



Cite this: RSC Adv., 2019, 9, 32010

 Received 6th September 2019
 Accepted 1st October 2019

DOI: 10.1039/c9ra07129f

rsc.li/rsc-advances

Oxadiazole derivatives as bipolar host materials for high-performance blue and green phosphorescent organic light-emitting diodes†

 Yanming Wang, ^{ab} Keke Duan,^a Guoxiang Li^a and Gewen Yu^a

By combining two n-type groups, pyridine and oxadiazole, with one p-type carbazole group, two novel bipolar hosts, namely 2-(3-(9H-carbazol-9-yl)-[1,1'-biphenyl]-3-yl)-5-(pyridin-2-yl)-1,3,4-oxadiazole (**PyOxd-mCz**) and 2-(4'-(9H-carbazol-9-yl)-[1,1'-biphenyl]-3-yl)-5-(pyridin-2-yl)-1,3,4-oxadiazole (**PyOxd-pCz**) have been developed as hosts for blue and green phosphorescent organic light-emitting diodes (PhOLEDs). The two compounds exhibit similar HOMO levels of -5.64 eV for **PyOxd-mCz** and -5.63 eV for **PyOxd-pCz** and the same LUMO level of -2.60 eV. With a more twisted configuration due to *meta* connections, **PyOxd-mCz** possesses a higher triplet energy level ($E_T = 2.77$ eV) and more balanced carrier transport than **PyOxd-pCz** ($E_T = 2.60$ eV). **PyOxd-mCz** hosted devices achieve a peak current efficiency of 39.7 cd A⁻¹ and a maximum EQE of 20.8% with a low turn-on voltage of 3.5 V for FIrpic and 55.2 cd A⁻¹ and 16.4% for Ir(ppy)₃. Apart from the appropriate frontier molecular orbital levels and sufficiently high triplet energy of **PyOxd-mCz**, the more balanced carrier transport plays a key role for excellent device performance.

Introduction

Organic light emitting diodes (OLEDs) have been considered the third generation display technology after cathode-ray tubes (CRTs) and liquid crystal displays (LCDs). After decades of development, OLEDs have realized industrialization and are widely used in display fields, especially in portable electronic products.¹⁻³ The phosphorescent OLEDs (PhOLEDs) with a theoretical maximum internal quantum efficiency of 100% have received great research attention both from the academic world and industrial fields.⁴⁻⁶ In general, the PhOLEDs adopt the host-dopant structure to avoid concentration quenching and triplet-triplet annihilation^{7,8} and the hosts usually take up over 90% in the emitting layer. Therefore, the host materials are as important as the doped emitter in determining the overall performance of the whole device.⁹ Typically, an ideal host should possess appropriate frontier molecular orbital levels matching those of the adjacent layers, sufficiently high triplet energy (E_T) to prevent reverse energy transfer, decent thermal and morphological stabilities to extend the device's operational lifetime, and

balanced charge transport to broaden the recombination zone.¹⁰⁻¹⁷ One of the effective ways to obtain balanced charge fluxes is to utilize bipolar host materials.¹⁸⁻²³ But most bipolar hosts have much higher hole mobilities than electron transporting mobilities even by one or two orders of magnitude, which leads to unbalanced charge transport and narrow recombination zone in emitting layer and eventually leads to low emission efficiency. And in those cases, the ratio of electron-withdrawing groups (n-type) to electron-donating groups (p-type) was usually $1 : 1$. However, there have been a few reports about bipolar hosts which have balanced carrier transportation and high external quantum efficiencies (EQE) by adjusting the ratio of n-type groups to p-type groups from $1 : 1$ to $2 : 1$ or $3 : 1$. Chow group had developed a bipolar host POAPF by adjusting the ratio of n-type groups to p-type groups to $3 : 1$ and a POAPF-based device doped with 7 wt% FIrpic exhibited a very low turn-on voltage (2.5 V) and high electroluminescence efficiencies (20.6% and 36.7 lm W⁻¹).²⁴ Li group had synthesized several bipolar hosts (the ratio of n-type/p-type = $2 : 1$) and blue phosphorescent OLEDs based on them achieved high EQE over 20% .²⁵ In this paper, we have increased the proportion of n-type groups by attaching two n-type groups of pyridine and oxadiazole to one p-type carbazole group. In this way, we expect to increase the electron mobilities of the hosts and get more balanced charge transporting and high efficiency OLED. The position relationship of carbazole and n-type group on the phenylene bridge is varied as *para*- and *meta*-linking mode to tune the molecular conjugated chain length and thus the optoelectronic parameters of the host materials. The blue and green devices based on them exhibit excellent

^aSchool of Chemistry and Chemical Engineering, Inner Mongolia University of Science and Technology, 7 Aldine Street, Baotou 014010, Inner Mongolia, PR China. E-mail: wymygw@163.com

^bSchool of Chemistry, Dalian University of Technology, 2 Linggong Road, Dalian 116024, PR China

† Electronic supplementary information (ESI) available: TGA thermograms, chemical structures and energy level diagram, NMR spectra, absorption spectra and room-temperature photoluminescence. See DOI: 10.1039/c9ra07129f



performances. The **PyOxd-mCz**-hosted blue PhOLEDs exhibited a low turn-on voltage of 3.5 V and a peak current efficiency (η_c) of 39.7 cd A⁻¹, a peak power efficiency (η_p) of 25.0 lm W⁻¹ and a maximum EQE (η_{ext}) of 20.8%. The **PyOxd-mCz**-hosted green PhOLEDs exhibited high efficiencies of 55.2 cd A⁻¹ (29.8 lm W⁻¹ and 16.4%).

