

Cite this: *Chem. Sci.*, 2019, 10, 2336

All publication charges for this article have been paid for by the Royal Society of Chemistry

Vacancy-enhanced generation of singlet oxygen for photodynamic therapy†

Shanyue Guan,^{‡a} Li Wang,^{‡ab} Si-Min Xu,^{‡c} Di Yang,^b Geoffrey I. N. Waterhouse,^d Xiaozhong Qu^{ib*} and Shuyun Zhou^{*a}

Oxygen vacancy (OV) engineering in semiconductors can greatly enhance the separation of photo-induced electron-hole pairs, thereby enhancing the photocatalytic activity. Taking inspiration from this, we prepared a novel BiOBr-H/Rub₂d composite by functionalizing OV-rich BiOBr (named BiOBr-H) with a carboxyl functionalized ruthenium photosensitizer (Ru(bpy)₂C-pyCl₂, abbreviated as Rub₂d), which was then successfully applied for photodynamic therapy (PDT). Density functional theory (DFT) calculations confirmed efficient electron transfer from the Rub₂d complex to the intermediate energy level of BiOBr-H under visible light irradiation. *In vitro* and *in vivo* studies demonstrated that BiOBr-H/Rub₂d was a superior agent for photodynamic therapy compared with the free ruthenium complex. The theoretical and experimental data presented thus reveal for the first time that abundant OVs in BiOBr-H can significantly improve the photocatalytic activity of a photosensitizer, resulting in the generation of more reactive oxygen species to enhance PDT. The findings of this study thus offer a new strategy for the development of highly efficient cancer therapies.

Received 27th November 2018
Accepted 20th December 2018

DOI: 10.1039/c8sc05275a

rsc.li/chemical-science

Introduction

Cancer has always been one of the most common diseases in humans, which has been a threat to human beings.^{1,2} Photodynamic therapy (PDT) has been widely applied in cancer treatment due to its non-invasive properties, few side effects, easy procedure and short treatment time relative to surgery or chemotherapy.^{3,4} In the PDT system, the mechanism is driven *via* the excitation of a photosensitizer (PS) to its triplet state, from which energy is then transferred to triplet oxygen, leading to the production of singlet oxygen (¹O₂).^{5–7} Over the past decade, researchers have developed various PSs for PDT, including indocyanine green (ICG),⁸ chlorin e6 (Ce6),^{9–12} zinc phthalocyanine (ZnPc),¹³ and ruthenium complexes.^{14,15} However, traditional PSs suffer from photo-bleaching under irradiation, resulting in the recombination of electron-hole pairs, leading to low production rates of singlet oxygen.¹⁶

Therefore, it is highly desirable to realize the reaction mechanism and discover a method that can prevent the recombination of electron-hole pairs and improve the PDT therapeutic effect.

As far as we are concerned, oxygen vacancies (OVs), *i.e.*, the number of oxygen atoms expected in a compound is less (or missing) than what it should be in its perfect crystal lattice,¹⁷ can facilitate photogenerated charge separation in semiconductors.^{18,19} As a result, OVs can offer more carriers for surface reactions that lead to the generation of reactive oxygen species (ROS), such as ¹O₂, H₂O₂, ·O₂[–], ·OH, *etc.*^{20–22} OVs have always been considered as an effective electron trap since they are electron-deficient.^{23,24} Recently, many semiconductors have been engineered with abundant OVs, including TiO₂, MnO₂ and BiOBr.^{25,26} By virtue of the abundant OVs, low toxicity, moderate band gap and outstanding photostability properties of BiOBr,¹⁷ we meticulously designed a novel composite by functionalizing OV-rich BiOBr (denoted as BiOBr-H) with a PS (ruthenium complexes). Under this circumstance, the generation of ROS (not only singlet oxygen but also superoxide and hydroxyl radicals) can be considerably enhanced, leading to outstanding PDT performance. This method can not only be applied in this specific system but also provide a mechanism for other PSs, resulting in a better understanding of the oxygen vacancy engineering PDT process.

Herein, a BiOBr-H/Rub₂d composite was successfully prepared *via* the interaction between BiOBr-H and a ruthenium complex PS (di(2,2'-bipyridine) 2,2'-bipyridine-4,4'-dicarboxyl dichlororuthenium(II), denoted herein as Rub₂d), as detailed in

^aKey Laboratory of Photochemical Conversion and Optoelectronic Materials, Technical Institute of Physics and Chemistry, Chinese Academy of Sciences, Beijing, 100190, P. R. China. E-mail: zhou_shuyun@mail.ipc.ac.cn

^bCollege of Materials Science and Opto-Electronic Technology, University of Chinese Academy of Sciences, Beijing 100049, China. E-mail: quxz@iccas.ac.cn

^cState Key Laboratory of Chemical Resource Engineering, Beijing University of Chemical Technology, 100029, Beijing, P. R. China

^dSchool of Chemical Sciences, The University of Auckland, Auckland 1142, New Zealand

† Electronic supplementary information (ESI) available. See DOI: 10.1039/c8sc05275a

‡ These authors equally contributed to this work.

