Solvent effects on Li ion transference number and dynamic ion correlations in glyme- and sulfolane-based molten Li salt solvates†

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The Li⁺ transference number of electrolytes is one of the key factors contributing to the enhancement in the charge–discharge performance of Li secondary batteries. However, a design principle to achieve a high Li⁺ transference number has not been established for liquid electrolytes. To understand the factors governing the Li⁺ transference number, we investigated the influence of the ion–solvent interactions, Li ion coordination, and correlations of ion motions on the Li⁺ transference number in glyme (Gn, n = 1–4)- and sulfolane (SL)-based molten Li salt solvate electrolytes with lithium bis(trifluoromethansulfonyl)amide (LiTFSA). For the 1 : 1 tetraglyme-LiTFSA molten complex, [Li(G4)][TFSA], the Li⁺ transference number estimated using the potentiostatic polarisation method (tPP = 0.028) was considerably lower than that estimated using the self-diffusion coefficient data with pulsed filed gradient (PFG)-NMR (tNNR = 0.52). The dynamic ion correlations (i.e., cation–cation, anion–anion, and cation–anion cross-correlations) were determined from the experimental data on the basis of Roling and Bedrov's concentrated solution theory, and the results suggest that the strongly negative cross-correlations of the ion motions (especially for cation–cation motions) are responsible for the extremely low tPP of [Li(G4)][TFSA]. In contrast, tNNR is larger than tPP in the SL-based electrolytes. The high tNNR of the SL-based electrolytes was ascribed to the substantially weaker anti-correlations of cation–cation and cation–anion motions. Whereas the translational motions of the long-lived Li⁺(glyme)⁺ and [TFSA]⁻ dominate the ionic conduction for [Li(G4)][TFSA], Li ion hopping/exchange conduction was reported to be prevalent in the SL-based electrolytes. The unique Li ion conduction mechanism is considered to contribute to the less correlated cation–cation and cation–anion motions in SL-based electrolytes.

1. Introduction

The Li⁺ transference number, i.e., the ratio of Li ion mobility to the total ionic mobility, is one of the important parameters characterising the transport properties of electrolyte materials used in Li secondary batteries. With a growing demand for high-energy-density batteries with rapid charge–discharge capability, the significance of Li⁺ transference number has been revisited and emphasised upon for designing a highly Li ion conducting electrolytes, as recent studies on Li-ion battery modelling suggested a high-Li⁺ transference number electrolyte enables a fast charge–discharge capability even with relatively low ionic conductivity,1 and suppression of the lithium dendrite growth at the Li metal anode.2 All solid state batteries using sulphide-based solid electrolytes with Li⁺ transference number of unity demonstrated very fast charge–discharge within a few minutes3 and high-energy long-cycling with lithium metal anode.4 Molten Li salt solvates or liquid electrolytes with nearly saturated Li salt concentrations are also regarded as a prospective electrolyte material for Li secondary batteries; this is because various types of highly concentrated electrolyte systems, ranging from aqueous to non-aqueous and ionic liquid (IL), have shown improved rate capability, long-term cycling stability, and high Coulombic efficiency.5–13 In previous studies, we investigated the Li ion coordination structure and ionic transport properties of two types of molten Li salt solvates: oligoether glyme (CH₃O(CH₂)nCH₃)₁₄–₁₆ and sulfolane (SL)-based electrolytes.¹⁷,¹⁸ In the former case, a 1 : 1 molar mixture of tetraglyme (G₄) and lithium
bis(trifluoromethanesulfonyl)amide (LiTFSA) forms a thermally-stable molten Li salt solvate ([Li(G4)][TFSA]) comprising a crown-ether like [Li(G4)]⁻ complex cation and a TFSÀ anion, and it behaves like ILs. [Li(G4)][TFSA] was found to exhibit a Li⁺-limiting current density one order of magnitude higher than that of a binary mixture electrolyte of IL and Li salt, leading to the enhanced rate capability of the Li/LiCoO₂ cell.14-19 In the latter case, i.e., for SL-based molten Li salt solvates, the Li ion is observed to diffuse faster than SL and the anion; this indicates that the Li ion hopping/exchange conduction mechanism contributes significantly to the ionic conduction. A unique chain-like ⋯SL·Li⁺⋯SL·Li⁺⋯ coordination structure was suggested to be involved in the extraordinary Li ion conduction.17 The SL-based molten Li salt solvates showed a better rate capability of the Li/S cell compared with the cell using [Li(G4)][TFSA].18,20

