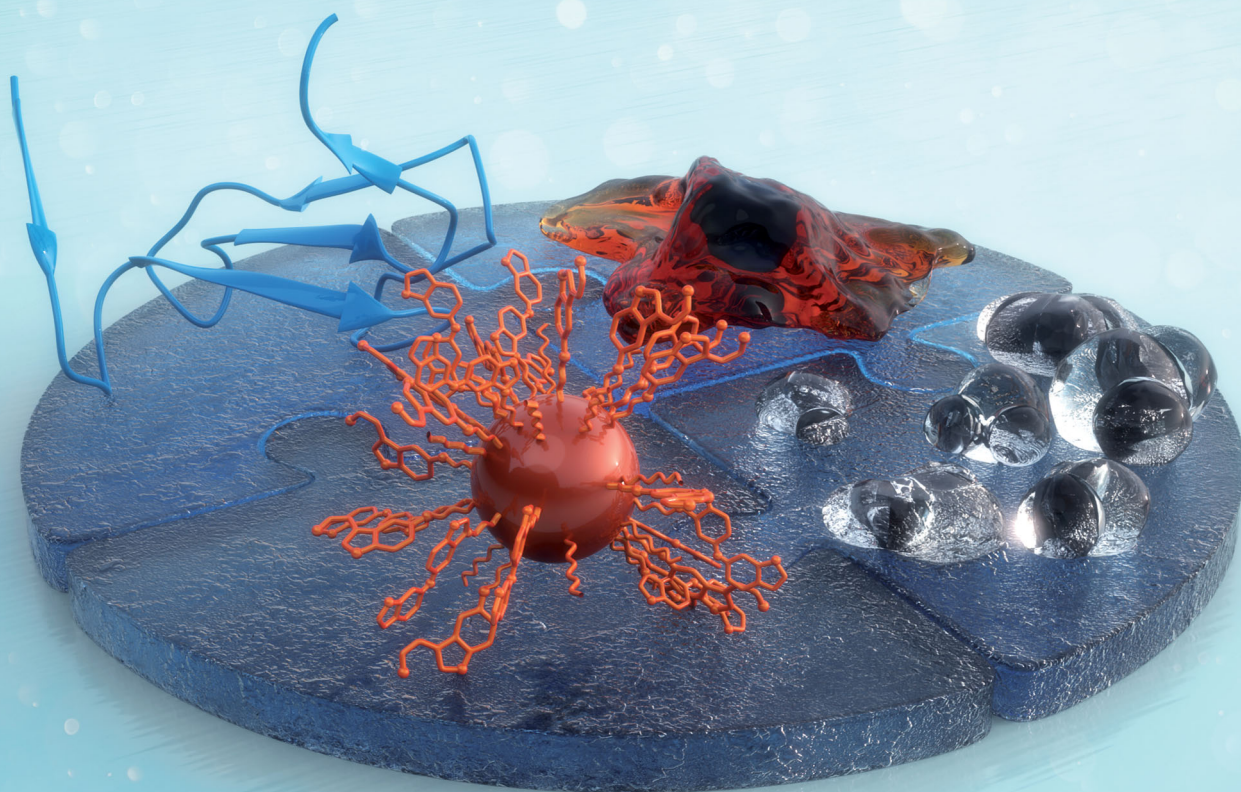


# Chem Soc Rev

Chemical Society Reviews

rsc.li/chem-soc-rev



ISSN 0306-0012

**REVIEW ARTICLE**

Håvard J. Haugen *et al.*  
Biological responses to physicochemical properties  
of biomaterial surface



Cite this: *Chem. Soc. Rev.*, 2020, **49**, 5178

## Biological responses to physicochemical properties of biomaterial surface

Maryam Rahmati,<sup>a</sup> Eduardo A. Silva,<sup>b</sup> Janne E. Reseland,<sup>a</sup> Catherine A. Heyward<sup>c</sup> and Håvard J. Haugen<sup>ib</sup> \*<sup>a</sup>

Biomedical scientists use chemistry-driven processes found in nature as an inspiration to design biomaterials as promising diagnostic tools, therapeutic solutions, or tissue substitutes. While substantial consideration is devoted to the design and validation of biomaterials, the nature of their interactions with the surrounding biological microenvironment is commonly neglected. This gap of knowledge could be owing to our poor understanding of biochemical signaling pathways, lack of reliable techniques for designing biomaterials with optimal physicochemical properties, and/or poor stability of biomaterial properties after implantation. The success of host responses to biomaterials, known as biocompatibility, depends on chemical principles as the root of both cell signaling pathways in the body and how the biomaterial surface is designed. Most of the current review papers have discussed chemical engineering and biological principles of designing biomaterials as separate topics, which has resulted in neglecting the main role of chemistry in this field. In this review, we discuss biocompatibility in the context of chemistry, what it is and how to assess it, while describing contributions from both biochemical cues and biomaterials as well as the means of harmonizing them. We address both biochemical signal-transduction pathways and engineering principles of designing a biomaterial with an emphasis on its surface physicochemistry. As we aim to show the role of chemistry in the crosstalk between the surface physicochemical properties and body responses, we concisely highlight the main biochemical signal-transduction pathways involved in the biocompatibility complex. Finally, we discuss the progress and challenges associated with the current strategies used for improving the chemical and physical interactions between cells and biomaterial surface.

Received 8th February 2020

DOI: 10.1039/d0cs00103a

[rsc.li/chem-soc-rev](http://rsc.li/chem-soc-rev)

### 1. Introduction

In the case of severe tissue injuries, the body is not able to successfully repair the injured tissues.<sup>1</sup> Owing to the emergence of tissue engineering and regenerative medicine fields, many promising treatment strategies are currently available for repairing and/or replacing damaged tissues.<sup>2,3</sup> However, there is no doubt that, if not all, most of these approaches are dependent on using chemical principles for designing biomaterials as biological substitutes that mimic and/or stimulate tissue functions. The American National Institutes of Health defines a biomaterial as “any substance or combination of substances, other than drugs, synthetic or natural in origin, which can be used for any period of time, which augments or replaces partially or totally any tissue, organ or

function of the body, to maintain or improve the quality of life of the individual”.<sup>4,5</sup> Biomaterials engineering is a highly interdisciplinary research field in which scientists (mostly with a background in chemistry) introduce biological alternatives for replacing or enhancing tissue and/or organ functions.<sup>1,6</sup> Over the past few decades, chemical scientists and engineers have achieved substantial progress in designing promising biomaterials as key diagnostic or therapeutic solutions for several disorders.<sup>7–9</sup>

Although considerable effort is devoted to developing biomaterials as successful tissue replacements for clinical applications, most of the suggested strategies fail to match the functional properties of targeted tissues *in vivo*, due to their poor biocompatibility.<sup>1</sup> The nature of the interaction of biomaterials with the surrounding biological microenvironment defines their biocompatibility. There is still a critical gap in successfully matching the biomaterial surface physicochemical characteristics to biochemical signal-transduction pathways *in vivo*. This gap could be owing to our poor understanding of biochemical signaling pathways, lack of reliable techniques for designing biomaterials with optimal physicochemical properties,

<sup>a</sup> Department of Biomaterials, Institute of Clinical Dentistry, University of Oslo, 0317 Oslo, Norway. E-mail: [h.j.haugen.odont.uio.no](mailto:h.j.haugen.odont.uio.no)

<sup>b</sup> Department of Biomedical Engineering, University of California, One Shields Avenue, Davis, CA 95616, USA

<sup>c</sup> Oral Research Laboratory, Institute of Clinical Dentistry, University of Oslo, 0317 Oslo, Norway



and/or poor stability of biomaterial properties after implantation.<sup>10–13</sup> The designed biomaterials for tissue engineering applications should have a strong affinity to targeted cells by sending chemical and physical signals to stimulate neo-tissue formation. Establishing strong positive interactions between the biomaterial surface and cells is entirely dependent on both the materials and targeted biological system chemistry.<sup>14</sup>

From the biochemical point of view, the features of cellular niche are highly important. The cellular niche is a highly complex tissue-specific microenvironment within a particular anatomic location providing physicochemical signals for cell communication.<sup>15</sup> Different mechanotransduction, macromolecular adsorption and biochemical signaling pathways, which can be dependent on the tissue type, play key roles in determining the material's success after implantation. The main

biochemical signaling pathways of local innate immune cells and their receptors, the other neighboring tissues or factors around the targeted tissue, and the biological systems a material might face are very diverse in different tissues.<sup>16,17</sup>

From the materials engineering point of view, each physicochemical property of the biomaterial surface (such as topographical features, stiffness, functional groups, and interfacial free energy) can profoundly affect biochemical mechanisms (Fig. 1). In addition, the commonly applied techniques and chemical strategies for modifying the surface properties can influence biomaterial–cell interactions.

Despite several reviews in the literature that address the importance of surface properties in regulating cell responses,<sup>10,11,13,18–20</sup> none of the recently published reviews have comprehensively discussed the vital roles of chemistry in



**Maryam Rahmati**

*Maryam Rahmati is a PhD Research Fellow in Tissue Regeneration at University of Oslo. She is currently working on the correlation between recently developed chemical and biological imaging techniques for analyzing body responses to biomaterials. She is doing her PhD in Prof. Haugen's research group and her projects are funded by European Training Network within the framework of Horizon2020 Marie Skłodowska-Curie Action (MSCA). She was awarded a master's degree in biomaterials science from Materials and Energy Research Center, Tehran, Iran, 2016. She worked as a biomaterials and tissue engineer at Iran University of Medical Sciences, Tehran, Iran, 2016–2018.*



**Eduardo A. Silva**

*Eduardo Silva is an Associate Professor of Biomedical Engineering at the University of California, Davis. Dr Silva obtained his degree in Metallurgical and Materials Science Engineering and PhD in Bioengineering from the University of Porto. Dr Silva has been studying polymeric biomaterials, including alginate and chitosan, for over 15 years. The long-term goal of his current research is to engineer biomaterials for controlled delivery of cells, drugs and/or genes. Dr Silva received several honors and awards, including the Hellman Family Fellow and the Biomaterials Emerging Investigator award. He has 8 patents or patent applications and Novartis recently licensed one of his patents.*



**Janne E. Reseland**

*Reseland became an associate professor in 2005 and a professor in Biomaterials at the Department of Biomaterials in 2007. She is currently (from 2010) the head of the Oral Research Laboratory.*

*Dr Janne E. Reseland obtained her PhD in biochemistry from the University of Oslo (UiO) in 1995. She was a postdoc fellow and research scientist at the Faculty of Medicine, UiO, for a period of 8 years. In 2003 Reseland started working on the development of stable extracellular matrices as novel therapeutics for biomimetic induction of hard tissue growth at the Institute for Clinical Dentistry, Faculty of Dentistry, UiO.*

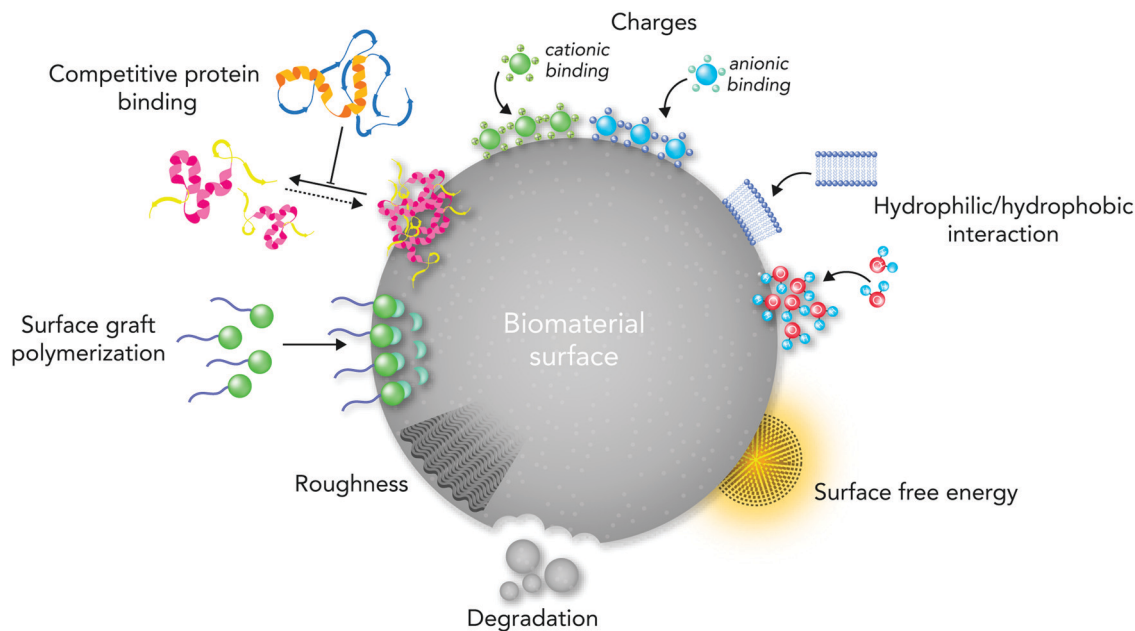


**Catherine A. Heyward**

*works at the Institute for Clinical Dentistry at the University of Oslo, where she is responsible for histology and imaging of a range of biomaterial samples.*

*Catherine A. Heyward completed her MBiochem at Oxford University in 2001, and PhD in lipid signalling under the supervision of Prof. Michael Wakelam at the University of Birmingham in 2006. Since then she has worked on live cell confocal imaging for the study of numerous proteins. She has also trained others in the preparation and imaging of samples at the Institute for Biological Sciences at the University of Oslo. She currently*





**Fig. 1** An illustration of the key surface physicochemical properties in directing biological responses to biomaterials. Biomaterials can manipulate molecular and cellular signaling pathways through their surface physicochemical properties (e.g. topography, stiffness, functional groups, biological moieties, ions, charges, and surface free energy).

regulating biological pathways, manipulating biomaterial surface properties, and directing molecular and cellular responses after biomaterial implantation. In addition, it is time to provide an updated state-of-the-art and future perspective for researchers in this field based on the recent chemical, physical and biological research findings. This review aims at emphasizing the key roles of chemistry in determining the biocompatibility of biomaterials by presenting an overview of both biochemical and chemical engineering principles and challenges in designing biocompatible systems. As biochemical signaling pathways of the immune system are critical factors in determining the success of

biomaterials, we briefly highlight the main biochemical signaling mechanisms and concepts of biocompatibility. Then, we address the current progress and challenges in biological responses to biomaterial surface physicochemical properties such as topographical features, functional groups, interfacial free energy, ion enrichment, and biological moieties. Although we use biomaterials in different tissue engineering applications, reliable evaluation of biological responses is still a big challenge. Altogether, this review provides an overview of the progress and challenges of each part to the readers; however, due to the complex nature of biological responses to biomaterials, not all related issues are possible to discuss here.



**Håvard J. Haugen**

*Professor Haugen is the leader of Biomaterials group, Faculty of Dentistry, University of Oslo. He received his master's degree in chemical engineering from Imperial College, UK, in 2001, and his PhD in biomaterials from Technische Universität München, Germany, in 2004. He worked as a scientist at the Central Institute for Medical Engineering, Munich, Helmholtz Institute for Biomedical Engineering, Aachen and the Tissue Engineering Centre of Imperial*

*College, London. Haugen has been awarded many research grants and innovation awards from both the European Research Council and the Research Council of Norway. Haugen was the past President of the Scandinavian Society for Biomaterials.*

## 2. Using biomaterials for tissue regeneration applications

The self-renewal potential of tissues decreases or completely disappears over time due to several reasons such as increasing age, reducing the amount and capability of host stem cell/progenitor populations, naturally poor repair potential of tissues, or undesirable inflammatory responses in damaged tissues and/or organs.<sup>21,22</sup> Tissue engineering and regenerative medicine approaches represent a clinically appealing and promising strategy to repair biological processes associated with injured tissues.<sup>22</sup> In the past few decades, scientists have used various cell types as key elements in different tissue regeneration therapies.<sup>23–25</sup> However, if cells are transplanted freely into the body, only a small proportion might reach the targeted tissue.<sup>26</sup>

Biomedical scientists use naturally occurring chemical processes as an inspiration to design new biomaterials. Different



classes of biomaterials are designed to offer suitable micro-environments for enhancing cell engraftment, including both naturally occurring and synthetic polymers, ceramics, metals and composites (Fig. 2).<sup>26,27</sup> Implanted biomaterials in tissue engineering are categorized generally into two main groups: (i) auto-, allo- or xeno-based cellularized or decellularized scaffolds known as natural/physiological polymers (*e.g.* proteins, polysaccharides and decellularized tissue matrices) and (ii) other materials such as synthetic polymers, implants, ceramics and composites.<sup>5</sup> Chemical strategies can be employed for designing a wide range of naturally occurring and synthetic biomaterials while stimulating cells to secrete and deposit the native extracellular matrix (ECM) locally.<sup>28–30</sup> The substantial progress in chemical and tissue engineering fields has led to the existence of smart biomaterials as promising therapeutic solutions for several devastating disorders. Nowadays, we clinically use biomaterials as valid therapeutic candidates for various tissue regeneration applications such as musculoskeletal system,<sup>31</sup> cardiovascular system,<sup>32</sup> neural system,<sup>33</sup> and skin.<sup>34</sup> In addition, biomaterials can affect the results of regenerative medicine strategies such as cell-based therapies, and engineered living tissues or organs.<sup>35</sup>

For successfully using biomaterials in the medicine world, designed biomaterials should be able to enhance the cell survival and functions after transplantation as well as stimulate autologous tissue growth.<sup>36,37</sup> The designed biomaterials for tissue regeneration applications should provide provisional mechanical support and mass transport to stimulate biochemical signaling pathway functions toward tissue healing.<sup>38</sup> Additionally, biomaterials could increase the success of tissue regeneration by sending physicochemical signals with spatiotemporal precision toward cells.<sup>39</sup> With this concept, a biomaterial is dynamically involved in providing some physicochemical cues to targeted cells resulting in neo-tissue formation.<sup>40</sup> For initiating biochemical signaling pathways, considering the presence of soluble signaling molecules such as growth factors and cytokines is also important.<sup>41</sup>

On the other hand, scaffolds designed from one material type would not be able to meet the requirements for tissue regeneration applications, which is owing to the absence of a

controlled degradation rate, optimal physicochemical properties, and stimulating ideal biochemical signaling pathways.<sup>42,43</sup> Thus, composite biomaterials designed by combining the chemistries of different materials tend to exhibit greater success in stimulating tissue regeneration after implantation.<sup>44,45</sup> Manipulating the biomaterial surface physico-chemistry based on the targeted site is essential for achieving optimal biological performance. Indeed, selecting biomaterials for tissue engineering applications is reliant on their physicochemical surface properties such as surface roughness,<sup>46</sup> architecture,<sup>47</sup> charge,<sup>48</sup> energy,<sup>49</sup> and functional groups.<sup>50</sup> Hence, the effects of each physicochemical surface property on the biological performance of biomaterials should be precisely investigated *in vitro* and *in vivo*.<sup>37</sup>

### 3. The evolution of the definition of host responses

In the early 20th century, a prodigious revolution took place in both therapeutic and diagnosis strategies through designing synthetic biomaterials by manipulating the chemistry of materials.<sup>51</sup> Since naturally occurring chemistry was used for designing biomaterials, modifying and/or proving their biological safety were among the most challenging issues in this field. Some decades ago, James Anderson defined foreign body reactions to biomaterials by demonstrating short- and long-term inflammatory responses to biomaterials and the substantial roles of macrophages in each step.<sup>16,52–54</sup> Owing to Anderson group's work, the biomedical scientists' understanding of molecular and cellular responses to biomaterials increased so that these days at the time of designing each biomaterial its effects on the foreign body responses determine its biocompatibility.

Although this definition favors non-degradable inert biomaterials, it cannot thoroughly define the body responses to the recently developed biomaterials with bioactive degradable surfaces suitable for tissue regeneration.<sup>10,11,13,55</sup> Owing to the advances in chemistry, the recently developed biomaterials are designed and formulated to stimulate different biochemical signaling pathways. In these cases, we could not define

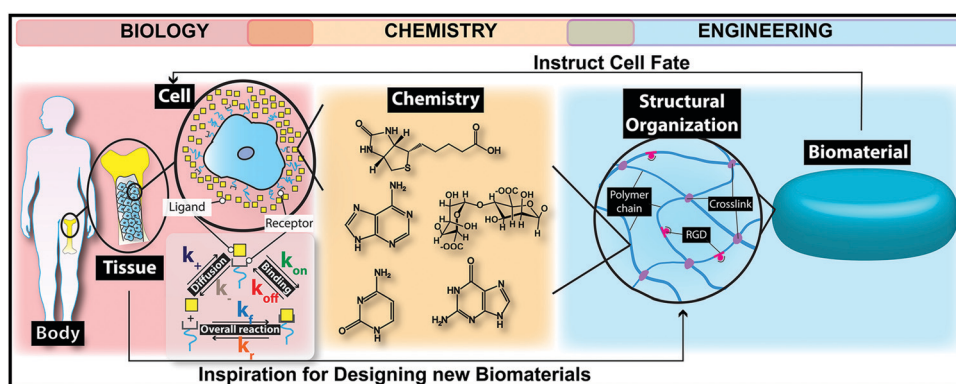


Fig. 2 An illustration of the role of chemistry in bridging the gap between biomaterials engineering and biology. The chemistry-driven processes in the body have inspired biomedical engineers to fabricate biomaterials using chemical strategies. Chemistry promotes tissue regeneration through designing different biomaterials for stimulating cells to deposit the native extracellular matrix (ECM).



biocompatibility as only not having any adverse effects.<sup>56</sup> The designed biomaterials should have a strong affinity for targeted cells to stimulate biochemical signaling pathways toward the neo-tissue formation. This ability is entirely dependent on the specific chemical characteristics of both the material system and the biological environment of targeted tissue.<sup>14</sup>

The biomaterial surface physicochemical properties such as charges, functional groups, biological moieties, and ion enrichment play key roles in directing biological responses to biomaterials.<sup>14,56</sup> From the biochemical perspective, different mechanotransduction, physiological, macromolecular adsorption and biochemical signaling pathways are crucial, which can be different from tissue to tissue. Because different tissues and cells have different chemical signals and physical characteristics, it is hard to say whether a material compatible with one tissue will make positive interactions with other cell types and tissues.<sup>10,13</sup> In addition, although both innate and adaptive immune systems respond to biomaterial implantation, their biochemical cues are different from each other, which requires evaluating their responses individually.<sup>57</sup>

## 4. The classical perspective of biological responses to biomaterials

The host responses to biomaterials mainly originate from biochemical signals and cues. As our knowledge in the biochemistry field has tremendously grown since the first definition of biocompatibility, we should update our definitions regarding host responses to biomaterials. Here we briefly discuss the out-of-date concept of cellular responses and biochemical signaling pathways involved in foreign body responses to biomaterials. Then, we provide an overview of the recently updated biochemical signaling pathways in the next sections.

Foreign body responses to implanted biomaterials are generally defined as a sequence of body reactions, which start instantly after biomaterial implantation.<sup>14,56</sup> With this concept, after biomaterial implantation, the tissue injury stimulates several chemical signaling cascades, which result in a sequence of acute and chronic inflammatory as well as wound healing responses.<sup>58,59</sup> Protein adsorption, neutrophils, and type 1 macrophages direct the acute inflammatory phase. This phase is essentially responsible for provisional matrix formation and wound site cleaning, which can take from some hours to days.<sup>60</sup>

After the release of some biochemical cues, blood vessels start expanding and consequently more blood flows into the injured area. Some blood and tissue proteins (such as fibronectin, fibrinogen, vitronectin, complement C3, albumin and growth factors) as well as leukocytes are released, which adhere to the blood vessel endothelium.<sup>59–61</sup>

Proteins are made from 20 natural amino acids. A linear chain of amino acid residues is called a polypeptide. A protein contains at least one long polypeptide. Short polypeptides, containing less than 20–30 residues, are rarely considered as proteins and are commonly called peptides, or sometimes oligopeptides. Each amino acid has a general backbone

network of  $\{-\text{NH}-\text{C}\alpha\text{HR}-\text{CO}-\}$ , where R describes a specific side group structure that gives the amino acid its specific functional properties. Based on the R structure, the amino acids are divided into three main types: nonpolar, polar, and charged amino acids, in which each class has an affinity to surfaces with unique physicochemical properties.<sup>14,62</sup> Furthermore, the size of proteins affects their adsorption to the biomaterial surface. Small proteins move faster and are responsible for the primary adsorption on surfaces. However, the larger proteins have higher affinity to the surface, which is owing to their greater surface area.<sup>63</sup>

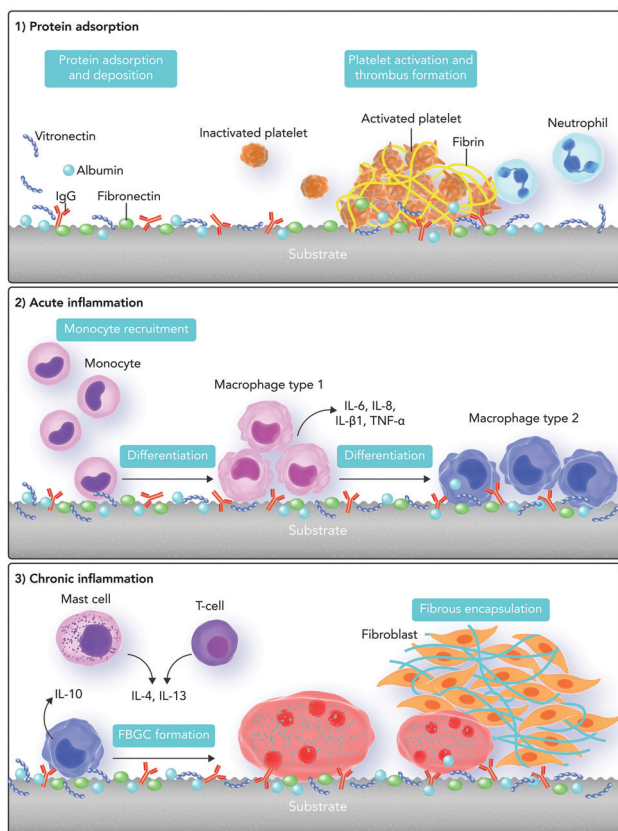
Moreover, the protein conformation defines its structure, bioactivity and communication with other biomolecules on the surface.<sup>64</sup> Most proteins have at least one active region to adsorb on the biomaterial surface, ligands, and receptors. The receptor domain of extracellular molecules accepts a signal from the upstream part and as a result changes its conformation, leading to stimulating the formation of a ligand binding domain.<sup>14</sup> These receptor binding proteins are connected into a chain, which transmit the biochemical signals across the cell membrane.<sup>14</sup> Researchers can achieve different protein–surface interactions through modifying the physicochemical properties of proteins.<sup>14</sup>

Monocytes released into the area differentiate into type 1 and type 2 macrophages. Type 1 macrophages are responsible for the acute inflammatory phase and release pro-inflammatory factors. On the other hand, type 2 macrophages are responsible for the chronic inflammatory phase and release anti-inflammatory factors.<sup>59,60,65,66</sup>

Monocytes, type 2 macrophages, and lymphocytes control the chronic inflammatory phase. During this phase, tissue granulation, fibroblast infiltration, and neovascularization occur, which can subsequently lead to the formation of blood vessels and connective tissue to allow wound healing to proceed.<sup>66,67</sup> In the wound healing phase, the proliferation of fibroblasts and vascular endothelial cells changes the fibrin clot into an extremely vascularized granulation tissue. The presence of several growth factors is important in this phase including platelet-derived growth factor, fibroblast growth factor, transforming growth factor- $\beta$ , transforming growth factor- $\alpha$ /epidermal growth factor, interleukin-1 (IL-1), and tumor necrosis factor.<sup>68–70</sup> Fibroblasts are also active in synthesizing collagen and proteoglycans, which lead to replacement of the granulation tissue with ECM (Fig. 3). Depending on the severity of injury at the implanted site, tissue type, and biomaterial properties, the acute phase takes less than one week and the chronic phase about two weeks.<sup>71,72</sup>

Based on this traditional definition of foreign body responses to biomaterials, the ability of a biomaterial to stimulate minimal inflammatory responses defines its success. Hence, the focus in designing biomaterials is on reducing foreign body responses through directing macrophage responses. However, because allowing natural body responses to occur is more useful for both biomaterial integration and function, this definition started to be redefined over the past few years.<sup>73–75</sup> The synchronization between inflammation and its resolution is essential for wound





**Fig. 3** An illustration of the traditional concepts regarding foreign body responses to the biomaterial surface. The foreign body responses are a combination of both acute and chronic phases of inflammation. The mechanism starts with protein adsorption and desorption (Vroman binding) on the surface of the biomaterial after its implantation. It continues with thrombin formation through activating platelets. After that, monocytes differentiate into type "1" macrophages which are responsible for the acute phase of inflammation. After some days, type "1" macrophages differentiate into type "2" macrophages which are responsible for chronic inflammation. T cells and mast cells also express cytokines that increase foreign body giant cell (FBGC) creation. In addition, FBGCs express fibroblast-recruiting factors and consequently by collagen deposition, a capsule starts forming around the biomaterial.

healing, which is dependent on the biochemical signaling pathways and cues.<sup>76</sup> To enhance the healing process, biomaterials are currently designed with a focus on improving their chemical interactions with immune system components.<sup>77–79</sup>

## 5. The role of innate and adaptive immune systems and biochemical cues in biological responses to biomaterials

The immune system is the main biological network, which releases biochemical cues responsible for protecting the body against foreign materials and keeping homeostasis. It consists of two main parts: innate and adaptive immune systems. Just after the instant recognition of foreign materials, the innate immune system causes a non-specific inflammatory response through a chain of biochemical reactions.<sup>16,77,80,81</sup>

Responsible cells in the innate immune system consist of phagocyte cells (including dendritic cells, monocytes, and macrophages) and lymphocytes (natural killer cells, gamma delta T-cells, and innate lymphoid cells).<sup>16,81,82</sup> However, the adaptive immune system is responsible for showing particular antigen responses and making a long-term memory through B and T lymphocytes.<sup>16,81,83</sup>

A suitable immune system response requires organized crosstalk between these two systems, where chemical cues are intrinsically present and play pivotal roles. After biomaterial implantation, the degradation products and subsequent chemical surface changes of biomaterials can stimulate the immune system.<sup>71</sup> The interactions between the surface and the immune system are reliant on the targeted tissue nearby the biomaterial causing tissue-specific biochemical responses.<sup>81</sup> After biomaterial implantation, the native vasculature is likely to be disrupted, which could induce interactions between blood and the implanted biomaterial.<sup>56</sup>

Depending on the biomaterial surface physicochemistry, the plasma constituents including proteins, lipids, sugars, and ions can be adsorbed on it.<sup>77</sup> Platelets, which through aggregation and coagulation form a fibrin-rich clot, are also a part of the blood exudate. The formed clot is a temporary provisional matrix for supporting cellular and molecular functions.<sup>84</sup> The adsorbed proteins elicit biochemical signaling pathways and make interactions with the innate immune system cells such as neutrophils, monocytes, fibroblasts and endothelial cells through their particular recognition sites including C-termini, N-termini, proline-histidine-serine-arginine-asparagine (PHSRN) and arginine-glycine-aspartic acid (RGD).<sup>85–87</sup>

Neutrophils are commonly the first responders to foreign materials. These cells are stimulated when the adsorbed proteins (RGD, PHSRN), microbes (pathogen associated molecular patterns or PAMPs), and/or dead cell residues (damage-associated molecular patterns or DAMPs) bind to their ligands through biochemical reactions.<sup>87–90</sup> The adsorbed proteins bind to macrophage type 1 antigen, lymphocyte function-associated antigen 1, and integrin alphaXbeta2. However, DAMPs and PAMPs bind to toll-like receptors (TLRs) and some specific pattern recognition receptors (PRRs), which also exist on the surface of macrophages and dendritic cells.<sup>87,91</sup>

Neutrophils stimulate the expression of cytokines as pro-inflammatory chemical mediators through sending biochemical signals.<sup>92,93</sup> These chemical mediators stimulate directed chemotaxis of other innate inflammatory cells and dendritic cells, which leads to the stimulation of adaptive immunity responses through B and T lymphocytes.<sup>77,94</sup>

DAMPs are endogenous molecules that under normal physiological conditions are sequestered intracellularly and cannot be recognized by the innate immune system.<sup>95</sup> Nevertheless, under cellular stress or injury conditions, they are released into the extracellular environment leading to the transmission of biochemical signals to cells for initiating inflammatory responses under sterile conditions.<sup>95</sup> The DAMP release from cells depends on the type of cell injury or death. Chromatin-associated protein, high-mobility group box 1, heat shock



proteins, and purine metabolites are prototypical DAMPs derived from damaged cells.<sup>95–98</sup> Furthermore, ECM degradation can send biochemical signals for releasing DAMPs. DAMPs can initiate inflammatory responses, and the lack of DAMPs in the environment leads to a decrease of inflammatory biochemical cues.<sup>99</sup>

There are different types of PRRs in the innate immune system that stimulate the expression of various types of pro-inflammatory cytokines and biochemical markers. According to the subcellular location of PRRs, they are classified into two main groups: (i) TLRs and C-type lectin receptors, which are transmembrane proteins, and (ii) RIG-I-like receptors (retinoic acid-inducible gene-I-like receptors, RLRs) and NOD-like receptors (NLR), which exist in the intracellular compartments. PAMPs and DAMPs activate these receptors and subsequently inflammasome complexes.<sup>100,101</sup>

The inflammasome complex contains a cytosolic sensor that can be a PRR of the NLR, absent in melanoma 2 (AIM2) receptors, and an effector protein.<sup>102</sup> There are various types of PRRs, which can form inflammasomes such as NLRP1, NLRP3, NLRC4 (the NLR family of intracellular proteins) and AIM2.<sup>103–105</sup>

In response to PAMPs and DAMPs, the pentameric or heptameric assembly of PRRs can oligomerize the caspase recruitment domain in filaments. This might cause the inflammasome formation through stimulating caspase-1.<sup>106</sup> The NLRP3 inflammasome is the most known inflammasome, which contains the NLRP3 scaffold, caspase recruitment domain adaptor protein, caspase-1, and accessory protein serine/threonine-protein kinase.<sup>107,108</sup> Monocytes, macrophages, granulocytes, dendritic cells, epithelial cells and osteoblasts mainly express this inflammasome.<sup>109</sup>

After cellular injury through biomaterial implantation, DAMPs and PAMPs activate the NLRP3 inflammasome through sending biochemical signals.<sup>110,111</sup> Examples of such stimuli from the DAMP group are crystalline matter such as asbestos, calcium influx, mitochondrial reactive oxygen species (ROS), and extracellular neurotransmitter adenosine triphosphate (ATP).<sup>112</sup> However, this process is not yet fully understood and needs further detailed studies.<sup>73</sup> The inflammasome can through subsequent control over the rest of immune response processes either cause the resolution of inflammation and tissue regeneration, or lead to chronic inflammation and fibrosis.<sup>113</sup> After inflammasome expression, the migrated monocytes/macrophages adhere to the temporary provisional matrix formed on the biomaterial surface.<sup>114</sup>

After 24 to 48 hours, the activated neutrophils die through apoptosis and release some vesicles and lipids through biochemical signals (*e.g.* lipoxins and resolvins), which have anti-inflammatory influences.<sup>87,115,116</sup> Hence, neutrophils through binding to PAMPs and DAMPs and initiating inflammasome responses are vital for activating type 1 macrophages and the acute inflammatory phase. Apoptotic neutrophils are also crucial for stimulating macrophage polarization from type 1 to type 2 and the following inflammation resolution. Therefore, if their lifespan is extended and/or if they increase in number at the biomaterial surface, chronic inflammation can occur at the site.<sup>117</sup>

After the polarization of type 1 macrophages to type 2, they locally release several growth factors (such as transforming growth factor beta and vascular endothelial growth factor) while stimulating fibroblast and endothelial cell migration and proliferation by sending biochemical signals. Fibroblasts produce collagen to form the ECM, whereas endothelial cells nourish the formation of new blood vessels to offer essential nutrients for neo-tissue formation as well as for waste removal.<sup>118</sup> In the chronic inflammatory phase, T lymphocytes, mainly helper T cells, play key roles in controlling the expression of pro- and anti-inflammatory mediators.<sup>119</sup> In this system, B lymphocytes are responsible for making antibodies (Fig. 4).<sup>120</sup> Immune-modulatory biomaterials should direct biochemical signaling pathways and cues, which are responsible for the functions of neutrophils, PAMPs, DAMPs, inflammasomes, endothelial cells, and mesenchymal stem cells (MSCs).<sup>121</sup> As describing the details of innate and adaptive immune system mechanisms and the responsible biochemical cues is out of the scope of this review, we refer the readers to the following seminal review papers.<sup>16,73,77,80,95</sup>

## 6. The roles of ion channels in regulating immune system responses

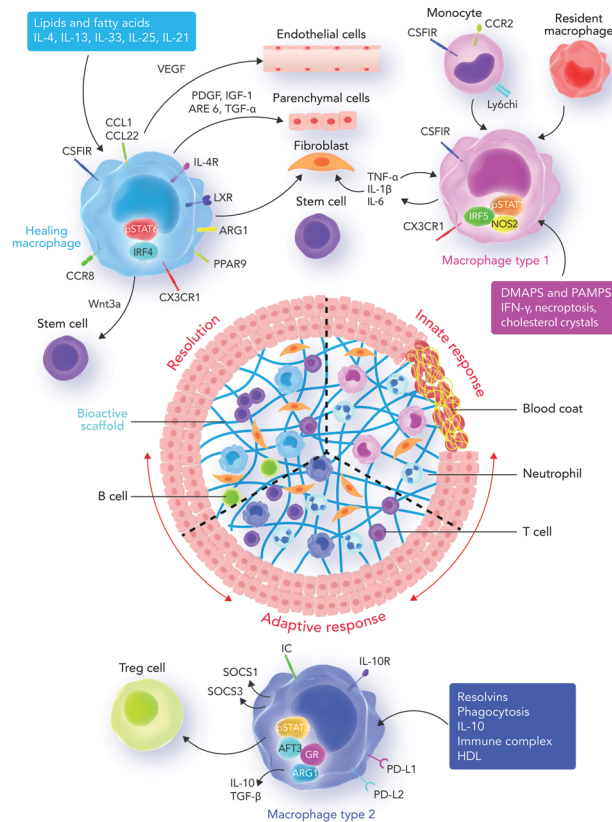
Cell surface receptors play key roles in receiving biochemical signals (from chemical substances such as hormones, growth factors, cell adhesion molecules, nutrients, and neurotransmitters) and initiating biochemical and/or biophysical signaling in the cells.<sup>123–126</sup> Ion channels are a class of surface receptors, which control many cellular signaling events in cells.<sup>127–129</sup> Ion exchanges between the intra and extracellular environments create the mechanisms essential for controlling the cell metabolism and activation state.<sup>130</sup> In addition, ion channels are important regulators of cell–cell communication. As a result, genes encoding proteins responsible for regulating membrane permeability to ions are also vital in most of the complex intra and extracellular signaling events.<sup>130</sup> Because of the key roles of immune cells in controlling foreign body responses, we discuss some ion channels that regulate innate and adaptive immune system responses here.