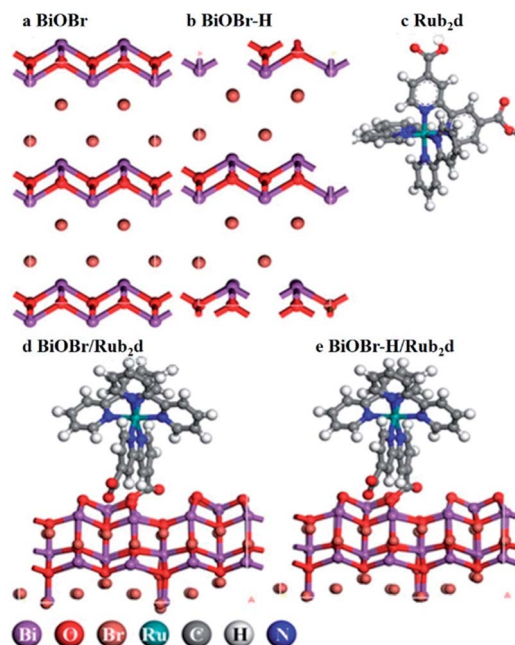
the ESI.[†] Compared with the RuBr₂d complex alone, the BiOBr-H/RuBr₂d agent can considerably improve the production of ROS under light irradiation, which can be verified by both ESR data and the DFT calculation. Specifically, the ¹O₂ yield of BiOBr-H/RuBr₂d (0.49) was more than twice that of RuBr₂d (0.22). Furthermore, both *in vitro* and *in vivo* studies confirmed that the BiOBr-H/RuBr₂d composite was a potent PDT agent for cancer treatment. Density functional theory (DFT) calculations established that the photogenerated electrons of RuBr₂d can be facily transferred to the intermediate energy level located in the forbidden zone of BiOBr-H, facilitating electron-hole separation in BiOBr-H/RuBr₂d and thus enhancing the ¹O₂ yield. To demonstrate the general applicability of this strategy, here we additionally designed and synthesized two BiOBr-H/PS composites that bind specifically with the photosensitizers indocyanine green (ICG) and zinc phthalocyanine (ZnPc). As illustrated in our strategy, compared with the PS complex, the BiOBr-H/PS composite significantly increased the generation of ¹O₂ under irradiation. Thus, this strategy of OV-enhanced generation of singlet oxygen holds great potential in the precise treatment of cancer.

Results and discussion

Structural and morphological characterization

BiOBr was fabricated *via* a hydrothermal method,²⁷ followed by heating in O₂ for 4 h (the product was denoted as BiOBr-H). The hydrothermally synthesized BiOBr without calcination was employed here for comparison purposes in PDT tests (and denoted simply as BiOBr). The optimized geometries of the models BiOBr, BiOBr-H, RuBr₂d, BiOBr/RuBr₂d, and BiOBr-H/RuBr₂d are displayed in Scheme 1. Ru(bpy)₂C-pyCl₂ (denoted as RuBr₂d) was prepared *via* a two-step reaction of ruthenium chloride with 2,2'-bipyridine and 2,2'-bipyridine-4,4'-dicarboxylic acid, the molecular structure of which is shown in Fig. S1.[†] BiOBr-H/RuBr₂d and BiOBr/RuBr₂d were synthesized *via* the electrostatic attraction between positively charged BiOBr-H (zeta potential +12.60 mV) and negatively charged Ru(bpy)₂C-pyCl₂ (zeta potential −0.97 mV). The zeta potential of the resulting composite product is measured to be +12.30 mV (Fig. S2[†]).

Powder X-ray diffraction (XRD) was used to characterize the structures of the various samples (Fig. 1a). BiOBr, BiOBr-H, BiOBr/RuBr₂d and BiOBr-H/RuBr₂d all showed diffraction patterns typical of BiOBr (JCPDS, PDF #73-2061). Transmission electron microscopy (TEM) and elemental mapping (EM) were used to examine the compositional uniformity of BiOBr-H/RuBr₂d. TEM (Fig. 1b and c) revealed that BiOBr-H/RuBr₂d possessed a plate-like shape with a diameter of around 200 nm. The lattice fringe spacing of 0.346 nm can be attributed to the (011) plane of BiOBr-H (Fig. 1b). Furthermore, TEM images of BiOBr-H were recorded after incubation in the cell culture medium (DMEM) or buffer solution (PBS) pH 6.5 with H₂O₂ (tumor microenvironment) for 24 h. As shown in Fig. S3[†] the morphology of BiOBr-H showed no apparent change after 24 h of incubation, indicating that BiOBr-H was stable in the acidic environment and cell culture medium. Atomic force microscopy



Scheme 1 Schematic illustration of the optimized geometries of BiOBr, BiOBr-H, RuBr₂d, BiOBr/RuBr₂d, and BiOBr-H/RuBr₂d. The detailed chemical structure of RuBr₂d is listed in Fig. S1.[†]

(AFM) determined the thickness of the sheets in BiOBr-H/RuBr₂d to be approximately 1.0 nm (Fig. S4[†]). Elemental mapping images demonstrate that the RuBr₂d complex was uniformly distributed over the BiOBr-H surface (Fig. 1d).

