

Cite this: *Chem. Sci.*, 2019, 10, 1098

All publication charges for this article have been paid for by the Royal Society of Chemistry

Host–guest selectivity in a series of isorecticular metal–organic frameworks: observation of acetylene-to-alkyne and carbon dioxide-to-amide interactions†

Jack D. Humby,^{‡a} Oguarabau Benson,^{‡b} Gemma L. Smith,^{‡a} Stephen P. Argent,^{‡c} Ivan da Silva,^{‡d} Yongqiang Cheng,^e Svemir Rudić,^{‡d} Pascal Manuel,^d Mark D. Frogley,^{‡f} Gianfelice Cinque,^f Lucy K. Saunders,^{‡f} Iñigo J. Vitórica-Yrezábal,^a George F. S. Whitehead,^{‡a} Timothy L. Easun,^{‡g} William Lewis,^{‡b} Alexander J. Blake,^{‡b} Anibal J. Ramirez-Cuesta,^{‡d} Sihai Yang^{‡*a} and Martin Schröder^{‡*a}

In order to develop new porous materials for applications in gas separations such as natural gas upgrading, landfill gas processing and acetylene purification it is vital to gain understanding of host–substrate interactions at a molecular level. Herein we report a series of six isorecticular metal–organic frameworks (MOFs) for selective gas adsorption. These materials do not incorporate open metal sites and thus provide an excellent platform to investigate the effect of the incorporation of ligand functionality *via* amide and alkyne groups on substrate binding. By reducing the length of the linker in our previously reported MFM-136, we report much improved CO₂/CH₄ (50 : 50) and CO₂/N₂ (15 : 85) selectivity values of 20.2 and 65.4, respectively (1 bar and 273 K), in the new amide-decorated MOF, MFM-126. The CO₂ separation performance of MFM-126 has been confirmed by dynamic breakthrough experiments. *In situ* inelastic neutron scattering and synchrotron FT-IR microspectroscopy were employed to elucidate dynamic interactions of adsorbed CO₂ molecules within MFM-126. Upon changing the functionality to an alkyne group in MFM-127, the CO₂ uptake decreases but the C₂H₂ uptake increases by 68%, leading to excellent C₂H₂/CO₂ and C₂H₂/CH₄ selectivities of 3.7 and 21.2, respectively. Neutron powder diffraction enabled the direct observation of the preferred binding domains in MFM-126 and MFM-127, and, to the best of our knowledge, we report the first example of acetylene binding to an alkyne moiety in a porous material, with over 50% of the acetylene observed within MFM-127 displaying interactions (<4 Å) with the alkyne functionality of the framework.

Received 14th August 2018
Accepted 1st October 2018

DOI: 10.1039/c8sc03622e

rsc.li/chemical-science

Introduction

Porous metal–organic frameworks (MOFs) have attracted considerable attention due to their promise in a wide range of

applications, notably in gas adsorption, selectivity and catalysis.^{1–6} Modulation of organic linkers and metal nodes allows fine-tuning of properties to tailor MOFs for the application of interest. In the field of gas adsorption, MOFs have shown advantages for selective uptake of specific guests, such as CO₂ (ref. 7–9) and C₂H₂.^{10–12} Capture and separation of CO₂ from flue gas streams *via* the utilization of selective and regenerable sorbents, such as MOFs, provides a promising alternative pathway to conventional amine-scrubbers.¹³ Moreover, MOF adsorbents also have applications in natural gas (containing ~15% CO₂) upgrading, and the selective separation of landfill gas (40–60% CO₂).^{14,15} Other industrially significant separations include the purification of C₂H₂, an important feedstock in the petrochemical industry. C₂H₂ is produced by the partial combustion of CH₄ and hence requires separation from CH₄ and CO₂ to obtain high-purity C₂H₂.¹⁶ However, the identical kinetic diameters of 3.30 Å and similar boiling points of –84 °C and –79 °C, for C₂H₂ and CO₂, respectively, make C₂H₂/CO₂

^aSchool of Chemistry, University of Manchester, Oxford Road, Manchester, M13 9PL, UK. E-mail: Sihai.Yang@manchester.ac.uk; M.Schröder@manchester.ac.uk

^bSchool of Chemistry, University of Nottingham, Nottingham, NG7 2RD, UK

^cDepartment of Chemistry, University of Warwick, CV4 7AL, UK

^dISIS Facility, STFC Rutherford Appleton Laboratory, Oxfordshire OX11 0QX, UK

^eOak Ridge National Laboratory, Oak Ridge, TN 37831, USA

^fDiamond Light Source, Harwell Science and Innovation Campus, Oxfordshire, OX11 0DE, UK

^gSchool of Chemistry, Cardiff University, Cardiff CF10 3XQ, UK

† Electronic supplementary information (ESI) available: Correspondence and requests for materials should be addressed to S. Y. and M. S. CCDC 1857732–1857743. For ESI and crystallographic data in CIF or other electronic format see DOI: 10.1039/c8sc03622e

‡ These authors contributed equally to this work.

separation under ambient conditions a highly challenging task.^{17–19}

Common strategies for enhancing host–guest interactions in MOFs include incorporation of open metal sites,²⁰ polar functional groups (e.g., $-\text{NH}_2$, $-\text{CONH}-$, $-\text{OH}$, $-\text{F}$)^{17,21–24} and narrowing pore channels by use of small ligands.²⁵ For example, polar nitrogen-containing groups remain a favored approach to enhancing CO_2 adsorption, as shown in a crystallographic study that visualized CO_2 molecules directly interacting with the amine group in a Zn–MOF.²⁶ However, we have previously reported neutron diffraction and scattering data revealing that the high CO_2 uptake in amide-functionalized MFM-136 is not solely due to guest–host interactions at the amide group, but is a combination of geometry, pore size and functionality that lead to improved gas sorption properties.²⁷ Whilst many MOFs have been reported for their gas sorption properties, it is often difficult to fully account for differences in performance owing to many variables such as surface area, porosity and pore geometry, functionality and presence of open metal sites. Thus, to aid design-based approaches for improved materials, thorough investigations of isorecticular series of MOFs such as the IRMOF,²⁸ UiO-66 (ref. 29) and MOF-74 (ref. 30) series are important.

Herein we present a comprehensive investigation into the roles of functional groups, pore geometry and porosity in enhancing selective gas binding through a series of six isostructural MOFs (MFM-126–128 and MFM-136–138; Table 1). MFM-137 and MFM-138 were designed by adapting previously reported MFM-136,²⁷ replacing the amide group with an ethynyl bond and phenyl ring, respectively. Further modification of MFM-136–138 was achieved by removal of the central phenyl unit of the linker to produce ‘shortened’ derivatives, MFM-126–128. This systematic approach allows us to isolate either the effect of varying pore size or functionality to rationalize the properties of these materials. Our approach has been to focus on the role of ligand sites for substrate binding and, significantly, we report herein the first example of binding of acetylene to the alkyne groups in a porous material at crystallographic resolution.

