


 Cite this: *RSC Adv.*, 2019, **9**, 22597

Improving optoelectronic and charge transport properties of D- π -D type diketopyrrolopyrrole-pyrene derivatives as multifunctional materials for organic solar cell applications†

 Ruifa Jin, ^{ab} Kexin Li^a and Xueli Han^a

A series of novel diketopyrrolopyrrole-pyrene-based molecules were designed for small molecule based organic solar cell (SMOSC) applications. Their electronic and charge transfer properties were investigated by applying the PBE0/6-31G(d,p) method. The absorption spectra were simulated using the TD-PBE0/6-31G(d,p) method. The results showed that the frontier molecular orbital (FMO) energy levels, reorganization energy, the energetic driving force, and absorption spectra can be tuned by the introduction of different aromatic heterocyclic groups to the side of diketopyrrolopyrrole fragments' backbones. Additionally, the designed molecules possess suitable FMOs to match those of typical acceptors PC₆₁BM and PC₇₁BM. Meanwhile, the designed molecules can act as good ambipolar charge transport materials in SMOSC applications. Meanwhile, the electron and hole reorganization energies of the designed molecules are smaller than those of the typical electron and hole transport materials, respectively. Moreover, the differences between electron and hole reorganization energies do not exceed 0.046 eV. Our results suggest that the designed molecules can act as promising candidates for donor and ambipolar charge transport materials in SMOSC applications.

 Received 8th June 2019
 Accepted 12th July 2019

DOI: 10.1039/c9ra04304g

rsc.li/rsc-advances

Introduction

Organic solar cells (OSCs), as promising candidates to address the global energy crisis, have attracted great attention all over the world owing to their outstanding merits, including low cost, light weight, and flexibility.^{1–6} In particular, the small molecule based OSCs (SMOSCs) have received tremendous interest due to their prominent advantages over polymer solar cells (PSCs), such as defined structures, relatively easy purification, and less batch to batch variation.^{7–10} With continuous efforts, significant advances have been demonstrated for OSCs. Recently, the power conversion efficiency (PCE) of PSCs and SMOSCs have exceeded 10%.^{11–13} However, unfortunately, the overall performance of SMOSCs still lags behind that of PSCs.^{14,15} Accordingly, it remains a great challenge to achieve high performance

SMOSCs. To address this issue, an ideal small molecule used as the donor material should meet several requirements: (i) lower band gap and strong absorption for harvesting the solar light effectively, (ii) high hole mobility for fast charge-carrier transport to lead to high short circuit current (J_{sc}), (iii) suitable frontier molecular orbitals (FMO) energy levels. A deep highest occupied molecular orbital (HOMO) energy of molecules ensures a high open circuit voltage (V_{oc}).^{16–18} Meanwhile, another crucial parameter is the energetic driving force (ΔE_{L-L}). The ΔE_{L-L} is determined by the lowest unoccupied molecular orbital (LUMO) energy difference between the donor and acceptor. The ΔE_{L-L} plays a major role in enhancing charge transfer, which should exceed the binding energy.^{19,20} Generally, the fullerene derivatives PC₆₁BM or PC₇₁BM ([6,6]-phenyl-C₆₁- (or C₇₁)-butyric acid methyl ester) were used as acceptors in OSCs.^{21,22} With the aim to enhance the PCE of SMOSCs, one efficient method is to design electron donor-acceptor (D-A) molecular structures.^{23,24} They can efficiently tune the FMOs energy levels through an intramolecular charge transfer, which leads to a narrow band gap, intense light absorption, and high PCEs.^{25–27} Additionally, the intermolecular stacking, film morphology, and charge transport properties can be determined by the type and length of side chains of D-A molecular structures.^{28,29} It has been reported that a variety of structures, including A-D-A, D-A-D, D- π -A, D- π -D, A- π -A, and D- π -A- π -D were designed.^{30,31} Among the various D-A type donor

^aCollege of Chemistry and Chemical Engineering, Chifeng University, Chifeng 024000, China. E-mail: Ruifajin@163.com

^bInner Mongolia Key Laboratory of Photoelectric Functional Materials, Chifeng 024000, China

† Electronic supplementary information (ESI) available: The following are available online at <http://www.mdpi.com/xxx/s1>, Table S1 calculated the longest absorption wavelengths λ_{abs} of parent molecule **1** by various methods with 6-31G(d,p) basis set, along with available experimental data. Table S2 the calculated E_{HOMO} and E_{LUMO} (in eV) for PC₆₁BM and PC₇₁BM at PBE0/6-31G(d,p) and B3LYP/6-31G (d,p) levels, along with available experimental data. Table S3 the differences between the E_{HOMO} of **1–8** and the E_{LUMO} of PC₆₁BM and PC₇₁BM at the PBE0/6-31G(d,p) level. See DOI: 10.1039/c9ra04304g

