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Designing conjugated porous polymers for visible light-driven photocatalytic chemical transformations

Jeehye Byun^{bc} and Kai A. I. Zhang  ^{ab}

Conjugated porous polymers (CPPs) have recently emerged as a new class of visible light-active, organic and heterogeneous photocatalysts for visible light-mediated photoredox reactions. The CPPs have been established as a potential alternative to resolve critical drawbacks of traditional molecular and homogeneous photocatalysts due to their structural durability, non-toxicity, low cost due to the absence of noble metals, and high designability. Tremendous attempts have been made toward the design and synthesis of CPPs for a variety of visible light-promoted photocatalytic chemical transformations. Nevertheless, the concomitant design protocols of CPPs have not been well structured so far. Herein, in this review, we aim to summarize the recent developments in controlling the structural, photophysical and electronic properties of CPPs, and thereby extract the underlying design principles. According to the principle of the photocatalytic process, key parameters for the molecular design of CPPs were described in three sections: (1) light absorbance by energy band gap, (2) charge separation and transport, and (3) electron transfer to the target substrate. The macroscopic features, *i.e.* morphology, porosity and chemical functionality, and processibility of CPPs were also presented for the enhancement of their photocatalytic activity.

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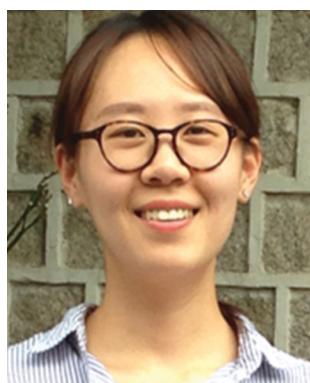
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1. Introduction

Sunlight is an enormous power reservoir which daily supplies our planet with an immense amount of energy.¹ For using solar energy as the most available renewable energy source, nature has shown its ability to utilize sunlight to drive chemical

^a Department of Materials Science, Fudan University, Shanghai 200433, P. R. China. E-mail: kai_zhang@fudan.edu.cn
^b Max Planck Institute for Polymer Research, Ackermannweg 10, 55128 Mainz, Germany. E-mail: kai.zhang@mpip-mainz.mpg.de
^c Water Cycle Research Center, Korea Institute of Science and Technology (KIST), Hwarangno 14-gil 5, Seongbuk-gu, 02792 Seoul, Republic of Korea



Jeehye Byun

Jeehye Byun attained her PhD from Korea Advanced Institute of Science and Technology (KAIST) in 2016, and worked as a postdoctoral fellow in the group of Dr Kai Zhang at Max Planck Institute for Polymer Research supported by the Alexander von Humboldt Foundation. She is currently a senior researcher in Korea Institute of Science and Technology (KIST). Her research interests are designing porous materials for environmental applications, *i.e.* CO₂ conversion and water treatment.



Kai A. I. Zhang

Kai Zhang did his PhD at the University of Cologne, Germany and partly at the University of Strathclyde, Glasgow, UK. After working as a postdoctoral researcher with Prof. Markus Antonietti at the Max Planck Institute of Colloids and Interfaces, Potsdam, Germany, he joined Max Planck Institute for Polymer Research in 2013 as a research group leader in the department of Physical Chemistry of Polymers under Prof. Katharina Landfester. In 2019, he joined as

full professor the Department of Materials Science, Fudan University, Shanghai, China. His research interests include the development of organic porous materials, macromolecular photocatalysts and their applications in organic photoredox reactions, fine chemistry, photobiocatalysis etc.

processes, *i.e.* photosynthesis. Inspired by nature processes, scientists have made remarkable efforts toward the development of efficient photocatalysts to facilitate chemical reactions with assistance from solar energy.^{2,3} Among the intensely investigated systems, homogeneous molecular photocatalysts have led the earlier discoveries of photocatalysis, where the established examples are transition metal complexes^{4–6} and organic dyes.⁷ Despite the performance, several drawbacks are still associated with homogeneous photocatalysts, to name but a few, toxicity of metals, high cost, and stability issues due to the photo-bleaching effect. Furthermore, the accompanying purification step of the homogeneous system is also a nettle-some obstacle that hinders their large-scale field applications.

Metal-free and heterogeneous photocatalysts have thus drawn attention in recent years for resolving the disadvantages of homogeneous photocatalytic systems. Carbon nitrides, a state-of-the-art example of metal-free photocatalysts, have been deeply studied for visible light-driven photoredox reactions such as water splitting,^{8–13} CO₂ reduction,^{14–16} bioimaging,^{17,18} humidity sensing,¹⁹ and homogeneous catalysis.²⁰ π -Conjugated polymers are another emerging class as an alternative to the traditional homogeneous photocatalysts as they have versatile photophysical and electronic properties that can be controlled by the diverse combination of building blocks. Moreover, by virtue of structural designability, the conjugated polymers can be further endowed with morphological traits and specific chemical functionalities, facilitating a wider range of photocatalytic reactions with enhanced efficiency. Conjugated porous polymers (CPPs), in this regard, which combine promising features with controllable porosity and surficial properties, have arisen as an ideal candidate for visible-light-driven photocatalysis.²¹ In 2013, CPPs were first demonstrated for visible light-induced chemical transformations, in which a series of poly-benzothiadiazoles could catalyze singlet oxygen (${}^1\text{O}_2$) generation under blue light illumination to convert α -terpinene into ascaridole.²² Since then, a considerable number of photoredox reactions have been conducted with CPPs as heterogeneous photocatalysts. Recent examples include selective oxidation of sulfides,²³ free radical polymerization,²⁴ dehalogenation of haloketones,²⁵ oxidative coupling of amines,²⁶ hydrogen^{27–31} and oxygen evolution,^{32,33} reduction of 4-nitrophenol,³⁴ metal-free Stille-type C–C coupling,³⁵ [2+2] cycloaddition,³⁶ and enantioselective α -alkylation³⁷ *etc.*

A variety of synthetic protocols are well established to prepare CPPs,³⁸ and the typical chemical routes are as follows; (i) metal-assisted cross coupling reactions (*e.g.* Suzuki,²⁴ Sonogashira,^{22,39} Heck,⁴⁰ Yamamoto⁴¹, Glaser,⁴² Negishi,⁴³ Kumada coupling reaction,⁴⁴ and oxidative polymerization^{45,46}), (ii) acid^{34,47–49} and base-catalyzed polymerization,^{27,50} and (iii) thermo-driven polymerization.^{9,51,52} Such a diversity of chemical reactions with various building blocks produces a number of CPPs for visible light-promoted photocatalytic applications.^{53,54} The π -conjugated skeleton and the corresponding properties are to be controlled by a suitable type of reaction and chemical composition, depending on the targeted applications.

Despite the performance of CPPs shown in recent studies, the associated design principles of CPPs for photocatalysis have

not been well defined, raising questions such as (i) how the molecular composition and geometry affect the photocatalytic activity of the CPPs, (ii) how the functionality and morphology of the CPPs can improve the catalytic efficiency, and (iii) how the CPPs can be utilized in a large-scale system. There is therefore a demand to reveal the structure–property relationship of the CPPs for photocatalytic application. Here, carbon nitrides, which belong to their own class with a number of comprehensive reviews,^{55–58} will not be much covered. This review is to summarize the latest investigations upon how the molecular structures and physicochemical properties of CPPs correlate with their photocatalytic behavior. In addition, a design strategy to scale up and process CPPs for potential field applications is also discussed. We believe that this review can provide a general design concept to fine-tune the structure and property of CPPs for targeted photocatalytic chemical transformations under visible light.

2. Design of CPPs for photocatalysis

The photocatalytic performance of CPPs is originated from strong light-harvesting ability arising from the extended conjugation and the corresponding energy band gap. A few early examples of organic polymer photocatalysts were linear chains with delocalized π -systems,^{59,60} followed by recent studies of linear polymers with considerable photocatalytic performance.^{61–65} By integrating morphological and surficial properties in three dimensional structures, additional strengths such as high stability, reusability and selectivity could be generated as add-on advantages, resulting into recent research activities on CPPs. In the following sections, our main focus will be firstly on the molecular design of CPPs according to the principle of photocatalysis with semiconductive materials. Then, macroscopic controls in morphology, porosity, and functionality of CPPs, and large-scale process design of CPPs will be presented in turn.

2.1. Structure design of CPPs on a molecular level

In principle, the main polymeric backbone of CPPs is sp^2 hybridized and the p_z orbitals are oriented perpendicular to the polymer chain, leading to π -delocalization and the concomitant energy band gap (E_g) by a HOMO (highest occupied molecular orbital)–LUMO (lowest unoccupied molecular orbital) gap.⁶⁶ The molecular energy level of CPPs can be empirically determined by cyclic voltammetry or by ultraviolet photoelectron spectroscopy *in vacuo*.⁶⁷ The use of computational calculations by density functional theory (DFT) has become useful to predict the energy level of CPPs.⁶⁸ The first step of heterogeneous photocatalysis by CPPs is light absorbance and the generation of charge carriers within the structure. Upon light irradiation, the semiconductive CPP absorbs photons with energy greater or equal to its band gap energy, generating electron–hole pairs. The fate of the photo-generated electron and hole could be either (volume/surface) recombination or migration to the surface for electron/energy transfer to substrates. At the surface of CPPs, the migrated electron can be donated to an electron acceptor as a reduction



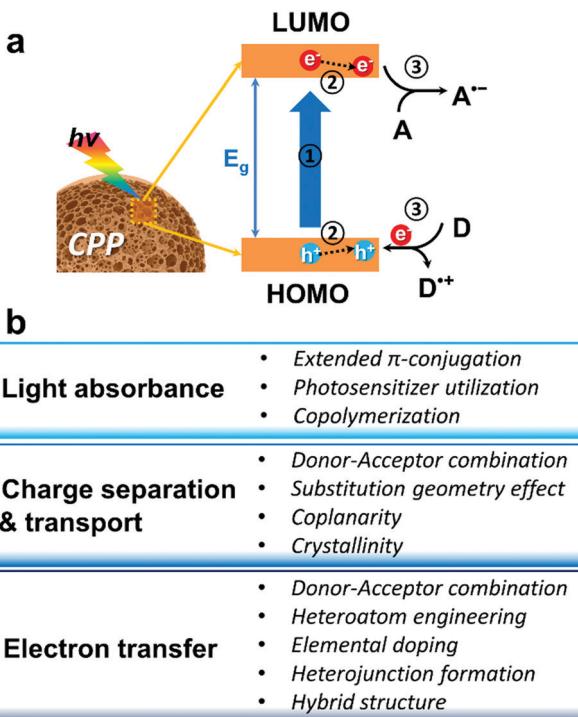


Fig. 1 (a) Photocatalytic reaction process using CPPs as photocatalysts; (1) light absorbance by the energy band gap, (2) charge separation and transport inside the CPP, and (3) electron transfer between the CPP and substrates. HOMO and LUMO indicate the highest occupied molecular orbital and lowest unoccupied molecular orbital of CPP, respectively. D and A represent electron donor and electron acceptor, respectively. (b) Molecular design strategy of CPPs based on the photocatalytic reaction processes.

process, and the hole can be combined with an electron from an electron donor as an oxidation process. Therefore, key parameters of CPPs for efficient photocatalysis can be considered as the following three steps (Fig. 1a); (1) light absorption in the visible light range with optimal energy band gap, (2) efficient dissociation of photogenerated charge carriers and their transport in CPPs, and (3) electron or energy transfer from the CPP to the target substrates associated between the band positions of the CPP and the redox potentials of the substrates. Fig. 1b summarizes the molecular design strategy of CPPs classified into three parts based on the photocatalytic reaction processes, which are described in the following sections.

2.1.1. Light absorbance of CPPs. The energy band gap of CPPs determines what portion of the solar spectrum absorbs, thus the CPPs should have a narrow band gap of < 3.18 eV to absorb visible light (390–700 nm).⁶⁹ A long extension of conjugation within CPPs allows the maximized delocalization of π electrons over the CPPs, making the optical band gap narrower. A typical approach to extend π conjugation is to utilize a longer phenylene linker within the CPPs. Zhang *et al.* have shown olefin-bridged porous organic polymers (OB-POPs) produced *via* a metal-free Knoevenagel condensation reaction using different phenylene linkers as both donor and spacer²⁷ (Fig. 2). When the number of phenylene units increased, the absorption band became red-shifted due to the extended π -conjugation. When used for H_2

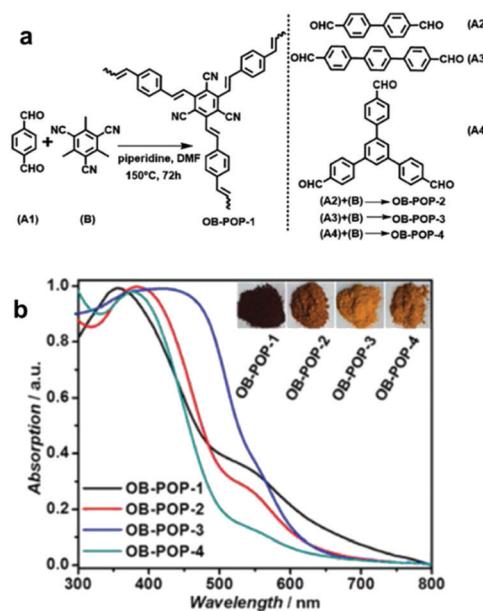


Fig. 2 Enhanced light absorption using longer phenylene linkers in CPPs. (a) Synthetic process of olefin-bridged porous organic polymers (OB-POPs) and (b) their UV/vis diffuse reflectance spectra with photographs of the resulting polymers (inset). Reproduced with permission.²⁷ Copyright 2017 Wiley-VCH.

evolution, the H_2 production rate was well in accordance with the conjugation length of the polymers, enabling a broader light harvesting in the visible region. A similar phenomenon was observed by Cooper *et al.*, of which covalent triazine-based frameworks (CTFs) showed red-shift and narrower optical band gap from 2.95 eV to 2.48 eV with respect to the increasing phenylene spacer length from phenyl to quaterphenyl, respectively.⁷⁰ The hydrogen productivity of the CTFs, however, was inconsistent with the length of the phenylene linker, where the biphenyl-linked CTF-2 showed the highest performance. The mechanistic study further proved that there was a trade-off between the light absorption and thermodynamic driving force for the oxidation of sacrificial agents. Tuning of phenylene linkers on CTFs has been recently applied for photocatalytic oxygen evolution in the UV-vis region, showing a decrease in the optical band gap from 2.98 eV to 2.36 eV upon increasing the number of phenyl units in the structure.⁷¹ The authors suggested that the combinational consideration upon optical absorbance and oxidation potential is critical for O_2 evolution.

