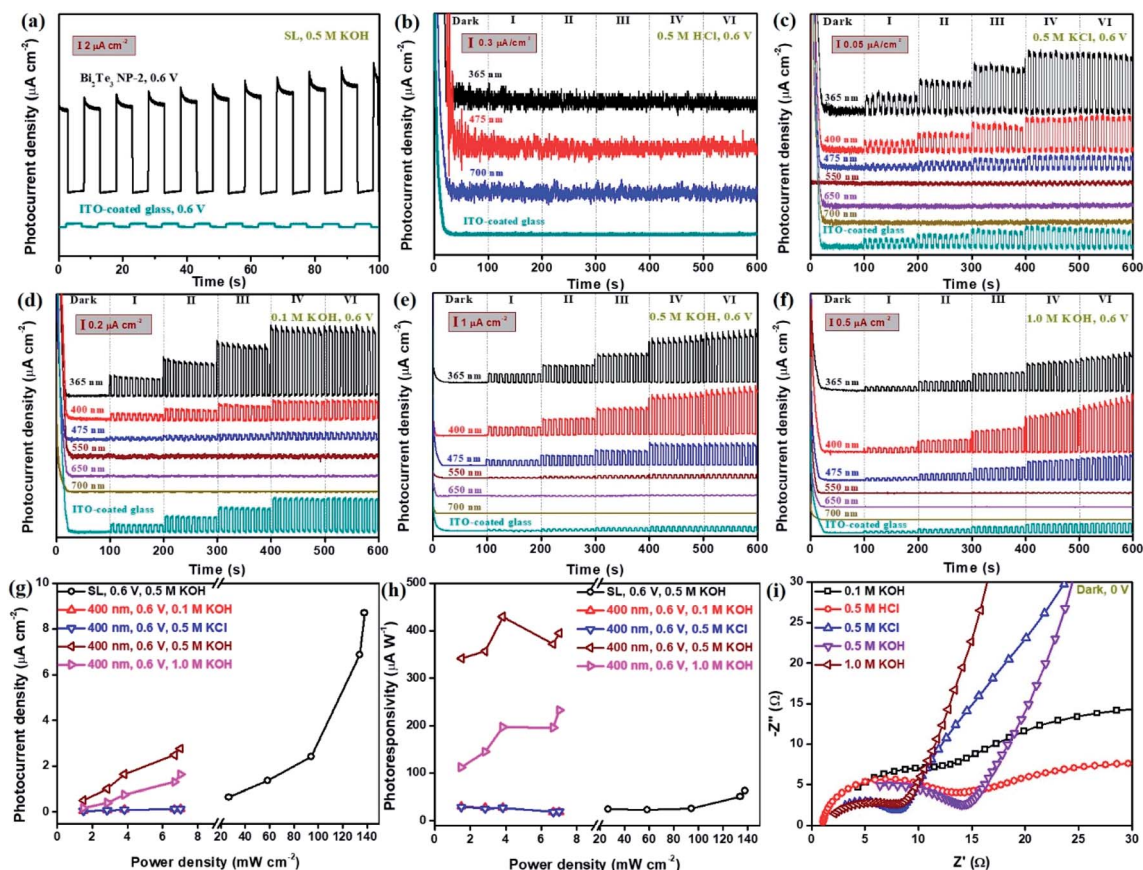


Fig. 3 The photoresponse behaviours of  $\text{Bi}_2\text{Te}_3$  NP-2-based photodetectors in various electrolytes under a SL and light with various wavelengths. (a) The on/off switching behaviours triggered by a SL in 0.5 M KOH at 0.6 V at  $134 \text{ mW cm}^{-2}$ . (b–f) The on/off switching behaviours in various electrolytes under light with wavelengths of 365, 400, 475, 550, 650, and 700 nm at 0.6 V in (b) 0.5 M HCl, (c) 0.5 M KCl, (d) 0.1 M KOH, (e) 0.5 M KOH, and (f) 1.0 M KOH. For clarity, the photoresponse profile of naked ITO-coated glass irradiated by using a SL was added. (g)  $P_{\text{ph}}$  as a function of light power density under a SL and light with various wavelengths. (h)  $R_{\text{ph}}$  as a function of light power density under a SL and irradiation with various wavelengths. (i) EIS patterns of  $\text{Bi}_2\text{Te}_3$  NPs-2 in various electrolytes under dark environments at 0.6 V.