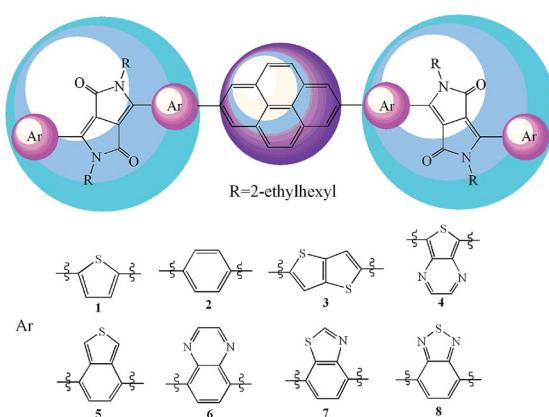


building blocks, diketopyrrolopyrrole (DPP)-based molecules have attracted great attention in terms of their outstanding features, such as good planarity, strong light absorption, high carrier mobility, good photochemical stability, and facile synthesis.^{32,33} Moreover, pyrene is another promising donor building block. The pyrene derivatives can enhance the absorption due to their large aromatic hydrocarbon ring and planarity.^{34,35} The planarity of molecules can improve the charge transport and interconnectivity.³⁶ Therefore, the planar DPP and pyrene have combined each other to develop novel donor materials.³⁷ Furthermore, it is worth noting that the charge mobility can be improved through introducing extended aromatic bridges to the DPP derivatives.³⁸ Recently, DPP-based DDP–pyrene–DDP type molecules have been reported for use as donor material for SMOSCs. It was found that these molecules exhibited high hole mobility and PCEs for SMOSCs applications.^{39,40}

Taking into account these merits of DPP and pyrene derivatives mentioned above, we designed eight D–A–D type molecules as the acceptor materials for improving the efficiency of SMOSCs (Scheme 1). In order to study the influence of the DPP moiety on the optical, electronic, charge transfer properties, different aromatic heterocyclic groups have been introduced into the side of DPP in molecules. Applying density functional theory (DFT) approach, the properties of the designed molecules such as HOMO and LUMO energies (E_{HOMO} and E_{LUMO}), HOMO–LUMO gap (E_g), $\Delta E_{\text{L-L}}$, the reorganization energy (λ), adiabatic ionization potential (AIP) and electron affinity (AEA) were systematically investigated. The absorption spectra of the designed molecules were predicted using the time dependent DFT (TD-DFT) method. The charge transfer properties (reorganization energy, λ) were also simulated.

Computational methods

All the electronic structure calculations were carried out performed using the Gaussian 09 program suite.⁴¹ The molecular structures were optimized by the DFT method. Then the frequency calculations were performed at the same level of theory to confirm that all geometries are stable structures.



Scheme 1 Molecular structures of the investigated molecules.

Under the optimized structures, their absorption spectra were simulated at the TD-DFT level of theory. The all calculations were carried out with the 6-31G(d,p) basis set. The solvent effects on the optical property in chloroform solution (dielectric constant: 4.7113) were also analyzed with the polarised continuum model (PCM).⁴² For the FMOs energy E_{HOMO} and E_{LUMO} , the optimized geometries were used to obtain E_{HOMO} . However, it is difficult to obtain the E_{LUMO} since the virtual orbitals can not be described definitely by theoretical method.⁴³ Therefore, the E_{LUMO} were be predicted by the sum of the E_{HOMO} and the first vertical excited energy.^{44,45} It is important to choose an appropriate method in the theoretical investigations. In order to select reasonable method, six representative functionals, B3LYP,⁴⁶ PBE0,⁴⁷ LC- ω PBE,⁴⁸ ω B97XD,⁴⁹ M062X,⁵⁰ and CAM-B3LYP⁵¹ were used to optimize the geometries of molecule 1, and then using the corresponding TD-DFT method, the absorption spectra were predicted in chloroform, respectively. The calculated data are shown in Table S1 in the ESI.† As showing Table S1,† the calculated longest wavelengths of absorption (λ_{abs}) using PBE0 (583 nm) approach compares very satisfactorily with the experimental result (589 nm).³⁹ Furthermore, in order to make further investigation of the validity of the selected approach, both PBE0 and B3LYP methods were also used to obtain the E_{HOMO} and E_{LUMO} of acceptors PC₆₁BM and PC₇₁BM, and the calculated results are listed in Table S2 in the ESI.† The predicted E_{HOMO} and E_{LUMO} of PC₆₁BM (−5.98 and −3.99 eV) and PC₇₁BM (−5.92 and −3.82 eV) using PBE0 approach reproduce well the experimental data (−6.00 and −3.80 eV, −6.00 and −3.95 eV, respectively).^{52,53} Comprehensively, the PBE0 method arises as the best choice for characterizing the systems studied in our work.

It is worth noting that the λ plays an important role in determining the charge transfer. Usually, lower λ value is beneficial for improving the charge transfer.^{54,55} Meanwhile, the external λ is much small compared with its internal λ in organic condensed phases.^{56,57} Therefore, we only consider the internal λ in this work. The electron and hole reorganization energy (λ_e and λ_h) were be predicted by:⁵⁸

$$\lambda_e = (E_0^- - E_-) + (E_-^0 - E_0^0) \quad (1)$$

$$\lambda_h = (E_0^+ - E_+^+) + (E_+^0 - E_0^0) \quad (2)$$

where, $E_0^+(E_0^-)$ is the energy of the cation (anion) structure with the optimized neutral geometry molecule. $E_+^0(E_-^0)$ corresponds to the energy of the neutral structure with the optimized cation (anion) geometry structure. $E_+^+(E_-^-)$ represents the energy of the cation (anion) calculated with the optimized cation (anion) structure, while E_0^0 is the energy of the neutral molecule at ground state. Furthermore, it is quite clear that the AIP and AEA play an important role in determining the charge transfer behavior for materials. In this work, the AIP is obtained by the energy difference between cation and neutral molecules. The AEA is determined by the energy difference between neutral and anion molecules. The λ_e , λ_h , AIP, and AEA were calculated using PBE0/6-31G(d,p) method.