Incorporation of photosensitizer molecules within CPPs could be a simple approach to increase the number of conjugation units for light harvesting with the lower energy band gap. Bottom-up approaches of making polymeric structures with known photosensitizing units were recently reported. Dye molecules such as Rose Bengal,⁷² BODIPY^{73–76} and Eosin Y,⁷⁷ metal complexes,^{78,79} metalloporphyrins,^{80,81} and metallo-phthalocyanine⁸² were merged into the CPP skeleton to generate porous materials with photophysical attributes. Cooper *et al.* integrated a Rose Bengal dye into conjugated microporous polymers *via* Pd-catalyzed Sonogashira–Hagihara polycondensation, showing a broad light absorption in the

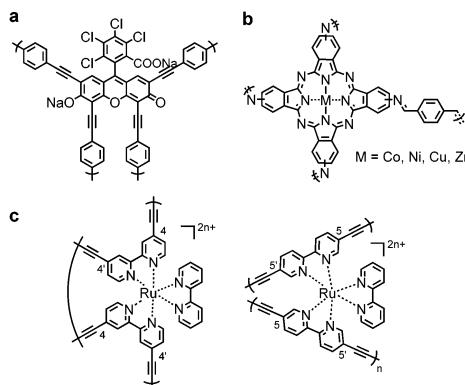


Fig. 3 Incorporation of photosensitizers in CPPs for light harvesting. (a) Conjugated microporous polymer with Rose Bengal Dye (RB-CMP1) and (b) metallophthalocyanine (MPc-CMP, M = Co, Ni, Cu, Zn). (c) Ru(bpy)₃²⁺-homocoupled polymers containing high Ru(bpy)₃²⁺ loading up to 90%.

range from 350 nm to 700 nm (Fig. 3a).⁷² The obtained polymer RB-CMP1 showed high porosity ($>830 \text{ m}^2 \text{ g}^{-1}$) and reactivity for heterogeneous photocatalytic aza-Henry reaction. Similarly, a BODIPY chromophore was employed to generate polymer photocatalyst CMPBDP by Liras *et al.*,⁷⁴ showing an intense light absorption at around 550 nm. The utilization of BODIPY into the polymer further enabled complexation with ruthenium, forming a hybrid structure for the improved catalytic efficiency.⁷⁵ The incorporation of metallophthalocyanine photosensitizer in CPPs allowed light absorption in the long-wavelength visible to far-red regions (Fig. 3b).⁸² The attained polymer MPc-CMP could efficiently generate singlet oxygen ($^1\text{O}_2$) at 700 nm (M = Co, Ni, Cu, Zn), where the zinc- and copper-coordinated structures showed the highest efficiency among the four MPc-CMPs. Meanwhile, the binding motif of monomers and their stoichiometry determine the amount of photosensitizers loaded on CPPs. Lin *et al.* showed that the homocoupling of $[\text{Ru}(\text{bpy})_3]^{2+}$ maximized the loading of chromophores into the polymer up to $\sim 90\%$,⁷⁹ comparable to the cross-coupled non-conjugated structures with 2.2% of Ru photosensitizer (Fig. 3c).⁸³ The higher loading of the photoactive unit resulted in enhanced visible light absorption and electron transfer. The Ru(bpy)₃²⁺-homocoupled polymer was almost nonporous with small surface area, however, it showed superior photocatalytic efficiency for organic transformations, *i.e.* aza-Henry reaction, oxidative coupling of amines, and reductive dehalogenation reaction, owing to efficient excited state migration through Ru chromophores.

Copolymerization of two or more monomers having different energy levels is the dominating synthetic route to control the photophysical properties of CPPs. The energy band gap of CPPs can be adjusted by the molecular hybridization between the monomers for intramolecular π electron interaction, resulting in the reduced band gap of CPPs. A representative example of the copolymerized CPPs for the band gap control was made by Cooper *et al.*²⁸ A series of pyrene-based conjugated microporous polymers (CP-CMP n , $n = 1$ to 15) was prepared by statistical copolymerization between benzene and pyrene. The tunable

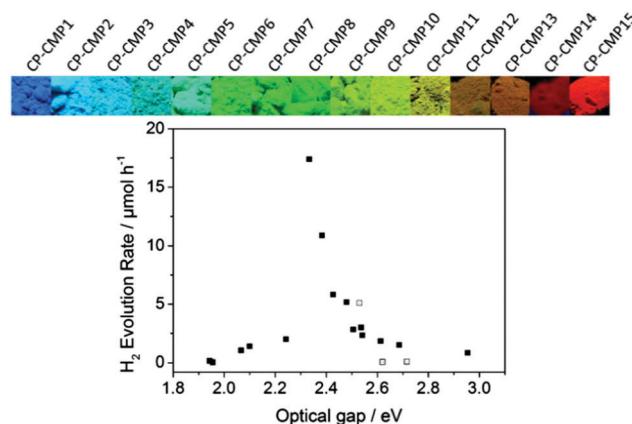


Fig. 4 Statistical copolymerization of comonomers for band gap engineering of CPP. Pyrene-based conjugated microporous polymers and their photocatalytic H₂ production efficiencies. Reproduced with permission.²⁸ Copyright 2015 American Chemical Society.

band gaps by increasing pyrene composition were found to be altered from 2.95 to 1.94 eV along with the photoluminescent peak from 445 to 588 nm, respectively (Fig. 4). Time-dependent density functional theory (TD-DFT) revealed that the narrower band position of the pyrene-containing system than that of the benzene-containing system contributed to the optical gap shifts of pyrene-dominant structures to higher wavelengths. When applied for photocatalytic H₂ evolution, CP-CMP10 with the optical band gap of 2.33 eV displayed the highest amount of H₂ produced among the structures, clearly showing that the electronic structure and optical property should be well aligned for the enhanced photocatalytic activity.

2.1.2. Charge separation and transport within CPPs. Upon the light absorption, the electron–hole pair is generated by photoexcitation, and the efficient charge separation in CPPs is often achieved by having a donor–acceptor (D–A) interface within the structure. When the CPPs consist of an electron-rich donor and electron-poor acceptor, the photoexcited donor unit transfers the electron to the acceptor unit with higher electron affinity, preventing the fast charge recombination. Zhang and Shi *et al.* reported a copolymerization of aromatics with graphitic carbon nitride ($\text{g-C}_3\text{N}_4$) to give a series of intramolecular D–A conjugated polymers, showing the enhanced charge carrier separation and mobility.⁸⁴ Under light irradiation, the $\text{g-C}_3\text{N}_4$ possessed photoexcited electrons and holes in the same triazine ring, where the electron located on carbon and the hole on nitrogen, inducing the charge recombination with high probability. However, by the incorporation of quinoline on $\text{g-C}_3\text{N}_4$ by the nucleophilic substitution reaction, the photogenerated electrons on the HOMO of the nitrogen-based donor unit were transferred to the LUMO of the quinoline acceptor unit by charge transfer transition, and thus the electron and holes could locate at the separated units for delayed charge recombination. The improved electronic conductivity and photocurrent further confirmed that the quinoline-based conjugated polymer showed efficient charge separation and transfer, leading to four times higher H₂ production rate than



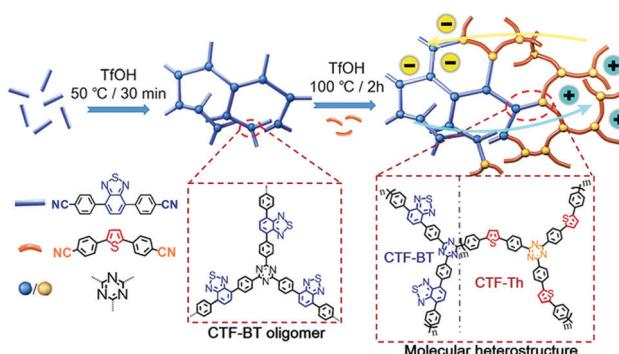


Fig. 5 Enhanced charge separation within a covalent triazine framework by creating an intramolecular heterostructure via sequential copolymerization strategy. Reproduced with permission.⁸⁶ Copyright 2018 Wiley-VCH.

the pure $\text{g-C}_3\text{N}_4$ photocatalyst. Similarly, CPP with alternating heptazine (D) and benzothiadiazole (A) moieties showed enhanced H_2 evolution compared to $\text{g-C}_3\text{N}_4$ due to the stabilization of the photoexcited charge carriers in the D-A structure.⁸⁵

Li *et al.* reported on the design of molecular donor-acceptor heterostructures with covalent triazine frameworks (Fig. 5).⁸⁶ Within the polymer network, benzothiadiazole and thiophene moieties were selectively incorporated as electron-withdrawing and electron-donating units by sequential copolymerization, respectively. The authors showed enhanced charge-carrier separation within the heterostructures, and therefore the photocatalytic H_2 production could be efficiently facilitated, showing 4–6 times higher H_2 evolution rate ($6.6 \text{ mmol g}^{-1} \text{ h}^{-1}$) than the single component structures. Likewise, Zhang *et al.* showed the copolymerization of benzene with thiophene as a strong donor and benzothiadiazole as a strong acceptor to demonstrate that the D-A compositions in CPPs could improve the charge separation and mobility.⁸⁷ In particular, the structure made out of benzene and benzothiadiazole (BBT) showed the most intense signal, indicating that the photogenerated charge separation was efficient when with a weak donor and strong acceptor combination. The BBT structure thereby gave the highest photocatalytic efficiency for $\text{C}=\text{C}$ bond activation of styrene to generate benzaldehyde as the main product. Similarly, three CPPs were synthesized with a triphenyltriazine core and three linkers of thiophene, benzene, and benzothiadiazole for photocatalytic reduction of CO_2 .⁸⁸ Electrochemical impedance and photocurrent measurements confirmed that the CPP with the benzothiadiazole linker (CPs-BT) exhibited the minimum Nyquist radius and an improved photocurrent response, indicating higher charge separation and mobility. The CPs-BT thus gave a stable photoconversion of CO_2 to CO with high selectivity, leading to a high quantum yield of 1.75% at 405 nm. Bojdys *et al.* recently showed a library of sulfur and nitrogen containing porous polymers, where the thiophene-based monomers played a role as an electron donor and the triazine-containing units as an electron acceptor.⁸⁹ The authors demonstrated the optimal band gaps of the polymers for proton reduction and the oxidation of the sacrificial donor lay at 2.1–2.3 eV formed by the D-A interaction. More importantly, when the donor

and the acceptor were spaced with a phenyl linker, the push-pull effect by D-A units was rather weakened to yield less charge mobility within the polymers. The polymer consisting of benzothiophene and triazine exhibited the superior H_2 evolution rate of $3158 \mu\text{mol h}^{-1} \text{ g}^{-1}$ with 4.5% of apparent quantum efficiency at 420 nm due to the D-A induced charge transfer.