Ion channels direct immune responses mostly by regulating endosomal pH and intracellular calcium concentrations.<sup>131,132</sup> Regulating the intracellular calcium amounts is dependent on the biophysical properties of the ion channels and their ability to control the calcium passage across the membrane.<sup>130</sup> The calcium permeability can be changed by activating particular ligands, feedforward responses to the calcium release from intracellular stores, changes in cell polarization, and the strength of sodium driving force.<sup>130</sup>

In adaptive immune system cells (B- and T-lymphocytes), regulating the intracellular calcium amount is important as releasing calcium from intracellular stores activates the immune response pathways.<sup>133</sup> In addition, the CAV1 (caveolin 1) ion channel (as a subfamily of L-type voltage-gated calcium channels) is vital in activating B- and T-lymphocytes.<sup>134,135</sup>







**Fig. 4** Innate and adaptive immune system responses. Recruited and resident macrophages start experiencing marked phenotypic and functional changes in response to damage-associated molecular patterns (DAMPs), pathogen associated molecular patterns (PAMPs), growth factors, cytokines, and other mediators released into the interface area. The main phenotypic changes regulate inflammation, tissue repair, regeneration, and resolution. Macrophages then express different types of factors which can direct different functions in fibroblasts, epithelial cells, endothelial cells, and stem and progenitor cells to promote tissue repair. During the final stages of the healing process, a regulatory pro-resolving phenotype, which confirms the suppression of the tissue-damaging inflammatory response, is expected. If the process does not successfully proceed, persistent inflammation and/or maladaptive repair processes can cause tissue-destructive fibrosis. Sometimes, the recruited monocytes lead to the formation of a resident macrophage phenotype in tissues.<sup>122</sup> Abbreviations: DAMP, damage-associated molecular pattern; PAMP, pathogen-associated molecular pattern; Treg cell, regulatory T cell; IRF5, interferon regulatory factor 5; NOS2, nitric oxide synthase 2; LXR, liver X receptor; AREG, amphiregulin; Arg1, arginase-1; IRF4, interferon regulatory factor 4; PPARγ, peroxisome proliferator-activated receptor γ; FGF, fibroblast growth factor; GAL-3, galectin-3; TGF, transforming growth factor; GR, glucocorticoid receptor; ATF3, activating transcription factor 3; SOCS, silencer of cytokine signaling.

Increasing ROS activates transient receptor potential melastatin (TRPM) 2 ion channels.<sup>136</sup> The TRPM2 activation causes the release of calcium from immune cells. In addition, TRPM2 has a key role in activating the NLRP3 inflammasome causing the expression of cytokines and chemokines from immune cells.<sup>136</sup>

Some studies have reported the importance of ion channels in regulating microglia functions as the resident macrophage cells of the central nervous system.<sup>137,138</sup> In microglia, the P2X and *N*-methyl-D-aspartate (NMDA) receptor families respond

to the neurotransmitters adenosine triphosphate (ATP) and glutamate, respectively.<sup>139</sup> By regulating the intracellular calcium concentration, these receptors can affect microglial activation. P2X receptors (P2XRs) are trimeric plasma membrane channels, permeable to small inorganic cations (e.g. Na<sup>+</sup>, K<sup>+</sup>, and Ca<sup>2+</sup>).<sup>127,140</sup> However, some P2XR channels are permeable to both cationic and anionic organic ions.<sup>141</sup> Ferreira *et al.*<sup>142</sup> studied the Ca<sup>2+</sup>-activated K<sup>+</sup> channel (KCa3.1)-dependent responses in microglia under ROS.<sup>142</sup> They concluded that increasing the cyclic guanosine monophosphate (cGMP) concentration leads to protein kinase activation and, subsequently, ROS formation in mitochondria. The ROS formation causes endoplasmic reticulum calcium release, which subsequently binds to calmodulin to activate the KCa3.1 channel.<sup>142</sup>

Connexin and pannexin cell–cell channels, unopposed hemichannels as well as P2 receptors are essential in initiating and regulating the inflammatory responses.<sup>143</sup> For instance, the activation of connexin and pannexin channels leads to the release of ATP and other metabolites to the extracellular media. Extracellular ATP can stimulate intracellular signaling pathways by acting on P2 receptors, which leads to inflammation.<sup>143</sup>

Overall, the activation of ion channels can be “danger” signals propagating the inflammatory responses of immune systems.<sup>143</sup> Their biochemical effects on cell homeostasis affect the immune system functions.<sup>133</sup> Therefore, the activation of ion channels can be vital in directing the host responses to biomaterials. However, more research on understanding the ion channel roles in regulating signaling pathways and directing cell–biomaterial interactions is vital.

## 7. The role of mesenchymal stem cells in biological responses to biomaterials

MSCs have many roles in modulating the immune system responses to implanted biomaterials, particularly in biochemical signaling pathways responsible for stimulating the innate immune system.<sup>144,145</sup> These cells can have immunosuppressive roles by releasing several soluble biochemical factors responsible for controlling the functions of lymphoid and myeloid cells.<sup>145–147</sup> For example, prostaglandin E2 (PGE2) synthesized by MSCs can stimulate macrophages to have an adapted directing phenotype by increasing IL-10 and decreasing tumor necrosis factor-α and IL-12 expression.<sup>148</sup> The biochemical soluble factors released by MSCs (e.g. IL-10, PGE2 and IL-1b) can play vital roles in the crosstalk between MSCs and macrophages, mainly in macrophage type 1 to 2 polarization.<sup>149</sup> In addition, interferon gamma and tumor necrosis factor-alpha cytokines expressed from T cells can stimulate macrophage polarization by stimulating MSCs to release cyclooxygenase-2 and indoleamine 2,3-dioxygenase.<sup>150</sup>

MSCs can also control T regulatory lymphocytes (Tregs) and T helper-based immunosuppressive activities by releasing heme oxygenase-1 and its metabolic by-product carbon monoxide.<sup>145,151</sup> Because type 2 macrophage polarization is linked with Tregs stimulation, MSCs are vital in controlling the crosstalk between



innate and adaptive immune systems.<sup>145,151</sup> These cells can down-regulate the expression of some lymphocyte growth factors, differentiation of antigen presenting cells and effector T cells as well as epithelial cell proliferation by releasing PGE2.<sup>145</sup> The readers can find a good level of details concerning the role of MSCs in regulating immune system responses to foreign materials in these review papers.<sup>145,148,152</sup>

## 8. The role of mechanotransduction pathways in biological responses to biomaterials

The local microenvironment and physical forces surrounding the cells can play a crucial role in several physiological mechanisms including embryonic development, in adult physiology, and in a wide variety of different disorders and diseases. For example, in tissue development, the local physical forces can control dorsal closure, epithelial morphogenesis and skeletal growth, ECM remodeling, vascular inflammation as well as tissue regeneration processes.<sup>153–155</sup> In addition, at the cellular scale, cell-generated contractile forces can dictate both the cytoskeleton assembly and cellular architecture formation.<sup>155,156</sup>

Conversely, cells translate these mechanical stimuli into biochemical responses in a process that is typically referred to as mechanotransduction.<sup>157,158</sup> Therefore, the cell ability to sense the mechanical properties surrounding them is a key decision-making factor influencing cellular responses to biomaterials.<sup>159,160</sup> Fig. 5 shows how myosin motors and mechanosensors play a role in mechanotransduction pathways.<sup>11,161</sup> The cellular membrane is the main location of force transmission from the ECM to cells. When cells encounter a stiff substrate, several multiprotein complexes known as focal adhesions are activated and become the central hub of cell–ECM interactions.

The mechanosensing activity of focal adhesions includes recognizing and transporting mechanical signals from the ECM to the cellular cytoskeleton. Many of the focal adhesion complexes have both transmembrane and intracellular components. The intracellular layer is an interface between the transmembrane components and the actin cytoskeleton.<sup>157,158,162</sup> The molecular composition of the focal adhesion core can be very diverse and is mainly sensitive to the ECM composition and mechanics.<sup>163,164</sup> Focal adhesions are created after the assembly of transmembrane proteins for physical interactions with ECM components.

Chen W. *et al.*<sup>165</sup> reported that the “inside-outside signaling” mechanism inside cells or extracellular mechanical stimuli control integrin affinity for its ECM ligand.<sup>165</sup> The activated integrins assemble and strengthen the molecular links at the cell–matrix interface.<sup>166</sup> The ECM structure can elicit the expression of certain integrin subsets, which in combination with other biochemical signaling pathways can lead to particular cellular responses to physical forces.<sup>167</sup> Artola *et al.*<sup>168</sup> revealed that cells can adjust their force production to be ideal at tissues with different physiological conditions by controlling the expression of various integrin types.<sup>168</sup>

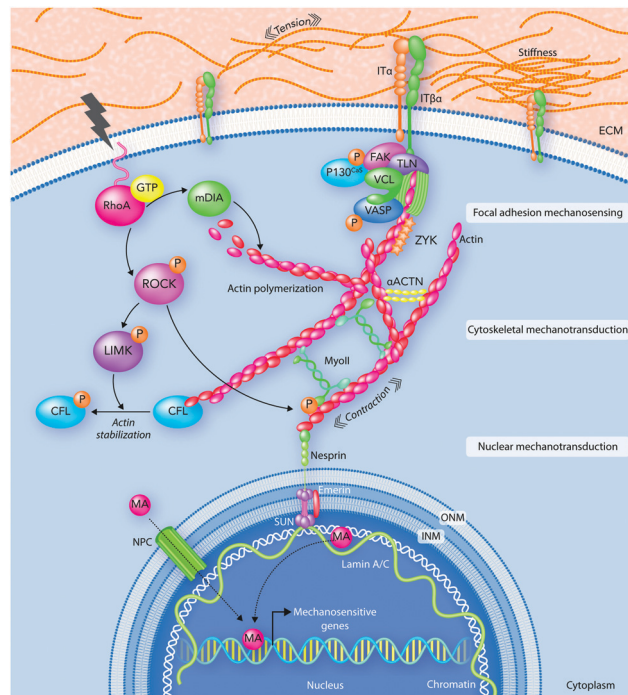


Fig. 5 An illustration of mechanotransduction pathways. Focal adhesions (FAs) interpret extracellular physical stimuli at the cell–ECM interface. Then, the received signals spread through the cytoskeleton and move to the nucleus where mechanosensors (MA) activate mechanosensitive genes. ACTN, actinin; CFL, cofilin; FAK, focal adhesion kinase; INM, inner nuclear membrane; IT, integrin; LIMK, LIM kinase; mDia, diaphanous-related formin-1; MyoII, myosin II; NPC, nuclear pore complex; ONM, outer nuclear membrane; PAX, paxillin; PS, perinuclear space; ROCK, Rho-associated protein kinase; TLN, talin; VASP, vasodilator-stimulated phosphoprotein; ZYX, zyxin.

In addition, cells can control their own mechanical properties through changing their cytoskeletal architecture, which is a dynamic network of filamentous and cross-linking proteins.<sup>169</sup> Cytoskeleton networks contain three main components including actin fibers, microtubules and intermediate filaments.<sup>170</sup> The F-actin sliding on the motor protein myosin II provides the cytoskeleton contractility.<sup>171</sup>

In summary, the mechanical properties of the cell micro-environment display a direct effect on several cellular functions after their translation to biochemical signals. Therefore, it is undeniable that the mechanotransduction pathways and their following biochemical signals play critical roles in directing host responses to biomaterials.<sup>11,161</sup> We will discuss the effects of biomaterial surface physicochemistry on the biochemical signals caused by mechanotransduction pathways in more detail in the subsequent sections.

## 9. Biomaterial strategies for directing biological responses

### 9.1. Impact of biomaterial surface physical properties on biological responses

The physical properties of the biomaterial surface can direct biophysical and biochemical signaling pathways involved in



cellular responses to the surface. However, the mechanisms involved in cell responses to these surface properties are not yet fully understood. In the following sections, we provide a brief overview of the current progress and challenges in this field.

**9.1.1. Biological responses to biomaterial surface topography.** Since 1945, the contact guidance term has been used to emphasize that topographical features of the biomaterial surface can control biochemical and biophysical signaling pathways.<sup>172</sup> Surface topographical parameters including both surface patterns and roughness<sup>173</sup> can affect the orientation of cells and stress fibers, which we discuss in detail here.

**9.1.1.1. Biological responses to feature size and geometry of biomaterial surface.** In the natural processes of tissue healing and/or regeneration, the curvature or topographical features of other surrounding cells and ECM guide the injured cell functions.<sup>174,175</sup> Hence, the surface topography of scaffolds can affect the cell fate determination, adhesion, polarization, and migration through manipulating physicochemical signaling pathways.<sup>176</sup> Topographical features including shape, size, and geometric structures can direct cellular functions through influencing either the cytoskeleton organization and protein orientation or protein unfolding.

Actin filaments can spread out on the 2D structure of flat surfaces; however, the curved surfaces offer a 3D network for cells to grow inside the material.<sup>177</sup> Rianna *et al.*<sup>178</sup> studied the mechanotransduction pathways and biochemical factors behind cell responses to topographical patterns through investigating the mechanical properties of peripheral and nuclear regions of cultured NIH-3T3 cells on azopolymer scaffolds with various topographical patterns. They designed micrometer scale patterns in either parallel ridge or square lattice geometry and then studied the mechanical cell responses by atomic force microscopy (AFM). Their results indicated that surface topographical features stimulate the cytoskeleton network to generate some forces, which affect nucleus functions.<sup>178</sup> Scientists and engineers used a wide range of strategies for improving the topographical features of biomaterials and therefore controlling cell functions in a non-invasive manner.<sup>179–181</sup> The topographical patterns can be represented either anisotropically by grooves and ridges or isotropically through random spreading of protrusions and pits.<sup>182</sup>

Regarding anisotropic patterns, studies investigate the alignment of cells alongside the anisotropic direction regardless of the topography scale. However, in isotropic topographies, which have more effects on cell signaling pathways, the topography size plays a key role in controlling cell responses to patterns.<sup>183,184</sup> Among the topographic parameters, the size of designed patterns, in both micro- and nano-scales, can play a vital role in controlling cell functions. The micro-scale patterns can profoundly influence the overall cell morphology; however, the nano-scale topographies are more critical in controlling the molecular and subcellular physicochemical sensing pathways.<sup>182,185</sup>

Padmanabhan *et al.*<sup>186</sup> investigated the role of surface topography size and stiffness of metallic glass nanorod arrays on cell–cell fusion. They revealed that the topographic features in nano-scale size can dominate biochemical signals in decreasing fusion through controlling cytoskeletal remodeling-associated signaling pathways.<sup>186</sup> Researchers focus on manipulating protein adsorption mechanisms and signaling pathways *via* designing substrates with nano-scale surface topography.<sup>187,188</sup> In addition, the nano-scale substrates can be used to answer basic questions concerning protein adsorption/desorption at the nano-scale. Wang *et al.*<sup>189</sup> developed a patterned poly(2-(dimethylamino)ethyl methacrylate) (PDMAEMA) brush with sub-100 nm structures over large areas by combining block copolymer micelle lithography and surface-initiated atom transfer radical polymerization (ATRP).<sup>189</sup> The PDMAEMA brushes were neutralized and collapsed at pH 9, while positively charged and swollen at pH 4. The authors studied bovine serum albumin (BSA) adsorption on PDMAEMA brushes using laser scanning confocal microscopy, AFM, and quartz crystal microbalance with dissipation (QCM-D). Because of the steady sub-100 nm topography of the patterned brushes, the authors could observe the protein adsorption mechanisms inside and outside of brushes using the AFM technique (Fig. 6).<sup>189</sup>

However, there are still some contradictions between the research results, which could be owing to the following reasons:

(i) Considering one individual defined scale for topographical features while biological environments are rich in physicochemical gradients.

(ii) Mostly, researchers focus on cell responses to one individual physical or chemical property. However, we recommend considering the synergistic effects of topographical and chemical properties of the surface on each other.

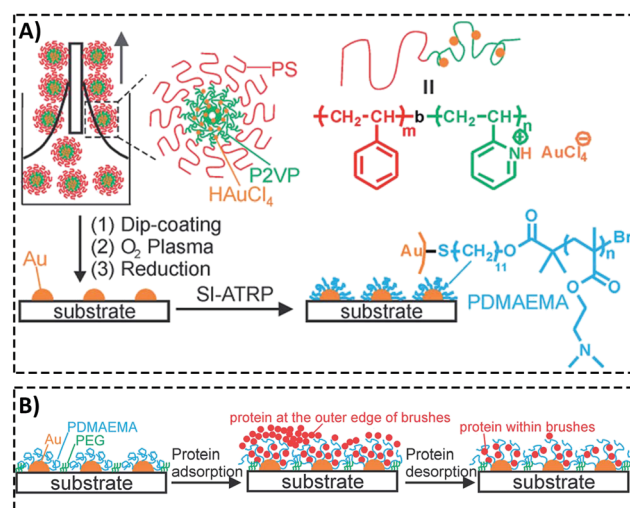


Fig. 6 (A) A representative graphic of fabricating patterned poly(2-(dimethylamino)ethyl methacrylate) (PDMAEMA) brushes with sub-100 nm structures over large areas using a combination of block copolymer micelle lithography and surface-initiated atom transfer radical polymerization (ATRP). (B) A schematic illustration of protein adsorption mechanisms inside and outside of brushes with a nanostructured surface. Reproduced from ref. 189 published by The Royal Society of Chemistry.



Surface nanofunctionalization has attracted much attention as a promising strategy for enhancing cell responses to biomaterials, which we will address later in this paper.

Liu *et al.*<sup>190</sup> designed some surface topography gradients through surface immobilization of gold nanoparticles in a density-dependent manner. They modified the surface chemistry of scaffolds *via* coating a thin plasma polymerized film with allylamine (AA) or acrylic acid (AC) chemical composition on the surface (Fig. 7A). They revealed that surface topography plays the main role in stimulating the initial cell adhesion and spreading. However, both topographical and chemical properties of the surface govern cell differentiation.<sup>190</sup> After culturing osteoblast-like SaOS-2 cells on surfaces, surface topography could enhance the stimulating effects of allylamine chemical treatment on osteogenic differentiation (Fig. 7B). Furthermore, in the natural *in vivo* conditions, the biological interactions with surface topographical features occur in an inhomogeneous and dynamic environment. These inhomogeneous dynamic interactions between micro/nano topographical patterns and molecules are complicated because local changes in other surface features, mainly chemistry, control attractive and repulsive forces on the surface.<sup>191</sup> Some strategies are available to design dynamic topographical features without disturbing environmental conditions or affecting the surface chemistry of scaffolds.

Hernandez *et al.*<sup>181</sup> suggested an *in vitro* approach for stimulating cells with dynamic topographical features of

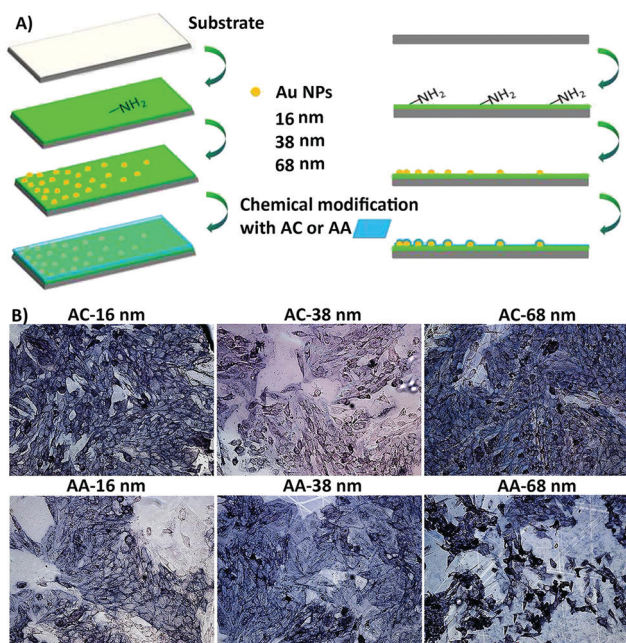


Fig. 7 (A) An illustration of the process to design the biomaterial with different surface topography gradient designed from three different sized gold nanoparticles (16, 38, 68 nm) coated with acrylic acid (AC) and allylamine as uppermost surface chemistry modification (AA) (*i.e.* AC 16, AC 38, and AC 68). (B) They cultured osteoblast-like SaOS-2 cells for seven days. The surface topography could improve the influence of AA chemical treatment on osteogenic differentiation, especially on AA 68 surfaces. The ALP expression of cells was highest at position 8 mm of AA. Scale bar = 100 μm. Reprinted from ref. 190. Copyright © 2018, Elsevier.

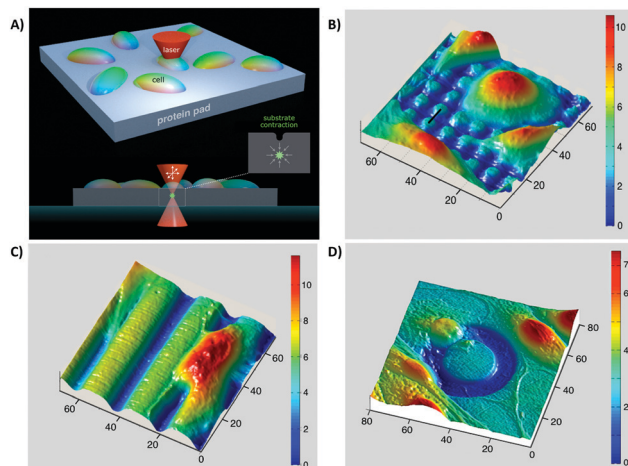


Fig. 8 *In situ* imprinting of topographical properties on protein-based hydrogel scaffolds. (A) Multiphoton excitation of the photosensitizer in the scaffold through a pulsed near-IR laser beam enhanced hydrogel contraction. Through limiting the excitation to 3D defined volumes, the authors could achieve imprinting without exposing cell or material surfaces to optical destruction. (B–D) *In situ* imprinting under cultured cells. The authors imprinted different topographies in the presence of cells such as micropost arrays (B), grooves (C), and annular depressions (D). All numbers are in units of μm. Reprinted with permission from ref. 181. Copyright © 2018, American Chemical Society.

protein-based hydrogel surfaces. They modified scaffolds *in situ* in real time by positioning a pulsed near-infrared laser focus inside a hydrogel, which leads to enhancement of the chemical cross-linking and consequently local contraction of the protein matrix. Fig. 8 shows that exposing hydrogels to a series of scan patterns can generate, remove, or retreat topographical patterns without having destructive effects on other surface properties or cell functions.<sup>181</sup> By using a laser-based confocal microscope technique, researchers can also design a synthetic scaffold capable of controlling cell orientation and migration in time and space without affecting surface chemistry.<sup>192</sup> However, stimulating spatiotemporal dynamic topographic parameters without affecting surface chemistry is still in its infancy and needs more research.

**9.1.1.2. Biological responses to biomaterial surface roughness.** Surface roughness relates to the texture of the biomaterial surface and is commonly represented with the roughness value ( $R_a/S_a$ ), which quantitatively represents the roughness grade.<sup>176,193</sup> The broadly used techniques for producing and controlling surface protrusions and/or depressions are blasting, electropolishing, nanoparticle/fiber formation, and nanofabrication technologies such as photolithography.<sup>176,194,195</sup> Using chemical surface treatments such as acid etching can further increase the surface roughness when compared with traditional machining techniques.<sup>196</sup> The biomaterial surface roughness can directly dictate specific cellular responses. For example, Barr *et al.*<sup>197</sup> studied the biocompatibility of 13 commercially available breast implants by focusing on macrophage responses to their roughness feature. They showed that macrophage responses to surface roughness determine the biocompatibility of implants.<sup>197</sup>



By considering the current assumptions about the effects of surface roughness on biological functions, one may identify several challenges:

(i) The first challenge relates to the other suggested parameters, which besides  $R_a$  can play key roles in determining surface roughness. Anselme and co-workers<sup>198–200</sup> showed that the fractal dimension ( $D$ ) and the developed surface can also be key parameters in determining surface roughness. The fractal dimension parameter can be useful in measuring surface disorder; however, the developed surface factor is related to the surface detachment index.<sup>198</sup> The mean distance between peaks ( $RS_m$ ), the sum of the average height of the five highest profile peaks and the average depth of the 5 deepest profile valleys calculated from the parallel line to the mean line ( $S_z$ ), kurtosis ( $S_{ku}$ ), skewness ( $S_{sk}$ ) and fluid core index ( $S_c$ ) are important roughness parameters.<sup>201–205</sup>

(ii) We cannot decide about host responses to biomaterials based on improving only one physicochemical property or (iii) testing with one or two cell types *in vitro*. However, several research groups reported designing biocompatible materials by only improving their roughness through mainly focusing on  $R_a$  measurement and *in vitro* testing with one or two cell types. Table 1 shows that increasing the surface roughness of one biomaterial can have positive effects on one cell type or protein (a phenomenon called rugophilia); however, it can have negative influences on other types. Increasing the surface roughness can decrease or not affect the proliferation and/or differentiation of leukocytes, keratinocytes, and monocytes; however, it can improve osteoblast proliferation on the surface.<sup>206–208</sup>

(iv) Even in one cell type, the surface roughness can affect different cell functions including cell adhesion, migration, proliferation, and differentiation in various ways. Increasing the surface roughness reduces the proliferation and increases the differentiation of osteoblasts.<sup>209–211</sup> When we evaluate host responses to biomaterials in tissues containing different cell types, the differences between responses of various cells to surface roughness can pose many challenges.<sup>212,213</sup>

(v) On the other hand, there is a considerable contradiction between the literature outcomes when it comes to one particular cell type response to surface roughness. These contradictions could arise from the current misuse of biocompatibility definition by ignoring critical biochemical signal transduction pathways, which indeed might play vital roles in determining cell responses.<sup>11,13</sup> For example, Saldana *et al.*<sup>214</sup> studied the role of mechanotransduction pathways in controlling human-MSC (hMSCs) responses to stainless steel surface roughness. In this study, the surface  $R_a$  was manipulated with 2 nm or 0.9  $\mu\text{m}$  for smooth and rough samples, respectively. Fig. 9A provides an overall perspective of the multitude of receptors and proteins that were involved in this study. It shows that under static conditions, improving stainless steel roughness increases the expression of biochemical markers including PGE2, vascular endothelial growth factor (VEGF), and receptor activator of nuclear factor kappa-B ligand (RANKL) as well as the phosphorylation of focal adhesion kinase (FAK) to its active form (Tyr-397).<sup>214</sup> Moreover, Fig. 9B indicates that applying tensile forces to the

plasma membrane of hMSCs improves VEGF secretion on smooth surfaces as well as PGE2 amounts and osteoprotegerin/RANKL proportion on both smooth and rough surfaces. Although mechanical stretch does not affect smooth surfaces, it stimulates FAK phosphorylation at Tyr397 on rough surfaces. Overall, showing the influence of stretch in enhancing FAK phosphorylation at Tyr397 on rough surfaces ( $R_a = 0.9 \mu\text{m}$ ) is evidence for the current hypothesis that mechanotransduction pathways can be substantial factors in directing cell responses to surface roughness.<sup>214</sup>

(vi) Although surface roughness can have positive or negative effects on cell responses in short-term periods (less than 48 h), its influence can change over time. Therefore, it is recommended to consider both short- and long-term cell responses to surface roughness. For example, Lee *et al.*<sup>209</sup> suggested a new strategy for controlling the sensitivity of different cell functions to surface roughness through using shape memory materials. They designed a shape memory (meth) acrylate copolymer with thermomechanical properties, which had a time-dependent dynamic surface change from smooth to rough under cell culture conditions. They used soft lithography techniques for making rough surfaces and then by applying compression decreased the surface roughness to generate smooth areas. Their results showed that under static conditions, surface roughness does not affect osteoblast amount, alkaline phosphatase specific activity (ALP), as well as osteoprotegerin and VEGF expression; however, it enhances osteocalcin expression. After three days of culture of cells on rough surfaces under dynamic conditions, surface roughness caused a decrease in DNA content and an increase in osteocalcin and osteoprotegerin expression.<sup>209</sup>

(vii) Another challenge involves the determination of the critical roughness value for each specific biomaterial type. The reported roughness value is different from study to study. Although a few statistical studies have revealed the critical roughness value on different surfaces,<sup>198,215</sup> it is still difficult to define one critical roughness value, which can properly guide cell functions. Therefore, researchers suggest decreasing dissimilarities between cell responses by considering an average roughness gradient, rather than individual numbers.<sup>201,216</sup> For instance, Zhou *et al.*<sup>216</sup> designed some polydimethylsiloxane (PDMS) substrates with surface roughness gradients by using a combination of microfluidics and photopolymerization techniques. They grafted *N*-isopropylacrylamide (NIPAM) with concentration gradients onto PDMS substrates, which produced a gradient of roughness ranging from  $2.6 \pm 0.7 \text{ nm}$  to  $163.6 \pm 11.7 \text{ nm}$  on the surface (Fig. 10A). The applied gradient improves cell attachment on the surface in both MSCs and hepatocellular carcinoma cell lines (HepG-2) (Fig. 10B).<sup>216</sup>

(viii) Designing biomaterials with multiscale surface roughness in both micro- and nano-scale can also improve cell responses. A roughness gradient in both micro- and nano-scale can improve interactions between various cell or protein types on the surface.<sup>217</sup>

(ix) The synergistic effects of roughness with other physical and chemical properties, which are different depending on the

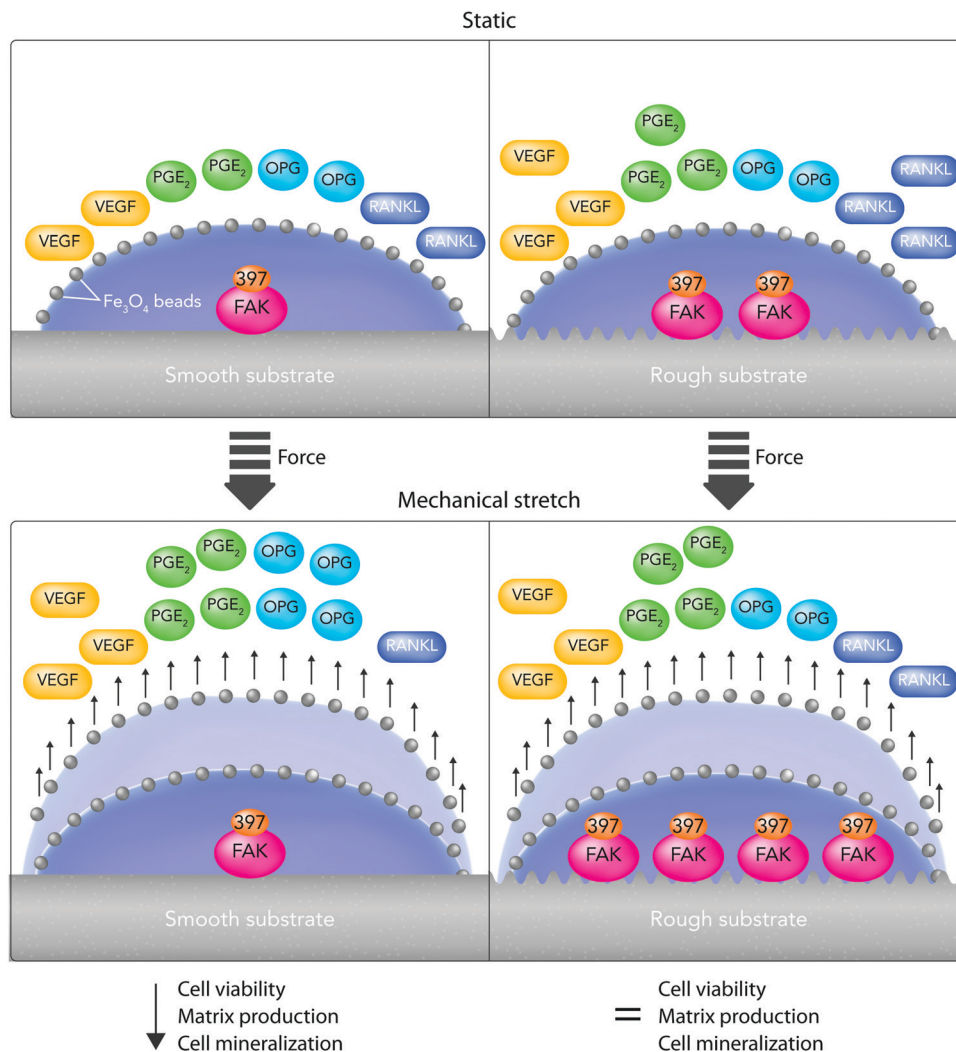


Table 1 An overview of the most recent studies on cell responses to the surface roughness of different biomaterials

Materials	$R_a$	Other modified physicochemical properties	Cell responses	Ref.
Silicon	1.07–80.03 $\mu\text{m}$	NA	Surface roughness in this range induces poor macrophage polarization and an innate potential to increase the pro-inflammatory response. Depending on the amount, $R_a$ has variable influences on inflammatory factors.	197
Polystyrene	~121, 505, 867 nm	NA	Rough surfaces with nano-meter dimensions make E-cadherin junctions and human gingival keratinocytes to grow gradually or imperfectly.	208
PCL	654 $\pm$ 91 nm	Hydrophilic surface	Increasing the initial cell attachment, proliferation, and differentiation of MG-63 cells.	223
PCL	~0.5–4.7 $\mu\text{m}$ & $RS_m \sim 214$ –33 $\mu\text{m}$	NA	Depending on the surface roughness gradient, faster, slower, or similar osteogenic commitment and gene expression of MSCs can be observed on rough surfaces in comparison with smooth surfaces.	201
PCL	1, 1.3, 2 $\mu\text{m}$	HA coating	Increasing the surface roughness increases osteoblast attachment and differentiation. Decreasing the activity of the osteoclast marker, tartrate-resistant acid phosphatase, by decreasing roughness.	213
PDMS & PNIPAM	2.6 $\pm$ 0.7–163.6 $\pm$ 11.7 nm	NA	Increasing the roughness causes a decrease in both amount and regions of MSC and HepG-2 attachment.	216
PLLA	1 $\times$ 1–20 $\times$ 20 $\mu\text{m}$	NA	Increasing the surface roughness increases fibroblast proliferation; however, it decreases osteoblast proliferation.	224
PHB	Pristine PHB = 32.9 nm Treated PHB = 270 nm	Laser surface treatment	Surface roughness can play a leading role in determining NIH 3T3 responses.	225
PEO & PI	$R_a$ = NA $R_q$ = 4–5 nm	Hydrophobicity	VGP WM115 cell adhesion forces are higher on surfaces with various hydrophobicity and roughness in comparison with WM266-4 cells.	226
Zirconia	1.7 & 3 $\mu\text{m}$	NA	1.7 $\mu\text{m}$ $R_a$ shows better osteoblast responses both <i>in vitro</i> and <i>in vivo</i> compared to 3 $\mu\text{m}$ $R_a$ .	227
PEEK/n-HA/CF	~0.09–2.95 $\mu\text{m}$ . $R_q \sim 0.17$ –3.64 $\mu\text{m}$ $R_a = 0.46$ –2.29 $\mu\text{m}$ , $R_q = 0.67$ –2.78 $\mu\text{m}$ , $R_z = 4.68$ 12.52 $\mu\text{m}$ & 0.9–1.7 $\mu\text{m}$ .	Surface treatment	Moderate surface roughness increases MG-63 cell attachment/proliferation. Suitable surface roughness increases bioactivity and osseointegration <i>in vivo</i> .	228
$\alpha$ -Tricalcium phosphate and $\alpha$ TCP Cap		1.5 wt% or 3.0 wt% of $C_2S$	Increasing MSC adhesion and proliferation by increasing surface roughness.	202
TiAl6V4	~0.30–1.80 $\mu\text{m}$	Chemical treatment with HA and $\beta$ -TCP	The influence of surface roughness depends on surface chemistry. HA causes higher mineralizing activity of MSCs at $R_a = 1.5$ $\mu\text{m}$ .	229
Ti6Al4V	0.114, 0.277, 0.316 $\mu\text{m}$	Surface treatment and melatonin	$\beta$ -TCP increases the osteogenic differentiation of MSCs when $R_a = 1.7$ $\mu\text{m}$ . Surfaces with $R_a$ in the 0.50–1.00 $\mu\text{m}$ range increase MC3T3-E1 cell proliferation.	230
Titanium	100–400 nm	NA	Increasing the surface roughness increases osteoblast adhesion after 24 h cell culture. Adding melatonin to the surface increases osteoblast proliferation.	196
Titanium	0.02–3.63 $\mu\text{m}$	NA	Increasing the surface roughness increases osteoblast differentiation, macrophage tendency to polarize to type 1 macrophages, and osteogenic ability.	231
Titanium	100 nm	NA	Surface roughness can have a combinational influence on osteoclast genesis and osteogenic differentiation of macrophages.	212
Stainless steel	SS = 2 nm & RS = 0.9 $\mu\text{m}$	NA	Increasing the surface roughness inhibits MC3T3-E1 cell functions through decreasing cell attachment, proliferation, and calcification ability.	232
TiAl6V4 & 316 L stainless steel	0.01–0.1 $\mu\text{m}$	NA	Mechanotransduction pathways play key roles in determining MSC responses to surface roughness.	214
			Surface roughness in this range has no effect on osteoblast adhesion mechanisms when roughness is isotropic and groove width is lower than a critical amount. It can only affect osteoblast orientation on wider grooves.	215

Abbreviations: smooth surface (SS), rough surface (RS), poly( $\epsilon$ -caprolactone) (PCL), not available (NA), a human osteosarcoma cell line (MG-63 cells), mean distance between peaks ( $RS_m$ ), CFPEEK-nanohydroxyapatite ternary composites (PEEK/n-HA/CF), root mean square average roughness ( $R_q$ ), a mouse osteoblast-like cell line (MC3T3-E1), N-isopropylacrylamide (NIPAM), polydimethylsiloxane (PDMS), a human liver cancer cell line (HepG2), tricalcium (TCP), dicalcium silicate ( $C_2S$ ), poly(L-lactide) (PLLA), mouse embryonic fibroblasts (NIH 3T3), human bone osteosarcoma cell (U-2 OS), polyhydroxybutyrate (PHB), two melanoma cell lines (VGP WM115 cells and (WM266-4 cells), polyethylene oxide (PEO), 1,4-polyisoprene (PI), calcium phosphates (CaP), hydroxyapatite (HA).