Experimental section

Instruments and methods

The ¹H NMR and ¹³C NMR spectra were recorded on Bruker AVANCE 500 MHz spectrophotometer and 126 MHz spectrophotometer. The mass spectra were taken on an HP1100LC/MSD MS spectrometer. The fluorescence and UV-vis absorption spectra measurements were performed on a PerkinElmer LS55 spectrometer and a PerkinElmer Lambda 35 spectrophotometer, respectively. The phosphorescence spectra were measured on an Edinburgh FLS920 spectrometer at 77 K in 2-MeTHF. Thermogravimetric analyses (TGA) and differential scanning calorimetry (DSC) measurements were carried out using a PerkinElmer thermos gravimeter (Model TGA7) and a Netzsch DSC 201 at a heating rate of 10 °C min⁻¹ under a nitrogen atmosphere, respectively. The electrochemical measurements were carried out by using a conventional three-electrode configuration and an electrochemical workstation (BAS100B, USA) at a scan rate of 100 mV s⁻¹. A glass carbon working electrode, a Pt-wire counter electrode, and a saturated calomel electrode (SCE) reference electrode were used. All the electrochemical measurements were carried out at room temperature on samples dissolved in dichloromethane or *N,N*-dimethylformamide (DMF), deoxygenated with argon, and with 0.1 M [nBu₄N]PF₆ as the electrolyte.

The electron density distribution of HOMO/LUMO orbitals were optimized by the density functional theory (DFT) using Beck's three-parameterized Lee-Yang-Parr exchange functional (B3LYP) with 6-31G(d) basis sets. There were no imaginary frequencies for both the optimized structures. All these calculations were performed using Gaussian 03.²⁶

Compounds synthesis

The chemical structures and synthetic routes of compounds **PyOxd-mCz** and **PyOxd-pCz** are depicted in Scheme 1. The intermediate 2-(3-bromophenyl)-5-(pyridin-2-yl)-1,3,4-oxadiazole (**1**) was synthesized according to the literatures.²⁷⁻²⁹ (3-(9H-Carbazol-9-yl) phenyl)boronic acid (**2**) and (4-(9H-carbazol-9-yl) phenyl)boronic acid (**3**) were used as received from commercial sources. Through Suzuki cross-coupling reactions, the target compounds **PyOxd-mCz** and **PyOxd-pCz** were prepared at yields of 68% and 81%, respectively. The two compounds both have good solubility in common organic solvents, such as dichloromethane, tetrahydrofuran and ethyl acetate, so that they could be thoroughly purified by column chromatography and repeated recrystallization to reach a high purity for OLED applications. Their chemical structures were fully characterized by ¹H NMR, ¹³C NMR spectroscopy, mass spectrometry, and elemental analysis.

2-(3'-(9H-Carbazol-9-yl)-[1,1'-biphenyl]-3-yl)-5-(pyridin-2-yl)-1,3,4-oxadiazole (**PyOxd-mCz**)

A mixture of 2-(3-bromophenyl)-5-(pyridin-2-yl)-1,3,4-oxadiazole (1.51 g, 5 mmol), (3-(9H-carbazol-9-yl)phenyl)boronic acid (1.58 g, 5.5 mmol), toluene (20 mL), ethanol (5 mL), aqueous sodium carbonate (2 M, 4 mL), and tetrakis(-triphenylphosphino)palladium(0) (17.4 mg, 0.15 mmol) were refluxed at 80 °C under nitrogen atmosphere overnight. After cooling and filtrating, the filtrate was evaporated under reduced pressure. The residue was purified by column chromatography over silica using petroleum ether/ethyl acetate (5 : 1) as eluent to give pure **PyOxd-mCz** as a white solid (1.6 g, 68% yield).

¹H NMR (500 MHz, CDCl₃) δ = 8.79 (d, *J* = 7.5 Hz, 1H), 8.48 (s, 1H), 8.30 (d, *J* = 8.0 Hz, 1H), 8.22 (d, *J* = 7.5 Hz, 1H), 8.15 (d, *J* = 7.5 Hz, 2H), 7.87 (m, 2H), 7.81 (d, *J* = 7.5 Hz, 1H), 7.77 (d, *J* = 8.0 Hz, 1H), 7.71 (t, *J* = 7.8 Hz, 1H), 7.60 (m, 2H), 7.45 (t, 2H), 7.44–7.40 (m, 2H), 7.29 (t, *J* = 7.8 Hz, 2H).

¹³C NMR (126 MHz, CDCl₃) δ = 165.38, 163.95, 150.25, 143.54, 141.76, 141.22, 140.87, 138.46, 137.23, 130.69, 130.48, 129.75, 126.60, 126.33, 126.04, 125.88, 125.84, 125.82, 124.39, 123.46, 123.31, 120.35, 120.06, 109.75.

TOF-EI-MS (*m/z*): 464.1644 [M]⁺. Anal. calcd for C₃₁H₂₀N₄O: C, 80.15; H, 4.34; N, 12.06. Found: C, 80.18; H, 4.29; N, 12.10.

2-(4'-(9H-Carbazol-9-yl)-[1,1'-biphenyl]-3-yl)-5-(pyridin-2-yl)-1,3,4-oxadiazole (**PyOxd-pCz**)

This compound was synthesized using 2-(3-bromophenyl)-5-(pyridin-2-yl)-1,3,4-oxadiazole and (4-(9H-carbazol-9-yl)phenyl)boronic acid as reagents following the above procedure and obtained white solid (81% yield).

