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Preparation of copper phthalocyanine/SiO₂ composite particles through simple, green one-pot wet ball milling in the absence of organic dispersants

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In order to improve the dispersibility, thermal stability and pH adaptability of organic pigments in water, submicrometer copper phthalocyanine (CuPc)/SiO₂ composite particles (CPs) were prepared through a simple one-pot wet ball-milling process under acidic conditions without using any organic surfactant. In the as-obtained CPs, the surface of the CuPc particles was homogeneously decorated with $SiO₂$ nanoparticles (NPs) through hydrogen bonding interactions. Due to the surface-attached $SiO₂$ NPs, the CuPc/SiO₂ CPs present a high aqueous dispersibility and a pH-dependent colloidal stability. Furthermore, both the thermal stability and color intensity of CuPc were increased by encapsulation of CuPc particles within SiO₂ NPs. PAPER
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Introduction

Organic pigments are used as colorants for printing inks, coatings, plastics, and cosmetics because of their high tinting strength, brilliance, broad chromatogram, and good thermal stability.¹⁻⁵ However, the wide applications of hydrophobic organic pigments are somehow limited by their poor aqueous dispersibility. Modification of pigment particles with polymeric, hybrid, and inorganic materials is often made to improve the aqueous dispersibility, colloidal stability at different pH values or ionic strengths of the medium, and even color intensity of pigment particles.⁶–⁹ For example, various pigment particles have been encapsulated with (co)polymers through emulsion, miniemulsion, or precipitation polymerization.¹⁰⁻¹²

Stabilization of organic pigments by inorganic materials such as $SiO₂$ and $TiO₂$ may also improve the aqueous dispersibility of organic pigments, as well as enhance their weatherability, thermal stability, and color intensity.¹³⁻¹⁵ Composite particles (CPs) consisting of pigment/inorganic materials may be prepared by coating an inorganic layer onto pigment particles modified with cationic polyelectrolytes or surfactants via a sol–gel process of inorganic precursors. For example, Yuan *et al.* prepared pigment/SiO₂ and pigment/TiO₂ CPs by deposition of inorganic materials onto pigment particles modified with cationic polyelectrolyte using a sol-gel process.

The resulting CPs displayed good UV-shielding property, weatherability, and enhanced thermal stability.^{16,17} Similarly, Yin *et al.* coated six types of pigment particles with a $SiO₂$ layer through a sol-gel process. Because of the $SiO₂$ layer, the pigment core– $SiO₂$ shell CPs displayed vivid color and fast response to electric fields, thus giving them potential in applications for full-color electrophoretic displays. Fabjan et al. coated pigment particles stabilized by cetyltrimethylammonium bromide with a $SiO₂$ layer through a solgel process of potassium water glass to improve the photostability of the pigment particles.¹⁸ In summary, two steps are required for the preparation of pigment/inorganic CPs through the aforementioned techniques: pulverization of pristine pigment aggregates into pigment particles using cationic polymeric or small molecular dispersants and coating of inorganic moieties onto the modified pigment particles through a sol-gel process. This multistep process is not straight forward and sometimes time consuming. In addition, the presence of organic dispersants may influence the downstream applications of the organic pigments.

Preparation of pigment/inorganic CPs through a simple physical milling process may be straightforward and efficient. Lan et al. prepared pigment/clay CPs by simple wet-pulverizing organic pigments in combination with smectite clays.¹⁹ Ichimura et al. prepared a nanosized $SiO₂$ core-pigment shell CPs by mechanical dry milling of organic pigments with surface silane-modified $SiO₂$ nanoparticles (NPs).^{20,21} The resulting CPs can be easily dispersed in propylene glycol monomethyl ether acetate by ball milling. However, the aqueous dispersibility of the CPs was limited because of their hydrophobic surface. The pigment particles were attached by clay particles. The surface

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ionic charges and unique plate-like geometric shape of the clay particles have been utilized to obtain pigment/clay CPs with good colloidal stability. However, this technique is limited to colloidal stabilizers with a unique geometric shape that can induce geometric-shape in homogenerity, which prevents particle aggregation.^{22,23} Although there are many commercially available spherical inorganic NPs, to our knowledge, the preparation of submicrometer pigment particles stabilized solely by spherical inorganic NPs through a simple one-pot wet ballmilling process has not been reported yet.

