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MoS₂ nanosheet incorporated α -Fe₂O₃/ZnO nanocomposite with enhanced photocatalytic dye degradation and hydrogen production ability†

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We have synthesized MoS₂ incorporated α -Fe₂O₃/ZnO nanocomposites by adapting a facile hydrothermal synthesis process. The effect of incorporating ultrasonically exfoliated few-layer MoS₂ nanosheets on the solar-light driven photocatalytic performance of α -Fe₂O₃/ZnO photocatalyst nanocomposites has been demonstrated. Structural, morphological and optical characteristics of the as-synthesized nanomaterials are comprehensively investigated and analyzed by performing Rietveld refinement of powder X-ray diffraction patterns, field emission scanning electron microscopy and UV-visible spectroscopy, respectively. The photoluminescence spectra of the as-prepared nanocomposites elucidate that the recombination of photogenerated electron–hole pairs is highly suppressed due to incorporation of MoS₂ nanosheets. Notably, the ultrasonicated MoS₂ incorporated α -Fe₂O₃/ZnO nanocomposite manifests 91% and 83% efficiency in degradation of rhodamine B dye and antibiotic ciprofloxacin respectively under solar illumination. Active species trapping experiments reveal that the hydroxyl (·OH) radicals play a significant role in RhB degradation. Likewise the dye degradation efficiency, the amount of hydrogen produced by this nanocomposite *via* photocatalytic water splitting is also considerably higher as compared to both non-ultrasonicated MoS₂ incorporated α -Fe₂O₃/ZnO and α -Fe₂O₃/ZnO nanocomposites as well as Degussa P25 titania nanoparticles. This indicates the promising potential of the incorporation of ultrasonicated MoS₂ with α -Fe₂O₃/ZnO nanocomposites for the generation of carbon-free hydrogen by water splitting. The substantial increase in the photocatalytic efficiency of α -Fe₂O₃/ZnO after incorporation of ultrasonicated MoS₂ can be attributed to its favorable band structure, large surface to volume ratio, effective segregation and migration of photogenerated electron–hole pairs at the interface of heterojunctions and the plethora of exposed active edge sites provided by the few-layer MoS₂ nanosheets.

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1 Introduction

In the last few decades, the unprecedented growth in population and industry has tremendously elevated the necessity of finding a clean and sustainable alternative source of energy to overcome the global energy crisis and increasing environmental pollution. Hydrogen (H₂) can be considered as an ideal source of energy for the future owing to its high energy capacity along with non-polluting nature.^{1,2} Notably, semiconductor photocatalysed splitting of water by solar illumination is an effective

and economical route to produce H₂ as an inexhaustible source of carbon-free fuel for numerous applications.^{3,4} Since 1972, after the first successful use of n-type titania (TiO₂) photo-electrodes in solar H₂ production *via* water splitting,⁵ metal oxide semiconductors have been extensively investigated and employed both in solar H₂ generation and photocatalytic decontamination and in the removal of toxic dyes from wastewater owing to their satisfactory photocatalytic efficiency and good stability under the electrochemical reaction conditions.^{6,7} Moreover, in recent years, semiconductor photocatalysts are being employed to mineralize and eliminate second and third generation antibiotics such as tetracycline chloride, ciprofloxacin and their by-products from wastewater.^{8–10} Especially, nanoparticles of metal oxide semiconductors have achieved significant research interest in the last few decades compared to their bulk counterparts because of their high surface to volume ratio which enables a plenty of photons to be incident upon the surface.^{6,11} In the past decades, titanium dioxide (TiO₂) (bandgap \sim 3.2 eV) has been studied the most as

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a semiconductor photocatalyst because of its low production cost, high chemical stability and non-toxic attributes.^{12,13} Recently, zinc oxide (ZnO), an n-type semiconductor oxide photocatalyst has emerged as a suitable alternative to TiO₂ since it possesses almost same bandgap energy (\sim 3.37 eV)¹⁴ but exhibits higher quantum and absorption efficiency when compared to TiO₂.¹⁵ Nevertheless, the fast recombination rate of photoexcited electron–hole (e–h) pairs in ZnO along with photo-corrosion degrades its photocatalytic efficiency significantly.¹⁶ To enhance the solar energy conversion efficiency by absorbing a broader range of the solar spectrum, hematite (α -Fe₂O₃), a visible light responsive photocatalyst¹⁷ with moderate band gap (\sim 2.2 eV), has emerged as a promising candidate to form nanocomposite with ZnO nanoparticles. In some previous investigations,^{18,19} a photocatalyst combining the excellence of both ZnO and α -Fe₂O₃ has been achieved through the development of α -Fe₂O₃/ZnO nanocomposite which has manifested good performance in photocatalytic decomposition of different organic dyes. However, the solar H₂ production potential of this nanocomposite is mostly unexplored and the extant synthesis procedures are quite robust involving multiple steps, high temperature, especial solvents and additives.

Notably, graphene-like two-dimensional (2D) layered transition metal dichalcogenides (TMDs) with enormous surface area and agile carrier transport properties can further improve the photocatalytic abilities of metal oxides having optimal optical absorption potential.^{20–24} Molybdenum disulfide (MoS₂) is a typically used TMDs which has been reported to introduce incredible improvements in optical absorption, electron mobility, surface morphology and adsorption capabilities of pristine metal oxide nanoparticles by forming heterojunctions with them.^{25,26} Especially, MoS₂ nanosheets can effectively catalyze the H₂ evolution reaction by utilizing their sulphur edge sites while keeping basal planes catalytically inert.²⁷ In a recent investigation,²⁸ we have adapted a facile ultrasound driven exfoliation technique to obtain MoS₂ nanosheets from bulk MoS₂ powder. We have observed that few-layer MoS₂ nanosheets exfoliated from bulk MoS₂ powder *via* such a facile one-step ultrasonication technique²⁹ demonstrate better photocatalytic performance than their bulk counterpart.

Inspired by such improvements, in this investigation, we have synthesized α -Fe₂O₃/ZnO nanocomposite and afterward, incorporated it with both bulk MoS₂ powder and few-layer MoS₂ nanosheets distinctly. For brevity, non-ultrasonicated MoS₂ and ultrasonicated MoS₂ will be referred to as NMS and UMS respectively, from now on. We have adapted a step-wise low temperature hydrothermal synthesis technique³⁰ to obtain α -Fe₂O₃ nanoparticles as well as nanocomposites *i.e.* α -Fe₂O₃/ZnO, NMS incorporated α -Fe₂O₃/ZnO and UMS incorporated α -Fe₂O₃/ZnO. Structural, morphological and optical characterizations of all the as-synthesized samples have been carried out and their photocatalytic performance has been compared to that of commercially available titania, TiO₂ (Degussa P25) nanoparticles in photodegradation of a representative organic pollutant rhodamine B (RhB) dye by solar irradiation. We have observed that UMS incorporated α -Fe₂O₃/ZnO nanocomposite exhibits considerably better performance in dye degradation

than α -Fe₂O₃ nanoparticles, α -Fe₂O₃/ZnO, NMS incorporated α -Fe₂O₃/ZnO nanocomposites and titania nanoparticles. This nanocomposite has also photodegraded the antibiotic ciprofloxacin in aqueous medium with high efficiency. Further, the UMS incorporated α -Fe₂O₃/ZnO nanocomposite has demonstrated higher potential to generate H₂ *via* solar-driven water-splitting in comparison with other synthesized nanocomposites under scrutiny. We have also performed active species trapping experiments to evaluate the role of different reactive species in the degradation of RhB dye. Finally, a model has been proposed to critically assess the underlying mechanism behind the superior photocatalytic performance of UMS incorporated α -Fe₂O₃/ZnO nanocomposite.

2 Experimental details

2.1 Materials

Analytical grade zinc nitrate hexahydrate [Zn(NO₃)₂·6H₂O], ferric nitrate nonahydrate [Fe(NO₃)₃·9H₂O], sodium hydroxide [NaOH], ammonium hydroxide [NH₄OH] and molybdenum disulfide [MoS₂] powder (Sigma-Aldrich, 99% trace metals basis) were used as chemical reagents.

2.2 Sample preparation

Synthesis of α -Fe₂O₃ nanoparticles. The α -Fe₂O₃ nanoparticles were prepared using the hydrothermal reaction technique^{30,31} by dissolving 1 mmol of Fe(NO₃)₃·9H₂O in 50 mL deionized (DI) water. By adding NH₄OH, the pH value of the solution was maintained at 9. After stirring rigorously for 50 minutes using a magnetic stirrer, the solution was placed in a Teflon-lined autoclave³⁰ and put in a programmable oven for 10 hours. The temperature of the oven was maintained at 120 °C. The obtained suspension was then cooled down to room temperature (RT) followed by centrifugation using centrifuge machine (Hettich Universal 320) and consequent rinsing with DI water and ethanol and afterward, was dried at 80 °C for 10 hours.³⁰ The synthesized α -Fe₂O₃ nanoparticles were used to prepare α -Fe₂O₃/ZnO nanocomposite by adapting a similar hydrothermal technique.