Experimental

Neutron powder diffraction (NPD) of gas-loaded MFM-126 and MFM-127

NPD data were collected at WISH beamline at ISIS Muon and Neutron facility. Acetone-exchanged MFM-126 and MFM-127 were loaded into 11 mm diameter vanadium cans, heated at 393 K and degassed at 1×10^{-7} mbar for 2 days to activate the sample. Data for the bare framework were collected after placing the sample can into a liquid helium cryostat and cooling to 7 K. CO_2 , C_2D_2 and CD_4 gases were dosed into the system after warming the system to 250 and 150 K. The gases were dosed volumetrically from a calibrated volume and, to ensure gases were fully adsorbed into the sample without condensation elsewhere, the system was cooled to 7 K slowly over 3 h. Structures of gas-loaded MFM-126 and MFM-127 were solved by

sequential difference Fourier analyses followed by Rietveld refinements using TOPAS software (see ESI Section 4†).

Inelastic neutron scattering of CO_2 -loaded MFM-126

INS data were collected on TOSCA beamline at ISIS Muon and Neutron facility. MFM-126 was loaded into an 11 mm diameter vanadium can and outgassed at 10^{-7} mbar at 393 K for 2 days. After placing the sample into a He-cooled cryostat, INS data of the bare framework were collected at 7 K. A loading of 1.0 CO_2/Cu was dosed volumetrically from a calibrated volume at room temperature and gradually cooled to 7 K to allow the guest species to fully adsorb into MFM-126. INS data of 1.0 CO_2/Cu of MFM-126 were collected at 7 K. Experimentally obtained INS data were compared with modelled data obtained *via* density functional theory (DFT) calculations and simulated using OClmax software (see ESI for further details†).

Synchrotron FTIR microspectroscopy of CO_2 -loaded MFM-126

Single crystals of MFM-126 were loaded onto a ZnSe slide and placed into a Linkam FTIR600 variable temperature gas-tight cell fitted with ZnSe windows. The MOF sample was activated *in situ* under a flow of N_2 whilst heating the Linkam stage to 413 K for 6 h. Partial pressures of zeolite-dried gases N_2 and CO_2 were controlled by varying the volumetric flow of the two gases *via* separate mass flow controllers. FTIR spectra were collected at the B22 MIRIAM beamline at Diamond Light Source using a polarized and highly bright synchrotron IR source connected to a Bruker Hyperion 3000 IR microscope with a $15\times$ objective and MCT detector (liq. N_2 cooled). Spectra (256 scans) were measured at room temperature with a $20 \times 20 \mu\text{m}$ beam, in the spectral range of $4000\text{--}650 \text{ cm}^{-1}$ (4 cm^{-1} resolution).

Gas breakthrough

Breakthrough experiments were performed using a Hiden Isochema Automated Breakthrough Analyzer with integrated mass spectrometer. MFM-126 (950 mg) was packed into a stainless steel column and the sample was activated by heating to 393 K under a flow of He (100 mL min^{-1}) overnight. Breakthrough experiments were conducted using gas mixtures of CO_2/N_2 (15 : 85) and equimolar CO_2/CH_4 which were flowed over MFM-126 at a total flow rate of 10 mL min^{-1} at 298 K and 1.0 bar (see ESI† for further detail).

Results and discussion

Structures

Solvothermal reactions of isophthalate–pyrimidyl linkers H_2L^1 – H_2L^6 (Table 1) with $\text{Cu}(\text{NO}_3)_2$ affords isostructural (3,6)-connected MOFs of *eea* topology. All these frameworks are constructed from $\text{Cu}(\text{II})$ cations bridged by four carboxylate groups from four independent linkers and capped by two pyrimidyl nitrogen donors to form elongated octahedral $[\text{Cu}_2(\text{RCOO})_4(\text{NR})_2]$ nodes (Table 1, Fig. 1). The isophthalate units bridge adjacent $\{\text{Cu}_2\}$ paddlewheels to form a *Kagomé* lattice net (Fig. 1d), and these are connected by coordination of a pyrimidyl nitrogen center to an axial position of the $\{\text{Cu}_2\}$



Table 1 Structures of linkers, coordination environments, cage lengths, BET surface areas and experimentally and calculated pore volumes for MFM-126–128 and MFM-136–138. Colors: C, grey; H, white; O, red; N, blue; Cu, teal

	MFM-126	MFM-127	MFM-128	MFM-136 ^a	MFM-137	MFM-138
Structures of linkers						
Coordination environment						
Cage sizes						
Long cage (A) length/Å	15.4	16.4	20.2	24.9	26.1	29.9
Short cage (B) length/Å	12.3	12.7	14.1	16.2	16.4	18.1
BET surface area/m ² g ⁻¹	1004	1557	1491	1634	1749	1590
Pore volume (N ₂ isotherm)/cm ³ g ⁻¹	0.47	0.52	0.57	0.65	0.61	0.60
Pore volume (single crystal)/cm ³ g ⁻¹	0.52	0.57	0.57	0.64	0.67	0.62

^a Previously reported.²⁷

paddlewheels. The capping of the {Cu₂} paddlewheels at both axial positions results in the absence of any open Cu(II) sites. Lu *et al.* recently reported HHU-2,³¹ a (3,4,6)-connected Cu(II)-based MOF using H₂L¹, but its structure has both pyrimidine N atoms (instead of just one as in MFM-126) bound to Cu(II) ions and therefore does not have free N centers pointing into the pore as observed in MFM-126 and all MOFs in this current series. MFM-

137 and MFM-138 are formed by replacement of the amide group of MFM-136 with alkyne and phenyl moieties, respectively. The removal of a central phenyl ring from MFM-136–138 affords the corresponding shortened derivatives MFM-126–128, respectively.

The MOFs in this series all incorporate two types of cages, the larger of which, cage A (Fig. 1a) is comprised of six ligands



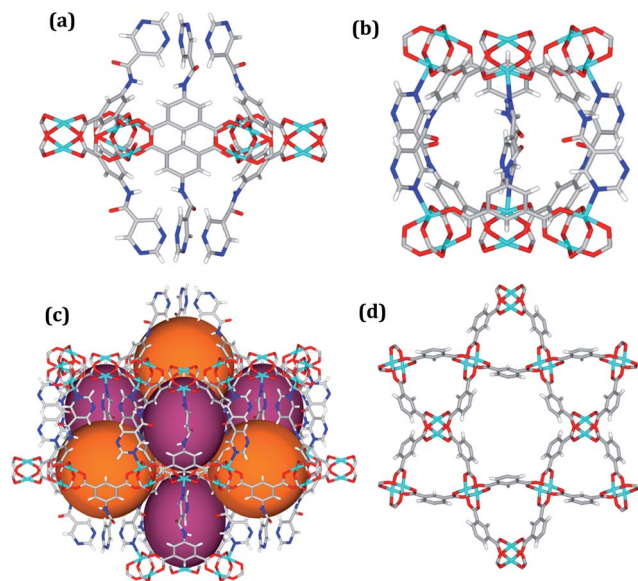


Fig. 1 Views of crystal structure of MFM-126. (a) Cage A; (b) cage B. (c) View of the alternate packing of cages A (void space coloured orange) and B (void space coloured plum). (d) View along the *c*-axis of the *Kagomé* lattice in MFM-126. Colours: C, grey; H, white; O, red; N, blue; Cu, teal.

and six $[\text{Cu}_2(\text{RCOO})_4(\text{NR})_2]$ paddlewheel units forming a hexagonal bipyramid. The six $\{\text{Cu}_2\}$ units form the six equatorial vertices of this cage and the hexagonal window of the *Kagomé* lattice (Fig. 1d). Six pyrimidyl units form the apical vertices, whereby ligands form six of the twelve faces of the hexagonal bipyramid. The smaller cage B (Fig. 1b) is constructed from six ligands and six $\{\text{Cu}_2\}$ paddlewheels forming a ditrigonal scale-nohedral cage, whereby two sets of three $\{\text{Cu}_2\}$ paddlewheels bridged by three linker isophthalate units form triangular windows of these MOFs being comprised of discrete cages A and B, which are packed in an alternating manner (Fig. 1c), giving highly porous and robust 3D frameworks. Views of the structure along the principal crystallographic axes of MFM-126 are shown in Fig. S3†. The phase purity of all material samples was confirmed by powder X-ray diffraction (PXRD) data (Fig. S4†).

