

Cite this: *Nanoscale Adv.*, 2020, 2, 798

Understanding the role of zinc dithiocarbamate complexes as single source precursors to ZnS nanomaterials

Husn-Ubayda Islam,^{ab} Anna Roffey,^a Nathan Hollingsworth,^a Wim Bras,^{bc} Gopinathan Sankar,^a Nora H. De Leeuw^d and Graeme Hogarth^{de*}

Zinc sulfide is an important wide-band gap semi-conductor and dithiocarbamate complexes $[\text{Zn}(\text{S}_2\text{CNR}_2)_2]$ find widespread use as single-source precursors for the controlled synthesis of ZnS nanoparticulate modifications. Decomposition of $[\text{Zn}(\text{S}_2\text{CN}^i\text{Bu}_2)_2]$ in oleylamine gives high aspect ratio wurtzite nanowires, the average length of which was increased upon addition of thiuram disulfide to the decomposition mixture. To provide further insight into the decomposition process, X-ray absorption spectroscopy (XAS) of $[\text{Zn}(\text{S}_2\text{CNMe}_2)_2]$ was performed in the solid-state, in non-coordinating xylene and in oleylamine. In the solid-state, dimeric $[\text{Zn}(\text{S}_2\text{CNMe}_2)_2]_2$ was characterised in accord with the single crystal X-ray structure, while in xylene this breaks down into tetrahedral monomers. *In situ* XAS in oleylamine (RNH_2) shows that the coordination sphere is further modified, amine binding to give five-coordinate $[\text{Zn}(\text{S}_2\text{CNMe}_2)_2(\text{RNH}_2)]$. This species is stable to ca. 70 °C, above which amine dissociates and at ca. 90 °C decomposition occurs to generate ZnS. The relatively low temperature onset of nanoparticle formation is associated with amine-exchange leading to the *in situ* formation of $[\text{Zn}(\text{S}_2\text{CNMe}_2)(\text{S}_2\text{CNHR})]$ which has a low temperature decomposition pathway. Combining these observations with the previous work of others allows us to propose a detailed mechanistic scheme for the overall process.

Received 21st October 2019
Accepted 6th January 2020

DOI: 10.1039/c9na00665f

rsc.li/nanoscale-advances

Introduction

Zinc sulfide (ZnS) is a wide band-gap (3.7 eV) semiconductor that finds use in optoelectronic devices due to its unique photoluminescence and electroluminescence properties.¹ Nano-dimensional ZnS is especially important as both its optical and electronic properties can be tailored by controlling the size and shape of the nanostructures and over the past decade a range of different nanoparticulate forms of ZnS have been prepared through judicious choice of synthetic methods and control over the synthetic parameters,² however, many of these methods employ either toxic gases such as H_2S and H_2 and/or have to be carried out in an inert atmosphere. More synthetically appealing is the single source precursor (SSP) approach,³ especially when the precursors are air and moisture stable, since this allows control of nanoparticulate shape-size-structure

without the need for elaborate experimental facilities. Dithiocarbamate complexes are easily prepared from readily available starting materials⁴ and consequently find use as SSPs towards a range of metal-sulfides.^{5–10} Zinc dithiocarbamates, $[\text{Zn}(\text{S}_2\text{CNR}_2)_2]$, are particularly attractive SSPs since they are air and moisture stable solids which are easily prepared from the requisite secondary amine in water.⁴ They have been used extensively in the production of ZnS either as pure nanomaterials,^{10–19} quaternary zinc-containing sulfide nanomaterials²⁰ or as shells on quantum dots.²¹ In this contribution we detail our use of $[\text{Zn}(\text{S}_2\text{CNMe}_2)_2]$ (**1**) and $[\text{Zn}(\text{S}_2\text{CN}^i\text{Bu}_2)_2]$ (**2**) (Chart 1) as SSPs towards ultra-thin wurtzite nanowires and also probe mechanistic aspects of the precursor decomposition process by *in situ* X-ray absorption spectroscopy (XAS) studies of **1**. By piecing together our *in situ* findings with previous NMR studies^{22,23} we have put together a plausible reaction mechanism which will allow others to better rationalise their own findings and logically plan synthetic routes to ZnS nanomaterials.

Results and discussion

(i) Synthesis

Dithiocarbamate complexes $[\text{Zn}(\text{S}_2\text{CNMe}_2)_2]$ (**1**) and $[\text{Zn}(\text{S}_2\text{CN}^i\text{Bu}_2)_2]$ (**2**) were prepared by standard methods. Decomposition of SSPs is normally carried out in a high boiling point

^aDepartment of Chemistry, University College London, 20 Gordon Street, London WC1H 0AJ, UK

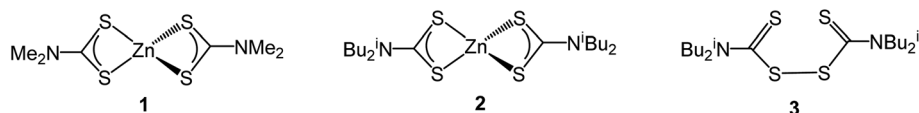
^bNetherlands Organisation for Scientific Research DUBBLE@ESRF, 38043 Grenoble, France

^cChemical Sciences Division, Oak Ridge National Laboratory, Oak Ridge, Tennessee 37831, USA

^dSchool of Chemistry, Cardiff University, Main Building, Park Place, Cardiff, CF10 3AT, UK

^eDepartment of Chemistry, King's College London, Britannia House, 7 Trinity Street, London SE1 1DB, UK. E-mail: graeme.hogarth@kcl.ac.uk





solvent which can act both as a heat sink and capping agent, with long chain primary amines such as oleylamine being widely utilised.²⁴ In order to ensure that our synthetic and *in situ* studies (see later) were carried out under similar conditions, we prepared ZnS from the decomposition of $[\text{Zn}(\text{S}_2\text{CN}^i\text{Bu}_2)_2]$ (**2**) in oleylamine, being chosen over **1** due to its greater solubility in this solvent. A 5 mM solution of **2** in oleylamine was heated to 230 °C and held for 1 h. Following this, the mixture was slowly cooled to room temperature and methanol added to precipitate the generated nanoparticles, which were isolated by centrifugation, washing being repeated three times and the material produced air dried. Solid samples had an off-white appearance and were characterised by powder X-ray diffraction (PXRD) as wurtzite (Fig. 1). Interestingly, the peak at 28.5° 2θ for the (002) lattice plane is more intense than in the reference pattern for bulk wurtzite (ICDD card no. 36-1450) which suggests preferential growth along the *c* axis.

Transmission Electron Microscopy (TEM) showed that high aspect ratio nanowires were produced (Fig. 2) with an average

particle length of 34.3 nm (SD 26.2 nm) and width of 3.3 nm (SD 0.8 nm) and HRTEM the shows *d*-spacings of 3.17 Å, corresponding to the (002) lattice plane in wurtzite (ICDD card no. 36-1450), being consistent with PXRD data. These dimensions are similar to the nanowires generated upon heating $[\text{Zn}(\text{S}_2\text{CNET}_2)_2]$ (**3**)¹¹ or $[\text{Zn}(\text{S}_2\text{CNBz}_2)_2]$ (**4**)¹² in oleylamine.

In recent work on the decomposition of $[\text{Fe}(\text{S}_2\text{CNR}_2)_3]$ we found that addition of thiuram disulfide, $(\text{R}_2\text{NCS}_2)_2$, led to the formation of different phases of iron sulfides.²⁵ As $[\text{Zn}(\text{S}_2\text{CNR}_2)_2]$ are known to add dithiocarbamate to generate the tris(dithiocarbamate) species, $[\text{Zn}(\text{S}_2\text{CNR}_2)_3]^-$,²⁶ then potentially thiuram disulfide addition may slow down the decomposition process (oleylamine provides a reducing environment). Decomposition of **2** in the presence of 10 mM of the thiuram disulfide **3** (Chart 1) also afforded wurtzite nanowires, being of the same width (3.2 nm ± 0.8 nm) but slightly longer (45.0 nm ± 25.1 nm).