is less than 550,  $\text{Bi}_2\text{Te}_3$  NPs show a strong PEC signal in KCl and KOH and the signal gradually increases with the decrease of the  $\lambda$  value (Fig. 3c–f), which can be attributed to the relatively high laser energy and  $P_{\lambda}$  (Table S1†). However, when  $\lambda \geq 550$  nm, a negligible photoresponse signal of  $\text{Bi}_2\text{Te}_3$  NPs can be observed, different from the results in absorption spectra, which is attributed to the very weak laser energy employed for PEC measurements in this work. In addition, it should be pointed out that the signal of naked ITO-coated glass irradiated by using a SL with much higher  $P_{\lambda}$  is obviously lower than that of  $\text{Bi}_2\text{Te}_3$  NPs (Fig. 3c–f) irradiated by using single-wavelength lasers with shorter  $\lambda$ , e.g., 365 nm, while higher than that irradiated by using single-wavelength lasers with longer  $\lambda$ , e.g., 650 nm and 700 nm, reconfirming the truth of the PEC signal of  $\text{Bi}_2\text{Te}_3$  NPs. Furthermore, it shows the same trend when the  $P_{\lambda}$  gradually increases from dark to VI in both KCl and KOH, that is, the PEC signal of  $\text{Bi}_2\text{Te}_3$  NPs increases with the  $P_{\lambda}$  (Fig. 3c–f). To quantitatively evaluate the photoresponse performance of  $\text{Bi}_2\text{Te}_3$  NPs, the photocurrent density ( $P_{\text{ph}}$ ) and photoresponsivity ( $R_{\text{ph}}$ ) can be obtained by:<sup>43,44</sup>

$$P_{\text{ph}} = (I_{\text{light}} - I_{\text{dark}})/S \quad (1)$$

$$R_{\text{ph}} = P_{\text{ph}}/P_{\lambda} \quad (2)$$

where,  $I_{\text{light}}$  and  $I_{\text{dark}}$  are the drain current with and without light, respectively;  $P_{\lambda}$  and  $S$  are the light power density and effective area of the  $\text{Bi}_2\text{Te}_3$  NPs on ITO-coated glass, respectively. The bias voltage dependent on the PEC performance was also studied (Fig. 3e and S1†). The  $P_{\text{ph}}$  of  $\text{Bi}_2\text{Te}_3$  NPs-2 in 0.5 M KOH gradually increases with the applied bias voltage, i.e., the  $P_{\text{ph}}$  of  $\text{Bi}_2\text{Te}_3$  NPs-2 irradiated by using a 365 nm laser increases from  $44.8 \text{ nA cm}^{-2}$  (0 V), to  $96.4 \text{ nA cm}^{-2}$  (0.3 V), to  $2.52 \mu\text{A cm}^{-2}$  (0.6 V), similar to that in 0.5 M KCl (Fig. 3c and S2†). Photocurrent generation of the  $\text{Bi}_2\text{Te}_3$  NPs at 0 V means that the  $\text{Bi}_2\text{Te}_3$  NP-based photodetector is able to display self-powered PEC performance in KOH, yet additional external bias voltage can strengthen the photocurrent generation, which can be ascribed to the fact that the external bias voltage across the photoelectrode can construct a potential gradient within  $\text{Bi}_2\text{Te}_3$  NPs and enhance the separation of photo-generated holes and electrons.<sup>45</sup> Therefore, we think that the photoresponse mechanism of  $\text{Bi}_2\text{Te}_3$  NPs is similar to that of bismuthene and tellurene: (i) formation of electron ( $e^-$ )-hole