UV-vis absorption spectra of BiOBr-H, BiOBr, RuBr₂d, BiOBr/RuBr₂d and BiOBr-H/RuBr₂d are presented in Fig. S5.[†] BiOBr and BiOBr-H displayed weak absorption across the 300–900 nm

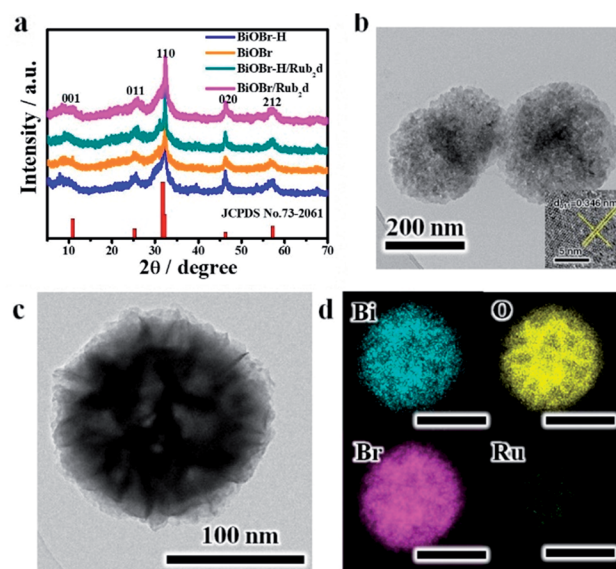


Fig. 1 (a) XRD patterns for BiOBr, BiOBr/RuBr₂d, BiOBr-H and BiOBr-H/RuBr₂d; the standard PDF card for BiOBr is also shown (JCPDS no. 73-2061). (b) TEM image (inset: HRTEM image) and (c and d) elemental maps for BiOBr-H/RuBr₂d.



region. After functionalization with the Rub₂d complex, a strong absorption signal at 460 nm appeared, which can readily be attributed to metal-to-ligand charge-transfer (MLCT) transitions in the Rub₂d complex.²⁸ Fourier transform infrared (FT-IR) spectroscopy was utilized to analyze the functional groups present in the various samples. For pristine BiOBr and BiOBr-H (Fig. S6†), peaks at 527 cm⁻¹ and 3445 cm⁻¹ were observed and assigned to the Bi-O stretching and O-H stretching modes, respectively. The peak at 2870 cm⁻¹ is a C-H stretching vibration of polyvinyl pyrrolidone (PVP).²⁹ This feature was much weaker for BiOBr-H. The Rub₂d complex showed a strong peak around 1700 cm⁻¹ due to the carboxylate groups of the C-py ligand³⁰ (Fig. S6†). The spectra of BiOBr/Rub₂d and BiOBr-H/Rub₂d also show the IR peaks of the Rub₂d complex, confirming the successful loading of the Rub₂d complex onto these materials.

The chemical composition of BiOBr, BiOBr-H, Rub₂d, and BiOBr-H/Rub₂d was further probed by X-ray photoelectron spectroscopy (XPS) (Fig. S7–S10 and Table S1†). Comparing the spectra of BiOBr and BiOBr-H, the most obvious difference is the absence of the C 1s and N 1s signals in the survey spectrum of BiOBr-H. This is explained by the decomposition of PVP during the calcination step used to synthesize BiOBr-H. The spectrum of the Rub₂d complex displayed signals due to Ru, O, N and C, whilst that of the BiOBr-H/Rub₂d complex contained signals due to C, O, Br, Ru, N and Bi elements (Fig. S7–S8†). High-resolution Bi 4f XPS spectra of BiOBr and BiOBr-H (Fig. S9†) showed two peaks at 158.9 and 164.2 eV in a 4 : 3 area ratio, which can readily be assigned to the Bi 4f_{7/2} and 4f_{5/2} peaks of Bi³⁺.²⁶ For BiOBr-H, two additional peaks are seen at 157.8 and 163.5 eV which are assigned to the Bi 4f_{7/2} and 4f_{5/2} peaks of Bi³⁺ near oxygen vacancies.²⁶ The O 1s spectrum (Fig. S10†) of BiOBr-H was deconvoluted into two peaks at 529.7 and 530.9 eV. The intense peak at 529.7 eV is typical of lattice oxygen (O²⁻) in BiOBr-H, whereas the weaker peak at higher binding energy is due to the adsorbed hydroxyl species or oxygen species at vacancy sites (O⁻).¹⁹ The combination of the Bi 4f and O 1s data provides good evidence for the presence of OVs in the BiOBr-H sample.

Due to the intrinsic fluorescence properties of the Rub₂d complex, we probed the fluorescence properties of Rub₂d, BiOBr-H/Rub₂d and BiOBr/Rub₂d. All samples displayed an obvious fluorescence signal centered at 650 nm (similar to Rub₂d at a concentration of 250 µg mL⁻¹) (Fig. S11†). To further characterize the fluorescence performance of the samples, the two-photon fluorescence intensities of BiOBr, BiOBr-H, Rub₂d, BiOBr/Rub₂d and BiOBr-H/Rub₂d were examined under 800 nm irradiation (Fig. S12†). Rub₂d exhibited a higher up-conversion photoluminescence (UCPL) intensity than BiOBr-H/Rub₂d, whilst no two-photon signal was seen for BiOBr and BiOBr-H. For BiOBr-H/Rub₂d, a quadratic relationship was found between the laser power and the fluorescence intensity, confirming that the up-conversion photoluminescence emission of the sample was due to a two-photon excitation process (Fig. S13†). The emission lifetime of Rub₂d and BiOBr-H/Rub₂d was further examined under 520 nm excitation. The emission lifetime of BiOBr-H/Rub₂d was 309.44 ns, longer than that of

pristine Rub₂d in solution (234.47 ns) (Fig. S14 and Table S2†). This proves conclusively that the triple state of Rub₂d was enhanced by interaction with BiOBr-H.^{31,32}