**Fig. 9** A summary of hMSC changes cultured on stainless steel substrates with different roughness, with 2 nm or 0.9  $\mu\text{m}$  for smooth and rough samples, respectively, after mechanical stimulation. The authors treated cultured cells with  $\text{Fe}_3\text{O}_4$  beads (black circles) and then subjected them to mechanical stretching (+Force). (A) Under static conditions, improving the roughness of stainless steel causes greater prostaglandin E2 (PGE2) and vascular endothelial growth factor (VEGF) expression. Focal adhesion kinase (FAK) phosphorylation to the active form (Tyr-397) is also improved. (B) The mechanical stimulation improves VEGF expression on smooth samples and PGE2 amounts as well as osteoprotegerin/receptor activator of nuclear factor kappa-B ligand (OPG/RANKL) proportion on both surfaces. The mechanical stimulation of cells enhances FAK phosphorylation degrees on rough surfaces; however, it does not affect smooth surfaces.

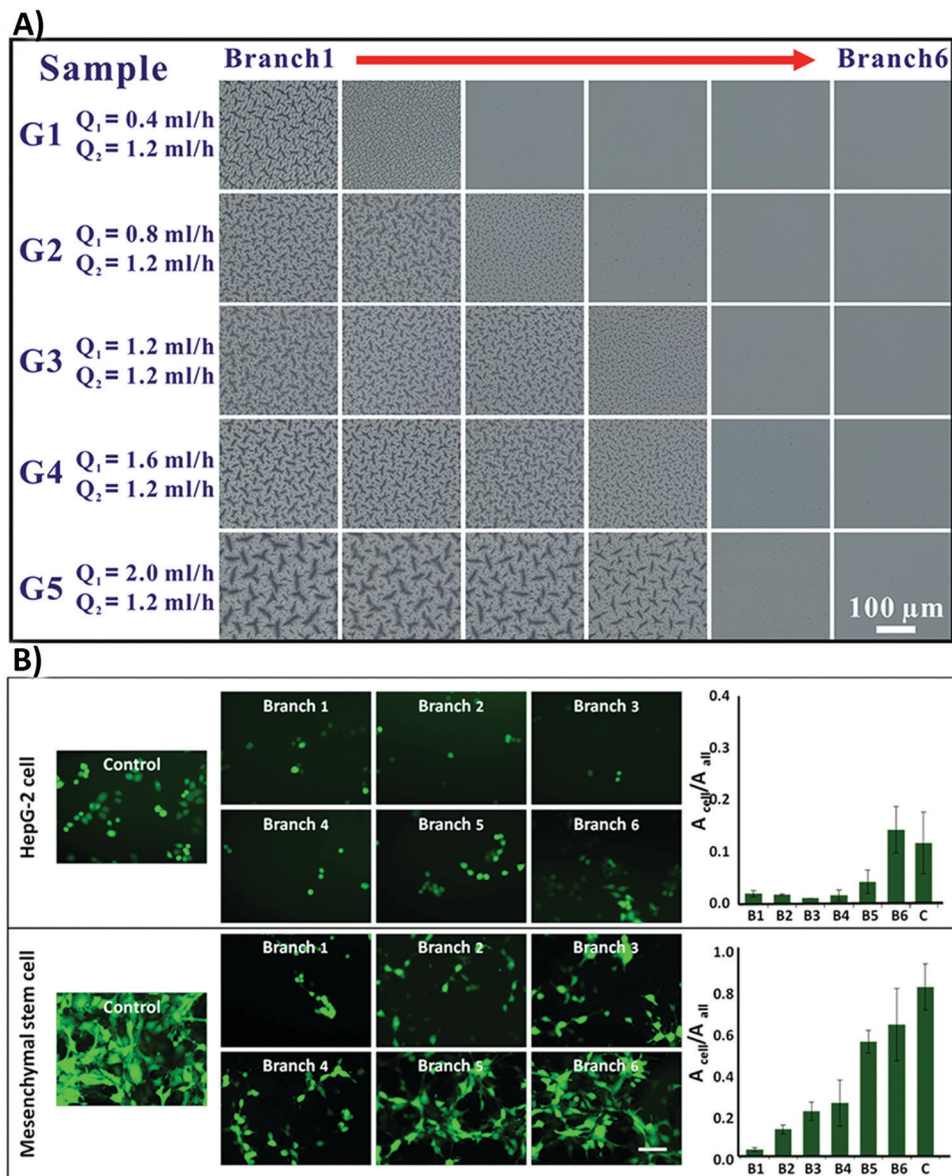
situation, can also cause contradictions. Rough surfaces with different topographies or stiffness have various influences on cell functions.<sup>218</sup> The same roughness value in two different types of materials can affect cell responses differently, which can be owing to the differences in their chemistry.<sup>219–222</sup> Fukuda *et al.*<sup>220</sup> investigated the osseointegration ability of a poly(ether ether ketone) implant by enhancing its surface roughness and/or surface chemistry (Fig. 11A). Fig. 11B shows that phosphorylation of the surface enhances cell responses on both smooth and rough surfaces, with more improvement on rough surfaces. However, only modifying the surface roughness cannot improve MSC responses. Fig. 11C shows the substantial effects of the combined surface modification strategies on bone regeneration in the rabbit tibia.<sup>220</sup> Both chemistry and roughness properties of the surface play roles in neo-tissue formation.

### 9.1.2. Biological responses to biomaterial surface stiffness.

The biomaterial stiffness defines the quantity of vital force, which is required for making some changes in the biomaterial surface and the surrounding environment. Because in the tissue engineering field stiffness and elasticity are often used interchangeably, here we use the stiffness term for discussing the mechanical properties of biomaterial surface. The surface stiffness can play key roles in regulating biochemical signaling pathways and cell behaviors including cell adhesion, spreading, migration, differentiation, and proliferation (Table 2).<sup>214,233</sup>

Navarrete *et al.*<sup>234</sup> studied the substantial effects of biomaterial surface stiffness in determining the MSC fate by investigating the MSC differentiation to osteoblasts and chondrocytes as two narrowly interconnected cell phenotypes. They designed four methyl acrylate/methyl methacrylate scaffolds





**Fig. 10** (A) The optical images of the polydimethylsiloxane (PDMS) surface grafted with *N*-isopropylacrylamide (NIPAM) with concentration gradients. A total of five samples were represented with flexible Q1 and fixed Q2 (microchannels were injected with red (Q1) and green (Q2) dye solutions to improve the contrast). Branch one contained the highest NIPAM concentration and branch six had the lowest. (B) Fluorescence images of MSC and HepG-2 cell attachment on the surface with various roughness grades. The authors stained HepG-2 cells with cell tracker dye (green) and MSCs were transfected with GFP (green).  $A_{\text{cell}}$  was applied to determine the total cell adhesion region; however,  $A_{\text{all}}$  was the total area of the vision ( $A_{\text{cell}}/A_{\text{all}}$  is the proportion between the total area of cell adhesion and the captured vision). The scale bar is 50  $\mu\text{m}$ . Reprinted with permission from ref. 216. Copyright © 2015, American Chemical Society.

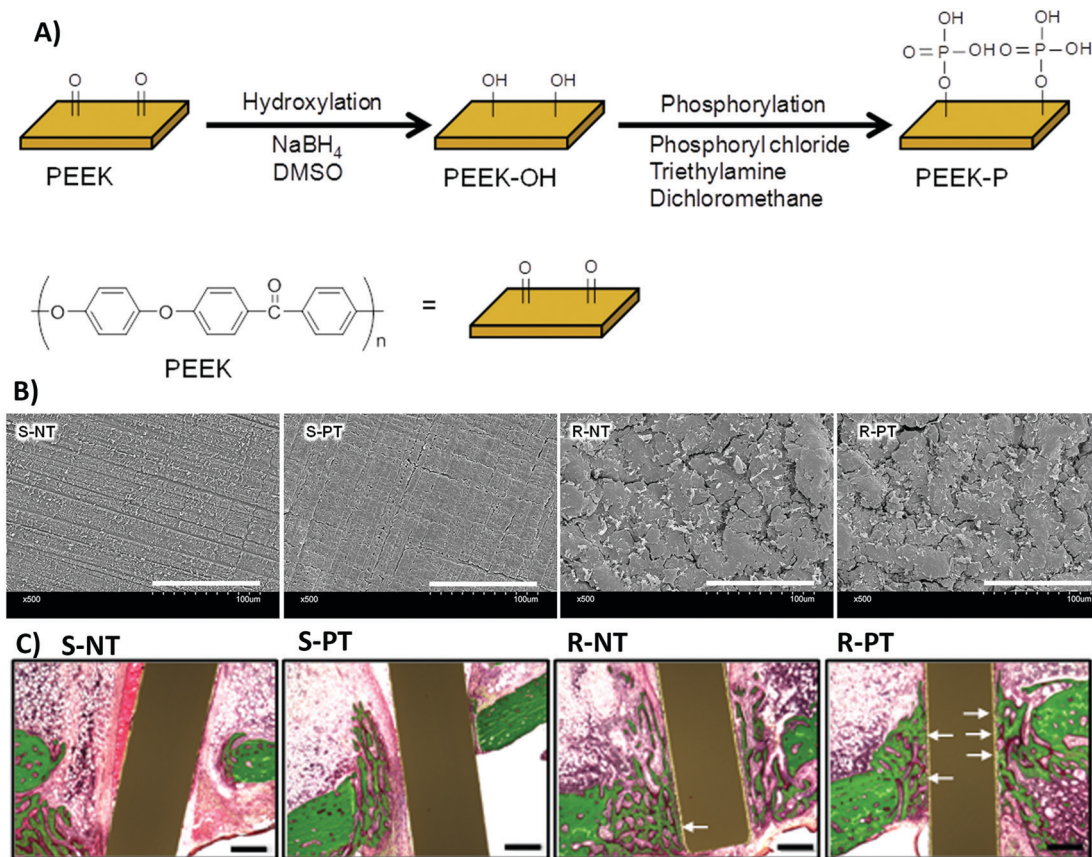
with elastic moduli ranging from 0.1 MPa to 310 MPa and then cultured cells on them. They reported that on softer surfaces, MSCs tend to increase the expression of chondrogenesis factors such as aggrecan, SOX9 (a chondrogenic transcription factor), type II collagen, and proteoglycan amount. However, the expression of osteogenesis factors including Runt-related transcription factor 2, ALP specific activity, osteocalcin, and osteoprotegerin decreases on softer surfaces. The expression of integrin subunits  $\alpha 1$ ,  $\alpha 2$ ,  $\alpha 5$ ,  $\alpha v$ ,  $\beta 1$ , and  $\beta 3$  is important in starting signaling pathways in response to the surface stiffness.<sup>234</sup>

Although it is clear that modifying the surface stiffness can affect cell responses, there are still some challenges, as described in the following paragraphs.

(i) The mismatch between surface stiffness and cells is a reason for foreign body responses. Moshayedi and co-workers<sup>235</sup> investigated the reason for glial cell activation and encapsulation of implanted electrodes. They reported that nervous glial cells are highly sensitive to the surface mechanical properties and adjusting the stiffness leads to decreased adverse reactions.<sup>235</sup> This is because of differences in the mechanical properties of various tissues (soft, like brain or ECM, to stiff, like bundled







**Fig. 11** (A) A schematic of poly(ether ether ketone) (PEEK) surface phosphorylation. (B) Scanning electron microscopy (SEM) images of the biomaterial surface. S-NT: unmodified PEEK, S-PT: phosphorylated PEEK with a smooth surface, R-NT: sandblasted PEEK, R-PT: phosphorylated PEEK with a sandblasted surface. (C) Biomaterial implantation in the rabbit tibia. Illustrative histological images of neo-tissue formation and osseointegration of samples four weeks after surgery. The white arrows show neo-tissue formation at the interface with substrates. Scale bars, 500  $\mu\text{m}$ . Reprinted from ref. 220. Copyright © 2018, Springer Nature in accordance with Creative Commons Attribution License.

collagen or bone).<sup>236</sup> At molecular levels, the stiffness of ECM components is also different from each other so that single collagen fibers are more rigid than fibrillary collagen structures.<sup>237</sup>

(ii) There are synergistic effects between stiffness, topography and/or chemistry of biomaterials so that surface stiffness also depends on the surface composition and topography.<sup>155</sup> The surface stiffness of biomaterials can be adjusted through directly altering the polymer ratio and cross-linker solution or treatment temperature and/or period.<sup>155</sup> More recently, researchers designed several covalently cross-linked hydrogels to investigate the synergistic effects of surface chemistry and stiffness in controlling cell adhesion and migration.<sup>238</sup> They made some scaffolds by synthetic coupling of ECM proteins to the surface of hydrogels. The results revealed that the surface biochemistry of substrates could influence cell responses to surface stiffness.<sup>238</sup>

(iii) The surface chemistry of biomaterials could explain the contradictions between research results as different types of biomaterials have different chemistries and consequently different stiffness. Li *et al.*<sup>239</sup> compared the roles of both bulk and interfacial stiffness of PDMS and polyacrylamide (PAAm)

scaffolds in guiding A549 cell behaviors.<sup>239</sup> They cultured cells on the surfaces of scaffolds with bulk stiffness ranging from 0.1 kPa to 40 kPa. On PAAm scaffolds, bulk stiffness directly affects the cell spreading speed. However, on PDMS scaffolds, the coated silica layer on the surface directs cell functions.<sup>239</sup> Because of the synergistic effects of biochemical and mechanical cues on cell responses, there are some challenges in clarifying the individual effects of each surface feature on cell responses.

Nii *et al.*<sup>240</sup> designed a 3D combinatorial hydrogel with individually adjustable biochemical and mechanical features to investigate the influence of interactive niche cues on the osteogenesis ability of adipose tissue-derived stem cells.<sup>240</sup> Their results indicated that stiffness and biochemical cues interact in a non-linear manner, emphasizing their complex synergistic influence on cell behaviors.<sup>240</sup>

(iv) Most of the suggested approaches for stiffness generation and/or regulation are restricted to designing materials with static stiffness for dynamic cells. Wang *et al.*<sup>241</sup> synthesized some polyelectrolyte multilayer films, whose mechanical properties are controlled dynamically through slight stimuli.<sup>241</sup> They designed the films *via* different deposition of poly-L-lysine



Table 2 An overview of the most recent studies on the effects of stiffness gradients on cell responses to scaffolds

Materials	Stiffness values	Stiffness modification approach	Cell responses	Ref.
PDMS	~ 2–85 kPa	Adding cross-linkers with different concentrations	The NIH3T3 adhesion is related to the exposure of cell-binding motif of fibronectin. Motif exposure depends on the surface stiffness. MQMs: NA or difficult to summarize.	247
PDMS	0.6–2.7 MPa	Changing Sylgard 527 and 184 concentration	Surface stiffness does not affect MC3T3-E1 cell spreading; however, it can influence MC3T3-E1 osteogenic differentiation, but there is no rule like “the stiffer, the better”. MQMs: (I) the vinculin expression on surfaces with 2.7 MPa stiffness is almost 1.5 times higher than that on surfaces with the lowest stiffness. (II) After three weeks of cell culture, the expression of osteogenesis markers on surfaces with 0.6 MPa stiffness is almost 1.5 times higher than that on surfaces with 2.7 MPa stiffness. Increasing the surface stiffness increases the spreading speed of A549 cells. In PAAm, the bulk stiffness affects cell behaviors; however, in PDMS, the surface stiffness plays a key role in regulating cell responses.	249
PDMS or PAAm	0.1–40 kPa	UV cross-linking	MQMs: the cell spreading speed on stiff surfaces is almost 4-fold higher than that on soft surfaces. MQMs: (I) softer surfaces support the expression of heparin and maintenance of hepatocyte phenotype 5 times better than rough or stiff surfaces. (II) Cell density on soft and stiff heparin gels is 690 cells per mm <sup>2</sup> and 540 cells per mm <sup>2</sup> , respectively.	239
Hep-SH & PEG-DA	10–110 kPa	Changing the concentration of the precursor solution	MQMs: (I) softer surfaces support the expression of heparin and maintenance of hepatocyte phenotype 5 times better than rough or stiff surfaces.	250
PAAm	6.1 or 46.7 kPa	Changing the concentration of bisacrylamide	Compared to topography and dimension, surface stiffness and/or dimension are predominant in controlling MSC proliferation. Stiffness supports the osteogenic or neuronal differentiation of rBMSCs on a stiff or soft surface, respectively. MQMs: cell proliferation on stiff and soft surfaces is $3.49 \pm 0.96$ and $2.50 \pm 0.42$ , respectively.	251
PAAm	100 Pa, 10 or 30 kPa	NA	Increasing the surface stiffness values increases foreign body responses (primary rat microglial cells and astrocytes) to surfaces. MQMs: compared to soft surfaces, stiff surfaces show a significant increase in the inflammatory response ( $P = 1.9 \times 10^{-6}$ ), immune cell trafficking ( $P = 9.6 \times 10^{-5}$ ), cellular growth and proliferation ( $P = 7.5 \times 10^{-5}$ ), cell-mediated immune response ( $P = 2.2 \times 10^{-4}$ ), and antigen presentation ( $P = 1.1 \times 10^{-4}$ ) of microglia.	235
PAAm	0.51, 3.7, and 22 kPa	Changing monomer concentration	Nuclear localization of osteogenesis transcription factors is dependent on surface stiffness. ALP expression is improved only in MSCs, which are not only adhered to stiffer surfaces, but also have a direct contact with other cells. Both cell-cell contact and stiffness are important in guiding the cell fate. MQMs: NA or difficult to summarize.	252
PAAm	13–16, 35–38, 48–53, & 62–68 kPa	Changing bisacrylamide concentration	Increasing surface stiffness increases UC-MSC adhesion. Tendency for adipogenic differentiation on softer surfaces, for muscle differentiation on surfaces with moderate stiffness, and for osteogenesis on high-stiffness surfaces. MQMs: regarding cell proliferation, after 2 days, the percentages of S phase cells on matrices of 13-16, 35-38, 48-53, and 62-68 kPa are 26.82, 26.64, 24.43, and 22.39%, respectively.	253
PAAm	1–25 kPa	Changing monomer concentration	ECM proteins can influence NIH 3T3 responses to surface stiffness. NIH 3T3 exhibits durotaxis on fibronectin coated stiffness gradients but not on the lamin coated surfaces. MQMs: NA or difficult to summarize.	238
PAAm	0.5, 1.7, 2.9, 4.5, 6.8 & 8.2 kPa	Changing monomer and cross-linker concentration	Some stiffness values might be non-durotactic for hASC cells, other values can be durotactic and cause cell migration and differentiation. MQMs: NA or difficult to summarize.	245
PAAm	0.04–0.95 kPa	Changing monomer and cross-linker concentration	Schwann cells exhibit durotaxis in response to stiffness gradients in the biological stiffness range of peripheral nerve tissue. Median cell velocity on even surfaces is $0.67 \mu\text{m min}^{-1}$ , on surfaces with low gradient stiffness is $1.55 \mu\text{m min}^{-1}$ , and on surfaces with high gradient stiffness is $1.38 \mu\text{m min}^{-1}$ .	244
PAAm	0.167 or 49.6 kPa	Changing monomer and cross-linker concentration	Hypoxia can affect MSC cell responses to surface stiffness. MQMs: NA or difficult to summarize.	254
PAAm	5–60 kPa	Changing monomer and cross-linker concentration	Primary breast cancer cells experience significant phenotypic changes after culturing on surfaces with different stiffness.	255





Table 2 (continued)

Materials	Stiffness values	Stiffness modification approach	Cell responses	Ref.
ECM-derived polymers	15–194 kPa	linker concentration Using PEGDA with different MW	MQMs: (i) an approximate 4-fold decrease in FBXW7 gene expression in cells seeded onto soft surfaces compared to glass. (ii) A 10-fold overexpression of CYP1A1 in MDA-MB-453 cells on soft surfaces compared to glass. (iii) A 3-fold decrease of cell apoptosis on soft surfaces. Surface biochemical and mechanical cues synergize only at particular mixtures to increase bone differentiation of ADSCs. For osteocalcin gene expression, intermediate stiffness (55 kPa) and low concentration of fibronectin are optimal. MQMs: (i) a 58-fold and 46-fold osteocalcin expression at 20% w/v PEGDA for both MW 3400 and MW 5000, respectively, compared to the control group. (ii) Increasing fibronectin from 10 to 25 $\mu\text{g ml}^{-1}$ decreases osteocalcin to 12–47-fold of the control.	240
Glass slides	0.5–110 MPa	Using a weak PEM system and EDC cross-linker	Compared to chemistry and wettability, surface stiffness is a stronger driving force in determining HDF cell fate. MQMs: (i) after 6 days of cell culture, a 5-fold increase in cell number on soft regions compared to the control group. (ii) However, a 10-fold increase in cell number on stiff regions compared to the control group.	256
Self-assemble peptide nanofibers	22.9 $\pm$ 5 or 7.3 $\pm$ 0.9 kPa	Supramolecular interactions	NIH 3T3 neuronal polarity and maturity are faster on softer nanofiber surfaces compared to stiff surfaces. Surface stiffness can affect neuronal growth by adjusting its dynamics. MQMs: after 20 h cell culture, 67.1 $\pm$ 6.2% neurons on stiff surfaces reach developmental stage 2; however, 60.9 $\pm$ 2.6% neurons on soft substrates reach developmental stage 3 at the same time.	257
MeHA	2.1 or 24 kPa	UV cross-linking	Stiffer surfaces with $E \sim 24$ kPa better support HSC cell adhesion and differentiation compared to softer ones.	258
MeHA	5, 12, and 23 kPa	Changing HA concentration	MQMs: after 28 days of HSC cell culture, the cell mean area on soft and stiff surfaces is $\sim 200$ and $5500 \mu\text{m}^2$ , respectively. HMSC cell volume or size has a potential influence on foreign responses to surface stiffness.	248
Gtn-HPA	$G' = 570\text{--}2750$ Pa	Changing $\text{H}_2\text{O}_2$ and Gtn-HPA concentration	MQMs: greater cell volume leads to more focal adhesion by increasing stiffness (62%, 38%, and 10% focal adhesion formation on surfaces with 2800, 3600, and $6000 \mu\text{m}^3$ cell volumes, respectively).	259
PEG	130 & 3170 kPa	Changing PEG-DA MW and concentration	Cartilage regeneration is dependent on surface stiffness values of scaffolds. Scaffolds with medium stiffness (1000 Pa) better guide chondrocyte cell responses toward cartilage repair compared to softer or stiffer surfaces. MQMs: (i) after 10 days of cell culture, a hydrogel with medium stiffness produces the highest level of sGAG, which is twice that of low stiffness hydrogels. (ii) The collagen expression increases by increasing stiffness, with 1.5-fold difference.	260
collagen-GAG	5.05–2.85 kPa	Carbodiimide cross-linking and benzophenone photoimmobilization chemistries	Surface stiffness plays a key role in regulating MSC differentiation. RGD nanospacing influences MSC spreading and differentiation irrespective of surface stiffness. Both surface stiffness and nano-scale spatial organization of cell-adhesive ligands play key roles in regulating stem cell fate.	261
Cellulose	76–448 kPa	Changing glyoxal cross-linker concentration	MQMs: cell density and area as well as F-actin intensity are almost 30% higher on stiff surfaces compared to soft surfaces. The stiffest surfaces guide the osteogenesis of ADSCs irrespective of the presence or absence of growth factors.	262
Silk fibroin	3–7.4, 12–25.9, and 35.6–58.4 kPa	Using freezing Temperature	Cell proliferation is enhanced on surfaces with moderate stiffness. MQMs: NA or difficult to summarize.	263
PLL/HA-SH	60–210 kPa	Cross-linking through oxidation of thiol groups	Surface stiffness can regulate MG-63 morphology. MQMs: NA or difficult to summarize.	241
DEGDMA/nOM	25–4700 kPa	Changing n-OM monomer concentration	BMSC responses are sensitive to even subtle changes in the surface stiffness values. The ideal vascularization ability of the surface is in the 3–7.4 kPa stiffness range. MQMs: 5.7 kPa stiffness is optimal for neo-vessel formation with a vessel density of 37, 70, and 45 vessels per $\text{mm}^2$ after 1, 2, and 3 weeks, respectively, compared to other stiffness values. Increasing the stiffness increases fibroblast adhesion. MQMs: surface stiffness increases the cell metabolic activity and adhesion by 30% compared to soft surfaces.	264

Table 2 (continued)

Materials	Stiffness values	Stiffness modification approach	Cell responses	Ref.
PCL, PLA, PGA	62, 128, and 204 MPa	By testing different materials with different stiffness properties	Surface stiffness and topographical cues influence MSC morphology and aggregation at the earlier phase of MSC chondrogenic differentiation. Softer pillar surfaces stimulate the formation of hyaline-like cartilage with middle/deep zone cartilage features. Stiffer nanopillar areas increase the formation of hyaline/fibro/hypertrophic cartilage.	246
MA/MMA	0.1–310 MPa	Changing monomer concentration	MQMs: collagen type I expression is 2–3 times higher on stiff nanopillar surfaces compared to soft surfaces. Integrin subunit expression alters depending on the stiffness value of the surface and cell types. Surface stiffness values can determine and change the fate of MSCs. MQMs: NA or difficult to summarize.	234

Abbreviations: mouse fibroblasts (NIH3T3), thiolated heparin (Hep-SH), diacrylated poly(ethylene glycol) (PEG-DA), polyacrylamide (PAAm), adipose-derived stem cells (ADSCs), poly(ethylene glycol)-diacrylate (PEGDA), molecular weight (MW), human dermal fibroblasts (HDFs), 1-ethyl-3-(3-dimethylaminopropyl)carbodiimide (EDC), polyelectrolyte multilayer (PEM), hepatic stellate cells (HSCs), methacrylated hyaluronic acid (MeHA), gelatin-hydroxyphenylpropionic acid (Gtn-HPA), storage modulus ( $G'$ ), glycosaminoglycans (GAGs), poly-L-lysine (PLL), thiol group modified hyaluronan (HA-SH), poly(glycerol sebacate) (PGS), alkaline phosphates (ALP), *n*-octyl methacrylate–diethylene glycol dimethacrylate (DEGDMA/nOM), valvular interstitial cell (VIC), alpha smooth muscle actin (aSMA), polycaprolactone (PCL), polylactide (PLA), polyglycolide (PGA), umbilical cord mesenchymal stem cells (UC-MSCs), methyl acrylate/methyl methacrylate (MA/MMA), methacrylated hyaluronic acid (MeHA), human adipose derived stem cell (hASC), major quantitative measurements (MQMs), sulfated glycosaminoglycans (sGAG).

and thiol group treated hyaluronan (Fig. 12). This method increases the surface stiffness resulting in enhanced fibroblasts cell adhesion. However, using glutathione dynamically decreases surface stiffness leading to reduced cell adhesion.<sup>241</sup>

(v) The native ECM contains complex mechanical pathways, such as time-dependent dynamic manners and nonlinear stiffness. Deciding about the mechanotransduction pathways based on the hydrogels, as commonly used materials for investigating mechanical cues, which have simple linear elastic mechanics can lead to biomaterial failure after implantation.<sup>153</sup> While cell spreading depends on the absolute stiffness of the surface, their alignment and migration depend on the stiffness gradient.<sup>242</sup>

Researchers typically pattern stiffness gradients by using different cross-linking densities, through either introducing chemical gradients in cross-linkers or different exposure of photosensitive cross-linkers.<sup>243</sup> Evans *et al.*<sup>244</sup> studied the migration and morphodynamics of Schwann cells on polyacrylamide scaffolds containing stiffness gradients on their surfaces.<sup>244</sup> The cells could track the slope of stiffness gradients on the surfaces through the durotaxis mechanism, which supports the hypothesis that Schwann cells are extremely sensitive indicators of mechanical gradients.<sup>244</sup> Additionally, scientists designed polyacrylamide hydrogels with stiffness gradients at their surface through controlling the differential diffusion distance of free cross-linkers and monomers into a prepolymerized hydrogel environment.<sup>245</sup> They showed that lower gradients allow detecting more unknown stem cell responses, such as the concentration-dependent rather than switch-like reactions of mechanosensitive proteins (such as yes-associated protein) to some gradient amounts.<sup>245</sup>

(vi) We should also address the synergistic influence of other physicochemical properties with stiffness on cell responses. For example, Wu *et al.*<sup>246</sup> studied the synergistic influence of surface nano-topography, chemistry and stiffness in controlling MSC chondrogenesis.<sup>246</sup> They designed three polyesters (poly-epsilon-caprolactone, polylactic acid, polyglycolide) with different stiffness values and then generated nano-grating or pillar patterns of the same scale on surfaces. They also coated chondroitin sulphate on the surfaces to increase the chance of cell adhesion. They revealed that both surface stiffness and topographical features affect MSC morphology and aggregation. Softer pillar surfaces induce hyaline-like cartilage formation with middle/deep zone cartilage features; however, stiffer nanopillar surfaces stimulate hyaline/fibro/hypertrophic cartilage formation. Nano-grating of lower stiffness values causes fibro/superficial zone-like cartilage formation; nevertheless, greater stiffness with the same topography cannot stimulate chondrogenesis.<sup>246</sup> Hence, different cell functions are affected by the simultaneous influence of various physicochemical properties, which can also up- or down-regulate each other's influence.<sup>246</sup>

(vii) As the mechanobiology field is still in its early stages, before making firm decisions about the role of stiffness and applying it, further biochemistry research on how cells translate mechanical signals to biochemical ones is essential. To



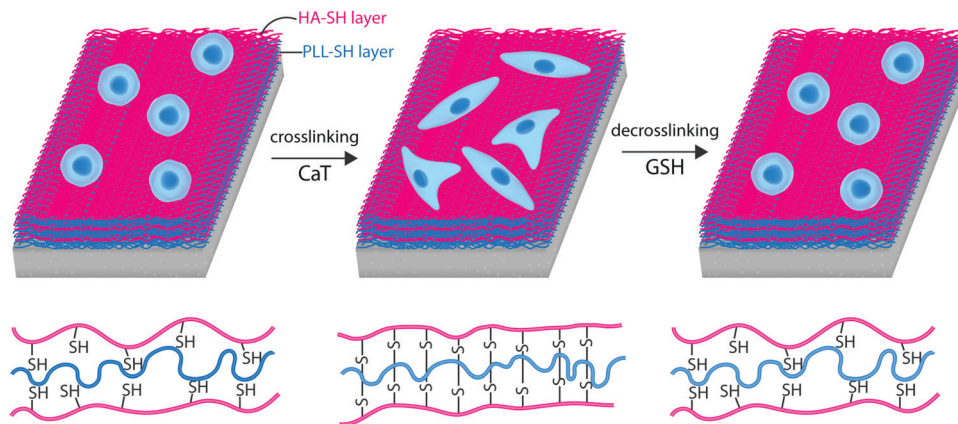


Fig. 12 A schematic representation of NIH/3T3 cell responses to cross-linking and decross-linking of poly-L-lysine (PLL) and thiol group treated hyaluronan (PLL/HA-SH) multilayer substrates.

investigate the mechanotransduction pathways involved in cell responses to materials, some studies tracked phenotypic changes of cells cultured on PDMS or acrylamide substrates with stiffness gradients by patterning ECM proteins.<sup>243,247</sup> Tseng *et al.*<sup>243</sup> developed an approach for having precise, decoupled control of the ECM pattern and local surface stiffness to cells by chemical modification of PDMS surfaces.<sup>243</sup> After culturing MC-3T3 cells on PDMS surfaces with different stiffness gradients and ECM patterns (including X, square, and I), the actin cytoskeleton polarizes to make interactions with the PDMS surface.<sup>243</sup>

(viii) The accurate evaluation of cell responses to mechanical properties of the surface can provide useful information about mechanotransduction pathways involved in cell responses to the surface.