¹H NMR (500 MHz, CDCl₃) δ = 8.84 (d, *J* = 4.5 Hz, 1H), 8.55 (s, 1H), 8.36 (d, *J* = 7.5 Hz, 1H), 8.26 (d, *J* = 8.0 Hz, 1H), 8.16 (d, *J* = 8.0 Hz, 2H), 7.94–7.88 (m, 4H), 7.70–7.66 (m, 3H), 7.51–7.48 (m, 3H), 7.44 (t, *J* = 8.0 Hz, 2H), 7.31 (t, *J* = 8.0 Hz, 2H).

¹³C NMR (126 MHz, CDCl₃) δ = 165.56, 163.96, 150.26, 143.56, 141.39, 140.80, 138.82, 137.62, 137.38, 130.66, 129.78, 128.74, 127.47, 126.44, 126.05, 125.92, 124.37, 123.53, 123.41, 120.37, 120.10, 109.81.

TOF-EI-MS (*m/z*): 464.1648 [M]⁺. Anal. calcd for C₃₁H₂₀N₄O: C, 80.15; H, 4.34; N, 12.06. Found: C, 80.21; H, 4.30; N, 12.12.

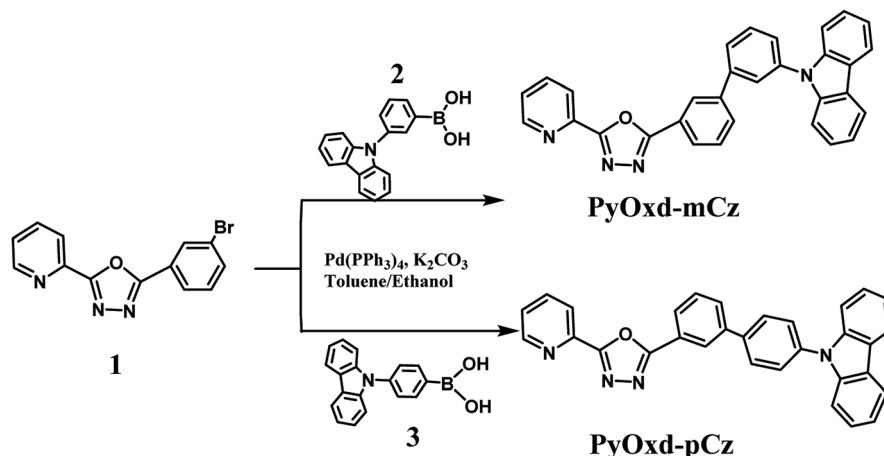
The ¹H NMR and ¹³C NMR spectra of the compounds are shown as Fig. S3–S8 in the ESI.†

Results and discussion

Thermal properties

Thermal properties of the two materials were investigated by thermogravimetric analysis (TGA) and differential scanning calorimetry (DSC) and the data are summarized in Table 1. The two compounds exhibit high thermal decomposition temperatures (*T_d*, corresponding to a 5% weight loss) over 380 °C (Fig. S1†). And as shown in Fig. 1, the glass transition temperatures (*T_g*) for **PyOxd-mCz** and **PyOxd-pCz** are 90 °C and 98 °C, respectively, which are at least 30 °C higher than that of traditional host material of mCP (*T_g* = 60 °C). Such high decomposition temperatures and glass transition temperatures will definitely benefit the performances of the emitting devices.





Scheme 1 Chemical structures and synthetic routes of PyOxd-mCz and PyOxd-pCz.

Photophysical properties

Fig. 2 shows the UV-vis absorption, fluorescence spectra in dilute dichloromethane solutions at 293 K and the LT PL spectra measured in a frozen 2-methyltetrahydrofuran (2-Me-THF) matrix at 77 K of **PyOxd-mCz** and **PyOxd-pCz**. The UV/Vis spectra of both **PyOxd-mCz** and **PyOxd-pCz** include three main absorption at the wavelength of 243 nm, 293 nm and 340 nm, corresponding to the transitions of phenyl, carbazole-centered $n-\pi^*$ transition and $\pi-\pi^*$ transitions of extended π conjunction between the carbazole unit and the central phenyl, respectively.³⁰ The 340 nm attribution can be verified by the fact that the absorption of 340 nm for **PyOxd-pCz** is stronger than that of **PyOxd-mCz**, because the former with less space hinder has longer conjunction line which leads to stronger extended $\pi-\pi^*$ transition and absorption.

Upon optical excitation at the absorption maxima, **PyOxd-mCz** and **PyOxd-pCz** emit purple-blue fluorescence with emission peaks at 435 nm and 441 nm, respectively. The small red shift of 6 nm from **PyOxd-mCz** to **PyOxd-pCz** is understandable since the reduced steric hindrance and less twisted molecular backbone for **PyOxd-pCz** will definitely lead to lower-energy S_1 excited state. The low-temperature photoluminescence (LT PL) spectra were measured in the frozen 2-methyltetrahydrofuran glass at 77 K. The triplet energies are estimated from the highest-energy vibronic sub-band of the phosphorescence spectra as *ca.* 2.77 eV and 2.60 eV for **PyOxd-mCz** and **PyOxd-pCz**, respectively. It is obvious that the E_T of **PyOxd-mCz** is higher than that of **PyOxd-pCz**. This can be explained that the less twisted **PyOxd-pCz** has longer conjugation and thus the

lower triplet excited state compared with more steric **PyOxd-mCz**. Their E_T are both high enough for the typical green emitter $\text{Ir}(\text{ppy})_3$ ($E_T = 2.37$ eV, corresponding to an emission peak at 524 nm),³¹ but **PyOxd-mCz** is more appropriate to be the host for traditional blue emitter FIrpic ($E_T = 2.63$ eV, corresponding to an emission peak at 472 nm).^{32,33}

Electrochemical properties and theoretical calculations

To investigate the electrochemical properties of the two compounds, cyclic voltammetry (CV) had been performed by using a three-electrode cell setup with 0.1 M $n\text{-Bu}_4\text{NPF}_6$ as the supporting electrolyte and saturated calomel electrode (SCE) as