Electron Spin Resonance (ESR) was applied to quantify vacancy-induced ¹O₂ generation. As shown in Fig. 2a, the pristine BiOBr possessed few vacancies. However, the ESR spectrum of BiOBr-H revealed a greatly enhanced OV signal, resulting from the calcination step used in its synthesis. In comparison, BiOBr-H/Rub₂d contained relatively few vacancies (as did BiOBr/Rub₂d). Using 2,2,6,6-tetramethylpiperidine (Temp) as an ¹O₂ trap (Fig. 2b), BiOBr, BiOBr-H, BiOBr/Rub₂d, BiOBr-H/Rub₂d, and Rub₂d all gave similar signals prior to irradiation. Following irradiation, the ¹O₂ signal of BiOBr-H/Rub₂d increased significantly, whereas the signals of BiOBr/Rub₂d and Rub₂d did not change much. The ¹O₂ signal of irradiated BiOBr-H/Rub₂d was twice that of Rub₂d (Fig. 2c). Clearly, the abundant OVs in BiOBr-H/Rub₂d are effective for the capture of electrons,^{23,33} thereby suppressing electron-hole pair recombination in the semiconductor. Accordingly, the excited electrons transferred from Rub₂d under xenon lamp irradiation are captured by the OVs in BiOBr-H, leading to enhanced generation of ¹O₂.

To determine the optimal Rub₂d loading capacity, a series of BiOBr-H/Rub₂d (x%) were prepared and their ESR spectra were collected under xenon lamp irradiation (Fig. 2d). The ESR signal increased from BiOBr-H/Rub₂d (25%) to BiOBr-H/Rub₂d (200%). The signal intensity gradually decreased as the loading was increased from BiOBr-H/Rub₂d (200%) to BiOBr-H/Rub₂d (400%) (Fig. 2d and S15†). Therefore, BiOBr/Rub₂d (200%) displayed the optimum performance and we used this Rub₂d loading in all subsequent experiments. In addition, to further realize this oxygen-vacancy engineering mechanism, we further prove the generation of other radical species, e.g., [•]O₂⁻ and [•]OH of ROS. As expected, the ESR signal of [•]O₂⁻ and [•]OH was also

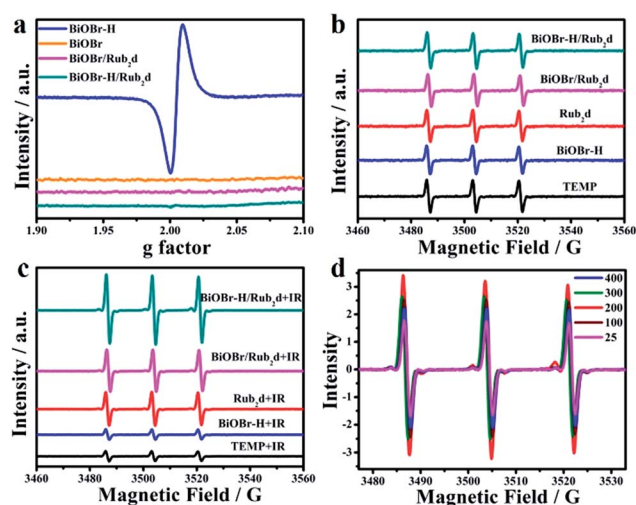


Fig. 2 (a) ESR spectra of BiOBr-H, BiOBr, Rub₂d and BiOBr-H/Rub₂d (500 µg mL⁻¹) without irradiation and (b) ESR spectra of BiOBr-H, BiOBr, BiOBr/Rub₂d and BiOBr-H/Rub₂d (500 µg mL⁻¹) without irradiation and (c) with irradiation using a xenon lamp (100 mW cm⁻²) for 10 min. (d) ESR spectra of BiOBr-H/Rub₂d with different loadings after irradiation for 10 min (Rub₂d concentration 500 µg mL⁻¹).



significantly enhanced as the irradiation time increased from 0 min to 10 min (Fig. S16†). This can further support our oxygen-vacancy engineering generation of ROS, which can be applied as a tool for the improvement of cancer therapy.

DFT calculations were used to gain deeper insights about the impact of oxygen-vacancy engineering on the generation of singlet oxygen. The schematic illustration of the structures of BiOBr and BiOBr-H is displayed in Fig. S17.† The surface energies of the low-index facets of BiOBr were first calculated and are listed in Table S3.† It was found that the (011) facet of BiOBr possesses the smallest surface energy (0.428 J m^{-2}), revealing that the (011) facet will be preferentially exposed on the surface of BiOBr and BiOBr-H, in good agreement with the HRTEM findings (Fig. 1b). The density of states of both BiOBr and BiOBr-H is subsequently calculated and displayed in Fig. 3a and b. The band gap energy of BiOBr was calculated to be 2.70 eV, in good accord with that estimated from the UV-vis absorbance spectrum (Fig. S4†). For BiOBr-H, an intermediate energy level appears in the forbidden zone, resulting from the presence of abundant oxygen vacancies.^{34,35} The density of states of BiOBr-H indicates that the intermediate energy level is mainly composed of Bi-6p orbitals (Fig. S18†). Using the work functions determined for BiOBr and BiOBr-H, the positions of their conduction band minimum (CBM) and valence band maximum (VBM) were calculated (Fig. 3c). The HOMO and LUMO positions of the Rub₂d complex were also calculated and are displayed in Fig. 3c. The calculations reveal that the photogenerated electrons in Rub₂d can be efficiently transferred to the intermediate energy level of BiOBr-H with a large driving force of 1.311 eV, thereby facilitating rapid electron-hole separation. Similar defect-induced charge separation has also been reported in previous work.^{36,37} Furthermore, the binding energy between BiOBr-H and Rub₂d was determined to be -4.482 eV , confirming the strong interaction between BiOBr-H and Rub₂d.