The higher Li⁺ transference number of the concentrated electrolytes has been considered to be the key reason for the enhanced rate capability of Li secondary batteries. Nevertheless, the relationship between the molecular structure and Li⁺ transference number remains to be clarified. Furthermore, there is no established guideline for designing liquid electrolytes with a high Li⁺ transference number. For solid polymer electrolytes, Balsara and Newman systematically studied cross-correlations of ion motions by introducing Stefan–Maxwell diffusion coefficients and their relevance to the Li⁺ transference number.21-23 Woolf and Harris have also investigated the case, i.e., for SL-based molten Li salt solvates, the Li ion is observed to diffuse faster than SL and the anion; this indicates that the Li ion hopping/exchange conduction mechanism contributes significantly to the ionic conduction. A unique chain-like ⋯SL·Li⁺⋯SL·Li⁺⋯ coordination structure was suggested to be involved in the extraordinary Li ion conduction.17 The SL-based molten Li salt solvates showed a better rate capability of the Li/S cell compared with the cell using [Li(G4)][TFSA].18,20

Two different Li⁺ transference numbers were determined by either potentiostatic polarisation method or self-diffusion coefficients obtained from PFG-NMR. For the potentiostatic polarisation method, the experiments were performed using an Li symmetric cell. A porous glass filter paper (Advantec, GA55, 17 mm in diameter) impregnated with the liquid electrolyte was placed between two Li foil electrodes (Honjo Metal, 16 mm in diameter) in a 2032-coin-type cell. Impedance spectra were measured in the frequency range of 100 mHz–1 MHz with a perturbation voltage amplitude of 10 mV, using a potentiostat (Solartron Analytical, ModuLab XM ECS). The impedance measurements were preliminarily performed every hour prior to the potentiostatic polarisation to ensure the stabilisation of Li electrode interface. The time dependence of current was monitored during the potentiostatic polarisation, with a potential step of 10 mV. After the current reached the steady state, impedance spectra were measured again, as prescribed in the literature.28,29 The experimental details for PFG-NMR are available in our previous report.15 The salt diffusion coefficients (Dsalt) used for calculating Onsager transport coefficients were determined using the same Li symmetric coin cell, as described in the literature.22 The cell was polarised to 10 mV in the potentiostatic mode and the relaxation of the cell potential was recorded after the applied potential was removed. The potential relaxation was fitted using a single exponential decay function to determine Dsalt (see ESI,† in detail). The electrode potentials of Li/Li⁺ were measured with the help of the reference electrode (Li/Li⁺ in 1 mol dm⁻³ Li[TFSA]/G3) using a multi-compartment concentration cell.10 Vycor glass was used for the junction between the reference electrolyte and the sample electrolytes. The potential was recorded using a multi-channel potentiostat (BioLogic, VSP300). A slope dQ/dm(\*i) at given concentration was obtained from the slope of the concentration dependence of the concentration cell potential (Fig. S1, ESI†).26 All the above measurements were performed at 30 °C, unless otherwise noted.

3. Results and discussion

3.1 Estimation of Li⁺ transference number

Several methods have been proposed for estimating Li⁺ transference number of Li ion conducting electrolytes, such as potentiostatic polarisation, galvanostatic polarisation, very-low-frequency impedance spectroscopy, concentration cell method, PFG-NMR, and electrophoretic NMR, although there are inherent restrictions for each method (e.g., sample limitations and cumbersome procedures).25,31,32 In previous publications, we reported the apparent Li⁺ transference number derived from PFG-NMR (\( \eta^\text{MR}_i \)) for the glyme- and SL-based molten Li salt solvates electrolytes.15,18 Here, \( \eta^\text{MR}_i \) is simply

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**2. Experimental**

**2.1 Materials**

Purified diglyme (G2), triglyme (G3), and tetruglyme (G4) (water content <50 ppm) were supplied by Nippon Nyukazai. Mono-glyme (G1, battery grade <50 ppm) and sulfolane (SL, battery grade <50 ppm) were purchased from Kishida Chemical. Lithium bis(trifluoromethanesulfonyl)amide (LiTFSA), >99.9%, water content <100 ppm) was provided by Solvay Japan. All molten Li salt solvates were prepared in an inert argon-filled glove box ([H₂O] < 1 ppm; [O₂] < 1 ppm).
defined as the ratio of self-diffusion coefficients of Li ion ($D_{\text{Li}}$) and TFSA anion ($D_{\text{anion}}$),

$$t_{\text{Li}}^{\text{NMR}} = \frac{D_{\text{Li}}}{D_{\text{Li}} + D_{\text{anion}}} \quad (1)$$

The PFG-NMR method relies on Nernst–Einstein equation, assuming that the ions are completely dissociated and move independently, as postulated for ideal dilute electrolyte solutions. This would not rigorously hold for the molten Li salt solvate electrolytes with extremely high salt concentrations, and therefore, $t_{\text{Li}}^{\text{NMR}}$ may be dissimilar to the realistic Li$^+$ transferance number.

