

# Materials Advances

Volume 1  
Number 7  
October 2020  
Pages 2141–2546

[rsc.li/materials-advances](https://rsc.li/materials-advances)



ISSN 2633-5409

**COMMUNICATION**

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Simple chemical synthesis of intermetallic Pt<sub>2</sub>Y bulk  
nanopowder

Simple chemical synthesis of intermetallic Pt<sub>2</sub>Y bulk nanopowder†Yasukazu Kobayashi,<sup>a</sup> Shohei Tada<sup>b</sup> and Ryuji Kikuchi<sup>c</sup>Cite this: *Mater. Adv.*, 2020, 1, 2202Received 15th June 2020,  
Accepted 3rd August 2020

DOI: 10.1039/d0ma00419g

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**Intermetallic Pt<sub>2</sub>Y bulk nanopowder (2.9 m<sup>2</sup> g<sup>−1</sup>, 28 nm) was chemically obtained from H<sub>2</sub>PtCl<sub>6</sub>·6H<sub>2</sub>O and Y<sub>2</sub>O<sub>3</sub> that were finally reduced and alloyed in molten LiCl–CaH<sub>2</sub> under Ar at 600 °C. It is a novel simple approach to obtain nanopowders with no gloveboxes, vacuum systems, or even organic solvents.**

Platinum is the most widely used oxygen reduction reaction (ORR) catalyst in fuel cells due to its excellent catalytic performance.<sup>1,2</sup> Since Pt is expensive and rare, alloying catalysts have been intensively studied in order to decrease the platinum amount required, and to improve the performances of activity and stability. Among them, Pt–Y alloy is one of the most attractive and promising candidates due to the improved performances confirmed by several research groups.<sup>3,4</sup> However, it is a significantly challenging task to prepare alloy nanostructures for actual applications, mainly because yttrium has a very strong oxygen affinity to form Y<sub>2</sub>O<sub>3</sub> readily and the oxide is hardly reducible.<sup>5</sup> Therefore, most successful reports of Pt–Y nanoparticles have been made *via* a physical top-down approach, such as advanced sputtering techniques,<sup>6,7</sup> in which the nanoparticles were commonly prepared from the pure metals under a highly clean environment of oxygen-/moisture-tight conditions.<sup>8</sup>

On the other hand, a more challenging chemical approach has recently been attempted ardently to obtain Pt–Y nanostructures because the approach has a more scalable potential. Kanady *et al.* successfully prepared intermetallic Pt<sub>3</sub>Y nanoparticles of 5–20 nm by a molten salt reducing approach with molten borohydride (KEt<sub>3</sub>BH) both as a strong reducing agent and a reaction medium.<sup>9</sup> Roy *et al.* synthesized carbon-supported Pt<sub>x</sub>Y nanoparticles by a hydrogen-reduction of YCl<sub>3</sub> at 600–800 °C,

while keeping very low ppm-range concentrations of O<sub>2</sub> and H<sub>2</sub>O in the system.<sup>10</sup> Hu *et al.* reported intermetallic Pt<sub>3</sub>Y nanoparticles prepared from the reduction of an oxygen-free yttrium carbodiimide precursor that is insensitive to O<sub>2</sub> and H<sub>2</sub>O.<sup>11</sup> Reported chemical approaches are well summarized in a current reviewed paper in more detail.<sup>12</sup> Thus, the total reduction of yttrium species in a highly clean environment is a key strategy to obtain Pt–Y alloy nanostructures successfully.

In this study, we report a novel chemical approach to obtain intermetallic Pt<sub>2</sub>Y bulk nanopowder *via* a simple molten salt reduction.<sup>13–16</sup> In the method, Pt/Y<sub>2</sub>O<sub>3</sub> precursor was totally reduced under argon flow in a physical mixture with LiCl and CaH<sub>2</sub> powders at 600 °C, at which temperature the mixed LiCl was in a molten salt form. Apart from the previous molten salt approach,<sup>9</sup> LiCl and CaH<sub>2</sub> are commonly approachable powder chemicals at room temperature and allow for easy handling without any special facilities, such as glove boxes and vacuum systems. CaH<sub>2</sub> was additionally mixed as a reducing agent to extract oxygen from yttrium oxides in the reduction process. No organic solvents, dangerous chemicals, or explosive gases were used in the method. Even in critical comparison with previous chemical methods, our proposed approach in this study is the simplest method ever to prepare Pt–Y nanostructures, thus leading to a scalable application.