Measurements of singlet oxygen

We further investigated $^1\text{O}_2$ generation *via* a chemical trapping experiment using 1,3-diphenylisobenzofuran (DPBF) as

a trapping agent and Rose Bengal (RB) as a standard photosensitizer ($^1\text{O}_2$ quantum yield $\Phi_{\text{RB}} = 0.86$ in ethanol). The absorption of DPBF at 410 nm was monitored under 520 nm irradiation, to follow the decay rate of the photosensitizing process. The detailed calculations used in this analysis are provided in Fig. S19.† The absorbance of the DPBF solution at 410 nm decreased quite quickly with irradiation time in the presence of BiOBr-H/Rub₂d, confirming the fast generation of $^1\text{O}_2$. In comparison, the decrease in the absorbance at 410 nm was relatively slow in the presence of BiOBr/Rub₂d and Rub₂d, again confirming that the OV in BiOBr-H/Rub₂d facilitate $^1\text{O}_2$ generation. The $^1\text{O}_2$ quantum yield of BiOBr-H/Rub₂d was calculated to be 0.49, which is approximately 2-fold higher than that of Rub₂d alone (0.22). To further evaluate the benefits of the oxygen vacancies in BiOBr-H for promoting $^1\text{O}_2$ generation, we loaded the photosensitizers zinc phthalocyanine (ZnPc) and indocyanine green (ICG) on BiOBr-H and studied the generation of $^1\text{O}_2$ *via* UV-vis spectroscopy. The singlet oxygen yields of BiOBr-H/ZnPc and BiOBr-H/ICG were 0.50 and 0.28, respectively (Table S4†), much higher than those of ZnPc (0.31) and ICG (0.15) (Fig. S20 and S21†). Photocurrent response measurements provided further evidence for the excellent charge transfer properties of BiOBr-H/Rub₂d. As expected, Rub₂d and BiOBr-H showed only a weak photocurrent response, whereas BiOBr-H/Rub₂d showed a remarkable photocurrent response (Fig. S22†). Therefore, it can be concluded that the incorporation of the probe Rub₂d into BiOBr-H can accelerate the generation of $^1\text{O}_2$ from the Rub₂d. To summarize, adsorption of Rub₂d on OV-rich BiOBr-H enhances the population of the Rub₂d triplet state (evidenced by a phosphorescence lifetime increase). Facile electron transfer occurs from excited Rub₂d to BiOBr-H.

In vitro anticancer activity

The internalization and *in vitro* cellular bio-imaging of the BiOBr-H/Rub₂d was assessed on the Hela cell line using a Laser Scanning Confocal Microscope (LSCM) with an excitation wavelength of 488 nm. After incubation with BiOBr-H/Rub₂d for 24 h, the nucleus of the cells was stained with DAPI which gives a blue color emission. Fig. S23† shows that the BiOBr-H/Rub₂d was internalized by cells through endocytosis³⁸ and was mainly located in the cytoplasm. Furthermore, from the TEM observation (Fig. S24†), the BiOBr-H/Rub₂d samples can be internalized by the Hela cells and then enter mitochondria while the others accumulate in lysosomes.³⁸ In addition, elemental mapping can vividly monitor the Bi, Br and Ru distribution during the cell culture and the results confirmed the stability of BiOBr-H/Rub₂d inside Hela cells (Fig. S25†). We further investigated the two-photon imaging properties of BiOBr-H/Rub₂d using three various cell lines: Hela, HepG-2 and MCF-7 cells. As shown in Fig. S26,† the cells exhibit fluorescence in the green and red channels under 800 nm excitation. This confirms the potential of BiOBr-H/Rub₂d as an imaging agent for the detection of cancer cells *via* two-photon fluorescence imaging techniques.

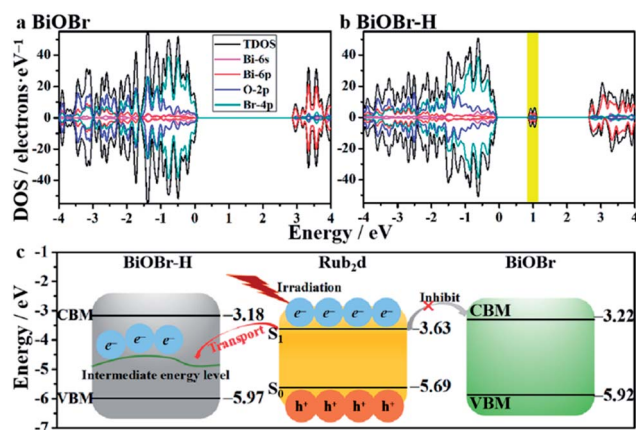


Fig. 3 Band structures of BiOBr (a) and BiOBr-H (b). The band edge positions of BiOBr, BiOBr-H and Rub₂d with the vacuum level set at 0 eV (c).