The potentiostatic polarisation combined with electrochemical impedance spectroscopy is facile and a relatively versatile method, and has been adopted for various electrolytes, ranging from polymeric to liquid systems. The Li$^+$ transferance number based on this method ($t_{\text{Li}}^{\text{pp}}$) was obtained as follows:

$$t_{\text{Li}}^{\text{pp}} = \frac{I_{\text{SS}}(V_{\text{DC}} - I_{\text{Ohm}}R_{i,0})}{I_{\text{Ohm}}(V_{\text{DC}} - I_{\text{SS}}R_{i,SS})} \quad (2)$$

where $V_{\text{DC}}$ is the constant applied voltage, $I_{\text{Ohm}}$ and $I_{\text{SS}}$ are initial and steady-state current during the polarisation, and $R_{i,0}$ and $R_{i,SS}$ are the initial and steady-state interfacial resistance at the Li metal electrodes, respectively. In previous studies, $t_{\text{Li}}^{\text{pp}}$ has often been calculated using an initial current value, $I_{\text{0}}$, determined using the initial data point of the polarisation current curve. However, as noted in recent research by Balsara et al.$^{23}$ and Roling et al.$^{25}$ $I_{\text{0}}$ may not be accurate in some cases since it strongly depends on the data sampling time. In accordance with the report by Balsara et al.$^{23}$ we used another initial current calculated by the Ohm’s law, $I_{\text{Ohm}} = V_{\text{DC}}/(R_{\text{bulk}} + R_{i,0})$, for the calculation of $t_{\text{Li}}^{\text{pp}}$. Specifically, $t_{\text{Li}}^{\text{pp}}$ can also be approximated as the realistic Li$^+$ transferance number only for dilute and completely dissociated electrolyte systems. However, $t_{\text{Li}}^{\text{pp}}$, also referred to as “current fraction” in the literature,$^{23}$ is practically useful for characterising the Li ion transport property of electrolytes because this value is measurable at any salt concentrations and is determined under anion-blocking conditions, similar to the saturation in lithium ion batteries.

$t_{\text{Li}}^{\text{pp}}$ affects the degree of concentration polarisation in the electrochemical cell with a small DC potential, and lower $t_{\text{Li}}^{\text{pp}}$ implies a larger concentration gradient of Li salt in the steady state.

With the results of polarisation current curves ($I_{\text{Ohm}}$ and $I_{\text{SS}}$) and impedance spectra ($R_{i,0}$ and $R_{i,SS}$) for the Li symmetric cell as shown in Fig. 1, $t_{\text{Li}}^{\text{pp}}$ was found to be 0.68 for a 1:2 molar mixture of LiTFSA and SL ([Li(SL)$_2$][TFSA]). Similarly, $t_{\text{Li}}^{\text{pp}}$ for [Li(G4)][TFSA] was found to be 0.028. These values reasonably agree with the reported $t_{\text{Li}}$ values for [Li(SL)$_2$][TFSA] (0.71$^{26}$) and for [Li(G4)][TFSA] (0.026$^{25}$ and 0.03$^{20}$) determined by very-low-frequency impedance method, confirming the validity of $t_{\text{Li}}^{\text{pp}}$ measured by the potentiostatic polarisation method.

Here, we studied the influence of ion–solvent interaction and Li ion coordination structure on the Li$^+$ transferance numbers for the glyme- and SL-based molten Li salt solvate electrolytes. For the glyme systems, we examined the molten Li salt solvates with different glyme chain lengths (G1 to G4), while adjusting the molar ratio of oxygen atoms of the solvent molecules to Li ions ([O]/[Li$^+$]) to be 4 or 5.$^{16}$ [Li(SL)$_2$][TFSA] and [Li(SL)$_2$][TFSA] showing a unique Li ion transport via a dominant Li ion hopping/exchange mechanism$^{17,18}$ were also studied for comparison.