Spherical $SiO₂$ NPs have been widely used in stabilizing hetero phase polymerization systems, such as Pickering emulsions, Pickering miniemulsions, and Pickering dispersion polymerization systems for preparation of organic–inorganic hybrid NPs.²⁴⁻²⁶ In the present papers, commercial unmodified SiO2 NPs were used as the sole colloidal stabilizer to prepare copper phthalocyanine $(CuPc)/SiO₂$ CPs through a simple onepot wet ball-milling process without using any organic dispersant. The CuPc/SiO₂ CPs formed by hydrogen bonding between CuPc and $SiO₂$ NPs. The particle size of the CuPc/SiO₂ CPs can be reduced to \sim 330 nm by controlling the SiO₂/CuPc weight ratio and the pH value of the medium. The $CuPe/SiO₂$ CPs displayed reversible colloidal stability that was dependent on the pH value of the medium. Compared with pristine CuPc, the CuPc/SiO2 CPs showed enhanced thermal stability and improved color intensity. Paper

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Experimental

Materials

CuPc (C.I. Pigment Blue 15, chemical structure shown in Fig. 1) was obtained from Hangzhou Baihe Chemical Co. Ltd. Ludox HS-30 colloidal $SiO₂$ (30 wt%, average particle size 12 nm, pH 9.8) was purchased from Sigma–Aldrich Corporation. Colloidal SiO₂ (30 wt%, pH 9.9, average particle size 100 nm) was purchased from Zhejiang Yuda Chemical Co., Ltd. Hydrochloric acid (HCl, 37 wt%, Quzhou Juhua Chemical Reagent Co. Ltd.), sodium hydroxide (NaOH, AR grade, Quzhou Juhua Chemical Reagent Co. Ltd.), and sodium dodecylsulfate (SDS, AR grade, Shanghai No. 2 Chemistry Reagent Co. Ltd.) were used as received. Deionized water was used in all experiments.

Fig. 1 Chemical structure of CuPc.

Preparation of the aqueous dispersion of $CuPe/SiO₂CPs$

An aqueous dispersion of $SiO₂$ NPs (25 wt%), CuPc, and water were added to a 100 mL of glass cylinder. The pH value of the medium was adjusted by addition of HCl aqueous solution (3 mol L^{-1}) or NaOH aqueous solution (3 mol L^{-1}) . The mixture was intensively milled with1 mm $ZrO₂$ beads in a bead miller at an agitation rate of 1000 rpm for 30 min at room temperature. The recipes of the prepared dispersions of the $CuPe/SiO₂CPs$ are listed in Table 1.

Preparation of the aqueous dispersion of CuPc/SDS particles

SDS, CuPc and water were added to a 100 mL of glass cylinder. The pH value of medium was adjusted by addition of HCl aqueous solution (3 mol L^{-1}) to 2. The mixture was intensively milled with1 mm $ZrO₂$ beads in a bead miller at an agitation rate of 1000 rpm for 30 min at room temperature. The recipe of the prepared dispersion of the CuPc/SDS particles is listed in Table 1, run 2.

Evaluation of pH-dependent, reversible colloidal stability of CuPc/SiO₂ CPs

One gram of the CuPc/SiO₂ CP dispersion prepared in run 2 was diluted by 9 g of water. Its pH value was adjusted to 1.3 through addition of HCl aqueous solution, and then the pH value of medium was adjusted to 12.4 through addition of NaOH aqueous solution. Subsequently, the pH value of medium was re-adjusted to 1.3 through addition of 3 mol L^{-1} HCl aqueous solution. The particle sizes of the $CuPc/SiO₂$ CPs at various pH values during the pH-adjusting process were measured by dynamic light scattering (DLS) to evaluate the pH-dependent colloidal stability.

Adsorption efficiency and adsorbed amount of $SiO₂$ NPs onto the CuPc particles

Five grams of the dispersions of $CuPc/SiO₂CPs$ prepared in runs 3–8 were centrifuged at 12 000 rpm for 30 min, respectively. The supernatant was dried to measure the mass of free $SiO₂$ NPs. The adsorption efficiency of $SiO₂$ NPs (W%) was calculated by formula (1), in which m_1 and m_2 refer to the mass of SiO₂ NPs in the supernatant and the overall mass of $SiO₂$ NPs in 5 g of the CuPc/SiO₂ CP dispersion. The adsorbed amount of $SiO₂$ NPs on the CuPc particles (m) can be calculated by formula (2) , in which m_3 refers to the overall mass of SiO₂ NPs in 40 g of the CuPc/SiO₂ CP dispersion.

$$
W\% = 1 - \frac{m_1}{m_2} \tag{1}
$$

$$
m = m_3 \times W\% \tag{2}
$$

Table 1 Recipes of the prepared dispersions of the CuPc/SDS or CuPc/SiO₂ CPs (g)