Synthesis of α -Fe₂O₃/ZnO nanocomposite. To synthesize α -Fe₂O₃/ZnO nanocomposite, 1 mmol of α -Fe₂O₃ nanoparticles and 1 mmol of Zn(NO₃)₂·6H₂O were dissolved in 50 mL of 5 M NaOH solution in stoichiometric proportions. After stirring properly for 4 hours, the mixture was transferred to the autoclave and heated for 2 hours at 190 °C. Then the sample was cooled down to RT naturally. Afterward, centrifugation was applied and consequent rinsing was performed using DI water and ethanol. After this purification process, the solution was dried for 10 hours at 80 °C to get the desired α -Fe₂O₃/ZnO nanocomposite.

Synthesis of few-layer MoS₂ nanosheets. Ultrathin MoS₂ nanosheets were synthesized from bulk MoS₂ powder (referred earlier as NMS) using ultrasonication assisted exfoliation technique.²⁸ Initially, 300 mg of MoS₂ powder was dissolved in 60 mL of isopropanol and ultrasonicated for 90 minutes.^{28,29} The temperature in the ultrasonic bath was maintained at 50 ±



2 °C by adjusting the temperature control knob. After sonication, the resultant solution was allowed to sediment for about 4 hours to remove the residue powder and micron size thick sheets and to obtain MoS₂ nanosheets by collecting 35% of the mass as supernatant. Prior to further characterization, these supernatant (ultrasonicated MoS₂ nanosheets, referred earlier as UMS) were dried at 80 °C for 5 hours.

Synthesis of NMS incorporated α -Fe₂O₃/ZnO and UMS incorporated α -Fe₂O₃/ZnO nanocomposites. Hydrothermal synthesis technique was also used for incorporating NMS and UMS individually with α -Fe₂O₃/ZnO nanocomposites. To prepare NMS incorporated α -Fe₂O₃/ZnO nanocomposite, 1 mmol of α -Fe₂O₃/ZnO and 0.05 mmol of non-ultrasonicated MoS₂ powder were added into 50 mL of DI water and stirred rigorously for 4 hours. The mixture was then placed in an autoclave and heated for 14 hours at 150 °C. Finally, the desired samples were collected following the aforementioned steps *i.e.* centrifugation, rinsing and drying for 10 hours at 80 °C. Following this same technique, ultrasonicated few-layer MoS₂ nanosheets *i.e.* UMS incorporated α -Fe₂O₃/ZnO nanocomposite was synthesized.

2.3 Characterization techniques

The crystallographic parameters of the synthesized samples were determined by analyzing their powder X-ray diffraction (XRD) patterns obtained using a diffractometer (PANalytical Empyrean) with Cu X-ray source. The experimentally obtained powder XRD patterns was further refined by Rietveld method using the FULLPROF package.³² The surface morphology of the samples was observed by performing field emission scanning electron microscopy (FESEM) imaging using a scanning electron microscope (XL30SFEG; Philips, Netherlands). UV-visible diffuse reflectance spectra (DRS) of the synthesized materials were recorded for wavelengths ranging from 200 nm to 800 nm at RT using a UV-vis spectrophotometer (UV-2600, Shimadzu).³³ Steady-state photoluminescence (PL) spectroscopy of the synthesized materials was carried out at RT using Spectro Fluorophotometer (RF-6000, Shimadzu) to qualitatively evaluate their rate of photogenerated e-h recombination during photocatalysis. PL measurements were carried out with an excitation wavelength of 371 nm for all the samples except NMS incorporated α -Fe₂O₃/ZnO nanocomposite for which the applied excitation wavelength was 347.5 nm. The excitation wavelengths were chosen by studying the absorbance spectra of the respective material determined by UV-vis spectrophotometer.

2.4 Photocatalytic degradation and hydrogen evolution

The photocatalytic performance of all the synthesized nanomaterials was evaluated by photodegradation of RhB^{30,34} dye in aqueous medium. A mercury-xenon (Hamamatsu L8288, 500 W) lamp having irradiance value of 100 mW cm⁻² was used as a solar simulator. The amount of RhB dye in the suspension, before and after illumination in presence of photocatalyst nanomaterials, was measured by determining the intensity of the absorbance peak of the mixture by UV-visible spectroscopy. The absorbance measurements were carried out for 4 hours at 1

hour interval. For testing the reusability of the photocatalyst, the photodegradation experiment was run 4 times. After each run, the residue nanomaterial was extracted from the solution and again used as photocatalyst in the next photodegradation cycle.⁶ The photocatalytic performance of UMS incorporated α -Fe₂O₃/ZnO in degradation of ciprofloxacin was also evaluated by following the same experimental technique.¹⁰

A photocatalytic hydrogen production experiment was conducted using typical method as was described in details in our previous investigation.³⁰ For this test, 40 mg sample was dispersed into 60 mL DI water and then irradiated by the 500 W mercury-xenon lamp.

2.5 Active species trapping experiment

In order to investigate the photocatalytic mechanism of RhB dye degradation, active species trapping experiments were performed with different scavengers *i.e.* isopropanol (1 mmol L⁻¹), acrylamide (1 mmol L⁻¹), ethylenediaminetetraacetic acid disodium salt dihydrate (EDTA-2Na, 1 mmol L⁻¹) and potassium dichromate (K₂Cr₂O₇, 1 mmol L⁻¹).^{35,36} Notably, the experimental conditions were kept same for all photocatalytic degradation tests.

3 Results and discussions

3.1 Structural characterization

Crystal structure analysis. Fig. 1 presents the observed and Rietveld refined powder XRD patterns of all the synthesized nanomaterials. Fig. 1(a) demonstrates the XRD pattern of the α -Fe₂O₃ nanoparticles. All diffraction peaks of Fe₂O₃ can be indexed to alpha (α)-phase having a trigonal crystal structure and $R\bar{3}c$ space group. The XRD pattern of the α -Fe₂O₃/ZnO nanocomposite is shown in Fig. 1(b) in which along with the diffraction peaks of α -Fe₂O₃, some additional peaks can be observed. These peaks are matched with the peaks of hexagonal wurtzite ZnO crystal structure which belongs to $P6_3mc$ space group. The obtained results are congruent with the ones reported in the literature.³⁷⁻³⁹ Rietveld refinement of the powder XRD patterns reveals high phase purity of both α -Fe₂O₃ nanoparticles and α -Fe₂O₃/ZnO nanocomposite since no undesired peaks were detected. The success achieved in the suppression of impurity phases might be credited to the meticulous regulation of the stoichiometry of precursor materials.

The Rietveld refined XRD spectra of NMS incorporated α -Fe₂O₃/ZnO and UMS incorporated α -Fe₂O₃/ZnO nanocomposites are depicted in Fig. 1(c) and (d) respectively. Three separate phases corresponding to α -Fe₂O₃, ZnO and MoS₂ are observed in their XRD patterns which reveals the successful formation of pristine MoS₂ incorporated α -Fe₂O₃/ZnO nanocomposite without any impurity phases. The crystallographic phase corresponding to MoS₂ is found to have a hexagonal crystal structure belonging to $P6_3/mmc$ space group which is consistent with the results of previous investigation.²² Notably, as shown in Fig. 1(c), the peak introduced by MoS₂ at around $2\theta = 14.48^\circ$ has significantly high intensity in the NMS incorporated sample which is substantially lower in few-layer MoS₂ *i.e.*