The formation of wurtzite nanowires upon decomposition of zinc dithiocarbamate complexes has previously been reported.^{11–14} Thus Lieber and co-workers prepared wires with a diameter of *ca.* 18 nm upon decomposition of $[\text{Zn}(\text{S}_2\text{CNET}_2)_2]$ by a nanocluster-catalysed vapour-liquid-solid method,¹³ while Wang and co-workers prepared ultra-thin nanowires with diameters of *ca.* 3–5.5 nm from decomposition of the same precursor in oleylamine at 300 °C;¹¹ even thinner nanowires (*ca.* 2 nm) have been prepared from $[\text{Zn}(\text{S}_2\text{CN}^n\text{Bu}_2)_2]$ in dodecylamine at 280 °C.¹⁴

(ii) X-ray absorption spectroscopy (XAS) studies of $[\text{Zn}(\text{S}_2\text{CNMe}_2)_2]$ (**1**)

For the XAS studies we chose $[\text{Zn}(\text{S}_2\text{CNMe}_2)_2]$ (**1**) as its simple structure enables easier interpretation of data, whilst still rendering relevant information on the class of SSP. We sequentially investigated the structure of **1** in the solid-state, the non-coordinating (high boiling solvent) xylene and oleylamine. The results of these studies are summarised in Fig. 3 and 4.

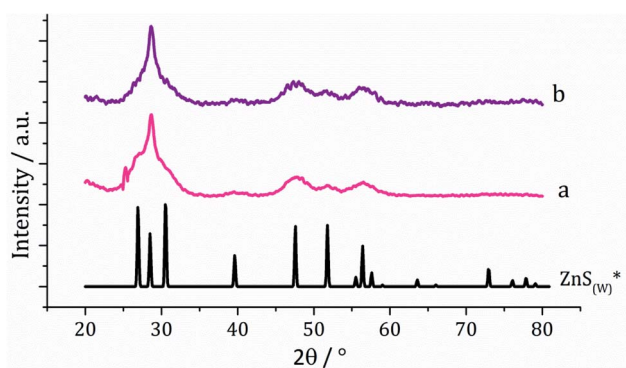


Fig. 1 PXRD patterns using Cu K_α radiation for samples prepared from (a) **2** and (b) **2** with added **3**, with reference pattern for bulk wurtzite (2H) (ICDD card no. 36-1450).

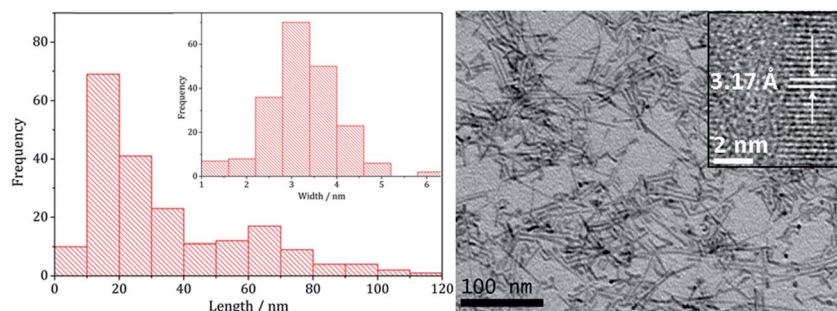


Fig. 2 Particle length histogram with width histogram insert (left) and TEM image with HRTEM insert (right), of the sample prepared from **2**.



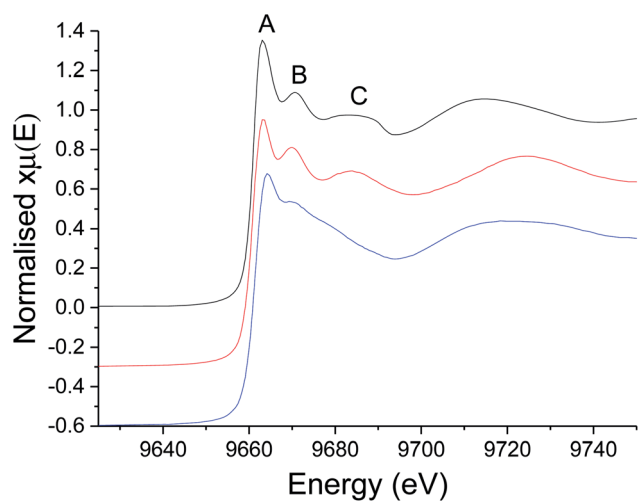


Fig. 3 XANES spectra of **1** as a solid (black), dissolved in xylene (red), and dissolved in oleylamine (blue). The relevance of peaks marked (A–C) is discussed in the text.

We first investigated **1** in the solid-state. While in the gas phase it is monomeric with four equivalent Zn–S distances of 2.28 Å,²⁷ a single crystal X-ray diffraction study shows it is dimeric in the solid-state containing two bridging dithiocarbamate ligands giving an overall boat-type conformation with non-degenerate Zn–S bond distances.²⁸ The XANES spectrum exhibits a straight edge jump followed by a multi-featured region including white line intensity **A**, and two further peaks **B** and peak **C**. The EXAFS of **1** can be modelled accurately to the crystal structure. Thus there are two Zn–S distances at 2.32 Å arising from bonding of the zinc centre to the terminal bidentate dithiocarbamate ligand; XRD shows these bonds to be 2.31 and 2.33 Å. There are also two Zn–S distances at 2.38 and 2.41 Å resulting from the bridging ligands; found at 2.37 and 2.43 Å in the crystal structure.²⁸ A fifth Zn–S distance is detected at 2.95 Å. This is a typical feature of dimeric zinc dithiocarbamates and arises in **1** at 3.03 Å.²⁸

Spectra of **1** in xylene, while comparable to that in the solid-state with similar features after the edge position, also show some significant differences; specifically the ratio between **A** and **B**, and the shape of **C** differ. Indeed, the EXAFS can be best

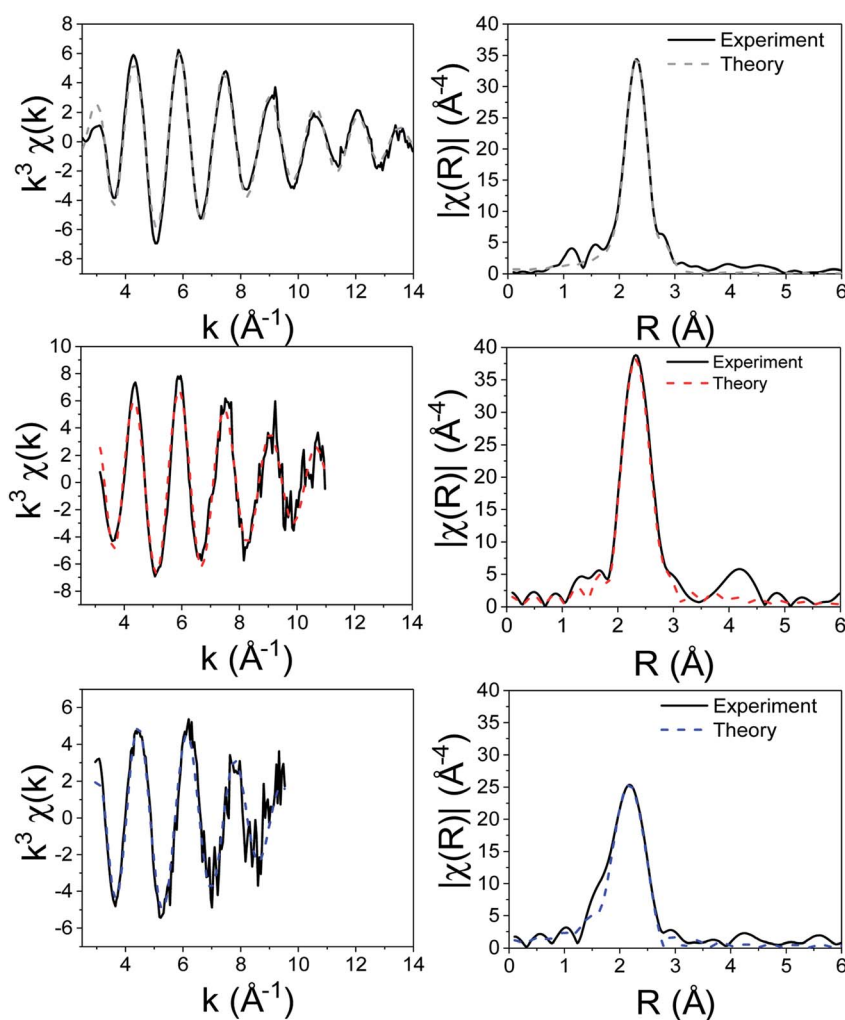


Fig. 4 k^3 -Weighted EXAFS and Fourier transformed (real space) data of **1** solid (top), in xylene (middle) and in oleylamine (bottom).