( $h^+$ ) pairs by photoexcitation and (ii) photoinduced charge transportation.<sup>10,11</sup> Surprisingly, it should be noted that the  $P_{ph}$  and  $R_{ph}$  of  $Bi_2Te_3$  NPs can reach up to  $8.68 \mu A cm^{-2}$  and  $395 \mu A W^{-1}$  (Fig. 3g and h), respectively, both of which largely outperform the reported bismuthene-based or tellurene-based photodetectors,<sup>10,11</sup> which could be attributed to the unique property of the topological insulator,  $Bi_2Te_3$ . Notably, the  $P_{ph}$  of  $Bi_2Te_3$  NPs in this work is also remarkably superior to those of ZnO homojunction nanowires ( $\sim 0.28 nA cm^{-2}$ )<sup>46</sup> and GaN nanowires ( $\sim 0.45 nA cm^{-2}$ ),<sup>47</sup> considering the same effective area of the measured samples ( $2.2 cm^2$ ). Moreover, it is observed (Fig. 3g and h) that electrolyte concentration has a great effect on the PEC signal of  $Bi_2Te_3$  NPs, that is, the  $P_{ph}$  and  $R_{ph}$  of  $Bi_2Te_3$  NPs irradiated by using a 400 nm laser increase in the range of KOH concentration from 0.1 M to 0.5 M while both of them decrease from 0.5 M to 1.0 M, possibly ascribed to the slight electrochemical reaction at both bias voltage and high electrolyte concentration.<sup>48</sup> The resistance ( $R$ ) at the interface between the electrolyte and electrode gradually decreases with the increase of KOH concentration,  $R_{0.1 M}$  ( $16.3 \Omega$ )  $>$   $R_{0.5 M}$  ( $13.0 \Omega$ )  $>$   $R_{1.0 M}$  ( $8.09 \Omega$ ), while in 0.5 M electrolytes, the fact that  $R_{HCl}$  ( $13.3 \Omega$ )  $>$   $R_{KOH}$  ( $13.0 \Omega$ )  $>$   $R_{KCl}$  ( $7.25 \Omega$ ) (Fig. 3i) can be due to the different functionalities between  $Bi_2Te_3$  NPs and electrolytes, similar to the reported results of black phosphorus nanosheets<sup>49</sup> and bismuth sulfide(III) nanosheets.<sup>45</sup>

Due to the size-dependent  $E_g$  of  $Bi_2Te_3$  NPs (Fig. 2g), the photoresponse behaviours of the  $Bi_2Te_3$  NPs-1 and  $Bi_2Te_3$  NPs-2 irradiated by using a SL and three kinds of lasers (365, 400, and 475 nm) were studied to understand the influence of the size of  $Bi_2Te_3$  NPs on the PEC performance, as shown in Fig. 4. It can be seen that the PEC signals of both  $Bi_2Te_3$  NPs-1 and

$Bi_2Te_3$  NPs-2 increase with the  $P_\lambda$  (Fig. 4a) and show the same trend as those of the  $Bi_2Te_3$  NPs-2, *i.e.*, the PEC signal declines as the  $\lambda$  value increases (Fig. 4b). In addition, it is noted that the PEC signal of  $Bi_2Te_3$  NPs-2 is obviously stronger than that of  $Bi_2Te_3$  NPs-1, no matter which laser was employed (Fig. 4a and b), *e.g.*, the  $P_{ph}$  of  $Bi_2Te_3$  NPs-2 irradiated by using a 400 nm laser at  $4.65 mW cm^{-2}$  is  $2.52 \mu A cm^{-2}$  while that of  $Bi_2Te_3$  NPs-1 is only  $0.729 \mu A cm^{-2}$ . This could be attributed to the synergistic effect of the suitable  $E_g$  and the number of accessible active sites on the  $Bi_2Te_3$  NPs-2. Since the  $E_g$  of  $Bi_2Te_3$  NPs inversely correlates with size, the larger  $Bi_2Te_3$  NPs-1 have stronger absorption under incident light. However, with the size decrease of  $Bi_2Te_3$  NPs, the specific surface area becomes higher and accessible active sites on the PEC performance of  $Bi_2Te_3$  NP-based photodetectors, similar to previously reported results.<sup>10,44,50</sup> Therefore, the size of nanomaterials has a great influence on the PEC performance;  $Bi_2Te_3$  NPs tend to be larger, resulting in higher efficiency, which provides fundamental acknowledgement of the design and optimization of PEC-type devices.