Modulation of porosity and CO_2 adsorption properties

All MOFs were solvent-exchanged with acetone or ethanol before heating at 393 K under dynamic vacuum to produce desolvated materials. The N_2 (77 K) isotherms (Fig. S6†) reveal that the desolvated MOFs have BET surface areas of 1000–1750 $\text{m}^2 \text{g}^{-1}$ and pore volumes of 0.47–0.65 $\text{cm}^3 \text{g}^{-1}$ (Table 1). As expected, MFM-136–138 have higher BET surface areas compared to their isostructural shorter derivatives MFM-126–128. The higher BET surface area and pore volume of MFM-127 ($\text{C}\equiv\text{C}$ linker) compared with MFM-126 ($\text{C}(\text{O})\text{--NH}$ linker) can be attributed to the presence of the space-efficient alkyne group in the linker, which has been shown previously to increase the hypothetical maximum surface area of MOFs.³² The measured pore volumes compare favorably with those calculated directly

from single crystal structures, confirming complete desolvation and phase purity of these materials.

CO_2 adsorption isotherms were measured to 20 bar at 273 and 298 K for all MOFs (Fig. S5†). Throughout the series, it was found that the CO_2 uptake at 20 bar increased with increasing pore volume, indicating that the pore functionality had little effect on high pressure gas adsorption, where porosity is the dominant factor. For example, a 10% increase in CO_2 uptake at 20 bar is observed in MFM-127 compared with MFM-126 (11.3 and 10.2 mmol g^{-1} at 298 K, respectively), corresponding well with the 11% pore volume increase from MFM-126 to MFM-127. However, at 1.0 bar and 298 K, comparing phenyl-functionalized MFM-138 and MFM-128 (2.89 and 3.19 mmol g^{-1} , respectively), it was found that the CO_2 uptake is greater in MFM-128, indicating a stronger interaction with the MFM-128 framework at ambient pressure. This suggests that pore geometry has an important role in low pressure gas uptake, with MFM-126–128 all having greater uptake at 1.0 bar CO_2 than the comparative extended derivatives, MFM-136–138 (Fig. 2a). The influence of functionality on CO_2 uptake can be assessed by comparing MFM-136 and MFM-137 which show similar porosity (Table 1). MFM-136 and MFM-137 show CO_2 uptakes of 7.30 and 5.76 mmol g^{-1} (273 K), respectively, suggesting that amide-functionalized MFM-136 possesses stronger affinity for CO_2 compared to the alkyne groups in MFM-137.

Crucially, however, MFM-126, with the smallest pore volume, has the greatest CO_2 uptake at low pressure across the series, reaching 4.63 mmol g^{-1} at 1 bar and 298 K, presumably due to the greater overlap of attractive interactions between gas molecules and the host framework. This is exemplified further at 0.15 bar, the partial pressure of CO_2 in flue gas streams, where MFM-126 has a 52% higher uptake of CO_2 compared with the next best-performing MOF, MFM-136 (2.94 and 1.94 mmol g^{-1} , respectively) in this series.

Selective sorption in MFM-126 and MFM-127

All the MOFs in this series show highly selective uptakes of CO_2 with respect to CH_4 and N_2 (Fig. 2, Table 2). The selective uptake of CO_2 over the other substrates studied is also evidenced by the analysis of heats of adsorption (Table 2 and Fig. S19†). MFM-126 exhibits the highest CO_2 uptake at low pressures (<1 bar) as well as relatively low CH_4 uptake in the same pressure region (Fig. 2b). The selectivity values were estimated from the single-component isotherms using ideal adsorbed solution theory (IAST) (Table 2).³³ Considering the relative gas concentrations found in natural gas and flue gas streams, MFM-126 exhibits the highest selectivity values across the series for binary mixtures of both CO_2/CH_4 (equimolar) and CO_2/N_2 (15 : 85 composition) with selectivity values of 20.2 and 65.4, respectively at 1 bar and 273 K (Fig. 2d). MFM-126 also exhibits the highest adsorption enthalpy for CO_2 of 30.7 kJ mol^{-1} , which is significantly higher than 17.3 kJ mol^{-1} for CH_4 (Table 2). Conversely, MFM-136 has the lowest CO_2/CH_4 selectivity value of 4.1 and exhibits similar values for the isosteric heat (Q_{st}) for CO_2 and CH_4 adsorption, 20.1 and 18.9 kJ mol^{-1} , respectively. When compared with other