(ix) Not only cell type but also cell volume or size is a key player in determining cell responses to surface stiffness. Bao *et al.*<sup>248</sup> cultured hMSCs in separate 3D micro niches with various volumes (2800, 3600, and 6000  $\mu\text{m}^3$ ) on methacrylated hyaluronic acid hydrogels with different stiffness (5, 12, and 23 kPa).<sup>248</sup> They revealed that cell volume has a strong influence on the hMSC responses to surface stiffness. Cells with ideal volume can form obvious stress fibers and focal adhesions on all surfaces with different stiffness. However, in small volumes, stiffness does not have any effects on stress fiber formation and yes-associated protein/PDZ-binding motif localization.<sup>248</sup>

## 9.2. Impact of biomaterial surface chemical properties on biological responses

In the following sections, we provide an overview of the current progress and challenges related to chemical surface modification of biomaterials for improving their interactions with cells and proteins. We also provide a concise overview of the most common chemical techniques used for applying these strategies.

**9.2.1. Biological responses to biomaterial surface functional groups and charges.** The surface wettability of a biomaterial (known as hydrophobicity and hydrophilicity) plays key roles in directing cell responses through affecting protein

adsorption mechanisms such as protein conformation and adsorption force.<sup>265–267</sup> The surface topography and/or functional groups can adjust this surface feature.<sup>268</sup> Researchers use many functional groups with different charges as promising candidates for adjusting biomaterial surface wettability [such as phosphorylcholine (zwitterionic/hydrophilic), trimethylammonium (cationic/hydrophilic), sulfonate (anionic/hydrophilic), hydroxyl (nonionic/hydrophilic), and *n*-butyl groups (nonionic/hydrophobic)] (Fig. 13).<sup>269</sup> Enriching a surface with favorable functional groups and charges, based on the targeted molecule and cell type, is a broadly used chemical approach for improving cell–biomaterial interactions (Table 3).<sup>270–272</sup>

The surface functional groups can direct cell functions through covalent conjugation with functional groups of cell surface lipids, proteins and glycans.<sup>273</sup> Bygd *et al.*<sup>274</sup> systematically studied how using various types of functional groups could affect macrophage reprogramming and polarization *in vivo* by working on poly(*N*-isopropylacrylamide-*co*-acrylic acid) nanoparticle surfaces.<sup>274</sup> They also used a quantitative

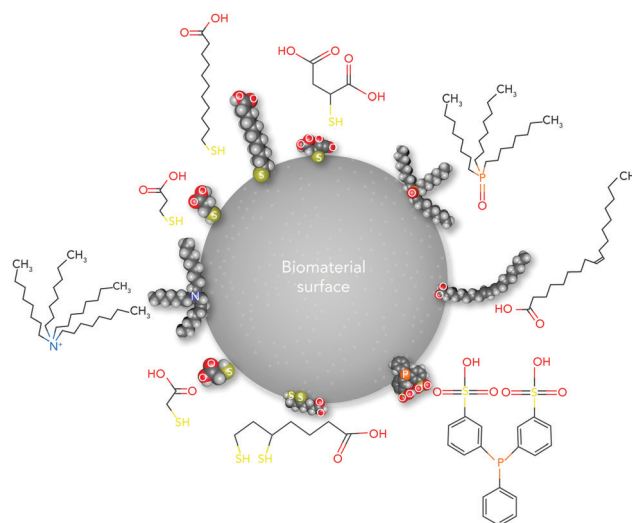


Fig. 13 A schematic representation of different hydrophilic and hydrophobic functional groups used for adjusting the surface wettability.



Table 3 An overview of the most recent studies on the effects of surface functional groups on cell responses to biomaterials

Functional group	Materials	Other surface features or comment	Cell responses	Ref.
NH <sub>2</sub>	Stainless steel	PDAM/H/D coating	NH <sub>2</sub> improves HUVBC cell responses. MQMs: (I) the ratio of apoptotic cells cultured on treated surfaces is ~2.9% compared to NC (~17.8%). (II) The cell migration is 3.9 times greater on treated surfaces compared to untreated surfaces.	286
NH <sub>2</sub>	MBG	NA	Higher expression of osteogenic genes and anti-inflammatory factors as well as fewer TRAP <sup>+</sup> multinuclear cells of BMSCs and macrophages in modified surfaces compared to untreated surfaces. MQMs: (I) after 4 weeks of implantation in the maxillary of rabbits, alizarin red labeling on treated surfaces is 1.01 ± 0.13% compared to untreated surfaces (0.70 ± 0.11%). (II) After 8 weeks healing, calcein labeling on treated surfaces is 3.62 ± 0.16% compared to untreated surfaces (2.89 ± 0.14%).	287
NH <sub>2</sub>	TCP, PLLA	Designing the surfaces in two different topography (flat and fibrous). APTES	Surface topography has more effects on cell proliferation and gene expression of hMSCs than surface chemistry. MQMs: (I) topography makes a significant difference ( $F = 39.27$ , $df = 1$ , $p < 0.001$ , GLM) in cell proliferation. (II) Chemistry does not significantly affect cell proliferation ( $F = 0.03$ , $df = 1$ , $p = 0.868$ , GLM).	288
NH <sub>2</sub>	MBG	NA	NH <sub>2</sub> improves MC3T3-E1 cell proliferation.	289
NH <sub>2</sub>	UHMWPE	Allylamine coating, using oxygen and nitrogen for making the surface more hydrophilic	MQMs: cell proliferation on treated surfaces is almost twice that on untreated surfaces. NH <sub>2</sub> improves human foreskin fibroblast stability and morphology.	290
OH	PHBVDB	NA	MQMs: cell viability on treated surfaces is (~70%) higher than NC.	291
SNO	Polyester	NA	OH treated surfaces do not show significant cytotoxicity for hMSCs. MQMs: cellular metabolic activity on treated surfaces is almost twice that on untreated surfaces.	292
SO <sub>3</sub> <sup>-</sup>	PEEK	NA	Fibroblast cell response tests indicate a lack of toxic leachates at cytotoxic concentrations. MQMs: cell viability on treated surfaces is 100% higher than negative control.	293
PO <sub>4</sub> H <sub>2</sub>	OPF	0%, 20%, or 40% concentration of the functional group	SO <sub>3</sub> <sup>-</sup> improves BMSC adhesion, spreading, proliferation, and osteogenic differentiation. MQMs: cell adhesion and proliferation on treated surfaces are ~1.5 times higher than that on untreated surfaces.	294
OH, CH <sub>3</sub>	ONPs	Different sizes of functional groups	MQMs: a higher bone volume on 20% (35 ± 13%) and 40% (34 ± 13%) phosphate surfaces compared to untreated surfaces (17 ± 10%).	295
OH, COOH	SWCNTs	Different material lengths (1–3 μm and 5–30 μm)	The surface functional groups play key roles in directing HeLa cellular uptake of nanoparticles under both serum and serum-free conditions. The cell response trend can be predicted partly by surface lipophilicity. MQMs: the cellular uptake index can increase to ~80% by increasing the carbon chain of ONPs.	296
NH <sub>2</sub> , COOH	Polystyrene NPs	NA	The functional groups might have different effects on HepG2 responses in long and short SWCNTs. OH treated surfaces might be safer in comparison with other surfaces. MQMs: NA or difficult to summarize.	297
NH <sub>2</sub> , SH	MBG	NA	The functional groups do not affect cell viability and the expression of type 1 macrophage markers. However, regarding type 2 macrophages, both surfaces hinder the expression of scavenger receptor CD163 and CD200R, as well as IL-10. MQMs: (I) the exposure of type 1 macrophages to PS-COOH can significantly reduce the <i>E. coli</i> uptake by ~30%; however, it does not have any effects on <i>E. coli</i> phagocytosis by type 2 macrophages. (II) PS-NH <sub>2</sub> can significantly reduce the phagocytosis of both types of macrophages by ~20%.	271
OH, CH <sub>3</sub> , NH <sub>2</sub>	Glass coverslip	Surfaces modified with silane.	The functional groups increase hBMSC cell adhesion, proliferation and differentiation. MQMs: cell differentiation on the NH <sub>2</sub> -treated surface is almost 1.5 times higher than that on the SH-treated surface. Surfaces treated with CH <sub>3</sub> and NH <sub>2</sub> stimulate mitochondria-mediated apoptosis in MDA-MB-231 cells through suppressing the PTEN/P13K/AKT pathway.	298
OH, CH <sub>3</sub> , NH <sub>2</sub>	Au-spattered silica wafers	SAM	MQMs: almost 28.10% and 19.07% of the cells exhibit apoptotic behavior after two days of cell culture on the CH <sub>3</sub> - and NH <sub>2</sub> -treated surfaces, respectively, compared to untreated surfaces (13.85%) and OH-treated surface (12.82%).	265
			Fibronectin adsorption force on SAMs follows a chemistry-dependence of -NH <sub>2</sub> > -CH <sub>3</sub> >> -OH. Fibronectin adsorption force and conformation can control the late osteoblast adhesion and subsequent reorganization of adsorbed proteins.	265





Table 3 (continued)

Functional group	Materials	Other surface features or comment	Cell responses	Ref.
CH <sub>3</sub> , NH <sub>2</sub> , COOH	Allylamine	Comparing surface modification with (i) functional groups, (ii) coating with COL-1 or immobilization of the integrin adhesion peptide sequence RGD, and (iii) treatment with plasma containing argon/oxygen gas	MQMs: the fibronectin adsorption force is $1.47 \pm 0.43$ , $25.56 \pm 7.47$ , and $14.10 \pm 4.12$ for OH-, NH <sub>2</sub> -, and CH <sub>3</sub> -treated surfaces respectively. Only surfaces containing amino groups could chemically mask the microgrooves and block the microtopography-mediated guidance of MG-63 cells. MQMs: NA or difficult to summarize.	299
NH <sub>2</sub> , COOH, OH	AuNPs	Containing different surface charge	Positive charges increase hBMSC cellular uptake. COOH decreases ALP activity and calcium deposition. MQMs: NA or difficult to summarize.	272
NH <sub>2</sub> , COOH, OH	InP/ZnS quantum dots	NA	All the functional groups increase HCC-15 and RLE-6TN cell apoptosis and intracellular ROS generation. All functional groups could enter the cells, with greater uptake efficiency for surfaces modified with COOH and NH <sub>2</sub> at low amounts. Modifying surfaces with COOH and NH <sub>2</sub> causes more toxicity than surfaces containing OH. MQMs: (I) the HCC-15 cellular uptake efficiency of COOH-, NH <sub>2</sub> -, and OH-treated surfaces is $87.4 \pm 2.67\%$ , $89.0 \pm 2.15\%$ , and $74.5 \pm 1.89\%$ , respectively. (II) And for RLE-6TN cells it is $67.1 \pm 0.95\%$ , $48.6 \pm 2.03\%$ , and $32.6 \pm 2.14\%$ , respectively.	300
PO <sub>4</sub> H <sub>2</sub> , COOH, OH	PEEK	NA	The functional groups increase MC3T3-E1 adhesion, spreading, proliferation, and osteointegration. MQMs: NA or difficult to summarize.	301
PO <sub>4</sub> H <sub>2</sub> , COOH, OH	PEEK	Vinyl-terminated silanization layers generated on the hydroxylation-pretreated substrate surface.	Apatite forms homogeneously on the functionalized surface and strongly attaches to the surface. The functional groups improve MC3T3-E1 attachment, spreading and proliferation. MQMs: cell adhesion and proliferation on treated surfaces is almost 1.25 times higher than that on untreated surfaces.	302
CH <sub>3</sub> , NH <sub>2</sub> , COOH, OH	Au-coated glass slides	SAM	Endothelial cell migration depends on surface functional group types in the order CH <sub>3</sub> > NH <sub>2</sub> > OH > COOH. MQMs: (I) cell migration distance of CH <sub>3</sub> , NH <sub>2</sub> , and COOH SAMs is $369.5 \pm 27.9$ $\mu\text{m}$ , $325.6 \pm 26.3$ $\mu\text{m}$ , and $213.6 \pm 10.7$ $\mu\text{m}$ , respectively, NA for OH SAMs. (II) Wound repair duration of CH <sub>3</sub> , NH <sub>2</sub> , OH, COOH treated and untreated surface is $17.5 \pm 1$ h, $19 \pm 1$ h, $25 \pm 1.5$ h, $26.5 \pm 1.5$ h, and $22 \pm 1.5$ h, respectively. The functional groups direct MSC differentiation through controlling protein adsorption, nonspecific cell adhesion and cell spreading. Free neutral surfaces (-CH <sub>3</sub> and -OH) cause less protein adsorption, cell spreading and adhesion but more chondrogenic induction than charged surfaces (-COOH and -NH <sub>2</sub> ). MQMs: NA or difficult to summarize.	276
CH <sub>3</sub> , NH <sub>2</sub> , COOH, OH	PEG	Coating surfaces with alkanethiols with one of the functional groups and then experiencing 9 days of chondrogenic induction.		303
CH <sub>3</sub> , NH <sub>2</sub> , COOH, OH	Au-coated titanium implants	SAM	NH <sub>2</sub> increases the hDPSC adhesion, proliferation, and osteo/odontogenesis differentiation compared to other functional groups. MQMs: after 7 and 21 days, alizarin red labeling on NH <sub>2</sub> -treated surfaces is almost 1.5 times higher than other groups.	304
OH, COOH, NH <sub>2</sub> , CONH <sub>2</sub>	Carbon nanowall	Designing surfaces from hydrophobic to hydrophilic with the incorporation of different concentrations	Changing surface chemistry by plasma treatment has effects on macrophage response <i>in vitro</i> , regardless of surface wettability features. MQMs: NA or difficult to summarize.	305

Table 3 (continued)

Functional group	Materials	Other surface features or comment	Cell responses	Ref.
$[(\text{CH}_3)_3\text{N}^+\text{CH}_2\text{CH}_2\text{PO}_4^-]$ , $(\text{C}_6\text{H}_5)_2\text{CO}$	Photoreactive methacrylate	of carbon, oxygen and nitrogen functional groups Using amphiphilic polymers PMH and poly(MPC-co- <i>n</i> -butyl methacrylate-co-MHPBP)	The functional groups increase HeLa cell responses to surfaces. MQMs: NA or difficult to summarize.	306
OH, COOH, biotin, azide, nitrilotriacetic acid	AuNPs	Conjugation to functional groups through using PEG	After 2 h incubation at physiological temperature, active BEAS-2B cellular uptake can occur only on citrate-stabilized and COOH-treated surfaces. MQMs: NA or difficult to summarize.	307

Abbreviations: dopamine and hexamethylenediamine (PDAM/HD), amine ( $\text{NH}_2$ ), carboxylic groups (COOH), human umbilical vein endothelial cells (HUVECs), self-assembled monolayers (SAMs), S-nitrosothiols (RSNOs), poly(etheretherketone) (PEEK), pre-osteoblast cell (MC3T3-E1), organic nanoparticles (ONPs), thiol (SH), mesoporous bioactive glass (MBG), human bone marrow-derived mesenchymal stem cells (hBMSCs), osteocalcin (OCN), bone sialoprotein (BSP), bone morphogenic protein 1 (BMP1), collagen type I (COL-1), gold nanoparticles (AuNPs), ultra-high molecular weight polyethylene (UHMWPE), sulfonate ( $\text{SO}_3\text{H}$ ), phosphorylcholine  $[(\text{C}_6\text{H}_5)_3\text{N}^+\text{CH}_2\text{CH}_2\text{PO}_4^-]$  benzophenone  $[(\text{C}_6\text{H}_5)_2\text{CO}]$ , 3-methacryloyloxy-2-hydroxypropyl-4-oxobenzophenone (MHPBP), poly(2-methacryloyloxyethyl phosphorylcholine (MPC)-co-MHPBP) (PMH), human cervical cancer cells (HeLa), macrophage inflammatory protein-1 $\alpha$  (MIP-1 $\alpha$ ), colony-stimulating factor (CSF), poly(3-hydroxybutyrate-co-3-hydroxyvalerate-co-2,3-dihydroxybutyrate) [PHBVDB], Arg-Gly-Asp (RGD), poly(ethylene glycol) (PEG), human dental pulp stem cells (hDPSCs), human lung cancer cell (HCC-15), alveolar type II epithelial cell (RLE-6TN), polystyrene (TCP), poly(L-lactide) (PLLA), octyl ( $\text{C}_8\text{H}_{17}$ ), fetal bovine serum (FBS), sulfonate ( $\text{SO}_3^-$ ), osteopontin (OPN), oligo[(polyethylene glycol) fumarate] (OPF), human breast cancer cell line (MDA-MB-231), B-cell lymphoma 2 (Bel-2), single-walled carbon nanotubes (SWCNTs), human bronchial epithelial cell line (BEAS-2B), 3-aminopropyltriethoxysilane (APTES), major quantitative measurements (MQMs).

structured activity relationship (QSAR) method as an analytical predictive tool for measuring the macrophage responses to various surface chemistries. Based on the surface functional groups, they observed a spectrum of macrophage phenotypes.<sup>274</sup>

However, we should keep in mind that not only the surface properties of biomaterials but also the local cell microenvironment affects the proliferation of macrophages.<sup>275</sup> Shen and colleagues studied the effects of different functional groups ( $\text{CH}_3$ ,  $\text{NH}_2$ ,  $\text{COOH}$ ,  $\text{OH}$ ) prepared by self-assembled monolayers on endothelial cell responses through studying the expression of key proteins in integrin-induced signaling pathways.<sup>276</sup> By observing the differences in the expression of focal adhesion components and Rho GTPases, they showed that endothelial cell migration is highly dependent on the type of surface functional groups in the order  $\text{CH}_3 > \text{NH}_2 > \text{OH} > \text{COOH}$ .<sup>276</sup>

Rashad *et al.*<sup>277</sup> designed wood-derived cellulose nanofibril hydrogels containing two different surface functional groups to address the influence of surface chemistry on fibroblast cell functions.<sup>277</sup> They used 2,2,6,6-tetramethylpiperidine-1-oxyl radical (TEMPO)-mediated oxidation or carboxymethylation pretreatments on the surface. The TEMPO-oxidized surface induces favorable cell morphology and spreading, whereas the carboxymethylated surface inhibits these processes.<sup>277</sup>

Here we summarize some of the recent challenges in this area:

(i) Using different functional groups on the surface of biomaterials can stimulate different biochemical signaling pathways.<sup>278</sup> Zhang *et al.*<sup>278</sup> investigated the potential pro-inflammatory influences of surface functional groups on human pulmonary epithelial cells and macrophages by using PEGylated CdSe/ZnS quantum dots (QDs) containing an amphiphilic polymer coating (PEG-pQDs) (Fig. 14A).<sup>278</sup> The pro-inflammatory effects of this surface depend on the functional groups ( $-\text{COOH}$ ,  $-\text{NH}_2$ ,  $-\text{OH}$ , and  $-\text{OCH}_3$ ) at the end of the PEG chain.  $\text{COOH}$ -PEG-pQDs cause the highest pro-inflammatory effects followed by  $\text{NH}_2$ -PEG-pQDs,  $\text{HO}$ -PEG-pQDs and  $\text{CH}_3\text{O}$ -PEG-pQDs. Fig. 14B shows that  $\text{COOH}$  containing areas internalize *via* lipid raft- and class A scavenger receptor-mediated endocytosis and subsequently stimulate the NF- $\kappa\text{B}$  signaling pathway. However, lipid raft-mediated endocytosis and stimulation of p38 MAPK/AP-1 signaling pathways are affected by surfaces containing  $\text{NH}_2$  and  $\text{HO}$ .<sup>278</sup>

Hence, researchers suggest using surfaces containing multiple functional groups to affect different biological pathways simultaneously. For instance, Wang *et al.*<sup>279</sup> designed a poly(ether sulfone) (PES) surface containing multiple bio-functional groups such as sodium carboxylic, sodium sulfonic and amino groups to act as an antithrombotic bio-interface. They introduced functional groups onto the surface in three steps: (1) making PES with carboxylic groups (CPES) and water-soluble PES with sodium sulfonic and amino groups (SNPES); (2) presenting carboxylic groups onto the PES membrane by mixing CPES with PES; (3) and grafting SNPES onto CPES/PES membranes by coupling amino and carboxyl groups on the surface (Fig. 15A).<sup>279</sup> Their results indicated that the treated





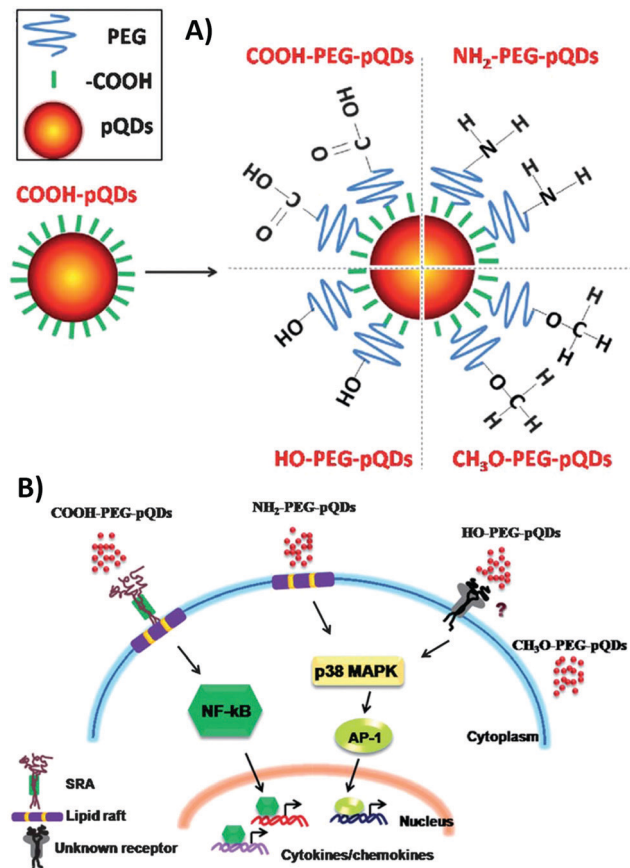


Fig. 14 (A) A schematic representation of the structure of water-soluble CdSe/ZnS QDs bearing carboxyl groups (COOH-pQDs) and the derivative PEGylated pQDs. (B) The association between the terminal functional group-dependent endocytic pathways and the pro-inflammatory responses induced by PEGylated quantum dots. Reprinted from ref. 278. Copyright © 2013, Royal Society of Chemistry.

surfaces could cause an excellent hindrance to platelet adhesion and activation, extend clotting times, and block blood-related complement and leukocyte-related complement receptor activation. Fig. 15B shows that owing to the synergistic enhancement of the functional groups, endothelial cell proliferation improves on treated surfaces.<sup>279</sup>

(ii) Similar to physical properties, research groups suggest using gradients of functional groups on the biomaterial surface to more precisely control biological mechanisms.<sup>280</sup> Liu *et al.*<sup>280</sup> designed a surface chemical gradient of amine functional groups through tuning the gas composition of 1,7-octadiene (OD) and allylamine of plasma phase.<sup>280</sup> Under standard culture conditions (with serum), hASC adhesion and spreading area improve toward the allylamine side of the gradient surface. However, there is no difference in cell behaviors in the absence of serum, which supports the idea that surface functional groups affect hASC response through cell-adhesive serum proteins, rather than directly influencing cell functions. In addition, osteogenic differentiation is enhanced on the allylamine side of the gradient, while the adipogenic differentiation is reduced. Differences between the cell differentiation in different chemical

gradients of surfaces disappear *via* blocking the extracellular signal-regulated kinase 1/2 signaling pathway activation *via* PD98059 (a specific inhibitor of the mentioned signaling pathway).<sup>280</sup>

(iii) The presence or absence of protein serum in the experimental media can affect cell responses and research outcomes. Shahabi *et al.*<sup>281</sup> studied the cellular uptake of five various single or multifunctionalized fluorescent silica nanoparticles (FFSNPs) to address the role of surface charge in directing cell responses by using different concentrations of sulfonate and amino groups (Fig. 16).<sup>281</sup> They set the zeta potential values of the surfaces from extremely positive to extremely negative, whereas other surface properties remained nearly constant. Depending on the surface charge and on the presence or absence of protein serum, two reverse trends for FFSNP cellular uptake exist. In the absence of serum, human osteoblasts can better accumulate positively charged nanoparticles than negatively charged surfaces. However, in the serum-containing medium, osteoblasts can better internalize anionic particles. Under physiological conditions, sulfonate-functionalized silica nanoparticles are the preferred choice to have a high rate of nanoparticle internalization.<sup>281</sup>

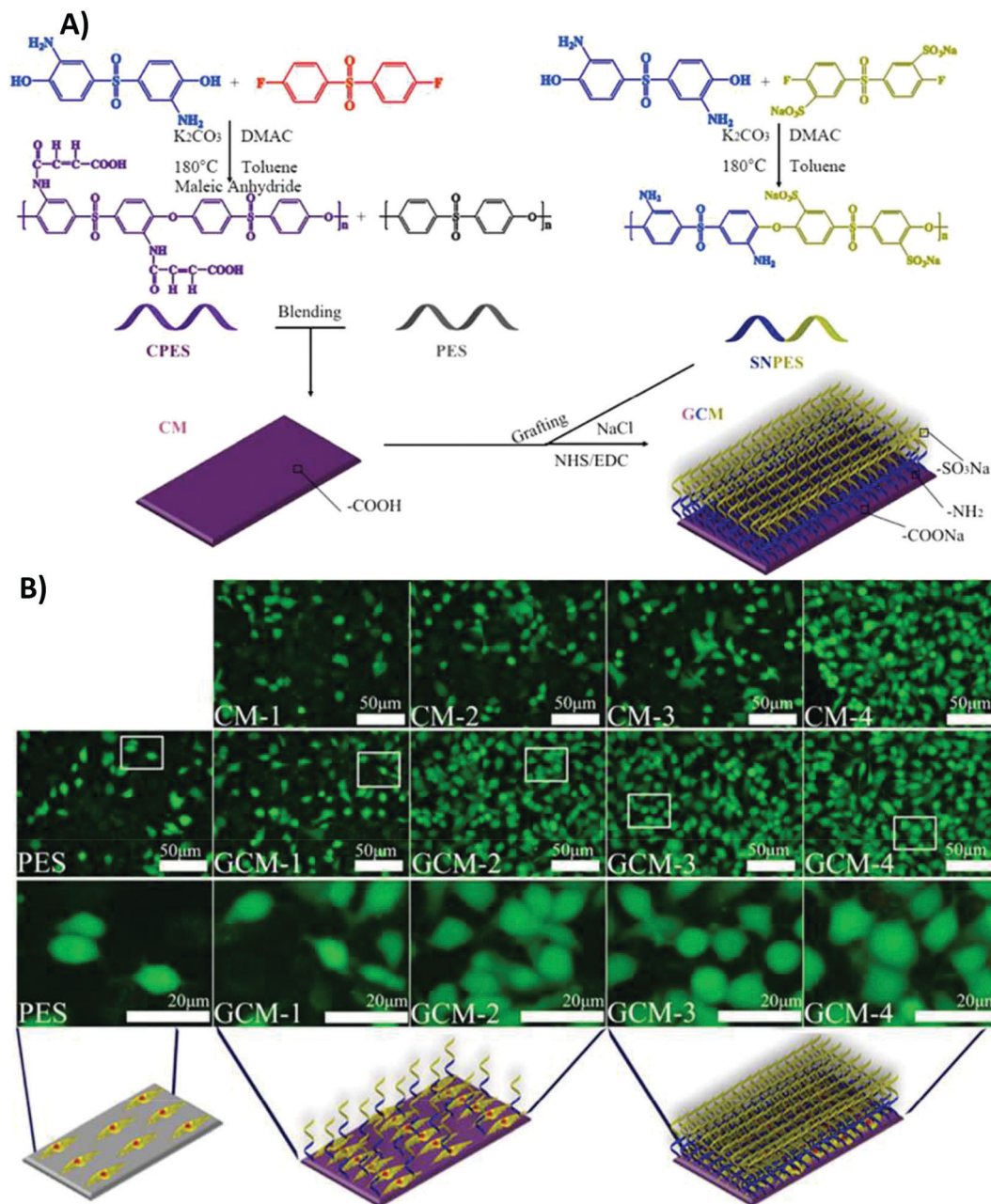
(iv) Hasan *et al.*<sup>282</sup> studied the effects of media conditions on protein and L929 mouse fibroblast cell responses to five dissimilar nano-scaled surfaces treated with functional groups including amine, octyl, mixed, hybrid, and carboxylic.<sup>282</sup> They studied the protein and cell responses under three dissimilar conditions consisting of (1) foetal bovine serum (FBS) in media, (2) pre-adsorbed FBS on surfaces, and (3) partial media without FBS. Regardless of the functional groups used on the surfaces, surfaces with pre-adsorbed FBS show the highest L929 fibroblast adhesion rate and cell spread area. However, surfaces placed in partial media show minimum adhesion rate, poor cell spreading and inappropriate morphology.<sup>282</sup>

(v) The functional group density can also influence protein adsorption mechanisms.<sup>283</sup> Meder *et al.*<sup>283</sup> studied the adsorption of three model proteins, bovine serum albumin, lysozyme and trypsin, on colloidal alumina surfaces. They functionalized the surfaces with SO<sub>3</sub>H in densities ranging from 0 to 4.7 SO<sub>3</sub>H nm<sup>-2</sup>.<sup>283</sup> Their results indicated that the functional group surface density affects the adsorption of all three proteins. Simply changing the density of functional groups on the surface can cause a continuous tuning of protein adsorption from nearly no adsorption to a theoretical monolayer.<sup>283</sup>

(vi) Although there are different strategies for using functional groups to direct cell responses, the biochemical signaling pathways which respond to each functional group are not yet fully detected.

(vii) Other factors such as cell lines, topography, stiffness as well as the complex ECM and physiological metabolism of cells may also be involved in directing cell responses to surface functional groups.<sup>268</sup> Researchers designed a series of model surfaces with controlled surface nanotopography ranging from 16, 38, to 68 nm.<sup>284</sup> They functionalized surfaces with amine, carboxyl, or methyl groups to investigate the primary neutrophil and macrophage responses. In these chemically modified





**Fig. 15** (A) Fabricating poly(ether sulfone) (PES) with carboxylic ( $-\text{COOH}$ ) groups (CPES) and water-soluble PES with sodium sulfonic ( $-\text{SO}_3\text{Na}$ ) groups and amino ( $-\text{NH}_2$ ) groups (SNPES) and grafting of SNPES. (B) Fluorescence staining (FITC) images of cultured vein endothelial cells on PES, CM-1, CM-2, CM-3, CM-4, GCM-1, GCM-2, GCM-3 and GCM-4 after 6 days (CM: carboxylated surface with the PES/CPES ratios of 10/0, 9/1, 8/2, 7/3 and 6/4 is termed PES, CM-1, CM-2 and CM-3, and CM-4, respectively; GCM: functionalized surfaces with sodium carboxylic ( $-\text{COONa}$ ) groups, sodium sulfonic ( $-\text{SO}_3\text{Na}$ ) groups and amino ( $-\text{NH}_2$ ) groups, the grafted CMs with PES/CPES ratios of 10/0, 9/1, 8/2, 7/3 and 6/4 are named as PES, GCM-1, GCM-2, GCM-3, and GCM-4, respectively). Reprinted from ref. 279. Copyright © 2017, Royal Society of Chemistry.

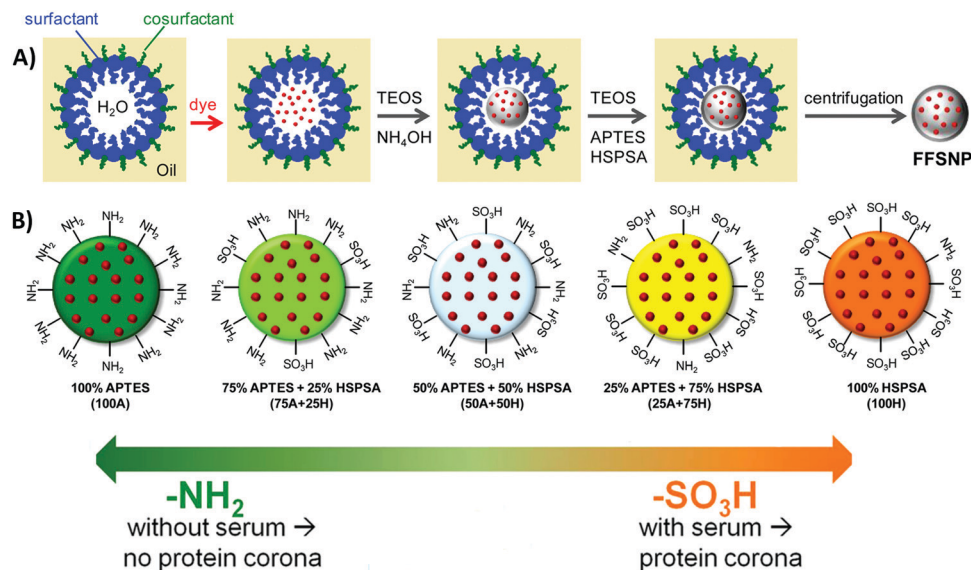
surfaces, the surface nanotopography can decrease matrix metalloproteinase 9 expression in neutrophils as well as the concentration of IL-6 and IL-1 $\beta$  in macrophages.<sup>284</sup> Surface chemistry and nanotopography can, in a synergistic manner, control the osteo-immune environment functions such as the production of inflammatory cytokines, osteoclastic activities, as well as osteogenic, angiogenic, and fibrogenic factors.<sup>285</sup>

(viii) Table 3 shows that  $\text{CH}_3$ ,  $\text{NH}_2$ ,  $\text{COOH}$ , and  $\text{OH}$  are the commonly used functional groups used for modifying the

surface chemistry. Here we suggest also considering the other existing functional groups present in nature.

**9.2.2. Biological responses to ion enrichment of biomaterial surface.** After the immersion of a biomaterial in a medium *in vitro* or *in vivo*, the surface charge, stability, and ions are key players affecting cell responses.<sup>308</sup> McCarthy *et al.*<sup>309</sup> investigated how cobalt ions direct collagen matrix formation and cell responses to the surface by using 200 ppm cobalt ions, as the highest reported cobalt dose at the injured site.<sup>309,310</sup> They found that the presence





**Fig. 16** (A) An illustration of the synthesis and concurrent functionalization of various single or multifunctionalized fluorescent silica nanoparticles (FFSNPs) by 3-aminopropyl-triethoxysilane (APTES) and 3-(trihydroxysilyl)-1-propanesulfonic acid (HSPSA). (B) An illustration of the designed particles by modulating the primary molar ratio of amino and sulfonate functional groups. Reprinted from ref. 281, copyright © 2013, Royal Society of Chemistry. (DOI: 10.1021/acsami.5b01900). Further permissions related to the material excerpted should be directed to ACS.

of cobalt ions could cause local variations in collagen density, which leads to an improvement in localized mechanical properties but a decrease in the bulk stiffness of the material. Fig. 17 shows that cobalt ions can make interactions with the hydroxyl group in the carboxy terminus of the collagen fibril and inhibit the formation of essential stabilizing bonds within the collagen network. Hence, the presence of these ions in collagen matrices confers substantial changes in how cells interact with the collagen matrix.<sup>309</sup> Several research groups suggest using different ions on the surface of biomaterials for improving cell–biomaterial interactions through enhancing physicochemical properties of the surface and/or directing the physiological pathways involved in host responses (Table 4).<sup>311</sup>

Engineers use ion implantation, a standard technique in semiconductor processing, for improving the surface cell adhesion or wettability through modulating surface mechanical properties, wear resistance, and corrosion resistance.<sup>312,313</sup> Ion implantation can improve the surface properties of metallic implants, polymers and ceramics.<sup>314–318</sup> This technique includes the bombardment of ionized species and their implantation into the topmost layers of a solid material.<sup>319</sup> It needs an ion generation source, an electrostatic acceleration system, and a vacuum chamber. Physical sources in a discharge chamber make ions and then precursors convert them into vapor.