Run	CuPc	SiO ₂ NPs dispersion	ratio	CuPe : SiO ₂ weight	Water	SDS	Medium pН
1	1.0	$\bf{0}$	0:1.0		39.0	0	2.0
2	1.0	$\mathbf{0}$			38.0	1.0	2.0
3	1.0	4.0	1.0:1.0		35.0	0	2.0
4	1.0	0.4	0.1:1.0		38.6	0	2.0
5	1.0	1.2	0.3:1.0		37.8	0	2.0
6	1.0	2.0	0.5:1.0		37.0	0	2.0
$\overline{7}$	1.0	3.0		0.75:1.0	36.0	$\mathbf{0}$	2.0
8	1.0	8.0	2.0:1.0		31.0	0	2.0
9	1.0	4.0	1.0:1.0		35.0	0	1.3
10	1.0	4.0	1.0:1.0		35.0	0	5.1
11	1.0	4.0	1.0:1.0		35.0	$\mathbf{0}$	7.2
12	1.0	4.0	1.0:1.0		35.0	$\mathbf{0}$	9.0
13	1.0	4.0	1.0:1.0		35	$\mathbf{0}$	12.4
14	1.0	$4.0_{(100 \text{ nm})}$	1.0:1.0		35.0	$\mathbf{0}$	2.0
	Characterization	Transmission electron microscopy (TEM) Particle morphologies of the samples were observed on a JSM- 1200EX transmission electron microscope operated at 80 kV. The preparation of TEM samples was as follows: a drop of	Fourier transform infrared spectroscopy (FTIR) FTIR spectra were recorded on a FTIR-430 spectrophotometer (VETEX 70, Bruker). The sample (1 mg) was mixed with 100 mg of FTIR-grade KBr. The powder mixture was pressed into a pellet for FTIR measurements.				
		dispersion was diluted with 5 mL of water; a drop of the diluted sample was placed on a 230 mesh Formvar copper grid and then	Ultraviolet-visible (UV-vis) spectroscopy				
	allowed to air dry at room temperature.		UV-vis absorption spectra of the samples in the spectral range of 400-700 nm were recorded on a Lambda 900 UV-vis spectro- photometer. The samples for analysis were prepared by diluting 50 μL of each sample with 5 mL of deionized water and taking				
	Dynamic light scattering (DLS)						
		Particle sizes and polydispersities (PDIs) of the CuPc/SiO ₂ CPs			2 mL of the diluted dispersion for measurement.		
		were analyzed by DLS (Zetasizer Nanoseries, Malvern) at 25 °C					
		under a scattering angle of 90° at a wavelength of 633 nm. The	X-ray photoelectron spectroscopy (XPS)				
preparation of the DLS samples was as follows: five grams of the $CuPe/SiO2$ CPs dispersion were centrifuged at 12 000 rpm for				Samples used for XPS measurements were extensively dried in a veguum, than the newday of camples were existinte a goot			

Characterization

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Particle sizes and polydispersities (PDIs) of the $CuPc/SiO₂$ CPs were analyzed by DLS (Zetasizer Nanoseries, Malvern) at 25 °C under a scattering angle of 90° at a wavelength of 633 nm. The preparation of the DLS samples was as follows: five grams of the CuPc/SiO2 CPs dispersion were centrifuged at 12 000 rpm for 30 min to remove the free $SiO₂$ NPs; the supernate was removed and 5 g of water was added to redisperse the collected solid samples; the centrifugation–redispersion cycle was repeated once; the aqueous dispersion of $CuPe/SiO₂$ CPs were treated by sonication at 500 W for 15 min; finally, the aqueous dispersion was transferred to a polystyrene cuvette. Particle sizes were reported as the average of three measurements.

Scanning electron microscopy (SEM)

Particle morphologies of the $CuPc/SiO₂$ CPs were also observed on a Hitachi SU8010 field emission scanning electron microscopy operated at 3 kV. The preparation of SEM samples was as follows: one gram of the $CuPe/SiO₂$ CP dispersion was centrifuged at 12 000 rpm for 30 min to remove the free $SiO₂$ NPs; the supernate was removed and 5 g of water was added to redisperse the collected solid samples; the centrifugation–redispersion cycle was repeated once; the aqueous dispersion of $CuPe/SiO₂$ CPs were treated by sonication at 500 W for 15 min; a drop of dispersion was diluted with 5 mL of water; a drop of the diluted sample was placed on a silicon chip and then allowed to air dry at room temperature.

Fourier transform infrared spectroscopy (FTIR)

Ultraviolet-visible (UV-vis) spectroscopy

X-ray photoelectron spectroscopy (XPS)

Samples used for XPS measurements were extensively dried in a vacuum, then the powders of samples were crush into a spot which the size is 250 μ m \times μ m for test. All survey scans sweeping over 0–1300 eV electron binding energy with a resolution of 1 eV were averaged. Sample charging was minimized by an electron flood gun operated at 3 eV.

Thermogravimetric analysis (TGA)

Weight losses of the pristine CuPc and $CuPe/SiO₂$ CPs were evaluated by TGA on a PerkinElmer Pyris I thermogravimetric analyzer by heating from 100 to 700 \degree C at a rate of 10 \degree C min⁻¹ under a nitrogen flow.

Results and discussion

Preparation of the aqueous dispersion of the $CuPe/SiO₂CPs$

In the absence of colloidal stabilizer, the particle size of the pristine CuPc particles in the aqueous dispersion prepared by ball milling was larger than $7 \mu m$ (Table 1, run 1). Moreover, the dispersion of the pristine CuPc particles was not stable, and obvious precipitates were formed after a short storage time period. SDS, as a commonly-used surfactant, was used as the colloidal stabilizer to prepared CuPc particles (Table 1, run 2).