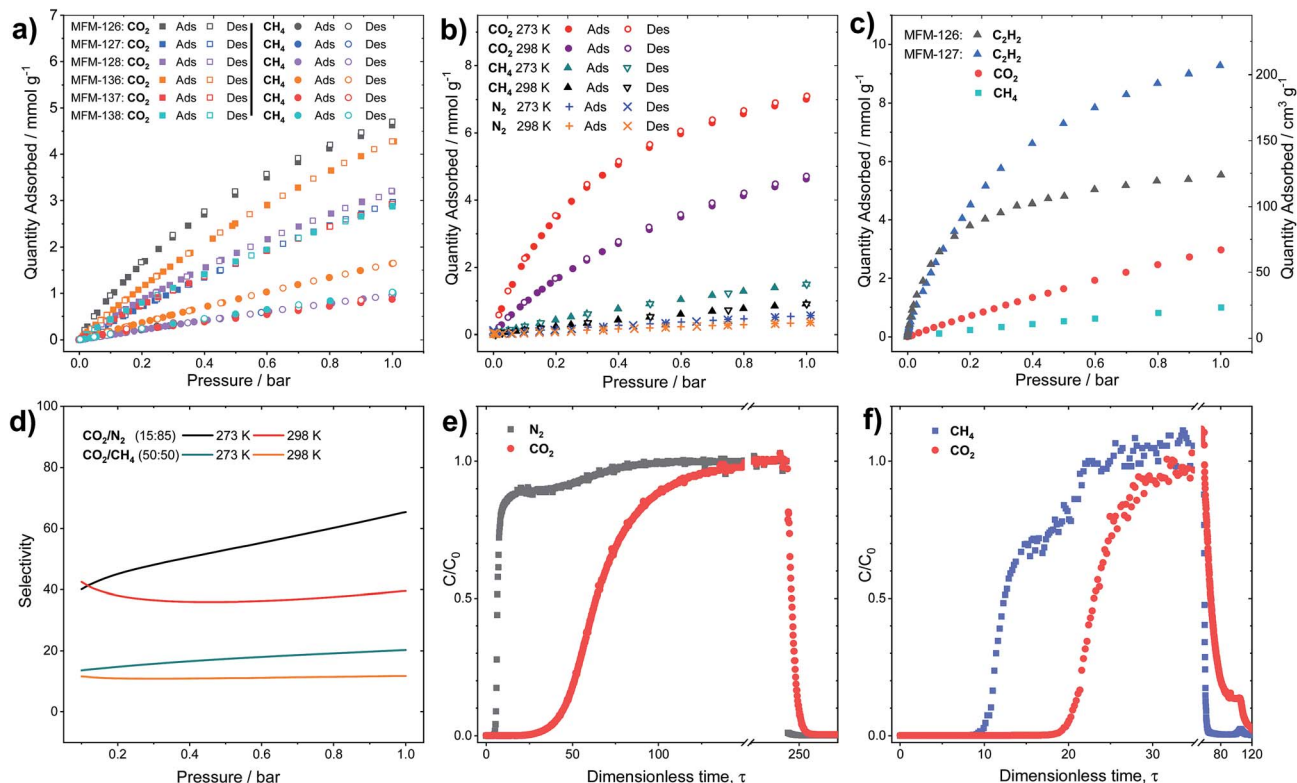


Fig. 2 (a) Sorption isotherms for CO₂ and CH₄ in MFM-126–128 and MFM-136–138 at 298 K. (b) Sorption isotherms for CO₂, CH₄ and N₂ in MFM-126 at 273 and 298 K. (c) Sorption isotherms for C₂H₂, CO₂ and CH₄ in MFM-127 and for C₂H₂ in MFM-126 at 273 K. (d) IAST selectivity for CO₂/N₂ (15 : 85) and equimolar CO₂/CH₄ in MFM-126 at 273 and 298 K. (e and f) Experimental breakthrough curves for the adsorption of CO₂/N₂ (15 : 85) and equimolar CO₂/CH₄ mixtures flowing through a 0.9 mL fixed-bed of MFM-126 at 298 K with a total gas flow of 10 mL min^{−1} at atmospheric pressure. After breakthrough of both species, a He purge was applied leading to desorption of all components.

Table 2 Summary of adsorption data for MFM-126–128 and MFM-136–138

MOF	CO ₂ uptake/ mmol g ^{−1} (1 bar)		CH ₄ uptake/ mmol g ^{−1} (1 bar)		S _{CO₂/CH₄} (50 : 50, 1 bar)		S _{CO₂/N₂} (15 : 85, 1 bar)		Q _{st} /kJ mol ^{−1} (virial method 1)	
	273 K	298 K	273 K	298 K	273 K	298 K	273 K	298 K	CO ₂	CH ₄
MFM-126	7.00	4.63	1.50	0.897	20.2	11.7	65.4	39.6	30.7	17.3
MFM-127	5.72	2.97	1.71	0.991	5.08	3.33	10.6	7.65	25.8	13.0
MFM-128	5.76	3.20	1.77	0.953	5.46	4.53	35.8	18.9	20.4	22.7
MFM-136	7.29	4.28	2.93	1.64	4.07	3.35	37.0	23.2	20.1	18.9
MFM-137	5.76	2.92	1.41	0.870	6.09	4.08	27.6	15.7	19.2	17.1
MFM-138	6.08	2.89	1.75	1.02	5.42	3.87	17.1	15.5	30.0	18.8

leading MOFs, MFM-126 has respectable CO₂/CH₄ and CO₂/N₂ selectivity values (Tables S18 and S19†).

To evaluate the dynamic separation performance of MFM-126, breakthrough experiments were performed with CO₂/N₂ (15 : 85) and equimolar CO₂/CH₄ mixtures (Fig. 2e and f). The gas mixtures were flowed over a fixed-bed packed with MFM-126 with a total flow of 10 mL min^{−1} at 298 K (1 bar). These breakthrough experiments confirmed the separation potential for both gas mixtures as predicted by IAST selectivity calculations (Fig. 2d).

Although the amide-functionalised MFM-126 exhibits higher CO₂ uptake compared with MFM-127, the functional groups

were also investigated for their effect on acetylene adsorption. Interestingly, alkyne-functionalised MFM-127 exhibits a 68% higher acetylene uptake than MFM-126 at 273 K and 1 bar (9.28 and 5.54 mmol g^{−1}, respectively, Fig. 2c). IAST analysis for MFM-127 reveals selectivity values for equimolar mixtures of C₂H₂/CO₂ and C₂H₂/CH₄ to be 3.7 and 21.2, respectively (Fig. S20†). This result is comparable to that observed for [Cu₂(EBTC)] (H₄EBTC = 1,1'-ethynebenzene-3,3',5,5'-tetra-carboxylic acid),³⁴ which shows enhanced acetylene uptake compared with the non-alkyne bridged analogue MOF-505/MFM-100.^{35,36} Whilst the presence of C≡C⋯C≡C interactions between acetylene gas and an alkyne organic linker may be

postulated, this has not yet been confirmed or observed experimentally.

In situ neutron powder diffraction (NPD) of gas-loaded MFM-126 and MFM-127

The absence of open metal sites in this series of MOFs affords an excellent platform to study the role of functional group on gas binding. *In situ* neutron powder diffraction (NPD) has been applied to determine the preferred binding sites of CO₂ and CD₄ in MFM-126 and MFM-127 at loadings of 1.2 CO₂/Cu and 1.0 CD₄/Cu, and C₂D₂ in MFM-127 at a loading of 1.0 C₂D₂/Cu. NPD data for the desolvated samples confirm the complete removal of guest solvents and an absence of structural distortion in the parent solvated material. Relatively low loadings of 1.0–1.2 adsorbate/Cu were used to assess the strongest binding sites within the material without involving notable adsorbate–adsorbate interactions. These loadings represent typically 25–40% of the saturated capacities of each adsorbate. Fourier difference map analysis of the NPD patterns afforded the location of guest CO₂, CD₄ and C₂D₂ molecules which, after further development by Rietveld refinement, allowed unambiguous

determination of gas positions, orientations and crystallographic occupancies within each sample.