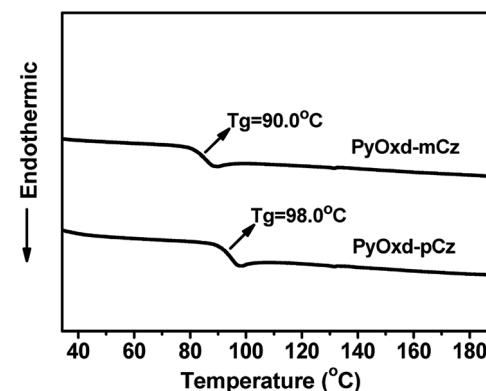


Fig. 1 DSC traces (in the second heating cycle) of **PyOxd-mCz** and **PyOxd-pCz** recorded at a heating rate of $10\text{ }^\circ\text{C min}^{-1}$ under a nitrogen atmosphere.

Table 1 Physical properties of **PyOxd-mCz** and **PyOxd-pCz**

Compound	Abs λ_{max}^a (nm)	PL λ_{max}^a (nm)	E_T^b (eV)	E_g^c (eV)	LUMO (eV)	HOMO (eV)	T_d^d (°C)	T_g (°C)
PyOxd-mCz	243, 293, 340	435	2.77	3.04	-5.64	-2.60	395	90
PyOxd-pCz	243, 293, 340	441	2.60	3.03	-5.63	-2.60	384	98

^a Absorption and fluorescence wavelengths in dilute dichloromethane solutions. ^b Measured in 2-Me-THF at 77 K. ^c Determined using electrochemical potentials. ^d T_d is the thermal decomposition temperature corresponding to 5% weight loss.



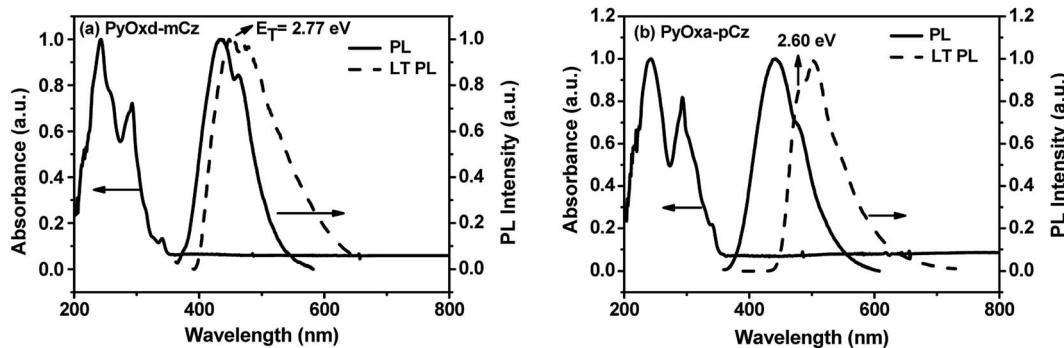


Fig. 2 UV-vis absorption, room-temperature photoluminescence (PL) spectra in dilute dichloromethane solutions, and LT PL spectra in frozen 2-methyltetrahydrofuran matrix at 77 K of PyOxd-mCz (a) and PyOxd-pCz (b).

the internal standard. As shown in Fig. 3, **PyOxd-mCz** and **PyOxd-pCz** both have distinct oxidation and reduction behaviors, which should arise from the electron-donating carbazole unit and electron-withdrawing pyridine-oxadiazole unit, respectively.

The HOMO energies are determined from the onset potential of the first oxidation wave ($E_{\text{ox}}^{\text{onset}}$) according to the equation of $E_{\text{HOMO}} = -e(E_{\text{ox}}^{\text{onset}} + 4.4)$ as -5.64 eV and -5.63 eV for **PyOxd-mCz** and **PyOxd-pCz**, respectively. The HOMO levels of these compounds are close to that of the widely used hole transport material 1,1-bis[(di-4-tolylamino)phenyl]cyclohexane (TAPC, -5.60 eV). The LUMO energies are estimated from the half-wave potentials of the reduction curves to be -2.60 eV for both **PyOxd-mCz** and **PyOxd-pCz**.³⁴ The LUMO energies are close to the adjacent electron transport layer 1,3,5-tri(*m*-pyrid-3-yl-phenyl)benzene (TmPyPb, -2.63 eV) which will lead to less barrier. Comparing with host materials with one n-type group of oxadiazole (LUMO, -2.30 eV),⁷ the two compounds have much lower LUMO energy which indicates that the directly linking pyridine unit to oxadiazole effectively pulls down the LUMO energy of the two hosts. The energy gap (E_g) of **PyOxd-mCz** and **PyOxd-pCz** are determined to be 3.04 eV and 3.03 eV , respectively, by calculating the energy differences between the LUMO and HOMO.

The detailed electrochemical and electronic data of the two molecules are listed in Table 1.

B3LYP/6-31G calculations were performed to investigate the influence of the molecular structure on electronic properties. The HOMO/LUMO distributions for the two compounds are given in Fig. 4. Their HOMOs are mainly contributed on the hole-transporting carbazole moieties and the adjacent benzene, while the LUMOs are mainly localized on the pyridine and oxadiazole moieties and the adjacent benzene. For **PyOxd-mCz**, its HOMO and LUMO are completely separated due to larger steric hindrance of *meta*-substitution; for **PyOxd-pCz**, there is a tiny overlap on the central phenyl between HOMO and LUMO, which means it has longer π -conjugation that leads to lower E_T . The sufficient spatial separation of the HOMO and LUMO indicates the bipolar charge transporting feature of the two molecules.