In addition, the response time ( $t_{res}$ ) and recovery time ( $t_{rec}$ ) of the  $Bi_2Te_3$  NP-based photodetector were ascribed to the time interval for the rise and decay from 10% to 90% and from 90% to 10% of its peak value, respectively.<sup>48,51</sup> It can be observed that regardless of the size of  $Bi_2Te_3$  NPs and the type of electrolytes, the  $Bi_2Te_3$  NP-based photodetector always shows fast  $t_{res}$  (0.001–0.09 s) as well as  $t_{rec}$  (0.001–0.07 s) (Fig. 5), both of which are superior to those of the bismuthene-based photodetector ( $t_{res} = 0.2 s$ ,  $t_{rec} = 0.2 s$ ),<sup>11</sup> tellurene-based photodetector ( $t_{res} = 0.2 s$ ,  $t_{rec} = 0.2 s$ ),<sup>10</sup> ZnO homojunction nanowires ( $t_{res} = \sim 50 s$ ,  $t_{rec} = \sim 200 s$ )<sup>46</sup> and GaN nanowires ( $t_{res} = 0.003 s$ ,  $t_{rec} = 0.003 s$ ).<sup>47</sup> This could be attributed to the unconventional phase of  $Bi_2Te_3$  quantum matter. This

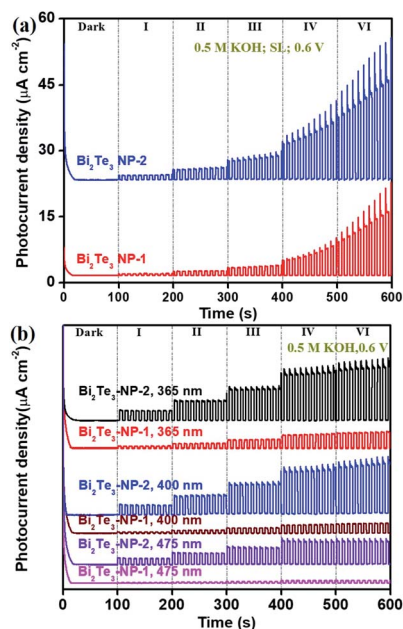


Fig. 4 Size effect of  $Bi_2Te_3$  NPs on the photoresponse performance in 0.5 M KOH under (a) a SL and (b) three different lasers (365, 400, and 475 nm).

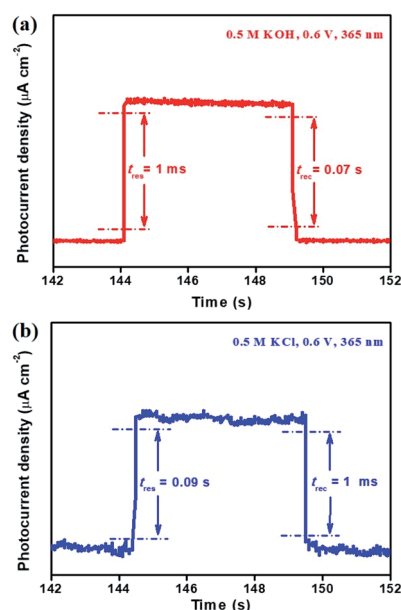


Fig. 5 The  $t_{res}$  and  $t_{rec}$  of (a)  $Bi_2Te_3$  NPs-1 in 0.5 M KOH and (b)  $Bi_2Te_3$  NPs-2 in 0.5 M KCl, irradiated by using a 365 nm laser.



indicates that the  $\text{Bi}_2\text{Te}_3$  NP-based photodetector has appealing potential in the field of PEC-type devices.