MFM-126 displays four binding sites for CO₂, 1–4 (in decreasing order of occupancy; Fig. 3). CO₂-1 is situated in small cage B and exhibits co-operative binding between crystallographically equivalent CO₂ molecules [$C_{CO_2} \cdots O_{CO_2} = 3.30(3) \text{ \AA}$] where the linear bodies of the CO₂ molecules lie parallel with an interaction to an adjacent amide [$O_{CO_2} \cdots N_{amide} = 3.86(5) \text{ \AA}$, $\angle C=O \cdots N = 111^\circ$]. In addition, there are short contacts between a pyrimidyl ring [$C-H_{pyrimidine} \cdots O_{CO_2} = 2.32(5) \text{ \AA}$, $\angle C-H \cdots O = 143^\circ$] and an isophthalate ring [$O_{CO_2} \cdots \text{centroid}_{isophthal.} = 3.15(4) \text{ \AA}$, $\angle C=O \cdots \text{centroid} = 111^\circ$]. CO₂-2 is located in the triangular window of cage B, with a close contact to a C–H group of the isophthalate unit [$O_{CO_2} \cdots H-C_{isophthal.} = 1.71(10) \text{ \AA}$, $\angle O \cdots H-C = 152^\circ$] and a side-on interaction with two adjacent pyrimidine rings [$O_{CO_2} \cdots H-C_{pyrimidine} = 2.17(5) \text{ \AA}$, $\angle O \cdots H-C = 128^\circ$]. CO₂-3 is situated in a pocket between cages A and B with two short contacts to amido N–H units [$O_{CO_2} \cdots H-N_{amide} = 3.77(6) \text{ \AA}$, $\angle O \cdots H-N = 124^\circ$ and $O_{CO_2} \cdots H-N_{amide} = 3.98(8) \text{ \AA}$, $\angle O \cdots H-N = 138^\circ$]. Further to this, there are two other close contacts with a pyrimidyl ring [$O_{CO_2} \cdots C_{pyrimidine} = 2.44(5) \text{ \AA}$,

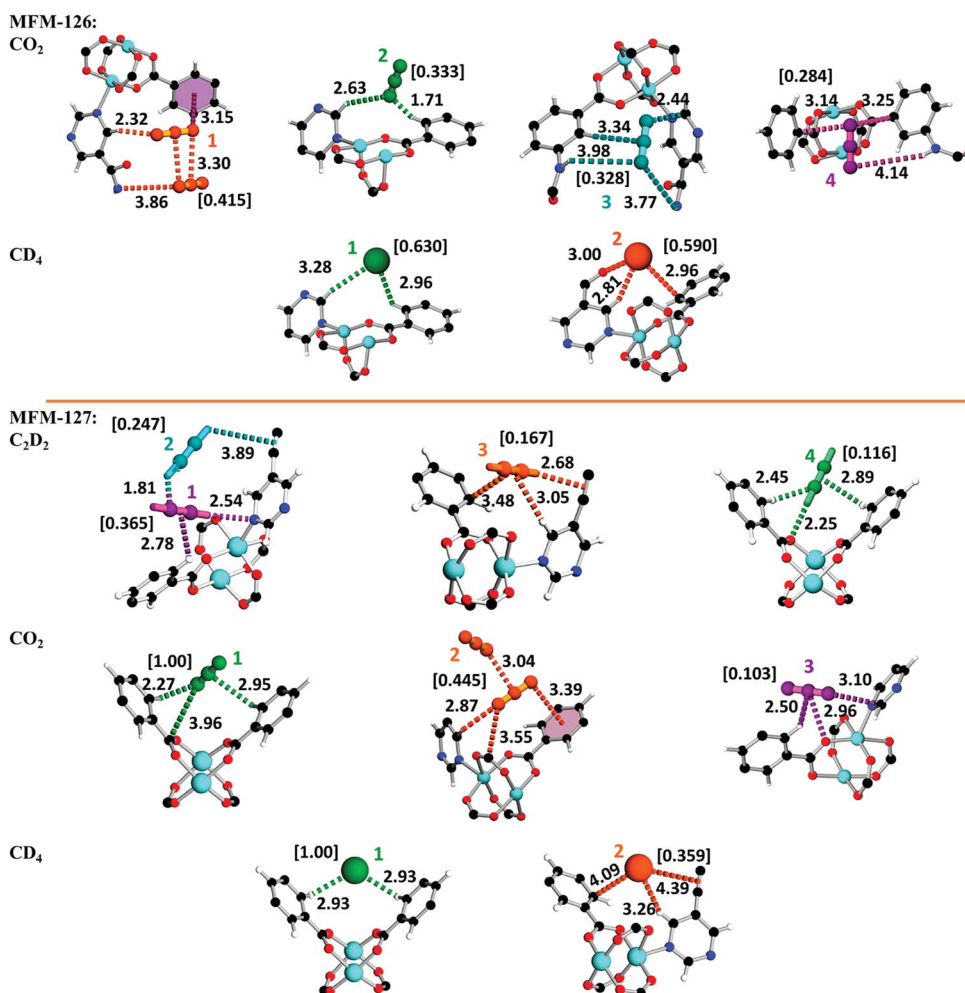


Fig. 3 Binding sites of guests in MFM-126 and MFM-127 derived from Rietveld refinement of NPD data with quoted distances in Å. Colors: C, black; H, white; O, red; N, blue; Cu, teal. Refined occupancies of guest molecules are shown in square brackets.



$\text{C}=\text{O}\cdots\text{C} = 115^\circ$] and a side-on interaction with an isophthalate ring [$\text{C}_{\text{CO}_2}\cdots\text{H}-\text{C}_{\text{isophthal.}} = 3.34(12) \text{ \AA}$, $\angle\text{C}\cdots\text{H}-\text{C} = 172^\circ$]. $\text{CO}_2\text{-4}$ is positioned at the periphery of larger cage **A** with a hydrogen bond to an amido N-H [$\text{O}_{\text{CO}_2}\cdots\text{H}-\text{N} = 4.14(9) \text{ \AA}$, $\angle\text{O}\cdots\text{H}-\text{N} = 149^\circ$] as well as lying in a crevice between two isophthalate units [$\text{O}_{\text{CO}_2}\cdots\text{C}_{\text{isophthal.}} = 3.14(11) \text{ \AA}$, $\text{C}=\text{O}\cdots\text{C} = 81^\circ$ and $\text{O}_{\text{CO}_2}\cdots\text{C}_{\text{isophthal.}} = 3.25(10) \text{ \AA}$, $\text{C}=\text{O}\cdots\text{C} = 96^\circ$].

Analysis of CD_4 -loaded MFM-126 reveals seven sites 1–7, the first two of which have significantly higher crystallographic occupancies of 0.630 and 0.590 than the rest (Fig. 3). $\text{CD}_4\text{-1}$, resides in cage **B** in an equivalent site to that of $\text{CO}_2\text{-2}$, also with close interactions to isophthalate and pyrimidyl rings [$\text{D}_4\text{C}\cdots\text{H}-\text{C}_{\text{isophthal.}} = 2.96(11) \text{ \AA}$, $\angle\text{C}\cdots\text{H}-\text{C} = 122^\circ$ and $\text{D}_4\text{C}\cdots\text{H}-\text{C}_{\text{pyrimidine}} = 3.28(4) \text{ \AA}$, $\angle\text{C}\cdots\text{H}-\text{C} = 159^\circ$]. $\text{CD}_4\text{-2}$ is found in an identical site to that of $\text{CO}_2\text{-1}$ located in the center of cage **B** with two key interactions to an isophthalate ring and an amido carbonyl of the framework [$\text{D}_4\text{C}\cdots\text{H}-\text{C}_{\text{isophthal.}} = 2.81(4) \text{ \AA}$, $\angle\text{C}\cdots\text{H}-\text{C} = 146^\circ$ and $\text{D}_4\text{C}\cdots\text{O}=\text{C}_{\text{amide}} = 3.00(5) \text{ \AA}$, $\angle\text{C}\cdots\text{O}=\text{C} = 140^\circ$].

