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Self-assembly of two robust 3D supramolecular organic frameworks from a geometrically non-planar molecule for high gas selectivity performance†

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The synthesis of highly porous frameworks has received continuous research interest, but achieving the ability to target stable and selective materials remains challenging. Herein, by utilizing a 'direction-oriented' strategy and modulating reaction conditions, two novel 3D porous supramolecular organic framework (SOF) materials (JLU-SOF2 and JLU-SOF3, as isomers) are assembled from a non-planar building block (TMBTI = 2,4,6-trimethyl benzene-1,3,5-triyl-isophthalic acid) and they display permanent porosity, high thermal stability, and good recyclability. It is worth mentioning that the CO₂ uptake values of JLU-SOF2 and JLU-SOF3 rank among the highest values for SOF-based materials under ambient conditions. Furthermore, these two materials exhibit preferential adsorption of CO₂ over N₂ and CH₄, and can effectively separate the mixtures of light hydrocarbons. These studies indicate the possible application of JLU-SOF2 and JLU-SOF3 in trapping greenhouse gases and upgrading natural gas. In addition, this synthetic strategy introduces an effective method for developing remarkable 3D SOFs among other framework materials.

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Introduction

Alleviating the escalation of concentration of atmospheric CO₂ is an ongoing research topic.^{1–3} To solve this issue, adsorption-based separation technology will likely be a feasible method until more effective technologies can be developed, which can realize the post-combustion carbon capture and purification of natural gas.^{3–5} For gas adsorbents, permanent porosity, sufficient void-space, high selectivity and good recyclability are crucial prerequisites for their use in practical applications. Over the past few decades, some novel porous framework materials, such as metal-organic frameworks (MOFs),^{6,7} covalent organic frameworks (COFs),^{8,9} and other crystalline porous materials,^{10–12} have been explored as potential adsorbents.

Supramolecular organic frameworks (SOFs, also called HOFs, abbreviation of hydrogen-bonded organic frameworks) constructed by hydrogen-bonding and π - π interactions are

attractive gas adsorbents thanks mainly to their low framework density and regeneration by simple recrystallization.^{10,13–19} However, the stability of SOFs is still a challenge although a number of SOFs with permanent porosity have been reported.^{20–35} Considering their highly predictable hydrogen-bonding motif, some (carboxylic acid)-containing organic units were selected to construct SOF materials by the groups of Zentner, Moorthy, Hisaki and others.^{24,36–41} The vast majority of these SOFs are 2D materials due to the limited propagation ability of planar blocks. Thus, synthesis of new building blocks with a non-planar configuration, that is, utilizing a 'direction-oriented' strategy, may be an effective method to construct 3D SOF materials. Li *et al.* reported the first solution-phase 3D periodic SOF in water, and they pointed out that by utilizing self-assembly of tetrahedral building blocks and cucurbit[8]uril ordered 3D frameworks could be achieved.⁴² Pillarene-based 3D SOF materials were constructed by Yang and co-workers, and their construction is mainly attributed to the spatial configuration and abundant electron donor-acceptor components of macrocyclic molecules.⁴³ For organic small-molecules, modifying the central component of the planar organic entity to make it rotate or replacing it with a flexible group³⁸ is a feasible way to break the planarity of building blocks. On the basis of the above considerations, a non-planar organic building block, 2,4,6-trimethyl benzene-1,3,5-triyl-isophthalic acid (TMBTI), is synthesized by modifying the central phenyl ring of

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† Electronic supplementary information (ESI) available: Synthesis procedures of TMBTI, structure descriptions, materials characterization, and crystal data for JLU-SOF2 and JLU-SOF3. CCDC 1842179 and 1861545. For ESI and crystallographic data in CIF or other electronic format see DOI: 10.1039/c9sc00290a



3,3',3'',5,5',5''-benzene-1,3,5-triyl-hexabenzic acid (H_6BHB)⁴⁴ with methyl groups (Scheme 1). In terms of the 'direction-oriented' strategy, desired 3D frameworks would be assembled from TMBTI because it inherently exhibits a high propensity to form multiple hydrogen bonds in different directions (*i.e.*, beyond two dimensions).