As shown in Fig. 2a, the CuPc/SDS particles displayed an irregular granular morphology. The Z-average particle size of the CuPc/SDS particles was \sim 420 nm. Although the CuPc/SDS particles could stably dispersed in aqueous dispersion, the properties of the CuPc particles, such as heat resistance and color intensity, could not be obviously enhanced. Moreover, as aforementioned in the introduction, SDS, as a low molecularweight surfactant, may limited the downstream applications of organic pigments.¹⁹

Therefore, in the present work, $SiO₂$ NPs were used as the sole colloidal stabilizer to disperse the CuPc pigment in water. The particle size of the used $SiO₂$ NPs was \sim 25 nm, displaying a monodisperse particle size distribution (Fig. 2b). A deep blue, stable dispersion of the CuPc/SiO₂ CPs was prepared by simple one-pot wet ball-milling process of CuPc with $SiO₂$ NPs under an acidic condition (Fig. 2c, inset). The Z-average particle size and PDI of the CuPc/SiO₂ CPs as determined by DLS were \sim 350 nm and 0.39, respectively. Many $SiO₂$ NPs attached to the CuPc particles to improve the colloidal stability of the pigment particles (Fig. 2d). It should be pointed out some free $SiO₂$ NPs were also presented in this sample (Fig. 2c).

SiO2/CuPc weight ratio

Aqueous dispersions of the CuPc/SiO₂ CPs with various $SiO₂$ / CuPc weight ratios were prepared (Fig. 3a). The colloidal stability of the aqueous dispersion depended substantially on the $SiO_2/CuPc$ weight ratio. When the $SiO_2/CuPc$ ratio exceeded $0.3:1$, visually stable CuPc/SiO₂ CP aqueous dispersions can be prepared (Fig. 3a). The Z-average particle sizes and PDIs of CuPc/SiO₂ CPs at various SiO₂/CuPc weight ratios as measured by DLS (Fig. 3b) show that with the increase in weight ratio from

Fig. 2 TEM images of the CuPc/SDS particles ((a) Table 1, run 2), the $SiO₂$ NPs (b), the CuPc/SiO₂ CPs (c and d) at various magnifications ((c) 10 000 \times ; (d) 25 000 \times ; Table 1, run 3). Photograph of the aqueous dispersion of the CuPc/SiO₂ CPs (c, inset).

Fig. 3 (a) Photograph of the aqueous dispersions of $CuPc/SiO₂$ CPs with various SiO₂/CuPc weight ratios (Table 1, runs 1, 3-8). (b) Particle sizes and PDIs of the CuPc/SiO₂ CPs with various $SiO₂/CuPc$ weight ratios (Table 1, runs 3-8). (c) Adsorption efficiency of the SiO₂ NPs and the adsorbed $SiO₂$ amount onto the surface of CuPc particles at various SiO₂/CuPc weight ratios (Table 1, runs 3-8).

 $0.1:1$ to $1:1$, the particle sizes and PDIs of the CuPc/SiO₂ CPs substantially decreased. Further increase in the weight ratio to 2 : 1 led to a decrease in particle size and PDI to \sim 330 nm and 0.35, respectively. When the $SiO₂/CuPc$ weight ratio fell below $0.5:1$, the Z-average particle sizes of the CuPc/SiO₂ CPs were larger than 1000 nm. These results suggest the poor dispersibility of the CuPc particles in these systems due to the insufficient amount of $SiO₂$ NPs used. Therefore, it is necessary to keep the $SiO₂/CuPc$ weight ratio above $0.5:1$ to obtain a homogenous and colloidally stable dispersion of $CuPe/SiO₂$ CPs with a submicrometer size.

The reason for the decrease in the particle size of $CuPe/SiO₂$ CPs with the increase of the $SiO₂$: CuPc ratio can be explained by the adsorption efficiency and adsorbed amount of $SiO₂$ NPs. When the $SiO_2/CuPc$ weight ratio was below $0.3:1$, almost all the $SiO₂$ NPs were adsorbed on the surface of CuPc particles (Fig. 3c). It meant that when the particle size of the CuPc particles decreased to some extent, only a few free $SiO₂$ NPs were available to stabilize the newly-produced surface of the CuPc particles, and thus the particle size of the $CuPe/SiO₂$ CPs could not be further reduced by ball milling. With the increase of the $SiO₂/CuPc$ weight ratio, the amount of $SiO₂$ NPs could stabilize more newly-produced surface of the CuPc particles (Fig. 3c), leading to the smaller particle size of the $CuPc/SiO₂$ CPs.

pH value of the medium

The particle sizes and PDIs of the $SiO₂$ NPs at various pH values were determined by DLS (Fig. 4a). When the pH value of the mediums was below 10.0, the particle sizes and PDIs of $SiO₂$ NPs kept constant. Further increase of the pH value to 12 or above, the particle sizes and PDIs of the $SiO₂$ NPs increased obviously, indicative of the aggregation of the $SiO₂$ NPs.