Long-term stability measurements of the photoresponse of  $\text{Bi}_2\text{Te}_3$  NP-based photodetectors are of great importance for practical application. Fig. 6 gives the stability profiles of the  $\text{Bi}_2\text{Te}_3$  NP-1-based photodetector irradiated by using a SL in 0.1 M KOH at a bias voltage of 0.6 V. The 600 cycles with 5 s intervals of on/off switching were traced (Fig. 6a), and the 481–500<sup>th</sup> cycles were chosen to evaluate the PEC stability of the  $\text{Bi}_2\text{Te}_3$  NP-1-based photodetector (Fig. 6b). It should be pointed out that no obvious change is observed by visual inspection of  $\text{Bi}_2\text{Te}_3$  NPs-1 specimens after PEC stability measurements in 0.1 M KOH (Fig. S3<sup>†</sup>), indicating the excellent PEC stability of the as-fabricated  $\text{Bi}_2\text{Te}_3$  NP-1-based photodetector. A notable on/off switching behaviour can be observed even after one month, suggesting the long-term PEC stability of  $\text{Bi}_2\text{Te}_3$  NPs under ambient conditions. Furthermore, it is calculated from Fig. 6b that the  $P_{\text{ph}}$ s of the fresh  $\text{Bi}_2\text{Te}_3$  NPs-1 in the 481–500<sup>th</sup> cycles is  $879 \text{ nA cm}^{-2}$  in 0.1 M KOH, declining to  $439 \text{ nA cm}^{-2}$  after one month. An approximate reduction of 50.1% of  $P_{\text{ph}}$  was obtained, comparable to that of the bismuthene-based photodetector<sup>11</sup> and tellurene-based photodetector.<sup>10</sup> The decline of  $P_{\text{ph}}$  could be ascribed to the weak electrochemical reaction at high bias voltage (0.6 V) under a SL with high  $P_{\lambda}$  ( $134 \text{ mW cm}^{-2}$ ) and slight peel-off during long-running measurements, which can be efficiently solved by coating conductive polymers, such as polyaniline and polypyrrole, onto the surface of  $\text{Bi}_2\text{Te}_3$  NPs to remarkably lower the electrochemical reaction of  $\text{Bi}_2\text{Te}_3$  NPs and employ stronger binders to make  $\text{Bi}_2\text{Te}_3$  NPs more strongly fixed on the surface of ITO-coated glass to avoid the slight peel-off during long-running measurements.

## Conclusions

In summary, topological insulator  $\text{Bi}_2\text{Te}_3$  NPs were successfully synthesized by a solvothermal approach and the size of  $\text{Bi}_2\text{Te}_3$  NPs can be readily controlled by simply tuning the reaction time. The hexagonal structure of  $\text{Bi}_2\text{Te}_3$  NPs was well-characterized, and UV-Vis spectra revealed a broadband absorption range from 260 nm to 1500 nm. The as-synthesized  $\text{Bi}_2\text{Te}_3$  NPs were, for the first time, employed as a working material in a PEC-type photodetector. The PEC result not only shows that the  $P_{\text{ph}}$  and  $R_{\text{ph}}$  significantly improve but also exhibits faster  $t_{\text{res}}$  and  $t_{\text{rec}}$ , compared to those of bismuthene-based or tellurene-based photodetectors. It was also shown that  $\text{Bi}_2\text{Te}_3$  NPs-2 displayed better PEC performance, attributed to the synergistic effect of optical absorbance and the number of accessible active sites on a  $\text{Bi}_2\text{Te}_3$  NP. In addition, good PEC stability of the  $\text{Bi}_2\text{Te}_3$  NP-based photodetector was obtained in 0.1 M KOH after one month without any protection, comparable to the bismuthene-based or tellurene-based photodetector. Because of the facile synthesis, easy size control of  $\text{Bi}_2\text{Te}_3$  NPs, excellent photoresponse performance, and good long-term stability of the  $\text{Bi}_2\text{Te}_3$  NP-based photodetector, we believe that  $\text{Bi}_2\text{Te}_3$  can pave a new way for the design of bismuthene or tellurene nano-material-based high-performance PEC-type devices with practical applicability.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

This research was supported by the National Natural Science Foundation of China (Grant Nos 61805147, 61435010, and 61675135), and the Science and Technology Innovation Commission of Shenzhen (Grant No. JCYJ20180305125141661, and KQTD2015032416270385).