Due to the tremendous freedom of hydrogen bonds, the self-assembly of the same constituent could also allow for achievement of different SOFs based on diverse hydrogen-bonding modes.^{24,45–49} To the best of our knowledge, though some SOF materials have been reported to have hydrogen-bonded isomers, most of these polymorphs were formed simultaneously under the same conditions.^{24,47} Therefore, design strategies, including precise modulation of reaction conditions, are vitally important to yield unique or desired structures as phase-pure materials.

Herein, as an experimental proof-of-concept, two 3D SOF isomers, **JLU-SOF2** and **JLU-SOF3**, with high thermal stability and permanent microporous nature are assembled by subtly modulating reaction conditions. It is notable that some MOFs have been synthesized using the TMBTI ligand, but all of them are isomorphous.^{50–52} **JLU-SOF2** and **JLU-SOF3** exhibit excellent CO_2 capture and separation ability, which surpasses that of the majority of SOF materials. Besides, these two materials show salient performance for purification of natural gas under ambient conditions.

Results and discussion

Structure descriptions

Single-crystal X-ray diffraction (SCXRD) analysis reveals that **JLU-SOF2** and **JLU-SOF3** crystallize in the trigonal $P\bar{3}1c$ and hexagonal $P6_2c$ space group, respectively. As illustrated in Fig. 1a and b, each TMBTI block existing in both of them is connected with six other organic units through six pairs of intermolecular $-COOH\cdots HOOC-$ hydrogen bonds. As expected, these two SOFs exhibit 3D hydrogen-bonded structures, which are mainly attributed to the propagation ability of the geometrically non-planar block in different directions. **JLU-SOF3** is assembled by the classic carboxylic acid dimer motif exclusively, and the $O-H\cdots O$ distances are 2.605(2), 2.615(7), 2.630(6), and 2.628(3) Å (Table S3†). Meanwhile, two types of hydrogen-bonding motifs are present in **JLU-SOF2**: one is the typical carboxylic acid dimer formed between two

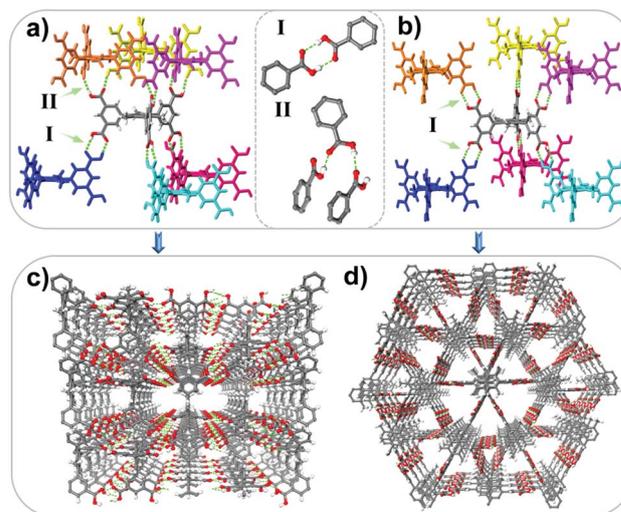
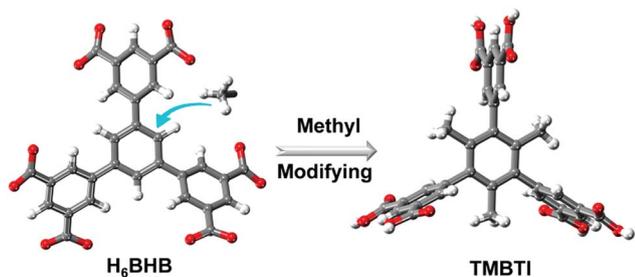


Fig. 1 Hydrogen-bonding interactions of **JLU-SOF2** and **JLU-SOF3** and their different intermolecular $-COOH\cdots HOOC-$ hydrogen-bonding motifs (a and b); ball-and-stick models of **JLU-SOF2** and **JLU-SOF3** along the x -axis and z -axis, respectively (c and d).

