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Theoretical investigations of thermoelectric phenomena in binary semiconducting skutterudites

Muhammad Saeed, pa Banaras Khan,*bcd Iftikhar Ahmad,bc Awais Siddique Saleemi,e Najeebur Rehman,f H. A. Rahnamaye Aliabad pg and Sarir Uddind

In this study, we explored the thermoelectric properties of the host thermoelectric materials (TM), namely, binary skutterudites, using a combination of simulations based on density functional theory and post-DFT Boltzmann's semiclassical theory. The calculations were performed close to the Fermi surface for the Seebeck coefficient and other thermoelectric parameters. Our results demonstrated that CoSb₃ exhibited the highest Seebeck value at room temperature among all the compounds (CoP₃, CoAs₃, CoSb₃, IrP₃, IrAs₃, IrSb₃, RhAs₃, and RhSb₃), which confirmed that this compound is an ideal host material for thermoelectric applications. Furthermore, the calculated electrical conductivity values show that RhAs₃ has the largest value of $3.736 \times 10^5 \,\Omega^{-1} \,\mathrm{m}^{-1}$. However, at high temperatures, the Seebeck values for all of these compounds are almost constant due to the activation of the minority charge carriers.

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Introduction

The demand for energy has caused a remarkable unrest across the globe and climate change due to the combustion of fossil fuels is increasing and creating a drastic impact on the social and economic life of human civilization.1 In this context, the conversion of waste heat from various sources to useful electrical energy can play a key role in the energy crisis as well as growing environmental problem. The heating process of buildings, heat from the automobile exhausts as well as industries, all generate a huge amount of waste heat that could be converted to electricity using thermoelectric (TE) generators. Thermoelectric generators are solid devices having no moving parts, make no noise, reliable and scalable, which make them ideal for small amounts of electric power generation.^{1,2} It is estimated that two-third of used energy is lost as waste heat, so there is a need for good TE materials that can be used to convert heat to electrical energy and vice versa.3

Certain materials are identified for their high efficiency of energy conversion from heat to electricity with a high TE figure Skutterudites have attracted significant attention as high temperature TE materials due to their high figure of merit (zT) in the temperature range from 300 °C to 550 °C.⁷ The experimental observations show that skutterudite semiconductors possess attractive transport properties and have good potential for achieving zT values. For p-type skutterudites, low hole effective mass, high carrier mobility, low electrical resistivity and moderate Seebeck coefficient were obtained; however, the larger electron effective mass and Seebeck coefficients make the n-type skutterudites more attractive candidates for TE applications.⁸ The Seebeck coefficient values of all skutterudites are high and therefore can be used as thermoelectric materials.⁹ However, CoSb₃ has attracted enormous attention for its conversion applications of waste heat to electricity due to its reasonable band gap, high carrier mobility and the fact that it is

of merit above unity. These materials have good electrical conductivity, and ultimately maintain a low thermal conductivity. Their interesting results have attracted considerable attention and research is still in progress to further improve their TE properties.4 High-temperature bulk materials, such as skutterudites, clathrates, half-heusler alloys and complex chalcogenides, have been extensively studied for their TE applications.5 These materials usually have complex crystal structures and physical properties that are required for a good TE material. For example, the skutterudite-structured compounds and clathrates have cage-like structures that have empty spaces where filler atoms are inserted, which significantly lower the thermal conductivity of these compounds due to the filling atoms' ability to scatter phonons. These materials possess some unusual phenomena, such as heavy fermion superconductivity, an exciton-mediated superconducting state and Weyl fermions.6

^aCollege of Nuclear Science and Engineering, East China University of Technology, Nanchang, 330013, China

^bCenter for Computational Materials Science, University of Malakand, Chakdara, Pakistan. E-mail: banarasphysicist@gmail.com

Department of Physics, University of Malakand, Chakdara, Pakistan

^dDepartment of Physical and Numerical Sciences, Qurtuba University of Science & Information Technology, Peshawar, Pakistan