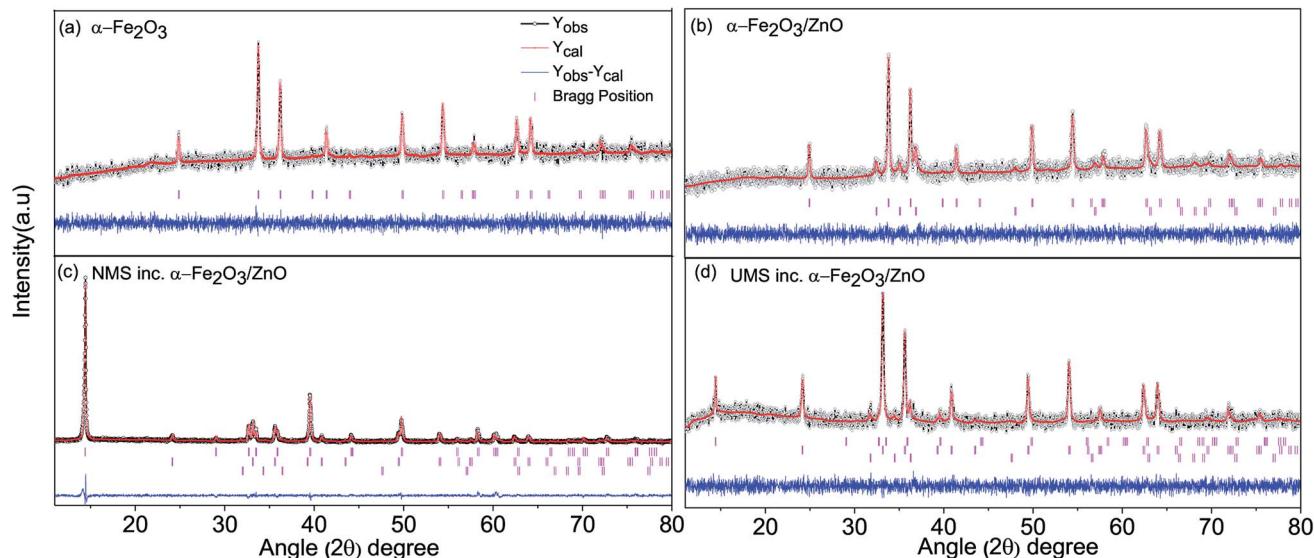


Fig. 1 Rietveld refined powder XRD patterns of (a) α -Fe₂O₃ nanoparticles, (b) α -Fe₂O₃/ZnO, (c) non-ultrasonicated MoS₂ incorporated α -Fe₂O₃/ZnO and (d) ultrasonicated MoS₂ incorporated α -Fe₂O₃/ZnO nanocomposites. Due to space constraint, 'incorporated' has been abbreviated to 'inc.' in legends of all the figures and tables.

UMS incorporated α -Fe₂O₃/ZnO nanocomposite (Fig. 1(d)). Recently, in a separate investigation,²⁸ we have observed a noticeable decrement in the peak intensity of XRD patterns of MoS₂ after ultrasonication. Hence, the powder XRD patterns indicate the successful incorporation of few-layer MoS₂ nanosheets with α -Fe₂O₃/ZnO nanocomposite.

The crystallographic phases, space groups, lattice parameters and unit cell volume (V) for all the constituents of the synthesized nanomaterials are listed in Table 1. Noticeably, the lattice parameters and volume of the phases corresponding to α -Fe₂O₃ and ZnO remained almost constant in every step of the synthesis process. Hence, it can be inferred that no distortion was introduced in the crystal structure of the individual phase during synthesis. The atomic coordinates, crystallographic phases (in wt%) of the constituent of synthesized nanomaterials and reliability (R) factors (R_{wp} , R_p , R_{Exp} and χ^2) are tabulated in ESI Table S1.† The diminutive values of R factors indicate excellent fit with the defined crystal structures.

Surface morphology analysis. The FESEM image of synthesized α -Fe₂O₃ nanoparticles (average size 50 to 60 nm) is shown

in Fig. 2(a) which demonstrates that the nanoparticles are well dispersed with less agglomeration and their surface is satisfactorily homogeneous and non-porous. It is also clearly visible that the nanoparticles are uniform in shape and size and all the nanoparticles have trigonal crystal structure³⁷ (marked in the figure) which justifies the results obtained by Rietveld refined XRD patterns. Fig. 2(b) depicts the FESEM image of α -Fe₂O₃/ZnO nanocomposite. It is apparent that some hexagonal shaped nanoparticles have been integrated with as-synthesized trigonal α -Fe₂O₃ nanoparticles which implies the successful formation of α -Fe₂O₃/ZnO nanocomposite since ZnO has a hexagonal crystal structure.³⁸ For getting an insight into the morphological change of MoS₂ due to ultrasonication, FESEM images of non-ultrasonicated and ultrasonicated MoS₂ have been investigated and depicted in Fig. 2(c) and (d), respectively. It is observed from Fig. 2(c) that non-ultrasonicated MoS₂ powder having comparatively rough surface morphology is composed of numerous stacked MoS₂ layers which cannot be detected individually. Interestingly, these stacked layers were successfully delaminated to well-dispersed ultra-thin few-layer MoS₂

Table 1 Crystallographic phases, space groups, lattice parameters (a , b and c) and unit cell volume (V) for all the constituents of the synthesized nanomaterials obtained from Rietveld refined powder XRD spectra. Numbers in the parentheses indicate errors on the last significant digit

Sample	Constituent	Crystallographic phase (space group)	$a = b$ (Å)	c (Å)	V (Å ³)
α -Fe ₂ O ₃	α -Fe ₂ O ₃	Trigonal ($R\bar{3}c$)	5.045(0)	13.779(1)	303.75(3)
α -Fe ₂ O ₃ /ZnO	α -Fe ₂ O ₃	Trigonal ($R\bar{3}c$)	5.036(0)	13.754(1)	302.14(3)
	ZnO	Hexagonal ($P6_3mc$)	3.250(0)	5.207(2)	47.67(3)
NMS inc. α -Fe ₂ O ₃ /ZnO	α -Fe ₂ O ₃	Trigonal ($R\bar{3}c$)	5.036(0)	13.767(1)	302.93(3)
	ZnO	Hexagonal ($P6_3mc$)	3.229(1)	5.227(1)	47.20(4)
	MoS ₂	Hexagonal ($P6_3/mmc$)	3.164(0)	12.305(0)	106.70(0)
UMS inc. α -Fe ₂ O ₃ /ZnO	α -Fe ₂ O ₃	Trigonal ($R\bar{3}c$)	5.040(0)	13.764(0)	302.88(2)
	ZnO	Hexagonal ($P6_3mc$)	3.252(0)	5.203(3)	47.68(3)
	MoS ₂	Hexagonal ($P6_3/mmc$)	3.161(1)	12.300(3)	106.46(6)



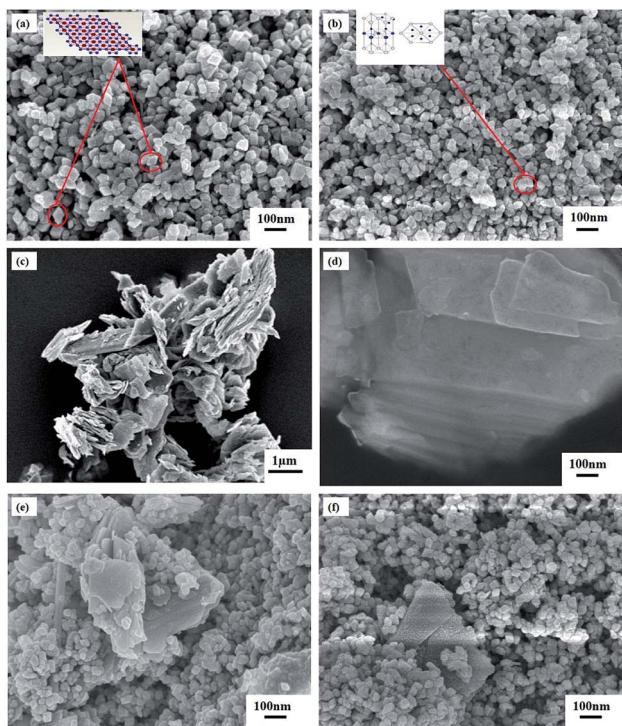


Fig. 2 FESEM images of (a) α -Fe₂O₃ nanoparticles, (b) α -Fe₂O₃/ZnO nanocomposite, (c) non-ultrasonicated bulk MoS₂ powder, (d) ultrasonicated few-layer MoS₂ nanosheets, (e) non-ultrasonicated MoS₂ incorporated α -Fe₂O₃/ZnO and (f) ultrasonicated MoS₂ incorporated α -Fe₂O₃/ZnO nanocomposites.

nanosheets due to ultrasonication which is evident from Fig. 2(d). To provide a better view, additional FESEM images of ultrasonicated MoS₂ sample with different magnifications have been presented in ESI Fig. S1(a) and (b).[†] From Fig. S1(a),[†] using ImageJ software, the thickness of an individual MoS₂ nanosheet has been estimated to be in the range of 8–15 nm which corresponds to 11–21 monolayers of MoS₂²⁸ since single-layer MoS₂ has a typical thickness of \sim 0.7 nm.⁴⁰ Furthermore, the FESEM image of synthesized NMS incorporated α -Fe₂O₃/ZnO nanocomposite is presented in Fig. 2(e). It can be seen that α -Fe₂O₃/ZnO nanocomposite have been scattered over the surface of bulk MoS₂ powder albeit they are not well distributed and are agglomerated to a large extent. However, as demonstrated in Fig. 2(f), in UMS incorporated α -Fe₂O₃/ZnO sample, the particles of α -Fe₂O₃/ZnO are more uniformly distributed over the ultra-thin MoS₂ nanosheets as compared to the NMS incorporated sample. The small particle size of α -Fe₂O₃/ZnO nanocomposite along with the nanoscopic thickness of ultrasonicated 2D MoS₂ sheets is anticipated to provide an increased surface area to volume ratio which is one of the essential features of an efficient photocatalyst.³⁰