carboxyl groups and the distance of $O(4)-H(4)\cdots O(3)\#4$ is 2.642(2) Å; but the other is an uncommon motif formed among three carboxyl groups and the distance of $O(2)-H(2)\cdots O(1)\#3$ is 2.6606(19) Å (Table S4†). The different hydrogen-bonding interactions and corresponding packing modes of TMBTI in **JLU-SOF2** and **JLU-SOF3** were caused by the differences in reaction systems in terms of solvent polarity, solubility and temperature. Nevertheless, these two SOFs display similar pore sizes. **JLU-SOF2** contains homogeneous open channels with window sizes of about $5.6\text{ \AA} \times 3.8\text{ \AA}$ along the x -axis and y -axis, respectively, considering the van der Waals radius (Fig. 1c and S3†). For **JLU-SOF3**, two types of triangular cavities (5.8 Å and 3.2 Å in diameter) co-exist in the framework along the z -axis (Fig. 1d and S5†). Interestingly, the whole structure of **JLU-SOF3** contains three equivalent interwoven frameworks which are linked by $C-H\cdots\pi$ interactions, as supported by the measured distance of 3.39 Å between the H atoms and the centroids of phenyl rings of the isophthalic acid groups (Fig. 3d, S6 and S7†). There exist typical honeycomb channels in a single net with an internal dimension of about 19.3 Å along the z -axis, considering the van der Waals radius (Fig. 3b and S8†). However, it is expected that the stability of **JLU-SOF3** will be significantly increased after three-fold interpenetration.

In the crystal structure of **JLU-SOF2**, the central phenyl ring could be described as a 3-connected node, and the isophthalic acid group could be simplified as a 4-connected node (by three hydrogen bonds) (Fig. 2 and S4†). Consequently, the structure of **JLU-SOF2** can be rationalized as a new (3,4)-connected network with a point symbol of $\{6^3\}\{6^5 \times 10\}^3$, which possesses four types of tiles $[6^5]$, $[10^3]$, $[6^2 \times 10^2]$ and $[6^3 \times 10^3]$. Similarly, the single net of **JLU-SOF3** can be reduced into a (3,3)-connected etc net topology with an $\{8^3\}$ point symbol, and the total framework consists of three-fold interpenetrated networks (Fig. 3 and S9†). The potential solvent-accessible volume of **JLU-SOF2** and **JLU-SOF3** is 43.7% and 51.3%, respectively, estimated using PLATON after removal of disordered 1,4-dioxane (DOA) solvent.



Scheme 1 Synthesis of the non-planar TMBTI block by modifying H_6BHB with methyl groups.



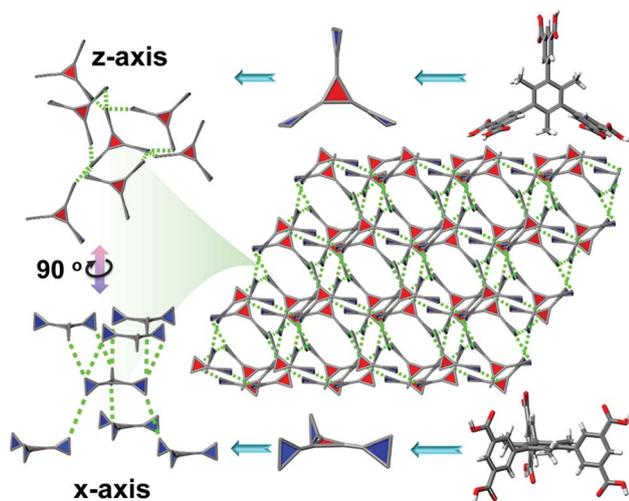


Fig. 2 Schematic representation of the (3,4)-connected net of JLU-SOF2 showing the simplified organic building block and hydrogen bonds along the x-axis and z-axis.