Results and discussion

Match between the donor and acceptor

It is noticeable that the electronic energy levels play important role roles to increase the PCE of SMOSCs. Small molecule donors should possess suitable energy levels with the acceptors to ensure efficient charge separation and to maximize the J_{sc} and V_{oc} . The deep HOMO and high LUMO levels are beneficial for improving the V_{oc} and for efficient exciton dissociation and charge separation, respectively. Moreover, small molecule donors should exhibit lower E_g to guarantee high harvest sunlight efficiency and enhance the J_{sc} . Moreover, in order to ensure the efficient electron transfer, the ΔE_{L-L} should at least amount to 0.3 eV.^{59,60} We choose PC₆₁BM and PC₇₁BM as acceptors in our work. Additionally, the explicit molecular orbital contributions from the individual fragments to the FMOs have been investigated by partial density of states (PDOS) analysis and the results are listed in Table 1. With the aim of evaluating whether the FMOs energy of designed donors match with that of the acceptors, the ΔE_{L-L} are estimated. The calculated values of ΔE_{L-L} for **1–8** at the PBE0/6-31G(d,p) level are given in Table 2. The results presented in Table 2 reveal that the E_{LUMO} of **1–8** are positioned above those of PC₆₁BM and PC₇₁BM, respectively. Meanwhile, the ΔE_{L-L} of **1–8** are 0.61–1.35 and 0.45–1.18 eV for PC₆₁BM and PC₇₁BM as acceptors respectively, which are all beyond 0.3 eV. It indicates that the charge transfer from **1–8** to PC₆₁BM and PC₇₁BM efficiently. Additionally, the E_{HOMO} of **1–8** are larger than those of the E_{LUMO} of PC₆₁BM and PC₇₁BM about 1.07–1.40 and 1.24–1.57 eV, respectively (see Table S3 in the ESI†). Both the sequences of difference values are **3 < 1 < 4 < 7 < 5 < 6 < 2 < 8**. It suggests that **1–8** possess suitable FMOs to match those of typical acceptors PC₆₁BM and PC₇₁BM for SMOSCs.

Frontier molecular orbitals level and band gaps

To gain insight into the influence of the FMO energies on the optical and electronic properties of the designed molecules, the contour plots of HOMOs and LUMOs have been examined (see Fig. 1). Inspection of the distribution of FMO reveals clearly that the HOMOs and LUMOs reveal π characteristic and exhibit

Table 1 The HOMOs and LUMOs contributions (%) of **1–8** at the PBE0/6-31G(d,p) level

Molecules	HOMOs		LUMOs	
	PYR ^a	DPP ^b	PYR	DPP
1	6.7	93.3	11.0	89.0
2	3.8	96.2	6.2	93.8
3	4.6	95.4	6.9	93.1
4	8.6	91.4	9.9	90.1
5	3.0	97.0	5.0	95.0
6	1.8	98.2	3.9	96.1
7	1.1	98.9	4.7	95.3
8	3.4	96.6	4.9	95.1

^a PYR: pyrene moieties. ^b DPP: diketopyrrolopyrrole moieties.

Table 2 Evaluation of calculated FMO energies, HOMO–LUMO gap E_g , and energetic driving force ΔE_{L-L} for **1–8** at the PBE0/6-31G(d,p) level

Molecules	E_{HOMO}	E_{LUMO}	E_g	ΔE_{L-L}^a	ΔE_{L-L}^b
1	−5.11	−2.93	2.18	1.06	0.89
2	−5.33	−2.71	2.62	1.28	1.11
3	−5.06	−2.98	2.08	1.01	0.84
4	−5.18	−3.38	1.80	0.61	0.44
5	−5.23	−2.82	2.41	1.17	1.00
6	−5.27	−3.00	2.27	0.99	0.82
7	−5.19	−2.64	2.55	1.35	1.18
8	−5.39	−3.37	2.02	0.62	0.45

^a Energetic driving force for PC₆₁BM as acceptor. ^b Energetic driving force for PC₇₁BM as acceptor.

strong delocalization features. For molecules **1–5**, the electronic cloud distributions of the FMOs spread over the whole molecules with minor contributions from π -bridge fragment (pyrene). For molecules **6** and **7**, the electronic cloud distributions of HOMOs are mainly delocalized on the one side DPP fragments, whereas the corresponding the LUMOs mainly reside at another side DPP fragments of molecules. For molecules **8**, the HOMO is delocalized throughout the molecule, while the electron density of LUMO is mainly localized on the one side DPP fragment of molecule. This is favorable for the charge transfer across these conjugated donor materials. It suggests that the introduction of the extended aromatic bridges to the DPP derivatives significantly affect on the distributions of FMOs for the molecules.

The evaluations of E_{HOMO} , E_{LUMO} , and E_g for the designed molecules as well as PC₆₁BM and PC₇₁BM are plotted in Fig. 2. From Fig. 2 one can find that the E_{HOMO} values are in the order of **3 > 1 > 4 > 7 > 5 > 6 > 2 > 8**, and the sequence of E_{LUMO} values is **7 > 2 > 5 > 1 > 3 > 6 > 8 > 4**. Clearly, comparing the E_{HOMO} and E_{LUMO} of **2–8** with those of **1**, molecules **2**, **5**, and **7** can lower/raise the E_{HOMO}/E_{LUMO} , whereas molecule **3** can raise/lower the E_{HOMO}/E_{LUMO} , respectively. On the other hand, molecules **4**, **6**, and **8** can lower both the E_{HOMO} and E_{LUMO} , respectively. The trend of the E_g is **2 > 7 > 5 > 6 > 1 > 3 > 8 > 4**. It reveals that molecules **2**, **5**, **6**, and **7** can higher, whereas molecules **3**, **4**, and **8** can narrower the E_g than that of **1**, respectively. The above results suggest that the E_{HOMO} , E_{LUMO} , and E_g can be tuned through the introduction of different aromatic heterocyclic groups to the molecules.