Charge-transporting properties

To evaluate the charge transporting properties of **PyOxd-mCz**, **PyOxd-pCz**, the hole-only devices with the structure of ITO/HATCN (5 nm)/TAPC (5 nm)/host (100 nm)/TAPC (10 nm)/Al (150 nm) and electron-only devices with the structure of ITO/TmPyPB (10 nm)/host (100 nm)/TmPyPB (5 nm)/LiF (1 nm)/Al (150 nm) were fabricated, respectively. In hole-only devices, 1,4,5,8,9,11-hexaaazatriphenylene hexacarbonitrile (HATCN) was used as a hole-injection layer (HIL), 4,4'-cyclohexylidenebis[*N,N*-bis(*p*-tolyl)aniline] (TAPC) was used as hole-transportation layer (HTL) and TAPC/Al interface was designed to prevent electron injection due to the large electron barrier of 2.3 eV. In electron-only devices, a thin layer of TmPyPB was inserted between ITO anode and the host layer to prevent hole injection due to its deep HOMO level (-6.68 eV). The current density *versus* voltage curves of these single-carrier devices are shown in Fig. 5. Each device exhibited certain current density, implying that both of the compounds possess hole-transporting and electron-transporting abilities, which are the bipolar charge-transporting characters. **PyOxd-pCz** has much higher electron current densities (109.9 mA cm^{-2} at 15 V) and hole current densities (3.3 mA cm^{-2} at 30 V) than those of **PyOxd-mCz** (0.015 mA cm^{-2} at 15 V for electron and 0.08 mA cm^{-2} for hole at 30 V). This can be explained by their different geometrical characteristics. **PyOxd-pCz** has less twisted configuration and more coplanar conformation, which probably

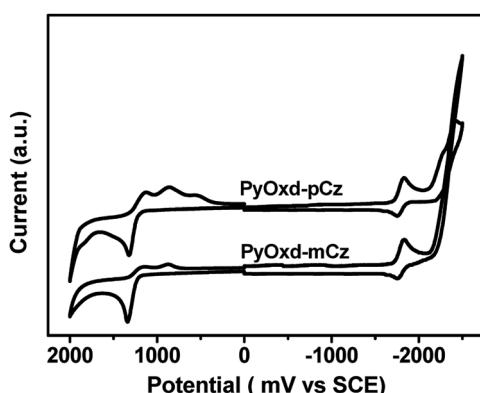


Fig. 3 Cyclic voltammograms of **PyOxd-mCz** and **PyOxd-pCz**. The measurement of oxidation potentials were performed in CH_2Cl_2 and the reduction CV were performed in DMF with 0.1 M $n\text{Bu}_4\text{NPF}_6$ as the supporting electrolyte at a scan rate of 100 mV s^{-1} .



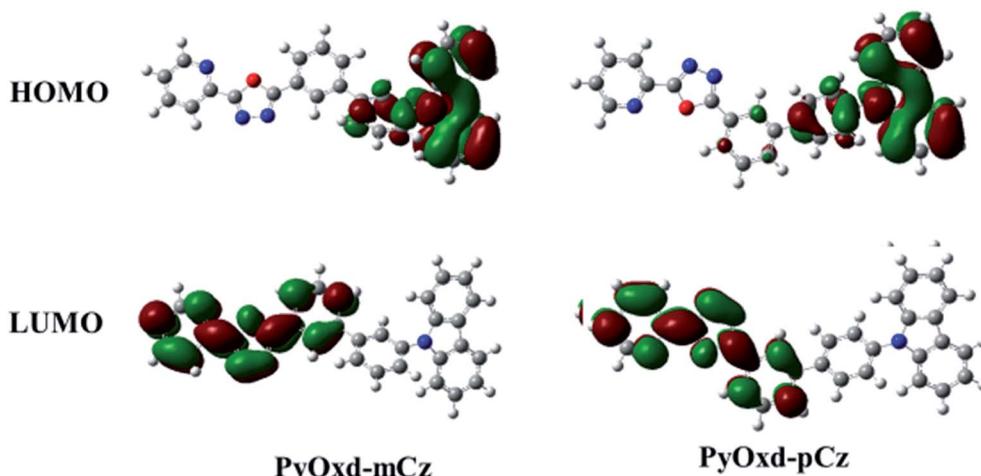


Fig. 4 HOMO and LUMO distributions of PyOxd-mCz and PyOxd-pCz.

leads to intermolecular stacking that are favorable for more rapid charge transportation. But its electron current densities are much higher than hole current densities which will lead to unbalanced charge transportation and inferior device performances. For both **PyOxd-mCz** and **PyOxd-pCz**, their electron current densities are higher than hole current densities indicates that attaching an extra electron-efficient pyridine to the host can effectively facilitate the electron transportation.

EL performance of PHOLEDs

In order to verify the contribution of the double n-type units strategy of the host materials to device performance, we fabricated FIripic-based blue electrophosphorescent (EL) devices B1 and B2 with the configuration of ITO/HATCN (50 nm)/TAPC (20 nm)/host: 6 wt% FIripic (20 nm)/TmPyPB (40 nm)/LiF (1 nm)/Al (200 nm). In the two devices, TAPC as the hole-transporting layer (HTL) and TmPyPB as electron-transporting layer (ETL) and hole-blocking layer (HBL), and HATCN and LiF were used as hole- and electron-injecting layers, respectively. Chemical structures of related materials and energy level diagrams of these devices are shown in Fig. S2.† The current density–voltage–brightness (*J*–*V*–*B*) characteristics and efficiency curves