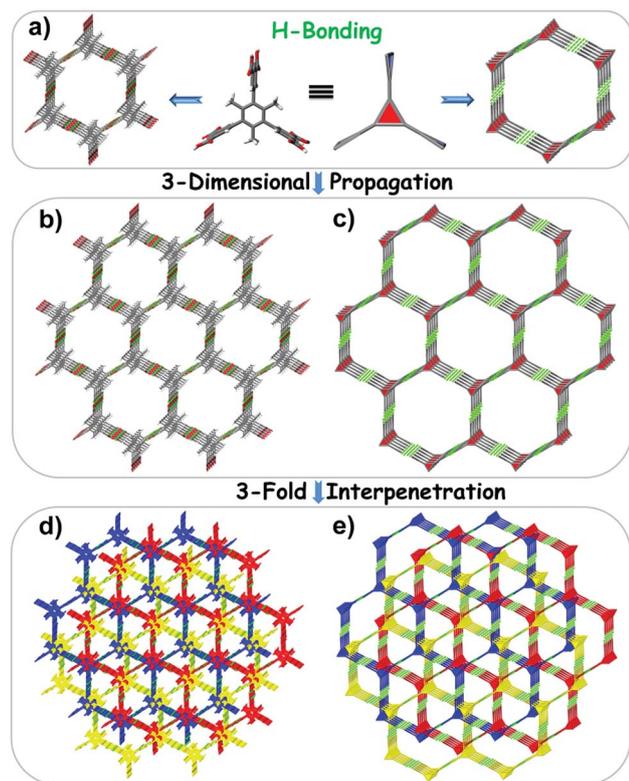


Fig. 3 Stick and schematic representation of JLU-SOF3 showing the simplified building block and hydrogen bonds for a single channel (a), single net (b and c) and three-fold interpenetrated nets (d and e) along the z-axis.

Stability studies

Thermogravimetric analyses (TGA) reveal that all solvent is removed at about 230 °C with a weight loss of 23% for JLU-SOF2, followed by a relatively steady plateau until 420 °C, after that the

material begins to decompose (Fig. S10†). JLU-SOF3 exhibits a weight loss of 24% before 210 °C, which corresponds to the loss of guest molecules, and then the material begins to decompose at 430 °C. The crystalline phase purity of as-synthesized JLU-SOF2 and JLU-SOF3 was identified from the consistency between the measured powder X-ray diffraction (PXRD) pattern and the simulated one from the SCXRD data (Fig. 4 and S12a†). To investigate their permanent porosity, N₂ sorption isotherms of these two compounds were studied. The isotherms exhibit a sharp rise in uptake at the low P/P_0 region corresponding to the natural behavior of microporous materials (Fig. S13†). Brunauer–Emmett–Teller surface areas of JLU-SOF2 and JLU-SOF3 are calculated to be 937 and 1141 m² g⁻¹, respectively, which are higher than those of most SOF-based materials^{16,53–55} (Tables S5 and S6†). The experimental pore volumes of JLU-SOF2 and JLU-SOF3 (0.47 and 0.63 cm³ g⁻¹) are consistent with the theoretical values (0.47 and 0.63 cm³ g⁻¹, the detailed calculation method is provided in the ESI†) estimated from the crystal structures, demonstrating that the cavities of the frameworks have been activated adequately.

In view of practical applications, the reusability and stability of these materials were explored. The N₂ sorption was tested for the used materials that were placed in air for one week after adsorption (named JLU-SOF2R and JLU-SOF3R). The BET surface areas are 884 and 1005 m² g⁻¹ for JLU-SOF2R and JLU-SOF3R, which are slightly lower than those of the pristine samples (Fig. S13 and Table S7†). It indicates that the reused materials remain intact basically. Variable-temperature PXRD and corresponding N₂ sorption were performed to test their thermal stability. As shown in Fig. 4 and S11,† the primary peaks are consistent with those of as-synthesized JLU-SOF2, and some changes can be observed, especially for the 350 °C treated sample. The N₂ uptake of JLU-SOF2 did not decrease significantly below 250 °C (Fig. S14 and Table S8†). Increasing the temperature above 300 °C lowered the BET surface area and a noticeable up-tail emerged at high P/P_0 . Yet, the N₂ isotherms