To confirm the anticancer activity of BiOBr-H/Rub₂d was solely due to the photodynamic therapy, the cytotoxicity of BiOBr and BiOBr-H was checked by the CCK8 assay in Hela cells (Fig. S27†). After 24 h of incubation, BiOBr and BiOBr-H showed negligible toxicity towards the Hela cells even at concentrations up to 500 $\mu\text{g mL}^{-1}$. Further, Rub₂d, BiOBr/Rub₂d, and BiOBr-H/Rub₂d also showed negligible toxicity (Fig. 4a). Rub₂d and BiOBr/Rub₂d slightly inhibited the growth of the Hela cells, following irradiation for 10 min (Fig. 4a). Remarkably, the growth of Hela cells was significantly inhibited when incubated with BiOBr-H/Rub₂d and irradiated at 520 nm (100 mW cm^{-2}) for 10 min. Similar results were obtained for the MCF-7 and HepG-2 cell lines (Fig. S28 and S29†). To visualize the PDT effect on the cells, the dead and live cells were stained with Propidium Iodide (PI) and calcein-AM, respectively (Fig. 4b). It is apparent that the majority of cells were dead after being treated with the BiOBr-H/Rub₂d and irradiated, whereas they remained in relatively good condition without irradiation. The effect of illumination on Rub₂d treated cells is less obvious. Again, the data point to a synergistic effect resulting from the adsorption of Rub₂d on BiOBr-H. To validate the benefits of OV-engineering for ROS generation *in vitro*, 2',7'-dichlorofluorescein diacetate (DCFH-DA) was used. Following diffusion into cells, DCFH-DA is deacetylated by cellular esterases to give a non-fluorescent compound, which can be oxidized by ROS into 2',7'-dichlorofluorescein (DCF), which has characteristic excitation and emission maxima of 488 and 525 nm, respectively.³⁹ Hela

cells treated with phosphate-buffered saline (PBS) and irradiated at 520 nm with a luminous power of 100 mW cm^{-2} for 10 min showed a typical Hela cell green fluorescence signal (Fig. S30†). Cells incubated with BiOBr-H/Rub₂d and Rub₂d showed a significantly enhanced green fluorescence due to the generation of DCF. The intensity of the fluorescence signal was strongly dependent on the incubation time. However, at all incubation times BiOBr-H/Rub₂d demonstrated the strongest green fluorescence signal.

Fluorescence imaging and *in vivo* antitumor assay

Encouraged by the *in vitro* PDT efficacy of BiOBr-H/Rub₂d, *in vivo* PDT experiments were subsequently performed involving monitoring of the volume of the tumors after irradiation. Hela tumor-bearing mice were divided into 5 groups (control group, BiOBr-H/Rub₂d, BiOBr + IR, Rub₂d + IR, and BiOBr-H/Rub₂d + IR), with each group having 5 mice (Fig. 4c). No significant therapeutic effect was seen for the BiOBr-H/Rub₂d group without irradiation and BiOBr with irradiation. In contrast, the tumor volumes for the Rub₂d + IR group decreased. Notably, the tumor volume of the BiOBr-H/Rub₂d + IR group was dramatically reduced compared with that of the Rub₂d + IR group (Fig. 4d), demonstrating that the high local concentration of ROS generated by BiOBr-H/Rub₂d under irradiation greatly improved the effectiveness of PDT therapeutic effects. In addition, no apparent body weight losses were found during the whole treatment (Fig. 4e). Imaging studies on the excised major organs and the tumor showed the strong fluorescence of the isolated tumor tissue after the i.v. injection of BiOBr-H/Rub₂d for 10 min and 24 h, respectively (Fig. S31 and S32†). To further investigate the therapeutic effect of each treatment group, the tumor tissues were analysed by hematoxylin and eosin (H&E) staining after the treatment. As shown in Fig. S33,† the main organs including the heart, liver, spleen, lungs and kidneys showed no obvious damage after BiOBr-H/Rub₂d + IR treatment, while there was serious necrosis in the tumor tissues after the treatment. These results demonstrate the negligible toxicity of the BiOBr-H/Rub₂d PDT treatment and excellent therapeutic potential of BiOBr-H/Rub₂d for cancer therapy.

Conclusions

In summary, under the hypothesis that abundant OVs in BiOBr-H enable the improvement of the photocatalytic activity of a photosensitizer, we developed a novel theranostic agent by functionalizing the BiOBr-H with a PS (Rub₂d). The resultant BiOBr-H/Rub₂d indeed demonstrates outstanding therapeutic properties for PDT, which originated from the efficacy of OVs in the enhancement of ROS generation in tumor cells during light irradiation, conferring strong ablation effects both *in vitro* and *in vivo*. The phenomenon can be explained using density functional theory (DFT) calculations. All results suggest the feasibility of the concept (OVs can enhance ROS generation). Therefore, this work highlights the potential of vacancy engineering in the development of effective theranostic agents for cancer therapy.