Intermetallic Pt<sub>2</sub>Y bulk nanopowder was prepared by reducing a Pt/Y<sub>2</sub>O<sub>3</sub> precursor in the molten LiCl–CaH<sub>2</sub> system. First, H<sub>2</sub>PtCl<sub>6</sub>·6H<sub>2</sub>O (Wako Pure Chem. Corp.) was dissolved in distilled water, and after mixing the solution well, Y<sub>2</sub>O<sub>3</sub> (Sigma-Aldrich Co. LLC) was suspended into the solution in a molar ratio of Pt/Y = 3/1. The platinum-rich ratio was required to obtain a Pt<sub>2</sub>Y phase. While stirring the suspension, it was kept heating at 110 °C overnight. The dried powder was then preliminarily heated at 300 °C and finally at 500 °C in air for 2 h in order to obtain the Pt/Y<sub>2</sub>O<sub>3</sub> precursor. Next, the precursor was mixed with CaH<sub>2</sub> (JUNSEI Chem. Co. Ltd) and LiCl (Wako Pure Chem. Corp.) in a mortar in a weight ratio of precursor/CaH<sub>2</sub>/LiCl = 1/2/1. The mixed powder was then loaded into a stainless steel reactor and heated at 600 °C for 2 hours under an argon gas flow.

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† Electronic supplementary information (ESI) available. See DOI: 10.1039/d0ma00419g

Finally, the reduced precursor was crushed in a mortar and rinsed with 0.1 M  $\text{NH}_4\text{Cl}$  aqueous solution to dissolve impure species, such as  $\text{CaH}_2$ ,  $\text{CaO}$ , and  $\text{LiCl}$ , and finally by distilled water in order to obtain the final reduced powder.

The crystal structure was analysed by X-ray diffraction (XRD, SmartLab (3 kW), Rigaku) with  $\text{CuK}\alpha$  radiation at 40 kV and 45 mA. The porosity was examined by  $\text{N}_2$  adsorption at  $-196^\circ\text{C}$  (BELLSORP mini-II, Microtrac-BEL). Before the measurement, the sample powder was pretreated at  $200^\circ\text{C}$  for 30 min under a vacuum, and the contained water was removed. The morphology was observed by scanning electron microscopy (SEM, JSM-7800F, JEOL Ltd) and transmission electron microscopy (TEM, a Tecnai Osiris, FEI system) with energy dispersive X-ray spectrometry (EDS) for elemental analysis. Metallic/oxidation states of platinum and yttrium in the final sample were confirmed by X-ray photoelectron spectroscopy (XPS, PS-9010, JEOL Ltd). The spectra were corrected by referencing the binding energy to carbon (C 1s 284.6 eV). We used the XPSPEAK4.1 to analyse the obtained spectra. Commercial Pt/C (type STD, Wako Pure Chem. Corp.) and  $\text{Y}_2\text{O}_3$  (Wako Pure Chem. Corp.) were used for comparison.

Fig. 1 shows the XRD patterns of the precursor and the final reduced powder. For the precursor, the observed peaks were totally identified to Pt metal. No clear peaks of  $\text{Y}_2\text{O}_3$  were observed, but a very broad signal was slightly obtained over  $30\text{--}40^\circ$  that could be assigned to amorphous yttrium oxide. For the reduced powder, the observed peaks were attributed mainly to an intermetallic  $\text{Pt}_2\text{Y}$  and partly to Pt metal at  $46.2^\circ$ . Since the precursor had a Pt/Y ratio of 3/1, the reduced powder could be composed of  $\text{Pt}_2\text{Y}$  and Pt phases. A peak corresponding to a (111) facet at  $20.2^\circ$  seemed higher than the reference one, probably indicating that the facet had grown preferentially in the intermetallic  $\text{Pt}_2\text{Y}$  phase.