The charge/mass selective mode and linear acceleration mode are the key operative modes of ion implantation. In the charge/mass selective mode, the ionized species are pre-accelerated prior to entering a quadrupole magnet. The filtered beam is then postaccelerated and concentrated onto the biomaterial surface. However, in the linear acceleration type, all ionized species in the discharge chamber speed toward the biomaterial surface.<sup>319</sup>

Ion implantation has several advantages including making a treated surface without delamination problems or changing the bulk properties of the biomaterial.<sup>320</sup> Any kind of dopant can be introduced into any solid biomaterial in the defined areas of the surface, and the process temperature is low. Engineers use this approach for transferring energy into the surface layer of biomaterials, which results in changing the surface properties without any variations in the chemical state of biomaterials.<sup>320</sup>

However, at the time of ion bombardment all targeted areas need to be orthogonally exposed to the ion beam, which is a big limitation when it comes to surfaces with complex surface geometries. Hence, plasma-immersion ion implantation (PIII) is an effective approach in ion bombardment of inhomogeneous surfaces.<sup>313</sup> Park *et al.*<sup>321</sup> investigated the effects of tantalum ion immersion on the poly(lactic acid) (PLA) surface for improving its osteoblast affinity by using PIII in combination with conventional direct current magnetron sputtering. They revealed that tantalum ion-doped surfaces have twice the surface roughness and improve adhesion stability in comparison with tantalum-coated PLA surfaces. Tantalum doped ions improve the osseointegration and osteogenesis of implanted PLA surfaces *in vivo* through improving surface hydrophilic properties.<sup>321</sup>

Researchers also studied macrophage polarization responses to titanium implants doped with magnesium (0.1–0.35%).<sup>322</sup> Doping magnesium ions on the surface causes a greater expression of type 2 macrophages, anti-inflammatory cytokines (such as IL-4 and IL-10) as well as genes encoding bone morphogenetic protein 2 (BMP2) and VEGF.<sup>322</sup> There are still challenges about using ions on biomaterial surface for improving cell responses:

(i) As different ions can affect cell functions differently, simultaneously doping different types of ions on the surface



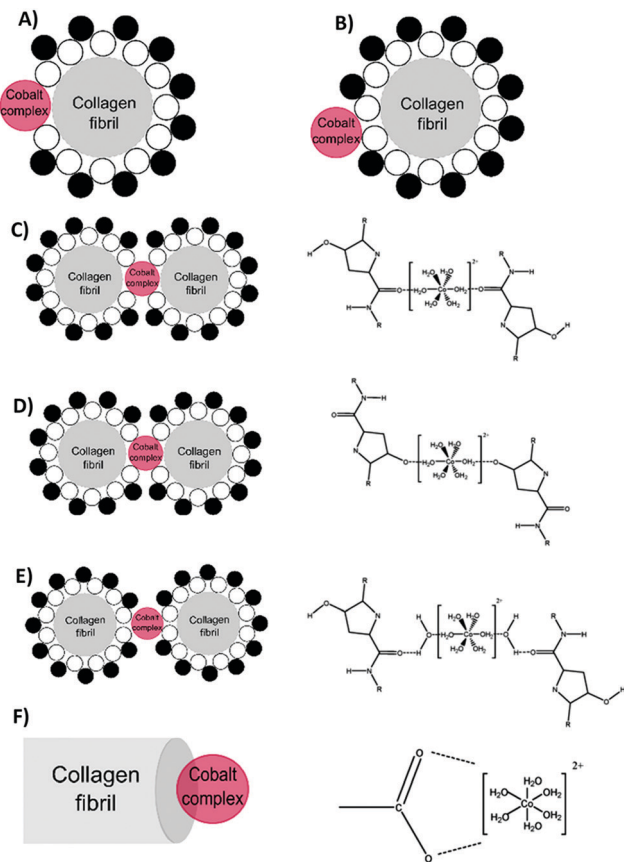


Fig. 17 A schematic representation of the binding sites of a cobalt complex with a collagen fibril. (A) The direct interaction of the cobalt complex with the collagen fibril. (B) The interaction of the cobalt complex with the non-freezing water layer surrounding the collagen fibril. Suggested binding sites of a cobalt complex with a collagen fibril, where interactions occur with the carboxylic group (C), hydroxyl group (D), forming a water bridge (E), or with the hydroxyl group in the carboxy terminus (F). Reprinted from ref. 309, copyright © 2018, Royal Society of Chemistry. Further permissions related to the material excerpted should be directed to the ACS.

of biomaterials is effective for providing multifunctional materials. Yu *et al.*<sup>323</sup> doped both zinc and magnesium ions on the titanium surfaces by using PIII. They detected that the dual implantation of ions enhances rat bMSCs' initial adhesion and spreading through the upregulation of the gene expression of integrin  $\alpha 1$  and integrin  $\beta 1$ .<sup>323</sup> Zn/Mg-PIII can also increase zinc and magnesium ion concentrations in the cells *via* enhancing the influx of both ions and hindering the outflow of zinc ions. Upregulating the expression of magnesium transporter 1 in human umbilical vein endothelial cells improves the magnesium ion influx, which leads to promotion of angiogenesis.<sup>323</sup>

(ii) The concentration of doped ions plays a key role in determining their influences on surface properties and biological pathways.<sup>309</sup> This indicates the crucial demand for precise systematic studies to find the optimal concentration of ions depending on the purpose.

### 9.2.3. Surface functionalization by using biological moieties.

Chemistry plays a crucial role in developing bioactive materials by

functionalizing the biomaterial surface with biomolecules to stimulate and/or mimic biological functions.<sup>344,345</sup> Surface functionalization with proteins through the functional groups, present on both the biomaterial surface and protein, has some advantages including its predictable biological effects, its production and analysis simplicity, as well as low cost.<sup>346,347</sup>

Surface functionalization with cell-specific molecules such as antibodies and receptor-targeting peptides improves the biomaterial surface affinity and specificity to the cell membrane.<sup>348</sup> Interactions between biomaterial surface and proteins can be extremely site-specific or non-specific (Fig. 18).

Adsorption/physorption is the mainly used strategy for non-site-specific conjugation of proteins to the surface, in which the protein–surface interactions are mainly directed through hydrophobic, hydration and electrostatic forces.<sup>349</sup> The incubation of a designed biomaterial with the targeted protein in buffer solutions is the main principle of this strategy. It is the most straightforward strategy for tethering proteins on the biomaterial surface.<sup>344,350,351</sup> However, non-covalently attached proteins can detach from the biomaterial surface after implantation and lose their functionality.<sup>352</sup> After implantation, the attached proteins can be replaced by adsorbed blood serum proteins through the Vroman effect.<sup>353</sup>

Therefore, there are several strategies for enhancing protein adsorption to surfaces including plasma treatments or using a transitional “sticky” layer such as mussel adhesive proteins.<sup>354</sup> Plasma treatment can induce different functional groups on the surface, which improves the protein conjugation.<sup>355</sup> Because aldehyde groups can trigger nucleophilic groups in lysine, arginine, asparagine and glutamine residues to make covalent imine bonds, plasma polymers, which have aldehyde groups on their surface, provide a reactive surface for covalent binding of proteins.<sup>356</sup>

In addition, engineers immobilize several proteins on the biomaterial surface by using PIII without changing their conformation. PIII is a one-step immobilization method, which through vapor-phase precursors produces radical species for covalently bonding with amino acid residues in proteins.<sup>357,358</sup> Using PIII, Tan *et al.*<sup>359</sup> developed a bioactive vascular graft coating with the macrophage polarizing cytokine IL-4 to control the macrophage phenotype and subsequent local inflammatory responses.<sup>359</sup> When subcutaneously implanted in mice, they observed that bioactive IL-4 surfaces could enhance the polarization of macrophages to their anti-inflammatory M2 phenotype. The functionalized surfaces could also positively regulate the local cytokine environment and reduce the foreign body responses.<sup>359</sup>

Mussel adhesive proteins are a coating layer for intermediating the biomolecule–surface interactions through their (*S*)-2-amino-3-(3,4-dihydroxyphenyl)propanoic acid containing a benzene ring with two neighboring hydroxyl groups.<sup>360</sup> However, due to the instability of non-specific attachment approaches, chemical conjugation methods have gained more attention for protein tethering on the biomaterial surface. In site-specific approaches, protein conjugation occurs through interactions between specific chemical groups in protein molecules and biomaterial surface,



Table 4 An overview of some of the recent studies on cell responses to ion implantation on the biomaterial surface

Materials	Doped ions	Other surface modifications	Main outcomes	Ref.
PEEK	Calcium	NA	Increasing rat bMSC adhesion, proliferation, and osteo-differentiation.	324
	Nitrogen	NA	MQMs: after 7 days of cell culture, cell proliferation on treated surfaces is almost 40% higher than that of untreated surfaces. Increasing the MG-63 responses and antibacterial activity of PEEK.	325
	Nitrogen	Coating tropoelastin	MQMs: after 3 days of cell culture, cell proliferation on treated surfaces could be almost twice that on untreated surfaces. Increasing SAOS-2 cell attachment, spreading, proliferation, and bone nodule formation.	326
Titanium	Copper	Coating PPAAm	MQMs: (I) after 7 days of cell culture, cell proliferation on PIII treated surfaces is 70 ± 8% higher than that on untreated surfaces. (II) After 6 days in osteogenic media, osteopontin expression on PIII treated surfaces, with and without tropoelastin, is 6-fold higher than that on untreated surfaces.	327
	Oxygen	NA	Decreasing the inflammatory responses of macrophages, antigen-presenting cells, and T lymphocytes to treated surfaces. MQMs: (I) decreasing the number of activated NK cells on copper-treated surfaces from a median of $0.91 \times 10^{-3}$ on day 7 to $0.46 \times 10^{-3}$ on day 56. (II) After 56 days, higher numbers of mast cells on copper-treated surfaces with median $2.29 \times 10^{-3}$ compared to Ti-Cu-PPAAm surfaces with median $1.97 \times 10^{-3}$ .	314
	Oxygen	NA	Improving fibronectin adsorption as well as hbMSC cell adhesion, migration, proliferation, mineralization, and differentiation. MQMs: (I) after 7 days of cell culture, cell proliferation on treated surfaces with different oxygen concentrations has a median of 0.840–1.041 compared to untreated surfaces with a median of 0.737. (II) Collagen type I expression on treated and untreated surfaces has a median of 129.70–154.91 and 100.00, respectively.	328
	Nitrogen	NA	Increasing responses of blood cells and antibacterial activities. MQMs: after 10 min, the optical density of blood (related blood clot formation) on treated surfaces is 50% less than that of the untreated group.	329
	Nitrogen	NA	Increasing blood clot formation, the adhesion of platelets, as well as adhesion and proliferation of HNEpC cells. MQMs: NA or difficult to summarize.	330
	Calcium & magnesium	NA	Increasing hbMSC cell adhesion, proliferation, and mineralization. MQMs: after 1 h incubation, the cell spreading area order of groups is high-dose oxygen > low-dose oxygen > untreated surfaces. After 7 days, cell proliferation on treated surfaces is 1.3-fold higher than that on untreated surfaces.	331
	Magnesium	NA	Increasing initial hbMSC cell attachment in ion-doped surfaces. There is no difference in cell proliferation among different groups. RUNX2 expression is greater on the magnesium ion-implanted surface. Osteocalcin expression is less on the calcium ion-implanted surface. MQMs: NA or difficult to summarize.	322
	Nitrogen & copper	NA	Increasing the number of type 2 macrophages and anti-inflammatory cytokine expression. MQMs: (I) after 4 days, the percentage of CD206, as a surface marker of type 2 macrophages, has the order Ti < Mg30 < Mg90 < Mg120. (II) The expression of C-C motif chemokine receptor 7, as the surface marker of type 2 macrophages, has the order Ti > Mg30 > Mg90 > Mg120.	332
	Silver	NA	Increasing angiogenic abilities of treated surfaces (especially dual ion treated surfaces) in response to HUVECs. MQMs: the number of cell migration on dual ion treated surfaces is almost 1.5-fold higher than that of untreated surfaces.	333
Titanium-nickel PU	Carbon	Coating DLC	Increasing implant stability <i>in vivo</i> as well as MG-63 cell responses toward neo-tissue formation, bone mineral density, and trabecular pattern. MQMs: bone implant contact for 30 min, 60 min, 90 min-Ag PIII treated surfaces as well as untreated surfaces are $73.18 \pm 5.23$ , $69.92 \pm 4.10$ , $66.05 \pm 3.97$ , and $61.99 \pm 4.66$ , respectively.	334
	Nitrogen	Coating collagen	Increasing MG-63 cell morphology, viability and spreading. MQMs: after 24 h cell incubation, the optical density value of treated surfaces is ~1.8-fold higher than that of untreated surfaces. Decreasing acute inflammatory responses (macrophages at day 28) in ion-collagen modified surfaces.	335
	Nitrogen	NA	MQMs: after 1, 3, 7, 14 or 28 days, the cell numbers in the surrounding capsule of treated implants is 42%, 41%, 51%, 43%, and 45% higher than that of untreated surfaces.	336
Polystyrene	Nitrogen & oxygen	NA	Increasing endothelial cell attachment and proliferation. MQMs: after 5 days of cell proliferation, the average cell density on treated surfaces is almost 3-fold higher than that of untreated surfaces.	337
		NA	Treated surfaces can be used for printing multiprotein micropatterns on the surface to investigate local effects on mouse pancreatic $\beta$ cell morphology and protein production. MQMs: after 11 days, the amount of immobilized BSA on treated surfaces is 30% lower than that of untreated surfaces.	337



Table 4 (continued)

Materials	Doped ions	Other surface modifications	Main outcomes	Ref.
Co-Cr	Tantalum	NA	Increasing endothelialization, platelet activation, and blood coagulation. MQMs: after 1 day cell culture, the cell spreading area and surface coverage of treated surfaces are 195% and 209%, respectively, higher than those of untreated surfaces.	338
	Oxygen	NA	Increasing endothelial cell viability after 7 days. MQMs: NA or difficult to summarize.	339
	Nitrogen	NA	Increasing the MSC osteogenic differentiation. MQMs: increasing integrin-binding sialoprotein expression and mineralization levels on treated surfaces 30-fold higher than those of untreated surfaces after 21 days of cell incubation.	340
Stainless steel	Nitrogen	Coating hydroxyapatite	Increasing hydroxyapatite growth and human oral fibroblast viability. MQMs: after 24 h cell culture, the percentage of cell growth on treated surfaces can be twice that of untreated surfaces.	341
	Silver	NA	Increasing antibacterial activity and osteogenic differentiation of hbMSCs. MQMs: NA or difficult to summarize.	316
Silicone	Tantalum	NA	Increasing hydrophilicity and human primary dermal fibroblast affinity. Decreasing fibrous capsule formation and contracture <i>in vivo</i> .	342
PLA	Tantalum	NA	MQMs: after 4 days, the cell spreading area on treated surfaces is more than twice that of untreated surfaces. Increasing MC3T3 cell adhesion, osseointegration and osteogenesis.	321
			MQMs: ALP activity on Ta-implanted PLA surfaces is 3.5 and 4.3 times more than Ta-coated and untreated PLA surfaces, respectively.	
PDMS	Oxygen	NA	Increasing Chinese hamster ovarian cell responses to surfaces. MQMs: after 2 days of cell incubation, the relative dead/live cells ratios are $1 \pm 0.037$ , $0.653 \pm 0.026$ , and $0.425 \pm 0.033$ on untreated, plasma treated, and PIII treated surfaces, respectively.	343

Abbreviation: poly ether ether ketone (PEEK), bone mesenchymal stem cells (bMSCs), plasma polymerized allylamine film (PPAAm), polyurethane (PU), extracellular matrix (ECM), primary human nasal epithelial cell (HNEpC), Runt-related transcription factor 2 (RUNX2), poly(lactic acid) (PLA), pre-osteoblast line (MC3T3), bovine serum albumin (BSA), cobalt-chromium (Co-Cr), human osteoblast-like cell line (SAOS-2), diamond-like carbon (DLC), human umbilical vein endothelial cells (HUVECs), polydimethylsiloxane (PDMS), bone morphogenetic protein 2 (BMP2), vascular endothelial growth factor (VEGF), major quantitative measurements (MQMs).



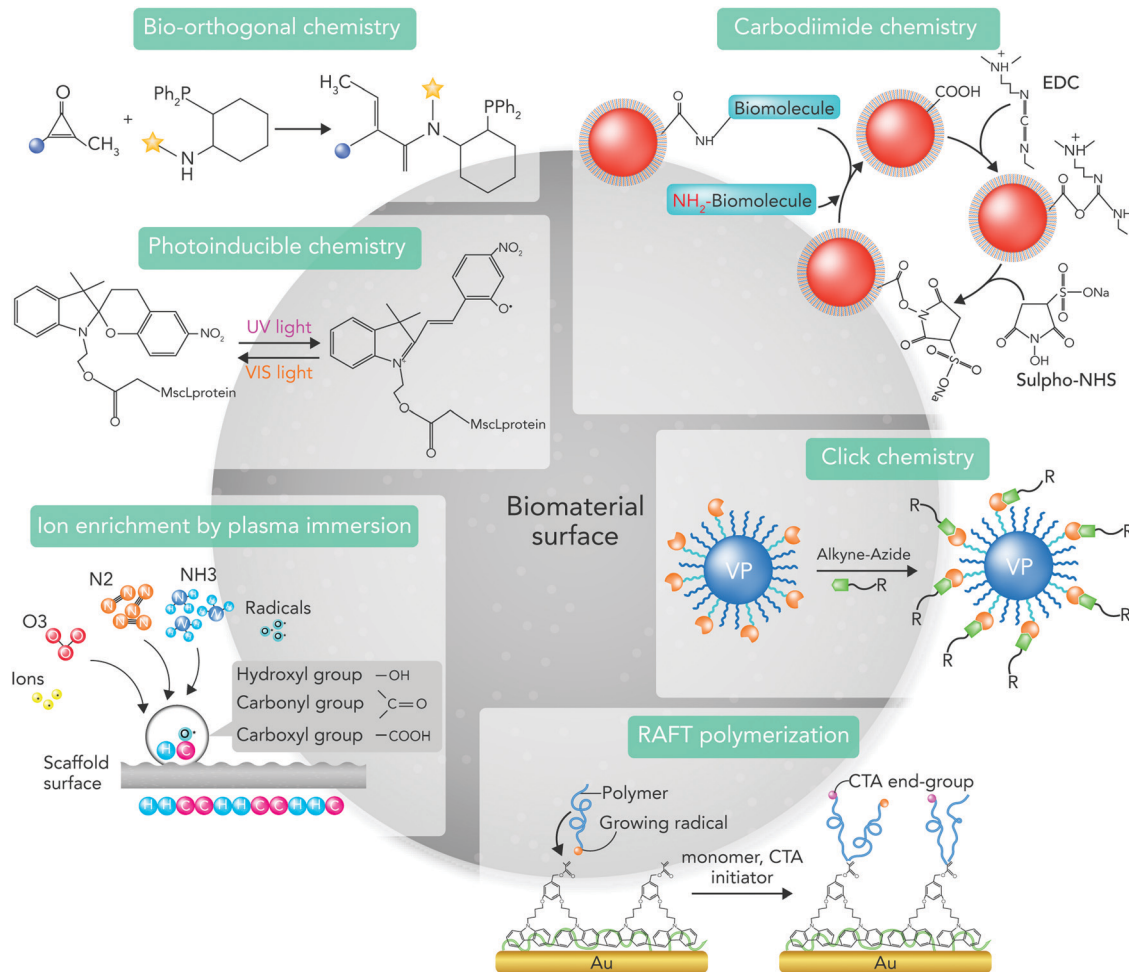


Fig. 18 The commonly used chemical strategies for engineering the biomaterial surface to manipulate biological responses.

which leads to proper protein orientation with high stability on the surface.<sup>344</sup>

Primary amine, carboxyl, sulfhydryl and carbonyl are the main chemical groups used for improving cell–biomaterial interactions.<sup>347</sup> Controlling chemo- and regio-selectivity and also the selection of treatment methods based on the surface properties of both protein and biomaterial are the main requirements of using site-specific strategies.<sup>361</sup>

Antibody conjugation on the biomaterial surface demands one of three functionalities in the antibody: (i) lysine amino acids, (ii) 12 cysteine residues, or (iii) 2–5 carbohydrate moieties in the Fc stem.<sup>362</sup> Amine conjugation occurs through an amine functionality, which is available on the surface of proteins or biomaterials.<sup>363</sup> Using covalent cross-linkers is one of the main strategies for establishing chemical interactions between proteins and cell surface, due to their ability to control the immobilization and accessibility of substrate surface biomolecules.<sup>364</sup>

Among cross-linkers, 1-ethyl-3-(3-dimethylaminopropyl)-carbodiimide/*N*-hydroxysuccinimide (EDC/NHS) can more efficiently make chemical interactions between amine functional groups of biomolecules and carboxylic groups of the

biomaterial surface.<sup>365</sup> The cross-linkers' non-cytotoxicity and water solubility of waste products make EDC/NHS a favorable candidate for mediating chemical interactions between cells and biomaterial surface.<sup>366</sup>

Additionally, the interaction between maleimide chemical compound ( $\text{H}_2\text{C}_2(\text{CO})_2\text{NH}$ ) and thiol ( $\text{R-SH}$ , where R represents an alkyl or aryl group) attracts much attention for establishing cell–biomaterial interactions. As the maleimide chemical compound can selectively react with cysteine residues in the protein, researchers use the thiol–maleimide conjugation reaction or immobilization of thiol-containing surfaces.<sup>273,367</sup> Cell surface thiols are present in oxidized disulfide bridges or reduced thiol group formations and can be labeled with a broad number of accessible reagents.<sup>368,369</sup> Some studies suggest using mono- and di-bromomaleimides as an additional reversible cysteine modification treatment on maleimide-based conjugation reaction.<sup>370</sup>

Reversible addition fragmentation chain transfer (RAFT) as a living radical polymerization reaction can also conjugate biomolecules to polymeric biomaterials.<sup>371,372</sup> RAFT-mediated bioconjugation can increase the chance of having a well-defined, site-directed bioconjugate architecture. The combination of



“Click” chemistry reactions with RAFT increases the effectivity and selectivity of protein conjugation on the substrate surface without interfering with the protein functionality.<sup>344,373</sup>

“Click” chemistry is defined as a class of small molecule chemical reactions, which joins a biomolecule and a reporter molecule, allowing coupling of substrates of choice with specific biomolecules. The reactions are fast, spontaneous, flexible, and very selective.<sup>374</sup> Over the past two decades, copper-based “click” chemistry reactions have been a common strategy for synthesizing hydrogels.<sup>375</sup> In addition, bio-orthogonal “click” chemistry is used to describe the way of generating products by joining small biomolecules such as proteins that occur in the presence of macromolecules such as proteins or cells.<sup>375</sup> Bio-orthogonal “click” chemistry is recognized as a promising molecular labeling approach, which does not affect normal biochemical processes.<sup>376</sup> Using these approaches, researchers can synthesize novel polymers and multifunctional hydrogels for biomedical applications.<sup>377,378</sup> “Click” chemistry synthesis occurs under stable physiological conditions, with highly stereo-specific and simple product separation. It does not produce any toxic end products. It is also insensitive to oxygen and water.<sup>375</sup>

Owing to its copper ion toxicity and ROS generation, copper-based “click” chemistry has been recently replaced with copper-free “click” chemistry strategies.<sup>379</sup> Copper-free “click” chemistry proceeds at a lower activation barrier and is free of cytotoxic catalysts or end products after gel formation.<sup>374,375</sup> There are many copper-free “click” chemistry strategies for designing hydrogels such as strain-promoted azide–alkyne cycloaddition “click” hydrogels,<sup>380</sup> Diels–Alder “click” chemistry hydrogels,<sup>381</sup> thiol–ene,<sup>382</sup> oxime,<sup>383</sup> and thiol–yne.<sup>384</sup> The available “click” chemistry-based functional hydrogels play key roles in fabricating 3D tissue and organ models by using 3D bioprinting.<sup>385,386</sup>

“Click” chemistry reactions improve biomolecule–surface conjugation in a controlled manner.<sup>387</sup> However, choosing or introducing suitable functional groups on proteins to mediate their attachment in a controlled manner without influencing their activity is still a big challenge in this field.<sup>344,361</sup>

**9.2.4. Biological responses to biomaterials interfacial free energy.** The interfacial free energy value is a critical factor affecting cell responses to biomaterial surface.<sup>388,389</sup> For example, Tojo *et al.*<sup>390</sup> revealed how the surface energy value of biomaterials direct MSC mechanosensitivity by culturing the cells on collagen-coated hydrophobic PDMS and hydrophilic PEO–PDMS.<sup>390</sup> They designed PDMS-based scaffolds and mechanically adjusted them within the stiffness range from 70 Pa to 2.3 MPa by using a surface energy gradient, without influencing the bulk properties and collagen topology of biomaterials. Their results indicated that the surface energy-driven ligand self-assembly could change the cell fate on soft surfaces. By adjusting the surface energy value, cells can spread and differentiate based on PDMS stiffness.<sup>390</sup>

Interfacial free energy between a solid and a liquid is a material’s property that is determined by its surface structure and chemical composition. It measures the disruption of intermolecular bonds, which occurs while creating a surface.<sup>388,389</sup>

Transferring an atom from the bulk of materials to the surface in response to liquids changes the surface energy value. The migrated atoms at the surface have fewer nearest neighbors than the same atoms positioned in the bulk. Consequently, the surface atoms have a greater energy state than atoms in the bulk, a phenomenon known as coordinative unsaturation of bonds.<sup>389</sup>

Thus, the type and amount of existing dangling bonds at the surface represent changes in the surface free energy value, which can be primary-(ionic, covalent, and metallic) as well as secondary-(van der Waals) type bonds.<sup>389</sup> If the dangling bonds are of the secondary type, the surface free energy is low and has a non-polar nature. However, if the bonds are mainly of primary type, a substantial Lewis acid and base contribute to the total surface free energy and its value is high.<sup>389</sup> Most surfaces have a combination of all these chemical bonds, which makes the surface interactions with the biological compounds complex.<sup>389</sup>

The surface energy value for high energy surfaces (*e.g.* metals and oxides) can be in the range from 500 to 5000 mN m<sup>-1</sup>; however, for low energy surfaces (such as molecular crystals and polymers) it is from 5 to 50 mN m<sup>-1</sup>.<sup>389</sup> Researchers commonly modify the surface free energy through changing the crystallographic termination of surfaces or adding ions using plasma treatments.<sup>391</sup>

The common technique for measuring the biomaterial surface energy and wettability is contact angle measurement.<sup>392–396</sup> The equilibrium state for liquids on a surface is reliant on both the thermodynamic equilibrium at interfaces and the total length/area of the phases in contact. Consequently, contact angle measurements can provide information about both surface free energy and wettability through the calculation of surface free energy from droplet geometry, or surface geometry from surface free energy.<sup>397–399</sup> Changes in both surface chemical composition and wettability can affect the obtained surface energy value.<sup>400–402</sup> Therefore, researchers should know the surface roughness and wettability before characterizing the surface energy using contact angle measurements.

As mentioned above, the intermolecular bonds at a biomaterial surface determine its surface free energy values. After biomaterial implantation, these chemical bonds interact with small molecules such as proteins. The chemical bonds also determine which molecules firstly adsorb, their orientation, conformation and bioactivity.<sup>389</sup>

Measuring wettability is a common method to determine the adsorption potential of proteins to biomaterial surfaces. However, because the relative wettability of many surfaces can be equal while their surface chemistries are different, these terms cannot precisely determine the surface interactions with proteins and cells.<sup>389</sup> It is commonly accepted that surfaces with high surface energy values and wettability enhance cell responses.<sup>403–408</sup> However, there is no direct correlation between hydrophilicity and surface free energy. Thus, investigating the effects of surface free energy on protein adsorption and cell responses, irrespective of surface wettability, is important.<sup>389</sup>





### 9.3. Nanofunctionalization of biomaterial surface

The ECM is composed of multifunctional nanostructures. Binding of cells to the ECM plays a vital role in regulating cell signaling pathways.<sup>409</sup> Therefore, bio-inspired nanofunctionalization, which combines biomimicry and nanotechnology, has attracted considerable attention as a promising strategy to modify the biomaterial surface.<sup>410–412</sup> Using patterning approaches, biomedical engineers can accurately position different biomolecules on a nano-scale surface to achieve site-specific attachment of biomolecules in some areas, while reducing undesirable surface interactions in other areas.<sup>413</sup> Techniques for producing nanostructures are commonly divided into two categories: (1) *in situ* surface nanofunctionalization, (2) nanocoating and film deposition. Electron beam, laser etching, acid and alkali treatments, anodic oxidation, and ion implantation are the most common *in situ* nanofunctionalization techniques.<sup>413,414</sup> Many nanocoating and film deposition techniques are also available including plasma spraying, plasma-immersion ion implantation and deposition, chemical or physical vapor deposition, cold spraying, lithography and self-assembly.<sup>413,414</sup>

By combining the above-mentioned techniques, more complex hybrid nanostructures can be designed.<sup>413</sup> For instance, Wang *et al.*<sup>415</sup> studied the synergistic effects of micro/nanostructure and bioactive ions on murine osteoblast responses to titanium surfaces containing bioactive ions ( $Zn^{2+}$  and  $Sr^{2+}$ ). They created surfaces by using a combination of sandblasting, acid etching, alkali-heat treatment, and ion exchange techniques. Compared to polished titanium surfaces, the micro/nanostructured functionalized surfaces could significantly enhance cell spreading, proliferation, and differentiation.<sup>415</sup> Furthermore, Frey *et al.*<sup>416</sup> used micelle nanolithography and soft micro-lithography to develop PEG-based hydrogels with a micro-grooved surface. They also incorporated gold nanoparticles on the surface to improve the binding of adhesive ligands. Compared to conventional micro-grooved surfaces, the nanofunctionalized surface could improve human fibroblasts' contact guidance and regulate cell signaling more accurately.<sup>416</sup>

Christo *et al.*<sup>417</sup> developed a biomaterial surface with controlled nanotopography and chemistry by combining plasma polymerization and electrostatic self-assembly techniques to evaluate the inflammasome responses.<sup>417</sup> They assessed the innate immune responses by using bone marrow derived macrophages harvested from genetically engineered mice deficient in apoptosis-associated speck-like protein containing CARD, NLRP3 and AIM2 inflammasome components. Their results showed that the macrophage adhesion changes on all controlled nanotopography surfaces irrespective of their surface chemistry or nanotopography scale. Although other studies reported that different surface chemistries influence the initial binding of serum proteins and cell attachment,<sup>418,419</sup> Christo and colleagues could not detect any difference between groups. However, both chemistry and nanotopography could change the macrophage functionality in the absence of main inflammasome components suggesting that these components could be key players in macrophage responses to surface nanofunctionalization.<sup>417</sup>

In another study, Marzaioli *et al.*<sup>420</sup> studied the effects of silica nanoparticle functionalization on inflammasome signaling pathways using murine bone marrow-derived dendritic cells and a mouse model of mild allergic inflammation.<sup>420</sup> They showed that non-functionalized surfaces activated the NLRP3 inflammasome response leading to the expression of inflammatory cytokines and chemokines. However, surface functionalization with phosphate or amino groups reduced the inflammasome activation.<sup>420</sup>

Owing to their chemical and structural features, proteins, peptides, and ligands play key roles in surface nanofunctionalization. Researchers commonly design biomaterials with integrin-specific ligands on their nanostructured surface for enhancing cell responses.<sup>421</sup> In addition, the ligand clustering on biomaterial surface can regulate intracellular signaling events that affect cellular phenotype. Karimi *et al.*<sup>421</sup> developed RGD-functionalized copolymers using RAFT polymerization to study the effects of nano-scale clustering of integrin-binding ligands on endothelial cell functions (Fig. 19).<sup>421</sup> They used the synthesized copolymers to prepare random and nano-clustered surfaces spanning different global and local RGD densities. Their results indicated that nano-clustering ligands on the surface could promote endothelial cell adhesion and migration.<sup>421</sup>

Proteins can be also patterned onto a surface by micro-contact printing or dip-pen nanolithography.<sup>422–424</sup> After immobilizing biomolecules, the surface arrays can display a selective capture of proteins from a mixture like serum.<sup>425</sup> Additionally, when the captured proteins are patterned on a desirable surface, the specific bound target proteins can be transferred to another surface by microcontact printing, which is a simple method for fabricating surface arrays of different proteins.<sup>426</sup>

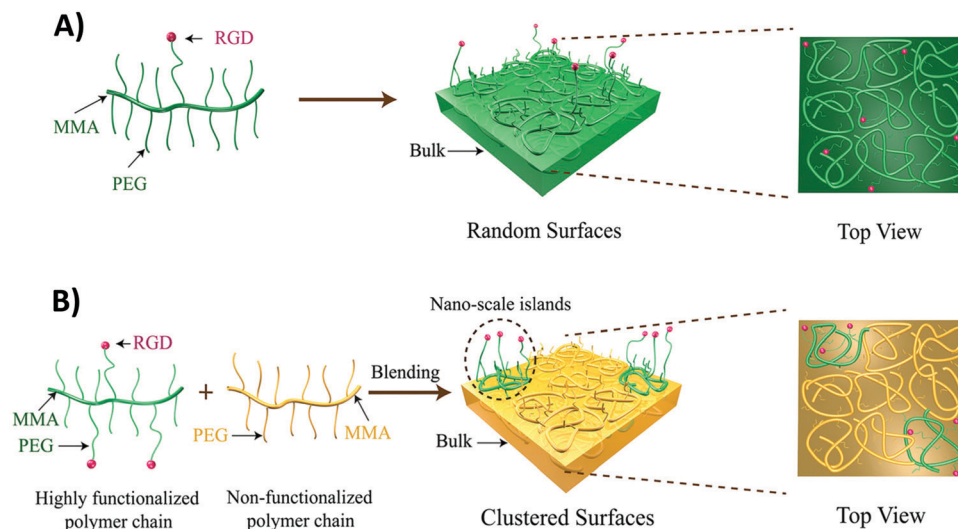
Despite the substantial progress in this field, there are still some challenges to be solved. Thoroughly investigating the thermodynamics and kinetics of protein adsorption on nanostructures is vital.<sup>427</sup> In addition, proteins have unfolding potential upon adsorption on nanostructures leading to tissue damage. Researchers should more profoundly study both unfolding and stabilization potential of proteins on nanostructures before their implantation in the body.<sup>427</sup>

## 10. Chemical surface analysis

Owing to the advances in analytical chemistry, biomedical researchers have achieved in depth information about the surface composition and structure of both biomaterials and biological molecules.<sup>428–430</sup> The commonly used analytical techniques for biomaterial surface characterization include secondary-ion mass spectrometry (SIMS), time of flight-SIMS (ToF-SIMS), X-ray photoelectron spectroscopy (XPS), auger electron spectroscopy (AES), (Raman), infrared (IR) spectroscopy, X-ray scattering and spectroscopy techniques, scanning electron microscopy-energy dispersive X-ray (SEM-EDX), atomic force microscopy (AFM) and streaming potential measurements/electroosmosis.<sup>431–434</sup>

Among these techniques, XPS, ToF-SIMS, Raman, and IR spectroscopy have gained substantial attention for characterizing





**Fig. 19** Fabricating random and clustered surfaces using nanofunctionalization strategies. (A) Random surfaces were developed by film casting lightly functionalized polymer chains. (B) Clustered surfaces were developed by film casting blends of highly functionalized polymer chains (green) with non-functionalized polymer chains (orange) to fabricate nano-scale islands of high peptide density defined by the random coils of the polymer molecule. Reprinted from ref. 421. Copyright © 2017, Royal Society of Chemistry.

the surface composition after applying surface treatments.<sup>431,434,435</sup> XPS and ToF-SIMS have been the main techniques used for evaluating the biological responses to surface chemistry because of several advantages. (I) These techniques can provide information about the interfacial intermolecular forces that control the biomaterial-biomolecule interactions through analyzing the surface composition and structure over the depth scales of a few nanometers.<sup>434</sup> (II) They can be useful for detecting surface contamination, which can be critical when interpreting cell responses to biomaterials. (III) They can provide high quality images of the spatial distribution of surface composition by combining imaging software technologies with narrowly focused ion guns. (IV) The recent progress in imaging technologies has led to using these techniques in the nanotechnology field for analyzing the designed patterned surfaces with more sensitivity and resolution.<sup>434,436</sup>

These two techniques are complementary and both are required to thoroughly analyze the physicochemical composition of biomaterial surface, because they can overcome each other's drawbacks.<sup>436</sup> For instance, XPS cannot differentiate isotopes, while ToF-SIMS can do it. Although quantification is commonly done by XPS, it is difficult to do it by using ToF-SIMS.<sup>436</sup>

Oteri *et al.*<sup>428</sup> designed a biomaterial based on calcium triphosphate and hydroxyapatite microgranules for bone regeneration.<sup>428</sup> They used ToF-SIMS and XPS to analyze the chemical composition and the micromorphological structure of the surface and confirm its osteoconductivity.<sup>428</sup> In another study, Stevanovic *et al.*<sup>437</sup> coated a layer of chitosan hydroxyapatite and gentamicin on pure titanium plates.<sup>437</sup> They used XPS and FTIR to confirm that the coating layer was connected to the titanium surface through intermolecular hydrogen bonds.<sup>437</sup> Moreover, Metoki *et al.*<sup>438</sup> studied the calcium phosphate electrodeposition on a titanium alloy covered with SAMs regarding the chain length, end-group charge, and anchoring group.<sup>438</sup> They used ToF-SIMS

to confirm that calcium-rich phases were formed primarily in the presence of SAMs.