Multiple domain D-A compositions in a single CPP led to enhanced photoinduced charge separation *via* an intramolecular energy transfer. Zhang *et al.* reported an asymmetric covalent triazine framework (asy-CTF) consisting of four different molecular D-A domains in comparison to symmetric ones (Fig. 6a and b).⁹⁰ Each domain showed different electron densities on the HOMO/LUMO levels (Fig. 6c), allowing the energy transfer cascade. Time-resolved photoluminescence (TRPL) spectra revealed that asy-CTF (1.27 ns) showed shorter fluorescence lifetime compared to the symmetric CTF-Th (1.66 ns). When in the selected time trace as shown in Fig. 6d and e, asy-CTF showed much higher and broader energy emission peak than CTF-Th, suggesting multiple overlapping emissive bands which

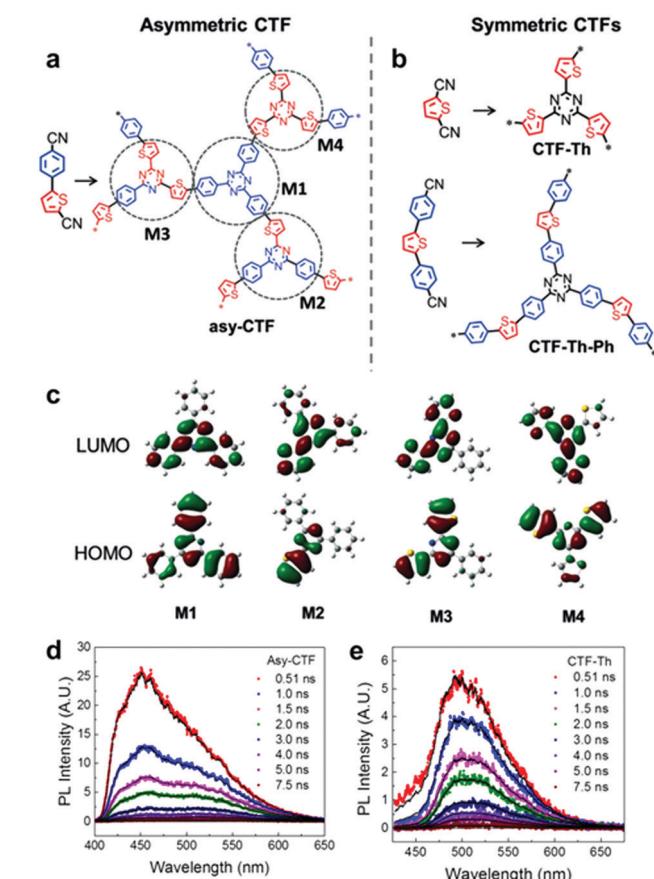


Fig. 6 Multidomain D-A compositions in CPP for energy transfer cascade. (a) Representative structure of an asymmetric covalent triazine framework (asy-CTF) containing four different molecular D-A domains (M1–M4), (b) its symmetric counterparts CTF-Th and CTF-Th-Ph with single D-A domains, and (c) the calculated electron densities over HOMO/LUMO levels of the four D-A domains in asy-CTF. TRPL spectra of (d) asy-CTF and (e) CTF-Th at an excitation wavelength of 400 nm measured in the selected time trace of 0.51–7.5 ns. Adapted with permission.⁹⁰ Copyright 2018 Wiley-VCH.

red-shift to lower energy by successive excitation energy transfer. Compared to the symmetric CTFs, asy-CTF exhibited a suppressed energy band gap of 2.3 eV and a higher photocurrent, indicating the improved light absorbance and charge mobility, respectively. Owing to the upshift of HOMO/LUMO level, asy-CTF could have a higher reductive overpotential toward an oxidant molecule ($E_{\text{red}} = -0.78$ V vs. SCE) than the symmetric CTFs, leading to the highest photocatalytic efficiency for synthesizing benzophosphole oxide from secondary phosphine oxide and alkyne in the presence of *N*-ethoxy-2-methylpyridinium tetrafluoroborate as the oxidant.

The photogenerated electrons and holes migrate through the π -conjugated CPP chains and take part in photocatalytic reactions, and thus the longer lifetime of electron–hole pairs by the efficient charge transport can improve the photocatalytic activity of CPPs. Zhang *et al.* suggested the geometry design principle of benzoxadiazole-based CPPs by merely changing the substitution positions on the central phenyl group with divalent benzoxadiazole linker (Fig. 7).²⁶ When the benzoxadiazole linker was connected to the 1,3,5-position on the phenyl unit, the CPP showed a large increase in the EPR (electron paramagnetic resonance) spectrum under light, indicating the longer living of charge carriers. On the other hand, the CPPs with 1,2,4- and 1,2,4,5-substitutions led to a marginal response in the EPR spectra, likely due to the lower charge mobility. The difference in structural configurations also resulted in the band position change in CPPs, and the CPP with 1,3,5-substitution gave the highest photocatalytic activity in the oxidative coupling of amines with the proper band alignment and the longer living electron–hole pair. The authors further showed a series of poly(benzothiadiazole)s according to the variation of substitution sites on the phenyl core with benzothiadiazole linker.³¹ The polymer

with 1,4-substitution (B-BT-1,4) exhibited the narrowest band gap of 2.17 eV and enhanced photocurrent owing to the extended π -conjugation. The linear type B-BT-1,4 yielded a higher photocatalytic H_2 production rate compared to the three-dimensional structures with 1,3,5-, 1,2,4-, and 1,2,4,5-substitution of the benzothiadiazole unit on the phenyl core, supporting the good charge transfer ability of B-BT-1,4. In the same vein, Zhang *et al.* mentioned that the *meso*-positions of the phenyl core were unfavorable for π -delocalization compared to the *para*-positions.²⁷

The effect of structure conformation on charge transport was further investigated by using a variety of comonomers on CPPs. Müllen *et al.* employed different sizes of phenylene derivatives as building blocks in poly(azomethine) networks.⁴⁷ As the size of the monomer increased (*i.e.* from benzene to anthracene), the optical band gap of the resulting polymers became narrower due to the extended π -conjugation as shown in Section 2.1.1. When the anthracene monomer was connected by the 9,10-position, however, the network linkage created a high torsion angle ($\sim 30^\circ$), interrupting the transport of charge carriers and thereby lowering the photocatalytic activity for H_2 production. With the connection to the 2,6-position on anthracene, the polymer network showed a planar structure with the torsion angle less than 2° , and the H_2 evolution rate was far increased, demonstrating a clear deviation of planarity in the connecting struts to affect the photoactivity of CPPs. Similarly, Cai *et al.* synthesized CPPs with perylene diimide and bipyridyl linker for photocatalytic H_2 evolution and dye degradation, where the coplanar structure exhibited better charge transport.⁹¹ A series of porous conjugated polymers was designed by copolymerizing electron-rich chromophores with bipyridyl comonomers.⁹² The connection between the bipyridyl unit and DBD (4,8-di(thiophen-2-yl)benzo[1,2-*b*:4,5-*b'*]dithiophene) as a strong electron-donor provided a coplanar structure of the CPP, leading to a favorable charge transport for photocatalytic H_2 evolution. A clear example for the effect of a planar structure in CPP-based photocatalysis has been unveiled by Lotsch *et al.*⁴⁹ Two-dimensional azine-based covalent organic frameworks (N_x -COFs) were produced using a series of triphenylarylaldehydes to vary the number of nitrogen atoms ($x = 0$ to 3) on the central aryl ring (Fig. 8). The increase of nitrogen content on the COF directly

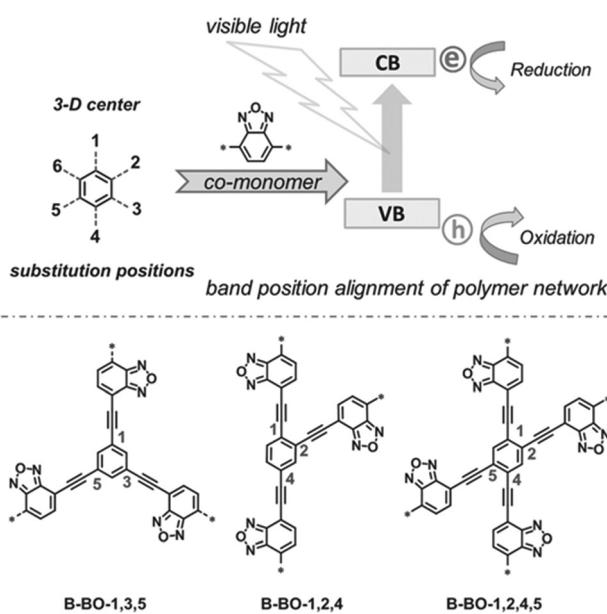


Fig. 7 Substitution geometry effect for charge transport in CPPs. Structural design of poly(benzoxadiazole)s by changing the substitution position on the central phenyl ring with benzoxadiazole linkers. Adapted with permission.²⁶ Copyright 2015 Wiley-VCH.

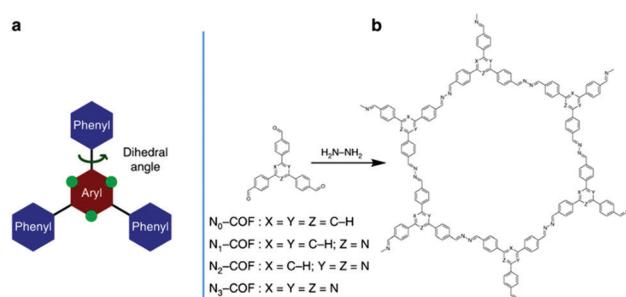


Fig. 8 Planarity control in CPPs for charge mobility. (a) Schematic illustration of the triphenylarene core and (b) synthetic procedure for a tunable azine covalent organic framework. Adapted with permission.⁴⁹ Copyright 2015 Macmillan Publishers Limited.



corresponded to the higher H_2 production rate. The more nitrogen content on the central aryl ring resulted in less dihedral angle to improve crystallinity and stack two-dimensional layers of COF, leading to the enhanced charge mobility and light harvesting.

A series of two-dimensional planar CPPs was lately reported for enhanced charge transport. Xu *et al.* have achieved overall water splitting using a 1,3-diyne-linked CPP, generating a stoichiometric amount of H_2 and O_2 under visible light.³³ The ultrathin layer of CPPs allowed the excitons to directly migrate to the polymer surface, which minimized charge recombination to yield the high quantum efficiency of 10% at 420 nm. Besides, layer-structured covalent triazine frameworks (CTF) were proven to show higher photocatalytic activities *via* the improved charge mobility and the suppressed charge recombination. Jin and Tan *et al.* presented the CTFs constructed by the polycondensation reaction between aldehyde and amidine monomers under mild conditions, and the resulting CTF-HUSTs showed extremely high efficiency for photocatalytic H_2 evolution.³⁰ The same authors were further able to control the crystallinity of CTFs by decreasing the nucleation rate during the polycondensation process. Instead of directly using the aldehyde monomer, the authors chose alcohol monomer to be *in situ* oxidized to the aldehyde monomer for slowing down the initial nucleation of CTF (Fig. 9a). By the stepwise increase of the reaction temperature,

CTF-HUST exhibited good crystallinity, leading to the improved charge transport in the framework (Fig. 9b). Indeed, the crystalline CTF-HUST-C1 gave 3.5 times higher H_2 production rate than the layered CTF-HUST-1, showing $5100 \mu\text{mol h}^{-1} \text{g}^{-1}$ and $1460 \mu\text{mol h}^{-1} \text{g}^{-1}$, respectively. The crystalline CTFs and their photocatalytic behavior were also demonstrated by Thomas *et al.*⁹³ *Via* a two-step synthetic process combining the acid-catalyzed trimerization and ionothermal method, CTFs could show the enhanced crystallinity even in much shorter reaction time (max. 30 min) than the ionothermal synthesis (40 h). The H_2 evolution efficiency of CTFs was increased with respect to the longer time of ionothermal reaction up to 10 min, likely due to higher crystallinity and polymerization degree of CTFs. When heated longer than 10 min, the H_2 production rate of CTFs started to decrease owing to the partial carbonization of the structures. Very recently, Cooper *et al.* revealed that sulfone-containing covalent organic frameworks with high crystallinity outperformed the semicrystalline and the amorphous conjugated polymer for photocatalytic H_2 evolution.⁹⁴ The crystallinity of the polymer layers facilitated the π -delocalization through π - π stacking as well as the enhanced charge carrier transport within the polymer, leading to 9 times higher H_2 production rate than the amorphous counterpart.

2.1.3. Electron transfer from CPPs to target substrates. The photogenerated electrons and holes in CPPs cause a redox reaction by the interaction with target substrates. Most importantly, the band positions of CPPs should correspond to the redox potentials of the substrates, *i.e.* the bottom level of the LUMO in CPPs has to be more negative than the reduction potential of the electron acceptor, while the top level of the HOMO should be more positive than the oxidation potential of the electron donor. The photo-induced electrons and holes with thermodynamically sufficient potentials are then able to reduce and oxidize the electron acceptor and donor, respectively. The HOMO/LUMO levels of CPPs can be altered in various ways, for instance, by the changing the conjugation length, copolymerization ratio of comonomers, and substitution positions of linkers, as we mentioned in the earlier sections. The most promising way of engineering the band positions is to introduce alternating composition of donor and acceptor moieties into CPPs. By the molecular hybridization and the intramolecular charge transfer between donor and acceptor, CPPs could have a compressed energy band gap and the associated band positions are dependent on the intrinsic attributes of the donor and acceptor units,⁹⁶ where the HOMO of CPPs is the characteristic of the HOMO of the donor and the LUMO of CPPs is the characteristic of the LUMO of the acceptor.⁹⁷ For instance, a recent study from Zhang *et al.*⁹⁶ showed that the combination of strong donor and weak acceptor could raise the reduction potential of organic photocatalysts as high as the reductive potential of an aryl halide substrate ($E_{\text{red}} \geq 2.0 \text{ V vs. SCE}$) for an aromatic C-C bond formation reaction under visible light. We again note that, however, energy band gap is highly dependent on the degree of π -delocalization, and thus the conjugation length and structural conformation of CPPs could affect the band positions along with the D-A compositions.