Fig. 2 and Fig. S1 (ESI<sup>†</sup>) show SEM images of the reduced  $\text{Pt}_2\text{Y}$  powder. The appearance looks quite porous and the porous structure seems composed of very small nanoparticles

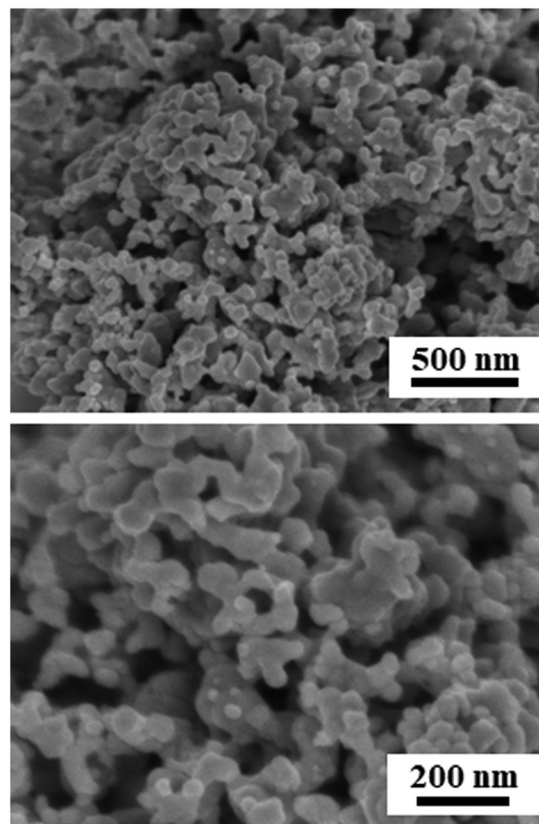


Fig. 2 SEM images of the reduced  $\text{Pt}_2\text{Y}$  bulk nanopowder.

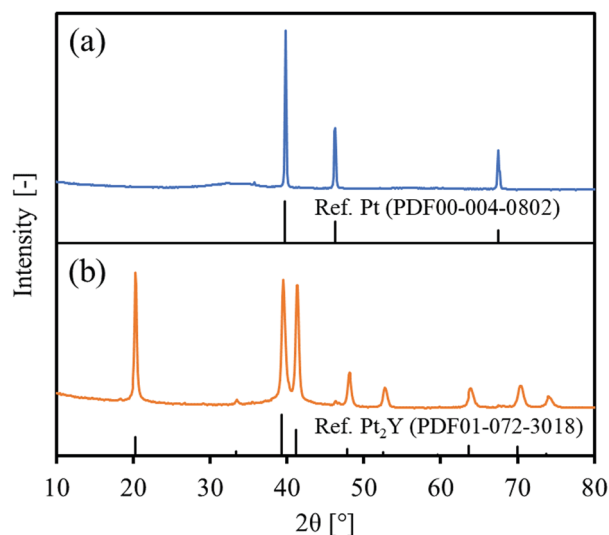


Fig. 1 XRD patterns of (a) a precursor and (b) the reduced  $\text{Pt}_2\text{Y}$  bulk nanopowder with references.

interconnected with one another. The particle size distribution is relatively broad, but nano-sized spheres of  $<100$  nm are clearly seen from the images. The nano-sized morphology was also confirmed by TEM observations (Fig. S2, ESI<sup>†</sup>), and it was likely to be polycrystalline compounds, from the magnified images (Fig. S3, ESI<sup>†</sup>). According to the results of elemental analyses by SEM-EDS (Fig. S4 and S5, ESI<sup>†</sup>) and TEM-EDS (Fig. 3 and Fig. S6, ESI<sup>†</sup>), uniform distributions are seen for both Pt and Y and they correspond with each other well. The obtained Pt/Y molar ratios were 4.5/1.0 and 3.8/1.0, respectively. Some particles were also observed as indicated in Fig. S7 (ESI<sup>†</sup>), and they were platinum metals in most cases, according to the EDS analyses. Small amounts of calcium were detected by EDS analyses of both, but they were negligible enough, so it indicated that any calcium-related impurity species were appropriately washed out by the post-rinsing treatments by  $\text{NH}_4\text{Cl}$  aqueous solution.