Analytical chemistry can also be useful to provide information about the chemical characterization of surface bound proteins.<sup>430,433,435</sup> Proteins are recognized as key players in determining biomaterial-cell interactions and many biological pathways.<sup>14</sup> The protein structure, which is the 3D arrangement of atoms in an amino acid-chain, can break down into primary structural units, secondary structural units, tertiary structural units, and quaternary structural units.<sup>433</sup> However, because proteins are extremely flexible and can easily restructure when they meet a surface, analyzing the surface structure after binding to large proteins is challenging and requires combining experimental and simulation techniques.<sup>439</sup> Furthermore, analyzing the well-defined, smaller subunits of larger proteins (such as peptides) could be a promising solution toward developing better methodologies for this purpose.<sup>433</sup> XPS, surface plasmon resonance, and quartz-crystal microbalance with dissipation can be used for characterizing peptides and protein surface coverage in single component protein solutions.<sup>433</sup> However, on surfaces covered with a mixture of proteins, these techniques can only determine the total protein surface coverage.

ToF-SIMS can determine the unique signatures from different proteins adsorbed on these surfaces.<sup>440</sup> Nevertheless, when the protein films become more complex, ToF-SIMS can only provide a qualitative measurement about the total composition of the films.<sup>433,441</sup> ToF-SIMS can also provide information about the overall protein orientation using selected peaks from asymmetrically distributed amino acids in the protein structure.<sup>442,443</sup> Additionally, ToF-SIMS and sum frequency generation techniques can give molecular level details about the orientation of surface bound peptides and proteins.<sup>444</sup> The obtained molecular level data from these techniques can be used to govern the molecular



dynamics simulations, which provide atomic-level structural information about surface bound peptides.<sup>445</sup>

For instance, Foster *et al.*<sup>435</sup> studied the adsorption of single-component bovine serum albumin, bovine fibrinogen, and bovine immunoglobulin G films as well as multicomponent bovine plasma films onto two different gold surfaces.<sup>435</sup> They used a multitechnique method based on XPS, ToF-SIMS, principal component analysis (PCA), and visual molecular dynamics (VMD) to obtain new information about the structure of proteins. Using XPS, they could evaluate the adsorption isotherms and show the effects of protein solution amount on the surface nitrogen composition. Using a combination of ToF-SIMS, PCA, and VMD, they could also provide some information about the differences in adsorbed protein structure on different surfaces. They finally analyzed the amino acid distributions in the adsorbed proteins by using PCA and VMD.<sup>435</sup>

Even though substantial progress has been made in the analytical chemistry of biomaterial and biomolecule surfaces, there are still many challenges that should be addressed. Over the past few years, researchers have suggested new strategies to improve ToF-SIMS applicability such as applying multivariate analysis methods for data mining, as well as using cluster-ion beams and metal-assisted SIMS.<sup>436</sup> Nevertheless, all the current available techniques have their own drawbacks, which need to be addressed. For instance, both XPS and ToF-SIMS measure in dry state, need vacuum, and are contamination sensitive.<sup>434</sup> ToF-SIMS has also inadequate optical capabilities and problems in collecting positive or negative ion data.<sup>431</sup> XPS cannot detect hydrogen and helium.<sup>434</sup> The current techniques cannot provide atomic-level structural information about biomolecules.<sup>433</sup> Future research dealing with analytical techniques should emphasize on the combination of most developed imaging and analytical chemistry techniques to go beyond the current limitations.

## 11. Using the 3D bioprinting technique for improving cell–biomaterial interactions

Over the past few decades, 3D bioprinting technology has emerged as a promising approach for designing patient-specific scaffolds by depositing biomaterials and living cells layer-by-layer based on a digital model.<sup>446–448</sup> It draws knowledge from developmental biology, chemistry, computer science, and materials science to fabricate biological substitutes mimicking their native counterparts.<sup>449</sup> 3D bioprinting techniques can precisely print and control the geometrical localization of living cells into biomaterials.<sup>446</sup> Therefore, researchers use 3D printed tissues in many biomedical engineering fields including tissue engineering and regenerative medicine,<sup>450</sup> transplantation,<sup>451</sup> drug testing and high-throughput screening,<sup>452</sup> and cancer research.<sup>453</sup> The 3D bioprinted tissue models can be promising substitutes for the current 2D cell culture and animal models to test drugs and biomaterials *in vitro*.<sup>446</sup>

Materials used for constructing 3D bioprinted tissues include biomaterials, living cells, drugs, growth factors, and genes.<sup>446</sup> Thermoplastic polymers, hydrogels, and decellularized extracellular matrix (dECM) are three main types of biomaterials used in combining cells and biomaterials field research.<sup>448,454,455</sup> Thermoplastic polymers such as PCL, PU, and PLA can be used as structural supports owing to their high mechanical properties.<sup>448</sup> However, printing thermoplastic polymers needs either high temperatures or toxic solvents, leading to low cytocompatibility. Furthermore, combining these polymers with cell-supportive hydrogels is a challenge in the printing process.<sup>448</sup>

As reviewed by Gopinathan *et al.*,<sup>375</sup> owing to the emergence of “click” chemistry strategies, many natural and synthetic hydrogels are currently available to 3D bioprint tissue models.<sup>375</sup> Natural hydrogels such as chitosan, collagen, alginate, and gelatin can provide a native ECM-like microenvironment for cells.<sup>456</sup> However, compared to natural hydrogels, the mechanical properties and cell-adherent characteristics of synthetically derived hydrogels (such as methacrylated gelatin, PEG, and polyoxyethylene–polyoxypropylene triblock copolymers) can be more easily manipulated.<sup>448</sup> Although increasing the chemical concentrations and crosslink capacities leads to better printability and shape fidelity, they can cause smaller pore size and lower cell viability.<sup>457,458</sup>

Because dECM has the ECM constituents, it can overcome these challenges.<sup>459</sup> However, its low post-printing shape fidelity and ethical issues should be solved before translating the designed models to clinical settings.<sup>448</sup> Designing composite biomaterials can be a solution to overcome the weakness of each component regarding mechanical strength, printability, biocompatibility, and gelation properties.<sup>448</sup>

Although bioprinting has been a powerful tool for creating 3D tissue models, there are still several challenges that we should address, as follows:

(i) Constructing whole organs. Organs are complex structures containing different cell types and gradients of physicochemical properties.<sup>460</sup> In addition, native organs have adequate mechanical strength to keep their shape and integrity over time.<sup>461</sup> Therefore, researchers should enhance the printing resolution to create 3D organ models with internal complex networks.<sup>446</sup>

(ii) Vascularization. Vascular networks are essential for providing passage to nutrients, oxygen, and metabolic wastes.<sup>448</sup> Fabricating scale-up tissues or organs with a functional vascular network that enables effective local innervation remains a critical challenge for the field.

(iii) Cell culture technique. Cells are one of the key elements of bio-inks, which should be cultured in large numbers. By using the current available techniques, cell culture expansion processes take from weeks to months for each cell type.<sup>462</sup> It is vital to develop new techniques, which are capable of speeding up the cell expansion time without distorting cells.<sup>446</sup>

(iv) Fabricating functional tissues. Cell responses, scaffold stability, and ECM deposition are three main elements of a



functional tissue construct.<sup>463,464</sup> The 3D ECM microenvironment should allow differentiation and proliferation of printed cells. Designing new biomaterials and modifying the properties of available ones are essential to facilitate ECM–cell signaling.<sup>446</sup> In addition, controlling the biomaterial degradation rate is crucial to make sure the synthetic ECM degradation rate is proportional to the native ECM production.<sup>465</sup>

(v) Improving the mechanical strength of hydrogels as synthetic ECMs. There are two main strategies for enhancing the mechanical strength of hydrogels. (I) Decreasing the construction time can enhance the mechanical strength of printed structures. However, slower printing speed leaves the cells longer inside the ink, which may not be beneficial for cells. To overcome this issue, researchers introduced the 4D bioprinting concept. It allows a 3D printed structure to change its configuration and/or function over time in response to external stimuli such as temperature, light, and water, which makes 3D printing alive.<sup>446,466,467</sup> (II) Improving the mechanical strength of hydrogels by co-printing them with tough degradable biomaterials.<sup>462</sup>

## 12. State-of-the-art of evaluation of biological responses and future perspective

Based on the International Organization for Standardization (ISO) and national standards, we must prove the biocompatibility of any biomaterial or medical device by doing a series of tests regarding its genotoxicity, carcinogenicity, toxicity, sensitization, as well as acute and chronic systemic toxicity prior to doing any clinical trials in humans.<sup>468,469</sup> Since 1992, ISO has published and modified a series of international biocompatibility standards for medical devices (ISO 10993), as a living and regularly evolving document, which provides the general principles that we should bear in mind during both *in vitro* and *in vivo* evaluation of material–tissue interfaces.<sup>57,470–472</sup> Despite applying a wide range of chemical analyses for evaluating the biocompatibility of biomaterials based on ISO standards, there are still some challenges in the evaluation of biomaterials by following the standards.

In 2010, Poly Implant Prothèse (PIP), a French manufacturer of silicone gel breast implants, went bankrupt after using low-grade industrial silicone gel in their products.<sup>473</sup> It is clear that their implants should not have passed the defined biocompatibility tests of ISO standards 10993. Consequently, the European governments and competent authorities set an additional approval process named as Medical Device Regulation (MDR 2017/745) to address the biocompatibility evaluations of medical devices.<sup>473</sup>

The main purpose of MDR is to improve both the report quality and clinical trials data availability, rather than changing ISO standards. In 2021, the results of clinical trials will be available for the public on the European database for medical devices (EUDAMED),<sup>474,475</sup> which could be a helpful tool for the rapid reviewing of available medical devices' biocompatibility properties.<sup>476</sup> As it would be more expensive to bring new

medical devices to the market, one could argue that the new MDR will reduce the number of novel biomaterials in the future.<sup>477</sup> Thus, it is crucial to ask if there are any alternatives for improving the translation reliability between “pre-market” clinical trial requirements and the “post-market surveillance”. The “post market surveillance” is defined as the clinical follow-up of the Conformité Européenne (CE) approved medical devices, whereas the “pre-market clinical trials” are the required testing prior to CE mark. There is still a question that if the new MDR will not amend the *in vitro* biocompatibility tests, what would be the alternatives to determine the cellular and molecular pathways and responses to new biomaterials more accurately than ISO standards?<sup>478</sup>

Several studies pointed out the weakness of ISO 10993-5. The cell lines are commonly tested on extracts. Obviously, using extracts cannot provide us valid information about the biochemical transduction pathways and signals, which affect the biocompatibility of materials. Additionally, extracts of biomaterials may show cytotoxic effects on the cultured cells caused by changes in the ionic composition of the medium.

Although using cell lines might provide reproducible results, it is difficult to predict the cells' biochemical and biophysical signals in response to implanted biomaterials in the body based on these results. The biochemical transduction pathways cannot be examined and detected precisely through using such immortalized cell lines.<sup>479–482</sup>

Even though cytotoxicity is one of the most important indicators for biomaterial evaluation, all present cytotoxicity test methods have certain drawbacks.<sup>468</sup> Considering the other potential key biochemical signals which might be involved in biological responses to a material and can affect the evaluation criteria of material safety in the body is important.<sup>12</sup>

Traditionally biocompatibility was investigated by considering the wound healing processes that occur after biomaterial implantation. The *in vivo* evaluation of biological responses to biomaterials involves implanting biomaterials in the targeted tissues at certain time points and then studying the histological changes in the implanted tissue and its surroundings. However, in the case of tissue-regenerated biomaterials, one would keep in mind not only wound healing processes, but also several other physiological, mechanical and biochemical pathways, which might be involved in determining the material's success or failure. In addition, there is a lack of enough valid chemical techniques for evaluating the biochemical features of surface biocompatibility *in vivo*. The main biochemical signaling pathways involved in host responses should be evaluated based on the properties of biomaterials and targeted cells.<sup>11</sup> As the biological responses to biomaterials is dependent on the cell type used, we emphasize using suitable cell lines for *in vitro* cytotoxicity testing.<sup>57</sup>

Regarding animal studies, the choice and design of animal models for testing biomaterials are complex and there are few guidelines. To evaluate both intended and unintended effects, we use several criteria to evaluate implants, including biological relevance, biofunctionality, biocompatibility/safety, and clinical relevance/efficacy. Safety studies often use smaller



species (e.g. rodents and rabbits) to detect local tissue damage and systemic toxicity from degradable or leachable products.<sup>483,484</sup> However, tissue reactions to specific biomaterials are tested typically in larger animals to provide more detailed biological information.<sup>476,484</sup> The anatomy, physiology and pathogenicity of experimental models should relate as much as possible to those of patients in order to demonstrate the safety and efficacy of new biomaterials.<sup>485</sup>

Knock-out methods are among the most common systems used for investigating molecular pathways in animal models. However, we should consider that these knock-out animal models are not able to target several natural pathways that might be alongside responsible for one mechanism.<sup>57</sup> There is an argument in the literature over communal animal implant models for evaluating host material response and trying to link common host responses across various species in response to biomaterials. Because of the complexity of the human biological pathways (including biochemical, biophysical, mechanical and physiological pathways), choosing the right animal model based on the aim of the study is a big issue in this field. Hence, as Grainger<sup>486</sup> stated in his review “*Surprisingly, little consensus or consistency is found in published literature for host-implant integration metrics for implant healing versus foreign body response.*” The translation of preclinical animal data into successful clinic products is typically poor.<sup>487–489</sup>

Consequently, we should use different types of animal models for each newly designed biomaterial to mimic all the potential biological pathways involved in human body responses to biomaterials, which increases the ethical issues and complexity in this field. If we assess the preclinical animal data shortcomings more systematically, we can achieve more reliable preclinical *in vivo* data. Typical deficiencies that must be overcome are poor design of animal models, acute models opposed to chronic ones, poor blinding and randomization, and statistically underpowered studies.<sup>490</sup>

Over the past few years, researchers have suggested using organoids as an alternative technology for biocompatibility assessment.<sup>491</sup> An organoid is a 3D structure, grown from stem and/or progenitor cells, consisting of organ-specific cell types, which organizes itself through cell sorting and spatially restricted lineage commitment. These models can provide systematic information about chemical and genetic perturbation of 3D tissue. Such models have substantial potential in bridging the gap between *in vitro* biocompatibility and animal models and might be a valid alternative to *in vivo* animal studies. The great advantage of using organoids could be discovering chemical-transduction pathways, which cannot be achieved by current *in vivo* techniques. Conventional *in vitro* cell cultures are based on 2D systems, which often fail to induce full proliferation and differentiation potential of cells. There is a massive discrepancy between 2D and 3D *in vitro* cell culture environments, where the latter system allows cell-to-cell and cell-to-biomaterial interactions, more similar to the body environment. This disparity between 2D and 3D systems makes it difficult for 2-D cell-based *in vitro* models to be reliable for investigating disease mechanisms and drug screening.<sup>492</sup>

Although it is doubtful that organoids will completely replace animal studies, we should evaluate the obvious advantages of using organoids in biocompatibility tests.<sup>493</sup> As Bredeboord *et al.*<sup>494</sup> stated: “*We suggest that the use of organoids is complementary to, rather than in competition with, these classical research methodologies.*”<sup>494</sup>

*In silico* bioinformatics are computational frameworks, which are currently playing a key role in the discovery and validation of new chemical identities. Therefore, using these computational models for also predicting the biochemical responses to biomaterials could be a revolution in biological response evaluation. Such computational models of tissue engineering processes attracted much attention as new non-invasive evaluation techniques, which can decrease or replace *in vivo* animal studies without having any ethical issues.<sup>495–498</sup> Although these computational models might strongly help in evaluating and predicting biomaterial–host complex interactions and decrease the number of unnecessary animal studies, without doing any animal studies we cannot make well-founded decisions regarding the safety of biomaterials and directly implant them in the human body with confidence.<sup>12</sup>

Considering more quantitative and statistical analyses can help us to make conclusions that are more convincing. Due to the immense development of new biomaterials for tissue engineering applications, researchers design more and more tissue-engineered templates with different chemistry. More reliable analytical techniques should be established alongside to provide atomic-level structural information about biomolecules. Combining imaging and analytical chemistry techniques can be a promising strategy to go beyond the current limitations of analytical chemistry.

### 13. Conclusions

It is undeniable that the medicine world owes chemistry science for the current existence of promising biomaterials as therapeutic solutions. By using the chemical principles present in nature, researchers have made substantial progress in biomaterials and tissue engineering fields. Nevertheless, there are still critical challenges regarding their biocompatibility that need to be further studied and are typically neglected when describing new therapeutic strategies using biomaterials. The biological responses to biomaterials should not be defined as only not having any adverse effects. They should have a strong affinity for targeted cells to stimulate neo-tissue formation, which depends on the chemical characteristics of both the biomaterials and biological environments. The host responses to biomaterials mainly originate from biochemical signaling pathways and factors. As our knowledge in the biochemistry field has grown tremendously since the first definition of biocompatibility, we should update our definitions about biological responses to biomaterials. In addition, from the materials engineering point of view, biomaterial surface physicochemistry can profoundly affect biophysical and biochemical responses. Providing more reliable analytical techniques and standards for



biochemical interface characterization of biomaterials is highly important. Nevertheless, the chemical basis of biological responses to biomaterials is a domain where much remains to be studied.

## Abbreviations

$R_a/S_a$	Roughness value
PDMS	Polydimethylsiloxane
NIPAM	<i>N</i> -Isopropylacrylamide
HepG-2	Hepatocellular carcinoma cell line
PAAm	Polyacrylamide
PES	Poly(ether sulfone)
CPES	PES with carboxylic groups
SNPES	PES with sodium sulfonic and amino groups
OD	1,7-Octadiene
FBS	Foetal bovine serum
PIII	Plasma-immersion ion implantation
PLA	Poly(lactic acid)
BMP2	Bone morphogenetic protein 2
RAFT	Reversible addition fragmentation chain transfer
ECM	Extracellular matrix
IL	Interleukin
PHSRN	Proline–histidine–serine–arginine–asparagine
RGD	Arginine–glycine–aspartic acid
PAMPs	Pathogen associated molecular patterns
DAMPs	Damage-associated molecular patterns
TLRs	Toll-like receptors
PRRs	Pattern recognition receptors
NLR	NOD-like receptor
ALP	Alkaline phosphatase
MSCs	Mesenchymal stem cells
PGE2	Prostaglandin E2
Tregs	T regulatory lymphocytes
hMSCs	Human-MSCs
VEGF	Vascular endothelial growth factor
FAK	Focal adhesion kinase
RANKL	Receptor activator of nuclear factor kappa-B ligand
SIMS	Secondary-ion mass spectrometry
ToF-SIMS	Time of flight-SIMS
XPS	X-ray photoelectron spectroscopy
IR	Infrared
SEM-EDX	Scanning electron microscopy-energy dispersive X-ray
PC	Principal component analysis
VMD	Visual molecular dynamics
FBGC	Foreign body giant cell
AIM2	Absent in melanoma 2
CAV1	Caveolin 1
ROS	Reactive oxygen species
TRPM	Transient receptor potential melastatin
NMDA	<i>N</i> -Methyl-D-aspartate
ATP	Adenosine triphosphate
P2XRs	P2X receptors

KCa3.1	Ca <sup>2+</sup> -Activated K <sup>+</sup> channel
cGMP	Cyclic guanosine monophosphate
AFM	Atomic force microscopy
PDMAEMA	Poly(2-(dimethylamino)ethyl methacrylate)
ATRP	Atom transfer radical polymerization
BSA	Bovine serum albumin
QCM-D	Quartz crystal microbalance with dissipation
SIMS	Secondary-ion mass spectrometry
ToF-SIMS	Time of flight-SIMS
XPS	X-ray photoelectron spectroscopy
IR	Infrared
SEM-EDX	Scanning electron microscopy-energy dispersive X-ray
PCA	Principal component analysis
VMD	Visual molecular dynamics

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

This work was supported by a project “Promoting patient safety by a novel combination of imaging technologies for biodegradable magnesium implants, MgSafe” funded by European Training Network within the framework of Horizon2020 Marie Skłodowska-Curie Action (MSCA) grant number no. 811226 (www.mgsafe.eu). Eduardo A. Silva was supported by the American Heart Association grant # 19IPLOI34760654/Eduardo Silva/2019. We would like to thank Prof. Silvia M. Rogers, the head of MEDWRITE Company, Basel, Switzerland, who kindly helped us with her professional comments on the scientific writing part of this paper. We also thank Dr Matilde Bongio (www.matildebongio.com), who designed some informative illustrations for this paper.

## References

- 1 F.-M. Chen and X. Liu, *Prog. Polym. Sci.*, 2016, **53**, 86–168.
- 2 M. Rahmati, C. P. Pennisi, A. Mobasheri and M. Mozafari, *Cell Biology and Translational Medicine*, Springer, 2018, vol. 3, pp. 73–89.
- 3 A. Khademhosseini and R. Langer, *Nat. Protoc.*, 2016, **11**, 1775–1781.
- 4 C. P. Bergmann and A. Stumpf, *Dental Ceramics*, Springer, 2013, pp. 55–65.
- 5 M. Rahmati, C. P. Pennisi, E. Budd, A. Mobasheri and M. Mozafari, *Cell Biology and Translational Medicine*, Springer, 2018, vol. 4, pp. 1–19.
- 6 S. J. Lee, J. J. Yoo and A. Atala, *Clinical Regenerative Medicine in Urology*, Springer, 2018, pp. 17–51.
- 7 S. N. Alhosseini, F. Moztaarzadeh, M. Mozafari, S. Asgari, M. Dodel, A. Samadikuchaksaraei, S. Kargozar and N. Jalali, *Int. J. Nanomed.*, 2012, **7**, 25.



- 8 V. Shabafrooz, M. Mozafari, D. Vashae and L. Tayebi, *J. Nanosci. Nanotechnol.*, 2014, **14**, 522–534.
- 9 K. Nazemi, F. Mozartzadeh, N. Jalali, S. Asgari and M. Mozafari, *BioMed Res. Int.*, 2014, **2014**, 898930, DOI: 10.1155/2014/898930.
- 10 D. F. Williams, *Biomaterials*, 2014, **35**, 10009–10014.
- 11 D. F. Williams, *ACS Biomater. Sci. Eng.*, 2016, **3**, 2–35.
- 12 D. F. Williams, *Tissue Eng., Part C*, 2017, **23**, 926–937.
- 13 D. F. Williams, *Prog. Biomed. Eng.*, 2019, **1**, 013001.
- 14 M. Rahmati and M. Mozafari, *Mater. Today Commun.*, 2018, **17**, 527–540.
- 15 C. J. Kirkpatrick, *Regener. Biomater.*, 2015, **2**, 267–272.
- 16 K. Sadtler, A. Singh, M. T. Wolf, X. K. Wang, D. M. Pardoll and J. H. Elisseeff, *Nat. Rev. Mater.*, 2016, **1**, 16040.
- 17 B. Reid, M. Gibson, A. Singh, J. Taube, C. Furlong, M. Murcia and J. Elisseeff, *J. Tissue Eng. Regener. Med.*, 2015, **9**, 315–318.
- 18 H. Rostam, S. Singh, N. Vrana, M. Alexander and A. Ghaemmaghami, *Biomater. Sci.*, 2015, **3**, 424–441.
- 19 A. Albanese, P. S. Tang and W. C. Chan, *Annu. Rev. Biomed. Eng.*, 2012, **14**, 1–16.
- 20 S. Bajpai, N. Y. Kim and C. A. Reinhart-King, *Int. J. Mol. Sci.*, 2011, **12**, 8596–8609.
- 21 F. M. Chen and X. Liu, *Prog. Polym. Sci.*, 2016, **53**, 86–168.
- 22 E. S. Place, N. D. Evans and M. M. Stevens, *Nat. Mater.*, 2009, **8**, 457–470.
- 23 B. Levi and M. T. Longaker, *Stem Cells*, 2011, **29**, 576–582.
- 24 B. Lindroos, R. Suuronen and S. Miettinen, *Stem Cell Rev. Rep.*, 2011, **7**, 269–291.
- 25 H. Mizuno, M. Tobita and A. C. Uysal, *Stem Cells*, 2012, **30**, 804–810.
- 26 S. Pina, J. M. Oliveira and R. L. Reis, *Adv. Mater.*, 2015, **27**, 1143–1169.
- 27 D. I. Braghirolli, D. Steffens and P. Pranke, *Drug Discovery Today*, 2014, **19**, 743–753.
- 28 M. J. Webber, E. A. Appel, E. Meijer and R. Langer, *Nat. Mater.*, 2016, **15**, 13–26.
- 29 F. Pati, D.-H. Ha, J. Jang, H. H. Han, J.-W. Rhie and D.-W. Cho, *Biomaterials*, 2015, **62**, 164–175.
- 30 G. S. Boersema, N. Grotenhuis, Y. Bayon, J. F. Lange and Y. M. Bastiaansen-Jenniskens, *BioRes. Open Access*, 2016, **5**, 6–14.
- 31 N. M. Seale, Y. Zeng and S. Varghese, *Developmental Biology and Musculoskeletal Tissue Engineering*, Elsevier, 2018, pp. 207–223.
- 32 M. Nightingale and M. R. Labrosse, *J. Biomech.*, 2018, **79**, 207–211.
- 33 P. Zarrintaj, A. Urbanska, S. S. Gholizadeh, V. Goodarzi, M. R. Saeb and M. Mozafari, *J. Colloid Interface Sci.*, 2018, **516**, 57–66.
- 34 M. Gholipourmalekabadi, A. Samadikuchaksaraei, A. M. Seifalian, A. Urbanska, H. Ghanbarian, J. G. Hardy, M. Omrani, M. Mozafari, R. L. Reis and S. C. Kundu, *Biomed. Mater.*, 2018, **13**(3), 035003.
- 35 A. S. Ayoub and L. A. Lucia, *Introduction to renewable biomaterials: first principles and concepts*, John Wiley & Sons, 2017.
- 36 M. Sekula and E. K. Zuba-Surma, *Biomaterials in Regenerative Medicine*, InTech, 2018.
- 37 P. Ducheyne, *Comprehensive biomaterials*, Elsevier, 2015.
- 38 S. J. Hollister, *Nat. Mater.*, 2005, **4**, 518–524.
- 39 R. C. Dutta and A. K. Dutta, *Biotechnol. Adv.*, 2009, **27**, 334–339.
- 40 X. Gu, F. Ding and D. F. Williams, *Biomaterials*, 2014, **35**, 6143–6156.
- 41 J. J. Rice, M. M. Martino, L. De Laporte, F. Tortelli, P. S. Briquez and J. A. Hubbell, *Adv. Healthcare Mater.*, 2013, **2**, 57–71.
- 42 M. A. Meyers, J. McKittrick and P. Y. Chen, *Science*, 2013, **339**, 773–779.
- 43 S. Mitragotri and J. Lahann, *Nat. Mater.*, 2009, **8**, 15–23.
- 44 J. M. Holzwarth and P. X. Ma, *Biomaterials*, 2011, **32**, 9622–9629.
- 45 F. Edalat, I. Sheu, S. Manoucheri and A. Khademhosseini, *Curr. Opin. Biotechnol.*, 2012, **23**, 820–825.
- 46 A. Ranella, M. Barberoglou, S. Bakogianni, C. Fotakis and E. Stratakis, *Acta Biomater.*, 2010, **6**, 2711–2720.
- 47 H.-I. Chang and Y. Wang, *Regenerative medicine and tissue engineering-cells and biomaterials*, InTechOpen, 2011.
- 48 M. P. Calatayud, B. Sanz, V. Raffa, C. Riggio, M. R. Ibarra and G. F. Goya, *Biomaterials*, 2014, **35**, 6389–6399.
- 49 M. Hoefling, F. Iori, S. Corni and K.-E. Gottschalk, *Langmuir*, 2010, **26**, 8347–8351.
- 50 F. Meder, T. Daberkow, L. Treccani, M. Wilhelm, M. Schowalter, A. Rosenauer, L. Mädler and K. Rezwani, *Acta Biomater.*, 2012, **8**, 1221–1229.
- 51 K. N. Ekdahl, J. D. Lambris, H. Elwing, D. Ricklin, P. H. Nilsson, Y. Teramura, I. A. Nicholls and B. Nilsson, *Adv. Drug Delivery Rev.*, 2011, **63**, 1042–1050.
- 52 J. M. Anderson and K. M. Miller, *Biomaterials*, 1984, **5**, 5–10.
- 53 J. M. Anderson, *Cardiovasc. Pathol.*, 1993, **2**, S33–S41.
- 54 R. Marchant, A. Hiltner, C. Hamlin, A. Rabinovitch, R. Slobodkin and J. M. Anderson, *J. Biomed. Mater. Res.*, 1983, **17**, 301–325.
- 55 D. F. Williams, *Biomaterials*, 2008, **29**, 2941–2953.
- 56 M. Rahmati and M. Mozafari, *ACS Biomater. Sci. Eng.*, 2019, **6**(1), 4–20.
- 57 J. M. Anderson, *Regener. Biomater.*, 2016, **3**, 73–77.
- 58 D. F. Williams, *Biomaterials*, 2008, **29**, 2941–2953.
- 59 P. M. Kou and J. E. Babensee, *J. Biomed. Mater. Res., Part A*, 2011, **96**, 239–260.
- 60 Z. Sheikh, P. J. Brooks, O. Barzilay, N. Fine and M. Glogauer, *Materials*, 2015, **8**, 5671–5701.
- 61 J. M. Morais, F. Papadimitrakopoulos and D. J. Burgess, *AAPS J.*, 2010, **12**, 188–196.
- 62 R. A. Latour, *Encycl. Biomater. Biomed. Eng.*, 2005, **1**, 270–284.
- 63 S. L. Hirsh, D. R. McKenzie, N. J. Nosworthy, J. A. Denman, O. U. Sezerman and M. M. Bilek, *Colloids Surf., B*, 2013, **103**, 395–404.
- 64 Q. Wei, T. Becherer, S. Angioletti-Uberti, J. Dzubiella, C. Wischke, A. T. Neffe, A. Lendlein, M. Ballauff and R. Haag, *Angew. Chem., Int. Ed.*, 2014, **53**, 8004–8031.