of these devices are shown in Fig. 6. The **PyOxd-mCz** hosted blue device B1 achieved a better performance with a turn-on voltage V_{on} (to deliver a brightness of 1 $cd\ m^{-2}$) of 3.5 V, a maximum brightness of 21 056 $cd\ m^{-2}$ at 8.5 V, a maximum current efficiency (η_c) of 39.7 $cd\ A^{-1}$, a maximum power efficiency (η_p) of 25.0 $lm\ W^{-1}$ and a maximum external quantum efficiency (η_{ext}) of 20.8% (5 V). The **PyOxd-pCz** hosted device B2 demonstrated a relatively moderate device performance with a maximum brightness of 15 572 $cd\ m^{-2}$ (8.5 V), a maximum η_c of 28.8 $cd\ A^{-1}$ (corresponding to a peak η_{ext} of 14.6% and a maximum η_p of 20.8 $lm\ W^{-1}$ at 3.5 V). The reason why the **PyOxd-mCz**-based device has much better performances can be elucidated from the following aspects. First, **PyOxd-mCz** possesses a high triplet energy E_T (2.77 eV) which is high enough to prevent the adverse back energy transfer from the guest FIripic (2.63 eV) to the host and efficiently confines the excitons in emissive layer (EML) and guarantees the high efficiencies. Secondly, **PyOxd-mCz** has more balanced carrier transporting and thus a wide charge recombination zone and high efficiency.

PyOxd-mCz and **PyOxd-pCz** were also used as hosts to fabricate green phosphorescent devices G1 and G2. The green devices had the same configuration as the above blue ones but with

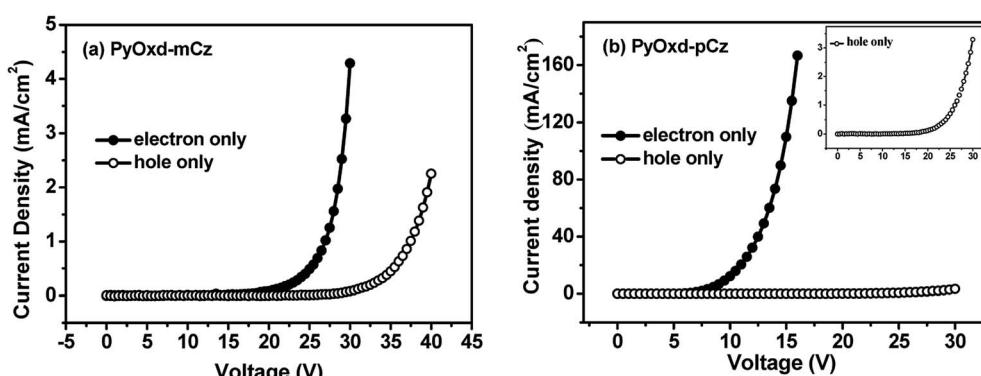


Fig. 5 Current density versus voltage curves of the hole-only and electron-only devices for the compounds PyOxd-mCz (a) and PyOxd-pCz (b).

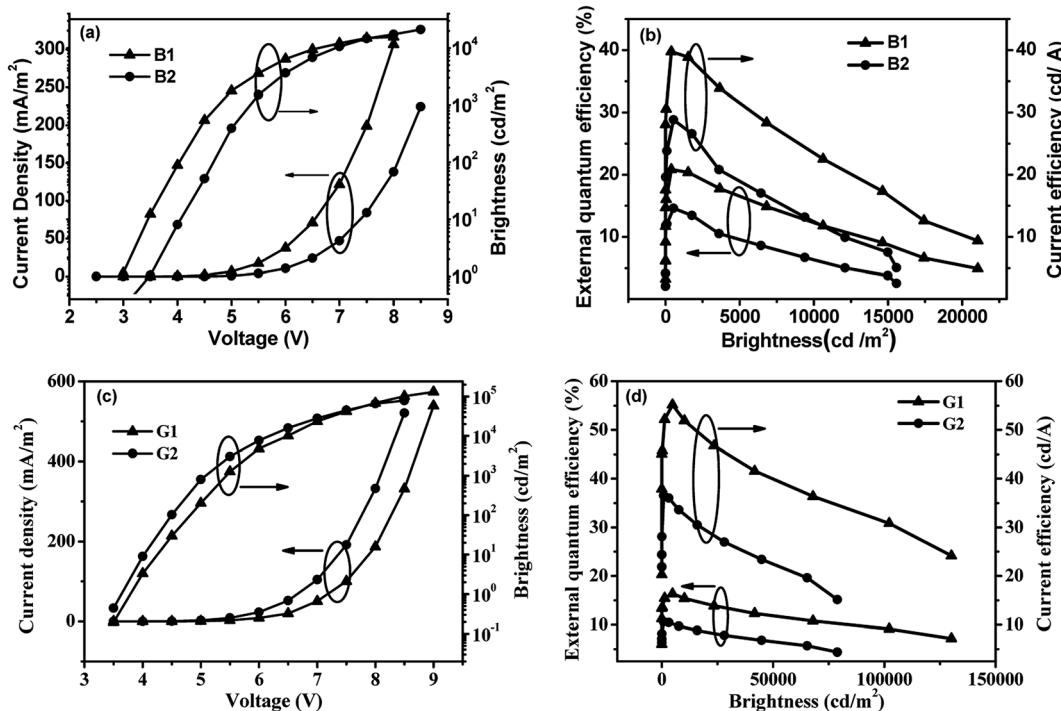


Fig. 6 Current density–voltage–brightness characteristics and efficiency curves for Flrpic-based blue PhOLEDs B1 and B2 (a and b) and for Ir(ppy)₃-based green PhOLEDs G1 and G2 (c and d).