Absorption and fluorescence spectra

Table 3 collected the longest absorption wavelengths λ_{abs} , oscillator strength f , and the main assignments of **1–8**. The corresponding simulated absorption spectra are presented in Fig. 3. Inspection of Table 3 reveals clearly that the λ_{abs} of molecules **1–5** correspond to the promotion of an electron from the HOMO \rightarrow LUMO and HOMO−1 \rightarrow LUMO+1 transitions, respectively. The 530 nm of molecule **6** essentially originates from the coupling of electronic transitions of HOMO \rightarrow LUMO, HOMO−1 \rightarrow LUMO, and HOMO \rightarrow LUMO+1. The shoulder at



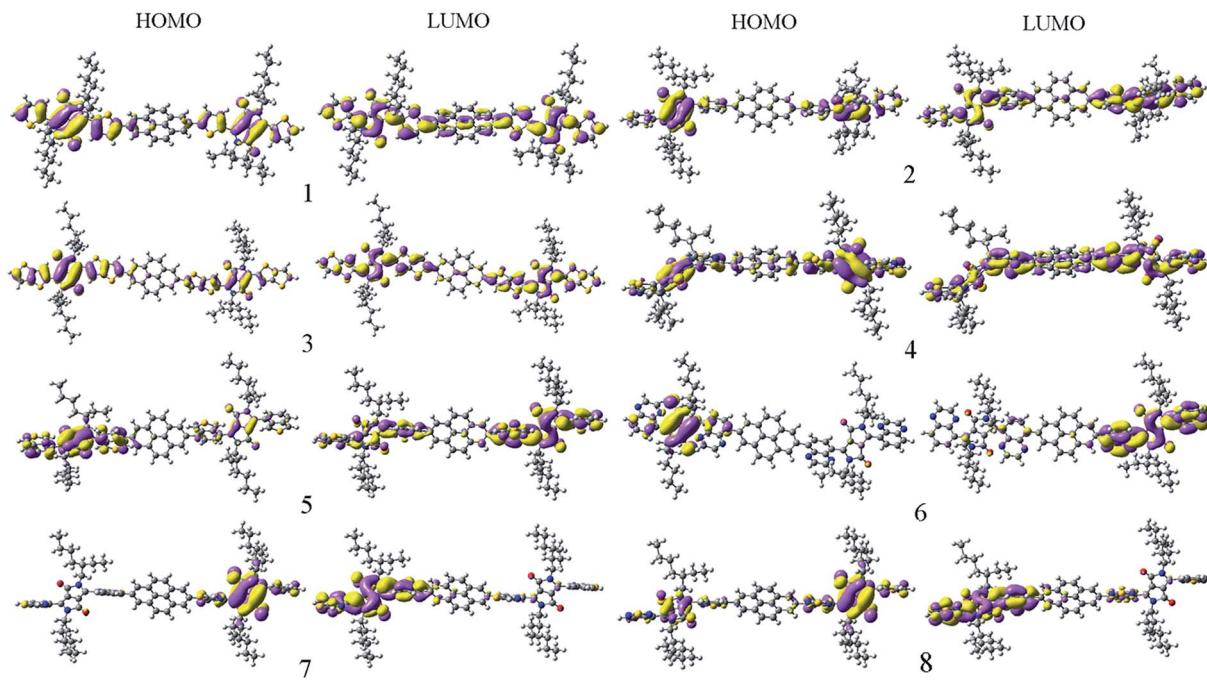


Fig. 1 The electronic density contours of the frontier orbital for the studied compounds at the PBE0/6-31G(d,p) level.

about 485 and 601 nm of molecules **7** and **8** are mainly derived from HOMO \rightarrow LUMO, HOMO-1 \rightarrow LUMO, and HOMO-1 \rightarrow LUMO+1 transitions, respectively. Comparing the λ_{abs} of molecules **2-8** with that of molecule **1**, molecules **3**, **4**, and **8** exhibit bathochromic shifts 31, 98, and 18 nm, respectively. However, molecules **2** and **5-7** show hypsochromic shifts 101, 59, 53, and 98 nm, respectively. The λ_{abs} sequence for **1-8** is **4** $>$ **3** $>$ **8** $>$ **1** $>$ **6** $>$ **5** $>$ **7** $>$ **2**. It suggests that the λ_{abs} of molecules **3**, **4**, and **8** exhibit bathochromic shifts, whereas the λ_{abs} of molecules **2**, **5**, **6**, and **7** show hypsochromic shifts compared with that of molecule **1**, respectively. Furthermore, the designed molecules show large f values (0.90–2.33). Molecules **2-5** have slightly

larger f values than those of molecules **6-8**, respectively. This indicates that molecules **2-5** exhibited slightly larger absorption intensity than those of molecules **6-8**. Apparently, through the introduction of different aromatic heterocyclic groups to the side of DPP molecules backbones, their absorption spectrum of

Table 3 The electronic transition, absorption wavelengths λ_{abs} (in nm), the oscillator strength f , and main assignments (coefficient) of **1-8** in chloroform at the TD-PBE0/6-31G(d,p)/PBE0/6-31G(d,p) level, along with available experimental data