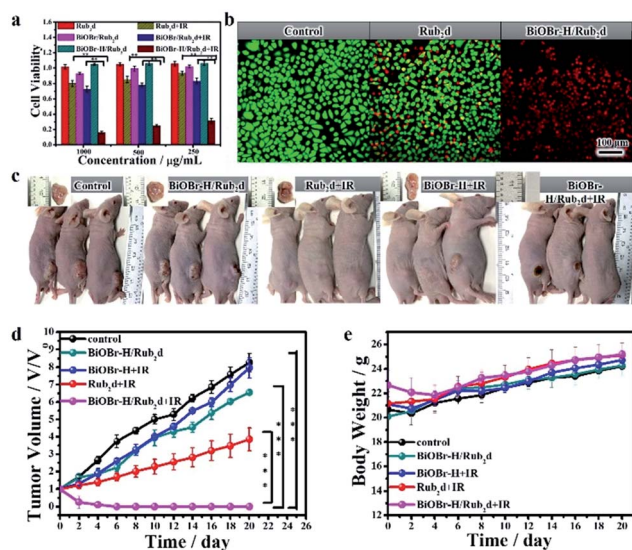


Fig. 4 (a) Cell viability of Hela cells incubated with Rub₂d, BiOBr-H/Rub₂d or BiOBr/Rub₂d for 24 h with and without irradiation (xenon lamp, 100 mW cm^{-2}), respectively. The viability was the average of six measurements ($n = 6$). (b) Fluorescence imaging of Rub₂d and BiOBr-H/Rub₂d with and without IR therapy, respectively. Live/dead Hela cells are green/red (calcein AM/PI), respectively. (c) *In vivo* photographs of various groups of mice administrated with 200 μL of PBS, BiOBr-H, Rub₂d or BiOBr-H/Rub₂d with or without 520 nm irradiation for 10 min (xenon lamp, 100 mW cm^{-2}). (d) Tumor size and (e) body weight. The tumor volume was the average of five measurements ($n = 5$) and the error bars indicate the standard deviation. * $p < 0.05$, ** $p < 0.01$, and *** $p < 0.001$.



Conflicts of interest

There are no conflicts to declare.

Acknowledgements

Shanyue Guan, Li Wang and Si-Min Xu contributed to this work equally. The authors are grateful for the financial support from the National Natural Science Foundation of China (21805293 and 51473169), the Joint NSFC-ISF Research Program by National Natural Science Foundation of China and Israel Science Foundation (51561145004) and the Director Foundation of the Technical Institute of Physics and Chemistry, Chinese Academy of Sciences (Grant No. 2018-GSY). All animal procedures were performed in accordance with the Guidelines for Care and Use of Laboratory Animals of Technical Institute of Physics and Chemistry, Chinese Academy of Sciences, and the experiments were approved by the Animal Ethics Committee of Technical Institute of Physics and Chemistry, Chinese Academy of Sciences.