Table 1 EXAFS fittings for **1** as a solid and dissolved in xylene and oleylamine

<i>Ex situ</i> XAS						
Sample	Scatter	<i>N</i>	<i>R</i> _{XRD} (Å)	<i>R</i> _{EXAFS} (Å)	σ^2 (Å ²)	<i>F</i>
1 solid	S	2	2.31–2.33	2.32 ± 0.02	0.010	20
	S	1	2.37	2.38 ± 0.02	0.010	
	S	1	2.43	2.41 ± 0.02	0.010	
	C	1	2.77	2.74 ± 0.02	0.006	
	S	1	3.03	2.95 ± 0.03	0.015	
	C	1	3.02	3.00 ± 0.03	0.006	
1 in xylene	S	2	—	2.30 ± 0.03	0.007	25
	S	2	—	2.39 ± 0.02	0.007	
	C	2	—	2.73 ± 0.02	0.008	
1 in oleylamine	N	1	—	2.06 ± 0.02	0.020	27
	S	4	—	2.25 ± 0.03	0.020	

fit to a monomeric structure with two sets of Zn–S bond distances at 2.30 and 2.39 Å. The crystal structure of $[\text{Zn}(\text{S}_2\text{CNBz}_2)_2]$ shows a monomeric species containing two inequivalent Zn–S bond distances of 2.33 and 2.36 Å.²⁹ A further two Zn–C interatomic distances are found at 2.73 Å, being similar to the Zn–C distance of 2.74 Å in the solid-state.

The XANES spectrum of **1** in oleylamine is quite distinct from **1** in the solid-state or in xylene; peaks **B** and **C** are absent suggesting significant changes in the local structure. The EXAFS was also modified with oleylamine, showing a Zn–N interatomic distance of 2.06 Å, similar to other Zn–N literature values,^{30,31} suggesting that oleylamine is bound to the metal centre to give five-coordinate $[\text{Zn}(\text{S}_2\text{CNMe}_2)_2(\text{RNH}_2)]$ (**1.RNH₂**). The structure of the five-coordinated complex shows a Debye–Waller factor that is far higher than values found for **1** in the solid-state or dissolved in xylene. This is likely due to the dynamic nature of the five-coordinated complex which can interchange between trigonal bipyramidal and square-based pyramid geometries. Table 1 summarises the results of EXAFS modelling studies.

These studies show that the structure of **1** (Scheme 1) and analogues are sensitive to the nature of surrounding coordination environment. Most importantly they show that under the standard decomposition conditions, which take place in a coordinating solvent such as oleylamine, then it is actually a species of the form $[\text{Zn}(\text{S}_2\text{CNR}_2)_2(\text{RNH}_2)]$ which undergoes decomposition. Indeed, as early as the 1960s, the reversible 1 : 1 complexation of **1** with a range of amines was reported by Saville and co-workers^{32,33} and subsequently the crystal structures of $[\text{Zn}(\text{S}_2\text{CNET}_2)_2(\text{NH}^1\text{Bu}_2)]$,³¹ $[\text{Zn}(\text{S}_2\text{CNET}_2)_2(\text{NHC}_5\text{H}_{10})]$ ³⁴ and related five-coordinate complexes¹⁶ have been reported. We have no evidence of a six-coordinate complex even in neat

oleylamine and as far as we are aware, while six-coordinate adducts with aromatic diamines such as $[\text{Zn}(\text{S}_2\text{CNR}_2)_2(2,2'\text{-bipy})]$ and $[\text{Zn}(\text{S}_2\text{CNR}_2)_2(1,10\text{-phen})]$ ³⁵ are well-known, fully authenticated examples of six-coordinate $[\text{Zn}(\text{S}_2\text{CNR}_2)_2(\text{amine})_2]$ remain elusive.

(iii) *In situ* XAS studies of the decomposition of **1**

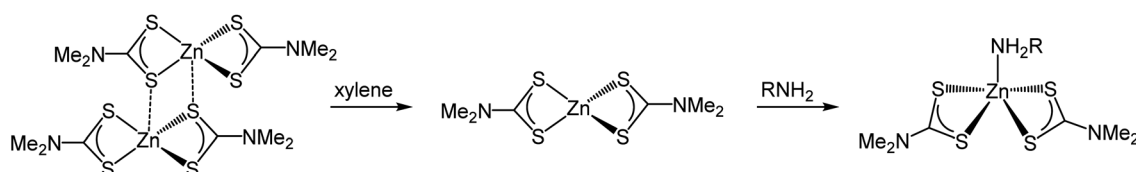
XAS spectra for the decomposition of **1** in oleylamine are shown in Fig. 5. XANES (Fig. 5a) spectra shows that $[\text{Zn}(\text{S}_2\text{CNMe}_2)_2(\text{RNH}_2)]$ (**1.RNH₂**) is stable up to 71 °C. Above this temperature, the peak previously labelled **B** decreases significantly along with a slight reduction in **C**. The edge position remains remarkably constant as a result of the stable oxidation state, while a slight drop in white line intensity may indicate a decrease in overall coordination.

EXAFS and FT data (Fig. 6) also suggest changes above *ca.* 71 °C (Table 2). Thus data best fits to 0.84 bound oleylamine ligands with an interatomic distance of 2.06 Å, in addition to four Zn–S bonds at 2.29 Å, and the disorder is slightly reduced. At 77 and 83 °C there are only 0.73 and 0.63 Zn–N bond distances respectively at 2.06 Å, and four Zn–S bond distances at 2.32 Å. By 89 °C, while the four Zn–S distances remain at 2.32 Å, zinc–oleylamine interactions are no longer detected, suggesting that coordination of the amine is lost. A further lengthening of the four Zn–S bond distances to 2.33 Å occurs at 94 °C the Zn–S bond distance of 2.33 Å (± 0.03 Å) being similar to those found in both wurtzite (2.334 Å) and sphalerite (2.347 Å).

Thus it appears that the decomposition takes place in two steps. Firstly, oleylamine dissociates from the zinc at *ca.* 70 °C, and following this decomposition occurs suddenly. In support of this low temperature transformation, Shen and co-workers have recently reported that decomposition of $[\text{Zn}(\text{S}_2\text{CNET}_2)_2]$ in BuNH_2 occurs at 60 °C, giving wurtzite nanosheets,¹⁸ although it is a slow process (*ca.* 10 h). Such low temperature decomposition (as discussed below) is associated with amine-exchange at the dithiocarbamate backbone.^{22,39}

(iv) Decomposition mechanism

Despite the widespread use of dithiocarbamate complexes as SSPs, relatively little is known regarding the decomposition mechanism(s).^{22,23,36–39} Here we first address the molecular transformations of the precursors. It is clear from our XAS measurements that dissolution of **1** leads to a breakdown of the dimeric solid-state structure to afford a tetrahedral monomer and in oleylamine this binds one equivalent of amine to afford 5-coordinate $[\text{Zn}(\text{S}_2\text{CNMe}_2)_2(\text{RNH}_2)]$ (**1.RNH₂**). As detailed above, Lee and co-workers studied the decomposition of **1** in

**Scheme 1** Changes to **1** upon dissolution in xylene and oleylamine.

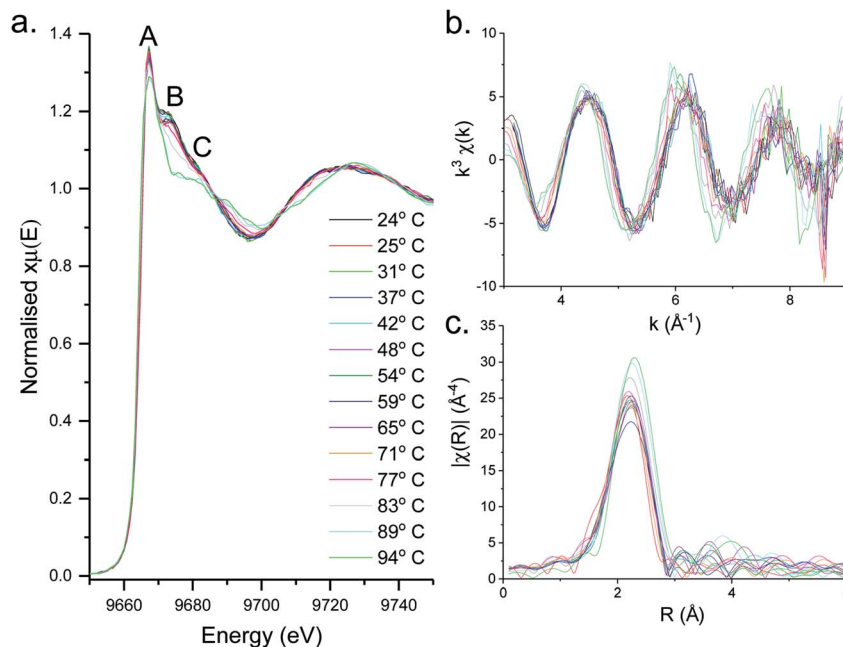


Fig. 5 *In situ* (a) XANES spectra, (b) EXAFS and (c) FT of the solvothermal decomposition of 1 in oleylamine up to 94 °C at a ramp rate of 1 °C min^{-1} . The relevance of the peaks marked (A–C) is discussed in the text.

octylamine by NMR spectroscopy²² and suggest that the key decomposition species is $[\text{Zn}(\text{S}_2\text{CNMe}_2)_2(\text{RNH}_2)_2]$ ($\text{R} = \text{octyl}$), in which free and bound octylamine are in rapid exchange. They isolated this as a waxy solid and studied its thermal decomposition in d^6 -dms $_o$ and d^8 -toluene by variable temperature NMR spectroscopy. We question assignment of this waxy solid as $[\text{Zn}(\text{S}_2\text{CNMe}_2)_2(\text{RNH}_2)_2]$ since we have no evidence of a bis(adduct) in our XAS work and favour formation of 5-coordinate $\mathbf{1.RNH}_2$ which at higher temperatures loses the amine; as might be expected on entropic grounds.