## References

- 1 L. Wu, W. Huang, Y. Wang, J. Zhao, D. Ma, Y. Xiang, J. Li, J. S. Ponraj, S. C. Dhanabalan and H. Zhang, *Adv. Funct. Mater.*, 2019, **29**, 1806346.
- 2 L. Lu, Z. Liang, L. Wu, Y. Chen, Y. Song, S. C. Dhanabalan, J. S. Ponraj, B. Dong, Y. Xiang, F. Xing, D. Fan and H. Zhang, *Laser Photonics Rev.*, 2018, **12**, 1700221.
- 3 L. Lu, W. Wang, L. Wu, X. Jiang, Y. Xiang, J. Li, D. Fan and H. Zhang, *ACS Photonics*, 2017, **4**, 2852–2861.
- 4 L. Lu, X. Tang, R. Cao, L. Wu, Z. Li, G. Jing, B. Dong, S. Lu, Y. Li, Y. Xiang, J. Li, D. Fan and H. Zhang, *Adv. Optical Mater.*, 2017, **5**, 1700301.
- 5 S. Lin, W. Li, Z. Chen, J. Shen, B. Ge and Y. Pei, *Nat. Commun.*, 2016, **7**, 10287.
- 6 L. Yang, Z.-G. Chen, M. S. Dargusch and J. Zou, *Adv. Energy Mater.*, 2018, **8**, 1701797.
- 7 A. Boukai, K. Xu and J. R. Heath, *Adv. Mater.*, 2006, **18**, 864–869.

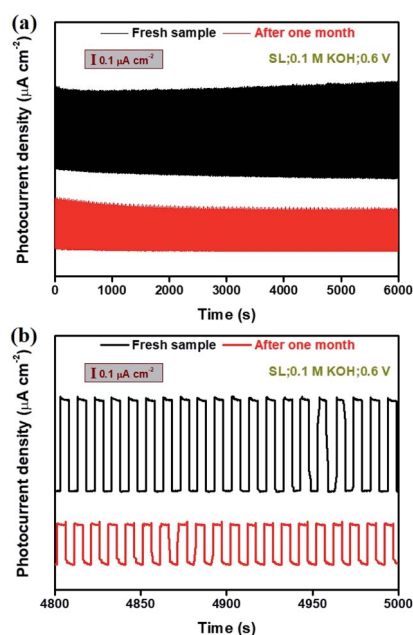


Fig. 6 Long-term stability of the photoresponse performance of the  $\text{Bi}_2\text{Te}_3$  NPs-1 under a SL in 0.1 M KOH (a) before and (b) after one month.