^eKey Laboratory of Optoelectronic Devices and Systems of Ministry of Education and Guangdong Province, College of Optoelectronic Engineering, Shenzhen University, Shenzhen 518060, Guangdong, China

Department of Chemistry, University of Malakand, Chakdara, Pakistan

^{*}Department of Physics, Hakim Sabzevari University, Sabzevar, Iran

composed of inexpensive and environmentally friendly elements as compared to other skutterudite materials. The thermal conductivity of pure $CoSb_3$ is, however, very high, which leads to a low zT and poor conversion efficiency for TE applications.¹⁰

The thermoelectric properties of the binary skutterudite have been studied extensively *via* experiments. However, some theoretical studies are available on the filled ones, but no detailed theoretical studies are available on the comparison of the binary skutterudites. The motivation of the present study is an attempt to understand the theory of TE properties of the binary skutterudite compounds.

Theory and calculations

The Boltzmann theory discusses a variation in the charge carriers' distribution function due to external fields, lattice phonon scattering, or various kinds of defect scattering. For the situation of an electric and magnetic fields and a thermal gradient, the electric current j can be written in terms of the tensor form as

$$j_i = \sigma_{ij}E_j + \sigma_{ij}kE_jBk + \nu_{ij}\nabla_jT \tag{1}$$

However, the complexity of various carrier scattering mechanisms causes difficulty in obtaining an exact solution of the Boltzmann transport equation.^{11,12} Mathematically, the relaxation time is given by

$$\frac{\delta f}{\delta t} \left(\frac{\delta f}{\delta t} \right)_{s} = -\frac{f - f_{0}}{\tau} \tag{2}$$

where f_0 and f are the equilibrium and perturbed carrier distribution functions, respectively, and τ is the relaxation time. eqn (2) describes the process of restoring f_0 from f via various scattering processes. The net effect is to restore f to f_0 exponentially with a relaxation time τ . Basically, τ is very complicated; however, in this study, τ has been approximated. Using this approximation, the thermoelectric parameters in the tensor form for a material can be written as

$$\sigma_{\alpha\beta}(T,\mu) = \frac{1}{\Omega} \int \overline{\sigma}_{\alpha\beta}(\varepsilon) \left[-\frac{\sigma f_0(T,\alpha,\mu)}{\delta s} \right] d\varepsilon \tag{3}$$

$$S_{\alpha\beta}(T,\mu) = \frac{1}{eT\Omega\sigma_{\alpha\beta}(T,\mu)} \int \overline{\sigma}_{\alpha\beta}(\varepsilon)(\varepsilon-\mu) \left[-\frac{\sigma f_0(T,\alpha,\mu)}{\delta s} \right] \mathrm{d}\varepsilon$$
(4)

where α and β are the tensor indices, and Ω , μ and f_0 are the volume of the unit cell, Fermi level of carriers, and the carrier Fermi–Dirac distribution function, respectively, and e is the electronic charge. The transport distribution function tensor given in σ and S is defined as:

$$\overline{\sigma}_{\alpha\beta}(\varepsilon) = \frac{e}{N} \sum_{i,k} \tau \nu_{\alpha}(i,k) \nu_{\beta}(i,k) \frac{\delta(\delta - \delta_{i,k})}{\delta_{s}}$$
 (5)

where k and i are the wave vector and band index, respectively, and N represents the number of k points. In the three dimensional tensor, $\nu\alpha(i,k)$ ($\alpha=x,y,z$) is the α -th component of the

group velocity $\nu(i,k)$ of carriers, where it can be derived directly from the band structure by