Refinement of NPD data of C_2D_2 -loaded MFM-127 unveiled five sites, 1–5 (Fig. 3). Interestingly, over 50% of C_2D_2 were found having short contacts ($<4 \text{ \AA}$) to the framework alkyne units. $\text{C}_2\text{D}_2\text{-1}$ is situated in a narrow window between cages **A** and **B**, with closest interactions to pyrimidyl and isophthalate rings [$\text{DC}_2\text{-D}\cdots\text{N}_{\text{pyrimidine}} = 2.54(5) \text{ \AA}$, $\angle\text{C}-\text{D}\cdots\text{N}_{\text{pyrimidine}} = 145^\circ$ and $\eta^2\text{-D}_2\text{C}_2\cdots\text{H}-\text{C}_{\text{isophthal.}} = 2.78(2) \text{ \AA}$, $\angle\text{C}-\text{H}\cdots\text{C}\equiv\text{C}_{\text{centroid}} = 108^\circ$]. Cooperative binding is visualized between $\text{C}_2\text{D}_2\text{-1}$ and $\text{C}_2\text{D}_2\text{-2}$ [$1\text{-D}_2\text{C}\equiv\text{C}\cdots\text{D}-\text{C}_2\text{D}-2 = 1.81(3) \text{ \AA}$, $\angle\text{C}^1\cdots\text{D}-\text{C}^2 = 127^\circ$]. $\text{C}_2\text{D}_2\text{-2}$ also binds to the framework alkyne [$\text{DC}_2\text{-D}\cdots\eta^2\text{-C}\equiv\text{C}_{\text{framework}} = 3.89(5) \text{ \AA}$, $\angle\text{C}-\text{D}\cdots\text{C}\equiv\text{C}_{\text{centroid}} = 123^\circ$] suggesting the framework alkyne has an integral role in the uptake of acetylene in MFM-127. $\text{C}_2\text{D}_2\text{-3}$ is located in the center of cage **B** with H-bonds to a framework alkyne moiety [$\text{DC}_2\text{-D}\cdots\eta^2\text{-C}\equiv\text{C}_{\text{framework}} = 2.68(5) \text{ \AA}$, $\angle\text{C}-\text{D}\cdots\text{C}\equiv\text{C}_{\text{centroid}} = 120^\circ$] as well as close contacts to pyrimidyl and isophthalate rings [$\eta^2\text{-D}_2\text{C}\equiv\text{C}\cdots\text{H}-\text{C}_{\text{pyrimidine}} = 3.05(8) \text{ \AA}$, $\angle\text{C}-\text{H}\cdots\text{C}\equiv\text{C}_{\text{centroid}} = 130^\circ$ and $\eta^2\text{-C}\equiv\text{CD}_2\cdots\text{C}_{\text{isophthal.}} = 3.48(3) \text{ \AA}$]. $\text{C}_2\text{D}_2\text{-4}$ occupies the hydrophobic triangular window of cage **B** with strong binding to an oxygen of the $\{\text{Cu}_2\}$ paddlewheel [$\text{DC}_2\text{-D}\cdots\text{O}_{\text{paddlewheel}} = 2.25(5) \text{ \AA}$, $\angle\text{C}-\text{D}\cdots\text{O} = 150^\circ$] as well as to adjacent isophthalate rings [$\text{D}_2\text{C}_2\cdots\text{H}-\text{C}_{\text{isophthal.}} = 2.45(2) \text{ \AA}$, $\angle\text{C}-\text{H}\cdots\text{C} = 120^\circ$ and $\eta^2\text{-C}\equiv\text{CD}_2\cdots\text{H}-\text{C}_{\text{isophthal.}} = 2.89(1) \text{ \AA}$, $\angle\text{C}-\text{H}\cdots\text{C}\equiv\text{C}_{\text{centroid}} = 124^\circ$].

Three CO_2 binding domains, 1–3, were found in MFM-127 (Fig. 3). $\text{CO}_2\text{-1}$ is situated in an equivalent site to $\text{C}_2\text{D}_2\text{-4}$ with a close contact to a carboxylate $\text{C}^{\delta+}$ [$\text{O}_{\text{CO}_2}\cdots\text{C}_{\text{COO}} = 3.96(6) \text{ \AA}$, $\angle\text{C}-\text{O}\cdots\text{C} = 100^\circ$] as well as side-on interactions with isophthalate rings [$\text{O}_{\text{CO}_2}\cdots\text{H}-\text{C}_{\text{isophthal.}} = 2.27(1) \text{ \AA}$, $\angle\text{C}-\text{H}\cdots\text{O} = 135^\circ$ and $\text{C}_{\text{CO}_2}\cdots\text{H}-\text{C}_{\text{isophthal.}} = 2.95(7) \text{ \AA}$, $\angle\text{C}-\text{H}\cdots\text{C} = 114^\circ$]. $\text{CO}_2\text{-2}$ occupies the corresponding site to $\text{CO}_2\text{-1}$ in MFM-126. As for MFM-126, cooperative binding is exhibited between crystallographically equivalent $\text{CO}_2\text{-2}$ molecules in cage **B** of MFM-127 [$\text{C}_{\text{CO}_2}\cdots\text{O}_{\text{CO}_2} = 3.04(2) \text{ \AA}$, $\angle\text{C}=\text{O}\cdots\text{C} = 128^\circ$]. Crucially, these results reveal the key role of the amide groups on selective adsorption of CO_2 in MFM-126 as well as confirming that alkyne moieties do not act as effective adsorption sites for CO_2 in MFM-127.

NPD data for CD_4 -loaded MFM-127 exposed only two noteworthy sites of interaction. Both sites correspond directly to CD_4

sites **1** and **2** observed in MFM-126, respectively. In MFM-127, $\text{CD}_4\text{-1}$ resides in the hydrophobic triangular windows of cage **B** [$\text{D}_4\text{C}\cdots\text{H}-\text{C}_{\text{isophthal.}} = 2.93(1) \text{ \AA}$, $\angle\text{C}-\text{H}\cdots\text{C} = 120^\circ$], further showing that this pocket provides an optimal environment for CD_4 . $\text{CD}_4\text{-2}$ is found in the center of cage **B** and interacts weakly to the framework alkyne [$\text{D}_4\text{C}\cdots\eta^2\text{-C}\equiv\text{C}_{\text{framework}} = 4.39(1) \text{ \AA}$] revealing that the alkyne moiety has little effect on CD_4 adsorption.

Thus, significantly, the NPD study reveals that it is a combination of cooperative binding as well as the amide functionality that leads to enhanced interaction of CO_2 with MFM-126. On the other hand, NPD data reveal that alkyne-functionalized MFM-127 exhibits much weaker interaction with CO_2 , but that the alkyne moieties play critical roles in acetylene binding, with over half of the adsorbed acetylene molecules exhibiting interactions ($<4 \text{ \AA}$) to alkyne moieties in the pore of MFM-127. This represents the first example of direct visualization of acetylene binding to an alkyne moiety in porous materials.