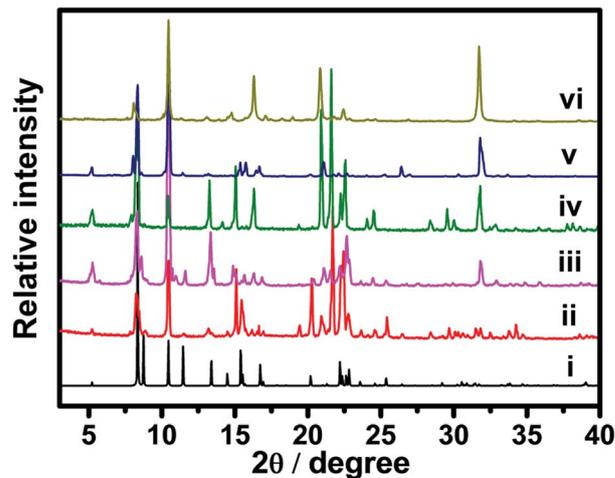


Fig. 4 PXRD patterns of simulated (i), as-synthesized (ii), exchanged in CH₃CN for one week (iii), regenerated after recrystallization of dissolved SOFs (iv), after CO₂ adsorption (v), and heated under 350 °C for 2 hours (vi) for JLU-SOF2.



preserved their type-I shape. These observations may be caused by the cleaving of hydrogen bonds and stacking of the samples.³⁵ These results indicate that the crystallinity and porosity of **JLU-SOF2** remain stable up to 250 °C, and are preserved partially up to 350 °C. Similar are made for **JLU-SOF3** (Fig. S12, S15 and Table S9†). The up-tail phenomenon was analysed in detail based on the pore volumes and scanning electron microscopy (SEM) images of these materials (Fig. S16, Tables S8 and S9,† the detailed analysis is provided in the ESI†). The ¹H NMR spectra and solubility of these materials indicate that the TMBTI blocks remained intact the samples treated at 350 °C (Fig. S17 and S18,† detailed information is presented in the ESI†). The high thermal stability of these two SOFs can be attributed to the strong and multiple hydrogen-bonding interactions. In addition, the CH₃CN-exchanged and regenerated samples still exhibit good crystallinity.

Gas adsorption and separation

The high pore volume and permanent porosity of these two materials inspired us to probe into their potential for application as gas adsorbents. Both **JLU-SOF2** and **JLU-SOF3** exhibit stepwise CO₂ adsorption at 195 K with a slight hysteresis loop, and the uptake values are 13.75 and 15.18 mmol g⁻¹, respectively (Fig. 5a). Furthermore, **JLU-SOF2** and **JLU-SOF3** also display notable CO₂ adsorption amounts at higher temperatures with 4.25 and 4.49 mmol g⁻¹ at 273 K, and 1.98 and 2.34 mmol g⁻¹ at 298 K, respectively (Fig. 5b). It is worth mentioning that the CO₂ uptake values of **JLU-SOF3** rank

among the values of activated carbon,⁵⁶ ZIF-100,⁵⁷ JLU-MOF50 (ref. 58) and the highest values for SOFs except for HOF-5a¹⁵ and IISERP-HOF1 (ref. 38) (Tables S10 and S11†). In contrast to the high CO₂ uptake values, much less N₂ is adsorbed, 0.24 and 0.36 mmol g⁻¹ at 273 K, and these values are decreased to 0.11 and 0.14 mmol g⁻¹ at 298 K, respectively (Fig. S20†). These observed differential adsorption behaviors between CO₂ and N₂ for **JLU-SOF2** and **JLU-SOF3** imply that these materials are probably potential separators for capture of CO₂ from flue-gas.

To predict their separation ability for CO₂/N₂, ideal adsorbed solution theory (IAST) was used to calculate the selectivity for the binary mixtures based on single-component adsorption data fitting using the dual-site Langmuir-Freundlich (DSLFF) equation at 298 K. As expected, the selectivity of **JLU-SOF2** was calculated to be 30.0 and 29.1, respectively, for the binary mixtures of CO₂/N₂ (v/v: 15/85 and 10/90) at 298 K and 1 bar and the values are 22.8 and 23.6 for **JLU-SOF3** under the same conditions (Fig. 5c–f). Encouragingly, these values are higher than those of considerable SOF materials, and exceed those of MOF materials such as PCN-88,³ UiO-67,⁵⁹ and NU-1000 (ref. 60) (Table S12†). The highly selective adsorption of CO₂ over N₂ may be explained by the following aspects: (1) the kinetic diameter of CO₂ (3.3 Å) is smaller than that of N₂ (3.64 Å); (2) the attractive and unique interactions formed by the quadrupole moment of CO₂ will increase the adsorption ability; and (3) the host cavities containing carboxyl groups exhibit stronger affinity behavior for CO₂ than for N₂, which can be identified using the isosteric adsorption enthalpy (*Q*_{st}). As shown in Fig. S21 and S22,† the *Q*_{st} of CO₂ and N₂ are 24.5 and 12.1 kJ mol⁻¹ for **JLU-SOF2**, and 21.8 and 17.0 kJ mol⁻¹ for **JLU-SOF3** at zero coverage, respectively, as calculated from the adsorption data at 273 and 298 K.