In Lee's experiments on the decomposition of putative $[\text{Zn}(\text{S}_2\text{CNMe}_2)_2(\text{RNH}_2)_2]$ zwitterionic complexes **X** and **Y**, resulting from nucleophilic addition of the primary amine to

the backbone carbon of the dithiocarbamate, are suggested to be observed by NMR spectroscopy (Scheme 2). From our work on the decomposition of $[\text{Ni}(\text{S}_2\text{CNBu}^i_2)_2]$ in oleylamine³⁹ we suggest that resonances assigned to **X** and **Y** are actually associated with the thioureas, $\text{Me}_2\text{NC}(\text{S})\text{NH}(\text{Oct})$ and $(\text{Oct})\text{NHC}(\text{S})\text{NH}(\text{Oct})$ respectively. This supposition is consistent with the work of Reedijk and co-workers who isolated thioureas, $\text{Me}_2\text{NC}(\text{S})\text{NH}(\text{Hex})$ and $(\text{Hex})\text{NHC}(\text{S})\text{NH}(\text{Hex})$, from the decomposition of **1** in hexylamine.²³ Nevertheless, the key premise of Lee and co-workers, namely that the amine attacks the backbone carbon and this leads to exchange NMe_2 for NHR , is in accord with the findings of Reedijk²² and our experiments on nickel.³⁹ Thus the key decomposition intermediates are

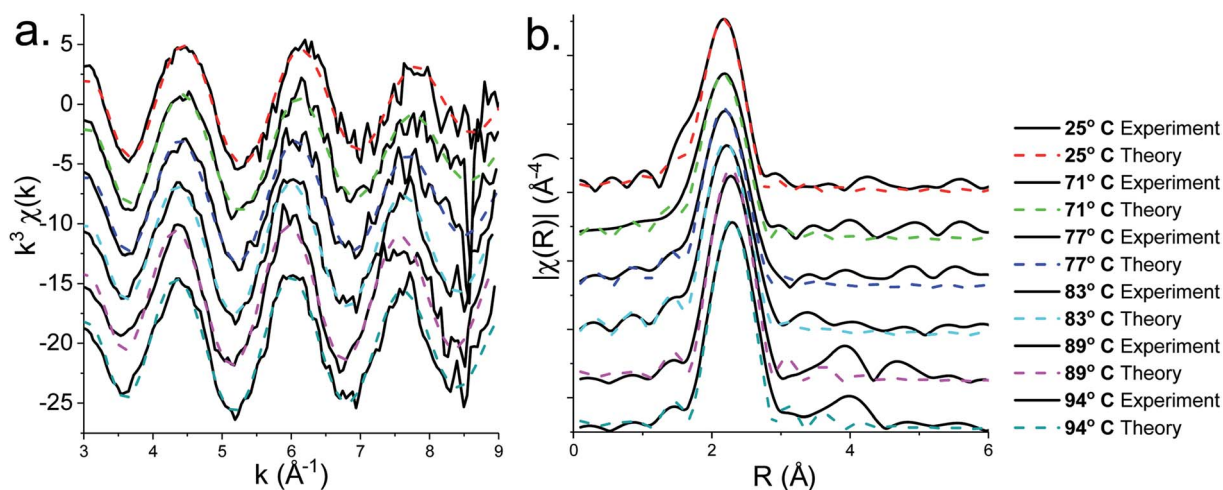


Fig. 6 (a) EXAFS and (b) FT fits for the solvothermal decomposition of 1 in oleylamine at 25 °C, 71 °C, 77 °C, 83 °C, 89 °C and 94 °C.



Table 2 EXAFS fitting for the *in situ* decomposition of 1 in oleylamine

Temperature (°C)	Scatter	<i>N</i>	<i>R</i> _{EXAFS} (Å)	2σ ² (Å ²)	<i>F</i>
71	N	0.84	2.06 ± 0.02	0.019	27
	S	4	2.29 ± 0.03	0.019	
77	N	0.73	2.06 ± 0.02	0.016	27
	S	4	2.30 ± 0.03	0.016	
83	N	0.63	2.06 ± 0.02	0.013	23
	S	4	2.32 ± 0.03	0.013	
89	S	4	2.32 ± 0.03	0.009	28
94	S	4	2.33 ± 0.03	0.009	26

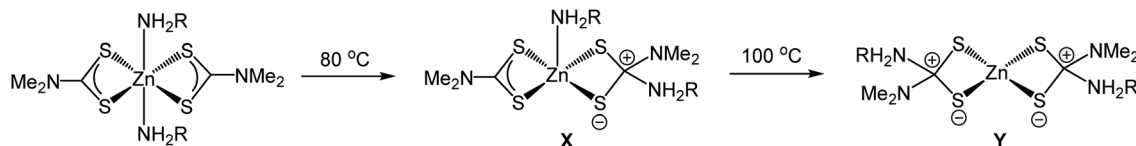
[Zn(S₂CNMe₂)(S₂CNHR)] and [Zn(S₂CNHR)₂] resulting from exchange of one or both NMe₂ groups. While the former have never been isolated, O'Brien and co-workers have prepared and crystallographically characterised [Zn(S₂CNHHex₂)₂] which decomposes to nanoparticulate zinc blende at 300 °C in hexadecylamine.¹⁹

Based on our current observations and those of others we propose a mechanism for the molecular transformations leading to the degradation of 1 (and other zinc dithiocarbamate complexes with secondary amine backbones) in solutions of primary amines (Scheme 3). Initial binding of the primary amine to zinc affords a 5-coordinate amine complex; which at higher temperatures (*ca.* 70 °C) is reversible. At *ca.* 90 °C, nucleophilic attack of primary amine at a backbone carbon leads to amine-exchange to afford [Zn(S₂CNMe₂)(S₂CNHR)] and (depending upon the rate of decomposition of this) possibly the bis(primary amine) derivatives [Zn(S₂CNHR)₂]. We have previously suggested that for related nickel complexes amine-exchange and decomposition are in competition but for zinc it appears that decomposition is rapid once a single amine-

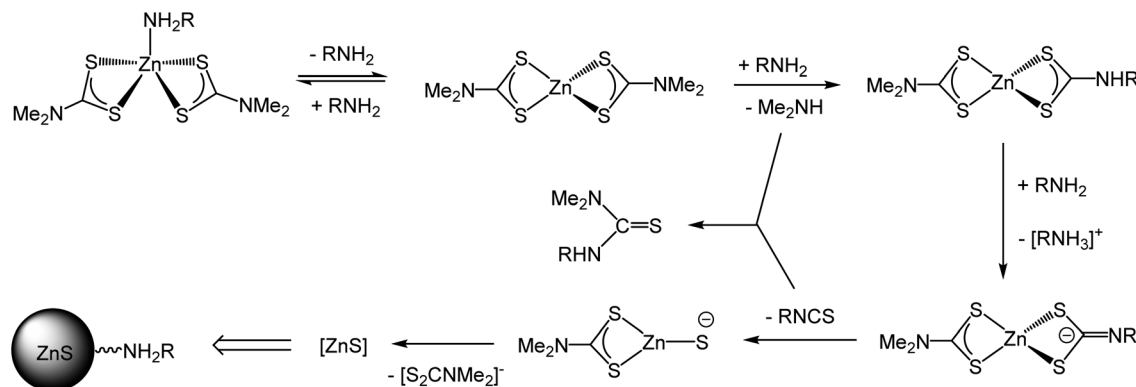
exchange has occurred.³⁹ Thus formation of [Zn(S₂CNMe₂)(S₂CNHR)] alone is enough to trigger ZnS formation; deprotonation of the dithiocarbamate leading to a zwitterionic dithiocarbamate which can extrude isothiocyanate.