$$\nu(i,k) = \frac{1}{h} \nabla_{k,s_{i,k}} \tag{6}$$

The so-called band-crossing17 may introduce problems in calculating v(i,k) from band structures. In these calculations, the band structures of these binary skutterudites were calculated in comparison to the optical band gaps obtained by experimental measurements, as given in Fig. 1. The electronic band structure calculations were performed using the improved Tran-Blaha modified Becke-Johnson potential (improved TBmBJ).21 This newly developed technique yields very accurate electronic band structures and band gaps for various types of semiconductors and insulators in comparison to the wellknown Green's function method (GW) and Tran-Blaha modified Becke-Johnson (TB-mBJ) technique. Furthermore, the details about the electronic structure of the binary skutterudites and improved TB-mBJ can be found in our previous study.22 To calculate the thermoelectric parameters, a dense k mesh of 56 000 k points was used in this study as per the method proposed by Madsen and Singh.¹⁷ By using eqn (2)-(5), the Seebeck coefficient and electrical conductivity of a compound can be estimated based on its band structure. The carrier relaxation time t is taken to be a constant with the value approximated to 10-13 s.23 The electrical conductivity and power factor were calculated with respect to τ; the Seebeck coefficient is independent of τ . This approach has been employed successfully in evaluating the electrical transport properties of thermoelectric compounds.14-18,24 Following similar procedures, we have interfaced this methodology with the Wien2k code.25

Results and discussions

Seebeck coefficient

Thermoelectric phenomenon in materials can be quantified by different parameters, one of which is the Seebeck coefficient, which has a unique importance because of its voltage capability for a given temperature gradient. A high Seebeck coefficient value usually leads to a good thermoelectric material. The Seebeck coefficient is a tool that provides a sensitive test of the

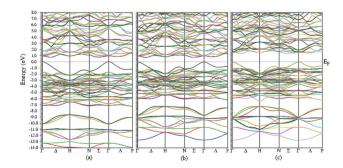


Fig. 1 Calculated band structures of (a) RhAs $_3$, (b) RhSb $_3$ and. (c) IrSb $_3$ by the improved TB-mBJ.

electronic structure in the vicinity of the Fermi energy. A different electronic structure results in different values for the Seebeck coefficient. The temperature dependence of the Seebeck coefficients *S* for the semiconducting binary skutterudites is shown in Fig. 2. Large values, with magnitudes on the order of $\mu V K^{-1}$, were observed in all samples. The calculated values of the Seebeck coefficients at room temperature were in close agreement with experimental values as listed in Table 1. In Fig. 2, the Seebeck coefficient increased with the increase in temperature up to 700 K and tended to flatten at higher temperatures. This is due to an increase in the number of thermally excited minority carriers, which tend to decrease the magnitude of the Seebeck coefficient. For all binary skutterudites an increase in temperature increases the energy per charge and ultimately increases the voltage (i.e., the Seebeck coefficient), and also exhibits positive S values, showing that the majority carriers are holes. In these calculations, the maximum observed value of the Seebeck coefficient was for CoSb₃, which was 215 μ V K⁻¹ at room temperature. The Seebeck coefficient at 50 K for CoP_3 in our calculation was 0.8782 μV K^{-1} . As the temperature rose, the calculated curve for CoP3 linearly increased with temperature to around 44 μ V K⁻¹ in about 300 K and this increase was continuous up to 800 K, at which point the Seebeck coefficient became 103 µV K⁻¹. The Seebeck curve of CoAs₃ first at 50 K was 41 µV K⁻¹, and then it increased to around 145 µV K⁻¹ at about 300 K, and finally increased to about 189 $\mu V K^{-1}$ at 800 K. Our calculated curves of CoSb₃ first increased to around 215 $\mu V~K^{-1}$ at about 300 K, and then this increase reached about 279 $\mu V~K^{-1}$ at 800 K. The calculated curve of IrP3 first abruptly increased to a value of around 199 µV K⁻¹ at about 300 K, and then showed a tendency of small increases and reached about 224 $\mu V~K^{-1}$ at about 800 K. Our calculated curves of IrAs3 estimated the value of the Seebeck coefficient to be about 44.6 µV K⁻¹ at about 50 K, and then increased up to a value of 200 μ V K⁻¹ at 800 K. However, at room temperature, the calculated Seebeck coefficient value was 148 μV K⁻¹. The trend followed by the Seebeck coefficient for IrSb₃ was the same as that for IrP3 and IrAs3, as is clear from Fig. 2. The values of the Seebeck coefficients at the temperatures 50,

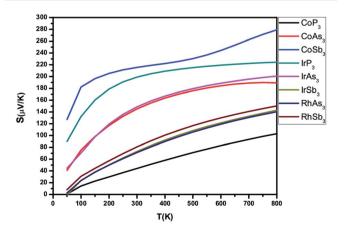


Fig. 2 Seebeck coefficient *versus* the temperature of the binary skutterudites.