Another particularly salient feature of **JLU-SOF2** and **JLU-SOF3** is the adsorption and separation ability for light hydrocarbons under ambient conditions. Single-component gas sorption isotherms measured at 298 K under 1 bar show that the adsorption amounts of CH₄ for **JLU-SOF2** and **JLU-SOF3** are 0.96 and 1.05 mmol g⁻¹, those for C₂H₆ are 4.04 and 4.41 mmol g⁻¹, and those for C₃H₈ are 4.11 and 4.70 mmol g⁻¹, respectively (Fig. 6a–c). Following a temperature decrease to 273 K, the maximum adsorption amounts are increased to 1.67 and 1.73 mmol g⁻¹ for CH₄, 5.10 and 5.79 mmol g⁻¹ for C₂H₆, and 4.55 and 5.51 mmol g⁻¹ for C₃H₈. The used materials exhibit a similar adsorption capacity for these gases to the pristine samples (Fig. S23 and Table S7†). The selectivity of **JLU-SOF2** for equimolar binary mixtures of C₂H₆/CH₄ and C₃H₈/CH₄ is 16.3 and 48.1 at 298 K and 1 bar, respectively (Fig. S24†). There are two primary reasons for the different adsorption ability and high selectivity: (1) the strength of mutual gas interaction decreases as the molecular size decreases,⁶¹ and (2) the hydrocarbon-building block dispersion interaction is higher for larger and more polarizable molecules,^{62,63} which can be proved from the gas affinity towards **JLU-SOF2** in the order of C₃H₈ > C₂H₆ > CH₄ according to the *Q*_{st} of 43.8, 32.3 and 16.6 kJ mol⁻¹ at zero coverage, respectively (Fig. 6d). With respect to **JLU-SOF3**, the selectivity for C₂H₆/CH₄ and C₃H₈/CH₄ is 17.8 and 89.2, respectively (Fig. 6e and f). Similarly, the gas affinity of **JLU-SOF3** shows a consistent trend for C₃H₈, C₂H₆, and CH₄ and

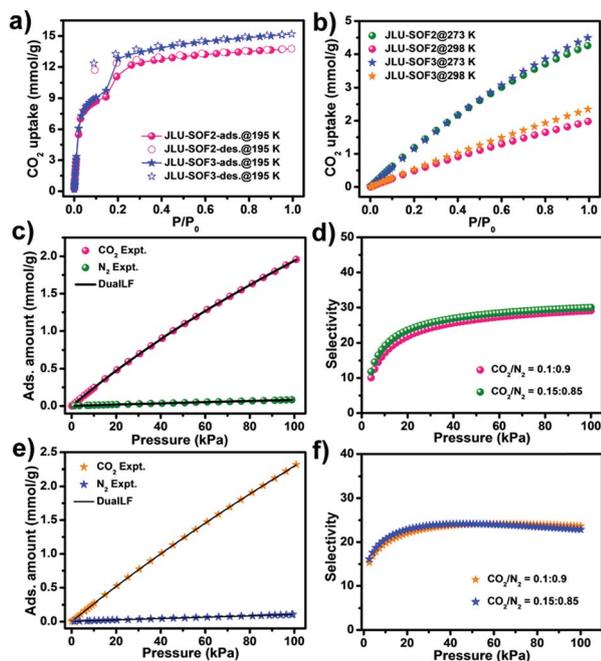


Fig. 5 CO₂ adsorption isotherms at 195 K (a) and at 273 and 298 K (b) for **JLU-SOF2** and **JLU-SOF3**; CO₂ and N₂ adsorption isotherms at 298 K under 1 bar along with the DSLF fits (c and e) and the adsorption selectivity of the binary mixtures predicted using IAST (d and f) for **JLU-SOF2** and **JLU-SOF3**.