(v) Nanoparticle morphology

A wide range of different nanoparticulate forms of ZnS have been prepared² and it is not obvious why decomposition of [Zn(S₂CNR₂)₂] in primary amines gives rise to high aspect nanowires. The amine (and indeed other solvents) plays multiple roles in the decomposition of SSPs including; interacting with molecular species, controlling the rate of growth and final particle size distribution, preventing particle aggregation and interacting with surface sites to passivate surface defects.⁴⁰ The XAS studies show that the amine binds to [Zn(S₂CNR₂)₂] precursors, and most likely all other molecular species, as Zn(II) is a good Lewis acid. Most likely the surface of the formed nanowires will also have close to 100% amine coverage as the small size of primary amines allows much higher surface coverage than other solvent-capping agents such as tri-*n*-octylphosphine oxide (TOPO).⁴¹ Three separate explanations have been put forward for the formation of high aspect ZnS nanowires from [Zn(S₂CNR₂)₂]/RNH₂ mixtures. Wang⁴¹ has proposed that nanowire formation is due to the selective absorption of the amine onto the (100) planes of the developing material, thus inhibiting growth in this dimension; supporting evidence coming from the formation of much wider wires upon addition of oleic acid to the decomposition mixture. Jia and Banin have attributed nanowire formation with the dissolution of monomers from the end facets of the initially formed nanoparticles, being followed by their secondary growth onto the side facets of the particles.¹² Certainly monomers on the end



Scheme 2 Lee's proposed intramolecular amine attack at the dithiocarbamate carbons.



Scheme 3 Proposed mechanism for the formation of ZnS from the thermal decomposition of 1 in primary amines.



of the wires will have lower coordination numbers than those in the bulk and hence might be expected to be more labile. Shen and co-workers offer a third view, namely that nanowires result from the loose aggregation of initially formed ellipse-like nanostructures,¹⁴ finding that formation of ultrathin nanowires is highly dependent upon the reaction medium; addition of PPh₃ to the amine giving longer nanowires, while in neat PPh₃ spherical nanoparticles result. We favour this explanation and note that in non-coordinating solvents, or in amine solutions at lower temperatures, the same precursors give spherical nanoparticles.^{15–17}

Summary and conclusions

The synthesis of ultrathin ZnS nanowires continues to attract considerable interest^{11–14,42,43} and the decomposition of zinc dithiocarbamate complexes in primary amines at temperatures of *ca.* 250 °C is proving to be a simple and valuable strategy to these materials.^{11–14} Here we add a new dithiocarbamate complex, [Zn(S₂CNⁱBu₂)₂] (2), to the available SSPs and show that its decomposition in oleylamine at 230 °C cleanly affords high aspect ratio wurtzite nanowires with an average particle length of 34.3 nm and width of 3.3 nm. Addition of the thiuram disulfide, (S₂CNⁱBu₂)₂ (3) results in a slight lengthening (*av.* 45.0 nm) of the wires, while the width remained unaffected. In order to gain further insight into the role of the amine in this process, *in situ* XAS studies were carried out on the simplest homologue, [Zn(S₂CNMe₂)₂] (1), showing that five-coordinate [Zn(S₂CNMe₂)₂(RNH₂)] (1.RNH₂) predominated in the amine solution, being stable up to 70 °C but decomposing above 90 °C. Combining our own observations with those of others^{22,23} has allowed us to piece together a molecular decomposition mechanism, the key feature of which is rate-determining amine-exchange at the backbone carbon of the dithiocarbamate to yield complexes [Zn(S₂CNR₂)(S₂CNHR)] and [Zn(S₂CNHR)₂] with one and two primary dithiocarbamate ligands respectively. Most likely once a single exchange has occurred decomposition, promoted *via* deprotonation of the ligand and extrusion of isothiocyanate, will be rapid. Thus onset of ZnS formation (*ca.* 90 °C) is far below that of the nanowire formation and this is likely a result of the aggregation of the initially formed spherical nanoparticles.

In previous work we have shown that the amine-exchange process is key to the facile decomposition of nickel SSPs, [Ni(S₂CNR₂)₂]³⁹ and others have also previously shown that the primary amine dithiocarbamate complexes, [M(S₂CNHR)_{*n*}], have significantly lower decomposition temperatures than their secondary amine counterparts.^{19,39} Thus, the rate of decomposition of different [Zn(S₂CNR₂)₂] SSPs in primary amines most likely depends upon their relative rates of secondary for primary amine exchange, which relates to both the electrophilicity of the backbone carbon and (likely most importantly) to the steric size of the secondary amine substituents as in the transition state the backbone carbon becomes sp³-hybridised.³⁹ Thus the primary amine plays multiple roles in the decomposition process; (i) solubilises the SSP, (ii) acts as a heat sink, (iii) coordinates to the Lewis acid Zn(II) in all molecular species, (iv)

acts as a nucleophile resulting in amine exchange, (v) acts as a base to deprotonate the primary amine dithiocarbamate, (vi) drives the reaction by combining with the extruded isothiocyanate and (vii) acts as a capping agent for the generated nanomaterials.

From this work and our previous work on nickel³⁹ it is clear that it is not the secondary amine dithiocarbamate complexes [M(S₂CNR₂)_{*n*}] that are the actual precursors to nanoparticulate metal sulfides but rather these are precursors to species with primary amine dithiocarbamate ligands. Indeed there is mounting evidence that this is the case⁴⁴ and given this then further study on the synthesis, stability and decomposition of primary amine dithiocarbamate complexes [M(S₂CNHR)_{*n*}] is warranted, and this is a direction of our current research. In this context an interesting recent development from Prasad and co-workers should be noted.⁴⁵ They generated zwitterionic [OleylNH₃][OleylNHCS₂] *in situ* upon addition of carbon disulfide to oleylamine solutions, it reacting rapidly with the metal salt to generate *in situ* primary amine dithiocarbamate complexes, [M(S₂CNHOLEY)_{*n*}], which decompose readily and cleanly to nanoparticulate metal sulfides.

Experimental

All ¹H and ¹³C{¹H} NMR spectra were obtained on either a Bruker Avance III 400 or Avance 600 spectrometer, the latter being equipped with a cryoprobe. All spectra were recorded using CDCl₃ which was dried and degassed over molecular sieves prior to use; ¹H and ¹³C{¹H} chemical shifts are reported relative to SiMe₄. The mass spectra were obtained using either Micromass 70-SE spectrometer using Electron Ionisation (EI) or a Thermo Finnigan MAT900xp spectrometer using Fast Atom Bombardment (FAB). Elemental analysis was carried using Elemental Analyser (CE-440) (Exeter Analytical Inc).

PXRD patterns were measured on a Bruker AXS D4 diffractometer using Cu Kα₁ radiation. The diffraction patterns obtained were compared to database standards. For TEM characterisation a 4 μL droplet of nanoparticle suspension (chloroform) was placed on a holey carbon-coated copper TEM grid and allowed to evaporate in air under ambient laboratory conditions for several minutes. TEM images were obtained using a JEOL-1010 microscope at 100 kV equipped with a Gatan digital camera. HRTEM measurements were collected using a Jeol 2100 TEM with a LaB₆ source operating at an acceleration voltage of 200 kV. Micrographs were taken on a Gatan Orius charge-coupled device (CCD).

Synthesis of precursors

[Zn(S₂CNMe₂)₂] (1)⁴⁶ and (S₂CNⁱBu₂)₂ (3)³⁹ were prepared by literature procedures. While [Zn(S₂CNⁱBu₂)₂] (2) is in the literature^{46,47} full synthetic details and characterisation data are not clearly presented. ⁱBu₂NH (2.62 mL, 15 mmol) was added to NaOH (0.60 g, 15 mmol) dissolved in water (100 mL). To this mixture CS₂ (0.90 mL, 15 mmol) was added dropwise over 10 minutes and the mixture stirred overnight. ZnSO₄·7H₂O (1.44 g, 5 mmol) was added yielding an off-white powder, which was



filtered, washed with water and dried. 2.09 g, 88%. Anal. calc. for $C_{18}H_{36}N_2S_4Zn$: C, 45.60; H, 7.65; N, 5.91. Found: C, 46.61; H, 7.97; N, 5.59. 1H NMR ($CDCl_3$): δ 0.97 (d, $J = 6.6$ Hz, 24H, CH_3), 2.39 (m, $J = 6.8$ Hz, 4H, CH), 3.70 (d, $J = 7.5$ Hz, 8H, CH_2). ^{13}C $\{^1H\}$ NMR ($CDCl_3$): 20.3 (CH_3), 27.1 (CH), 62.2 (CH_2), 204.5 (CS_2) ppm. MS: m/z 472 [M^+], 268 [$M^+ - S_2CN^iBu_2$], 383 [$ZnS(SCN^iBu_2)_2 - ^iBu$].