Molecules	λ_{abs}	f	Assignment
1	583	2.33	$\text{H} \rightarrow \text{L}$ (0.66) $\text{H}-1 \rightarrow \text{L}+1$ (0.25)
2	482	1.37	$\text{H} \rightarrow \text{L}$ (0.58) $\text{H}-1 \rightarrow \text{L}+1$ (0.30)
3	614	2.93	$\text{H} \rightarrow \text{L}$ (0.62) $\text{H}-1 \rightarrow \text{L}+1$ (0.32)
4	681	1.59	$\text{H} \rightarrow \text{L}$ (0.64) $\text{H}-1 \rightarrow \text{L}+1$ (0.27)
5	524	1.66	$\text{H} \rightarrow \text{L}$ (0.49) $\text{H}-1 \rightarrow \text{L}+1$ (0.33)
6	530	1.00	$\text{H} \rightarrow \text{L}$ (-0.36) $\text{H}-1 \rightarrow \text{L}$ (0.44) $\text{H} \rightarrow \text{L}+1$ (-0.32)
7	485	0.93	$\text{H} \rightarrow \text{L}$ (0.33) $\text{H}-1 \rightarrow \text{L}$ (0.61) $\text{H}-1 \rightarrow \text{L}+1$ (0.13)
8	601	0.90	$\text{H} \rightarrow \text{L}$ (0.56) $\text{H}-1 \rightarrow \text{L}$ (-0.36), $\text{H}-1 \rightarrow \text{L}+1$ (0.22)
Exp ^a	589		

^a Experimental results of **1** were taken from ref. 39.

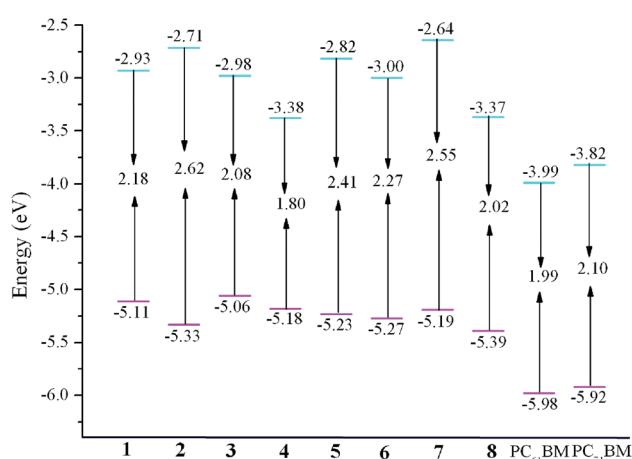


Fig. 2 Evaluation of calculated FMO energies for investigated molecules as well as FMO energies for PC₆₁BM and PC₇₁BM at the PBE0/6-31G(d,p) level.

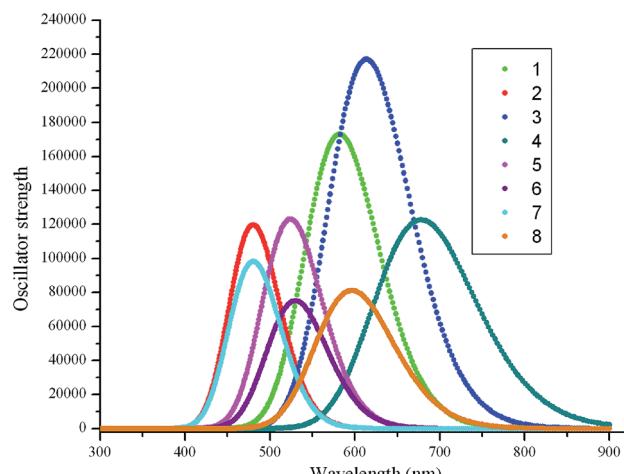


Fig. 3 The calculated absorption spectra of the investigated molecules (value of full width at half maximum is 3000 cm^{-1}).

the studied molecules can be tuned effectively. With the above considerations, these molecules can act as donor material with strong absorption spectra for SMOSCs.

Charge transfer properties

It is commonly known that the AIP, AEA, and λ are three key parameters that determine the charge transfer property. The AIP and AEA are good descriptors to estimate the energy barrier for the injection of holes and electrons, respectively. The λ is the extent to estimate the ability to carry the charge in solid. Usually, lower AIP and higher AEA are favorable for better charge transport ability for hole and electron, respectively.^{61,62} For λ , the lower λ_e and λ_h are beneficial for enhancement the electron and hole transfer rate, respectively.^{54,55}

The predictions of AIP, AEA, λ_h , and λ_e of the designed molecules are summarized in Table 4. It is clear that the AIP are in the order of **8** > **2** > **6** > **5** > **7** > **1** > **4** > **3**. The AIP of molecules **2** and **5-8** are larger, whereas the AIP of molecules **3** and **4** are smaller than that of molecule **1**, respectively. Namely, comparing the AIP of **2-8** with that of **1**, molecules **2** and **5-8** can raise the AIP, while molecules **3** and **4** can lower the AIP, respectively. At the same time, the AEA is in the order of **4** > **8** > **1** > **6** > **5** > **2** > **7**. Obviously, comparing the AEA of molecules **2-**

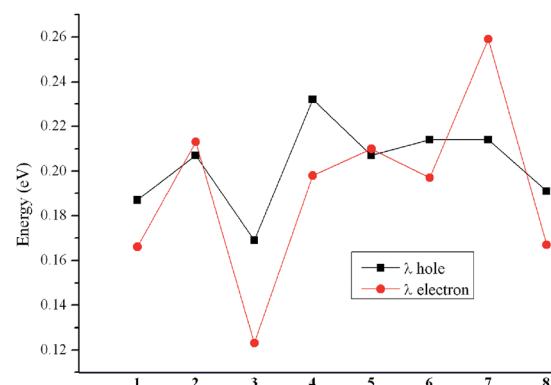


Fig. 4 Graphical representation of λ_e and λ_h calculated at the PBE0/6-31G(d,p) level of theory.