Inelastic neutron scattering (INS) studies of CO_2 -loaded MFM-126

To gain further understanding into CO_2 binding in MFM-126, inelastic neutron scattering (INS) was measured as a function of CO_2 -loading (Fig. S18†). The INS spectrum for the bare MOF exhibits numerous vibrational modes which have been assigned by comparison with a DFT-calculated INS spectrum (Fig. S17†). Interestingly, unlike previously reported with isostructural MFM-136,²⁷ MFM-126 exhibits significant interactions with guest CO_2 . Upon CO_2 loading, there is a shift in the peak at 82.8 meV (assigned to the in-plane bending of the amide group) to 84.1 meV signifying an interaction between adsorbed CO_2 and the amide group, such as those visualized in the structure model of CO_2 -loaded MFM-126. This result contrasts with that of MFM-136 (ref. 27) and confirms the importance of cooperativity between functional groups (in this case amide) and the pore geometries on guest binding.

In situ synchrotron FT-IR microspectroscopy of MFM-126

To investigate further the nature of host-guest interactions, *in situ* synchrotron FT-IR microspectroscopy was conducted on a single crystal of activated MFM-126. FTIR spectra were collected as a function of CO_2 -loading by increasing the partial pressure of CO_2 in N_2 from 0 to 1.0 bar (pp CO_2 ; Fig. 4). The $\nu_1 + \nu_3$ (3695 cm^{-1}) and $2\nu_2 + \nu_3$ (3590 cm^{-1}) combination bands of CO_2 (Fig. 4a) were used to monitor CO_2 sorption as the fundamental antisymmetric stretch at $\sim 2348 \text{ cm}^{-1}$ saturates at pp CO_2 above 0.2 bar. These combination bands are red-shifted from their free values (3714 and 3612 cm^{-1} , respectively) at all partial pressures of CO_2 loading and increase in intensity as a function of pp CO_2 , confirming that CO_2 interacts strongly with the framework (Fig. 4a). Activated MFM-126 has a $\nu(\text{N}-\text{H})$ stretch at 3306 cm^{-1} and upon increasing pp CO_2 from 0 to 1 bar, this band shifts by 36 cm^{-1} to 3270 cm^{-1} . In addition, the observed shift of the amide $\nu(\text{C}=\text{O})$ stretching band by 10 cm^{-1} (from 1684 cm^{-1} in the absence of CO_2 to 1674 cm^{-1} at 1.0 bar pp CO_2 ; Fig. 4c) further indicates CO_2 adsorption directed by the



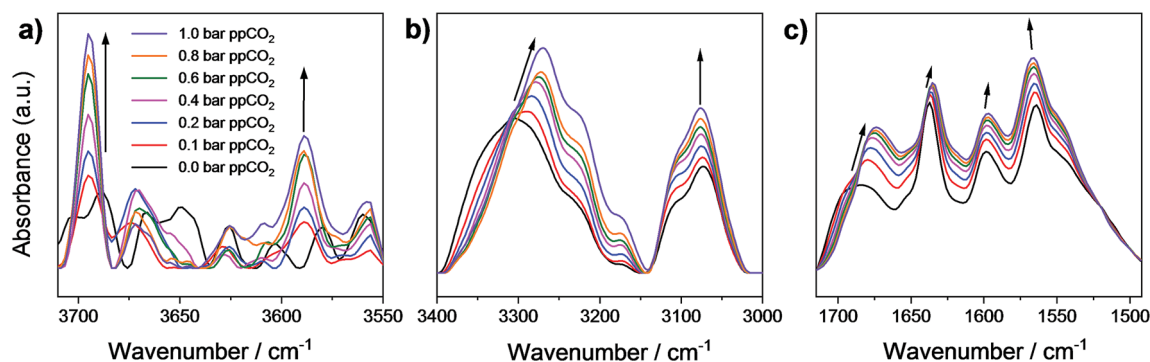


Fig. 4 Micro-FTIR spectra of a single crystal of MFM-126 as a function of CO₂-loading incrementally from 0 to 1 bar ppCO₂ focusing on (a) the CO₂ combination bands at 3695 and 3590 cm⁻¹; (b) ν(N-H) stretching band shifting from 3306 cm⁻¹ at 0 bar ppCO₂ (in N₂) to 3270 cm⁻¹ at 1 bar ppCO₂; (c) the ν(N-H) bending mode shifting from 1684 cm⁻¹ at 0 bar ppCO₂ (in N₂) to 1674 cm⁻¹ at 1 bar ppCO₂. All spectra were recorded referenced to the blank cell as a function of CO₂ loading to remove contributions of free gaseous CO₂.

amide group in MFM-126. Overall, these observations are highly consistent with the NPD and INS results.

Conclusions

A series of isorecticular (3,6)-connected pyrimidyl isophthalate Cu(II)-based MOFs, of rare *eea* topology have been synthesized and characterized. Although MFM-136 exhibits the highest CO₂ uptake at 20 bar, it was found that upon shortening the linker unit to form MFM-126, selectivity of CO₂ uptake at 1 bar increases dramatically. The relative ease of synthesis of MFM-126 coupled with its high selectivity for binary mixtures of CO₂/CH₄ and CO₂/N₂, position this MOF as a viable candidate for CO₂ separations. Furthermore, the CO₂ separation performance of MFM-126 has been confirmed by dynamic breakthrough experiments. NPD data of CO₂-loaded MFM-126 has revealed that cooperative binding of CO₂ (at position CO₂-1) and binding to amide groups that decorate cage **B** both have strong effects on the observed selective CO₂ binding. The *in situ* spectroscopic studies using INS and FTIR also establish that adsorbed CO₂ interacts strongly with the amide groups of MFM-126, with significant shifts of amide (C=O and N-H) vibrational bands on CO₂ loading. Replacing the amide-group in MFM-126 with an alkyne-group to give MFM-127 leads to a decrease in both CO₂ and CH₄ uptake capacities relative to MFM-126, but affords a 68% greater C₂H₂ capacity than MFM-126. NPD experiments reveal for the first time that acetylene interacts directly with alkyne moieties of MFM-127 in the pore. Notably, over 50% of the acetylene observed within MFM-127 displays strong interactions (<4 Å) with the alkyne functionality of the framework. The understanding gained in this study provides further insights into the development of materials showing improved gas binding *via* specific interaction to ligand sites within the MOF.

Conflicts of interest

The authors declare no competing financial interests.

Acknowledgements

We thank EPSRC (EP/I011870, EP/K038869, EP/P001386), ERC (AdG 742041), the Royal Society (6866) and the Universities of Manchester and Nottingham for funding. O. B. thanks Niger Delta University and TETFund, Nigeria for a PhD scholarship. We are especially grateful to STFC/ISIS Neutron Facility for access to the Beamlines TOSCA and WISH and to Diamond Light Source for access to the Beamlines I19 and B22. The computing resources were made available through the VirtuES and ICEMAN projects, funded by Laboratory Directed Research and Development program at ORNL.