Fig. 4 shows the XPS spectra of the reduced  $\text{Pt}_2\text{Y}$  bulk nanopowder. The detailed information on the deconvoluted peaks is summarized in Tables S1 and S2 (ESI<sup>†</sup>). In comparison with the  $\text{Y}_2\text{O}_3$  reference,  $\text{Pt}_2\text{Y}$  shows a broad spectrum corresponding to yttrium 3d. The spectrum observed over a high binding energy region that is nearly overlapped with  $\text{Y}_2\text{O}_3$  was identified to be  $3d_{5/2}$  and  $3d_{3/2}$  of  $\text{Y}_2\text{O}_3$ , whereas that corresponding to separated peaks at 155 eV and 157 eV could be  $3d_{5/2}$  and  $3d_{3/2}$  peaks of metallic yttrium, respectively.<sup>9</sup> Thus, it was suggested that the  $\text{Pt}_2\text{Y}$  surface was composed of metallic yttrium and yttrium oxide. The existence of oxides was also





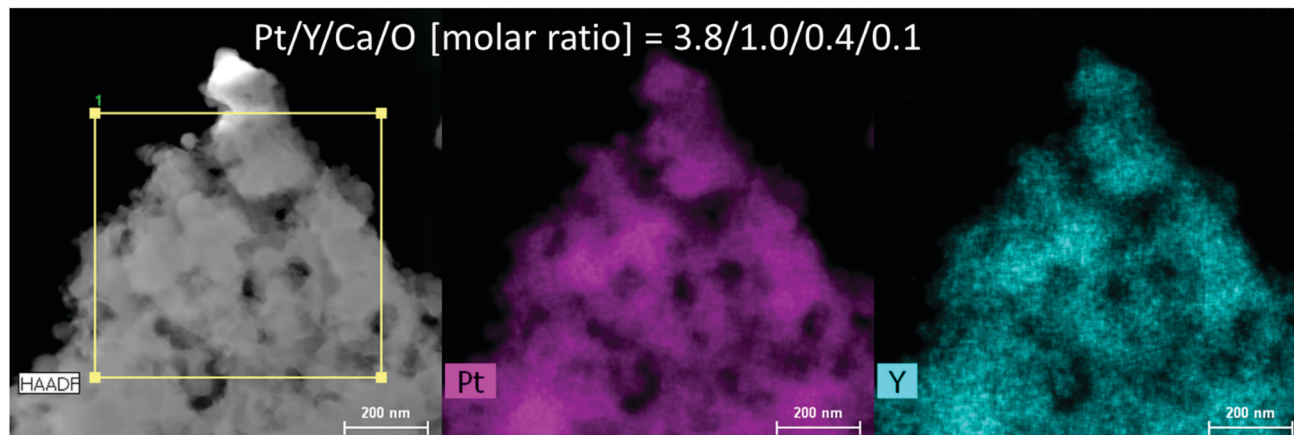


Fig. 3 Results of TEM-EDS for the reduced  $\text{Pt}_2\text{Y}$  bulk nanopowder. The EDS spectrum is depicted as Fig. S6 (ESI<sup>†</sup>).