Decomposition studies

$[Zn(S_2CN^iBu_2)_2]$ (5 mM) was added to oleylamine (20 mL) in a three-neck round bottom flask attached to a condenser and evacuated and refilled with nitrogen repeatedly for 15 minutes. The solution was heated to 230 °C and held there for 1 h. The mixture was allowed to cool to room temperature slowly, whereupon methanol (80 mL) was added with stirring. The mixture was centrifuged and then the solution decanted leaving behind the resultant nanoparticles. This procedure was repeated three times and finally the material was allowed to dry in air.

XAS studies

XAS spectra were acquired at the iron K-edge (7112 eV) on the Dutch-Belgian EXAFS beamline, BM26A.⁴⁸ Monochromatic radiation was supplied by a double Si(111) crystal, ion chambers were used to measure incident and transmitted beam intensities (I_0 and I_t), and X-ray fluorescence was measured using a 9 element germanium solid state detector. Solid **1** was diluted with polyvinylpyrrolidone, pelletized and placed in the beam. Solutions of **1** were held within *in situ* liquid cells. The *in situ* liquid cell used was developed at UCL for synchrotron based experiments on liquid samples. Cartridge heaters embedded into the conductive cell body allow temperatures to reach up to 200 °C, subject to pressure buildup of the system. The Kapton sealed reaction chamber holds 400 μ L of solution with a fixed path length of 2 mm. The *in situ* liquid microtron cell was developed by Sample Environment at the ESRF for liquid experiments up to 260 °C. With quartz cell windows and a 4 mm path length, the cell was used for measurements in transmission. XAS data was normalized and background subtracted using Horae Athena software.⁴⁹ Linear combination analyses were also performed using Horae Athena. Detailed EXAFS analyses were performed on Excurve Version 9.273.⁵⁰

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

We thank the EPSRC for funding and beamline staff at DUBBLE, ESRF for their technical support and the Netherlands Organisation for Scientific Research (NWO) for making beam time available on the DUBBLE beamlines at the ESRF. WB's contribution is based upon work supported by Oak Ridge National Laboratory, managed by UT-Battelle, LLC, for the U.S. Department of Energy.

References

- 1 X. Fang, T. Zhai, U. K. Gautam, L. Li, L. Wu, Y. Bando and D. Golberg, *Prog. Mater. Sci.*, 2011, **56**, 175–287.
- 2 (a) X. Fang, L. Wu and L. Hu, *Adv. Mater.*, 2011, **23**, 585–598; (b) L.-W. Yin, Y. Bando, J.-H. Zhan, M.-S. Li and D. Golberg, *Adv. Mater.*, 2005, **17**, 1972–1977; (c) S. Hamad, S. M. Woodley and C. R. A. Catlow, *Mol. Simul.*, 2009, **35**, 1015–1032; (d) A. Tiwari and S. J. Dhoble, *Cryst. Growth Des.*, 2017, **17**, 381–407; (e) A. Tiwari and S. J. Dhoble, *RSC Adv.*, 2016, **6**, 64400–64420; (f) Z. Jiang, H. Sun, Z. Qin, X. Jiao and D. Chen, *Chem. Commun.*, 2012, **48**, 3620–3622; (g) W.-H. Zhang and W.-D. Zhang, *Mater. Lett.*, 2013, **98**, 5–7; (h) C. Feigl, S. P. Russo and A. S. Barnard, *J. Mater. Chem.*, 2010, **20**, 4971–4980.
- 3 (a) M.-R. Gao, J. Jiang and S.-H. Yu, *Small*, 2012, **8**, 13–27; (b) D. Fan, M. Afzaal, M. A. Mallik, C. Q. Nguyen, P. O'Brien and P. J. Thomas, *Coord. Chem. Rev.*, 2007, **251**, 1878–1888; (c) M. Afzaal, M. A. Malik and P. O'Brien, *J. Mater. Chem.*, 2010, **20**, 4031–4040; (d) L. G. Bloor, C. J. Carmalt and D. J. Pugh, *Coord. Chem. Rev.*, 2011, **255**, 1293–1318; (e) J. van Embden, A. S. R. Chesman and J. J. Jasieniak, *Chem. Mater.*, 2015, **27**, 2246–2285; (f) K. Ramasamy, M. A. Malik, N. Revaprasadu and P. O'Brien, *Chem. Mater.*, 2013, **25**, 3551–3569; (g) N. L. Pickett and P. O'Brien, *Chem. Rec.*, 2001, **1**, 467–479; (h) P. O'Brien, M. A. Malik, M. Chunggaze, T. Trindade, J. R. Walsh and A. C. Jones, *J. Cryst. Growth*, 1997, **170**, 23–29; (i) M. A. Azad, M. Afzaal and P. O'Brien, *Chem. Rev.*, 2010, **110**, 4417–4446; (j) N. Pradhan, B. Katz and S. Efrima, *J. Phys. Chem. B*, 2003, **107**, 13843–13854; (k) M. Lazell, P. O'Brien, D. J. Otway and J.-H. Park, *Dalton Trans.*, 2000, 4479–4486.
- 4 G. Hogarth, *Prog. Inorg. Chem.*, 2005, **53**, 71–561.
- 5 (a) R. Nomura, K. Miyawaki, T. Toyosaki and H. Matsuda, *Chem. Vap. Deposition*, 1996, **2**, 174–179; (b) M. Kemmler, M. Lazell, P. O'Brien, D. J. Otway, J.-H. Park and J. R. Walsh, *J. Mater. Sci.: Mater. Electron.*, 2002, **13**, 531–535; (c) M. J. Almond, H. Redman and D. A. Rice, *J. Mater. Chem.*, 2000, **10**, 2842–2846; (d) A. Birri, B. Harvey, G. Hogarth, E. Subasi and F. Ugur, *J. Organomet. Chem.*, 2007, **692**, 2448–2455; (e) F. Srouji, M. Afzaal, J. Waters and P. O'Brien, *Chem. Vap. Deposition*, 2005, **11**, 91–94; (f) P. O'Brien, D. J. Otway and J.-H. Park, *Mater. Res. Soc. Symp. Proc.*, 2000, **606**, 133–138; (g) O. C. Monteiro, H. I. S. Nogueira and T. Trindade, *Chem. Mater.*, 2001, **13**, 2103–2111; (h) P. O'Brien, D. J. Otway and J. R. Walsh, *Thin Solid Films*, 1998, **315**, 57–61.
- 6 (a) D. M. Frifo, O. F. Z. Khan and P. O'Brien, *J. Cryst. Growth*, 1989, **96**, 989–992; (b) J. Wang, J. R. Hyde, J. W. Wilson, K. Mallik, P. J. Sazio, P. O'Brien, M. A. Malik, M. Afzaal, C. Q. Nguyen, M. W. George, S. M. Howdle and D. C. Smith, *Adv. Mater.*, 2009, **21**, 4115–4119; (c) B. Ludolph, M. A. Malik, P. O'Brien and N. Revaprasadu, *Chem. Commun.*, 1998, 1849–1850.
- 7 (a) W. Han and M. Y. Gao, *Cryst. Growth Des.*, 2008, **8**, 1023–1030; (b) Y. Zhang, Y. Du and Q. Wang, *CrystEngComm*, 2010,