Notes and references

- 1 Z. Yu, Y. Ge, Q. Sun, W. Pan, X. Wan, N. Li and B. Tang, *Chem. Sci.*, 2018, **9**, 3563–3569.
- 2 W. Yin, T. Bao, X. Zhang, Q. Gao, J. Yu, X. Dong, L. Yan, Z. Gu and Y. Zhao, *Nanoscale*, 2018, **10**, 1517–1531.
- 3 Z. Zhou, J. Song, R. Tian, Z. Yang, G. Yu, L. Lin, G. Zhang, W. Fan, F. Zhang, G. Niu, L. Nie and X. Chen, *Angew. Chem., Int. Ed.*, 2017, **56**, 6492–6496.
- 4 J. Ji, X. Li, T. Wu and F. Feng, *Chem. Sci.*, 2018, **9**, 5816–5821.
- 5 X. Li, S. Lee and J. Yoon, *Chem. Soc. Rev.*, 2018, **47**, 1174–1188.
- 6 W. Zhang, S. Li, X. Liu, C. Yang, N. Hu, L. Dou, B. Zhao, Q. Zhang, Y. Suo and J. Wang, *Adv. Funct. Mater.*, 2018, **28**, 1706375.
- 7 J. Ge, M. Lan, B. Zhou, W. Liu, L. Guo, H. Wang, Q. Jia, G. Niu, X. Huang, H. Zhou, X. Meng, P. Wang, C. S. Lee, W. Zhang and X. Han, *Nat. Commun.*, 2014, **5**, 4596.
- 8 W. S. Kuo, C. N. Chang, Y. T. Chang, M. H. Yang, Y. H. Chien, S. J. Chen and C. S. Yeh, *Angew. Chem., Int. Ed.*, 2010, **49**, 2711–2715.
- 9 Q. Chen, L. Feng, J. Liu, W. Zhu, Z. Dong, Y. Wu and Z. Liu, *Adv. Mater.*, 2016, **28**, 7129–7136.
- 10 M. Abbas, Q. Zou, S. Li and X. Yan, *Adv. Mater.*, 2017, **29**, 1605021.
- 11 D. He, L. Hai, X. He, X. Yang and H.-W. Li, *Adv. Funct. Mater.*, 2017, **27**, 1704089.
- 12 J. Kim, H. R. Cho, H. Jeon, D. Kim, C. Song, N. Lee, S. H. Choi and T. Hyeon, *J. Am. Chem. Soc.*, 2017, **139**, 10992–10995.
- 13 S. Cui, D. Yin, Y. Chen, Y. Di, H. Chen, Y. Ma, S. Achilefu and Y. Gu, *ACS Nano*, 2012, **7**, 676–688.
- 14 F. E. Poynton, S. A. Bright, S. Blasco, D. C. Williams, J. M. Kelly and T. Gunnlaugsson, *Chem. Soc. Rev.*, 2017, **46**, 7706–7756.
- 15 C. Mari, V. Pierroz, R. Rubbiani, M. Patra, J. Hess, B. Spingler, L. Oehninger, J. Schur, I. Ott, L. Salassa, S. Ferrari and G. Gasser, *Chem.–Eur. J.*, 2014, **20**, 14421–14436.
- 16 S. Guan, D. Yang, Y. Weng, H. Lu, X. Meng, X. Qu and S. Zhou, *Adv. Healthcare Mater.*, 2018, **7**, 1701123.
- 17 C. E. Rodríguez Torres, G. A. Pasquevich, P. M. Zélis, F. Golmar, S. P. Heluani, S. K. Nayak, W. A. Adeagbo, W. Hergert, M. Hoffmann, A. Ernst, P. Esquinazi and S. J. Stewart, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2014, **89**, 104411.
- 18 H. Li, J. Li, Z. Ai, F. Jia and L. Zhang, *Angew. Chem., Int. Ed.*, 2018, **57**, 122–138.
- 19 H. Wang, D. Yong, S. Chen, S. Jiang, X. Zhang, W. Shao, Q. Zhang, W. Yan, B. Pan and Y. Xie, *J. Am. Chem. Soc.*, 2018, **140**, 1760–1766.
- 20 H. Li, J. Shang, Z. Yang, W. Shen, Z. Ai and L. Zhang, *Environ. Sci. Technol.*, 2017, **51**, 5685–5694.
- 21 H. Li, F. Qin, Z. Yang, X. Cui, J. Wang and L. Zhang, *J. Am. Chem. Soc.*, 2017, **139**, 3513–3521.
- 22 P. Trogadas, J. Parrondo and V. Ramani, *ACS Appl. Mater. Interfaces*, 2012, **4**, 5098–5102.
- 23 H. Chen, S. He, X. Cao, S. Zhang, M. Xu, M. Pu, D. Su, M. Wei, D. G. Evans and X. Duan, *Chem. Mater.*, 2016, **28**, 4751–4761.
- 24 Y. Zhao, X. Jia, G. Chen, L. Shang, G. I. N. Waterhouse, L.-Z. Wu, C.-H. Tung, D. O'Hare and T. Zhang, *J. Am. Chem. Soc.*, 2016, **138**, 6517–6524.
- 25 Y. Yang, L. C. Yin, Y. Gong, P. Niu, J. Q. Wang, L. Gu, X. Chen, G. Liu, L. Wang and H. M. Cheng, *Adv. Mater.*, 2018, **30**, 1704479.
- 26 H. Li, J. Shang, Z. Ai and L. Zhang, *J. Am. Chem. Soc.*, 2015, **137**, 6393–6399.
- 27 X. Shi, X. Chen, X. Chen, S. Zhou, S. Lou, Y. Wang and L. Yuan, *Chem. Eng. J.*, 2013, **222**, 120–127.
- 28 L. N. Lameijer, D. Ernst, S. L. Hopkins, M. S. Meijer, S. H. C. Askes, S. E. Le Devedec and S. Bonnet, *Angew. Chem., Int. Ed.*, 2017, **56**, 11549–11553.
- 29 Z. Zhang, B. Zhao and L. Hu, *J. Solid State Chem.*, 1996, **121**, 105–110.
- 30 N. Liu, R. A. Assink and C. J. Brinker, *Chem. Commun.*, 2003, 370–371.
- 31 R. Tian, D. Yan, C. Li, S. Xu, R. Liang, L. Guo, M. Wei, D. G. Evans and X. Duan, *Nanoscale*, 2016, **8**, 9815–9821.
- 32 S. Guan, Y. Weng, M. Li, R. Liang, C. Sun, X. Qu and S. Zhou, *Nanoscale*, 2017, **9**, 10367–10374.
- 33 Y. Li, W. Zhang, J. Niu and Y. Chen, *ACS Nano*, 2012, **6**, 5164–5173.
- 34 A. Agrawal, R. K. Saroj, T. A. Dar, P. Baraskar, P. Sen and S. Dhar, *J. Appl. Phys.*, 2017, **122**, 195303.
- 35 M. Ichimura, H. Sakakibara, K. Wada and M. Kato, *J. Appl. Phys.*, 2013, **114**, 114505.
- 36 S. K. Cushing, F. Meng, J. Zhang, B. Ding, C. K. Chen, C.-J. Chen, R.-S. Liu, A. D. Bristow, J. Bright, P. Zheng and N. Wu, *ACS Catal.*, 2017, **7**, 1742–1748.
- 37 A. Sarkar, K. Karmakar and G. G. Khan, *J. Phys. Chem. C*, 2017, **121**, 25705–25717.
- 38 F. Zhao, Y. Zhao, Y. Liu, X. Chang, C. Chen and Y. Zhao, *Small*, 2011, **7**, 1322–1337.
- 39 Z. Lv, H. Wei, Q. Li, X. Su, S. Liu, K. Y. Zhang, W. Lv, Q. Zhao, X. Li and W. Huang, *Chem. Sci.*, 2018, **9**, 502–512.