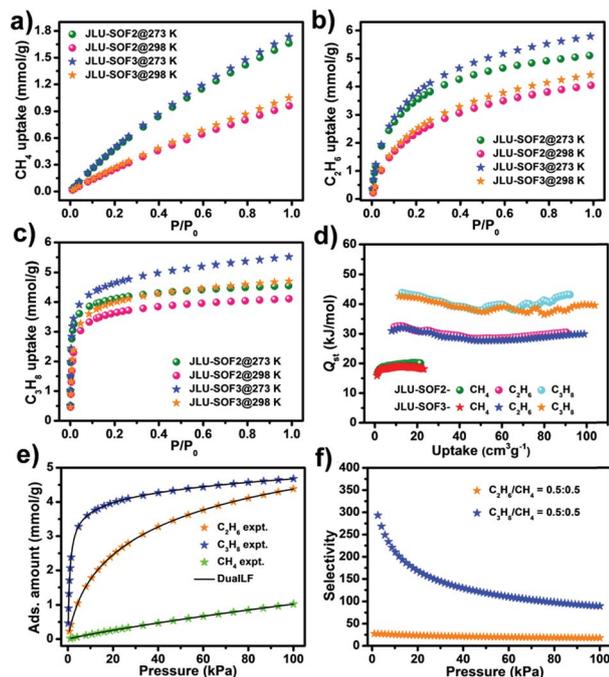


Fig. 6 CH_4 (a), C_2H_6 (b), and C_3H_8 (c) adsorption isotherms at 273 and 298 K as well as the corresponding Q_{st} (d) for JLU-SOF2 and JLU-SOF3; CH_4 , C_2H_6 , and C_3H_8 adsorption isotherms at 298 K under 1 bar along with DSLF fits (e) and the adsorption selectivity for equimolar binary mixtures of $\text{C}_2\text{H}_6/\text{CH}_4$ and $\text{C}_3\text{H}_8/\text{CH}_4$ (f) for JLU-SOF3.

the Q_{st} is 42.7, 31.0, and 15.8, respectively. These results imply that the suitable pore sizes of these two SOFs induce stronger host-guest interactions with larger hydrocarbons than with the smaller ones. It should be mentioned that only one SOF material, HOF-TCPB, has been reported to separate light hydrocarbons significantly, and the article mainly focuses on the separation of CH_4 from C_4 hydrocarbons.⁶⁴ In addition to this separation, the selectivity of our materials for CO_2/CH_4 mixtures (v/v: 50/50 and 5/95) was explored, and the values of which are 2.2 and 2.1 for JLU-SOF2 and 2.3 and 2.2 for JLU-SOF3 at 298 K and 1 bar, respectively (Fig. S25[†]). The above studies indicate the high potential for application of these two materials in selective capturing of light hydrocarbons and upgrading of natural gas.

Conclusions

In summary, by utilizing a 'direction-oriented' strategy and selectively modulating reaction conditions, two 3D SOF materials were synthesized based on different hydrogen-bonding modes from a singular pre-designed non-planar building block. JLU-SOF2 and JLU-SOF3 exhibit permanent porosity, high thermal stability, and good recyclability. It is worth mentioning that these two SOF materials display favourable CO_2 adsorption capacity at ambient temperature. More importantly, these two materials show preferential adsorption of CO_2 over N_2 , as well as CO_2 , C_2H_6 and C_3H_8 over CH_4 . These substantial results indicate that JLU-SOF2 and JLU-SOF3 could

serve as effective reservoirs and separators for greenhouse gas trapping and natural gas upgrading. Notably, the successful crystallization of these two materials offers a feasible method to fabricate 3D SOFs, and in principle, this strategy is not limited to the construction of SOF materials, but suitable for other porous framework materials also.