The Li$^+$ transferance numbers ($t_{\text{Li}}^{\text{NMR}}$ and $t_{\text{Li}}^{\text{pp}}$) of the molten Li salt solvates are listed in Table 1. Whereas the glyme-based electrolytes showed a relatively high $t_{\text{Li}}^{\text{NMR}}$ of roughly 0.5–0.6, irrespective of the chain length of the glyme molecules, $t_{\text{Li}}^{\text{pp}}$ was one order of magnitude lower for [Li(G3)][TFSA] and [Li(G4)][TFSA] with longer glymes compared with those for [Li(G1)][TFSA] and [Li(G1)(G2)][TFSA] with shorter glymes. This dependence of $t_{\text{Li}}^{\text{pp}}$ on the chain length appears to be relevant to the classification of the molten Li salt solvates into solvate ionic liquids. [Li(G3)][TFSA] and [Li(G4)][TFSA] were regarded as good solvate ionic liquids considering aspects such as the formation of crown ether-like, long-lived [Li(glyme)]$^+$ complex cation and negligible amounts of uncoordinated glyme; in contrast, the molten Li salt solvates with the shorter glyme, such as [Li(G1,2)][TFSA], were classified into concentrated solutions in which the complex cations are not stable owing to a weaker chelate effect.$^{16}$

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**Fig. 1** (a) Polarisarion current curve and (b) Nyquist plots of an Li/Li$^+$ symmetric cell using [Li(SL)$_2$][TFSA] before and after potentiostatic polarisation. The equivalent circuit used for fitting the Nyquist plots is shown in the inset.
### 3.2 Dynamic correlations of the ions

The dynamic ion correlations in the molten Li salt solvates were experimentally determined to understand their contribution to ionic conductivity and to clarify the difference in ionic transport behaviour in the molten Li salt solvates. In this work, the Onsager transport coefficients were employed to evaluate the interionic correlations according to Roling and Bedrov et al.\(^{25,26}\)

In the Onsager transport formalism, the ionic conductivity \(\sigma_{\text{ion}}\) is described as

\[
\sigma_{\text{ion}} = \sigma_{+} + \sigma_{-} - 2\sigma_{\text{diff}}
\]

Two transport coefficients for cation and anion, \(\sigma_{+}\) and \(\sigma_{-}\), can be further divided into self-terms and distinct terms, respectively, and the ionic conductivity \(\sigma_{\text{ion}}\) is given as follows:

\[
\sigma_{\text{ion}} = \sigma_{+}^{\text{self}} + \sigma_{-}^{\text{self}} + \sigma_{+}^{\text{distinct}} + \sigma_{-}^{\text{distinct}} - 2\sigma_{\text{diff}}
\]

All the Onsager transport coefficients (\(\sigma_{+}, \sigma_{-}\), and \(\sigma_{\text{diff}}\)) can be estimated from experimentally determined quantities, \(\sigma_{\text{ion}}\), \(D_{\text{salt}}\), and \(\Delta d\Phi/d\ln(c)\) and these experimental data were listed in Table S1 (ESI). The self-terms, \(\sigma_{+}^{\text{self}}\) and \(\sigma_{-}^{\text{self}}\), are obtained from the Nernst–Einstein equation using self-diffusion coefficients of Li ions \((D_{\text{Li}})\) and the anions \((D_{\text{anion}})\).\(^{26}\) and the distinct terms can be calculated by simply subtracting \(\sigma_{+}^{\text{self}}\) (and \(\sigma_{-}^{\text{self}}\)) from \(\sigma_{+}\) (and \(\sigma_{-}\)). The detailed procedure for estimating the transport coefficients is described in Supplementary Information. The sign of \(\sigma_{+}^{\text{distinct}}\) and \(\sigma_{-}^{\text{distinct}}\) is closely related to the dynamic cross-correlations of cation–cation, anion–anion, and cation–anion, respectively. The sign of the cross-correlations becomes positive and negative for correlated and anti-correlated ion motions, respectively, while the cross-correlations of zero represent non-correlated interionic dynamics as assumed in ideal electrolyte solutions.