In a typical system of D-A based CPPs, Zhang *et al.* introduced benzobisthiadiazole as a strong electron acceptor, and

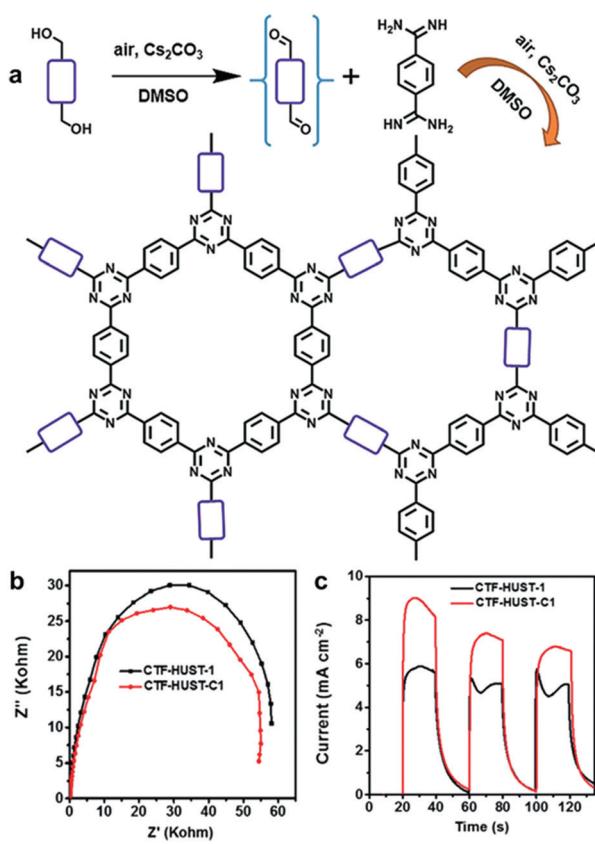


Fig. 9 Crystallinity of CPPs for charge transport. (a) Synthetic scheme of crystalline CTF-HUST-1 through *in situ* formation of aldehyde monomer from alcohol monomer. (b) Electrochemical impedance spectroscopy Nyquist plots and (c) photocurrent responses of the crystalline CTF-HUST-C1 and the layered CTF-HUST-1. Reproduced with permission.⁹⁵ Copyright 2018 Wiley-VCH.

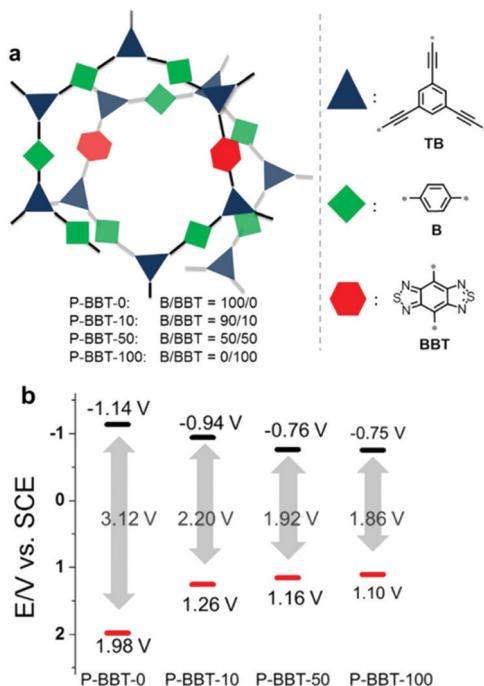


Fig. 10 HOMO/LUMO controls in CPPs by varying the amount of electron withdrawing units. (a) Synthetic compositions of D-A based conjugated nanoporous polymers, P-BBTs and (b) their redox potentials measured by cyclic voltammetry vs. SCE (Saturated Calomel Electrode). Reproduced with permission.⁹⁸ Copyright 2016 Wiley-VCH.

they found that varying the amount of the acceptor was effective to tune the band structure of CPPs for the oxidative cyclization of *N,N*-dimethylaniline with *N*-phenylmaleimide under visible-light irradiation.⁹⁸ As the amount of benzobisthiadiazole increased, the resulting HOMO–LUMO levels of the CPPs could be strategically aligned with narrower optical band gaps (Fig. 10). The reaction mechanism involved the oxidation of *N,N*-dimethylaniline ($E_{\text{Ox}} = 0.384$ V vs. SCE)⁹⁹ and the generation of perhydroxyl radicals from superoxide ($E_{\text{Red}} = -0.86$ V vs. SCE) to oxidize the cyclized intermediate. The CPP containing 10 mol% of benzobisthiadiazole (P-BBT-10) with the second highest redox potential to bracket the redox potentials of substrates therefore showed the best photocatalytic performance among the four structures. Despite the higher redox potential than P-BBT-10, the structure without benzobisthiadiazole (P-BBT-0) exhibited a large energy band gap up to 3.12 eV, yielding less conversion efficiency due to the limited visible light absorption.

Even a single change of heteroatom in the acceptor unit can alter the band position of CPPs, as displayed in Fig. 11a. Poly(benzochalcogenadiazole)s were constructed with different types of chalcogen atoms (*i.e.* O, S, and Se) on acceptor-type monomers, giving B-BO, B-BT, and B-BS, respectively.¹⁰⁰ It has been earlier reported that the stabilizing effect of a heteroatom on the LUMO can correlate with the ionization potential (IP) of the heteroatom, in other words, a smaller IP of the heteroatom leads to larger stabilization of the LUMO level in a macromolecule.¹⁰¹ The resulting CPPs exhibited a stabilizing LUMO level from B-BO to B-BS, since the degree of IP is decreasing

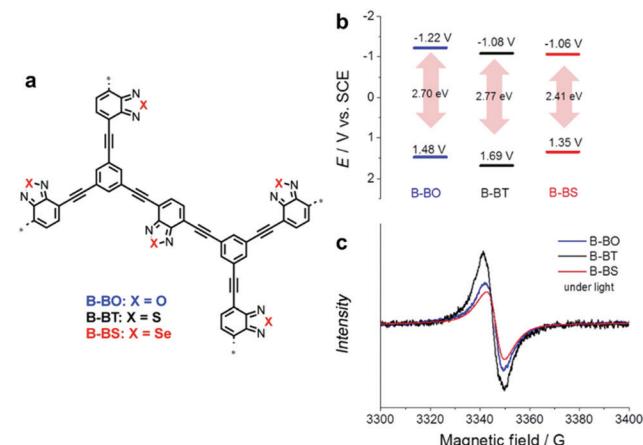


Fig. 11 Role of heteroatoms in electron acceptor units. (a) Chemical structures of poly(benzochalcogenadiazole)s, (b) their redox potentials, and (c) EPR spectra under light irradiation. Reproduced with permission.¹⁰⁰ Copyright 2015 Wiley-VCH.

from oxygen to selenium, resulting in the lower LUMO level of CPPs. Unexpectedly, however, the HOMO level of CPPs was significantly affected by the change in heteroatoms on the acceptor unit, making B-BS with sulfur of benzothiadiazole have the strongest oxidation potential. This implies that the band positions of CPPs cannot be simply predicted by the original properties of the D-A units;¹⁰² nonetheless, the higher surface area and the stronger response in the EPR spectrum of B-BS reflected the structural differences among the three CPPs for the discrepancy in their HOMO levels. The sulfur-containing B-BT showed the strongest oxidative potential, largest band gap, and most long-lived electron–hole pair production (Fig. 11b and c), resulting in the highest conversion for oxidative coupling of benzyl amines among the three CPPs.

In parallel to the acceptor control, donor-type monomers have been introduced to optimize the band structure of CPPs for target applications. Typically, carbazole-based porous organic framework (Cz-POF-1) has been successfully synthesized by Zhang *et al.*⁴⁵ The Cz-POF-1 exhibited strong electron-donating properties and extremely high surface area (~ 2065 $\text{m}^2 \text{ g}^{-1}$), accelerating photocatalytic reactions, *i.e.* dehalogenation of phenacyl bromides, hydroxylation of arylboronic acids, and α -alkylation of aldehydes. Similar carbazole-based conjugated microporous polymer (C-CMP) has been created by Loh *et al.*, where the C-CMP photocatalyst was utilized for oxidative coupling of amine, aerobic dehydrogenation of heterocycles, and oxidation of sulfides.¹⁰⁵ Recent trial of varying D-A ratio in the carbazolic copolymers (CzCP) showed a clear tendency of downshift in the HOMO/LUMO levels with respect to the increasing amount of acceptor units by the stabilizing effect, where the oxidation potential gradually increased from 1.23 V to 1.83 V vs. SCE for CzCP0 (D:A = 100:0) and CzCP100 (D:A = 0:100), respectively (Fig. 12a).¹⁰³ With the strongest oxidation capability, the CzCP100 consisting of 100% acceptor unit exhibited the highest efficiency for the oxidation of benzylic β -O-4 alcohols. For the reductive cleavage of β -O-4 ketones, on the other hand, the CzCP33

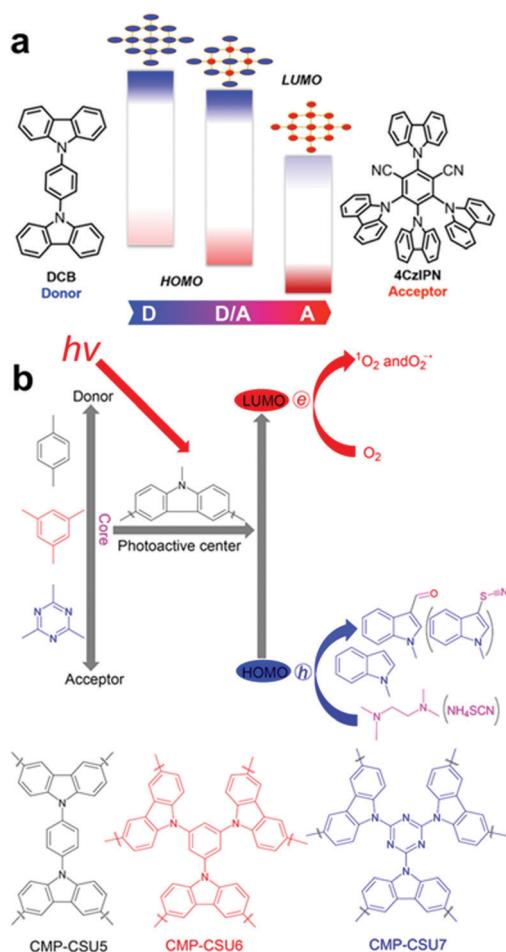


Fig. 12 HOMO/LUMO controls in CPPs by varying the amount of electron donating units. Schematic illustration of (a) D-A based carbazolic porous organic frameworks CzCP. Adapted with permission.¹⁰³ Copyright 2017 American Chemical Society. (b) Carbazole-based conjugated microporous polymers CMP-CSU. Adapted with permission.¹⁰⁴ Copyright 2018 American Chemical Society.

(D:A = 66:33) gave the highest yield owing to the largest reductive capability for the reduction of lignin oxide as a target substrate. Likewise, Pan and Yu *et al.* demonstrated that the oxidation potentials of carbazole-based conjugated microporous polymers could be altered by incorporating different D-A units as a core (Fig. 12b).¹⁰⁴ When consisting of all donor units with benzene as a core, the polymer exhibited the strongest oxidation potential of 1.42 V *vs.* SCE (CMP-SCU6), while the polymer with triazine as a core resulted in the oxidation potential of 1.19 V *vs.* SCE (CMP-SCU7). The stronger oxidation potential of CMP-SCU6 allowed the effective C-3 formylation and thiocyanation of indoles owing to the greater overpotential toward the target substrates.

Elemental doping on CPPs can change the chemical composition of D-A units, thus resulting in the shift of the band positions. The doping method on CPPs has been much conducted by using elemental sulfur. Su *et al.* reported sulfur-doped covalent triazine-based frameworks (CTFS_x, where x indicates the amount of injected sulfur, $x = 0, 5, 10, 20$ and 30 wt%), where the electronic

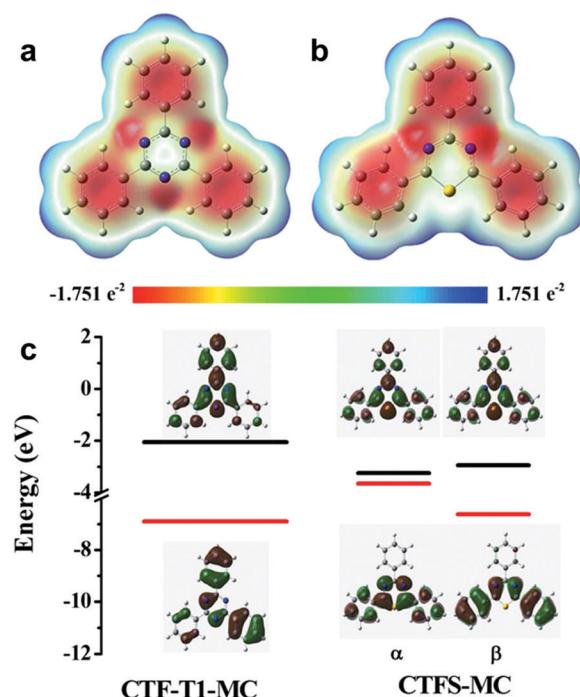


Fig. 13 Elemental doping on CPPs for controlling the energy band structure. Charge distributions of cluster models from structures of (a) covalent triazine framework (CTF-T1-MC) and (b) sulfur-doped CTF (CTFS-MC). (c) The calculated HOMO/LUMO orbitals of the cluster models. Adapted with permission.¹⁰⁶ Copyright 2016 The Royal Society of Chemistry.

structure and energy band positions of CTF were effectively modified by the loading of sulfur up to 0.52 atom% (Fig. 13).¹⁰⁶ The doped sulfur atom on CTF replaced the nitrogen atom on the triazine ring to form a C-S bond, and led to a narrower band gap of CTF by a drastic upshift of the HOMO level (being shifted from 1.61 V to 0.55 V *vs.* Ag/AgCl) through the electronic coupling between the sulfur-doped triazine ring as an acceptor and the neighboring phenyl rings as a donor. The smaller band gap of sulfur-doped CTF could harvest more visible light with a clear red shift in light absorbance, and the improved photocurrent was further observed on CTFS_x, giving 5 times higher H_2 evolution rate than the bare CTF structure. Similarly, Wang *et al.* produced a sulfur-doped polyimide (SPI) through thermal copolymerization with sublimed sulfur, giving 0.38 atom% sulfur content within the structure.¹⁰⁷ The nitrogen atom in polyimide was replaced with the doped sulfur to form a S-C bond, resulting in improved light absorbance. In contrast to the sulfur-doped CTF, however, the HOMO of SPI was lowered by 0.7 eV compared to the bare PI. The authors explained that PI consisted of pyromellitic dianhydride as an acceptor and triazine as a donor, and thus the sulfur doping on triazine promoted the stronger electron pull-push effect to lower the HOMO level of the SPI. The SPI thereby showed a faster degradation kinetic of organic dyes due to the stronger oxidation potential.