3.2 Optical characterization

Bandgap analysis. Diffuse reflectance spectra (DRS) of all the as-synthesized samples were acquired from UV-vis spectrophotometric measurements to determine their optical bandgaps

which denote the minimum energy of photons which can be absorbed by the respective material to produce e-h pairs *via* interband transition. It should be noted that both ZnO and α -Fe₂O₃ nanoparticles are widely regarded as direct bandgap materials.^{14,17} Hence, here we have applied Kubelka-Munk function and Tauc's law⁴¹ for direct bandgap material to produce Tauc plots from the DRS spectra of corresponding materials. Fig. 3 displays the Tauc plots *i.e.* $[F(R) \times h\nu]^2$ vs. $h\nu$ -curves of all as-synthesized nanomaterials which were used for optical bandgap calculation. The intercept of the tangent to the curve to the energy axis provided the optical bandgaps of corresponding materials. The direct optical bandgaps of α -Fe₂O₃ nanoparticles, α -Fe₂O₃/ZnO, NMS incorporated α -Fe₂O₃/ZnO and UMS incorporated α -Fe₂O₃/ZnO nanocomposites are found to be 1.87 eV, 1.90 eV, 1.83 eV and 1.96 eV respectively (Table 2). The energy bandgaps calculated for α -Fe₂O₃ nanoparticles and α -Fe₂O₃/ZnO nanocomposite are consistent with the values reported by Mishra and Chun (2015)¹⁷ and Wu *et al.* (2012).¹⁹ Notably, an earlier investigation¹⁴ on ZnO nanoparticles reported their optical bandgap to be 3.37 eV. Hence, the bandgaps demonstrated by all the synthesized nanocomposites are significantly smaller compared to ZnO nanoparticles. This redshift in the bandgap indicates that prepared nanocomposites are competent for absorbing photons in the visible range of the solar spectrum more effectively which may enhance their promising potential in photocatalytic applications. The slightly enhanced optical bandgap of UMS incorporated α -Fe₂O₃/ZnO nanocomposite as compared to NMS incorporated nanocomposite can be associated with the little increase in the bandgap of MoS₂ due to ultrasonication as reported in our previous investigation.²⁸ However, the observed variation among the bandgaps of the as-synthesized nanomaterials is very small. A previous investigation³⁰ demonstrated that the ascendancy of a certain nanomaterial as photocatalyst cannot be imputed only to its bandgap. Instead, some other determinants *e.g.* phase purity, surface homogeneity, superior potential

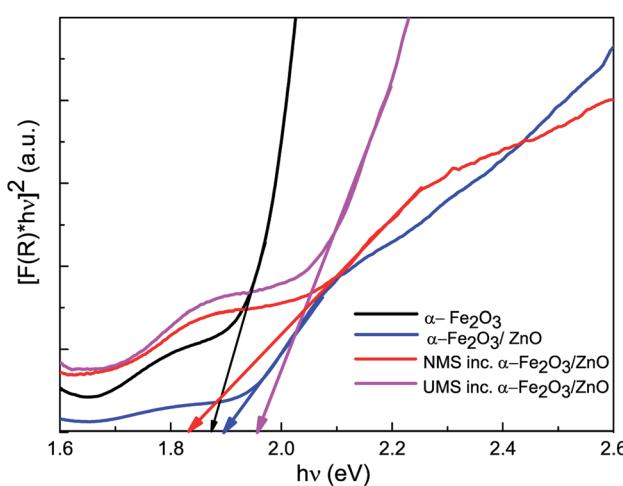


Fig. 3 Bandgap estimation of synthesized α -Fe₂O₃ nanoparticles, α -Fe₂O₃/ZnO, non-ultrasonicated MoS₂ incorporated α -Fe₂O₃/ZnO and ultrasonicated MoS₂ incorporated α -Fe₂O₃/ZnO nanocomposites.



Table 2 Calculated bandgap energy (E_g) of as-synthesized nano-materials, photodegradation rate constant (k) and degradation (%) of RhB dye after 4 h of solar illumination in the presence of the samples

Sample	E_g (eV)	Rate constant (k) $\times 10^{-1}$ (min $^{-1}$)	Degradation (%) after 4 h
α -Fe ₂ O ₃	1.87	0.984	37
α -Fe ₂ O ₃ /ZnO	1.90	1.66	50
NMS inc. α -Fe ₂ O ₃ /ZnO	1.83	1.94	53
UMS inc. α -Fe ₂ O ₃ /ZnO	1.96	6.25	91

to enhance charge-carriers' lifetime, favorable band structure etc. contribute to the performance of a good photocatalyst.

Analysis of steady-state photoluminescence (PL) spectra. To observe e-h recombination phenomena, we have performed steady-state PL spectroscopy of all the as-synthesized samples and presented their PL spectra in ESI Fig. S2.† As can be seen from the figure, the intensity of the PL signal of α -Fe₂O₃ nanoparticles is substantially higher than all other synthesized nanomaterials. Among all samples, UMS incorporated α -Fe₂O₃/ZnO nanocomposite has demonstrated the smallest PL peak intensity which indicates that a very small number of photo-excited e-h have been radiatively recombined in this semiconductor.³ Therefore, we may anticipate that the photoexcited charge carriers of α -Fe₂O₃/ZnO nanocomposite were successfully separated and transported by ultra-thin MoS₂ nanosheets and thus the recombination process was effectively inhibited. The outcome of our investigation is also in good agreement with previous studies where 2D MoS₂ nanosheets have been reported to efficiently quench photoluminescence of different semiconductor photocatalysts.^{42,43} However, the radiative recombination rate of NMS incorporated α -Fe₂O₃/ZnO is observed to be a bit higher than that of UMS incorporated nanocomposite, since the photoexcited e-h pairs cannot be satisfactorily segregated and transported by the stacked layers of bulk MoS₂ powder as compared to few-layer MoS₂ nanosheets.²⁸

3.3 Photocatalytic degradation activity and stability

The photocatalytic performance of α -Fe₂O₃ nanoparticles, α -Fe₂O₃/ZnO, NMS incorporated α -Fe₂O₃/ZnO and UMS incorporated α -Fe₂O₃/ZnO nanocomposites have been investigated by a colorant decomposition test of RhB dye under solar light irradiation.³⁰ For comparison, titania nanoparticles had been also subject to the photodegradation analysis of RhB dye in identical experimental setup. As a representative of synthesized nanomaterials, in Fig. 4(a) we have presented RhB dye's absorbance spectra measured in dark and light conditions at 1 hour interval when UMS incorporated α -Fe₂O₃/ZnO photocatalyst was employed. The intensity of absorbance peak of RhB decreases gradually with time which indicates the decomposition of RhB dye. Since RhB dye is highly resistive to decomposition, this result implies that UMS incorporated α -Fe₂O₃/ZnO nanocomposite has demonstrated high photocatalytic efficiency in RhB degradation. For determining the dye degradation efficiency of the as-prepared samples both in dark conditions and after 4 hours of irradiation, the maximum

intensity ratio C/C_0 vs. irradiation time in h is plotted in Fig. 4(b), where C and C_0 are obtained from the absorption spectra of RhB under stimulated solar illumination which respectively denote the highest intensity at initial time (0 hour) and at particular times (1–4 hours). For all the photocatalysts, we can observe that the degradation percentage is very small in dark conditions from which we can infer that the degradation of RhB molecules due to adsorption to the catalyst nanomaterials is quite negligible. Moreover, in order to evaluate the self-degradation rate of RhB, a blank test was run under 4 hours of illumination without using any photocatalyst and it was observed that only ~3% of RhB dye was degraded at the end of the experiment. On the other hand, after 4 hours of irradiation, α -Fe₂O₃ nanoparticles, α -Fe₂O₃/ZnO, NMS incorporated α -Fe₂O₃/ZnO and UMS incorporated α -Fe₂O₃/ZnO nanocomposites have respectively degraded 37%, 50%, 53% and 91% of RhB dye. Under identical experimental conditions, 65% degradation is achieved by titania nanoparticles. Since the adsorption process has very trivial contribution to the degradation of RhB as evident from the dark test, it may be inferred that the decolorization of RhB dye by irradiated catalysts has almost completely resulted from photocatalytic degradation. With a view to investigating the rate of degradation quantitatively, the data were fitted adapting a first-order model as expressed by $\ln(C_0/C) = kt$,⁴⁴ where C_0 and C are the same as mentioned earlier, k is the first-order rate constant which can be considered as a basic kinetic parameter of a photocatalyst. The pseudo-first order kinetics fitting data for the photocatalytic decomposition of RhB has been shown in Fig. 4(c). The slope of the straight line provides k . Blank RhB has a very small degradation rate of 4.31×10^{-3} min $^{-1}$. On the other hand, the degradation rates (k) are calculated to be 9.84×10^{-2} , 1.66×10^{-1} , 1.94×10^{-1} , 6.25×10^{-1} and 2.9×10^{-1} min $^{-1}$ for α -Fe₂O₃, α -Fe₂O₃/ZnO, NMS incorporated α -Fe₂O₃/ZnO, UMS incorporated α -Fe₂O₃/ZnO and titania respectively (Table 2). Notably, UMS incorporated α -Fe₂O₃/ZnO nanocomposite exhibits 277%, 222% and 116% greater degradation rate compared to that of α -Fe₂O₃/ZnO, NMS incorporated α -Fe₂O₃/ZnO and titania respectively.