Table 1 lists the normalised coefficients \(\tau_{\text{Li}}\) for each electrolyte. \(\tau_{\text{Li}}\) was considerably lower than \(\tau_{\text{C}}\) for \([\text{Li(G3)}][\text{TFSA}]\) and \([\text{Li(G4)}][\text{TFSA}]\). The recent MD simulation study also suggested that the anti-correlated motions of the ions can be mitigated in the shorter glyme systems since a more frequent solvent exchange between the unstable [Li(glyme)] complex cations contributes to the momentum conservation.\(^{26}\) This would result in \(\tau_{\text{PP}}\) being a little lower than \(\tau_{\text{PP}}^{\text{MD}}\) for \([\text{Li(G1)}][\text{TFSA}]\) and \([\text{Li(G1)}][\text{TFSA}]\). In contrast, the \(\tau_{\text{PP}}\) was higher than \(\tau_{\text{PP}}^{\text{MD}}\) for \([\text{Li(SL)}][\text{TFSA}]\) and \([\text{Li(SL)}][\text{TFSA}]\). The same trend was also observed for the SL-based concentrated electrolytes with lithium bis(fluorosulfonyl)amide.\(^{34,35}\) This implies that the dynamic ion correlations positively affected the Li\(^{+}\) transference number under anion blocking conditions \(\left(\tau_{\text{Li}}^{\text{MD}}\right)\) in the SL-based electrolytes. The effects of the dynamic ion correlations on the Li\(^{+}\) transference numbers are investigated in detail in the following section.

<table>
<thead>
<tr>
<th>Sample</th>
<th>(\tau_{\text{Li}}^{\text{NR}})</th>
<th>(\tau_{\text{PP}})</th>
<th>(\tau_{\text{PP}}^{\text{MD}})</th>
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<tbody>
<tr>
<td>([\text{Li(G3)}][\text{TFSA}])</td>
<td>0.61</td>
<td>0.028</td>
<td>0.10</td>
</tr>
<tr>
<td>([\text{Li(G4)}][\text{TFSA}])</td>
<td>0.52</td>
<td>0.028</td>
<td>0.06</td>
</tr>
<tr>
<td>([\text{Li(G1)}][\text{TFSA}])</td>
<td>0.54</td>
<td>0.35</td>
<td>0.30</td>
</tr>
<tr>
<td>([\text{Li(G1)}][\text{G2}][\text{TFSA}])</td>
<td>0.48</td>
<td>0.31</td>
<td>0.26</td>
</tr>
<tr>
<td>([\text{Li(SL)}][\text{TFSA}])</td>
<td>0.61</td>
<td>0.68</td>
<td>—</td>
</tr>
<tr>
<td>([\text{Li(SL)}][\text{TFSA}])</td>
<td>0.57</td>
<td>0.69</td>
<td>—</td>
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Molecular dynamics (MD) simulations by Bedrov et al. demonstrated a good correlation between the glyme chain length and the Li\(^{+}\) transference number \(\left(\tau_{\text{Li}}^{\text{MD}}\right)\) of the glyme-based molten Li salt solvates under anion-blocking conditions.\(^{26}\) The \(\tau_{\text{Li}}\) value decreases with increasing chain length of the glyme from G1 to G4 as seen in Table 1. The general tendency observed in our experimental \(\tau_{\text{Li}}^{\text{MD}}\) is in accordance with the glyme chain length dependence of \(\tau_{\text{Li}}^{\text{MD}}\) estimated by Bedrov et al.

### Table 1

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<td>([\text{Li(SL)}][\text{TFSA}])</td>
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<td>0.68</td>
<td>—</td>
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<tr>
<td>([\text{Li(SL)}][\text{TFSA}])</td>
<td>0.57</td>
<td>0.69</td>
<td>—</td>
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</table>
Although our experiments demonstrated the most negatively large value for $\sigma_{\text{distinct}}^{++}/\sigma_{\text{ion}}$ among the cross-correlations, $\sigma_{\text{distinct}}^{++}/\sigma_{\text{ion}}$ was comparable to $\sigma_{\text{distinct}}^{-}/\sigma_{\text{ion}}$ for $[\text{Li(G4)][TFSA}]$ in the reported MD simulations. Nevertheless, our experimental results on the transport coefficients are in reasonable agreement with the results in the MD simulations. All the distinct terms were found to be negative, and the dynamic cross-correlations of the ions with the same sign was negatively larger than the cation–anion cross-correlations for $[\text{Li(G4)][TFSA}]$. The anti-correlations of $[\text{Li(G4)][TFSA}]$ in $[\text{Li(G4)][TFSA}]$ was also experimentally verified by the electrophoretic NMR study. Significant anti-correlations of the ion motions derived from the momentum conservation have been reported for solvent-free ionic liquids and molten salts. Hence, the strongly anti-correlated motions found in $[\text{Li(G3)][TFSA}]$ and $[\text{Li(G4)][TFSA}]$ corroborate our previous classification of these molten Li salt solvates into a good solvate liquid on the basis of the robustness of $[\text{Li(glyme)]}$ complex cations (i.e., long-term coupled motions of Li ion and glyme by diffusivity measurement and MD simulations) and the scarcity of uncoordinated glyme estimated by Raman analysis.