Utilization of two (or more) semiconductors as a composite is a way to adjust the band positions of photocatalysts by the charge transfer between the photocatalysts. The formation of a

heterojunction in the hybrid structure facilitates the efficient dissociation of photo-generated charge carriers at the interface of the photocatalyst, leading to an increase in the number of holes and electrons in the photocatalytic system and the decrease in charge recombination rate, as described in Section 2.1.2. As a result, one photocatalyst could have much more photogenerated electrons collected on the LUMO level for enhanced reduction reaction, while the other photocatalyst with more holes on the HOMO level can play a role in an oxidation reaction. Huang *et al.* prepared an organic/organic hybrid photocatalyst with carbon nitride (C_3N_4) and linear-type poly(3-hexylthiophene) (P3HT) for H_2 evolution.¹⁰⁸ The photoluminescence (PL) quenching experiment of the composites revealed the electron transfer between two photocatalysts, showing a drastic decrease in PL intensity of $\text{g-C}_3\text{N}_4$ peak with the increasing amount of P3HT in the composite. The evolution of new PL peaks further confirmed the recombination of the photogenerated electron in P3HT with the hole in $\text{g-C}_3\text{N}_4$ and *vice versa*, indicating the charge transfer between two organic photocatalysts. By incorporating 3 wt% of P3HT on C_3N_4 , the H_2 production rate increased up to 300 times compared to bare $\text{g-C}_3\text{N}_4$, giving 2.9% of quantum yield at 420 nm. In the identical regime of P3HT/ C_3N_4 , Peng *et al.* demonstrated that the PL lifetime of the P3HT was significantly prolonged by adding C_3N_4 to the P3HT solution, demonstrating the electron transfer from P3HT to C_3N_4 .¹⁰⁹ With ascorbic acid as a sacrificial agent, the composite exhibited a quantum yield of H_2 evolution up to 77.4% at 420 nm. Three-dimensional poly(benzothiadiazole) was also introduced in the hybrid system with C_3N_4 (BBT- C_3N_4).¹¹⁰ The BBT- C_3N_4 composite exhibited enhanced oxidation and reduction ability for photocatalytic removal of sulfathiazole and Cr(vi) owing to the efficient charge carrier transfer for the band alignment to the target substrates. Recently, the van der Waals heterostructure of aza-conjugated microporous polymer (aza-CMP) and C_2N nanosheets was shown by Xiong and Xu *et al.*, accomplishing the Z-scheme overall water splitting to produce a 2:1 stoichiometric ratio of H_2 and O_2 (Fig. 14).¹¹¹ The proper alignment of the band structure was obtained by two polymer nanosheets, where the aza-CMP took part for water oxidation and C_2N for proton reduction, leading to the Z-scheme photocatalysis. The rapid charge carrier separation and transfer were also observed by PL quenching and photocurrent measurement, obviously due to the overlapping interfaces between the ultrathin layers of two polymers.

Inorganic/organic hybrid systems have been developed in a row. Zhang *et al.* immobilized Pd nanoparticles (5–10 nm in size, 3 wt%) on a conjugated microporous poly(benzoxadiazole) (B-BO_3) for photocatalytic Suzuki coupling between aryl halide and arylboronic acid (Fig. 15).¹¹² The formation of a hybrid heterojunction on the Pd metal and B-BO_3 interface resulted in a positively-charged region in the polymer and negatively-charged region in the metal nanoparticles due to the Schottky effect.¹¹³ When irradiated, the charge carrier generated within the B-BO_3 migrated into the Pd nanoparticles, and thus the excited electron could further amplify the Schottky effect to activate hard substrates such as iodobenzene with a C–I bond even at room

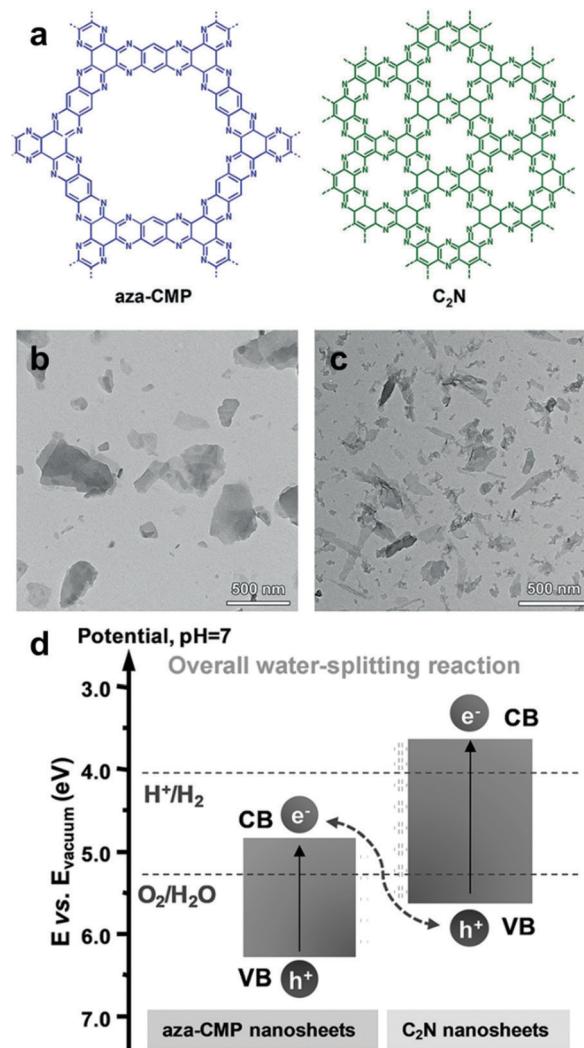


Fig. 14 Organic hybrid structure of CPPs for energy band alignment. (a) Chemical structure of aza-conjugated microporous polymer (aza-CMP) and C_2N nanosheets. TEM images of the exfoliated (b) aza-CMP and (c) C_2N nanosheet. (d) Electronic band structures of aza-CMP and C_2N nanosheet for a Z-scheme heterostructure. Reproduced with permission.¹¹¹ Copyright 2018 Wiley–VCH.

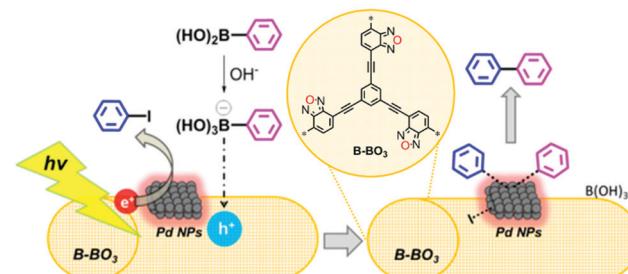


Fig. 15 Organic/inorganic hybrid structure of CPPs for electron transfer. Heterojunction formation between poly(benzoxadiazole) and palladium nanoparticles. Reproduced with permission.¹¹² Copyright 2015 American Chemical Society.

temperature. The pure B-BO_3 without Pd content exhibited no conversion, proving the indispensable role of Pd nanoparticles.



Yu *et al.* reported that the residual Pd content in CPPs after the polymer synthesis could affect the catalytic activity of CPPs, where the H₂ evolution rate could be increased by having max. ~2 wt% Pd remaining in the structure.⁹² Linear conjugated polymers shown by Kosco *et al.* demonstrated that the remaining Pd concentration in the polymers (~0.1 wt%) had a critical effect on the photocatalytic H₂ production rate, and the pure polymer after the complete removal of Pd species was inactive for H₂ evolution.¹¹⁴ This implies that the residual Pd species should be considered to evaluate the photocatalytic behavior of CPPs.

Incorporation of UV-active TiO₂ nanoparticles was recently carried out upon CPP photocatalysts. Banerjee *et al.* reported a hybrid system of porous porphyrin organic polymer (TpTph) and TiO₂ nanoparticles.¹¹⁵ The composite structure of TpTph-TiO₂ in a 1:2 ratio won over both bare TpTph and TiO₂ nanoparticles for photocatalytic H₂ production under visible-light illumination. The photo-excited electron generated by a porphyrin unit was transferred into the conduction band of TiO₂, which reduced a water molecule into H₂ in the presence of a Pt co-catalyst. The polymer/TiO₂ hybrid was also applied for photocatalytic organic transformations.¹¹⁶ Zhang *et al.* showed that poly(benzothiadiazole) containing 80 wt% of TiO₂ nanoparticles displayed an excellent photocatalytic behavior for oxidative coupling of amine and oxidation of sulfides, exceeding the efficiency of pristine poly(benzothiadiazole). More examples of inorganic/organic hybrid photocatalysts have been recently shown, for instance, MoO₃/polyimide,¹¹⁷ MoS₂/polyimide,¹¹⁸ WO₃/polyimide,¹¹⁹ and phosphotungstic acid/polyimide.¹²⁰

2.2. Macroscopic design of CPPs

The design of the CPP structure on a molecular level has induced intrinsic optical and electronic structure modulation of CPPs, affecting their photocatalytic behavior. Besides, there exist macroscopic design criteria of CPPs, *i.e.* morphology, porosity, and chemical functionalities, which can help to promote the photocatalytic reaction efficiency. Those criteria ought to be a secondary consideration after the structural control of CPPs, however, there have been a number of studies showing that the macroscopic factors boost up the photocatalytic efficiency of CPPs.

2.2.1. Morphology. Hollow architectures of CPP photocatalysts are of particular interest, in terms of the inner channels that harbor guest molecules and facilitate mass transfer. Son and coworkers reported a sulfonated hollow microporous organic polymer (S-HMOP) built by SiO₂ templating (83 ± 4 nm).¹²¹ The S-HMOP was utilized as a molecular carrier to load versatile homogeneous photocatalysts such as dye molecules, Ru complexes, and Zn-porphyrins. The utilization of hollow S-HMOP provided a platform to endow the molecular photocatalysts with high surface area, stability, and ease in separation. The effect of hollow morphology in photocatalysis has been clearly shown by Zhang *et al.* (Fig. 16a).³⁴ Nanoporous covalent triazine frameworks (CTF) were constructed *via* a super acid vapor-assisted solid phase synthetic method in the presence of SiO₂ nanoparticles (~300 nm in size). The resulting polymer CTF-BT consisted of photoactive benzothiadiazole units with interconnected hollow

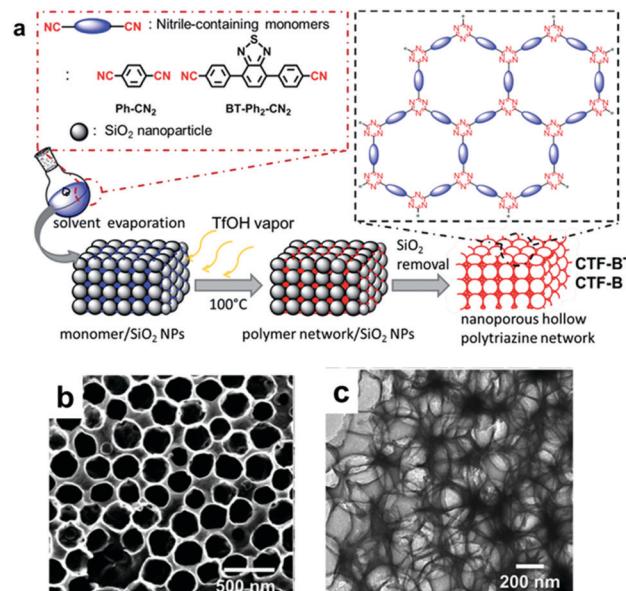


Fig. 16 Morphology control of CPPs by silica template. (a) Solid vapor synthesis of a hollow covalent triazine framework, CTF. (b) SEM and (c) TEM image of photoactive CTF-BT with a hollow structure. Reproduced with permission.³⁴ Copyright 2016 The Royal Society of Chemistry.

structure (Fig. 16b and c). When used for photocatalytic reduction of 4-nitrophenol, the CTF with a hollow structure exhibited excellent reduction behavior to produce 4-amino-phenol, while the bulk CTF and the ground CTF (to get rid of hollow voids) showed far lesser efficiency. It thus implied that the hollow structure of CTF-BT enhanced the photocatalytic activity due to efficient mass transfer and light reflection through the hollow channel. Zhang *et al.* generated a series of conjugated microporous polymer nanoparticles *via* Pd-catalyzed cross-coupling polycondensations in an oil-in-water miniemulsion.¹²² The transformation of a bulk photocatalyst into nanostructures provided high surface area, water dispersity, and processibility, which was demonstrated by the improved photo-activation of O₂ and oxidation of amines. The morphology of nanoparticles was dependent on the type of comonomers, varying the shapes *i.e.* nanosphere, nanorod, and nanoring. The authors further tried to control the morphology of nanoparticles by altering the amount of benzothiadiazole unit.⁶³ By increasing the benzothiadiazole moiety, the morphology of the resulting polymer nanoparticles was evolved from sphere to ring. The nanoring photocatalyst was highly efficient for photocatalytic inactivation of bacteria under visible light illumination. A nano-sized polymer dot photocatalyst could also be produced by post-synthetic nano-precipitation with linear benzothiadiazole polymer.¹²³ The polymer dot (Pdot) ranged in size from 30 to 50 nm with complete water dispersity, and showed the extended light absorption up to 550 nm with a narrow band gap of 2.38 eV. The effect of nano-size was proven by photocatalytic H₂ evolution behavior, in which the H₂ production efficiency of Pdot was 5-orders of magnitude higher than that of a pristine linear polymer, giving 8.3 ± 0.2 mmol h⁻¹ g⁻¹ of production rate in the absence of additional co-catalysts.