10 wt% Ir(ppy)₃ instead of Flrpic doped in the hosts as the emitting layers. The J-V-L characteristics and efficiency curves were shown in Fig. 6. Both **PyOxd-mCz** and **PyOxd-pCz** hosted devices G1 and G2 showed moderate performances, with a maximum brightness of 130 180 cd m⁻² at 9 V, a maximum current efficiency (η_c) of 55.2 cd A⁻¹ (6 V), a maximum power efficiency (η_p) of 29.8 lm W⁻¹ (5.5 V), and a maximum external quantum efficiency (η_{ext}) of 16.4% (6 V) for device G1, and 78 909 cd m⁻², 36.5 cd A⁻¹, 23.0 lm W⁻¹, and 10.6% for device G2, respectively. Similar to the case in blue devices, **PyOxd-mCz** based device has better performances than those of **PyOxd-pCz** based device. **PyOxd-pCz** also has enough high E_T (2.60 eV) which can effectively prevent reverse energy transfer from dopant Ir(ppy)₃ (E_T = 2.37 eV) to host as **PyOxd-mCz** does so the main reason can explain the inferior performances of **PyOxd-pCz**-hosted green device is the unbalanced electron and hole transportation which leads to narrow recombination zone where triplet–triplet annihilation and exciton–polaron annihilation are easier to take place and eventually leads to lower power efficiency.

The OLEDs performances are summarized in Table 2.

From Table 2, we can see that **PyOxd-mCz**-hosted devices has lower roll-off than **PyOxd-pCz**-hosted devices. For example, at a practical brightness of 1000 cd m⁻², EQE of **PyOxd-mCz** based device B1 still remains 20.6%, which corresponds to a reduce of 0.9% from the maximum values. But for **PyOxd-mCz** based device B2, EQE is 14.1% and equals to 3.4% roll-off at 1000 cd m⁻². There is also an interesting phenomenon as analyzing the performances of the four OLED, that is the blue OLEDs has higher EQE than green OLEDs which is contrary to the previous reports. It can be explained by the two facts. Firstly green emitter Ir(ppy)₃ has longer lifetime of 1.9 μ s (ref. 35) than Flrpic (τ = 1.1 μ s),³⁶ which will lead to higher possibility of triplet–triplet annihilation (TTA) and lower efficiency. Secondly there are two requirements for efficient energy transfer from host to guest: one is the overlap between the absorption spectra of guests and the emission spectra of the hosts, which is satisfied for both (see Fig. S9†); the other is the energy overlap between the hosts and the guests.^{37,38} From Fig. S2† we can see that

Table 2 Electroluminescence characteristics of blue and green OLEDs

V_{on} (V)	L_{max} (cd m ⁻²)	EQE (%)		η_c (cd A ⁻¹)		1000 cd m ⁻²	10000 cd m ⁻²	100000 cd m ⁻²
		Max.	1000 cd m ⁻²	Max.	1000 cd m ⁻²			
B1	3.5	21 056	20.8	20.6	12.4	39.7	39.3	23.6
B2	3.0	15 572	14.6	14.1	6.3	28.8	28.0	12.5
G1	3.8	130 180	16.4	15.1	15.4	55.2	50.5	52.0
G2	3.6	78 909	10.6	10.4	9.48	36.5	36.5	32.6

FIrpic (LUMO/HOMO: -2.8 eV/ -5.6 eV) has more energy overlap with the two hosts (-2.6 eV/ -5.6 eV) than Ir(ppy)₃ (-3.0 eV/ -5.4 eV) does and thus leads to more efficient energy transfer and better device performances.

Conclusion

Two bipolar host materials, namely **PyOxd-mCz** and **PyOxd-pCz**, have been designed and developed for application in blue and green PhOLEDs. Electron transportation can be effectively improved for both of the two compounds by adjusting the ratio of n-type groups to p-type groups from 1 : 1 to 2 : 1. With more twisted configuration by meta-position substitution of carbazole, **PyOxd-mCz** has higher ET (2.77 eV) and more balanced charge transportation. **PyOxd-mCz** hosted blue device shows excellent performances with maximum EQE, CE and PE of 20.84%, 39.7 cd A⁻¹, 25.0 lm W⁻¹.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

We acknowledge support from the National Natural Science Foundation of China (21463016) and Science and Technology Research Project of Inner Mongolia Autonomous Region University (NJZY149).