Fig. 4 Particle sizes and PDIs of SiO₂ NPs (a) and CuPc/SiO₂ CPs (b) prepared at various pH value of the medium (Table 1, runs 3, 9–13). SEM images of the CuPc/SiO₂ CPs prepared at pH 2 (c) and 12.4 (d) (Table 1, runs 3 and 13).

For the CuPc/SiO₂ CPs, submicrometer CuPc/SiO₂ CPs could only be prepared under acidic conditions (Fig. 4b). When the pH value of the medium increased to 7.2, the particle size and PDI of the CuPc/SiO₂ CPs significantly increased to \sim 2.2 µm and 0.61, respectively. Further increase of the pH to 12.4 resulted in an increase in the particle size of the CuPc/SiO₂ CPs to above 10 μ m. Considering the good colloidal stability of SiO₂ NPs in the pH range of 2–10, the dependence of the particle size and PDI of the CuPc/SiO₂ CPs on the pH value can be reasonably ascribed to the various intensities of interaction between the CuPc and $SiO₂$ NPs at different pH values. The interaction between CuPc and $SiO₂$ NPs under acidic conditions was expected to be stronger than that under neutral and basic conditions. A detailed discussion on these interactions is given the following sections.

The particle morphology of the CuPc/SiO₂ CPs prepared from run 3 and 13 was observed by SEM. As shown in Fig. 4c, most of the CuPc/SiO₂ CPs prepared in run 3 separately distributed on the silicon chip, displaying a good dispersibility. Many $SiO₂ NPs$ firmly attached to the surface of CuPc particles. In contrast, only macro-aggregates of the CuPc particles were observed in the sample prepared in run 13 (Fig. 4b). These results are consistent well with the DLS results in Fig. 4b.

Hydrogen bonding between the CuPc and $SiO₂$ NPs

In order to clarify the interaction between the CuPc and $SiO₂$ NPs, the $SiO₂$ NPs, pristine CuPc, a mixture of the CuPc and $SiO₂$ NPs, and $CuPc/SiO₂$ CPs were characterized by FTIR. One characteristic peak centered at 3470 cm^{-1} (Fig. 5a), which can be assigned to the stretching vibration of the surface silanol groups, can be observed in the FTIR spectrum of $SiO₂$ NPs. Pristine CuPc did not produce any obvious characteristic peak

Fig. 5 (a) FTIR spectra of the CuPc particles, $SiO₂$ NPs, mixture of the CuPc and $SiO₂$ NPs, and CuPc/SiO₂ CPs; (b) UV-vis spectra of aqueous dispersions of the SiO₂ NPs, CuPc/SDS particles (Table 1, run 2), CuPc/ $SiO₂$ CPs (Table 1, run 3), CuPc/SDS particles with post-added SiO₂ NPs (CuPc/SDS + SiO₂), and CuPc/SiO₂ CPs with post-added SDS $(CuPc/SiO₂ + SDS)$

in this spectral range. The characteristic peak of $SiO₂$ NPs at 3470 cm^{-1} observed in the spectrum of the mixture of CuPc and $SiO₂$ NPs suggests no interaction between the CuPc and $SiO₂$ NPs in this sample. In contrast, the characteristic peak of silanol groups in the $CuPe/SiO₂$ CPs distinctly shifted to \sim 3410 cm⁻¹. This shift is strong evidence of the formation of hydrogen bonds between CuPc and $SiO₂$ NPs in the CuPc/SiO₂ CPs.27,28

The interaction between the CuPc and $SiO₂$ NPs in the CuPc/ $SiO₂$ CPs was further characterized by UV-vis spectroscopy and X-ray photoelectron spectroscopy. The UV-vis spectra as shown in Fig. 5b, the aqueous dispersion of the $SiO₂$ NPs did not show any absorption in the spectral range of 400–700 (curve 1, Fig. 5b). The aqueous dispersion of CuPc/SDS particles showed absorption in the spectral range of 505–700 nm, centered at \sim 610 nm (curve II, Fig. 5b). The aqueous dispersion of the $CuPe/SiO₂$ CPs produced a strong absorption peak in the spectral range of 520–700 nm, which was centered at 625 nm (curve III, Fig. 5b). The red-shifted absorption of $CuPe/SiO₂$ CPs may be regarded as the second evidence for the formation of hydrogen bonding between the CuPc and $SiO₂$ NPs in the CuPc/SiO₂ $CPS.$ ^{20,21}

The more precise information of chemical environment of $CuPe/SiO₂$ CPs' atoms (run 3) was obtained by deconvoluting high resolution XPS spectra of C(1s), N(1s), Cu(2p) (Fig. 6).²⁹⁻³¹ It needs to be emphasized that the C(1s) and N(1s) peak of $SiO₂$ can be attributed to the adsorption of carbon dioxide $(CO₂)$, oxygen (O_2) in the air during drying progress.