References

- 1 A. Corma, H. García and F. X. Llabrés i Xamena, *Chem. Rev.*, 2010, **110**, 4606–4655.
- 2 L.-B. Sun, X.-Q. Liu and H.-C. Zhou, *Chem. Soc. Rev.*, 2015, **44**, 5092–5147.
- 3 T. L. Easun, F. Moreau, Y. Yan, S. Yang and M. Schröder, *Chem. Soc. Rev.*, 2017, **46**, 239–274.
- 4 X. Han, H. G. W. Godfrey, L. Briggs, A. J. Davies, Y. Cheng, L. L. Daemen, A. M. Sheveleva, F. Tuna, E. J. L. McInnes, J. Sun, C. Drathen, M. W. George, A. J. Ramirez-Cuesta, K. M. Thomas, S. Yang and M. Schröder, *Nat. Mater.*, 2018, **17**, 691–696.
- 5 Y. Yan, D. I. Kolokolov, I. da Silva, A. G. Stepanov, A. J. Blake, A. Dailly, P. Manuel, C. C. Tang, S. Yang and M. Schröder, *J. Am. Chem. Soc.*, 2017, **139**, 13349–13360.
- 6 A. H. Chughtai, N. Ahmad, H. A. Younus, A. Laypkov and F. Verpoort, *Chem. Soc. Rev.*, 2015, **44**, 6804–6849.
- 7 J. An, S. J. Geib and N. L. Rosi, *J. Am. Chem. Soc.*, 2010, **132**, 38–39.
- 8 C. A. Trickett, A. Helal, B. A. Al-Maythallony, Z. H. Yamani, K. E. Cordova and O. M. Yaghi, *Nat. Rev. Mater.*, 2017, **2**, 17045.
- 9 S. Chand, A. Pal and M. C. Das, *Chem.-Eur. J.*, 2018, **24**, 5982–5986.
- 10 J. W. Zhang, M. C. Hu, S. N. Li, Y. C. Jiang, P. Qu and Q. G. Zhai, *Chem. Commun.*, 2018, **54**, 2012–2015.



- 11 F. Chen, Y. Wang, D. Bai, M. He, X. Gao and Y. He, *J. Mater. Chem. A*, 2018, **6**, 3471–3478.
- 12 D.-M. Chen, C.-X. Sun, N.-N. Zhang, H.-H. Si, C.-S. Liu and M. Du, *Inorg. Chem.*, 2018, **57**, 2883–2889.
- 13 G. T. Rochelle, *Science*, 2009, **325**, 1652–1654.
- 14 J.-R. Li, J. Sculley and H.-C. Zhou, *Chem. Rev.*, 2012, **112**, 869–932.
- 15 A. Pal, S. Chand, S. M. Elahi and M. C. Das, *Dalton Trans.*, 2017, **46**, 15280–15286.
- 16 H. H. Storch, *Ind. Eng. Chem.*, 1934, **26**, 56–60.
- 17 R.-B. Lin, L. Li, H. Wu, H. Arman, B. Li, R.-G. Lin, W. Zhou and B. Chen, *J. Am. Chem. Soc.*, 2017, **139**, 8022–8028.
- 18 W. Yang, A. J. Davies, X. Lin, M. Suyetin, R. Matsuda, A. J. Blake, C. Wilson, W. Lewis, J. E. Parker, C. C. Tang, M. W. George, P. Hubberstey, S. Kitagawa, H. Sakamoto, E. Bichoutskaia, N. R. Champness, S. Yang and M. Schröder, *Chem. Sci.*, 2012, **3**, 2993–2999.
- 19 H.-M. Wen, H. Wang, B. Li, Y. Cui, H. Wang, G. Qian and B. Chen, *Inorg. Chem.*, 2016, **55**, 7214–7218.
- 20 D. Britt, H. Furukawa, B. Wang, T. G. Glover and O. M. Yaghi, *Proc. Natl. Acad. Sci. U. S. A.*, 2009, **106**, 20637–20640.
- 21 Q. Yan, Y. Lin, C. Kong and L. Chen, *Chem. Commun.*, 2013, **49**, 6873–6875.
- 22 B. Zheng, Z. Yang, J. Bai, Y. Li and S. Li, *Chem. Commun.*, 2012, **48**, 7025–7027.
- 23 S. Yang, J. Sun, A. J. Ramirez-Cuesta, S. K. Callear, W. I. F. David, D. P. Anderson, R. Newby, A. J. Blake, J. E. Parker, C. C. Tang and M. Schröder, *Nat. Chem.*, 2012, **4**, 887–894.
- 24 S. Yang, A. J. Ramirez-Cuesta, R. Newby, V. Garcia-Sakai, P. Manuel, S. K. Callear, S. I. Campbell, C. C. Tang and M. Schröder, *Nat. Chem.*, 2015, **7**, 121–129.
- 25 L. Du, Z. Lu, K. Zheng, J. Wang, X. Zheng, Y. Pan, X. You and J. Bai, *J. Am. Chem. Soc.*, 2013, **135**, 562–565.
- 26 R. Vaidhyanathan, S. S. Iremonger, G. K. H. Shimizu, P. G. Boyd, S. Alavi and T. K. Woo, *Science*, 2010, **330**, 650–653.
- 27 O. Benson, I. Da Silva, S. P. Argent, R. Cabot, M. Savage, H. G. W. Godfrey, Y. Yan, S. F. Parker, P. Manuel, M. J. Lennox, T. Mitra, T. L. Easun, W. Lewis, A. J. Blake, E. Besley, S. Yang and M. Schröder, *J. Am. Chem. Soc.*, 2016, **138**, 14828–14831.
- 28 N. L. Rosi, J. Eckert, M. Eddaoudi, D. T. Vodak, J. Kim, M. O’Keeffe and O. M. Yaghi, *Science*, 2003, **300**, 1127–1129.
- 29 S. J. Garibay and S. M. Cohen, *Chem. Commun.*, 2010, **46**, 7700–7702.
- 30 H. Deng, S. Grunder, K. E. Cordova, C. Valente, H. Furukawa, M. Hmadeh, F. Gandara, A. C. Whalley, Z. Liu, S. Asahina, H. Kazumori, M. O’Keeffe, O. Terasaki, J. F. Stoddart and O. M. Yaghi, *Science*, 2012, **336**, 1018–1023.
- 31 Z. Lu, J. Zhang, J. Duan, L. Du and C. Hang, *J. Mater. Chem. A*, 2017, **5**, 17287–17292.
- 32 O. K. Farha, I. Eryazici, N. C. Jeong, B. G. Hauser, C. E. Wilmer, A. A. Sarjeant, R. Q. Snurr, S. T. Nguyen, A. Ö. Yazaydin and J. T. Hupp, *J. Am. Chem. Soc.*, 2012, **134**, 15016–15021.
- 33 N. F. Cessford, N. A. Seaton and T. Düren, *Ind. Eng. Chem. Res.*, 2012, **51**, 4911–4921.
- 34 Y. Hu, S. Xiang, W. Zhang, Z. Zhang, L. Wang, J. Bai and B. Chen, *Chem. Commun.*, 2009, **45**, 7551–7553.
- 35 B. Chen, N. W. Ockwig, A. R. Millward, D. S. Contreras and O. M. Yaghi, *Angew. Chem., Int. Ed.*, 2005, **44**, 4745–4749.
- 36 X. Lin, J. Jia, X. B. Zhao, K. M. Thomas, A. J. Blake, N. R. Champness, P. Hubberstey and M. Schröder, *Angew. Chem., Int. Ed.*, 2006, **45**, 7358–7364.