8 with that of molecule **1**, molecules **3**, **4**, and **8** increase the AEA, whereas molecules **2** and **5-7** decrease the AEA, respectively. It is noticeable that the introduction of different aromatic heterocyclic groups to the molecules affects both their AIP and AEA values.

A graphical representation of λ_h and λ_e has been illustrated in Fig. 4 to represent the results more clearly. Generally, Alq3 (tris(8-hydroxyquinolinato)aluminum(III), $\lambda_e = 0.276\text{ eV}$) and TPD (*N,N*-diphenyl-*N,N*'-bis(3-methylphenyl)-(1,1'-biphenyl)-4,4'-diamine, $\lambda_h = 0.290\text{ eV}$) are taken as the typical electron and hole transport materials, respectively.^{63,64} Apparently, the λ_e and λ_h of the designed compounds are much smaller than those of Alq3 and TPD, respectively. It indicates that both the electron and hole transfer rates of **1-8** might be higher than those of Alq3 and TPD, respectively. The decreasing sequence of λ_h is **4** > **6** ≈ **7** > **2** ≈ **5** > **8** > **1** > **3**, and the values of λ_e are in the following order: **7** > **2** > **5** > **4** > **6** > **8** > **1** > **3**. Apparently, comparing the λ_h and λ_e of molecule **2-8** with those of **1**, molecules **2**, **4-8** can increase both λ_h and λ_e , whereas compound **3** can decrease both λ_h and λ_e , respectively. Additionally, molecule **3** has the smallest λ_h and λ_e values, indicating that molecule **3** has the largest hole and electron transport rates among the investigated molecules. Moreover, the differences between λ_e and λ_h values of **1-8** do not exceed 0.046 eV. It indicates that they exhibit better hole- and electron-transporting balance. Accordingly, these molecules can act as nice ambipolar charge transport materials for SMOSCs. Therefore, these molecules can be used as good candidates for ambipolar charge transport materials under the proper operating conditions.

Conclusions

In this article, we design a series of novel diketopyrrolopyrrole-pyrene-based molecules as the acceptor materials for SMOSCs applications. The optical, electronic, and charge transporting properties were investigated by applying DFT and TD-DFT methods. The parameters such as ΔE_{L-L} , λ , AIP, and AEA were calculated to preliminarily evaluate the performance of the solar cell. The results show that the E_{HOMO} , E_{LUMO} , E_g , ΔE_{L-L} , AIP, AEA, and λ can be tuned through the introduction of different

Table 4 Calculated molecular λ_e , λ_h , AIP, AEA and η (all in eV) of **1-8** at the PBE0/6-31G(d,p) level

Molecules	λ_h	λ_e	AIP	AEA	η
1	0.187	0.166	5.716	2.008	1.854
2	0.207	0.213	5.937	1.746	2.096
3	0.169	0.123	5.603	2.108	1.747
4	0.232	0.198	5.715	2.382	1.667
5	0.207	0.210	5.789	1.826	1.981
6	0.214	0.197	5.877	1.935	1.971
7	0.214	0.259	5.780	1.694	2.051
8	0.191	0.167	5.981	2.286	1.848



aromatic heterocyclic groups to the molecules. The absorption spectra were also simulated. Our results suggest that the designed molecules possess suitable FMOs to match those of typical acceptors PC₆₁BM and PC₇₁BM. Additionally, they can be used as ambipolar charge transport materials for SMOSCs.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

Financial supports from the NSFC (No. 21563002) are gratefully acknowledged.