For CuPc, we can see that the Cu $(2p)$ spectrum had two strong peaks (Fig. 6a), one located at 956.15 eV and another located at 936.12 eV, corresponding to the electron states of $Cu(2p_{1/2})$ and $Cu(2p_{3/2})$, respectively.^{32,33} The binding energy of copper atoms (in Cu $2p_{3/2}$ electron state) was 936.12 eV, showing that the copper atoms in CuPc thin film were in $Cu(n).³⁴$ This can be explained from the chemical structure of CuPc molecule (Fig. 1). In the CuPc molecule, the copper atom bonded with nitrogen atoms through coordinate bonds.

From the fine spectrum of C 1s in the raw surface state (Fig. 6c) we can find that, the main peak located at 285.6 eV,

Fig. 6 XPS spectra and fine spectra of (a) C(1s), (b) N(1s), (c) Cu(2p), (d) Si(2p) on the CuPc particles, $SiO₂$ NPs and CuPc/SiO₂ CPs.

which corresponds to the binding energy of the aromatic carbon atoms, i.e. the carbon atoms bonding with carbon and hydrogen atoms.35,36 The binding energy of carbon atoms in C–N bonds was observed at 287.9 eV. In the CuPc molecule, the nitrogen atoms were also in two kinds of chemical environment: four nitrogen atoms only bond with two carbon atoms and form C–N=C bonds, and the other four nitrogen atoms not only bond with carbon atoms but also bond with copper atom through coordination bond. We consider that the peak located at 400.1 eV corresponds to the binding energy of the nitrogen atoms in C–N= C bonds,^{37,38} and the peak observed at 401.3 eV may be due to the nitrogen atoms which bonding with carbon and copper atoms.

Compared to the CuPc, the binding energy of N(1s) on CuPc/ $SiO₂$ CPs has binding energy shift, which mean the extranuclear electron density decreased after ball milling. Stevens et al. has reported that when the carbon–nitrogen double bond $(C=N)$ form hydrogen bond with hydroxyl (OH), the binding energy of $N(1s)$ had a shift of +0.6 eV.³⁹ The reason of this change is possibly attributed to hydrogen bonding interactions between carbon–nitrogen double bond of CuPc and silicon hydroxyl. But as shown in $N(1s)$ XPS fine spectra of $N(1s)$, the binding energy shift of $N(1s)$ is attribute to the shift of the peak of nitrogen atoms in $C-N=C$ bonds. So, it can be guessed that the N atom which only bond with two carbon atoms and form $C-N=C$ bonds form hydrogen bonds with silicon hydroxyl.

As shown in Fig. 6c, the binding energy of two kind of C had a shift of $+0.4$ ev, 0.5 eV, respectively. The reason can be explained that the phthalocyanine was a conjugate structure, the cloud density of C was decrease because of the form of hydrogen bond. The same phenomenon was also can be seen in the Cu atom. In addition, the Si(2p) binding energy in composite decreased as compared to the value in pure $SiO₂$ binding energy, which indicates that the surface Si–OH binding

energy has shifted to the lower value. Therefore, the different changes of the $N(1s)$ of CuPc and Si-OH groups in the composites can be deduced that the formation of hydrogen bonding between the N atom in CuPc and the surface Si–OH group in $SiO₂$ NPs. The C(1s) binding energy of CuPc group comparing to pure CuPc increases and the Cu(2p) binding energy of $CuPe/SiO₂$ corresponding to $CuPe$ decreases, which also strongly confirms the formation of hydrogen bonding in the composite. The analyzed results of XPS spectra are also completely in accordance to that of FTIR and UV-vis spectra.

In conclusion, a plausible hydrogen bond between the CuPc and $SiO₂$ NPs is shown in Scheme 1 (the possible hydrogen bonds in fig shows by imaginary line). Hydrogen bonds between CuPc and $SiO₂$ NPs can form through Lewis acid-base interaction between the nitrogen atoms of CuPc particles and the silanols of $SiO₂$ NPs. There is large amounts of silanol on the surface of $SiO₂$ NPs, because the particle size of $SiO₂$ NPs is small. So it has a large probability that the silanol groups on the surface of $SiO₂$ NPs can form hydrogen bonds with CuPc molecule. It can also be happened that the $SiO₂$ NPs form the hydrogen bonds with each other.⁴⁰

In order to confirm the role of the ball milling in the formation of the interaction between the CuPc and $SiO₂$ NPs, the dispersion of $SiO₂$ NPs was dropwise added to the dispersion of the CuPc/SDS particles, and the resulted dispersion was named as the CuPc/SDS + $SiO₂$ dispersion. The UV-vis absorption spectrum of the CuPc/SDS + $SiO₂$ dispersion (curve IV, Fig. 5b) was almost the same as that of the dispersion of the CuPc/SDS CPs. It means that the newly-produced surface of the CuPc particles has already been occupied by SDS, and the simple mixing the CuPc/SDS particles with $SiO₂$ NPs could not form a strong interaction between the CuPC and $SiO₂$ NPs. Therefore, the co-milling process of CuPc and $SiO₂$ NPs played a crucial role in the formation of the hydrogen bonds between $CuPe$ and $SiO₂$ NPs. During co-milling of CuPc with zirconium beads, pristine CuPc aggregates were broken up under strong mechanical forces. Numerous new CuPc surfaces, which contain active nitrogen atoms, were produced in the milling process. The $SiO₂$ NPs were efficiently adsorbed by the CuPc particles by hydrogen bonding between nitrogen atoms of the CuPc particles and silanol groups of $SiO₂$ NPs.