300, 800 K are 18.086 μ V K $^{-1}$, 70.0 μ V K $^{-1}$ and 139.4565 μ V K $^{-1}$, respectively. The Seebeck coefficient values for RhAs $_3$ showed a regular increasing trend with temperature and the calculated values were 2.6473 μ V K $^{-1}$, 71.23732 μ V K $^{-1}$ and 140.29174 μ V K $^{-1}$ at 50, 300 and 800 K, respectively. Finally, the results obtained for RhSb $_3$ at the temperatures 50, 300 and 800 K were 8.15587 μ V K $^{-1}$, 80.79364 μ V K $^{-1}$ and 150.18459 μ V K $^{-1}$, respectively.

Electrical conductivity

The electrical conductivity is the passage of the charge carriers through a material. In a semiconductor, both the electrons and holes are responsible for the electrical conductivity. In these compounds, the electrical conductivity almost linearly increased with temperature, which was due to the increase in the number of carriers and potential, as shown in Fig. 3. The electrical conductivity of these compounds was of the order 10^4 $\Omega^{-1}~\rm m^{-1}$.

The calculated and experimental values of the electrical conductivity at room temperature are listed in Table 1 and our comparative study shows that the calculated values of the electrical conductivity agreed with the experimental values. The maximum value of the electrical conductivity in the binary skutterudites was obtained for RhAs3 for all calculated temperatures in the range from 50 to 800 K. The calculated electrical conductivity of CoP₃ at 50 K was $0.6645 \times 10^4 \Omega^{-1}$ m^{-1} , and then linearly increased to 2.06379 \times 10⁴ Ω^{-1} m^{-1} at room temperature. Subsequently, the electrical conductivity increased gradually and reached a peak value of 9.9643×10^4 Ω^{-1} m⁻¹ at 800 K. For CoAs₃ at 50 K, the electrical conductivity was $0.001592 \times 10^4 \, \Omega^{-1} \, m^{-1}$ and reached $0.32 \times 10^4 \, \Omega^{-1} \, m^{-1}$ at 300 K. This increase in the electrical conductivity was continuous up to 4.643 imes 10 4 Ω^{-1} m $^{-1}$ at 800 K. The calculated conductivity for CoSb₃ at 50 K was $0.2385 \times 10^4 \,\Omega^{-1} \,\mathrm{m}^{-1}$ and increased linearly with temperature to a value of 2.05247×10^4 Ω^{-1} m⁻¹ at 300 K. This increase continued until the electrical conductivity reached 9.35315 \times 10⁴ Ω^{-1} m⁻¹ at 800 K. The obtained results for IrP3 at 50 K for the electrical conductivity was $1.58836 \times 10^4 \, \Omega^{-1} \, \text{m}^{-1}$, and the increase in temperature of IrP_3 led to an electrical conductivity value of 3.8848 imes $10^4~\Omega^{-1}~m^{-1}$ at 300 K and 11.83157 \times 10⁴ Ω^{-1} m⁻¹ at 800 K. The electrical conductivity of IrAs₃ at 50 K was $1.4595 \times 10^4 \ \Omega^{-1} \ \text{m}^{-1}$, and then increased to 3.32611 \times 10⁴ Ω^{-1} m⁻¹. The increase in temperature caused a further increase in the electrical conductivity to a value of $8.88461 \times 10^4 \,\Omega^{-1} \,\mathrm{m}^{-1}$ at 800 K. The calculated value of the electrical conductivity for IrSb3 at 50 K was 2.363589 \times $10^5~\Omega^{-1}~m^{-1}$ and became 2.291882 \times $10^5~\Omega^{-1}$ m⁻¹ at 300 K, which was close to the experimental value at this temperature with the value of 2.33 \times ⁵ Ω^{-1} m⁻¹. A further increase in the temperature increased the electrical conductivity and reached 2.8827 \times 10⁵ Ω^{-1} m⁻¹ at 800 K. A calculation for the electrical conductivity of RhAs₃ gave the value of 3.5667 \times $10^5 \ \Omega^{-1} \ \text{m}^{-1}$ at 50 K, which changed by increasing the temperature and became 3.7361 \times 10⁵ Ω^{-1} m⁻¹ at a temperature of 300 K. Finally, the curve reached $4.2289 \times 10^5 \,\Omega^{-1} \, m^{-1}$ at 800 K. The electrical conductivity values for RhSb₃ showed the