The $\sigma_{\text{distinct}}^{++}/\sigma_{\text{ion}}$ and $\sigma_{\text{distinct}}^{-}/\sigma_{\text{ion}}$ of $[\text{Li(G1)x][TFSA}]$ and $[\text{Li(G1)(G2)][TFSA}]$ with shorter glymes are less negative than those of $[\text{Li(G3)][TFSA}]$ and $[\text{Li(G4)][TFSA}]$ (Fig. 2c and d). This may be due to the occurrence of much faster ligand (glyme) exchange in the shorter glyme-based molten Li salt solvates. Shorter glymes with a high exchange rate (or a short residence
time with the Li ion) enables the momentum exchange between the ions and the ligands to satisfy the momentum conservation, and thus the cation–cation and cation–anion correlations are less anti-correlated.\(^26\) Interestingly, \(\sigma_{\text{distinct}}/\sigma_{\text{ion}}\) and \(\sigma_{+-}/\sigma_{\text{ion}}\) of the SL-based electrolytes are even less negative than those of \([\text{Li(G1)}_2][\text{TFSA}]\) and \([\text{Li(G1)(G2)}][\text{TFSA}]\) and approach zero with a small increase in the SL content from \([\text{Li(SL)}_2][\text{TFSA}]\) to \([\text{Li(SL)}_3][\text{TFSA}]\) (Fig. 2e and f). This means that, despite high Li salt concentrations similar to the glyme systems, the cation–cation and cation–anion motions are found to be less anti-correlated for \([\text{Li(SL)}_3][\text{TFSA}]\) and nearly non-correlated for \([\text{Li(SL)}_2][\text{TFSA}]\). The postulated Li ion hopping/exchange condution mechanism through the unique SL-bridged, chain-like Li-ion coordination structure may be responsible for the insignificant anti-correlations even at high Li salt concentrations as well as a large contribution of \(\sigma_{\text{self}}\) to \(\sigma_{\text{ion}}\).\(^{17,18}\)

In addition, the \(\sigma_{\text{distinct}}/\sigma_{\text{ion}}\) of the shorter glyme-based and the SL-based electrolytes is more negative than that of \([\text{Li(G3)}][\text{TFSA}]\) and \([\text{Li(G4)}][\text{TFSA}]\). This suggests that the momentum of bulky TFSA anions is predominantly self-compensated in these molten Li salt solvates with the labile Li ion coordination structures. The negative contribution of \(\sigma_{\text{distinct}}\) significantly offsets the positive contribution of \(\sigma_{\text{self}}\), and thus, \(\sigma_{+-}\) has only a minor effect on \(\sigma_{\text{ion}}\) in these molten Li salt solvates.

### 3.3 Effects of dynamic correlations of the ions on Li\(^+\) transference number

Herein, we discuss the influence of the dynamic ion correlations in the molten Li salt solvates on the Li\(^+\) transference number. The Li\(^+\) transference number determined from the potentiostatic polarisation method (\(t_{\text{Li}}^{\text{PP}}\)) is considered to be equivalent to that estimated from the MD simulations under anion-blocking conditions (\(t_{\text{Li}}^{\text{b}}\)) and is rationalised with the Onsager transport coefficients as follows:\(^{25}\)

\[
t_{\text{Li}}^{\text{PP}} = \beta^2 - 2\alpha + 4\alpha^2 / 4(1-x)(\beta-1) \tag{5}
\]

\[
x = \frac{\sigma_{++}}{\sigma_{++} + \sigma_{--}}, \quad \beta = \frac{2\sigma_{+-}}{\sigma_{++} + \sigma_{--}} \tag{6}
\]

The parameter \(x\) must be equal to \(t_{\text{Li}}^{\text{b}}\) for non-correlated ideal systems, but is greatly affected by the cation–cation cross-correlation for the molten Li salt solvates, whereas the parameter \(\beta\) is dependent on the cation–anion cross-correlation. The parameters \(x\) and \(\beta\) for the molten Li salt solvates are listed in Table 2. Based on eqn (5), \(t_{\text{Li}}^{\text{PP}}\) was simulated versus \(\beta\) for different \(x\) values, as in Fig. 3; in this figure, the experimental values of \([\text{Li(G4)}][\text{TFSA}]\), \([\text{Li(G1)(G2)}][\text{TFSA}]\) and \([\text{Li(SL)}_3][\text{TFSA}]\) are also plotted.