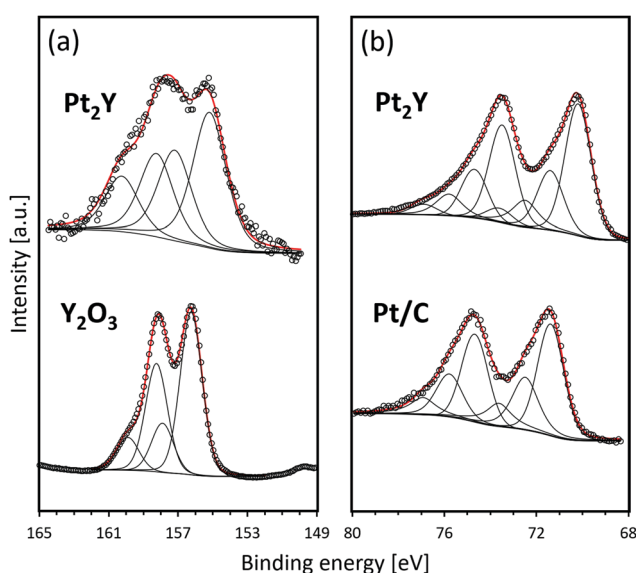


Fig. 4 XPS spectra of (a) Y 3d and (b) Pt 4f for the reduced  $\text{Pt}_2\text{Y}$  bulk nanopowder and the references of  $\text{Y}_2\text{O}_3$  and Pt/C, respectively. Open circles are experimental data, and red lines are the fitting curves separated into some peaks.

indicated by EDS analyses (Fig. S4 and S5, ESI<sup>†</sup>). The surface layer of yttrium oxide is considered to be a cumbersome issue in view of the actual catalysis application. Still, the easy removal has been confirmed by acid treatments afterward to activate the surface for superior catalytic performances.<sup>9,11</sup> As for platinum 4f, a similar spectrum to a reference Pt/C was observed. Of note, two extra peaks appear in a lower binding energy region (70.2 eV for  $4f_{7/2}$  and 73.5 eV for  $4f_{5/2}$ ). The shift indicates the electron donation from yttrium to platinum throughout the metallic bonds as reported.<sup>4,5</sup>

The average particle size of interconnected nanoparticles observed in the SEM and TEM images of the reduced  $\text{Pt}_2\text{Y}$  powder was estimated from the nitrogen adsorption experiment (Fig. S8, ESI<sup>†</sup>) and XRD measurement. As given in Table 1, the measured BET surface area was  $2.9 \text{ m}^2 \text{ g}^{-1}$ , and the estimated particle size from the surface area was 147.8 nm. Because the approximate particle size is less than a few hundred nanometers

Table 1 BET surface area (SA) and particle size of reduced  $\text{Pt}_2\text{Y}$  nanoparticles obtained by nitrogen adsorption and XRD measurement

BET SA [ $\text{m}^2 \text{ g}^{-1}$ ]	Particle size [nm]	
	$\text{N}_2$ ads. <sup>a</sup>	XRD <sup>b</sup>
2.9	147.8	28.2

<sup>a</sup> Assumed that the sample was composed of non-porous spherical  $\text{Pt}_2\text{Y}$  with the density of  $13.90 \text{ g cm}^{-3}$ . <sup>b</sup> Calculated by the Scherrer equation with a main peak at  $20.3^\circ$ .

as visible from the SEM and TEM images, the estimated size is considered reasonable and proper. In addition, the particle size is much larger than the crystallite size obtained by the Scherrer equation (28.2 nm), and this difference suggests that the  $\text{Pt}_2\text{Y}$  nanoparticles prepared in this study could be polycrystalline.

## Conclusions

Intermetallic  $\text{Pt}_2\text{Y}$  bulk nanopowder with a defined crystal structure and a nano-sized morphology was obtained *via* a simple chemical approach. No special facilities and chemicals were required to obtain the nanopowder but common chemicals, such as LiCl and  $\text{CaH}_2$ .

A part of this work was conducted at Advanced Characterization Nanotechnology Platform of the University of Tokyo, supported by “Nanotechnology Platform” of the Ministry of Education, Culture, Sports, Science and Technology (MEXT), Japan. We acknowledge Center for Instrumental Analysis, Ibaraki University for the XPS measurements.

## Conflicts of interest

There are no conflicts to declare.

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