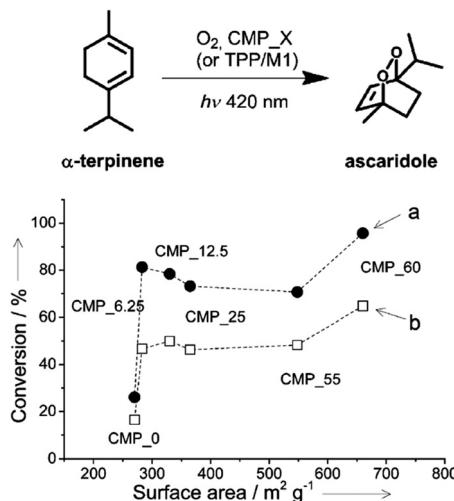


Fig. 17 Porosity control of CPPs by the silica template. Photocatalytic conversion of α -terpinene into ascaridole with poly(benzothiadiazole)s having difference surface areas. Photo-reactor set-up with (a) lower O_2 flow rate (5 mL min^{-1}) and (b) higher O_2 flow rate (10 mL min^{-1}). Adapted with permission.²² Copyright 2013 Wiley-VCH.

2.2.2. Porosity. *Via* porosity variation of CPPs, reaction rate could be controlled during photocatalysis.⁴⁵ High surface area and pore volume are thus compulsory to facilitate photocatalytic reactions and give higher yield. Vilela *et al.* utilized 12 nm-sized SiO_2 nanoparticles as a template to control the surface area of conjugated microporous poly(benzothiadiazole)s (CMP_ *X*, where *X* stands for the amount of SiO_2 added, *X* = 0, 6.25, 12.5, 25, 55, and 60 mg mL^{-1}).²² By increasing the concentration of SiO_2 nanoparticles, the surface area of the obtained polymer jumped from $270\text{ m}^2\text{ g}^{-1}$ to $660\text{ m}^2\text{ g}^{-1}$ for CMP_0 and CMP_60, respectively. With a better accessibility into high porosity, CMP_60 having the highest surface area showed a quantitative conversion of α -terpinene into ascaridole by photocatalytic singlet oxygen generation. The conversion efficiency increased almost proportionally to the increase in surface area of CMP_ *X* (Fig. 17). In order to more expand the porosity in a CPP photocatalyst, mesoporous silica SBA-15 was employed as a support by Zhang *et al.* (Fig. 18).¹²⁴ A thiophene-based covalent triazine framework (CTF-Th) was grown as a thin layer in the mesopores of SBA-15, resulting in a photoactive composite (CTF-Th@SBA-15) with high surface area ($548\text{ m}^2\text{ g}^{-1}$). CTF-Th@SBA-15 displayed superior photocatalytic activity for selective oxidation of benzyl alcohols. When the SBA-15 support was removed, the conversion efficiency of CTF-Th dropped down to one-fifth, indicating that the support-induced porosity played an important role in enhancing photocatalytic activity.

A large-scale porosity of CPPs was further achieved *via* polymerized high internal phase emulsion (polyHIPE) synthetic protocol. The π -conjugated polyHIPEs were firstly developed by Vilela and coworkers,¹²⁵ in which benzothiadiazole and fluorene moieties were copolymerized with 1,3,5-tribromobenzene in different ratios. The resulting polyHIPEs exhibited the BET surface areas of $35\text{--}50\text{ m}^2\text{ g}^{-1}$, and the monolithic structures could be utilized for a continuous flow set-up for photocatalytic

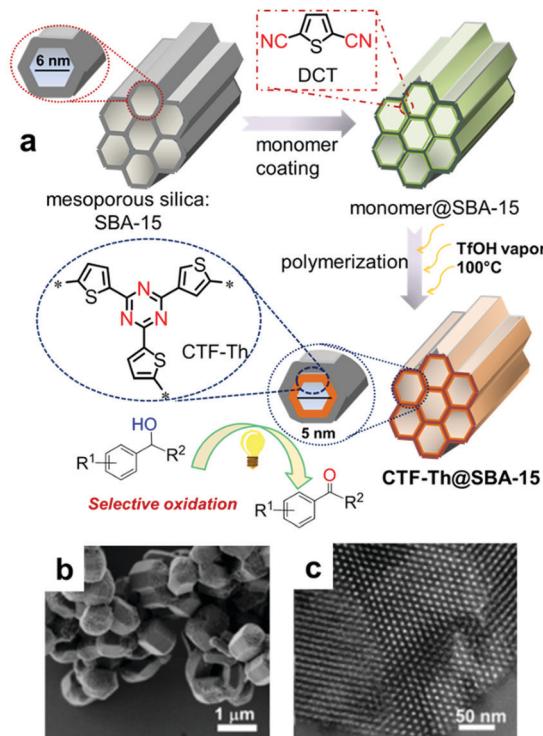


Fig. 18 Support-induced high porosity of CPPs. (a) Thiophene-based covalent triazine framework on mesoporous silica SBA-15 (CTF-Th@SBA-15). (b) SEM and (c) TEM image of CTF-Th@SBA-15. Reproduced with permission.¹²⁴ Copyright 2017 American Chemical Society.

oxidation of α -terpinene. Various CPP photocatalysts were produced in polyHIPE form, showing excellent performance for free radical polymerization²⁴ and reductive dehalogenation of haloketones.²⁵ The low surface area of polyHIPE-based CPP was reformed by thermal deprotection of a *tert*-butyl carboxylate (BOC) functional group in the polymer network (Fig. 19).²³ In a typical set-up, the π -conjugated polyHIPE (B-CB₃) showed 8 times increase in BET surface area from $27\text{ m}^2\text{ g}^{-1}$ to $230\text{ m}^2\text{ g}^{-1}$ by the removal of the BOC group. The improved porosity led to increase in conversion efficiency and selectivity for photocatalytic oxidation of sulfides.

2.2.3. Function integration. Making CPPs water-compatible has been reported as a remaining task for energy, bio, and environmental applications of CPPs.^{21,54} Post-modification of CPPs with hydrophilic functional groups is an efficient way without harming the π -conjugation of CPPs, but simply results in good water dispersity of CPPs. Poly(benzothiadiazole)s were decorated with 3-mercaptopropionic acid by thiol-ene chemistry to make water-dispersible CPPs.¹²⁶ With high water-compatibility, the resulting polymer was able to convert water-soluble 2-furoic acid to 5-hydroxy-2(5*H*)-furanone by creating singlet oxygen solely in water. Functional groups formed during the polymer formulation could also enhance water compatibility of CPPs. Zhang *et al.* reported the olefin-bridged conjugated porous polymer synthesized by a Knoevenagel reaction (OB-POP-4 as described in Section 2.1.2), and the cyano substituents in the polymer network helped to improve water wettability for photocatalytic H_2 production.²⁷

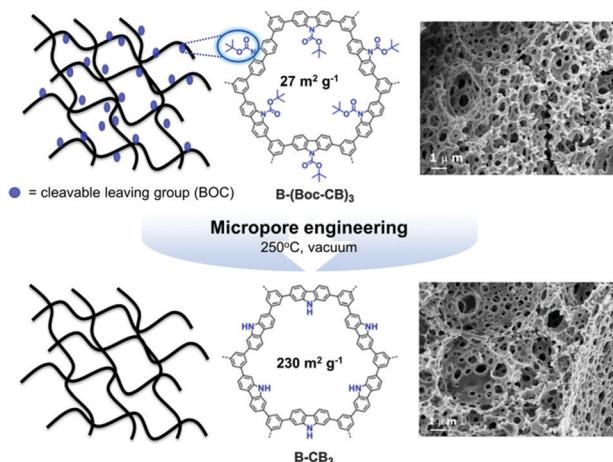


Fig. 19 Porosity control of CPPs by thermal deprotection of the functional group. Generation of additional porosity via a thermal deprotection of the BOC group in π -conjugated polyHIPEs. Reproduced with permission.²³ Copyright 2014 The Royal Society of Chemistry.

Similarly, Lotsch *et al.* found that phenyl-triazine oligomers produced *via* low-temperature ionothermal synthesis showed nice water dispersity due to the presence of nitrile units.¹²⁷ In the recent report by Cooper *et al.*, the introduction of sulfone groups in the CPPs produced much lower contact angles in water, leading to favourable interactions with water molecules and a sacrificial donor for photocatalytic H₂ evolution.⁹⁴ Ionic functional groups were further incorporated into CPPs for a better water compatibility. Zhang *et al.* attached ionic liquid moiety (1-alkyl-3-vinylimidazolium bromide) as a terminal group of the photoactive polymer (Fig. 20).¹²⁸ The water-dispersible polymer photocatalyst (P-FL-BT2) could be transformed into either nanoparticles or porous monoliths *via* a self-initiated radical crosslinking of vinylimidazolium units at the terminal. The resulting porous materials as a heterogeneous photocatalyst demonstrated excellent performance for photo-degradation of organic dyes and reduction of toxic Cr(vi). Wang *et al.* produced a photoactive polymer bearing imidazolium group with an oligo(ethylene glycol) moiety, allowing

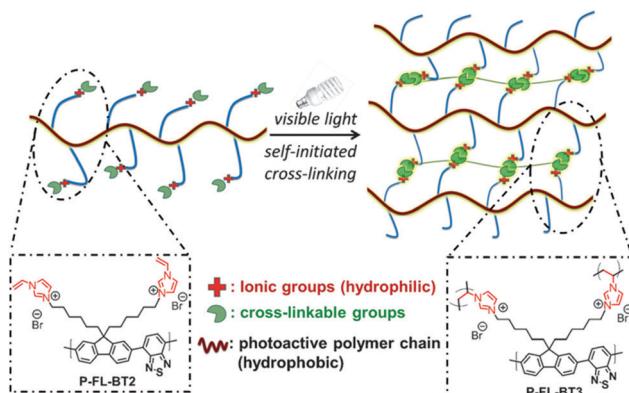


Fig. 20 Functionality control of CPPs for water compatibility. Conjugated polymer networks with vinylimidazolium group for self-initiated cross-linking. Adapted with permission.¹²⁸ Copyright 2015 Wiley-VCH.

both electrostatic binding upon the cell membrane and water compatibility for cell imaging.¹²⁹ Charge-assisted water-solubility of CPPs could be achieved by acid treatment of azulene-based conjugated microporous polymer.¹³⁰ The protonation of a cyclopentadiene ring of azulene under acidic conditions led to the formation of a stable aromatic tropylium cation, providing hydrophilicity toward the polymer network for photocatalytic water treatment.

2.3. Process design of CPPs

Future research direction of CPPs will rely on their integration into a physical device for field applications. At the current stage, most of the research is heavily focused on material development over process design. Early series of photo-reactors using CPPs were carried out with the π -conjugated polyHIPEs (shown in Section 2.2.2).^{23–25,125} In a typical condition, the porous polyHIPE monoliths were packed in a glass column, and the mixture of substrates was pumped through the column, which was irradiated with visible-light. The monolithic photocatalyst, however, has limitations in absorbing light through the thick body, so that a large number of photoactive sites hidden inside of the polymer are often wasted. A commercial photochemical flow reactor consisting of a coil reactor and LED module has also been used.⁷⁶ The photocatalyst dispersion and liquid substrate were pumped through the coil reactor repeatedly, thus improving the contact between the photocatalyst and substrate. But it may take more time and energy to treat a large amount of substrate in the coil reactor system. One way of fully utilizing the CPPs in the continuous flow system would be employing a packing support to create inner flow paths and expose the photoactive polymer on the surface. In this regard, a fixed-bed flow reactor was designed by Zhang *et al.*, incorporated with photoactive CPP-coated glass fiber (Fig. 21).³⁷ The CPP supported with glass fiber showed good light harvesting ability, which was attributed to the thin layer of the CPP (80 nm)

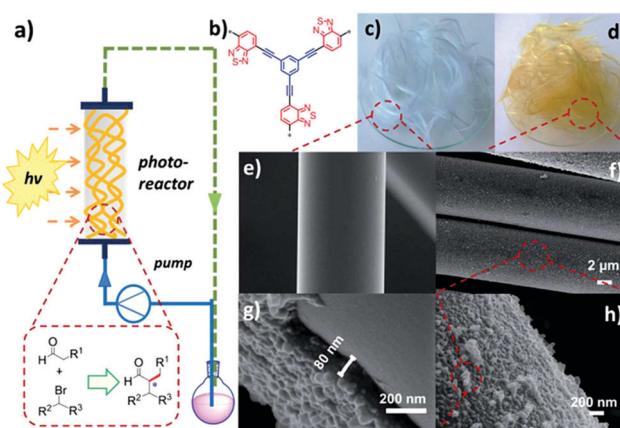


Fig. 21 Processing of CPPs for a flow column photo-reactor. (a) Schematic set-up of a fixed-bed flow photo-reactor with CPP-coated glass fibers. (b) The chemical structure of photoactive CPP. Photographs of (c) pure glass fiber and (d) CPP-coated glass fiber. SEM images of (e) pure glass fiber and (f–h) CPP-coated glass fiber. Adapted with permission.³⁷ Copyright 2017 The Royal Society of Chemistry.



on glass fiber, allowing facile light penetration through the fibers. The CPP-coated glass fiber showed an excellent photocatalytic activity for enantioselective α -alkylation of aldehyde and α -bromoketone in the photo-reactor. In consideration of the amount of polymer on the glass fiber (~ 3.2 wt%), the photocatalytic efficiency exceeded that from a flow system with a monolithic photocatalyst, using 15 times less amount of the photoactive polymer. Therefore, processing of CPPs would be of importance to maximize their photocatalytic performance for industrial applications. The coating of CPPs upon glass beads¹³¹ and films^{132,133} may also be good options to bring CPPs into well-designed photo-reactor systems.