- 65 C. D. Walkey, J. B. Olsen, H. Guo, A. Emili and W. C. Chan, *J. Am. Chem. Soc.*, 2012, **134**, 2139–2147.
- 66 Z. Xia and J. T. Triffitt, *Biomed. Mater.*, 2006, **1**, R1.
- 67 L. Tang and J. W. Eaton, *Am. J. Clin. Pathol.*, 1995, **103**, 466–471.
- 68 Y. Fong, L. L. Moldawer, G. T. Shires and S. F. Lowry, *Surg., Gynecol. Obstet.*, 1990, **170**, 363–378.
- 69 G. F. Pierce, T. A. Mustoe, B. W. Altmann, T. F. Deuel and A. Thomason, *J. Cell. Biochem.*, 1991, **45**, 319–326.
- 70 Y. Onuki, U. Bhardwaj, F. Papadimitrakopoulos and D. J. Burgess, *J. Diabetes Sci. Technol.*, 2008, **2**(6), 1003–1015.
- 71 J. M. Anderson, A. Rodriguez and D. T. Chang, *Semin. Immunol.*, 2008, **20**(2), 86–100.
- 72 J. M. Anderson, *ASAIO Trans.*, 1988, **34**, 101–107.
- 73 D. P. Vasconcelos, A. P. Aguas, M. A. Barbosa, P. Pelegrin and J. N. Barbosa, *Acta Biomater.*, 2019, **83**, 1–12.
- 74 B. N. Brown and S. F. Badylak, *Acta Biomater.*, 2013, **9**, 4948–4955.
- 75 B. D. Ratner and S. J. Bryant, *Annu. Rev. Biomed. Eng.*, 2004, **6**, 41–75.
- 76 S. A. Eming, T. Krieg and J. M. Davidson, *J. Invest. Dermatol.*, 2007, **127**, 514–525.
- 77 A. Vishwakarma, N. S. Bhise, M. B. Evangelista, J. Rouwkema, M. R. Dokmeci, A. M. Ghaemmaghami, N. E. Vrana and A. Khademhosseini, *Trends Biotechnol.*, 2016, **34**, 470–482.
- 78 S. F. Badylak, *Science*, 2016, **352**, 298.
- 79 S. A. Eming, M. Hammerschmidt, T. Krieg and A. Roers, *Semin. Cell Dev. Biol.*, 2009, **20**(5), 517–527.
- 80 N. A. Hotaling, L. Tang, D. J. Irvine and J. E. Babensee, *Annu. Rev. Biomed. Eng.*, 2015, **17**, 317–349.
- 81 E. Mariani, G. Lisignoli, R. M. Borzi and L. Pulsatelli, *Int. J. Mol. Sci.*, 2019, **20**, 636.
- 82 P. Krzyszczyk, R. Schloss, A. Palmer and F. Berthiaume, *Front. Physiol.*, 2018, **9**, 419.
- 83 P. J. Delves, S. J. Martin, D. R. Burton and I. M. Roitt, *Essential immunology*, John Wiley & Sons, 2017.
- 84 R. F. Diegelmann, *Front. Biosci.*, 2004, **9**, 283–289.
- 85 H. P. Felgueiras, N. S. Murthy, S. D. Sommerfeld, M. M. Bras, V. Migonney and J. Kohn, *ACS Appl. Mater. Interfaces*, 2016, **8**, 13207–13217.
- 86 M. W. Mosesson, *J. Thromb. Haemostasis*, 2005, **3**, 1894–1904.
- 87 B. O. O. Boni, L. Lamboni, T. Souho, M. Gauthier and G. Yang, *Mater. Horiz.*, 2019, **6**, 1122–1137.
- 88 L. W. Qian, A. B. Fourcaudot, K. Yamane, T. You, R. K. Chan and K. P. Leung, *Wound Repair Regen.*, 2016, **24**, 26–34.
- 89 M. E. Scarritt, R. Londono and S. F. Badylak, *The Immune Response to Implanted Materials and Devices*, Springer, 2017, pp. 1–14.
- 90 T. J. Koh and L. A. DiPietro, *Expert Rev. Mol. Med.*, 2011, **13**, e23.
- 91 D. Ramnath, E. E. Powell, G. M. Scholz and M. J. Sweet, *Semin. Cell Dev. Biol.*, 2017, **61**, 22–30.
- 92 M. Karin and H. Clevers, *Nature*, 2016, **529**, 307–315.
- 93 Z. Julier, A. J. Park, P. S. Briquez and M. M. Martino, *Acta Biomater.*, 2017, **53**, 13–28.
- 94 C. Esche, C. Stellato and L. A. Beck, *J. Invest. Dermatol.*, 2005, **125**, 615–628.
- 95 G. Y. Chen and G. Nunez, *Nat. Rev. Immunol.*, 2010, **10**, 826–837.
- 96 P. Scaffidi, T. Misteli and M. E. Bianchi, *Nature*, 2002, **418**, 191–195.
- 97 F. J. Quintana and I. R. Cohen, *J. Immunol.*, 2005, **175**, 2777–2782.
- 98 M. J. Bours, E. L. Swennen, F. Di Virgilio, B. N. Cronstein and P. C. Dagnelie, *Pharmacol. Ther.*, 2006, **112**, 358–404.
- 99 H. Kono and K. L. Rock, *Nat. Rev. Immunol.*, 2008, **8**, 279–289.
- 100 M. Lamkanfi and V. M. Dixit, *Cell*, 2014, **157**, 1013–1022.
- 101 B. K. Davis, H. Wen and J. P. Ting, *Annu. Rev. Immunol.*, 2011, **29**, 707–735.
- 102 J. von Moltke, J. S. Ayres, E. M. Kofoed, J. Chavarria-Smith and R. E. Vance, *Annu. Rev. Immunol.*, 2013, **31**, 73–106.
- 103 A. Denes, G. Coutts, N. Lenart, S. M. Cruickshank, P. Pelegrin, J. Skinner, N. Rothwell, S. M. Allan and D. Brough, *Proc. Natl. Acad. Sci. U. S. A.*, 2015, **112**, 4050–4055.
- 104 Y. Qu, S. Misaghi, K. Newton, A. Maltzman, A. Izrael-Tomasevic, D. Arnott and V. M. Dixit, *J. Exp. Med.*, 2016, **213**, 877–885.
- 105 W. K. Ip and R. Medzhitov, *Nat. Commun.*, 2015, **6**, 6931.
- 106 T. Strowig, J. Henao-Mejia, E. Elinav and R. Flavell, *Nature*, 2012, **481**, 278–286.
- 107 Y. He, M. Y. Zeng, D. Yang, B. Motro and G. Núñez, *Nature*, 2016, **530**, 354.
- 108 H. Shi, Y. Wang, X. Li, X. Zhan, M. Tang, M. Fina, L. Su, D. Pratt, C. H. Bu, S. Hildebrand, S. Lyon, L. Scott, J. Quan, Q. Sun, J. Russell, S. Arnett, P. Jurek, D. Chen, V. V. Kravchenko, J. C. Mathison, E. M. Moresco, N. L. Monson, R. J. Ulevitch and B. Beutler, *Nat. Immunol.*, 2016, **17**, 250–258.
- 109 S. Jha, W. J. Brickey and J. P.-Y. Ting, *Inflammasomes in myeloid cells: warriors within*, *Myeloid Cells in Health and Disease: A Synthesis*, American Society for Microbiology, Washington, DC, 2017, ch. 17, pp. 305–324.
- 110 D. Fong, P. Gregoire-Gelinas, A. P. Cheng, T. Mezheritsky, M. Lavertu, S. Sato and C. D. Hoemann, *Biomaterials*, 2017, **129**, 127–138.
- 111 J. I. Andorko and C. M. Jewell, *Bioeng. Transl. Med.*, 2017, **2**, 139–155.
- 112 P. Broz and V. M. Dixit, *Nat. Rev. Immunol.*, 2016, **16**, 407.
- 113 C. M. Artlett, *J. Pathol.*, 2013, **229**, 157–167.
- 114 H. H. Simms, R. D'Amico and K. I. Bland, *Ann. Surg.*, 1997, **225**, 757–763; discussion 763–755.
- 115 J. M. Anderson and S. Jiang, *The Immune Response to Implanted Materials and Devices*, Springer, 2017, pp. 15–36.
- 116 D. El Kebir and J. G. Filep, *Cells*, 2013, **2**, 330–348.
- 117 K. R. Martin, D. Ohayon and V. Witko-Sarsat, *Swiss Med. Wkly.*, 2015, **145**, w14056.
- 118 N. X. Landen, D. Li and M. Stahle, *Cell. Mol. Life Sci.*, 2016, **73**, 3861–3885.





- 119 W. G. Brodbeck, M. Macewan, E. Colton, H. Meyerson and J. M. Anderson, *J. Biomed. Mater. Res., Part A*, 2005, **74**, 222–229.
- 120 M. Aziz, N. E. Holodick, T. L. Rothstein and P. Wang, *Immunol. Res.*, 2015, **63**, 153–166.
- 121 F. Alegre, P. Pelegrin and A. E. Feldstein, *Semin. Liver Dis.*, 2017, **37**(02), 119–127.
- 122 T. A. Wynn and K. M. Vannella, *Immunity*, 2016, **44**, 450–462.
- 123 F.-X. Yu and K.-L. Guan, *Genes Dev.*, 2013, **27**, 355–371.
- 124 A. Lee, B. Fakler, L. K. Kaczmarek and L. L. Isom, *J. Neurosci.*, 2014, **34**, 15159–15169.
- 125 S. Asuthkar, P. A. Elustondo, L. Demirkhanyan, X. Sun, P. Baskaran, K. K. Velpula, B. Thyagarajan, E. V. Pavlov and E. Zakharian, *J. Biol. Chem.*, 2015, **290**, 2659–2669.
- 126 K. Takahashi, Y. Matsuda and K. Naruse, *AIMS Biophys.*, 2016, **3**, 63–74.
- 127 F. Di Virgilio, A. C. Sarti and F. Grassi, *Curr. Opin. Immunol.*, 2018, **52**, 51–59.
- 128 I. Hafner-Bratkovič and P. Pelegrin, *Curr. Opin. Immunol.*, 2018, **52**, 8–17.
- 129 L. Stokes, A. B. MacKenzie and R. Sluyter, *Front. Immunol.*, 2016, **7**, 48.
- 130 G. A. Ramirez, L. A. Coletto, C. Sciorati, E. P. Bozzolo, P. Manunta, P. Rovere-Querini and A. A. Manfredi, *Cells*, 2018, **7**, 70.
- 131 V. Schatz, P. Neubert, A. Schröder, K. Binger, M. Gebhard, D. N. Müller, F. C. Luft, J. Titze and J. Jantsch, *Pediatr. Nephrol.*, 2017, **32**, 201–210.
- 132 K. Salao, L. Jiang, H. Li, V. W.-W. Tsai, Y. Husaini, P. M. Curmi, L. J. Brown, D. A. Brown and S. N. Breit, *Biol. Open*, 2016, **5**, 620–630.
- 133 M. Vaeth and S. Feske, *Curr. Opin. Immunol.*, 2018, **52**, 39–50.
- 134 B. Davenport, Y. Li, J. W. Heizer, C. Schmitz and A.-L. Perraud, *Front. Immunol.*, 2015, **6**, 375.
- 135 L. L. Nohara, S. R. Stanwood, K. D. Omilusik and W. A. Jefferies, *Front. Immunol.*, 2015, **6**, 234.
- 136 S. Mortadza, S. Alawieyah, L. Wang, D. Li and L.-H. Jiang, *Front. Immunol.*, 2015, **6**, 407.
- 137 N. Liu, Y. Zhuang, Z. Zhou, J. Zhao, Q. Chen and J. Zheng, *Neurosci. Lett.*, 2017, **651**, 1–8.
- 138 Y. Mizoguchi, T. A. Kato, Y. Seki, M. Ohgidani, N. Sagata, H. Horikawa, Y. Yamauchi, M. Sato-Kasai, K. Hayakawa and R. Inoue, *J. Biol. Chem.*, 2014, **289**, 18549–18555.
- 139 M. J. Stebbing, J. M. Cottee and I. Rana, *Front. Immunol.*, 2015, **6**, 497.
- 140 R. A. North, *Physiol. Rev.*, 2002, **82**, 1013–1067.
- 141 F. Di Virgilio, G. Schmalzing and F. Markwardt, *Trends Cell Biol.*, 2018, **28**, 392–404.
- 142 R. Ferreira, R. Wong and L. C. Schlichter, *Front. Immunol.*, 2015, **6**, 153.
- 143 M. A. Retamal, M. V. Bennett, P. Pelegrin and R. Fernandez, *Mediators Inflammation*, 2016, **2016**, 6245731, DOI: 10.1155/2016/6245731.
- 144 Y. Wu, L. Chen, P. G. Scott and E. E. Tredget, *Stem Cells*, 2007, **25**, 2648–2659.
- 145 S. Mukherjee, S. Darzi, K. Paul, J. A. Werkmeister and C. E. Gargett, *Interface Focus*, 2019, **9**, 20180089.
- 146 T. S. Cheung and F. Dazzi, *Semin. Immunol.*, 2018, **35**, 59–68.
- 147 K. Le Blanc and L. C. Davies, *Immunol. Lett.*, 2015, **168**, 140–146.
- 148 E. Eggenhofer and M. J. Hoogduijn, *Transplant. Res.*, 2012, **1**, 12.
- 149 D. Kyurkchiev, I. Bochev, E. Ivanova-Todorova, M. Mourdjeva, T. Oreshkova, K. Belezmezova and S. Kyurkchiev, *World J. Stem Cells*, 2014, **6**, 552–570.
- 150 H. Lei, K. Schmidt-Bleek, A. Dienelt, P. Reinke and H. D. Volk, *Front. Pharmacol.*, 2015, **6**, 184.
- 151 L. C. Davies and P. R. Taylor, *Immunology*, 2015, **144**, 541–548.
- 152 H. Li, S. Shen, H. Fu, Z. Wang, X. Li, X. Sui, M. Yuan, S. Liu, G. Wang and Q. Guo, *Stem Cells Int.*, 2019, **2019**, 9671206.
- 153 J. D. Humphrey, E. R. Dufresne and M. A. Schwartz, *Nat. Rev. Mol. Cell Biol.*, 2014, **15**, 802–812.
- 154 C. Bonnans, J. Chou and Z. Werb, *Nat. Rev. Mol. Cell Biol.*, 2014, **15**, 786–801.
- 155 L. Li, J. Eyckmans and C. S. Chen, *Nat. Mater.*, 2017, **16**, 1164–1168.
- 156 J. Eyckmans, T. Boudou, X. Yu and C. S. Chen, *Dev. Cell*, 2011, **21**, 35–47.
- 157 F. Martino, A. R. Perestrelo, V. Vinarsky, S. Pagliari and G. Forte, *Front. Physiol.*, 2018, **9**, 824.
- 158 N. Wang, *J. Phys. D: Appl. Phys.*, 2017, **50**, 233002.
- 159 R. J. McMurray, M. J. Dalby and P. M. Tsimbouri, *J. Tissue Eng. Regener. Med.*, 2015, **9**, 528–539.
- 160 Y. Zhang, K. Liao, C. Li, A. C. K. Lai, J. J. Foo and V. Chan, *Bioengineering*, 2017, **4**, 72.
- 161 T. Iskratsch, H. Wolfenson and M. P. Sheetz, *Nat. Rev. Mol. Cell Biol.*, 2014, **15**, 825–833.
- 162 J. L. Alonso and W. H. Goldmann, *AIMS Biophys.*, 2016, **3**, 50–62.
- 163 E. A. Cavalcanti-Adam, T. Volberg, A. Micoulet, H. Kessler, B. Geiger and J. P. Spatz, *Biophys. J.*, 2007, **92**, 2964–2974.
- 164 H. B. Schiller and R. Fassler, *EMBO Rep.*, 2013, **14**, 509–519.
- 165 Z. J. Chen, W. P. Wang, Y. C. Chen, J. Y. Wang, W. H. Lin, L. A. Tai, G. G. Liou, C. S. Yang and Y. H. Chi, *J. Cell Sci.*, 2014, **127**, 1792–1804.
- 166 N. Strohmeier, M. Bharadwaj, M. Costell, R. Fässler and D. J. Müller, *Nat. Mater.*, 2017, **16**, 1262.
- 167 S. Seetharaman and S. Etienne-Manneville, *Biol. Cell.*, 2018, **110**, 49–64.
- 168 A. Elosegui-Artola, E. Bazellieres, M. D. Allen, I. Andreu, R. Oria, R. Sunyer, J. J. Gomm, J. F. Marshall, J. L. Jones, X. Trepas and P. Roca-Cusachs, *Nat. Mater.*, 2014, **13**, 631–637.
- 169 K. D. Webster, W. P. Ng and D. A. Fletcher, *Biophys. J.*, 2014, **107**, 146–155.
- 170 T. J. Chen, C. C. Wu, M. J. Tang, J. S. Huang and F. C. Su, *PLoS One*, 2010, **5**, e14392.
- 171 P. Naumanen, P. Lappalainen and P. Hotulainen, *J. Microsc.*, 2008, **231**, 446–454.
- 172 P. Weiss, *J. Exp. Zool.*, 1945, **100**, 353–386.



- 173 K. Metavarayuth, P. Sitasuwan, X. Zhao, Y. Lin and Q. Wang, *ACS Biomater. Sci. Eng.*, 2016, **2**, 142–151.
- 174 N. Gui, W. Xu, D. E. Myers, R. Shukla, H. P. Tang and M. Qian, *Biomater. Sci.*, 2018, **6**, 250–264.
- 175 J. Park, D. H. Kim and A. Levchenko, *Biophys. J.*, 2018, **114**, 1257–1263.
- 176 A. M. Ross, Z. Jiang, M. Bastmeyer and J. Lahann, *Small*, 2012, **8**, 336–355.
- 177 A. G. Harvey, E. W. Hill and A. Bayat, *Expert Rev. Med. Devices*, 2013, **10**, 257–267.
- 178 C. Rianna, M. Ventre, S. Cavalli, M. Radmacher and P. A. Netti, *ACS Appl. Mater. Interfaces*, 2015, **7**, 21503–21510.
- 179 F. F. B. Hulshof, B. Papenburg, A. Vasilevich, M. Hulsman, Y. Zhao, M. Levers, N. Fekete, M. de Boer, H. Yuan, S. Singh, N. Beijer, M. A. Bray, D. J. Logan, M. Reinders, A. E. Carpenter, C. van Blitterswijk, D. Stamatialis and J. de Boer, *Biomaterials*, 2017, **137**, 49–60.
- 180 F. Hulshof, C. Schophuizen, M. Mihajlovic, C. van Blitterswijk, R. Masereeuw, J. de Boer and D. Stamatialis, *J. Tissue Eng. Regen. Med.*, 2018, **12**, e817–e827.
- 181 D. S. Hernandez, E. T. Ritschdorff, J. L. Connell and J. B. Shear, *J. Am. Chem. Soc.*, 2018, **140**, 14064–14068.
- 182 A. T. Nguyen, S. R. Sathe and E. K. Yim, *J. Phys.: Condens. Matter*, 2016, **28**, 183001.
- 183 C. L. Gilchrist, D. S. Ruch, D. Little and F. Guilak, *Biomaterials*, 2014, **35**, 10015–10024.
- 184 B. K. Teo, S. T. Wong, C. K. Lim, T. Y. Kung, C. H. Yap, Y. Ramagopal, L. H. Romer and E. K. Yim, *ACS Nano*, 2013, **7**, 4785–4798.
- 185 S. Wang, J. Li, Z. Zhou, S. Zhou and Z. Hu, *Molecules*, 2018, **24**, 75.
- 186 J. Padmanabhan, M. J. Augelli, B. Cheung, E. R. Kinser, B. Cleary, P. Kumar, R. Wang, A. J. Sawyer, R. Li, U. D. Schwarz, J. Schroers and T. R. Kyriakides, *Sci. Rep.*, 2016, **6**, 33277.
- 187 T. Khampiang, V. Yamassatien, P. Ekabutr, P. Pavasant and P. Supaphol, *Adv. Polym. Technol.*, 2018, **37**, 2030–2042.
- 188 I. Firkowska-Boden, X. Zhang and K. D. Jandt, *Adv. Healthcare Mater.*, 2018, **7**, 1700995.
- 189 X. Wang, R. Berger, J. I. Ramos, T. Wang, K. Koynov, G. Liu, H.-J. Butt and S. Wu, *RSC Adv.*, 2014, **4**, 45059–45064.
- 190 X. J. Liu, Y. Xie, S. J. Shi, Q. L. Feng, A. Bachhuka, X. D. Guo, Z. D. She, R. W. Tan, Q. Cai and K. Vasilev, *Appl. Surf. Sci.*, 2019, **473**, 838–847.
- 191 M. S. Lord, M. Foss and F. Besenbacher, *Nano Today*, 2010, **5**, 66–78.
- 192 C. Rianna, L. Rossano, R. H. Kollarigowda, F. Formiggini, S. Cavalli, M. Ventre and P. A. Netti, *Adv. Funct. Mater.*, 2016, **26**, 7572–7580.
- 193 G. M. de Peppo, H. Agheli, C. Karlsson, K. Ekstrom, H. Brisby, M. Lenneras, S. Gustafsson, P. Sjoval, A. Johansson, E. Olsson, J. Lausmaa, P. Thomsen and S. Petronis, *Int. J. Nanomed.*, 2014, **9**, 2499–2515.
- 194 H. M. Khanlou, B. C. Ang, M. M. Barzani, M. Silakhori and S. Talebian, *Neural Comput. Appl.*, 2015, **26**, 1751–1761.
- 195 H. Chen, X. Huang, M. Zhang, F. Damanik, M. B. Baker, A. Leferink, H. Yuan, R. Truckenmuller, C. van Blitterswijk and L. Moroni, *Acta Biomater.*, 2017, **59**, 82–93.
- 196 M. F. Sola-Ruiz, C. Perez-Martinez, J. J. Martin-del-Llano, C. Carda-Batalla and C. Labaig-Rueda, *Med. Oral Patol. Oral Cir. Bucal.*, 2015, **20**, e88–e93.
- 197 S. Barr, E. W. Hill and A. Bayat, *J. Mech. Behav. Biomed. Mater.*, 2017, **75**, 75–81.
- 198 K. Anselme and M. Bigerelle, *Scanning*, 2014, **36**, 11–20.
- 199 V. Belaud, T. Petithory, A. Ponche, C. Mauclair, C. Donnet, L. Pieuchot, S. Benayoun and K. Anselme, *Biointerphases*, 2018, **13**, 06D408.
- 200 O. Raimbault, S. Benayoun, K. Anselme, C. Mauclair, T. Bourgade, A. M. Kietzig, P. L. Girard-Lauriault, S. Valette and C. Donnet, *Mater. Sci. Eng., C*, 2016, **69**, 311–320.
- 201 A. B. Faia-Torres, S. Guimond-Lischer, M. Rottmar, M. Charnley, T. Goren, K. Maniura-Weber, N. D. Spencer, R. L. Reis, M. Textor and N. M. Neves, *Biomaterials*, 2014, **35**, 9023–9032.
- 202 P. Mazón, D. García-Bernal, L. Meseguer-Olmo, F. Cragnochini and N. Piedad, *Ceram. Int.*, 2015, **41**, 6631–6644.
- 203 R. Olivares-Navarrete, S. L. Hyzy, M. E. Berg, J. M. Schneider, K. Hotchkiss, Z. Schwartz and B. D. Boyan, *Ann. Biomed. Eng.*, 2014, **42**, 2551–2561.
- 204 R. Xing, S. P. Lyngstadaas, J. E. Ellingsen, S. Taxt-Lamolle and H. J. Haugen, *Clin. Oral. Implants Res.*, 2015, **26**, 649–656.
- 205 S. F. Lamolle, M. Monjo, S. P. Lyngstadaas, J. E. Ellingsen and H. J. Haugen, *J. Biomed. Mater. Res., Part A*, 2009, **88**, 581–588.
- 206 F. S. L. Bobbert and A. A. Zadpoor, *J. Mater. Chem. B*, 2017, **5**, 6175–6192.
- 207 F. G. Torres, O. P. Troncoso, O. Gamucci, S. Corvaglia, V. Brunetti and G. Bardi, *Int. J. Biol. Macromol.*, 2015, **75**, 460–466.
- 208 C. Jin, G. Lee, C. Oh, H. J. Kim and H. M. Kim, *J. Periodontal Implant Sci.*, 2017, **47**, 116–131.
- 209 E. M. Lee, K. Smith, K. Gall, B. D. Boyan and Z. Schwartz, *Biomaterials*, 2016, **110**, 34–44.
- 210 O. Andrukhov, R. Huber, B. Shi, S. Berner, X. Rausch-Fan, A. Moritz, N. D. Spencer and A. Schedle, *Dent. Mater.*, 2016, **32**, 1374–1384.
- 211 K. Yusa, O. Yamamoto, H. Takano, M. Fukuda and M. Iino, *Sci. Rep.*, 2016, **6**, 29462.
- 212 Y. Zhang, S. E. Chen, J. Shao and J. van den Beucken, *ACS Appl. Mater. Interfaces*, 2018, **10**, 36652–36663.
- 213 D. O. Costa, P. D. Prowse, T. Chrones, S. M. Sims, D. W. Hamilton, A. S. Rizkalla and S. J. Dixon, *Biomaterials*, 2013, **34**, 7215–7226.
- 214 L. Saldana, L. Crespo, F. Bensiamar, M. Arruebo and N. Vilaboa, *J. Biomed. Mater. Res., Part A*, 2014, **102**, 128–140.
- 215 S. Giljean, M. Bigerelle and K. Anselme, *Scanning*, 2014, **36**, 2–10.
- 216 B. Zhou, X. Gao, C. Wang, Z. Ye, Y. Gao, J. Xie, X. Wu and W. Wen, *ACS Appl. Mater. Interfaces*, 2015, **7**, 17181–17187.



- 217 B. S. Moon, S. Kim, H. E. Kim and T. S. Jang, *Mater. Sci. Eng., C*, 2017, **73**, 90–98.
- 218 W. Wan, K. K. Bjorkman, E. S. Choi, A. L. Panepento, K. S. Anseth and L. A. Leinwand, *bioRxiv*, 2019, 682930.
- 219 S. Mei, L. Yang, Y. Pan, D. Wang, X. Wang, T. Tang and J. Wei, *Colloids Surf., B*, 2019, **174**, 207–215.
- 220 N. Fukuda, M. Kanazawa, K. Tsuru, A. Tsuchiya, J. Sunarso, R. Toita, Y. Mori, Y. Nakashima and K. Ishikawa, *Sci. Rep.*, 2018, **8**, 16887.
- 221 H. Begam, B. Kundu, A. Chanda and S. K. Nandi, *Ceram. Int.*, 2017, **43**, 3752–3760.
- 222 S. F. Lamolle, M. Monjo, M. Rubert, H. J. Haugen, S. P. Lyngstadaas and J. E. Ellingsen, *Biomaterials*, 2009, **30**, 736–742.
- 223 H. Jeon, H. Lee and G. Kim, *Tissue Eng., Part C*, 2014, **20**, 951–963.
- 224 C. Ribeiro, V. Sencadas, A. C. Areias, F. M. Gama and S. Lanceros-Mendez, *J. Biomed. Mater. Res., Part A*, 2015, **103**, 2260–2268.
- 225 P. Slepicka, I. Michaljanicova, S. Rimpelova and V. Svorcik, *Mater. Sci. Eng., C*, 2017, **76**, 818–826.
- 226 S. Prauzner-Bechcicki, J. Raczowska, J. Rysz, J. Wiltowska-Zuber, J. Pabijan, M. Marzec, A. Budkowski and M. Lekka, *Appl. Surf. Sci.*, 2018, **457**, 881–890.
- 227 M. Strickstroock, H. Rothe, S. Grohmann, G. Hildebrand, I. M. Zylla and K. Liefelth, *BioNanoMaterials*, 2017, **18**(1–2), 20160013.
- 228 Y. Deng, X. Liu, A. Xu, L. Wang, Z. Luo, Y. Zheng, F. Deng, J. Wei, Z. Tang and S. Wei, *Int. J. Nanomed.*, 2015, **10**, 1425–1447.
- 229 G. C. Machado, E. Garcia-Tunon, R. V. Bell, M. Alini, E. Saiz and M. Peroglio, *J. Eur. Ceram. Soc.*, 2018, **38**, 949–961.
- 230 C. Wu, M. Chen, T. Zheng and X. Yang, *Biomed. Mater. Eng.*, 2015, **26**(Suppl 1), S155–S164.
- 231 X. Li, Q. Huang, T. A. Elkhooley, Y. Liu, H. Wu, Q. Feng, L. Liu, Y. Fang, W. Zhu and T. Hu, *Biomed. Mater.*, 2018, **13**, 045013.
- 232 S. Migita and K. Araki, *AIMS Bioeng.*, 2017, **4**, 162–170.
- 233 K. H. Vining and D. J. Mooney, *Nat. Rev. Mol. Cell Biol.*, 2017, **18**, 728–742.
- 234 R. Olivares-Navarrete, E. M. Lee, K. Smith, S. L. Hyzy, M. Doroudi, J. K. Williams, K. Gall, B. D. Boyan and Z. Schwartz, *PLoS One*, 2017, **12**, e0170312.
- 235 P. Moshayedi, G. Ng, J. C. Kwok, G. S. Yeo, C. E. Bryant, J. W. Fawcett, K. Franze and J. Guck, *Biomaterials*, 2014, **35**, 3919–3925.
- 236 S. van Helvert, C. Storm and P. Friedl, *Nat. Cell Biol.*, 2018, **20**, 8–20.
- 237 A. D. Doyle, N. Carvajal, A. Jin, K. Matsumoto and K. M. Yamada, *Nat. Commun.*, 2015, **6**, 8720.
- 238 C. D. Hartman, B. C. Isenberg, S. G. Chua and J. Y. Wong, *Exp. Cell Res.*, 2017, **359**, 361–366.
- 239 J. Li, D. Han and Y. P. Zhao, *Sci. Rep.*, 2014, **4**, 3910.
- 240 M. Nii, J. H. Lai, M. Keeney, L. H. Han, A. Behn, G. Imanbayev and F. Yang, *Acta Biomater.*, 2013, **9**, 5475–5483.
- 241 L. M. Wang, H. Chang, H. Zhang, K. F. Ren, H. Li, M. Hu, B. C. Li, M. C. L. Martins, M. A. Barbosa and J. Ji, *J. Mater. Chem. B*, 2015, **3**, 7546–7553.
- 242 L. G. Vincent, Y. S. Choi, B. Alonso-Latorre, J. C. del Alamo and A. J. Engler, *Biotechnol. J.*, 2013, **8**, 472–484.
- 243 P. Tseng and D. Di Carlo, *Adv. Mater.*, 2014, **26**, 1242–1247.
- 244 E. B. Evans, S. W. Brady, A. Tripathi and D. Hoffman-Kim, *Biomater. Res.*, 2018, **22**, 14.
- 245 W. J. Hadden, J. L. Young, A. W. Holle, M. L. McFetridge, D. Y. Kim, P. Wijesinghe, H. Taylor-Weiner, J. H. Wen, A. R. Lee, K. Bieback, B. N. Vo, D. D. Sampson, B. F. Kennedy, J. P. Spatz, A. J. Engler and Y. S. Choi, *Proc. Natl. Acad. Sci. U. S. A.*, 2017, **114**, 5647–5652.
- 246 Y. Wu, Z. Yang, J. B. Law, A. Y. He, A. A. Abbas, V. Denslin, T. Kamarul, J. H. Hui and E. H. Lee, *Tissue Eng., Part A*, 2017, **23**, 43–54.
- 247 J. H. Seo, K. Sakai and N. Yui, *Acta Biomater.*, 2013, **9**, 5493–5501.
- 248 M. Bao, J. Xie, N. Katoele, X. Hu, B. Wang, A. Piruska and W. T. Huck, *ACS Appl. Mater. Interfaces*, 2018, **11**, 1754–1759.
- 249 Y. Zhang, Y. Xing, J. Li, Z. Zhang, H. Luan, Z. Chu, H. Gong and Y. Fan, *BioMed Res. Int.*, 2018, **2018**, 4025083.
- 250 J. You, S. A. Park, D. S. Shin, D. Patel, V. K. Raghunathan, M. Kim, C. J. Murphy, G. Tae and A. Revzin, *Tissue Eng., Part A*, 2013, **19**, 2655–2663.
- 251 Z. Li, Y. Gong, S. Sun, Y. Du, D. Lu, X. Liu and M. Long, *Biomaterials*, 2013, **34**, 7616–7625.
- 252 A. S. Mao, J. W. Shin and D. J. Mooney, *Biomaterials*, 2016, **98**, 184–191.
- 253 J. Xu, M. Sun, Y. Tan, H. Wang, H. Wang, P. Li, Z. Xu, Y. Xia, L. Li and Y. Li, *Differentiation*, 2017, **96**, 30–39.
- 254 D. A. Foyt, D. K. Taheem, S. A. Ferreira, M. D. A. Norman, J. Petzold, G. Jell, A. E. Grigoriadis and E. Gentleman, *Acta Biomater.*, 2019, **89**, 73–83.
- 255 S. H. Medina, B. Bush, M. Cam, E. Sevcik, F. W. DelRio, K. Nandy and J. P. Schneider, *Biomaterials*, 2019, **202**, 1–11.
- 256 I. Hopp, A. Michelmores, L. E. Smith, D. E. Robinson, A. Bachhuka, A. Mierczynska and K. Vasilev, *Biomaterials*, 2013, **34**, 5070–5077.
- 257 S. Sur, C. J. Newcomb, M. J. Webber and S. I. Stupp, *Biomaterials*, 2013, **34**, 4749–4757.
- 258 M. Guvendiren, M. Perepelyuk, R. G. Wells and J. A. Burdick, *J. Mech. Behav. Biomed. Mater.*, 2014, **38**, 198–208.
- 259 L. S. Wang, C. Du, W. S. Toh, A. C. Wan, S. J. Gao and M. Kurisawa, *Biomaterials*, 2014, **35**, 2207–2217.
- 260 K. Ye, X. Wang, L. Cao, S. Li, Z. Li, L. Yu and J. Ding, *Nano Lett.*, 2015, **15**, 4720–4729.
- 261 J. M. Banks, L. C. Mozdzen, B. A. Harley and R. C. Bailey, *Biomaterials*, 2014, **35**, 8951–8959.
- 262 J. C. Courtenay, C. Deneke, E. M. Lanzoni, C. A. Costa, Y. Bae, J. L. Scott and R. I. Sharma, *Cellulose*, 2018, **25**, 925–940.
- 263 X. Lu, Z. Ding, F. Xu, Q. Lu and D. L. Kaplan, *ACS Appl. Bio Mater.*, 2019, **2**(7), 3108–3119.
- 264 K. E. Coombs, A. T. Leonard, M. N. Rush, D. A. Santistevan and E. L. Hedberg-Dirk, *J. Biomed. Mater. Res., Part A*, 2017, **105**, 51–61.



- 265 M. Lin, H. Wang, C. Ruan, J. Xing, J. Wang, Y. Li, Y. Wang and Y. Luo, *Biomacromolecules*, 2015, **16**, 973–984.
- 266 C. Pinese, S. Jebors, P. E. Stoebner, V. Humblot, P. Verdie, L. Causse, X. Garric, H. Taillades, J. Martinez, A. Mehdi and G. Subra, *Mater. Today Chem.*, 2017, **4**, 73–83.
- 267 R. M. d. Nascimento, S. M. Ramos, I. H. Bechtold and A. C. Hernandez, *ACS Biomater. Sci. Eng.*, 2018, **4**, 2784–2793.
- 268 L. N. Chen, C. Yan and Z. J. Zheng, *Mater. Today*, 2018, **21**, 38–59.
- 269 S. Sakata, Y. Inoue and K. Ishihara, *Langmuir*, 2014, **30**, 2745–2751.
- 270 S. Kumar, S. Raj, E. Kolanthai, A. K. Sood, S. Sampath and K. Chatterjee, *ACS Appl. Mater. Interfaces*, 2015, **7**, 3237–3252.
- 271 S. Zhao, J. Zhang, M. Zhu, Y. Zhang, Z. Liu, Y. Ma, Y. Zhu and C. Zhang, *J. Mater. Chem. B*, 2015, **3**, 1612–1623.
- 272 J. J. Li, N. Kawazoe and G. Chen, *Biomaterials*, 2015, **54**, 226–236.
- 273 H. Amani, H. Arzaghi, M. Bayandori, A. S. Dezfuli, H. Pazoki-Toroudi, A. Shafiee and L. Moradi, *Adv. Mater. Interfaces*, 2019, **6**, 1900572.
- 274 H. C. Bygd, K. D. Forsmark and K. M. Bratlie, *Biomaterials*, 2015, **56**, 187–197.
- 275 S. J. Jenkins, D. Ruckerl, P. C. Cook, L. H. Jones, F. D. Finkelman, N. van Rooijen, A. S. MacDonald and J. E. Allen, *Science*, 2011, **332**, 1284–1288.
- 276 Y. Shen, M. Gao, Y. Ma, H. Yu, F. Z. Cui, H. Gregersen, Q. Yu, G. Wang and X. Liu, *Colloids Surf., B*, 2015, **126**, 188–197.
- 277 A. Rashad, K. Mustafa, E. B. Heggset and K. Syverud, *Biomacromolecules*, 2017, **18**, 1238–1248.
- 278 Y. Zhang, H. Pan, P. Zhang, N. Gao, Y. Lin, Z. Luo, P. Li, C. Wang, L. Liu, D. Pang, L. Cai and Y. Ma, *Nanoscale*, 2013, **5**, 5919–5929.
- 279 L. Wang, M. He, T. Gong, X. Zhang, L. Zhang, T. Liu, W. Ye, C. Pan and C. Zhao, *Biomater. Sci.*, 2017, **5**, 2416–2426.
- 280 X. Liu, S. Shi, Q. Feng, A. Bachhuka, W. He, Q. Huang, R. Zhang, X. Yang and K. Vasilev, *ACS Appl. Mater. Interfaces*, 2015, **7**, 18473–18482.
- 281 S. Shahabi, L. Treccani, R. Dringen and K. Rezwan, *ACS Appl. Mater. Interfaces*, 2015, **7**, 13821–13833.
- 282 A. Hasan, S. K. Pattanayek and L. M. Pandey, *ACS Biomater. Sci. Eng.*, 2018, **4**, 3224–3233.
- 283 F. Meder, C. Brandes, L. Treccani and K. Rezwan, *Acta Biomater.*, 2013, **9**, 5780–5787.
- 284 S. N. Christo, A. Bachhuka, K. R. Diener, A. Mierczynska, J. D. Hayball and K. Vasilev, *Adv. Healthcare Mater.*, 2016, **5**, 956–965.
- 285 Z. Chen, A. Bachhuka, S. Han, F. Wei, S. Lu, R. M. Visalakshan, K. Vasilev and Y. Xiao, *ACS Nano*, 2017, **11**, 4494–4506.
- 286 Y. Yang, P. Qi, Y. Ding, M. F. Maitz, Z. Yang, Q. Tu, K. Xiong, Y. Leng and N. Huang, *J. Mater. Chem. B*, 2015, **3**, 72–81.
- 287 D. Zeng, X. Zhang, X. Wang, Q. Huang, J. Wen, X. Miao, L. Peng, Y. Li and X. Jiang, *Artif. Cells, Nanomed., Biotechnol.*, 2018, **46**, 1425–1435.
- 288 Q. Li, B. Zhang, N. Kasoju, J. Ma, A. Yang, Z. Cui, H. Wang and H. Ye, *Int. J. Mol. Sci.*, 2018, **19**, 2344.
- 289 Y. Zhang, J. P. Luan, S. X. Jiang, X. Zhou and M. M. Li, *Composites, Part B*, 2019, **172**, 397–405.
- 290 G. Aziz, N. De Geyter, H. Declercq, R. Cornelissen and R. Morent, *Surf. Coat. Technol.*, 2015, **271**, 39–47.
- 291 C. Insomphun, J.-A. Chuah, S. Kobayashi, T. Fujiki and K. Numata, *ACS Biomater. Sci. Eng.*, 2016, **3**, 3064–3075.
- 292 J. P. Yapor, A. Lutzke, A. Pegalajar-Jurado, B. H. Neufeld, V. B. Damodaran and M. M. Reynolds, *J. Mater. Chem. B*, 2015, **3**, 9233–9241.
- 293 Y. Y. Zheng, L. H. Liu, Y. Ma, L. Xiao and Y. Liu, *Ind. Eng. Chem. Res.*, 2018, **57**, 10403–10410.
- 294 M. G. L. Olthof, M. A. Tryfonidou, X. Liu, B. Pouran, B. P. Meij, W. J. A. Dhert, M. J. Yaszemski, L. Lu, J. Alblas and D. H. R. Kempen, *Tissue Eng., Part A*, 2018, **24**, 819–829.
- 295 Y. Bai, H. Xing, P. Wu, X. Feng, K. Hwang, J. M. Lee, X. Y. Phang, Y. Lu and S. C. Zimmerman, *ACS Nano*, 2015, **9**, 10227–10236.
- 296 Z. Shen, J. Wu, Y. Yu, S. Liu, W. Jiang, H. Nurmammat and B. Wu, *Sci. Rep.*, 2019, **9**, 7557.
- 297 A. K. Fuchs, T. Syrovets, K. A. Haas, C. Loos, A. Musyanovych, V. Mailander, K. Landfester and T. Simmet, *Biomaterials*, 2016, **85**, 78–87.
- 298 J. Zhang, L. Li, Y. Peng, Y. Chen, X. Lv, S. Li, X. Qin, H. Yang, C. Wu and Y. Liu, *Biochim. Biophys. Acta, Mol. Cell Res.*, 2018, **1865**, 172–185.
- 299 C. Morke, H. Rebl, B. Finke, M. Dubs, P. Nestler, A. Airoudj, V. Roucoules, M. Schnabelrauch, A. Kortge, K. Anselme, C. A. Helm and J. B. Nebe, *ACS Appl. Mater. Interfaces*, 2017, **9**, 10461–10471.
- 300 T. Chen, L. Li, G. Xu, X. Wang, J. Wang, Y. Chen, W. Jiang, Z. Yang and G. Lin, *Front. Pharmacol.*, 2018, **9**, 763.
- 301 Y. Y. Zheng, L. H. Liu, C. D. Xiong and L. F. Zhang, *Mater. Lett.*, 2018, **213**, 84–87.
- 302 Y. Zheng, C. Xiong, S. Zhang, X. Li and L. Zhang, *Mater. Sci. Eng., C*, 2015, **55**, 512–523.
- 303 B. Cao, Y. Peng, X. Liu and J. Ding, *ACS Appl. Mater. Interfaces*, 2017, **9**, 23574–23585.
- 304 T. T. Yu, F. Z. Cui, Q. Y. Meng, J. Wang, D. C. Wu, J. Zhang, X. X. Kou, R. L. Yang, Y. Liu, Y. S. Zhang, F. Yang and Y. H. Zhou, *ACS Biomater. Sci. Eng.*, 2017, **3**, 1119–1128.
- 305 R. Ion, S. Vizireanu, C. E. Stancu, C. Luculescu, A. Cimpean and G. Dinescu, *Mater. Sci. Eng., C*, 2015, **48**, 118–125.
- 306 X. Lin, K. Fukazawa and K. Ishihara, *ACS Appl. Mater. Interfaces*, 2015, **7**, 17489–17498.
- 307 M. Vetten and M. Gulumian, *Toxicol. Appl. Pharmacol.*, 2019, **363**, 131–141.
- 308 T. Lu, Y. Qiao and X. Liu, *Interface Focus*, 2012, **2**, 325–336.
- 309 E. M. McCarthy, H. Floyd, O. Addison, Z. J. Zhang, P. G. Oppenheimer and L. M. Grover, *ACS Omega*, 2018, **3**, 10129–10138.
- 310 G. Socrates, *Infrared and Raman characteristic group frequencies: tables and charts*, John Wiley & Sons, 2004.
- 311 S. Staehlke, H. Rebl, B. Finke, P. Mueller, M. Gruening and J. B. Nebe, *Cell Biosci.*, 2018, **8**, 22.