Experimental

Materials and methods

The chemicals were purchased from commercial sources and used without further purification. The Fourier transform infrared (FT-IR) spectrum was recorded on a Bruker IFS-66v/S FT-IR spectrometer from 400 to 4000 cm^{-1} using KBr pellets. TGA analyses were performed on a TGA Q500 thermogravimetric analyzer with a heating rate of 10 $^\circ\text{C min}^{-1}$ under an air atmosphere. Elemental analyses were completed on a vario MICRO elemental analyzer. PXRD patterns were collected on a Rigaku D/max-2550 diffractometer ($\text{Cu-K}\alpha$ radiation, $\lambda = 1.5418 \text{ \AA}$). Nuclear magnetic resonance (NMR) spectra were obtained using a Varian 300 MHz NMR analyzer. The SEM images were measured using a HITACHI SU-8010.

Synthesis of TMBTI

The TMBTI building block was synthesized by Suzuki-Miyaura reaction according to the literature with some adjustments,⁵⁰ and the detailed information is given in the ESI.[†]

Preparation of JLU-SOF2

TMBTI (6 mg) dissolved in DOA (1 mL) was sealed in a vial, and then heated at 85 $^\circ\text{C}$ for 2 days. Colorless hexagonal prism-shaped crystals, JLU-SOF2, were obtained and dried in air with a yield of 75%. Elemental analysis (%) calcd for JLU-SOF2 ($\text{C}_{33}\text{H}_{24}\text{O}_{12} \cdot 2\text{C}_4\text{H}_8\text{O}_2$, TMBTI \cdot 2DOA): C 62.43, H 5.11; found: C 60.13, H 4.99.

Preparation of JLU-SOF3

TMBTI (6 mg) was dissolved in DOA (1 mL), and then acetonitrile was slowly diffused into the solution at room temperature for 1 day. The colorless hexagonal rod-like crystals, JLU-SOF3, were collected and air-dried with a yield of 72%. Elemental analysis (%) calcd for JLU-SOF3 ($\text{C}_{33}\text{H}_{24}\text{O}_{12} \cdot 2\text{C}_4\text{H}_8\text{O}_2$, TMBTI \cdot 2DOA): C 62.43, H 5.11; found: C 60.11, H 4.88.

Regeneration of JLU-SOF2 and JLU-SOF3

The prepared JLU-SOF2 (or JLU-SOF3) was dissolved in ethanol by ultrasonication, and then the solution was evaporated in a rotary evaporator to give the TMBTI building block. The regeneration procedures were the same as the original preparation of JLU-SOF2 (or JLU-SOF3).

X-ray crystallography

Crystallographic data of as-synthesized JLU-SOF2 and JLU-SOF3 were acquired on a Bruker Apex II CCD diffractometer using graphite-monochromated Mo- $\text{K}\alpha$ radiation ($\lambda = 0.71073 \text{ \AA}$) at



293 K and at 153 K, respectively. The structures of these two SOFs were resolved by a direct method and refined by full-matrix least-squares on F^2 using the SHELXTL program.⁶⁵ The non-hydrogen atoms were refined with anisotropic displacement parameters and the hydrogen atoms were located geometrically with isotropic parameters. In addition, the "SQUEEZE" command was utilized because of the significantly disordered solvent molecules in the pores. Topological analysis was carried out using TOPOS 4.0.⁶⁶ The final formulae of the structures were derived from the crystallographic data, TGA and elemental analyses data. The summarized crystallographic data and all the hydrogen-bonds data are presented in Tables S2–S4.† The supplementary crystallographic data for **JLU-SOF2** and **JLU-SOF3** (CCDC numbers: 1842179 and 1861545†) can be achieved free of charge from the Cambridge Crystallographic Data Centre upon request at www.ccdc.cam.ac.uk/data_request/cif.

Gas adsorption measurements

Before gas adsorption measurements, **JLU-SOF2** and **JLU-SOF3** were exchanged with acetonitrile 8 times for 2 days to fully remove the guest molecules, DOA, which can be identified by TGA analyses. Then, the exchanged samples were activated by using the 'outgas' function of the surface area analyzer at 100 °C for 10 h. Surface areas were calculated using N₂ adsorption isotherms at 77 K obtained using a Micromeritics ASAP 2040 instrument. CO₂ adsorption isotherms measured at 195 K were acquired on a Micromeritics ASAP 2020 instrument. CO₂, CH₄, C₂H₆, C₃H₈, and N₂ adsorption isotherms were collected on a Micromeritics 3-Flex and a Micromeritics ASAP 2020 PLUS HD88 instrument at 273 and 298 K.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

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