Notes and references

- 1 L. Lu, T. Zheng, Q. Wu, A. M. Schneider, D. Zhao and L. Yu, *Chem. Rev.*, 2015, **115**, 12666–12731.
- 2 W. Cao and J. Xue, *Energy Environ. Sci.*, 2014, **7**, 2123–2144.
- 3 K. Mazzio and C. K. Luscombe, *Chem. Soc. Rev.*, 2015, **44**, 78–90.
- 4 K. H. Hendriks, G. H. L. Heintges, V. S. Wienk, M. M. Gevaerts and R. A. J. Janssen, *Angew. Chem., Int. Ed.*, 2013, **52**, 8341–8344.
- 5 L. Ye, S. Zhang, L. Huo, M. Zhang and J. Hou, *Acc. Chem. Res.*, 2014, **47**, 1595–1603.
- 6 Q. R. Yin, J. S. Miao, Z. Wu, Z. F. Chang, J. L. Wang, H. B. Wu and Y. Cao, *J. Mater. Chem. A*, 2015, **3**, 11575–11586.
- 7 H. Yao, L. Ye, H. Zhang, S. Li, S. Zhang and J. Hou, *Chem. Rev.*, 2016, **116**, 7397–7457.
- 8 Y. Chen, X. Wan and G. Long, *Acc. Chem. Res.*, 2013, **46**, 2645–2655.
- 9 A. Mishra and P. Bäuerle, *Angew. Chem., Int. Ed.*, 2012, **51**, 2020–2067.
- 10 Y. Lin, Y. Li and X. Zhan, *Chem. Soc. Rev.*, 2012, **41**, 4245–4272.
- 11 B. Kan, M. Li, Q. Zhang, F. Liu, X. Wan, Y. Wang, W. Ni, G. Long, X. Yang, H. Feng, Y. Zuo, M. Zhang, F. Huang, Y. Cao, T. P. Russell and Y. Chen, *J. Am. Chem. Soc.*, 2015, **137**, 3886–3893.
- 12 J. H. Kim, C. E. Song, N. Shin, H. Kang, S. Wood, I. N. Kang, B. Kim, B. Kim, J. S. Kim, W. S. Shin and D. H. Hwang, *ACS Appl. Mater. Interfaces*, 2013, **5**, 12820–12831.
- 13 Z. He, B. Xiao, F. Liu, H. Wu, Y. Yang, S. Xiao, C. Wang, T. P. Russell and Y. Cao, *Nat. Photonics*, 2015, **9**, 174–179.
- 14 Y. Lin and X. Zhan, *Acc. Chem. Res.*, 2016, **49**, 175–183.
- 15 D. Ni, B. Zhao, T. Shi, S. Ma, G. Tu and H. Wu, *ACS Macro Lett.*, 2013, **2**, 621–624.
- 16 J. Li, K. H. Ong, S. L. Lim, G. M. Ng, H. S. Tan and Z. K. Chen, *Chem. Commun.*, 2011, **47**, 9480–9482.
- 17 J. E. Coughlin, Z. B. Henson, G. C. Welch and G. C. Bazan, *Acc. Chem. Res.*, 2014, **47**, 257–270.
- 18 Y. Huang, M. Zhang, H. Chen, F. Wu, Z. Cao, L. Zhang and S. Tan, *J. Mater. Chem. A*, 2014, **2**, 5218–5223.
- 19 M. Knupfer, *Appl. Phys. A*, 2003, **77**, 623–626.
- 20 I. Hill, A. Kahn, Z. Soos and R. Pascal Jr, *Chem. Phys. Lett.*, 2000, **327**, 181–188.
- 21 C. He, Q. G. He, X. D. Yang, G. L. Wu, C. H. Yang, F. L. Bai, Z. G. Shuai, L. X. Wang and Y. F. Li, *J. Phys. Chem. C*, 2007, **111**, 8661–8666.
- 22 M. Lenes, G. A. H. Wetzelaer, F. B. Kooistra, S. C. Veenstra, J. C. Hummelen and P. W. M. Blom, *Adv. Mater.*, 2008, **20**, 2116–2119.
- 23 M. Cheng, C. Chen, X. Yang, J. Huang, F. Zhang, B. Xu and L. Sun, *Chem. Mater.*, 2015, **27**, 1808–1814.
- 24 E. Wang, L. Hou, Z. Wang, S. Hellström, F. Zhang, O. Inganäs and M. R. Andersson, *Adv. Mater.*, 2010, **22**, 5240–5244.
- 25 A. Tang, C. Zhan and J. Yao, *Chem. Mater.*, 2015, **27**, 4719–4730.
- 26 E. Wang, W. Mammo and M. R. Andersson, *Adv. Mater.*, 2014, **26**, 1801–1826.
- 27 P. Zhou, D. Dang, M. Xiao, Q. Wang, J. Zhong, H. Tan, Y. Pei, R. Yang and W. Zhu, *J. Mater. Chem. A*, 2015, **3**, 10883–10889.
- 28 H. Qin, L. Li, F. Guo, S. Su, J. Peng, Y. Cao and X. Peng, *Energy Environ. Sci.*, 2014, **7**, 1397–1401.
- 29 S. Chen, L. Xiao, X. Zhu, X. Peng, W. K. Wong and W. Y. Wong, *Chem. Commun.*, 2015, **51**, 14439–14442.
- 30 Y. Li, R. G. Clevenger, L. Jin, K. V. Kilway and Z. Peng, *J. Mater. Chem. C*, 2014, **2**, 7180–7183.
- 31 J. L. Wang, F. Xiao, J. Yan, Z. Wu, K. K. Liu, Z. F. Chang, R. B. Zhang, H. Chen, H. B. Wu and Y. Cao, *Adv. Funct. Mater.*, 2016, **26**, 1803–1812.
- 32 K. Gao, J. Miao, L. Xiao, W. Deng, Y. Kan, T. Liang, C. Wang, F. Huang, J. Peng and Y. Cao, *Adv. Mater.*, 2016, **28**, 4727–4733.
- 33 J. H. Liu, B. Walker, A. Tamayo, Y. Zhang and T. Q. Nguyen, *Adv. Funct. Mater.*, 2013, **23**, 47–56.
- 34 X. Liu, G. Luo, X. Cai, H. Wu, S. J. Su and Y. Cao, *RSC Adv.*, 2015, **5**, 83155–83163.
- 35 J. Mei, K. R. Graham, R. Stalder, S. P. Tiwari, H. Cheun, J. Shim, M. Yoshio, C. Nuckolls, B. Kippelen, R. K. Castellano and J. R. Reynolds, *Chem. Mater.*, 2011, **23**, 2285–2288.
- 36 O. P. Lee, A. T. Yiu, P. M. Beaujuge, C. H. Woo, T. W. Holcombe, J. E. Millstone, J. D. Douglas, M. S. Chen and J. M. Fréchet, *Adv. Mater.*, 2011, **23**, 5359–5363.
- 37 K. A. Mazzio, M. J. Yuan, K. Okamoto and C. K. Luscombe, *ACS Appl. Mater. Interfaces*, 2011, **3**, 271–278.
- 38 Q. Wang, J. J. van Franeker, B. J. Bruijnaers, M. M. Wienk and R. A. J. Janssen, *J. Mater. Chem. A*, 2016, **4**, 10532–10541.
- 39 H. Huang, L. Xiao, B. Yang, T. Lai, L. Zhang, K. Gao, X. Peng and Y. Cao, *RSC Adv.*, 2016, **6**, 59218–59225.
- 40 J. Chen, L. Duan, M. Xiao, Q. Wang, B. Liu, H. Xia, R. Yang and W. Zhu, *J. Mater. Chem. A*, 2016, **4**, 4952–4961.
- 41 M. J. Frisch, G. W. Trucks, H. B. Schlegel, G. E. Scuseria, M. A. Robb, J. R. Cheeseman, G. Scalmani, V. Barone, B. Mennucci, G. A. Petersson, H. Nakatsuji, M. Caricato, X. Li, H. P. Hratchian, A. F. Izmaylov, J. Bloino, G. Zheng, J. L. Sonnenberg, M. Hada, M. Ehara, K. Toyota, R. Fukuda, J. Hasegawa, M. Ishida, T. Nakajima, Y. Honda, O. Kitao, H. Nakai, T. Vreven, J. A. Montgomery Jr,