In addition to the superior degradation ability, the photocatalysts need to be stable and recyclable under typical photocatalytic reaction conditions for practical applications. Therefore, UMS incorporated α -Fe₂O₃/ZnO nanocomposite had been subjected to a recyclability test in identical experimental conditions and the reusability graph is presented in Fig. 4(d). As can be observed from the graph, the synthesized nanocomposite shows satisfactory stability with negligible deterioration in the photocatalytic performance after 4 successive photodegradation cycles. Finally, with a view to investigating whether any toxic by-products are produced during this degradation process, we have performed an experiment using UMS incorporated α -Fe₂O₃/ZnO nanocomposite as photocatalyst under solar light illumination having wavelength ranging from 200 to 1100 nm. The result is presented in ESI Fig. S3† which shows that after degradation, no additional peaks have arisen in the absorbance spectra of RhB dye implying the absence of any foreign carbon assisted molecule or benzene radical in the



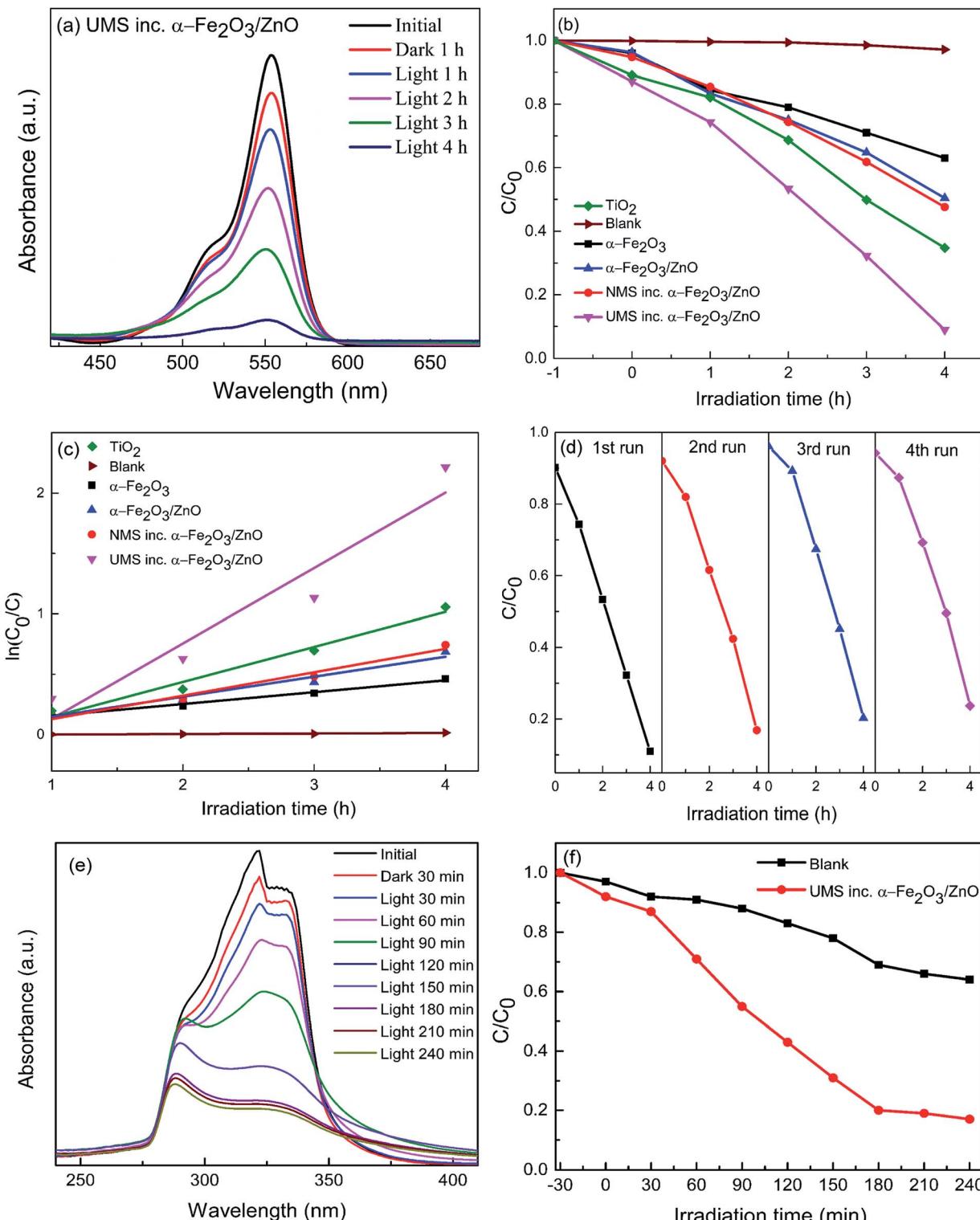


Fig. 4 (a) Time-dependent absorption spectra of RhB solution after being irradiated by solar light in the presence of ultrasonicated MoS_2 incorporated $\alpha\text{-Fe}_2\text{O}_3/\text{ZnO}$ nanocomposite; (b) degradation of RhB as a function of the irradiation time for titania, blank sample, $\alpha\text{-Fe}_2\text{O}_3$, $\alpha\text{-Fe}_2\text{O}_3/\text{ZnO}$, non-ultrasonicated MoS_2 incorporated $\alpha\text{-Fe}_2\text{O}_3/\text{ZnO}$ and ultrasonicated MoS_2 incorporated $\alpha\text{-Fe}_2\text{O}_3/\text{ZnO}$ nanomaterials; (c) pseudo-first order kinetics for RhB; (d) recyclability of ultrasonicated MoS_2 incorporated $\alpha\text{-Fe}_2\text{O}_3/\text{ZnO}$ for 4 successive runs of photocatalytic RhB dye degradation; (e) time-dependent absorption spectra of ciprofloxacin solution after being irradiated by solar light in the presence of ultrasonicated MoS_2 incorporated $\alpha\text{-Fe}_2\text{O}_3/\text{ZnO}$ nanocomposite; (f) degradation of ciprofloxacin as a function of irradiation time for blank sample and ultrasonicated MoS_2 incorporated $\alpha\text{-Fe}_2\text{O}_3/\text{ZnO}$ nanocomposite.

solution. Since UMS incorporated α -Fe₂O₃/ZnO nanocomposite has shown outstanding photocatalytic degradation ability and stability, we anticipate that it can be deemed a potential contender for practical applications such as removal of industrial pollutants from waste water, solar H₂ production *etc.*

Further, we have assessed the photocatalytic efficiency of UMS incorporated α -Fe₂O₃/ZnO nanocomposite in degradation of a colorless antibiotic, ciprofloxacin under solar illumination. The absorbance spectra of ciprofloxacin determined at 30 min interval in dark and light conditions are demonstrated in Fig. 4(e). The decrement of the absorbance peaks at 280 and 320 nm¹⁰ with time indicates the degradation of ciprofloxacin. Notably, after 4 hours of solar light illumination, almost 83% ciprofloxacin was photodegraded by UMS incorporated α -Fe₂O₃/ZnO nanocomposite as can be seen from Fig. 4(f).