- 8 Y. Wang, G. Qiu, R. Wang, S. Huang, Q. Wang, Y. Liu, Y. Du, W. A. Goddard III, M. J. Kim, X. Xu, P. D. Ye and W. Wu, *Nat. Electron.*, 2018, **1**, 228–236.
- 9 W. Wu, G. Qiu, Y. Wang, R. Wang and P. Ye, *Chem. Soc. Rev.*, 2018, **47**, 7203–7212.
- 10 Z. Xie, C. Xing, W. Huang, T. Fan, Z. Li, J. Zhao, Y. Xiang, Z. Guo, J. Li, Z. Yang, B. Dong, J. Qu, D. Fan and H. Zhang, *Adv. Funct. Mater.*, 2018, **28**, 1705833.
- 11 C. Xing, W. Huang, Z. Xie, J. Zhao, D. Ma, T. Fan, W. Liang, Y. Ge, B. Dong, J. Li and H. Zhang, *ACS Photonics*, 2018, **5**, 621–629.
- 12 Y. L. Chen, J. G. Analytis, J.-H. Chu, Z. K. Liu, S.-K. Mo, X. L. Qi, H. J. Zhang, D. H. Lu, X. Dai, Z. Fang, S. C. Zhang, I. R. Fisher, Z. Hussain and Z.-X. Shen, *Science*, 2009, **325**, 178–181.
- 13 Z. Rashid, A. S. Nissimagoudar and W. Li, *Phys. Chem. Chem. Phys.*, 2019, **21**, 5679–5688.
- 14 R. Singh, M. D. Anoop, R. K. Rathore, A. S. Verma, K. Awasthi, V. K. Saraswat and M. Kumar, *Appl. Phys. A*, 2018, **124**, 558.
- 15 J. L. Zhang, S. J. Zhang, J. L. Zhu, Q. Q. Liu, X. C. Wang, C. Q. Jin and J. C. Yu, *Phys. B*, 2017, **521**, 13–16.
- 16 S. A. Sapp, B. B. Lakshmi and C. R. Martin, *Adv. Mater.*, 1999, **11**, 402–404.
- 17 D. Kong, W. Dang, J. J. Cha, H. Li, S. Meister, H. Peng, Z. Liu and Y. Cui, *Nano Lett.*, 2010, **10**, 2245–2250.
- 18 A. L. Prieto, M. S. Sander, M. S. Martín-González, R. Gronsky, T. Sands and A. M. Stacy, *J. Am. Chem. Soc.*, 2001, **123**, 7160–7161.
- 19 T. Sun, X. B. Zhao, T. J. Zhu and J. P. Tu, *Mater. Lett.*, 2006, **60**, 2534–2537.
- 20 H. Mamur, O. F. Dilmac, H. Korucu and M. R. A. Bhuiyan, *Micro Nano Lett.*, 2018, **13**, 1117–1120.
- 21 Y. Liu, Q. Wang, J. Pan, Y. Sun, L. Zhang and S. Song, *Chem.–Eur. J.*, 2018, **24**, 9765–9768.
- 22 G. Zhang, B. Kirk, L. A. Jauregui, H. Yang, X. Xu, Y. P. Chen and Y. Wu, *Nano Lett.*, 2012, **12**, 56–60.
- 23 B. Zhou, Y. Zhao, L. Pu and J.-J. Zhu, *Mater. Chem. Phys.*, 2006, **96**, 192–196.
- 24 Z. Alpichshev, J. G. Analytis, J.-H. Chu, I. R. Fisher, Y. L. Chen, Z. X. Shen, A. Fang and A. Kapitulnik, *Phys. Rev. Lett.*, 2010, **104**, 016401.
- 25 T. Zhang, P. Cheng, X. Chen, J.-F. Jia, X. Ma, K. He, L. Wang, H. Zhang, X. Dai, Z. Fang, X. Xie and Q.-K. Xue, *Phys. Rev. Lett.*, 2009, **103**, 266803.
- 26 J. Gooth, B. Hamdou, A. Dorn, R. Zierold and K. Nielsch, *Appl. Phys. Lett.*, 2014, **104**, 243115.
- 27 H. Qiao, J. Yuan, Z. Xu, C. Chen, S. Lin, Y. Wang, J. Song, Y. Liu, Q. Khan, H. Y. Hoh, C.-X. Pan, S. Li and Q. Bao, *ACS Nano*, 2015, **9**, 1886–1894.
- 28 A. Parbatani, E. S. Song, J. Claypoole and B. Yu, *Nanotechnology*, 2019, **30**, 165201.
- 29 A. Sharma, A. K. Srivastava, T. D. Senguttuvan and S. Husale, *Sci. Rep.*, 2017, **7**, 17911.
- 30 J. Liu, Y. Li, Y. Song, Y. Ma, Q. Chen, Z. Zhu, P. Lu and S. Wang, *Appl. Phys. Lett.*, 2017, **110**, 141109.