Scheme 1 A possible hydrogen bond between the CuPc and $SiO₂$ NPs in the CuPc/SiO₂ CPs.

In order to confirm the interaction intensity between the CuPc and $SiO₂$ NPs, an aqueous solution of SDS was added to the dispersion of the CuPc/SiO₂ CPs, and the resulted dispersion was named as the CuPc/SiO₂ + SDS dispersion. The UV-vis absorption spectrum of the CuPc/SiO₂ + SDS dispersion (curve V, Fig. 5b) was almost the same as that of the dispersion of the CuPc/SiO₂ CPs. It means that the post-added SDS could not break the interaction between the CuPc and $SiO₂$ particles to replace the adsorbed $SiO₂$ NPs.

Many $\text{Si}(\text{OH}_2)^+$ or Si–OH groups on the surface of SiO_2 NPs could form hydrogen bonds with nitrogen atoms on the surface of the CuPc particles under acidic conditions. Most of silanols ionized to $Si-O^-$ under basic conditions, thus suppressing hydrogen bonding between CuPc and SiO₂ NPs. Therefore, hydrogen bonding between CuPc and $SiO₂$ NPs under acidic conditions is expected to be stronger than that under basic conditions. Consequently, colloidally stable dispersions of submicrometer $CuPe/SiO₂$ CPs were prepared only under acidic conditions.

pH-dependent, reversible colloidal stability of the prepared CuPc/SiO₂ CPs

As shown in Fig. 7, the colloidal stability of the prepared CuPc/ $SiO₂$ CPs can be tuned by pH value of the medium. The aqueous dispersions of the prepared $CuPe/SiO₂$ CPs under acidic and neutral conditions showed good colloidal stability (Fig. 7a, inset). Particle sizes and PDIs of the prepared $CuPe/SiO₂$ CPs increased slightly with the increase in the medium pH from 2.0 to 6.7 (Fig. 7a), and then obviously increased from pH 6.7 to 9.5 and 12.4. These changes indicate the aggregation of the prepared CuPc/SiO₂ CPs under basic conditions. Many precipitates were observed at the bottom of the dispersion at 12.4 (Fig. 7a, inset).

Macro-aggregates formed at pH 12.4 can be redispersed by gently shaking when the medium pH was adjusted to the neutral or acidic levels (Fig. 7b, inset). Particle sizes and PDIs of the CuPc/SiO₂ CPs decreased to \sim 390 nm and 0.34 at pH 1.3, respectively, indicating pH-dependent, reversible colloidal stability of the aqueous dispersion.

Fig. 7 Variation of particle sizes and PDIs of the prepared $CuPc/SiO₂$ CPs with (a) increasing and (b) decreasing the pH value of medium (Table 1, run 3). Visible changes in the aqueous dispersions of the prepared CuPc/SiO₂ CPs with (a, inset) increasing and (b, inset) decreasing the pH value of medium (Table 1, run 3).

As mentioned previously, the particle sizes of the $SiO₂$ NPs were \sim 25 nm when the pH value was in the range of 2–10 (Fig. 4a). However, when the pH value of the medium increased to 11.8, the particle size of $SiO₂$ NPs increased to 164 nm, showing the lowered colloidal stability of $SiO₂$ NPs. Visible aggregates of the $SiO₂$ NPs were observed when the medium pH increased to 12.4. These aggregates can be redispersed in the aqueous dispersion if the pH of the dispersion was restored to acidic levels. The $CuPe/SiO₂$ CPs were densely covered by many $SiO₂$ NPs (Fig. 2d and 4c). Therefore, the CuPc/SiO₂ CPs displayed a very similar pH-dependent, reversible colloidal stability to $SiO₂$ NPs.