- 12, 3658–3663; (c) M. Akhtar, J. Akhter, M. A. Malik, P. O'Brien, F. Tuna, J. Raftery and M. Helliwell, *J. Mater. Chem.*, 2011, **21**, 9737–9745; (d) M. Akhtar, A. L. Abdelhady, M. A. Malik and P. O'Brien, *J. Cryst. Growth*, 2012, **346**, 106–112.
- 8 (a) A. Adeogun, M. Afzaal and P. O'Brien, *Chem. Vap. Deposition*, 2006, **12**, 597–599; (b) Y. Lou, A. C. S. Samia, J. Cowen, K. Banger, X. Chen, H. Lee and C. Burda, *Phys. Chem. Chem. Phys.*, 2003, **5**, 1091–1095; (c) K. Ramasamy, V. L. Kuznetsov, K. Gopal, M. A. Malik, J. Raftery, P. P. Edwards and P. O'Brien, *Chem. Mater.*, 2013, **25**, 266–276; (d) M. Afzaal, C. L. Rosenberg, M. A. Malik, A. J. P. White and P. O'Brien, *New J. Chem.*, 2011, **35**, 2773–2780; (e) E. P. C. Higgins, S. G. McAdams, D. G. Hopkinson, C. Byrne, A. S. Walton, D. J. Lewis and R. A. W. Dryfe, *Chem. Mater.*, 2019, **31**, 5384–5391; (f) N. Zeng, D. G. Hopkinson, B. F. Spencer, S. G. McAdams, A. A. Tedstone, S. J. Haigh and D. J. Lewis, *Chem. Commun.*, 2019, **55**, 99–102.
- 9 (a) P. O'Brien and J. Waters, *Chem. Vap. Deposition*, 2006, **12**, 620–626; (b) H. Cui, R. D. Pike, R. Kershaw, K. Dwight and A. Wold, *J. Solid State Chem.*, 1992, **101**, 115–118; (c) G. H. Singhal, R. I. Botto, L. D. Brown and K. S. Colle, *J. Solid State Chem.*, 1994, **109**, 166–171; (d) X. Chen, Z. Wang, X. Wang, J. Wan, J. Liu and Y. Qian, *Chem. Lett.*, 2004, **33**, 1294–1295; (e) R. Chauhan, M. Trivedi, J. Singh, K. C. Molloy, G. Kociok-Köhn, U. P. Mulik, D. P. Amalnerkar and A. Kumar, *Inorg. Chim. Acta*, 2014, **415**, 69–74.
- 10 (a) M. A. Malik, N. Revaprasadu and P. O'Brien, *Chem. Mater.*, 2001, **13**, 913–920; (b) T. Trindade, P. O'Brien and X.-M. Zhang, *Chem. Mater.*, 1997, **9**, 523–530; (c) D. C. Onwudiwe and P. A. Ajibade, *Int. J. Mol. Sci.*, 2011, **12**, 5538–5551; (d) D. C. Onwudiwe and P. A. Ajibade, *Mater. Lett.*, 2011, **65**, 3258–3261; (e) P. A. Ajibade, D. C. Onwudiwe and M. J. Moloto, *Polyhedron*, 2011, **30**, 246–252; (f) E. Y. M. Lee, N. H. Tran and R. N. Lamb, *Appl. Surf. Sci.*, 2005, **241**, 493–496; (g) M. Motevalli, P. O'Brien, J. R. Walsh and I. M. Watson, *Polyhedron*, 1996, **15**, 2801–2808; (h) P. O'Brien, J. R. Walsh, I. M. Watson, L. Hart and S. R. P. Silva, *J. Cryst. Growth*, 1996, **167**, 133–142; (i) R. Nomura, T. Murai, T. Topyosaki and H. Matsuda, *Thin Solid Films*, 1995, **271**, 4–7; (j) P. O'Brien, J. R. Walsh, I. M. Watson, M. Motevalli and L. Henrikson, *J. Chem. Soc., Dalton Trans.*, 1996, 2491–2496; (k) D. Zeng, M. J. Hampden-Smith, T. M. Alam and A. L. Rheingold, *Polyhedron*, 1994, **13**, 2715–2730; (l) R. D. Pike, H. Cui, R. Kershaw, K. Dwight, A. Wold, T. N. Blanton, A. A. Wernberg and H. J. Gysling, *Thin Solid Films*, 1992, **224**, 221–226.
- 11 Y. Zhang, H. Xu and Q. Wang, *Chem. Commun.*, 2010, **46**, 8941–8943.
- 12 G. Jia and U. Banin, *J. Am. Chem. Soc.*, 2014, **136**, 11121–11127.
- 13 C. J. Barrelet, Y. Wu, D. C. Bell and C. M. Lieber, *J. Am. Chem. Soc.*, 2003, **125**, 11498–11499.
- 14 G. Zhu, S. Zhang, Z. Xu, J. Ma and X. Shen, *J. Am. Chem. Soc.*, 2011, **133**, 15605–15612.
- 15 L. D. Nyamen, A. A. Nejo, V. S. R. Pullabhotla, P. T. Ndifon, M. A. Malik, J. Akhtar, P. O'Brien and N. Revaprasadu, *Polyhedron*, 2014, **67**, 129–135.
- 16 B. A. Prakasam, M. Lahtinen, A. Peuronen, M. Muruganandham, E. Kolehmainen, E. Haapaniemi and M. Sillanpää, *Mater. Lett.*, 2015, **144**, 19–21.
- 17 (a) M. Hrubaru, D. C. Onwudiwe and E. Hosten, *J. Sulfur Chem.*, 2016, **37**, 37–47; (b) P. A. Ajibade, D. C. Onwudiwe and M. J. Moloto, *Polyhedron*, 2011, **30**, 248–252; (c) D. C. Onwudiwe and P. A. Ajibade, *Mater. Lett.*, 2011, **65**, 3258–3261.
- 18 G. Zhu, J. Yang, C. Bao, X. Zhang, Z. Ji, S. Wu and X. Shen, *J. Colloid Interface Sci.*, 2016, **468**, 138–144.
- 19 A. A. Memon, M. Afzaal, M. A. Malik, C. Q. Nguyen, P. O'Brien and J. Raftery, *Dalton Trans.*, 2006, 4499–4505.
- 20 (a) K. Ramasamy, M. A. Malik and P. O'Brien, *Chem. Commun.*, 2012, **48**, 5703–5714; (b) K. Ramasamy, M. A. Malik and P. O'Brien, *Chem. Sci.*, 2011, **2**, 1170–1172; (c) O. Stroyuk, A. Raevskaya, O. Selyshchev, V. Dzhagen, N. Gaponik, D. R. T. Zahn and A. Eychmüller, *Sci. Rep.*, 2018, **8**, 1–10; (d) A. Khare, A. W. Wills, L. M. Ammerman, D. J. Norris and E. S. Aydil, *Chem. Commun.*, 2011, **47**, 11721–11723; (e) M. D. Regulacio, C. Ye, S. H. Lim, M. Bosman, E. Ye, S. Chen, Q.-H. Xu and M.-Y. Han, *Chem.-Eur. J.*, 2012, **18**, 3127–3131; (f) B. D. Chernomordik, A. E. Beland, D. D. Deng, L. F. Francis and S. Eray, *Chem. Mater.*, 2014, **26**, 3191–3201; (g) A. Roffey, N. Hollingsworth and G. Hogarth, *Nanoscale Adv.*, 2019, **1**, 3056–3066; (h) B. D. Chernomordik, A. E. Béland, N. D. Trejo, A. A. Gunawan, D. D. Deng, K. A. Mkhoyan and E. S. Aydil, *J. Mater. Chem. A*, 2014, **2**, 10389–10395; (i) H. Zheng, X. Li, K. Zong, Y. Sun, J. Liu, H. Wang and H. Yan, *Mater. Lett.*, 2013, **110**, 1–3.
- 21 (a) J. R. Dethlefsen and A. Døssing, *Nano Lett.*, 2011, **11**, 1964–1969; (b) A. Polovitsyn, Z. Dang, J. L. Movilla, B. Martín-García, A. H. Khan, G. H. V. Bertrand, R. Brescia and I. Moreels, *Chem. Mater.*, 2017, **29**, 5671–5680; (c) H.-J. Pan, C.-W. Lai, S.-W. Chou and P.-T. Chou, *Mater. Express*, 2012, **2**, 224–232; (d) L. Xi, D.-Y. Cho, M. Duchamp, C. B. Boothroyd, J. Y. Lek, A. Besmehn, R. Waser, Y. M. Lam and B. Kardynal, *ACS Appl. Mater. Interfaces*, 2014, **6**, 18233–18242; (e) J. C. Bear, N. Hollingsworth, P. D. McNaughten, A. G. Mayes, M. B. Ward, T. Nann, G. Hogarth and I. P. Parkin, *Angew. Chem., Int. Ed.*, 2014, **53**, 1598–1601; (f) J. Ziegler, S. Xu, E. Kucur, F. Meister, M. Batentschuk, F. Gindele and T. Nann, *Adv. Mater.*, 2008, **20**, 4068–4073; (g) S. Xu, J. Ziegler and T. Nann, *J. Mater. Chem.*, 2008, **18**, 2653–2656; (h) X. Yang, D. Zhao, K. S. Leck, S. T. Tan, Y. X. Tang, J. Zhao, H. V. Demir and X. W. Sun, *Adv. Mater.*, 2012, **24**, 4180–4185; (i) D. Chen, F. Zhao, H. Qi, M. Rutherford and X. Peng, *Chem. Mater.*, 2010, **22**, 1437–1444; (j) W. Zhang, G. Chen, J. Wang, B.-C. Ye and X. Zhong, *Inorg. Chem.*, 2009, **48**, 9723–9731; (k) H. Wang, H. Nakamura, M. Ushara, Y. Yamaguchi, M. Miyazaki and