3. Conclusions and outlook

Given the designability, various efforts have been made to optimize the structure and properties of CPPs for target-specific photocatalytic reactions under visible light irradiation. Fine control over the molecular structure of CPPs has been implemented by considering the type, amount, and geometry of monomers for the optimal band structure and charge transport/transfer within CPPs. Morphology and porosity of CPPs could be intensively controlled when a template and porogen are utilized. Chemical functionality of CPPs offers hydrophilicity manipulation. The processing of CPPs to fit in industrial standard still has a long way to go, however, the embedment of CPPs with light and transparent supports could be an option for integrating CPPs into photo-reactors.

Future challenges in designing CPPs lie ahead. One of the open questions is how far we can tune the energy levels of CPPs to tackle highly demanding reactions such as CO₂ activation and methane oxidation. The next urgent task would be the process engineering of CPPs for large-scale photocatalysis, *i.e.* commercialization of CPPs. Keeping a balance between photocatalytic activity and cost of CPPs is thus becoming a key issue. New synthetic protocols of CPPs, cost effective and good in performance, have to be considered. Nevertheless, in terms of the photocatalytic activity along with tunable structure and properties, CPPs have demonstrated explosive growth in recent years and great promise for future applications. Through a limitless structural combination of CPPs, we envision that remaining challenges upon CPPs will be dissolved in the near future by best-tailoring CPPs with the design strategies that we have summarized in this review.

Conflicts of interest

There are no conflicts to declare.

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Notes and references

- 1 R. C. Evans, P. Douglas and H. D. Burrow, *Applied Photochemistry*, Springer, Netherlands, 2014.
- 2 K. R. Reddy, M. S. Jyothi, A. V. Raghu, V. Sadhu, S. Naveen and T. M. Aminabhavi, in *Nanophotocatalysis and Environmental Applications: Detoxification and Disinfection*, ed. I. Inamuddin, A. M. Asiri and E. Lichtfouse, Springer International Publishing, Cham, 2020, pp. 139–169, DOI: 10.1007/978-3-030-12619-3_6.
- 3 S. B. Patil, P. S. Basavarajappa, N. Ganganagappa, M. S. Jyothi, A. V. Raghu and K. R. Reddy, *Int. J. Hydrogen Energy*, 2019, **44**, 13022–13039.
- 4 J. M. R. Narayanan and C. R. J. Stephenson, *Chem. Soc. Rev.*, 2011, **40**, 102–113.
- 5 C. K. Prier, D. A. Rankic and D. W. C. MacMillan, *Chem. Rev.*, 2013, **113**, 5322–5363.
- 6 T. P. Yoon, M. A. Ischay and J. Du, *Nat. Chem.*, 2010, **2**, 527.
- 7 M. Neumann, S. Füldner, B. König and K. Zeitler, *Angew. Chem., Int. Ed.*, 2011, **50**, 951–954.
- 8 F. Guo, P. Yang, Z. Pan, X.-N. Cao, Z. Xie and X. Wang, *Angew. Chem., Int. Ed.*, 2017, **56**, 8231–8235.
- 9 X. Wang, K. Maeda, A. Thomas, K. Takanabe, G. Xin, J. M. Carlsson, K. Domen and M. Antonietti, *Nat. Mater.*, 2008, **8**, 76.
- 10 Y. Zhang, T. Mori, J. Ye and M. Antonietti, *J. Am. Chem. Soc.*, 2010, **132**, 6294–6295.
- 11 Z. Pan, G. Zhang and X. Wang, *Angew. Chem., Int. Ed.*, 2019, **58**, 7102–7106.
- 12 Y. Fang, Y. Xu, X. Li, Y. Ma and X. Wang, *Angew. Chem., Int. Ed.*, 2018, **57**, 9749–9753.
- 13 H. Ou, X. Chen, L. Lin, Y. Fang and X. Wang, *Angew. Chem., Int. Ed.*, 2018, **57**, 8729–8733.
- 14 J. Lin, Z. Pan and X. Wang, *ACS Sustainable Chem. Eng.*, 2014, **2**, 353–358.
- 15 K. Wang, Q. Li, B. Liu, B. Cheng, W. Ho and J. Yu, *Appl. Catal., B*, 2015, **176–177**, 44–52.
- 16 G. Zhang, G. Li, T. Heil, S. Zafeiratos, F. Lai, A. Savateev, M. Antonietti and X. Wang, *Angew. Chem., Int. Ed.*, 2019, **58**, 3433–3437.
- 17 J. Ji, J. Wen, Y. Shen, Y. Lv, Y. Chen, S. Liu, H. Ma and Y. Zhang, *J. Am. Chem. Soc.*, 2017, **139**, 11698–11701.
- 18 X. Zhang, X. Xie, H. Wang, J. Zhang, B. Pan and Y. Xie, *J. Am. Chem. Soc.*, 2013, **135**, 18–21.
- 19 H. Arazoe, D. Miyajima, K. Akaike, F. Araoka, E. Sato, T. Hikima, M. Kawamoto and T. Aida, *Nat. Mater.*, 2016, **15**, 1084.
- 20 C. Huang, J. Wen, Y. Shen, F. He, L. Mi, Z. Gan, J. Ma, S. Liu, H. Ma and Y. Zhang, *Chem. Sci.*, 2018, **9**, 7912–7915.
- 21 F. Vilela, K. Zhang and M. Antonietti, *Energy Environ. Sci.*, 2012, **5**, 7819–7832.



22 K. Zhang, D. Kopetzki, P. H. Seeberger, M. Antonietti and F. Vilela, *Angew. Chem.*, 2013, **125**, 1472–1476.

23 Z. J. Wang, S. Ghasimi, K. Landfester and K. A. I. Zhang, *Chem. Commun.*, 2014, **50**, 8177–8180.

24 Z. J. Wang, K. Landfester and K. A. I. Zhang, *Polym. Chem.*, 2014, **5**, 3559–3562.

25 Z. J. Wang, S. Ghasimi, K. Landfester and K. A. I. Zhang, *J. Mater. Chem. A*, 2014, **2**, 18720–18724.

26 Z. J. Wang, S. Ghasimi, K. Landfester and K. A. I. Zhang, *Adv. Mater.*, 2015, **27**, 6265–6270.

27 S. Bi, Z.-A. Lan, S. Paasch, W. Zhang, Y. He, C. Zhang, F. Liu, D. Wu, X. Zhuang, E. Brunner, X. Wang and F. Zhang, *Adv. Funct. Mater.*, 2017, **27**, 1703146.

28 R. S. Sprick, J.-X. Jiang, B. Bonillo, S. Ren, T. Ratvijitvech, P. Guiglion, M. A. Zwijnenburg, D. J. Adams and A. I. Cooper, *J. Am. Chem. Soc.*, 2015, **137**, 3265–3270.

29 V. S. Vyas, V. W.-h. Lau and B. V. Lotsch, *Chem. Mater.*, 2016, **28**, 5191–5204.

30 K. Wang, L.-M. Yang, X. Wang, L. Guo, G. Cheng, C. Zhang, S. Jin, B. Tan and A. Cooper, *Angew. Chem., Int. Ed.*, 2017, **56**, 14149–14153.

31 C. Yang, B. C. Ma, L. Zhang, S. Lin, S. Ghasimi, K. Landfester, K. A. I. Zhang and X. Wang, *Angew. Chem., Int. Ed.*, 2016, **55**, 9202–9206.

32 L. Wang, Y. Wan, Y. Ding, Y. Niu, Y. Xiong, X. Wu and H. Xu, *Nanoscale*, 2017, **9**, 4090–4096.

33 L. Wang, Y. Wan, Y. Ding, S. Wu, Y. Zhang, X. Zhang, G. Zhang, Y. Xiong, X. Wu, J. Yang and H. Xu, *Adv. Mater.*, 2017, **29**, 1702428.

34 W. Huang, Z. J. Wang, B. C. Ma, S. Ghasimi, D. Gehrig, F. Laquai, K. Landfester and K. A. I. Zhang, *J. Mater. Chem. A*, 2016, **4**, 7555–7559.

35 S. Ghasimi, S. A. Bretschneider, W. Huang, K. Landfester and K. A. I. Zhang, *Adv. Sci.*, 2017, **4**, 1700101.

36 R. Li, B. C. Ma, W. Huang, L. Wang, D. Wang, H. Lu, K. Landfester and K. A. I. Zhang, *ACS Catal.*, 2017, **7**, 3097–3101.

37 W. Huang, B. C. Ma, D. Wang, Z. J. Wang, R. Li, L. Wang, K. Landfester and K. A. I. Zhang, *J. Mater. Chem. A*, 2017, **5**, 3792–3797.

38 Y. Xu, S. Jin, H. Xu, A. Nagai and D. Jiang, *Chem. Soc. Rev.*, 2013, **42**, 8012–8031.

39 J.-X. Jiang, F. Su, A. Trewin, C. D. Wood, N. L. Campbell, H. Niu, C. Dickinson, A. Y. Ganin, M. J. Rosseinsky, Y. Z. Khimyak and A. I. Cooper, *Angew. Chem., Int. Ed.*, 2007, **46**, 8574–8578.

40 L. Sun, Z. Liang, J. Yu and R. Xu, *Polym. Chem.*, 2013, **4**, 1932–1938.

41 J. Schmidt, M. Werner and A. Thomas, *Macromolecules*, 2009, **42**, 4426–4429.

42 A. Li, H.-X. Sun, D.-Z. Tan, W.-J. Fan, S.-H. Wen, X.-J. Qing, G.-X. Li, S.-Y. Li and W.-Q. Deng, *Energy Environ. Sci.*, 2011, **4**, 2062–2065.

43 G. Lu, H. Yang, Y. Zhu, T. Huggins, Z. J. Ren, Z. Liu and W. Zhang, *J. Mater. Chem. A*, 2015, **3**, 4954–4959.

44 V. Senkovskyy, I. Senkovska and A. Kiriy, *ACS Macro Lett.*, 2012, **1**, 494–498.

45 J. Luo, X. Zhang and J. Zhang, *ACS Catal.*, 2015, **5**, 2250–2254.

46 J. Schmidt, J. Weber, J. D. Epping, M. Antonietti and A. Thomas, *Adv. Mater.*, 2009, **21**, 702–705.

47 M. G. Schwab, M. Hamburger, X. Feng, J. Shu, H. W. Spiess, X. Wang, M. Antonietti and K. Müllen, *Chem. Commun.*, 2010, **46**, 8932–8934.

48 L. Stegbauer, K. Schwinghammer and B. V. Lotsch, *Chem. Sci.*, 2014, **5**, 2789–2793.

49 V. S. Vyas, F. Haase, L. Stegbauer, G. Savasci, F. Podjaski, C. Ochsenfeld and B. V. Lotsch, *Nat. Commun.*, 2015, **6**, 8508.

50 Z. J. Wang, R. Li, K. Landfester and K. A. I. Zhang, *Polymer*, 2017, **126**, 291–295.

51 S. Chu, C. Wang, Y. Yang, Y. Wang and Z. Zou, *RSC Adv.*, 2014, **4**, 57153–57158.

52 Y. Kou, Y. Xu, Z. Guo and D. Jiang, *Angew. Chem., Int. Ed.*, 2011, **50**, 8753–8757.

53 Y. L. Wong, J. M. Tobin, Z. Xu and F. Vilela, *J. Mater. Chem. A*, 2016, **4**, 18677–18686.

54 G. Zhang, Z.-A. Lan and X. Wang, *Angew. Chem., Int. Ed.*, 2016, **55**, 15712–15727.

55 K. S. Lakhi, D.-H. Park, K. Al-Bahily, W. Cha, B. Viswanathan, J.-H. Choy and A. Vinu, *Chem. Soc. Rev.*, 2017, **46**, 72–101.

56 J. Liu, H. Wang and M. Antonietti, *Chem. Soc. Rev.*, 2016, **45**, 2308–2326.

57 Y. Wang, X. Wang and M. Antonietti, *Angew. Chem., Int. Ed.*, 2012, **51**, 68–89.

58 Z. Zhou, Y. Zhang, Y. Shen, S. Liu and Y. Zhang, *Chem. Soc. Rev.*, 2018, **47**, 2298–2321.

59 B. Muktha, G. Madras, T. N. Guru Row, U. Scherf and S. Patil, *J. Phys. Chem. B*, 2007, **111**, 7994–7998.

60 S. Yanagida, A. Kabumoto, K. Mizumoto, C. Pac and K. Yoshino, *J. Chem. Soc., Chem. Commun.*, 1985, 474–475.

61 T. Amaya, D. Hata, T. Moriuchi and T. Hirao, *Chem. – Eur. J.*, 2015, **21**, 16427–16433.

62 S. Ghosh, N. A. Kouamé, L. Ramos, S. Remita, A. Dazzi, A. Deniset-Besseau, P. Beaunier, F. Goubard, P.-H. Aubert and H. Remita, *Nat. Mater.*, 2015, **14**, 505–511.