3.4 Photocatalytic hydrogen generation

The hydrogen generation test^{30,45} has been performed using the as-synthesized nanomaterials under solar illumination. Under adapted experimental conditions, no H₂ evolution was observed when α -Fe₂O₃ nanoparticles were used as photocatalyst. Fig. 5 demonstrates the amount of H₂ gas ($\mu\text{mol H}_2$ per g catalyst) produced by α -Fe₂O₃/ZnO, NMS incorporated α -Fe₂O₃/ZnO and UMS incorporated α -Fe₂O₃/ZnO nanocomposites during 4 hours of solar light irradiation. For comparison, we have also performed the hydrogen generation test of titania nanoparticles and inserted the result in Fig. 5. The results demonstrate that after 4 hours of illumination, α -Fe₂O₃/ZnO nanocomposite produced \sim 480 $\mu\text{mol H}_2$ per g catalyst which was increased to \sim 522 $\mu\text{mol H}_2$ per g catalyst after the integration of non-ultrasonicated MoS₂. However, the generation of H₂ *via* water splitting was further enhanced by using UMS incorporated α -Fe₂O₃/ZnO nanocomposite as the photocatalyst. Notably, this nanocomposite generated \sim 614 $\mu\text{mol H}_2$ per g catalyst after 4

hours of solar light irradiation. The production rate is appreciably larger in comparison with all other synthesized materials as well as titania nanoparticles. The H₂ evolution performance of UMS incorporated α -Fe₂O₃/ZnO nanocomposite is also better than that of other analogous metal oxide and TMDs based photocatalyst nanocomposites reported in literature using similar experimental conditions.^{46,47} To better understand the H₂ generation performance level of this nanocomposite, we have provided a comparison between our work and previously reported investigations in ESI Table S2.† As shown in the Table S2,† a previous investigation has demonstrated that about 190 $\mu\text{mol H}_2$ can be produced by employing MoS₂/RGO nanohybrid cocatalyst under 4 hours of irradiation.⁴⁸ In another investigation on WO₃/g-C₃N₄ composite, only about 250 $\mu\text{mol H}_2$ per g catalyst evolution has been reported after 4 hours of solar light illumination.⁴⁶ The enhancement in photocatalytic H₂ generation due to incorporation of ultrasonicated MoS₂ in α -Fe₂O₃/ZnO nanocomposite can be associated with the exposed active edge sites and the accelerated charge transfer rate at the interface by few-layer MoS₂ nanosheets.⁴⁹ The outcome of this hydrogen production test suggests that the solar energy conversion efficiency of α -Fe₂O₃/ZnO nanocomposite was improved by incorporating few-layer MoS₂ nanosheets and can be further increased by integrating it with monolayer MoS₂ nanosheets. This is due to the fact that monolayer MoS₂ shows a number of unique photo-electro properties^{40,50} which may be favorable for accelerating H₂ evolution reaction albeit the synthesis of monolayer MoS₂ is quite complex and cumbersome.

3.5 Photocatalytic mechanism

Photocatalytic degradation of RhB dye is an electrochemical process in which photogenerated e-h from the semiconductor photocatalyst are transferred to the electrolyte (RhB solution) and then, they participate in a number of specific redox reactions to decompose the RhB solution. Two of these redox reactions are especially important and needed to be analyzed to explain the rationale behind the improved photocatalytic degradation performance of the synthesized nanocomposites.^{3,30} At first, the photoinduced holes yield hydroxyl 'OH radical species by reacting with the OH⁻ present in the solution ionized from water (redox potential: 2.38 V *vs.* NHE). Eventually, 'OH radicals oxidize and decompose RhB molecules. Meanwhile, the photogenerated electrons react with the surface-bound O₂ along with the O₂ in the aqueous solution (redox potential: -0.16 V *vs.* NHE) to produce superoxide O₂⁻ radicals which afterward cause degradation of RhB molecules.

To evaluate the contribution of different reactive species in the degradation of RhB dye, active species trapping experiments^{35,36} have been conducted with different scavengers in the presence of UMS incorporated α -Fe₂O₃/ZnO nanocomposite and the results are presented in Fig. 6. We have employed isopropanol, acrylamide, ethylenediaminetetraacetic acid disodium salt dihydrate (EDTA-2Na) and potassium dichromate (K₂Cr₂O₇) as scavengers for trapping hydroxyl 'OH radicals, superoxide O₂⁻ radicals, holes (h⁺) and electrons (e⁻)

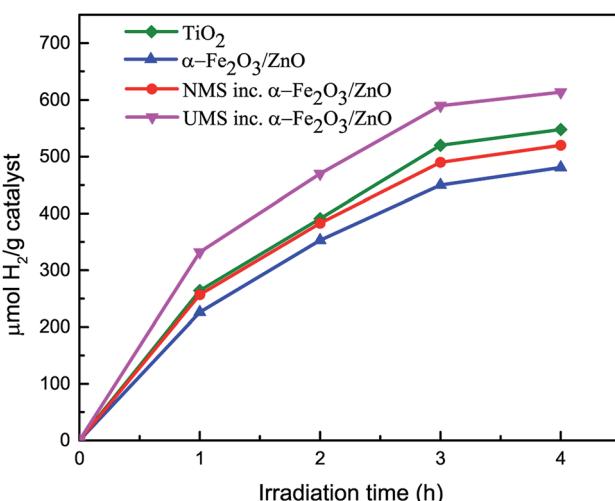


Fig. 5 Volume of H₂ generation *vs.* irradiation time during water splitting for titania nanoparticles, α -Fe₂O₃/ZnO, non-ultrasonicated MoS₂ incorporated α -Fe₂O₃/ZnO and ultrasonicated MoS₂ incorporated α -Fe₂O₃/ZnO nanocomposites.



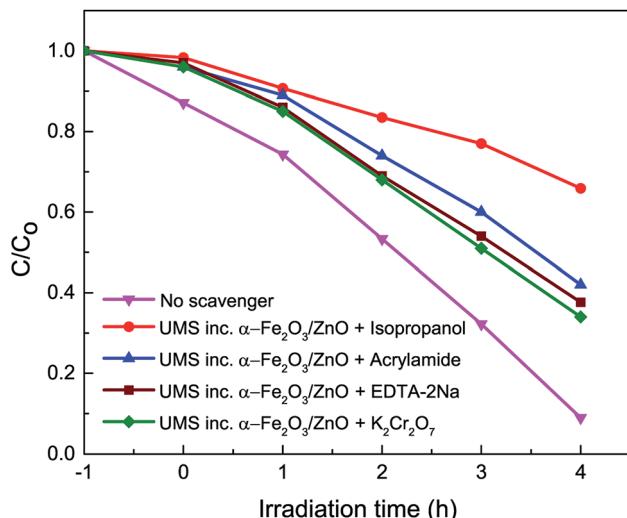


Fig. 6 Effect of the addition of different scavengers on the degradation of RhB in the presence of ultrasonicated MoS₂ incorporated $\alpha\text{-Fe}_2\text{O}_3/\text{ZnO}$ nanocomposite under solar irradiation.

respectively. It is evident from Fig. 6 that the degradation percentage of RhB dye drastically reduced to 34% from 91% due to the addition of isopropanol into the solution of RhB dye and photocatalyst nanocomposite which is much lower as compared to other scavengers. This implies that the hydroxyl 'OH radicals played the major role in the decomposition of RhB dye by the nanocomposite.^{8,35} However, the other scavengers also reduced the decomposition percentage moderately indicating that O_2^- , h^+ and e^- were the subsidiary active species in the photocatalytic process.³⁶

According to the thermodynamic requirements of photocatalytic reaction, to perform degradation, the potential of photoexcited electrons and holes must exceed the redox potential of the reduction and oxidation half-reaction respectively which implies that in the case of RhB dye degradation, a photocatalyst requires to possess a valence band maxima (VBM) > 2.38 V for executing the first redox reaction and conduction band minima (CBM) < -0.16 V to perform the second reaction.³ Therefore, to evaluate the photodegradation performance, the CBM and VBM potentials of the prepared photocatalysts needed to be investigated and so, we have calculated the Fermi energy, CBM and VBM values of $\alpha\text{-Fe}_2\text{O}_3$, ZnO and MoS₂ nanosheets using the absolute electronegativity theory^{51,52} and tabulated the values in ESI Table S3.† An energy band diagram is demonstrated in Fig. 7(a) consolidating the obtained values of band energetics along with the redox potentials of different redox half-reactions of our interest. It can be anticipated from the band structures that $\alpha\text{-Fe}_2\text{O}_3$ nanoparticles and MoS₂ nanosheets cannot take part in any of the redox reactions associated with RhB dye degradation whereas the CBM and VBM positions of ZnO enable it to perform in both the reduction and oxidation half-reactions. This anticipation is strongly validated by our experimental results since the very low dye degradation efficiency (37%) of $\alpha\text{-Fe}_2\text{O}_3$ nanoparticles was substantially enhanced to 50% after coupling ZnO with them.