	Table 1	Calculated values of the thermoelectric	parameters in co	mparison with ex	perimental values at roon	n temperature
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	$S \left(\mu V K^{-1}\right)$		$\sigma (1 \ \Omega^{-1} \ \mathrm{m}^{-1})$		$\kappa (W m^{-1} K^{-1})$	$\overline{PF \left(W \; \Omega^{-1} \; m^{-2} \; K^{-1}\right)}$
Comp.	Cal.	Exp.	Cal.	Exp.	Cal.	Cal.
CoP ₃	40	$50^a, 40^b$	$2.06 imes 10^4$	$2.1 imes 10^{4b}$	1.8691	$0.4042 imes 10^{-4}$
CoAs ₃	145	145^c , 140^d	$0.32 imes 10^4$	0.41×10^{4c} , 1.19×10^{4b}	0.1625	$0.6835 imes 10^{-4}$
$CoSb_3$	215	120^c , 220^e	2.05×10^4	$2.5 imes 10^{4c,e}, 5.28 imes 10^{4k}$	0.4451	9.5366×10^{-4}
IrP3	199	200^{f}	3.88×10^4	_	0.6892	15.3747×10^{-4}
$IrAs_3$	148	150 ^f	$3.32 imes 10^4$	_	0.5256	7.3490×10^{-4}
$IrSb_3$	73	77^g , 72^h , 80^c , 72^e	2.29×10^{5}	2.33×10^{5e} , 2.331×10^{5k}	1.9087	12.2335×10^{-4}
$RhAs_3$	71	70 ^f	3.736×10^{5}	3.81×10^{5l}	2.9087	18.9600×10^{-4}
$RhSb_3$	80	60 ^f , 80 ^j	2.93×10^5	$2.94 imes 10^{5e}$	2.3363	19.1806×10^{-4}

^a Ref. 26. ^b Ref. 27. ^c Ref. 28. ^d Ref. 29. ^e Ref. 30. ^f Ref. 31. ^g Ref. 32. ^h Ref. 33. ⁱ Ref. 34. ^j Ref. 35. ^k Ref. 36. ^l Ref. 37.

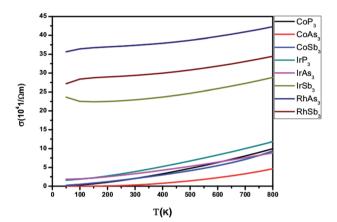


Fig. 3 Electrical conductivity versus the temperature of the binary skutterudites.

same trend and the values at 50, 300 and 800 K were 2.7174 \times $10^5 \,\Omega^{-1} \, \text{m}^{-1}$, 2.9383 × $10^5 \,\Omega^{-1} \, \text{m}^{-1}$ and 3.4463 × $10^4 \,\Omega^{-1} \, \text{m}^{-1}$, respectively.