References

- 1 S. R. Forrest, *Nature*, 2004, **428**, 911–918.
- 2 F. May, M. Al-Helwi, B. Baumeier, W. Kowalsky, E. Fuchs, C. Lennartz and D. Andrienko, *J. Am. Chem. Soc.*, 2012, **134**, 13818–13822.
- 3 W. Ji, J. Wang, Q. Zeng, Z. Su and Z. Sun, *RSC Adv.*, 2013, **3**, 14616–14621.
- 4 M. A. Baldo, D. F. O'brien, Y. You, A. Shoustikov, S. Sibley, M. E. Thompson and S. R. Forrest, *Nature*, 1998, **395**, 151.
- 5 C. Adachi, M. A. Baldo, M. E. Thompson and S. R. Forrest, *J. Appl. Phys.*, 2001, **90**, 5048–5051.
- 6 S. H. Cheng, S. H. Chou, W. Y. Hung, H. W. You, Y. M. Chen, A. Chaskar and K. T. Wong, *Org. Electron.*, 2013, **14**, 1086–1093.
- 7 M. A. Baldo, C. Adachi and S. R. Forrest, *Phys. Rev. B*, 2000, **62**, 10967–10977.
- 8 S. Gong, C. Zhong, Q. Fu, D. Ma, J. Qin and C. Yang, *J. Phys. Chem. C*, 2012, **117**, 549–555.
- 9 S. C. Dong, Y. Liu, Q. Li, L. S. Cui, H. Chen, Z. Q. Jiang and L. S. Liao, *J. Mater. Chem. C*, 2013, **1**, 6575–6584.
- 10 Y. Tao, Q. Wang, C. Yang, C. Zhong, K. Zhang, J. Qin and D. Ma, *Adv. Funct. Mater.*, 2010, **20**, 304–311.
- 11 S. H. Chou, W. Y. Hung, C. M. Chen, Q. Y. Liu, Y. H. Liu and K. T. Wong, *RSC Adv.*, 2013, **3**, 13891–13900.
- 12 E. Mondal, W. Y. Hung, Y. H. Chen, M. H. Cheng and K. T. Wong, *Chem.-Eur. J.*, 2013, **19**, 10563–10572.
- 13 P. Venkatakrishnan, P. Natarajan, J. N. Moorthy, Z. Lin and T. J. Chow, *Tetrahedron*, 2012, **68**, 7502–7508.
- 14 H. Huang, Y. Wang, B. Wang, S. Zhuang, B. Pan, X. Yang and C. Yang, *J. Mater. Chem. C*, 2013, **1**, 5899–5908.
- 15 E. Mondal, W. Y. Hung, H. C. Dai and K. T. Wong, *Adv. Funct. Mater.*, 2013, **23**, 3096–3105.
- 16 R. Braveenth, D. H. Ahn, J. H. Han, J. S. Moon, S. W. Kim, H. Lee and K. Y. Chai, *Dyes Pigm.*, 2018, **157**, 377–384.
- 17 J. Tagare, H. Ulla, M. N. Satyanarayan and S. Vaidyanathan, *J. Lumin.*, 2018, **194**, 600–609.
- 18 Y. T. Lee, Y. T. Chang, M. T. Lee, P. H. Chiang, C. T. Chen and C. T. Chen, *J. Mater. Chem. C*, 2014, **2**, 382–391.
- 19 Y. Tao, Q. Wang, C. Yang, C. Zhong, K. Zhang, J. Qin and D. Ma, *Adv. Funct. Mater.*, 2010, **20**, 304–311.
- 20 H. Xu, P. Sun, K. Wang, Y. Miao, T. Yang, H. Wang and W. Y. Wong, *Tetrahedron*, 2016, **72**, 4408–4413.
- 21 K. S. Vadagaonkar, C. J. Yang, W. H. Zeng, J. H. Chen, B. N. Patil, P. Chetti and A. C. Chaskar, *Dyes Pigm.*, 2019, **160**, 301–314.
- 22 L. Xu, Y. Zhao, G. Long, Y. Wang, J. Zhao, D. Li and X. W. Sun, *RSC Adv.*, 2015, **5**, 63080–63086.
- 23 W. Li, J. Li, F. Wang, Z. Gao and S. Zhang, *ACS Appl. Mater. Interfaces*, 2015, **7**, 26206–26216.
- 24 F. M. Hsu, C. H. Chien, C. F. Shu, C. H. Lai, C. C. Hsieh, K. W. Wang and P. T. Chou, *Adv. Funct. Mater.*, 2009, **19**, 2834–2843.
- 25 W. Li, J. Li, D. Liu and Q. Jin, *ACS Appl. Mater. Interfaces*, 2016, **8**, 22382–22391.
- 26 M. J. Frisch, G. W. Trucks, H. B. Schlegel, et al., *GAUSSIAN 03 (revision C.02)*, Gaussian, Inc., Wallingford, CT, 2004.
- 27 S. Megelski, J. S. Stephens, D. B. Chase and J. F. Rabolt, *Macromolecules*, 2002, **35**(22), 8456–8466.
- 28 H. Tashtoush and R. S. Al-Talib, *Magn. Reson. Chem.*, 1997, **35**, 549–552.
- 29 K. P. Kumar, M. G. Murali and D. Udayakumar, *Des. Monomers Polym.*, 2014, **17**, 7–18.
- 30 C. Fan, F. Zhao, P. Gan, S. Yang, T. Liu, C. Zhong, D. Ma, J. Qin and C. Yang, *Chem.-Eur. J.*, 2012, **18**, 5510–5514.
- 31 W. Li, J. Li, D. Liu, D. Li and F. Wang, *ACS Appl. Mater. Interfaces*, 2016, **8**, 21497–21504.
- 32 C. Adachi, R. C. Kwong, P. Djurovich, V. Adamovich, M. A. Baldo, M. E. Thompson and S. R. Forrest, *Appl. Phys. Lett.*, 2001, **9**, 2082–2084.
- 33 R. J. Holmes, S. R. Forrest, Y.-J. Tung, R. C. Kwong, J. J. Brown, S. Garon and M. E. Thompson, *Appl. Phys. Lett.*, 2003, **82**, 2422–2424.
- 34 X. K. Liu, C. J. Zheng, M. F. Lo, J. Xiao, Z. Chen, C. L. Liu and X. H. Zhang, *Chem. Mater.*, 2013, **25**, 4454–4459.
- 35 H. Yersin, *Highly efficient OLEDs with phosphorescent materials*, Wiley-VCH, Weinheim, 2008.
- 36 M. Sarma, W. L. Tsai, W. K. Lee, Y. Chi, C. C. Wu, P. T. Chou, S. H. Liu and K. T. Wong, *Chem.*, 2017, **3**, 461–476.
- 37 Y. Tao, C. Yang and J. Qin, *Chem. Soc. Rev.*, 2011, **40**, 2943–2970.
- 38 G. H. Kim, R. Lampande, J. H. Kong, J. M. Lee, J. H. Kwon, J. K. Lee and J. H. Park, *RSC Adv.*, 2015, **5**, 31282–31291.