We can propose a photocatalytic mechanism for the synthesized UMS incorporated $\alpha\text{-Fe}_2\text{O}_3/\text{ZnO}$ nanocomposite by illumination of solar light based on the outcome of our investigation. Fig. 7(b) represents a schematic illustration of the proposed model to give a better understanding of the possible photocatalytic activities. As shown in Fig. 7(b), $\alpha\text{-Fe}_2\text{O}_3$ with a narrow bandgap is easily stimulated by visible light whereas ZnO absorbs only the UV portion of the solar spectrum to photogenerate electrons and holes. It can be observed from the band diagram (Fig. 7(a)) that before forming heterojunction, both the conduction band edge and Fermi level of $\alpha\text{-Fe}_2\text{O}_3$ were lower than those of ZnO. When $\alpha\text{-Fe}_2\text{O}_3$ comes in contact with ZnO, the energy bands of ZnO shift downward whereas the energy bands of $\alpha\text{-Fe}_2\text{O}_3$ shift upward until their Fermi levels reach to an equilibrium and finally, a type-II heterojunction is formed at their interface where the CBM of ZnO is located between the CBM and the VBM of $\alpha\text{-Fe}_2\text{O}_3$.^{19,53} Therefore, under solar illumination, the electrons in the VB of $\alpha\text{-Fe}_2\text{O}_3$ will be photoexcited and transferred to the CB leaving behind holes in the VB and afterward, the photogenerated electrons will tend to move to the CB of ZnO because of the driving by the decreased potential energy. Hence, the photoinduced e-h pairs will be more effectively separated at the heterojunction which will mitigate the e-h recombination and increase the electron mobility and interfacial charge transfer rate. Consequently, the photocatalytic performance of $\alpha\text{-Fe}_2\text{O}_3/\text{ZnO}$ nanocomposite will be ameliorated owing to its enhanced charge separation and sunlight utilization efficiency.

Further, incorporation of ultrasonicated MoS₂ nanosheets increased the dye degradation capability of $\alpha\text{-Fe}_2\text{O}_3/\text{ZnO}$ nanocomposite. According to our proposed model (Fig. 7(b)), in UMS incorporated $\alpha\text{-Fe}_2\text{O}_3/\text{ZnO}$ nanocomposite, a heterojunction is formed between the MoS₂ and $\alpha\text{-Fe}_2\text{O}_3$ which promotes the photocatalytic activity by improving e-h pair separation. During the formation of this heterojunction the electrons from $\alpha\text{-Fe}_2\text{O}_3$ diffuse to the MoS₂ at the interface because of the carrier density gradient.⁵⁴ Similarly, holes from MoS₂ diffuse to the $\alpha\text{-Fe}_2\text{O}_3$. Hence, positive and negative charge regions are created in the $\alpha\text{-Fe}_2\text{O}_3$ and MoS₂ sides respectively resulting in the formation of an internal electrostatic field and band bending at interface. After forming the heterojunction and aligning the Fermi levels, both the CBM and VBM of $\alpha\text{-Fe}_2\text{O}_3$ lie under those of MoS₂. Hence, under solar illumination, due to the electrostatic field, the photogenerated electrons in the CB of MoS₂ will transfer to that of $\alpha\text{-Fe}_2\text{O}_3$ with higher mobility and similarly, photoexcited holes in the VB of $\alpha\text{-Fe}_2\text{O}_3$ will easily transport to the VB of MoS₂. As a result, the lifetime of the photogenerated e-h will increase which will finally result in improved photocatalytic performance. Along with modified band structure, the superiority in the photocatalytic performance of ultrasonicated MoS₂ incorporated nanocomposite also can be attributed to the enhanced specific surface area associated with its 2D sheet like structure and higher surface to volume ratio due to very few layer ultra-thin structure which facilitate the photocatalytic reactions by providing a good number of unsaturated surface condition sites to the RhB solution.^{25,28} The favorable effect of high surface to volume ratio on the photocatalytic activity can

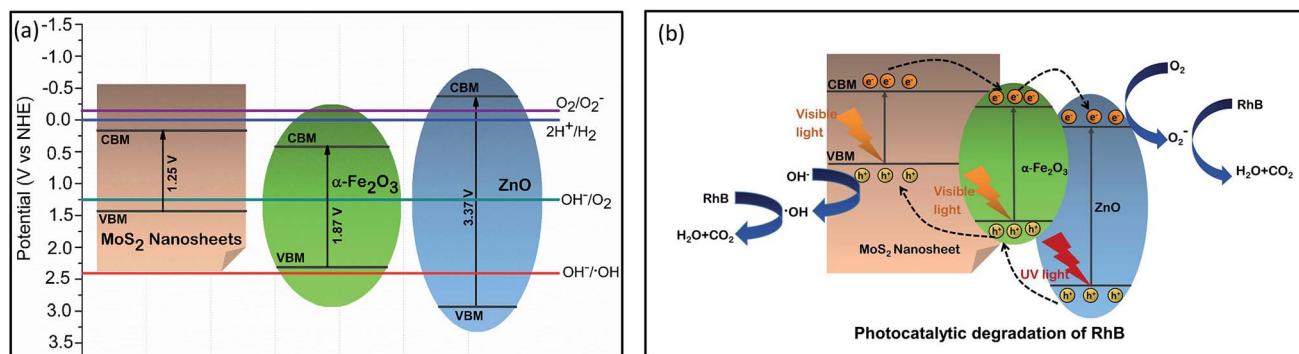


Fig. 7 (a) Energy band diagram of $\alpha\text{-Fe}_2\text{O}_3$, ZnO and MoS₂ nanosheets; (b) schematic of proposed photodegradation mechanism of RhB using ultrasonicated MoS₂ incorporated $\alpha\text{-Fe}_2\text{O}_3$ /ZnO nanocomposite as catalyst under solar illumination.

be further validated by analyzing the photocatalytic performance of NMS incorporated $\alpha\text{-Fe}_2\text{O}_3$ /ZnO which was not as satisfactory as the UMS incorporated $\alpha\text{-Fe}_2\text{O}_3$ /ZnO nanocomposite. This comparatively poor performance can be ascribed to the stacked layers of non-ultrasonicated bulk MoS₂ powder which is not able to provide sufficient catalytically active edge sites and also cannot assist the transportation of charge carriers to the electrolyte system.²⁸

A similar model can be suggested to delineate the underlying mechanism of the superior photocatalytic performance of ultrasonicated MoS₂ incorporated $\alpha\text{-Fe}_2\text{O}_3$ /ZnO nanocomposite in solar H₂ evolution through water splitting. The proposed model is demonstrated in ESI Fig. S4.† The photogenerated holes from the nanocomposite with sufficient potential can react with the water molecules and produce H⁺ along with O₂. Thereafter, the photoexcited electrons evolve H₂ gas from the H⁺.^{3,30} Likewise the enhanced RhB degradation efficiency, it can be anticipated that the higher H₂ evolution rate of UMS incorporated $\alpha\text{-Fe}_2\text{O}_3$ /ZnO nanocomposite is associated with its large specific surface area, efficient separation of photogenerated e-h pairs at the heterojunction, accelerated charge transportation and suppressed charge recombination. In addition to these, the unique structure of MoS₂ nanosheets may play a key role by yielding plenty of exposed edges with coordinately unsaturated S atoms as activity sites which can accept the accumulated photoinduced electrons on MoS₂ for reducing H⁺ to H₂ gas and thus, leads to the discerned high photocatalytic H₂ production potential.⁴⁹

4 Conclusions

We have successfully exfoliated few-layer 2D MoS₂ nanosheets from bulk MoS₂ powder *via* a facile ultrasonication technique and afterward, distinctively incorporated the non-ultrasonicated MoS₂ (NMS) and ultrasonicated MoS₂ (UMS) with $\alpha\text{-Fe}_2\text{O}_3$ /ZnO nanocomposites using low temperature hydrothermal reaction technique in order to investigate their photocatalytic performance. Notably, UMS incorporated $\alpha\text{-Fe}_2\text{O}_3$ /ZnO nanocomposite demonstrated considerably better photocatalytic performance as compared to other synthesized nanomaterials which can be associated with its enhanced

potential to inhibit charge carrier recombination as revealed by photoluminescence spectra, modified band structure and higher surface to volume ratio introduced by the 2D few-layer structure of MoS₂ nanosheets. Besides, the abundant active edge sites with coordinately unsaturated S atoms provided by the enormous specific surface area of MoS₂ nanosheets also might have played significant role in boosting the photocatalytic activities of this nanocomposite. Due to the observed superior photocatalytic performance over commercially available titania nanoparticles in RhB dye and antibiotic ciprofloxacin degradation as well as in hydrogen production *via* disintegration of water under solar illumination, the UMS incorporated $\alpha\text{-Fe}_2\text{O}_3$ /ZnO nanocomposite may be considered as an efficient photocatalyst. Research endeavour to incorporate MoS₂ monolayer with $\alpha\text{-Fe}_2\text{O}_3$ /ZnO nanocomposite is currently underway which may demonstrate further improved performance in numerous photocatalytic applications *e.g.* dye degradation, solar hydrogen production *via* water splitting, solar water disinfection *etc.*

Conflicts of interest

There are no conflicts to declare.