For the glyme-based electrolytes, the \(x\) value of \([\text{Li(G3)}][\text{TFSA}]\) and \([\text{Li(G4)}][\text{TFSA}]\) is significantly lower than that of \([\text{Li(G1)}_2][\text{TFSA}]\) and \([\text{Li(G1)(G2)}][\text{TFSA}]\). This is mainly attributed to the stronger anti-correlated motions of Li ions in the molten Li solvates with longer glymes (Fig. 2a and b). The decreased \(\sigma_{+-}\) due to the highly negative \(\sigma_{\text{distinct}}\) also contributes to higher \(x\) values for \([\text{Li(G1)}_2][\text{TFSA}]\) and \([\text{Li(G1)(G2)}][\text{TFSA}]\). The strength of the cation–anion anti-correlations is well manifested in the order of \(\beta\) values – \([\text{Li(G4)}][\text{TFSA}]\) with the most pronounced cation–cation anti-correlations has the lowest \(\beta\) value of \(-0.83\). Similar \(\beta\) values (\(-0.87\) for \([\text{Li(G4)}][\text{TFSA}]\) and \(-\sim 0.5\) for \([\text{Li(G1)}_2][\text{TFSA}]\) and \([\text{Li(G1)(G2)}][\text{TFSA}]\)) were reported in the MD simulation study.\(^{26,33}\) As seen in Fig. 3, both low \(x\) and \(\beta\) values are responsible for the extremely low \(t_{\text{Li}}^{\text{PP}}\) for \([\text{Li(G4)}][\text{TFSA}]\). Consequently, we experimentally confirmed Bedrov’s notion that \(t_{\text{Li}}^{\text{PP}}\) can increase for shorter glyme systems with the weaker Li ion–solvent interaction, in which the momentum conservation-derived ionic anti-correlations of cation–cation and cation–anion motions were suppressed to some extent by the fast solvent (momentum) exchange, leading to the higher \(x\) and \(\beta\) values.

For the SL-based electrolytes, the \(x\) values are comparable to that of the shorter glyme-based electrolytes, but the \(\beta\) values are higher than that of the shorter glyme-based electrolytes. The high \(x\) value arises from not only the large contribution of \(\sigma_{\text{self}}\) (due to higher \(D_{\text{Li}}\) than \(D_{\text{anion}}\)), but also the significantly less-pronounced cation–cation anti-correlations. In a comparison between the shorter glyme based electrolytes and the SL-based electrolytes with the high \(x\) values of \(-0.9\) (Fig. 3), the higher \(\beta\) value (due to weaker cation–anion anti-correlations) for the SL-based electrolytes exerts an efficient contribution to \(t_{\text{Li}}^{\text{PP}}\) becoming high. As a result, the Li ion hopping/exchange mechanism through the labile SL-bridged, chain-like Li-ion coordination structure may reduce the cation–cation and cation–anion anti-correlations, and impart the high \(x\) value and the \(\beta\) value closer to zero, enabling a high \(t_{\text{Li}}^{\text{PP}}\).

### Table 2 \(x\) and \(\beta\) of the molten Li salt solvates

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<thead>
<tr>
<th>Sample</th>
<th>(x)</th>
<th>(\beta)</th>
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<tbody>
<tr>
<td>([\text{Li(G3)}][\text{TFSA}])</td>
<td>0.25</td>
<td>-0.78</td>
</tr>
<tr>
<td>([\text{Li(G4)}][\text{TFSA}])</td>
<td>0.29</td>
<td>-0.83</td>
</tr>
<tr>
<td>([\text{Li(G1)}][\text{TFSA}])</td>
<td>0.88</td>
<td>-0.43</td>
</tr>
<tr>
<td>([\text{Li(G1)(G2)}][\text{TFSA}])</td>
<td>0.93</td>
<td>-0.37</td>
</tr>
<tr>
<td>([\text{Li(SL)}_3][\text{TFSA}])</td>
<td>0.89</td>
<td>-0.19</td>
</tr>
<tr>
<td>([\text{Li(SL)}_2][\text{TFSA}])</td>
<td>0.92</td>
<td>-0.18</td>
</tr>
</tbody>
</table>