- H. Maeda, *Adv. Funct. Mater.*, 2005, **15**, 603–608; (l) S. Shen, Y. Zhang, L. Peng, Y. Du and Q. Wang, *Angew. Chem., Int. Ed.*, 2011, **50**, 7115–7118; (m) M. A. Malik, P. O'Brien and N. Revaprasadu, *Chem. Mater.*, 2002, **14**, 2004–2010.
- 22 Y. K. Jung, J. I. Kim and J. K. Lee, *J. Am. Chem. Soc.*, 2010, **132**, 178–184.
- 23 A. Dirksen, P. J. Nieuwenhuizen, M. Hoogenraad, J. G. Haasnoot and J. Reedijk, *J. Appl. Polym. Sci.*, 2001, **79**, 1074–1083.
- 24 S. Mourdikoudis and L. M. Liz-Marzán, *Chem. Mater.*, 2013, **25**, 1465–1476.
- 25 A. Roffey, N. Hollingsworth, H.-U. Islam, W. Bras, G. Sankar, N. H. de Leeuw and G. Hogarth, *Nanoscale Adv.*, 2019, **1**, 2965–2978.
- 26 (a) C. C. Ashworth, N. A. Bailey, M. Johnson, J. A. McCleverty, N. Morrison and B. Tabbiner, *J. Chem. Soc., Chem. Commun.*, 1976, 743–744; (b) J. A. McCleverty, N. Spencer, N. A. Bailey and S. L. Shackleton, *J. Chem. Soc., Dalton Trans.*, 1980, 1939–1944.
- 27 K. Hagen, C. J. Holwill and D. A. Rice, *Inorg. Chem.*, 1989, **28**, 3239–3242.
- 28 H. Klug, *Acta Crystallogr.*, 1966, **21**, 536–546.
- 29 A. Decken, R. A. Gossage, M. Y. Chan, C. Lai and E. R. T. Tiekink, *Appl. Organomet. Chem.*, 2004, **18**, 101–102.
- 30 A. M. Malik, M. Motavalli and P. O'Brien, *Polyhedron*, 1999, **18**, 1259–1262.
- 31 A. V. Ivanov, A. S. Zaeva, E. V. Novikova, A. V. Gerasimenko and W. Forsling, *Russ. J. Coord. Chem.*, 2007, **33**, 233–243.
- 32 G. M. C. Higgins and B. Saville, *J. Chem. Soc.*, 1963, 2812–2817.
- 33 E. Coates, B. Rigg, B. Saville and D. Skelton, *J. Chem. Soc.*, 1965, 5613–5620.
- 34 A. V. Ivanov, A. S. Zaeva, A. V. Gerasimenko and T. A. Rodina, *Russ. J. Coord. Chem.*, 2008, **34**, 688–698.
- 35 (a) A. Manohar, V. Venkatachalam, K. Ramalingham, S. Thirumaran, G. Bocelli and A. Cantoni, *J. Chem. Crystallogr.*, 1998, **28**, 861–866; (b) S. Thirumaran, K. Ramalingam, G. Bocelli and A. Cantoni, *Polyhedron*, 1999, **18**, 925–930; (c) R. F. Klevtsova, L. A. Glinskaya, S. M. Zemskova and S. V. Larionov, *Polyhedron*, 1999, **18**, 3559–3565.
- 36 (a) G. Xie, Z.-P. Qiao, M.-H. Zeng, X.-M. Chen and S.-L. Gao, *Cryst. Growth Des.*, 2004, **4**, 513–516; (b) M. Green, L. Sandiford, K. M. Anderson and Y. Ma, *ChemPlusChem*, 2012, **77**, 192–195; (c) G. Barone, T. Chaplin, T. G. Hibbert, A. T. Kana, M. F. Mahon, K. C. Molloy, I. D. Worsley, I. P. Parkin and S. L. Price, *J. Chem. Soc., Dalton Trans.*, 2002, 1085–1092; (d) M. Chunggaze, J. McAlesse, P. O'Brien and D. J. Otway, *J. Chem. Soc., Chem. Commun.*, 1998, 833–834; (e) M. Chunggaze, M. A. Malik and P. O'Brien, *J. Mater. Chem.*, 1999, **9**, 2433–2437.
- 37 L. H. van Poppel, T. L. Groy and M. T. Caudle, *Inorg. Chem.*, 2004, **43**, 3180.
- 38 A. Roffey, N. Hollingsworth, H.-U. Islam, M. Mercy, C. R. A. Catlow, G. Sankar, N. H. de Leeuw and G. Hogarth, *Nanoscale*, 2016, **8**, 11067–11075.
- 39 N. Hollingsworth, A. Roffey, H.-U. Islam, M. Mercy, A. Roldan, W. Bras, M. Wolthers, C. R. A. Catlow, G. Sankar, G. Hogarth and N. H. de Leeuw, *Chem. Mater.*, 2014, **26**, 6281–6292.
- 40 C. Bullen and P. Mulvaney, *Langmuir*, 2006, **22**, 3007–3013.
- 41 M. Green, *J. Mater. Chem.*, 2010, **20**, 5797–5809.
- 42 S. Kumar, A. Sinhamahapatra, N. Sutradhar, H. C. Bajaj and A. B. Panda, *Chem. Commun.*, 2012, **48**, 850–852.
- 43 (a) Z. Deng, H. Yan and Y. Liu, *Angew. Chem., Int. Ed.*, 2010, **49**, 8695–8698; (b) S. K. Pahari, A. Sinhamahapatra, N. Sutradhar, H. C. Bajaj and A. B. Panda, *Chem. Commun.*, 2012, **48**, 850–852; (c) Y. Jaing, X. M. Meng, J. Liu, Z. R. Hong, C. S. Lee and S. T. Lee, *Adv. Mater.*, 2003, **15**, 1195–1198; (d) X. S. Fang, Y. Bando, G. Z. Shen, C. H. Ye, U. K. Gautam, P. M. F. J. Cosat, C. Y. Zhi, C. C. Tang and D. Golberg, *Adv. Mater.*, 2007, **19**, 2593–2596.
- 44 (a) N. Pradhan, B. Katz and S. Efrima, *J. Phys. Chem. B*, 2003, **107**, 13843–13854; (b) L. van Poppel, T. L. Groy and M. T. Caudle, *Inorg. Chem.*, 2004, **43**, 3180–3188; (c) A. W. Wills, M. S. Kang, A. Khare, W. L. Gladfelter and D. J. Norris, *ACS Nano*, 2010, **4**, 4523–4530; (d) J. J. Grenland, C. Lin, K. Gong, D. F. Kelley and A. M. Kelley, *J. Phys. Chem. C*, 2017, **121**, 7056–7061; (e) A. M. Munro, C. Chandler, M. Garling, D. Chai, V. Popovich, L. Lystrom and S. Kilina, *J. Phys. Chem. C*, 2016, **120**, 29455–29462.
- 45 (a) A. Bera, B. Mandal, P. N. Goswami, A. K. Rath and B. L. V. Prasad, *Langmuir*, 2018, **34**, 5788–5797; (b) A. Bera, B. Busupalli and B. L. V. Prasad, *ACS Sustainable Chem. Eng.*, 2018, **6**, 12006–12016.
- 46 A. M. Bond and A. F. Hollenkamp, *Inorg. Chem.*, 1990, **29**, 284–289.
- 47 (a) D. Oktavec, E. Beinrohr, B. Siles, J. Stefanec and J. Garaj, *Collect. Czech. Chem. Commun.*, 1980, **45**, 1495–1501; (b) M. Tarique, *E-J. Chem.*, 2011, **8**, 2020–2023.
- 48 S. Nikitenko, A. M. Beale, A. M. J. van der Eerden, S. D. M. Jacques, O. Leynaud, M. G. O'Brien, D. Detollenaere, R. Kaptein, B. M. Weckhuysen and W. Bras, *J. Synchrotron Radiat.*, 2008, **15**, 632–640.
- 49 B. Ravel and M. Newville, *J. Synchrotron Radiat.*, 2005, **12**, 537–541.
- 50 N. Binsted, *EXCURV98*, CCLRC Daresbury Laboratory Computer Program, 1998.