63 R. S. Sprick, B. Bonillo, R. Clowes, P. Guiglion, N. J. Brownbill, B. J. Slater, F. Blanc, M. A. Zwijnenburg, D. J. Adams and A. I. Cooper, *Angew. Chem., Int. Ed.*, 2016, **55**, 1792–1796.

64 R. S. Sprick, B. Bonillo, M. Sachs, R. Clowes, J. R. Durrant, D. J. Adams and A. I. Cooper, *Chem. Commun.*, 2016, **52**, 10008–10011.

65 M. Zhang, W. D. Rouch and R. D. McCulla, *Eur. J. Org. Chem.*, 2012, 6187–6196.

66 A. Pron and P. Rannou, *Prog. Polym. Sci.*, 2002, **27**, 135–190.

67 C. K. Song, B. J. Eckstein, T. L. D. Tam, L. Trahey and T. J. Marks, *ACS Appl. Mater. Interfaces*, 2014, **6**, 19347–19354.

68 T. M. McCormick, C. R. Bridges, E. I. Carrera, P. M. DiCarmine, G. L. Gibson, J. Hollinger, L. M. Kozycz and D. S. Seferos, *Macromolecules*, 2013, **46**, 3879–3886.

69 A. Kudo and Y. Miseki, *Chem. Soc. Rev.*, 2009, **38**, 253–278.

70 C. B. Meier, R. S. Sprick, A. Monti, P. Guiglion, J.-S. M. Lee, M. A. Zwijnenburg and A. I. Cooper, *Polymer*, 2017, **126**, 283–290.

71 Z.-A. Lan, Y. Fang, Y. Zhang and X. Wang, *Angew. Chem., Int. Ed.*, 2018, **57**, 470–474.

72 J.-X. Jiang, Y. Li, X. Wu, J. Xiao, D. J. Adams and A. I. Cooper, *Macromolecules*, 2013, **46**, 8779–8783.

73 S. Bandyopadhyay, A. G. Anil, A. James and A. Patra, *ACS Appl. Mater. Interfaces*, 2016, **8**, 27669–27678.

74 M. Liras, M. Iglesias and F. Sánchez, *Macromolecules*, 2016, **49**, 1666–1673.

75 M. Liras, M. Pintado-Sierra, M. Iglesias and F. Sanchez, *J. Mater. Chem. A*, 2016, **4**, 17274–17278.

76 J. M. Tobin, J. Liu, H. Hayes, M. Demleitner, D. Ellis, V. Arrighi, Z. Xu and F. Vilela, *Polym. Chem.*, 2016, **7**, 6662–6670.

77 C.-A. Wang, Y.-W. Li, X.-L. Cheng, J.-P. Zhang and Y.-F. Han, *RSC Adv.*, 2017, **7**, 408–414.

78 L. Pan, M.-Y. Xu, L.-J. Feng, Q. Chen, Y.-J. He and B.-H. Han, *Polym. Chem.*, 2016, **7**, 2299–2307.

79 J.-L. Wang, C. Wang, K. E. deKrafft and W. Lin, *ACS Catal.*, 2012, **2**, 417–424.

80 A. Nagai, X. Chen, X. Feng, X. Ding, Z. Guo and D. Jiang, *Angew. Chem., Int. Ed.*, 2013, **52**, 3770–3774.

81 Z. Xiao, Y. Zhou, X. Xin, Q. Zhang, L. Zhang, R. Wang and D. Sun, *Macromol. Chem. Phys.*, 2016, **217**, 599–604.

82 X. Ding and B.-H. Han, *Angew. Chem., Int. Ed.*, 2015, **54**, 6536–6539.

83 Z. Xie, C. Wang, K. E. deKrafft and W. Lin, *J. Am. Chem. Soc.*, 2011, **133**, 2056–2059.

84 X. Fan, L. Zhang, R. Cheng, M. Wang, M. Li, Y. Zhou and J. Shi, *ACS Catal.*, 2015, **5**, 5008–5015.

85 K. Kailasam, M. B. Mesch, L. Möhlmann, M. Baar, S. Blechert, M. Schwarze, M. Schröder, R. Schomäcker, J. Senker and A. Thomas, *Energy Technol.*, 2016, **4**, 744–750.

86 W. Huang, Q. He, Y. Hu and Y. Li, *Angew. Chem., Int. Ed.*, 2019, **58**, 8676–8680.

87 C. Ayed, L. Caire da Silva, D. Wang and K. A. I. Zhang, *J. Mater. Chem. A*, 2018, **6**, 22145–22151.

88 C. Yang, W. Huang, L. C. da Silva, K. A. I. Zhang and X. Wang, *Chem. – Eur. J.*, 2018, **24**, 17454–17458.

89 Y. S. Kochergin, D. Schwarz, A. Acharjya, A. Ichangi, R. Kulkarni, P. Eliášová, J. Vacek, J. Schmidt, A. Thomas and M. J. Bojdys, *Angew. Chem., Int. Ed.*, 2018, **57**, 14188–14192.

90 W. Huang, J. Byun, I. Rörich, C. Ramanan, P. W. M. Blom, H. Lu, D. Wang, L. Caire da Silva, R. Li, L. Wang, K. Landfester and K. A. I. Zhang, *Angew. Chem., Int. Ed.*, 2018, **57**, 8316–8320.

91 L. Li and Z. Cai, *Polym. Chem.*, 2016, **7**, 4937–4943.

92 L. Li, Z. Cai, Q. Wu, W.-Y. Lo, N. Zhang, L. X. Chen and L. Yu, *J. Am. Chem. Soc.*, 2016, **138**, 7681–7686.

93 S. Kuecken, A. Acharjya, L. Zhi, M. Schwarze, R. Schomäcker and A. Thomas, *Chem. Commun.*, 2017, **53**, 5854–5857.

94 X. Wang, L. Chen, S. Y. Chong, M. A. Little, Y. Wu, W.-H. Zhu, R. Clowes, Y. Yan, M. A. Zwijnenburg, R. S. Sprick and A. I. Cooper, *Nat. Chem.*, 2018, **10**, 1180–1189.

95 M. Liu, Q. Huang, S. Wang, Z. Li, B. Li, S. Jin and B. Tan, *Angew. Chem., Int. Ed.*, 2018, **57**, 11968–11972.

96 L. Wang, J. Byun, R. Li, W. Huang and K. A. I. Zhang, *Adv. Synth. Catal.*, 2018, **360**, 4312–4318.

97 A. Casey, S. D. Dimitrov, P. Shakya-Tuladhar, Z. Fei, M. Nguyen, Y. Han, T. D. Anthopoulos, J. R. Durrant and M. Heeney, *Chem. Mater.*, 2016, **28**, 5110–5120.

98 Z. J. Wang, S. Ghasimi, K. Landfester and K. A. I. Zhang, *Adv. Synth. Catal.*, 2016, **358**, 2576–2582.

99 R. N. Renaud, C. J. Stephens and G. Brochu, *Can. J. Chem.*, 1984, **62**, 565–569.

100 Z. J. Wang, K. Garth, S. Ghasimi, K. Landfester and K. A. I. Zhang, *ChemSusChem*, 2015, **8**, 3459–3464.

101 M. Heeney, W. Zhang, D. J. Crouch, M. L. Chabiny, S. Gordyev, R. Hamilton, S. J. Higgins, I. McCulloch, P. J. Skabara, D. Sparrowe and S. Tierney, *Chem. Commun.*, 2007, 5061–5063.

102 B.-G. Kim, X. Ma, C. Chen, Y. Ie, E. W. Coir, H. Hashemi, Y. Aso, P. F. Green, J. Kieffer and J. Kim, *Adv. Funct. Mater.*, 2013, **23**, 439–445.

103 J. Luo, X. Zhang, J. Lu and J. Zhang, *ACS Catal.*, 2017, **7**, 5062–5070.

104 W. Zhang, J. Tang, W. Yu, Q. Huang, Y. Fu, G. Kuang, C. Pan and G. Yu, *ACS Catal.*, 2018, **8**, 8084–8091.

105 C. Su, R. Tandiana, B. Tian, A. Sengupta, W. Tang, J. Su and K. P. Loh, *ACS Catal.*, 2016, **6**, 3594–3599.

106 L. Li, W. Fang, P. Zhang, J. Bi, Y. He, J. Wang and W. Su, *J. Mater. Chem. A*, 2016, **4**, 12402–12406.

107 C. Wang, Y. Guo, Y. Yang, S. Chu, C. Zhou, Y. Wang and Z. Zou, *ACS Appl. Mater. Interfaces*, 2014, **6**, 4321–4328.

108 H. Yan and Y. Huang, *Chem. Commun.*, 2011, **47**, 4168–4170.

109 X. Zhang, B. Peng, S. Zhang and T. Peng, *ACS Sustainable Chem. Eng.*, 2015, **3**, 1501–1509.

110 S. Wang, X. Yang, H. Hou, X. Ding, S. Li, F. Deng, Y. Xiang and H. Chen, *Catal. Sci. Technol.*, 2017, **7**, 418–426.

111 L. Wang, X. Zheng, L. Chen, Y. Xiong and H. Xu, *Angew. Chem., Int. Ed.*, 2018, **57**, 3454–3458.

112 Z. J. Wang, S. Ghasimi, K. Landfester and K. A. I. Zhang, *Chem. Mater.*, 2015, **27**, 1921–1924.

113 X.-H. Li and M. Antonietti, *Chem. Soc. Rev.*, 2013, **42**, 6593–6604.

114 J. Kosco, M. Sachs, R. Godin, M. Kirkus, L. Francas, M. Bidwell, M. Qureshi, D. Anjum, J. R. Durrant and I. McCulloch, *Adv. Energy Mater.*, 2018, **8**, 1802181.

115 G. Mukherjee, J. Thote, H. B. Aiyappa, S. Kandambeth, S. Banerjee, K. Vanka and R. Banerjee, *Chem. Commun.*, 2017, **53**, 4461–4464.

116 C. Ayed, W. Huang, R. Li, L. C. da Silva, D. Wang, O. Suraeva, W. Najjar and K. A. I. Zhang, *Part. Part. Syst. Charact.*, 2017, **35**, 1700234.

117 C. Ma, J. Zhou, H. Zhu, W. Yang, J. Liu, Y. Wang and Z. Zou, *ACS Appl. Mater. Interfaces*, 2015, **7**, 14628–14637.

118 C. Ma, H. Zhu, J. Zhou, Z. Cui, T. Liu, Y. Wang, Y. Wang and Z. Zou, *Dalton Trans.*, 2017, **46**, 3877–3886.

119 J. Dasgupta, J. Sikder, S. Chakraborty, U. Adhikari, V. P. Reddy B, A. Mondal and S. Curcio, *ACS Sustainable Chem. Eng.*, 2017, **5**, 6817–6826.

120 P. Meng, H. Heng, Y. Sun, J. Huang, J. Yang and X. Liu, *Appl. Catal., B*, 2018, **226**, 487–498.

121 N. Park, D. Kang, M. C. Ahn, S. Kang, S. M. Lee, T. K. Ahn, J. Y. Jaung, H.-W. Shin and S. U. Son, *RSC Adv.*, 2015, **5**, 47270–47274.

122 B. C. Ma, S. Ghasimi, K. Landfester, F. Vilela and K. A. I. Zhang, *J. Mater. Chem. A*, 2015, **3**, 16064–16071.

123 L. Wang, R. Fernández-Terán, L. Zhang, D. L. A. Fernandes, L. Tian, H. Chen and H. Tian, *Angew. Chem., Int. Ed.*, 2016, **55**, 12306–12310.

124 W. Huang, B. C. Ma, H. Lu, R. Li, L. Wang, K. Landfester and K. A. I. Zhang, *ACS Catal.*, 2017, **7**, 5438–5442.

125 K. Zhang, Z. Vobecka, K. Tauer, M. Antonietti and F. Vilela, *Chem. Commun.*, 2013, **49**, 11158–11160.

126 H. Urakami, K. Zhang and F. Vilela, *Chem. Commun.*, 2013, **49**, 2353–2355.

127 K. Schwinghammer, S. Hug, M. B. Mesch, J. Senker and B. V. Lotsch, *Energy Environ. Sci.*, 2015, **8**, 3345–3353.

128 S. Ghasimi, S. Prescher, Z. J. Wang, K. Landfester, J. Yuan and K. A. I. Zhang, *Angew. Chem., Int. Ed.*, 2015, **54**, 14549–14553.

129 Q. Cui, X. Wang, Y. Yang, S. Li, L. Li and S. Wang, *Chem. Mater.*, 2016, **28**, 4661–4669.

130 S. Ghasimi, K. Landfester and K. A. I. Zhang, *Chem. CatChem*, 2016, **8**, 694–698.

131 K. T. Oppelt, J. Gasiorowski, D. A. M. Egbe, J. P. Kollender, M. Himmelsbach, A. W. Hassel, N. S. Sariciftci and G. Knör, *J. Am. Chem. Soc.*, 2014, **136**, 12721–12729.

132 R. Irigoyen-Campuzano, M. González-Béjar, E. Pino, J. B. Proal-Nájera and J. Pérez-Prieto, *Chem. – Eur. J.*, 2017, **23**, 2867–2876.

133 D. J. Woods, R. S. Sprick, C. L. Smith, A. J. Cowan and A. I. Cooper, *Adv. Energy Mater.*, 2017, **7**, 1700479.