- 31 H. Liu and P. D. Ye, *Appl. Phys. Lett.*, 2011, **99**, 052108.
- 32 G. Xiao, C. Zhu, Y. Ma, B. Liu, G. Zou and B. Zou, *Angew. Chem., Int. Ed.*, 2014, **53**, 729–733.
- 33 R. Dey, T. Pramanik, A. Roy, A. Rai, S. Guchhait, S. Sonde, H. C. P. Movva, L. Colombo, L. F. Register and S. K. Banerjee, *Appl. Phys. Lett.*, 2014, **104**, 223111.
- 34 L. Yang, Z.-G. Chen, M. Hong, G. Han and J. Zou, *ACS Appl. Mater. Interfaces*, 2015, **7**, 23694–23699.
- 35 P. Yan, R. Lin, H. Chen, H. Zhang, A. Liu, H. Yang and S. Ruan, *IEEE Photonics Technol. Lett.*, 2015, **27**, 264.
- 36 H. Mu, Z. Wang, J. Yuan, S. Xiao, C. Chen, Y. Chen, Y. Chen, J. Song, Y. Wang, Y. Xue, H. Zhang and Q. Bao, *ACS Photonics*, 2015, **2**, 832–841.
- 37 Y. Chen, M. Wu, P. Tang, S. Chen, J. Du, G. Jiang, Y. Li, C. Zhao, H. Zhang and S. Wen, *Laser Phys. Lett.*, 2014, **11**, 055101.
- 38 J. Song, F. Xia, M. Zhao, Y. L. Zhong, W. Li, K. P. Loh, R. A. Caruso and Q. Bao, *Chem. Mater.*, 2015, **27**, 3471–3482.
- 39 G. Zhang, W. Wang, X. Lu and X. Li, *Cryst. Growth Des.*, 2009, **9**, 145–150.
- 40 B. K. Gupta, R. Sultana, S. Singh, V. Singh, G. Awana, A. Gupta, B. Singh, A. K. Srivastava, O. N. Srivastava, S. Auluck and V. P. S. Awana, *Sci. Rep.*, 2018, **8**, 9205.
- 41 M. Zhao, J. Zhang, N. Gao, P. Song, M. Bosman, B. Peng, B. Sun, C.-W. Qiu, Q.-H. Xu, Q. Bao and K. P. Loh, *Adv. Mater.*, 2016, **28**, 3138–3144.
- 42 P. Srivastava and K. Singh, *J. Exp. Nano.*, 2014, **9**, 1064–1074.
- 43 W. Huang, X. Jiang, Y. Wang, F. Zhang, Y. Ge, Y. Zhang, L. Wu, D. Ma, Z. Li, R. Wang, Z. N. Huang, X. Dai, Y. Xiang, J. Li and H. Zhang, *Nanoscale*, 2018, **10**, 20540–20547.
- 44 W. Huang, Z. Xie, T. Fan, J. Li, Y. Wang, L. Wu, D. Ma, Z. Li, Y. Ge, Z. N. Huang, X. Dai, Y. Xiang, J. Li, X. Zhu and H. Zhang, *J. Mater. Chem. C*, 2018, **6**, 9582–9593.
- 45 W. Huang, C. Xing, Y. Wang, Z. Li, L. Wu, D. Ma, X. Dai, Y. Xiang, J. Li, D. Fan and H. Zhang, *Nanoscale*, 2018, **10**, 2404–2412.
- 46 C.-L. Hsu, Y.-D. Gao, Y.-S. Chen and T.-J. Hsueh, *ACS Appl. Mater. Interfaces*, 2014, **6**, 4277–4285.
- 47 X. Wang, Y. Zhang, X. Chen, M. He, C. Liu, Y. Yin, X. Zou and S. Li, *Nanoscale*, 2014, **6**, 12009–12017.
- 48 W. Huang, Y. Zhang, Q. You, P. Huang, Y. Wang, Z. N. Huang, Y. Ge, L. Wu, Z. Dong, X. Dai, Y. Xiang, J. Li, X. Zhang and H. Zhang, *Small*, 2019, **15**, 1900902.
- 49 X. Ren, Z. Li, Z. Huang, D. Sang, H. Qiao, X. Qi, J. Li, J. Zhong and H. Zhang, *Adv. Funct. Mater.*, 2017, **27**, 1606834.
- 50 C. Backes, B. M. Szydłowska, A. Harvey, S. Yuan, V. Vega-Mayoral, B. R. Davies, P.-L. Zhao, D. Hanlon, E. J. G. Santos, M. I. Katsnelson, W. J. Blau, C. Gadermaier and J. N. Coleman, *ACS Nano*, 2016, **10**, 1589–1601.
- 51 Z. Li, H. Qiao, Z. Guo, X. Ren, Z. Huang, X. Qi, S. C. Dhanabalan, J. S. Ponraj, D. Zhang, J. Li, J. Zhao, J. Zhong and H. Zhang, *Adv. Funct. Mater.*, 2018, **28**, 1705237.