J. E. Peralta, F. Ogliaro, M. Bearpark, J. J. Heyd, E. Brothers, K. N. Kudin, V. N. Staroverov, T. Keith, R. Kobayashi, J. Normand, K. Raghavachari, A. Rendell, J. C. Burant, S. S. Iyengar, J. Tomasi, M. Cossi, N. Rega, J. M. Millam, M. Klene, J. E. Knox, J. B. Cross, V. Bakken, C. Adamo, J. Jaramillo, R. Gomperts, R. E. Stratmann, O. Yazyev, A. J. Austin, R. Cammi, C. Pomelli, J. W. Ochterski, R. L. Martin, K. Morokuma, V. G. Zakrzewski, G. A. Voth, P. Salvador, J. J. Dannenberg, S. Dapprich, A. D. Daniels, O. Farkas, J. B. Foresman, J. V. Ortiz, J. Cioslowski, and D. J. Fox, *Gaussian 09*, Gaussian, Inc., Wallingford, CT, 2009.

42 J. P. Cornard and C. Lapouge, *J. Phys. Chem. A*, 2006, **110**, 7159–7166.

43 J. Wu, Y. H. Kan, Y. Wu and Z. M. Su, *J. Phys. Chem. C*, 2013, **117**, 8420–8428.

44 J. Ku, Y. Lansac and Y. H. Jang, *J. Phys. Chem. C*, 2011, **115**, 21508–21516.

45 G. Zhang and C. B. Musgrave, *J. Phys. Chem. A*, 2007, **111**, 1554–1561.

46 C. Lee, W. Yang and R. G. Parr, *Phys. Rev. B*, 1988, **37**, 785–789.

47 C. Adamo and V. Barone, *J. Chem. Phys.*, 1999, **110**, 6158–6170.

48 Y. Tawada, T. Tsuneda, S. Yanagisawa, T. Yanai and K. Hirao, *J. Chem. Phys.*, 2004, **120**, 8425–8433.

49 J. D. Chai and M. Head-Gordon, *Phys. Chem. Chem. Phys.*, 2008, **10**, 6615–6620.

50 Y. Zhao and D. G. Truhlar, *Theor. Chem. Acc.*, 2008, **120**, 215–241.

51 T. Yanai, D. P. Tew and N. C. Handy, *Chem. Phys. Lett.*, 2004, **393**, 51–57.

52 I. Jeon, C. Delacou, T. Nakagawa and M. Y. Yutaka, *Chem.-Asian J.*, 2016, **11**, 1268–1272.

53 M. Chandrasekharan, M. A. Reddy, K. Ganesh, G. D. Sharma, S. P. Singh and J. L. Rao, *Org. Electron.*, 2014, **15**, 2116–2125.

54 R. A. Marcus, *Rev. Mod. Phys.*, 1993, **65**, 599–610.

55 R. A. Marcus, *Annu. Rev. Phys. Chem.*, 1964, **15**, 155–196.

56 S. Di Motta, E. Di Donato, F. Negri, G. Orlandi, D. Fazzi and C. Castiglioni, *J. Am. Chem. Soc.*, 2009, **131**, 6591–6595.

57 D. P. McMahon and A. Trois, *J. Phys. Chem. Lett.*, 2010, **1**, 941–946.

58 M. E. Köse, W. J. Mitchell, N. Kopidakis, C. H. Chang, S. E. Shaheen, K. Kim and G. Rumbles, *J. Am. Chem. Soc.*, 2007, **129**, 14257–14270.

59 M. C. Scharber, D. Wuhlbacher, M. Koppe, P. Denk, C. Waldauf, A. J. Heeger and C. L. Brabec, *Adv. Mater.*, 2006, **18**, 789–794.

60 J. L. Brédas, D. Beljonne, V. Coropceanu and J. Cornil, *Chem. Rev.*, 2004, **104**, 4971–5004.

61 C. C. Liu, S. W. Mao and M. Y. Kuo, *J. Phys. Chem. C*, 2010, **114**, 22316–22321.

62 G. Li, R. Zhu and Y. Yang, *Nat. Photonics*, 2012, **6**, 153–161.

63 B. C. Lin, C. P. Cheng, Z. Q. You and C. P. Hsu, *J. Am. Chem. Soc.*, 2005, **127**, 66–67.

64 N. E. Gruhn, D. A. da Silva Filho, T. G. Bill, M. Malagoli, V. Coropceanu, A. Kahn and J. L. Bredas, *J. Am. Chem. Soc.*, 2002, **124**, 7918–7919.