Formation process of $CuPe/SiO₂CPs$

The formation process of $CuPe/SiO₂$ CPs is shown in Scheme 2. One pristine CuPc particle is an aggregate of numerous primary particles. During the ball-milling process, the pigment aggregates were gradually broken up to produce many fresh surfaces. Under acidic conditions, strong hydrogen bonding between CuPc and $SiO₂$ NPs led to the formation of CuPc/SiO₂ CPs. Adsorption of $SiO₂$ NPs onto CuPc particles increased the hydrophilicity of CPs, thus increasing the aqueous dispersibility of the CuPc/SiO₂ CPs relative to that of pristine CuPc particles. The adsorbed $SiO₂$ NPs thus improved the colloidal stability of the dispersion of pigment particles, similar to stabilization observed in the well-known Pickering systems.¹⁹ Based on the above conclusions, if the particle size of the $SiO₂$ NPs is too large, there is not much chance of forming strong hydrogen bond between each other. The large size of $SiO₂$ NPs can't stabilize the system. The experimental results (run 14) also show that $SiO₂$ particles, which the average particle size is 100 nm, cannot form a stable dispersion under the same experimental conditions. BSC Advances

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Thermal stability of $CuPe/SiO₂CPs$

TGA thermograms of the CuPc particles, SiO₂ NPs, and CuPc/ $SiO₂$ CPs prepared at various pH values are shown in Fig. 8. The thermogram of CuPc displays a three-step weight loss. The first step starting at \sim 250 °C can be attributed to the disruption of

Scheme 2 Schematic representation of the formation process of the CuPc/SiO₂ CPs.

Fig. 8 TGA thermograms of CuPc, $SiO₂$ NPs and CuPc/SiO₂ CPs prepared at various medium pH (Table 1, runs 3, 9–13).

C=C and C=N bonds; the second step beginning at \sim 470 °C can be attributed to the disruption of C–C, C–N, and C–H bonds; the last step initiating at \sim 640 °C can be attributed to the decomposition of coordination bonds between copper and phthalocyanine.^{41,42}

Decomposition temperatures at 10% of degradation (T_{10}) of the different samples are summarized in Table 2. T_{10} of CuPc was \sim 470 °C, which is lower than that of the CuPc/SiO₂ CPs by at least 68 °C. This difference suggests enhanced thermal stability of the CuPc/SiO₂ CPs. This enhanced thermal stability may be reasonably ascribed to the encapsulation of CuPc by the $SiO₂$ NPs, which could retard the decomposition of the pigment.⁴ T_{10} of the CuPc/SiO₂ CPs decreased with the increase in the pH value of the medium. This reduction can be ascribed to the relatively weaker hydrogen bonding between the CuPc and $SiO₂$ NPs at higher pH value of the medium, which decreases the amount of adsorbed $SiO₂$ NPs.

Color-enhancing effect of $SiO₂$ NPs

As shown in Fig. 3a, the color of the aqueous dispersions of the $CuPe/SiO₂$ CPs changed from light blue to deep blue with the increase in the $SiO₂/CuPc$ weight ratio. UV-vis spectra (Fig. 9) indicate that all the dispersions of the CuPc/SiO₂ CPs strongly adsorbed in the wavelength range of 520–700 nm. The absorbance of the dispersions increased with the increase in the $SiO₂/$ CuPc weight ratio, consistent with the visual observations depicted in Fig. 3a. The higher absorbance may be attributed to the improved dispersibility of CuPc in water and the colorenhancing effect of $SiO₂$.⁴³ Two factors might have

Table 2 T_{10} of CuPc and CuPc/SiO₂ CPs prepared at various pH value of the medium

	$CuPe/SiO2$ CPs						
					Samples CuPc $pH = 1.3$ $pH = 2.0$ $pH = 5.1$ $pH = 7.2$ $pH = 12.4$		
T_{10} (°C) 470	633	631	614	584	538		

Fig. 9 UV-vis spectra of the aqueous dispersions of the CuPc/SiO₂ CPs with various $SiO₂/CuPc$ weight ratios (Table 1, runs 3–8).

concurrently contributed to the higher absorbance. The higher absorbance might be solely due to the color-enhancing effect of $SiO₂ NPs$, as the CuPc/SiO₂ CPs show good dispersibility in this range.

Conclusions

Aqueous dispersible $CuPc/SiO₂$ CPs were successfully prepared through one-pot wet ball milling of CuPc with $SiO₂$ NPs under acidic conditions in the absence of any organic dispersant. The $SiO₂$ NPs were attached to the surface of CuPc particles by hydrogen bonding. The colloidal stability and particle size of the CuPc/SiO₂ CPs can be tuned by changing the $SiO_2/CuPc$ weight ratio and pH value of the medium. A colloidally stable dispersion of the CuPc/SiO₂ CPs can be obtained when the $SiO₂/$ CuPc weight ratio was maintained at above 0.5 : 1 under acidic conditions. At higher $SiO₂/CuPc$ weight ratio, the particle size of CuPc/SiO₂ CPs decreased, reaching \sim 330 nm when the SiO₂/ CuPc weight ratio was $2:1$. CuPc/SiO₂ CPs displayed a pHdependent, reversible colloidal stability. Compared with CuPc, the CuPc/SiO₂ CPs showed a substantially enhanced thermal stability and greater color intensity. Our proposed technique based on the wet ball-milling process is green, convenient, and efficient, and may thus find wide applications in the preparation of various pigment/inorganic CPs. Paper

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Conflicts of interest

There are no conflicts to declare.

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