![Fig. 3 Plots of \(t_{\text{Li}}^{\text{PP}}\) versus \(\beta\) with different \(x\) values. The data for \([\text{Li(G4)}][\text{TFSA}]\), \([\text{Li(G1)(G2)}][\text{TFSA}]\) and \([\text{Li(SL)}_3][\text{TFSA}]\) are also plotted in the figure.](image-url)
The simulation of $t_{PP}^{\text{Li}}$ with different parameters of $x$ and $\beta$ (Fig. 3) provides an important guideline for designing liquid electrolytes with high $t_{PP}^{\text{Li}}$, and suggests that single ion conductors ($t_{PP}^{\text{Li}} \sim 1$) can be prepared even with low molecular weight solvents and Li salts when liquid electrolytes have appropriate values of $x$ ($> 0.5$) and $\beta$. For instance, one can expect an even higher $t_{PP}^{\text{Li}}$ for the SL-based electrolytes if the $\beta$ parameter approaches zero. This might be achieved either by decreasing the interactions between Li ions and SL solvents to facilitate solvent-derived momentum exchange (e.g., partial fluorination of SL), or by increasing the cation–anion interactions to some extent (e.g., use of more associative Li salt than LiTFSA). Both would contribute to an increase in $t_{PP}^{\text{Li}}$. Indeed, $t_{PP}^{\text{Li}}$ increased from 0.68 for [Li(SL)$_2$][TFSA] to 0.77 for [Li(SL)$_2$][BF$_4$] with repriming weakly coordinating, bulky TFSA anions by more strongly Lewis basic, smaller BF$_4$ anions.

4. Conclusions

$t_{PP}^{\text{Li}}$ of the glyme- and SL-based molten solvates was determined by potentiostatic polarisation method. The dynamic ion correlations of each electrolyte were also experimentally studied on the basis of Rolling and Bedrov’s concentrated solution theory, and the effects of these correlations on $t_{PP}^{\text{Li}}$ were discussed. For [Li(G3)][TFSA] and [Li(G4)][TFSA], the negatively large value of $\sigma_{\text{distinct}}/\sigma_{\text{ion}}$ and $\sigma_{\ldots/\sigma_{\text{ion}}}$ indicates strong anti-correlations of cation–cation and cation–anion motions, and may be attributed to the momentum conservation of the [Li(glyme)]$^+$ complex cation and TFSA$^-$ anions, preserved solely by the momentum exchange of the ions. These results verified that [Li(G3)][TFSA] and [Li(G4)][TFSA] showed ionic liquid-like ionic transport behaviour, which in turn led to the extremely low value of $t_{PP}^{\text{Li}}$. For [Li(G1)$_2$][TFSA] and [Li(G1)(G2)][TFSA], $\sigma_{\text{distinct}}/\sigma_{\text{ion}}$ and $\sigma_{\ldots/\sigma_{\text{ion}}}$ were found to be less negative, and the $t_{PP}^{\text{Li}}$ values was higher than those of [Li(G3)][TFSA] and [Li(G4)][TFSA]. This was probably due to the more frequent solvent exchange between the unstable [Li(glyme)]$^{\pm}$ complex cations. Overall, our experimental results on $t_{PP}^{\text{Li}}$ and the dynamic ion correlations agreed reasonably with the results obtained from the MD simulations. For the SL-based electrolytes, $\sigma_{\text{distinct}}/\sigma_{\text{ion}}$ and $\sigma_{\ldots/\sigma_{\text{ion}}}$ were even less negative and approached zero, indicative of the insignificant anti-correlations of cation–cation and cation–anion motions. It was suggested that the Li ion hopping/exchange conduction mechanism through the SL-brided, chain-like Li-ion coordination structure plays an important role in suppressing the cation–cation and cation–anion anti-correlations and displaying a high $t_{PP}^{\text{Li}}$. As a result, the difference in the ion–solvent interactions and the Li ion coordination structure of the molten Li salt solvates has a significant impact on the cross-correlations of ion motions, which in turn become a critical determinant for $t_{PP}^{\text{Li}}$. The present study on dynamic ion correlations unveiled the difference in the ion conduction mechanism in some of the molten Li salt solvate electrolytes and its relevance to Li$^+$ transference number. Further systematic studies on the ion dynamics in different electrolyte systems would offer insight into a rational design concept for single ion conducting liquid electrolytes.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

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