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References

1. J. Shi and L. Guo, *Prog. Nat. Sci.: Mater. Int.*, 2012, **22**, 592–615.
2. B. Wang, C. Li, D. Hirabayashi and K. Suzuki, *Int. J. Hydrogen Energy*, 2010, **35**, 3306–3312.
3. M. A. Basith, R. Ahsan, I. Zarin and M. A. Jalil, *Sci. Rep.*, 2018, **8**, 11090.



4 K. Lu, X. Xin, N. Zhang, Z. Tang and Y. Xu, *J. Mater. Chem. A*, 2018, **6**, 4590–4604.

5 A. Fujishima and K. Honda, *Nature*, 1972, **238**, 37–38.

6 R. Ahsan, A. Mitra, S. Omar, M. Z. Rahman Khan and M. A. Basith, *RSC Adv.*, 2018, **8**, 14258–14267.

7 S. Li, X. Shen, J. Liu and L. Zhang, *Environ. Sci.: Nano*, 2017, **4**, 1155–1167.

8 S. Li, J. Chen, W. Jiang, Y. Liu, Y. Ge and J. Liu, *J. Colloid Interface Sci.*, 2019, **548**, 12–19.

9 S. Li, J. Chen, Y. Liu, K. Xu and J. Liu, *J. Alloys Compd.*, 2019, **781**, 582–588.

10 S. Das, S. Ghosh, A. Misra, A. Tamhankar, A. Mishra, C. Lundborg and S. Tripathy, *Int. J. Environ. Res. Public Health*, 2018, **15**, 2440.

11 S. Li, S. Hu, W. Jiang, Y. Liu, Y. Zhou, J. Liu and Z. Wang, *J. Colloid Interface Sci.*, 2018, **530**, 171–178.

12 M. E. Borges, M. Sierra, E. Cuevas, R. D. Garcia and P. Esparza, *Sol. Energy*, 2016, **135**, 527–535.

13 G. B. Markad, S. Kapoor, S. K. Haram and P. Thakur, *Sol. Energy*, 2017, **144**, 127–133.

14 V. Srikant and D. Clarke, *J. Appl. Phys.*, 1998, **83**, 5447–5451.

15 S. Sakthivel, B. Neppolian, M. Shankar, B. Arabindoo, M. Palanichamy and V. Murugesan, *Sol. Energy Mater. Sol. Cells*, 2003, **77**, 65–82.

16 C. B. Ong, L. Y. Ng and A. W. Mohammad, *Renewable Sustainable Energy Rev.*, 2018, **81**, 536–551.

17 M. Mishra and D. M. Chun, *Appl. Catal., A*, 2015, **498**, 126–141.

18 Y. Liu, L. Yu, Y. Hu, C. Guo, F. Zhang and X. Wen Lou, *Nanoscale*, 2012, **4**, 183–187.

19 W. Wu, S. Zhang, X. Xiao, J. Zhou, F. Ren, L. Sun and C. Jiang, *ACS Appl. Mater. Interfaces*, 2012, **4**, 3602–3609.

20 H. Harada and T. Ueda, *Chem. Phys. Lett.*, 1984, **106**, 229–231.

21 K. Maeda and K. Domen, *J. Phys. Chem. C*, 2007, **111**, 7851–7861.

22 E. Kadantsev and P. Hawrylak, *Solid State Commun.*, 2012, **152**, 909–913.

23 X. Li and H. Zhu, *J. Materomics*, 2015, **1**, 33–44.

24 D. Voiry, M. Salehi, R. Silva, T. Fujita, M. Chen, T. Asefa, V. Shenoy, G. Eda and M. Chhowalla, *Nano Lett.*, 2013, **13**, 6222–6227.

25 Y. Tan, K. Yu, J. Li, H. Fu and Z. Zhu, *J. Appl. Phys.*, 2014, **116**, 064305.

26 M. Sabarinathan, S. Harish, J. Archana, M. Navaneethan, H. Ikeda and Y. Hayakawa, *RSC Adv.*, 2017, **7**, 24754–24763.

27 T. Jaramillo, K. Jorgensen, J. Bonde, J. Nielsen, S. Horch and I. Chorkendorff, *Science*, 2007, **317**, 100–102.

28 S. Das, A. M. Tama, S. Dutta, M. S. Ali and M. A. Basith, *Mater. Res. Express*, 2019, DOI: 10.1088/2053-1591/ab57dd.

29 M. A. Basith, D. T. Ngo, A. Quader, M. A. Rahman, B. L. Sinha, B. Ahmmad, F. Hirose and K. Molhave, *Nanoscale*, 2014, **6**, 14336–14342.

30 M. A. Basith, N. Yesmin and R. Hossain, *RSC Adv.*, 2018, **8**, 29613–29627.

31 F. Wang, X. Qin, Y. Meng, Z. Guo, L. Yang and Y. Ming, *Mater. Sci. Semicond. Process.*, 2013, **16**, 802–806.

32 J. Rodriguez-Carvajal, *Fullprof: A Program for Rietveld Refinement and Pattern Matching Analysis, Abstract of the Satellite Meeting on Powder Diffraction of the XV Congress of the IUCr*, Toulouse, France, 1990, p. 127.

33 L. Amirav and A. Alivisatos, *J. Phys. Chem. Lett.*, 2010, **1**, 1051–1054.

34 S. Li, S. Hu, W. Jiang, J. Zhang, K. Xu and Z. Wang, *J. Colloid Interface Sci.*, 2019, **556**, 335–344.

35 J. Feng, D. Zhang, H. Zhou, M. Pi, X. Wang and S. Chen, *ACS Sustainable Chem. Eng.*, 2018, **6**, 6342–6349.

36 J. Jin, Q. Liang, C. Ding, Z. Li and S. Xu, *J. Alloys Compd.*, 2017, **691**, 763–771.

37 N. Dzade, A. Roldan and N. de Leeuw, *Minerals*, 2014, **4**, 89–115.

38 R. Kumar, O. Al-Dossary, G. Kumar and A. Umar, *Nano-Micro Lett.*, 2014, **7**, 97–120.

39 W. Yan, H. Fan and C. Yang, *Mater. Lett.*, 2011, **65**, 1595–1597.

40 A. Splendiani, L. Sun, Y. Zhang, T. Li, J. Kim, C. Chim, G. Galli and F. Wang, *Nano Lett.*, 2010, **10**, 1271–1275.

41 N. Guy, K. Atacan, E. Karaca and M. Ozacar, *Sol. Energy*, 2018, **166**, 308–316.

42 S. Zhang, F. Tang, J. Liu, W. Che, H. Su, W. Liu, Y. Huang, Y. Jiang, T. Yao, Q. Liu and S. Wei, *Radiat. Phys. Chem.*, 2017, **137**, 104–107.

43 N. Tian, Z. Li, D. Xu, Y. Li, W. Peng, G. Zhang, F. Zhang and X. Fan, *Ind. Eng. Chem. Res.*, 2016, **55**, 8726–8732.

44 S. Ma, R. Li, C. Lv, W. Xu and X. Gou, *J. Hazard. Mater.*, 2011, **192**, 730–740.

45 M. Murdoch, G. Waterhouse, M. Nadeem, J. Metson, M. Keane, R. Howe, J. Llorca and H. Idriss, *Nat. Chem.*, 2011, **3**, 489–492.

46 H. Katsumata, Y. Tachi, T. Suzuki and S. Kaneco, *RSC Adv.*, 2014, **4**, 21405–21409.

47 L. Wu, S. Shi, Q. Li, X. Zhang and X. Cui, *Int. J. Hydrogen Energy*, 2019, **44**, 720–728.

48 S. Min and G. Lu, *J. Phys. Chem. C*, 2012, **116**, 25415–25424.

49 X. Qian, J. Zhang, Z. Guo, S. Liu, J. Liu and J. Lin, *Appl. Surf. Sci.*, 2019, **481**, 795–801.

50 P. Joensen, R. Frindt and S. Morrison, *Mater. Res. Bull.*, 1986, **21**, 457–461.

51 M. A. Butler and D. S. Ginley, *J. Mater. Sci.*, 1980, **15**, 1–19.

52 A. H. Nethercot Jr, *Phys. Rev. Lett.*, 1974, **33**, 1088–1091.

53 Q. Yin, R. Qiao, L. Zhu, Z. Li, M. Li and W. Wu, *Mater. Lett.*, 2014, **135**, 135–138.

54 M. Behtaj Lejbini and P. Sangpour, *Optik*, 2019, **177**, 112–117.