- 312 Y. Zhao, K. W. K. Yeung and P. K. Chu, *Appl. Surf. Sci.*, 2014, **310**, 11–18.
- 313 W. H. Jin and P. K. Chu, *Surf. Coat. Technol.*, 2018, **336**, 2–8.
- 314 C. H. Yang, Y. C. Li, W. F. Tsai, C. F. Ai and H. H. Huang, *Clin. Oral. Implants Res.*, 2015, **26**, 166–175.
- 315 W. H. Jin, G. S. Wu, H. Q. Feng, W. H. Wang, X. M. Zhang and P. K. Chu, *Corros. Sci.*, 2015, **94**, 142–155.
- 316 H. Qin, H. Cao, Y. Zhao, G. Jin, M. Cheng, J. Wang, Y. Jiang, Z. An, X. Zhang and X. Liu, *ACS Appl. Mater. Interfaces*, 2015, **7**, 10785–10794.
- 317 R. I. Iziyomov, A. Y. Beliaev, I. V. Kondyurina, I. N. Shardakov, A. V. Kondyurin, M. M. Bilek and D. R. McKenzie, *Mater. Sci. Eng.*, 2016, **123**(1), 012003.
- 318 C. R. Mariappan and N. Ranga, *Mater. Sci. Forum*, 2016, **860**, 25–28.
- 319 G. G. Fuentes, *Micromanufacturing Engineering and Technology*, 2010, pp. 459–486.
- 320 J. Jagielski, A. Piatkowska, P. Aubert, L. Thome, A. Turos and A. A. Kader, *Surf. Coat. Technol.*, 2006, **200**, 6355–6361.
- 321 C. Park, Y. J. Seong, I. G. Kang, E. H. Song, H. Lee, J. Kim, H. D. Jung, H. E. Kim and T. S. Jang, *ACS Appl. Mater. Interfaces*, 2019, **11**, 10492–10504.
- 322 B. Li, H. Cao, Y. Zhao, M. Cheng, H. Qin, T. Cheng, Y. Hu, X. Zhang and X. Liu, *Sci. Rep.*, 2017, **7**, 42707.
- 323 Y. Yu, G. Jin, Y. Xue, D. Wang, X. Liu and J. Sun, *Acta Biomater.*, 2017, **49**, 590–603.
- 324 T. Lu, S. Qian, F. Meng, C. Ning and X. Liu, *Colloids Surf., B*, 2016, **142**, 192–198.
- 325 K. Gan, H. Liu, L. Jiang, X. Liu, X. Song, D. Niu, T. Chen and C. Liu, *Dent. Mater.*, 2016, **32**, e263–e274.
- 326 E. A. Wakelin, G. C. Yeo, D. R. McKenzie, M. M. M. Bilek and A. S. Weiss, *APL Bioeng.*, 2018, **2**, 026109.
- 327 U. Walschus, A. Hoene, M. Patrzyk, S. Lucke, B. Finke, M. Polak, G. Lukowski, R. Bader, C. Zietz, A. Podbielski, J. B. Nebe and M. Schlosser, *J. Funct. Biomater.*, 2017, **8**, 30.
- 328 D. K. Shiau, C. H. Yang, Y. S. Sun, M. F. Wu, H. B. Pan and H. H. Huang, *Surf. Coat. Technol.*, 2019, **365**, 173–178.
- 329 Y. S. Sun, L. Zhang, H. Q. Zhu, W. E. Yang, M. Y. Lan, S. W. Lee and H. H. Huang, *J. Vac. Sci. Technol., A*, 2016, **34**, 041402.
- 330 H. H. Huang, D. K. Shiau, C. S. Chen, J. H. Chang, S. Wang, H. B. Pan and M. F. Wu, *Surf. Coat. Technol.*, 2019, **365**, 179–188.
- 331 S. Won, Y. H. Huh, L. R. Cho, H. S. Lee, E. S. Byon and C. J. Park, *Tissue Eng. Regener. Med.*, 2017, **14**, 123–131.
- 332 C. Xia, D. S. Cai, J. Tan, K. Q. Li, Y. Q. Qiao and X. Y. Liu, *ACS Biomater. Sci. Eng.*, 2018, **4**, 3185–3193.
- 333 S. Qiao, H. Cao, X. Zhao, H. Lo, L. Zhuang, Y. Gu, J. Shi, X. Liu and H. Lai, *Int. J. Nanomed.*, 2015, **10**, 653–664.
- 334 S. Viswanathan, L. Mohan, M. Chakraborty, C. Mandal, P. Bera, S. T. Aruna and C. Anandan, *Mater. Manuf. Processes*, 2018, **33**, 1121–1127.
- 335 X. Cheng, J. Fei, A. Kondyurin, K. Fu, L. Ye, M. M. M. Bilek and S. Bao, *Mater. Sci. Eng., C*, 2019, **99**, 863–874.
- 336 V. Chudinov, I. Kondyurina, V. Terpugov and A. Kondyurin, *Nucl. Instrum. Methods Phys. Res., Sect. B*, 2019, **440**, 163–174.
- 337 E. Kosobrodova, W. J. Gan, A. Kondyurin, P. Thorn and M. M. M. Bilek, *ACS Appl. Mater. Interfaces*, 2018, **10**, 227–237.
- 338 T. S. Jang, J. H. Lee, S. Kim, C. Park, J. Song, H. J. Jae, H. E. Kim, J. W. Chung and H. D. Jung, *Biomaterials*, 2019, **223**, 119461.
- 339 L. M. de Andrade, C. Paternoster, V. Montaña-Machado, G. Barucca, M. Sikora-Jasinska, R. Tolouei, S. Turgeon and D. Mantovani, *MRS Commun.*, 2018, **8**, 1404–1412.
- 340 K. Schrock, J. Lutz, S. Mandl, M. C. Hacker, M. Kamprad and M. Schulz-Siegmund, *J. Orthop. Res.*, 2015, **33**, 325–333.
- 341 M. A. Shafique, G. Murtaza, S. Saadat, M. K. H. Uddin and R. Ahmad, *Radiat. Eff. Defects Solids*, 2017, **172**, 590–599.
- 342 C. Park, S. W. Lee, J. Kim, E. H. Song, H. D. Jung, J. U. Park, H. E. Kim, S. Kim and T. S. Jang, *Biomater. Sci.*, 2019, **7**, 2907–2919.
- 343 L. Tong, W. Zhou, Y. Zhao, X. Yu, H. Wang and P. K. Chu, *Colloids Surf., B*, 2016, **148**, 139–146.
- 344 M. A. Wronska, I. B. O'Connor, M. A. Tilbury, A. Srivastava and J. G. Wall, *Adv. Mater.*, 2016, **28**, 5485–5508.
- 345 M. W. Tibbitt, C. B. Rodell, J. A. Burdick and K. S. Anseth, *Proc. Natl. Acad. Sci. U. S. A.*, 2015, **112**, 14444–14451.
- 346 A. V. Bryksin, A. C. Brown, M. M. Baksh, M. G. Finn and T. H. Barker, *Acta Biomater.*, 2014, **10**, 1761–1769.
- 347 A. Srivastava, L. B. O'Connor, A. Pandit and J. G. Wall, *Prog. Polym. Sci.*, 2014, **39**, 308–329.
- 348 T. Garg, G. Rath and A. K. Goyal, *J. Drug Targeting*, 2015, **23**, 202–221.
- 349 Q. Wei, T. Becherer, S. Angioletti-Uberti, J. Dzubiella, C. Wischke, A. T. Neffe, A. Lendlein, M. Ballauff and R. Haag, *Angew. Chem., Int. Ed.*, 2014, **53**, 8004–8031.
- 350 M. Aizawa, T. Matsuura and Z. Zhuang, *Biol. Pharm. Bull.*, 2013, **36**, 1654–1661.
- 351 T. Sarkar, Y. Gao and A. Mulchandani, *Appl. Biochem. Biotechnol.*, 2013, **170**, 1011–1025.
- 352 C. A. Custodio, C. M. Alves, R. L. Reis and J. F. Mano, *J. Tissue Eng. Regener. Med.*, 2010, **4**, 316–323.
- 353 S. L. Hirsh, D. R. McKenzie, N. J. Nosworthy, J. A. Denman, O. U. Sezerman and M. M. Bilek, *Colloids Surf., B*, 2013, **103**, 395–404.
- 354 P. Zucca and E. Sanjust, *Molecules*, 2014, **19**, 14139–14194.
- 355 B. R. Coad, M. Jasieniak, S. S. Griesser and H. J. Griesser, *Surf. Coat. Technol.*, 2013, **233**, 169–177.
- 356 B. R. Coad, K. Vasilev, K. R. Diener, J. D. Hayball, R. D. Short and H. J. Griesser, *Langmuir*, 2012, **28**, 2710–2717.
- 357 E. A. Wakelin, A. Fathi, M. Kracica, G. C. Yeo, S. G. Wise, A. S. Weiss, D. G. McCulloch, F. Dehghani, D. R. McKenzie and M. M. Bilek, *ACS Appl. Mater. Interfaces*, 2015, **7**, 23029–23040.
- 358 H. Main, J. Radenkovic, E. Kosobrodova, D. McKenzie, M. Bilek and U. Lendahl, *Cell. Mol. Life Sci.*, 2014, **71**, 3841–3857.
- 359 R. P. Tan, A. H. Chan, S. Wei, M. Santos, B. S. Lee, E. C. Filipe, B. Akhavan, M. M. Bilek, M. K. Ng and Y. Xiao, *JACC: Basic to Translational Science*, 2019, **4**, 56–71.



- 360 H. Lee, N. F. Scherer and P. B. Messersmith, *Proc. Natl. Acad. Sci. U. S. A.*, 2006, **103**, 12999–13003.
- 361 C. D. Spicer and B. G. Davis, *Nat. Commun.*, 2014, **5**, 4740.
- 362 M. P. Deonarain, G. Yahiolu, I. Stamati and J. Marklew, *Expert Opin. Drug Discovery*, 2015, **10**, 463–481.
- 363 A. Srivastava, C. Cunningham, A. Pandit and J. G. Wall, *Macromol. Biosci.*, 2015, **15**, 682–690.
- 364 S. Cao, M. N. Barcellona, F. Pfeiffer and M. T. Bernards, *J. Appl. Polym. Sci.*, 2016, **133**(40), 43985.
- 365 X. X. Yang, X. Y. Wang, F. Yu, L. L. Ma, X. H. Pan, G. J. Luo, S. Lin, X. M. Mo, C. L. He and H. S. Wang, *RSC Adv.*, 2016, **6**, 99720–99728.
- 366 K. A. Totaro, X. Liao, K. Bhattacharya, J. I. Finneman, J. B. Sperry, M. A. Massa, J. Thorn, S. V. Ho and B. L. Pentelute, *Bioconjugate Chem.*, 2016, **27**, 994–1004.
- 367 E. J. Park, T. N. Gevrek, R. Sanyal and A. Sanyal, *Bioconjugate Chem.*, 2014, **25**, 2004–2011.
- 368 C. A. Custodio and J. F. Mano, *ChemNanoMat*, 2016, **2**, 376–384.
- 369 P. Nacharaju, F. N. Boctor, B. N. Manjula and S. A. Acharya, *Transfusion*, 2005, **45**, 374–383.
- 370 M. E. Smith, F. F. Schumacher, C. P. Ryan, L. M. Tedaldi, D. Papaioannou, G. Waksman, S. Caddick and J. R. Baker, *J. Am. Chem. Soc.*, 2010, **132**, 1960–1965.
- 371 Z. Y. Zhang, N. Vanparijs, S. Vandewalle, F. E. Du Prez, L. Nuhn and B. G. De Geest, *Polym. Chem.*, 2016, **7**, 7242–7248.
- 372 N. Vanparijs, R. De Coen, D. Laplace, B. Louage, S. Maji, L. Lybaert, R. Hoogenboom and B. G. De Geest, *Chem. Commun.*, 2015, **51**, 13972–13975.
- 373 H. Kakwere, C. K. Chun, K. A. Jolliffe, R. J. Payne and S. Perrier, *Chem. Commun.*, 2010, **46**, 2188–2190.
- 374 Y. Zou, L. Zhang, L. Yang, F. Zhu, M. Ding, F. Lin, Z. Wang and Y. Li, *J. Controlled Release*, 2018, **273**, 160–179.
- 375 J. Gopinathan and I. Noh, *Tissue Eng. Regener. Med.*, 2018, **15**, 531–546.
- 376 K. Nagahama, Y. Kimura and A. Takemoto, *Nat. Commun.*, 2018, **9**, 2195.
- 377 B. K. Lee, J. H. Noh, J. H. Park, S. H. Park, J. H. Kim, S. H. Oh and M. S. Kim, *Tissue Eng. Regener. Med.*, 2018, **15**, 393–402.
- 378 G. Yi, J. Son, J. Yoo, C. Park and H. Koo, *Biomater. Res.*, 2018, **22**, 13.
- 379 S. L. Liu, M. J. Dong, Z. H. Zhang and G. D. Fu, *Polym. Adv. Technol.*, 2017, **28**, 1065–1070.
- 380 S. Fu, H. Dong, X. Deng, R. Zhuo and Z. Zhong, *Carbohydr. Polym.*, 2017, **169**, 332–340.
- 381 S. Li, L. Wang, X. Yu, C. Wang and Z. Wang, *Mater. Sci. Eng., C*, 2018, **82**, 299–309.
- 382 Y. L. Li, Y. Tan, K. Xu, C. G. Lu and P. X. Wang, *Polym. Degrad. Stab.*, 2017, **137**, 75–82.
- 383 J. Collins, Z. Y. Xiao, M. Mullner and L. A. Connal, *Polym. Chem.*, 2016, **7**, 3812–3826.
- 384 L. J. Macdougall, K. L. Wiley, A. M. Kloxin and A. P. Dove, *Biomaterials*, 2018, **178**, 435–447.
- 385 J. Gopinathan and I. Noh, *Biomater. Res.*, 2018, **22**, 11.
- 386 C. S. Jung, B. K. Kim, J. Lee, B. H. Min and S. H. Park, *Tissue Eng. Regener. Med.*, 2018, **15**, 155–162.
- 387 W. Tang and M. L. Becker, *Chem. Soc. Rev.*, 2014, **43**, 7013–7039.
- 388 M. Lotfi, M. Nejib and M. Naceur, *Adv. Biomater. Sci. Biomed. Appl.*, 2013, **8**, 208–240.
- 389 M. M. Gentleman and E. Gentleman, *Int. Mater. Rev.*, 2014, **59**, 417–429.
- 390 T. Razafiarison, C. N. Hostenstein, T. Stauber, M. Jovic, E. Vertudes, M. Loparic, M. Kawecki, L. Bernard, U. Silvan and J. G. Snedeker, *Proc. Natl. Acad. Sci. U. S. A.*, 2018, **115**, 4631–4636.
- 391 M. M. Gentleman and J. A. Ruud, *Langmuir*, 2010, **26**, 1408–1411.
- 392 R. J. Good, *J. Adhes. Sci. Technol.*, 1992, **6**, 1269–1302.
- 393 C. Della Volpe and S. Siboni, *J. Colloid Interface Sci.*, 1997, **195**, 121–136.
- 394 C. Della Volpe and S. Siboni, *J. Adhes. Sci. Technol.*, 2000, **14**, 235–272.
- 395 C. J. van Oss, W. Wu and R. F. Giese, *Lifshitz-van der Waals, Lewis acid–base and electrostatic interactions in adhesion in aqueous media*, 1998.
- 396 C. J. Vanoss, R. J. Good and M. K. Chaudhury, *Langmuir*, 1988, **4**, 884–891.
- 397 A. B. D. Cassie and S. Baxter, *Trans. Faraday Soc.*, 1944, **40**, 0546–0550.
- 398 R. N. Wenzel, *Ind. Eng. Chem.*, 1936, **28**, 988–994.
- 399 T. Young, *Philos. Trans. R. Soc. London*, 1805, **95**, 65–87.
- 400 B. Wang, P. Liu, Z. Liu, H. Pan, X. Xu and R. Tang, *Biotechnol. Bioeng.*, 2014, **111**, 386–395.
- 401 M. Nakamura, N. Hori, H. Ando, S. Namba, T. Toyama, N. Nishimiya and K. Yamashita, *Mater. Sci. Eng., C*, 2016, **62**, 283–292.
- 402 O. Savvova, G. Shadrina, O. Babich and O. Fesenko, *Chem. Chem. Technol.*, 2015, **9**(3), 349–354.
- 403 R. E. Baier, A. E. Meyer, J. R. Natiella, R. R. Natiella and J. M. Carter, *J. Biomed. Mater. Res.*, 1984, **18**, 337–355.
- 404 T. Groth and G. Altankov, *Biomaterials*, 1996, **17**, 1227–1234.
- 405 J. M. Schakenraad, H. J. Busscher, C. R. Wildevuur and J. Arends, *Cell Biophys.*, 1988, **13**, 75–91.
- 406 P. Parhi, A. Golas and E. A. Vogler, *J. Adhes. Sci. Technol.*, 2010, **24**, 853–888.
- 407 X. Liu, J. Y. Lim, H. J. Donahue, R. Dhurjati, A. M. Mastro and E. A. Vogler, *Biomaterials*, 2007, **28**, 4535–4550.
- 408 J. Y. Lim, X. Liu, E. A. Vogler and H. J. Donahue, *J. Biomed. Mater. Res., Part A*, 2004, **68**, 504–512.
- 409 C. Xie, *Biosurf. Biotribol.*, 2019, **5**, 83–92.
- 410 R. Sa, Y. Yan, Z. Wei, L. Zhang, W. Wang and M. Tian, *ACS Appl. Mater. Interfaces*, 2014, **6**, 21730–21738.
- 411 J. H. Liang, R. Song, Q. L. Huang, Y. Yang, L. X. Lin, Y. M. Zhang, P. L. Jiang, H. P. Duan, X. Dong and C. J. Lin, *Electrochim. Acta*, 2015, **174**, 1149–1159.
- 412 M. Nasrollahzadeh, S. M. Sajadi and M. Iqbal, *Nanomaterials and Plant Potential*, Springer, 2019, pp. 31–70.
- 413 L. Lunelli, C. Potrich, L. Pasquardini and C. Pederzoli, *Nanostructured Functionalized Surfaces*, *Encyclopedia of Nanotechnology*, Springer, Dordrecht, 2012, pp. 1760–1766.



- 414 R. Rasouli, A. Barhoum and H. Uludag, *Biomater. Sci.*, 2018, **6**, 1312–1338.
- 415 G. S. Wang, Y. Wan, B. Ren, T. Wang and Z. Q. Liu, *Adv. Eng. Mater.*, 2017, **19**, 1700299.
- 416 C. Frey, A. Sales, K. Athanasopulu, J. Spatz and R. Kemkemer, *Hydrogels with precisely nano-functionalized micro-topography for cell guidance*, Conference: 48th DGBMT Annual Conference, 2015, vol. 59. pp. S6–S9.
- 417 S. Christo, A. Bachhuka, K. R. Diener, K. Vasilev and J. D. Hayball, *Sci. Rep.*, 2016, **6**, 26207.
- 418 M. B. Hovgaard, K. Rechendorff, J. Chevallier, M. Foss and F. Besenbacher, *J. Phys. Chem. B*, 2008, **112**, 8241–8249.
- 419 J. Huang, S. V. Grater, F. Corbellini, S. Rinck, E. Bock, R. Kemkemer, H. Kessler, J. Ding and J. P. Spatz, *Nano Lett.*, 2009, **9**, 1111–1116.
- 420 V. Marzaioli, C. J. Groß, I. Weichenmeier, C. B. Schmidt-Weber, J. Gutermuth, O. Groß and F. Alessandrini, *Nanomaterials*, 2017, **7**, 355.
- 421 F. Karimi, T. G. McKenzie, A. J. O'Connor, G. G. Qiao and D. E. Heath, *J. Mater. Chem. B*, 2017, **5**, 5942–5953.
- 422 V. M. Aguilar and B. D. Cosgrove, *Skeletal Muscle Development*, Springer, 2017, pp. 75–92.
- 423 L. Filipponi, P. Livingston, O. Kaspar, V. Tokarova and D. V. Nicolau, *Biomed. Microdevices*, 2016, **18**, 9.
- 424 S. Arango-Santander, A. Pelaez-Vargas, S. C. Freitas and C. Garcia, *J. Nanotechnol.*, 2018, **2018**, 8624735, DOI: 10.1155/2018/8624735.
- 425 D. Hamelinck, H. Zhou, L. Li, C. Verweij, D. Dillon, Z. Feng, J. Costa and B. B. Haab, *Mol. Cell. Proteomics*, 2005, **4**, 773–784.
- 426 M. Woodson and J. Liu, *Phys. Chem. Chem. Phys.*, 2007, **9**, 207–225.
- 427 A. M. M. Brito, E. Belleti, L. R. Menezes, A. J. C. Lanfredi and I. L. Nantes-Cardoso, *An. Acad. Bras. Cienc.*, 2019, **91**(4), e20181236.
- 428 G. Oteri, M. Pisanom and M. Cicciu, *J. Appl. Spectrosc.*, 2016, **83**, 316–321.
- 429 S. Sen-Britain, D. M. Britain, W. L. Hicks, Jr. and J. A. Gardella, Jr., *Biointerphases*, 2019, **14**, 051003.
- 430 K. Gajos, A. Budkowski, P. Petrou, V. Pagkali, K. Awsuik, J. Rysz, A. Bernasik, K. Misiakos, I. Raptis and S. Kakabakos, *Appl. Surf. Sci.*, 2018, **444**, 187–196.
- 431 L. Yahia and L. Mireles, *Characterization of Polymeric Biomaterials*, Elsevier, 2017, pp. 83–97.
- 432 J. L. Calvo-Guirado, A. B. Montilla, P. N. De Aza, M. Fernandez-Dominguez, S. A. Gehrke, P. Cegarra-Del Pino, L. Mahesh, A. A. Pelegrine, J. M. Aragonese and J. E. M. S. de Val, *Materials*, 2019, **12**, 380.
- 433 D. G. Castner, *Surf. Interface Anal.*, 2018, **50**, 981–990.
- 434 P. Kingshott, G. Andersson, S. L. McArthur and H. J. Griesser, *Curr. Opin. Chem. Biol.*, 2011, **15**, 667–676.
- 435 R. N. Foster, E. T. Harrison and D. G. Castner, *Langmuir*, 2016, **32**, 3207–3216.
- 436 S. Cometa, M. A. Bonifacio, A. M. Ferreira, P. Gentile and E. De Giglio, *Materials*, 2020, **13**, 705.
- 437 M. Stevanović, M. Đošić, A. Janković, V. Kojic, M. Vukasinovic-Sekulic, J. Stojanović, J. Odović, M. Crevar Sakač, K. Y. Rhee and V. Mišković-Stanković, *ACS Biomater. Sci. Eng.*, 2018, **4**, 3994–4007.
- 438 N. Metoki, D. Mandler and N. Eliaz, *Cryst. Growth Des.*, 2016, **16**, 2756–2764.
- 439 E. T. Harrison, T. Weidner, D. G. Castner and G. Interlandi, *Biointerphases*, 2017, **12**, 02D401.
- 440 S. Muramoto, D. J. Graham, M. S. Wagner, T. G. Lee, D. W. Moon and D. G. Castner, *J. Phys. Chem. C*, 2011, **115**, 24247–24255.
- 441 M. S. Wagner, T. A. Horbett and D. G. Castner, *Biomaterials*, 2003, **24**, 1897–1908.
- 442 N. G. Welch, R. M. T. Madiona, J. A. Scoble, B. W. Muir and P. J. Pigram, *Langmuir*, 2016, **32**, 10824–10834.
- 443 J. Yu, Y. Zhou, M. Engelhard, Y. Zhang, J. Son, S. Liu, Z. Zhu and X. Y. Yu, *Sci. Rep.*, 2020, **10**, 3695.
- 444 T. Weidner and D. G. Castner, *Phys. Chem. Chem. Phys.*, 2013, **15**, 12516–12524.
- 445 M. Deighan and J. Pfaendtner, *Langmuir*, 2013, **29**, 7999–8009.
- 446 I. Donderwinkel, J. C. M. van Hest and N. R. Cameron, *Polym. Chem.*, 2017, **8**, 4451–4471.
- 447 S. Derakhshanfar, R. Mbeleck, K. Xu, X. Zhang, W. Zhong and M. Xing, *Bioact. Mater.*, 2018, **3**, 144–156.
- 448 B. Zhang, L. Gao, L. Ma, Y. C. Luo, H. Y. Yang and Z. F. Cui, *Engineering*, 2019, **5**, 777–794.
- 449 H. Lee and D. W. Cho, *Lab Chip*, 2016, **16**, 2618–2625.
- 450 S. Freeman, R. Ramos, P. Alexis Chando, L. Zhou, K. Reeser, S. Jin, P. Soman and K. Ye, *Acta Biomater.*, 2019, **95**, 152–164.
- 451 W. Lim, B. Kim and Y. L. Moon, *Ann. Joint*, 2019, **4**, 7.
- 452 E. Ning, G. Turnbull, J. Clarke, F. Picard, P. Riches, M. Vendrell, D. Graham, A. W. Wark, K. Faulds and W. Shu, *Biofabrication*, 2019, **11**, 045018.
- 453 Y. Luo, A. Elumalai, A. Humayun and D. K. Mills, *3D Bioprinting in Medicine*, Springer, 2019, pp. 163–189.
- 454 S. Vijayavenkataraman, N. Vialli, J. Fuh and W. F. Lu, *Int. J. Bioprint.*, 2019, **5**, 229.
- 455 C. Yu, X. Ma, W. Zhu, P. Wang, K. L. Miller, J. Stupin, A. Koroleva-Maharajh, A. Hairabedian and S. Chen, *Biomaterials*, 2019, **194**, 1–13.
- 456 W. L. Ng, J. M. Lee, M. Zhou and W. Y. Yeong, *Rapid Prototyping of Biomaterials*, Elsevier, 2020, pp. 183–204.
- 457 J. Yin, M. Yan, Y. Wang, J. Fu and H. Suo, *ACS Appl. Mater. Interfaces*, 2018, **10**, 6849–6857.
- 458 S. Massa, M. A. Sakr, J. Seo, P. Bandaru, A. Arneri, S. Bersini, E. Zare-Eelanjegh, E. Jalilian, B. H. Cha, S. Antona, A. Enrico, Y. Gao, S. Hassan, J. P. Acevedo, M. R. Dokmeci, Y. S. Zhang, A. Khademhosseini and S. R. Shin, *Biomicrofluidics*, 2017, **11**, 044109.
- 459 M. Hospodiuk, M. Dey, D. Sosnoski and I. T. Ozbolat, *Biotechnol. Adv.*, 2017, **35**, 217–239.
- 460 M. Rahmati, J. J. Blaker, S. P. Lyngstadaas, J. F. Mano and H. J. Haugen, *Mater. Today Adv.*, 2020, **5**, 100051.
- 461 Y. J. Seol, H. W. Kang, S. J. Lee, A. Atala and J. J. Yoo, *Eur. J. Cardiothorac. Surg.*, 2014, **46**, 342–348.
- 462 H. Gudapati, M. Dey and I. Ozbolat, *Biomaterials*, 2016, **102**, 20–42.



- 463 S. H. Park, C. S. Jung and B. H. Min, *Tissue Eng. Regener. Med.*, 2016, **13**, 622–635.
- 464 A.-V. Do, B. Khorsand, S. M. Geary and A. K. Salem, *Adv. Healthcare Mater.*, 2015, **4**(12), 1742–1762.
- 465 A. J. Engler, S. Sen, H. L. Sweeney and D. E. Discher, *Cell*, 2006, **126**, 677–689.
- 466 Y. C. Li, Y. S. Zhang, A. Akpek, S. R. Shin and A. Khademhosseini, *Biofabrication*, 2016, **9**, 012001.
- 467 X. Kuang, D. J. Roach, J. T. Wu, C. M. Hamel, Z. Ding, T. J. Wang, M. L. Dunn and H. J. Qi, *Adv. Funct. Mater.*, 2019, **29**, 1805290.
- 468 W. Li, J. Zhou and Y. Xu, *Biomed. Rep.*, 2015, **3**, 617–620.
- 469 R. Rai, T. Keshavarz, J. A. Roether, A. R. Boccaccini and I. Roy, *Mater. Sci. Eng., R*, 2011, **72**, 29–47.
- 470 S. Sinn, T. Scheuermann, S. Deichelbohrer, G. Ziemer and H. P. Wendel, *J. Mater. Sci.: Mater. Med.*, 2011, **22**, 1521–1528.
- 471 B. Swetha, S. Mathew, B. S. Murthy, N. Shruthi and S. H. Bhandi, *Int. Dent. Med. J. Adv. Res.*, 2015, **1**, 1–6.
- 472 A. K. Jain, D. Singh, K. Dubey, R. Maurya, S. Mittal and A. K. Pandey, *In Vitro Toxicology*, Elsevier, 2018, pp. 45–65.
- 473 C. Frumento, *Medical Writing*, 2017, **26**, 39–40.
- 474 A. Migliore, *Expert Rev. Med. Devices*, 2017, **14**(12), 921–923.
- 475 S. K. Gupta, *J. Young Pharm.*, 2016, **8**, 6–11.
- 476 M. N. Helmus, D. F. Gibbons and D. Cebon, *Toxicol. Pathol.*, 2008, **36**, 70–80.
- 477 M. Grennan and R. J. Town, *Penn Wharton Public Policy Initiative. Book 7*, 2016.
- 478 K. S. Jones, *Semin. Immunol.*, 2008, **20**, 130–136.
- 479 A. Przekora, *Mater. Sci. Eng., C*, 2019, **97**, 1036–1051.
- 480 A. Przekora, J. Czechowska, D. Pijocha, A. Ślósarczyk and G. Ginalska, *Open Life Sci.*, 2014, **9**, 277–289.
- 481 K. Klimek, A. Belcarz, R. Pazik, P. Sobierajska, T. Han, R. J. Wiglusz and G. Ginalska, *Mater. Sci. Eng., C*, 2016, **65**, 70–79.
- 482 W. De Jong, J. Carraway and R. Geertsma, *Biocompatibility and performance of medical devices*, Elsevier, 2012, pp. 120–158.
- 483 L. M. Wancket, *Vet. Pathol.*, 2015, **52**, 842–850.
- 484 P. H. Long, *Toxicol. Pathol.*, 2008, **36**, 85–91.
- 485 X. Struillou, H. Boutigny, A. Soueidan and P. Layrolle, *Open Dent. J.*, 2010, **4**, 37.
- 486 D. W. Grainger, *Nat. Biotechnol.*, 2013, **31**, 507.
- 487 H. B. van der Worp, D. W. Howells, E. S. Sena, M. J. Porritt, S. Rewell, V. O'Collins and M. R. Macleod, *PLoS Med.*, 2010, **7**, e1000245.
- 488 I. W. Mak, N. Evaniew and M. Ghert, *Am. J. Transl. Res.*, 2014, **6**, 114–118.
- 489 C. H. Evans, *Tissue Eng., Part B*, 2011, **17**, 437–441.
- 490 M. I. Martic-Kehl, R. Schibli and P. A. Schubiger, *Eur. J. Nucl. Med. Mol. Imaging*, 2012, **39**, 1492–1496.
- 491 V. Marx, *Nature*, 2015, **522**, 373–377.
- 492 M. A. Lancaster and J. A. Knoblich, *Science*, 2014, **345**, 1247125.
- 493 S. Kim, A. N. Cho, S. Min, S. Kim and S. W. Cho, *Adv. Ther.*, 2019, **2**, 1800087.
- 494 A. L. Bredenoord, H. Clevers and J. A. Knoblich, *Science*, 2017, **355**(6322), eaaf9414.
- 495 K. M. Conlee and A. N. Rowan, *Hastings Cent. Rep.*, 2012, (Suppl), S31–S34.
- 496 S. Schulze, S. G. Henkel, D. Driesch, R. Guthke and J. Linde, *Front. Microbiol.*, 2015, **6**, 65.
- 497 A. Denchai, D. Tartarini and E. Mele, *Front. Bioeng. Biotechnol.*, 2018, **6**, 155.
- 498 S. S. Shera, S. Sahu and R. M. Banik, *Advances in Polymer Sciences and Technology*, Springer, 2018, pp. 65–74.