Electronic thermal conductivity

Thermal conductivity is the flow of heat in materials and it arises from both the electronic and lattice parts of the compound, as shown by k = ke + kl. However, the BoltzTraP code can only estimate the electronic part of the thermal conductivity ke. The electronic thermal conductivity for the semiconducting binary skutterudites is tabulated in Table 1 as well as depicted in Fig. 4. The electronic thermal conductivity variation with temperature was prominent and changed abruptly with temperature. It was found for CoP₃ that the electronic thermal conductivity K at 50 K was 0.23945 W K⁻¹ m⁻¹, which then increased with temperature and became $1.86913~W~K^{-1}~m^{-1}$ at 300 K. The value was increased to 10.81445 W $\mathrm{K}^{-1}~\mathrm{m}^{-1}$ at 800 K. The CoAs_3 calculated electronic thermal conductivity at a temperature of 50 K was 0.1218 W K⁻¹ m^{-1} and was 0.1624 W K^{-1} m^{-1} at room temperature (300 K). With a further increase in the temperature to 800 K, the electronic thermal conductivity then became 5.4099 W K⁻¹ m⁻¹. The CoSb₃ electronic thermal conductivity obtained in our

calculations at 50 K was $0.00646 \text{ W K}^{-1} \text{ m}^{-1}$. At a temperature of 300 K, the thermal conductivity of $CoSb_3$ became 0.44515 W K⁻¹ m⁻¹, and then the electronic thermal conductivity further increased to a value of 9.71322 W K⁻¹ m⁻¹. The calculated values of the electronic thermal conductivity also showed an increase with temperature for IrP3. At the temperatures 50, 300 and 800 K, the thermal conductivity values were 0.02381, 0.68928 and 6.56185 W K⁻¹ m⁻¹, respectively. The calculations performed for IrAs₃ determined that the electronic thermal conductivity at 50 K was 0.02376 W K⁻¹ m⁻¹ and increased with the increase in temperature, by which the resultant values at 300 and 800 K were 0.5256 and 4.8096 W K^{-1} m⁻¹, respectively. The thermal conductivity findings for IrSb₃ showed a similar behavior as discussed for the previous skutterudite materials, and showed a continuous increase with temperature. The calculated values at the temperatures 0, 300 and 800 K were 0.29456, 2.01718 and 10.00149 W K⁻¹ m⁻¹, respectively. The RhAs₃ material had an electronic thermal conductivity value of 0.44776 W K⁻¹ m⁻¹ at 50 K, and then became 2.90874 W K⁻¹ m^{-1} at 300 K and finally 11.8569 W K^{-1} m^{-1} at 800 K. Our calculations for RhSb₃ at 50 K estimated the electronic thermal conductivity value at 0.03378 W K⁻¹ m⁻¹. Increasing the temperature to 300 K, the electronic thermal conductivity value showed an increase to the value 0.68953 W K⁻¹ m⁻¹. At 800 K, the thermal conductivity became 5.87658 W K⁻¹ m⁻¹.

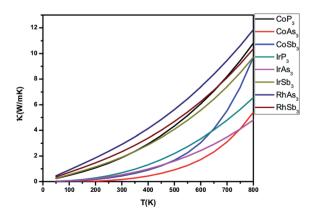


Fig. 4 Variation of electronic thermal conductivity of skutterudites with temperature.

Power factor

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The power factor is one of the parameters used to measure the thermoelectric efficiency of a compound. Mathematically, it is given as PF = $S^2\sigma$, where S is the Seebeck coefficient and σ is the electrical conductivity of the material. The power factor, instead of S or σ , is a comprehensive parameter for the electrical performance of a compound. The calculated values of the power factor for the binary skutterudite compounds are tabulated in Table 1. The maximum power factor in a skutterudite semiconductor system is for RhAs₃, with the peak value of 83.233 × 10^{-4} W Ω^{-1} m⁻² K⁻¹ at 800 K, as shown in Fig. 5. For the compound CoP₃, the power factor at 50 K temperature is very low with a value of 5.1228 × 10^{-10} W Ω^{-1} m⁻² K⁻¹. With the increase in temperature to 300 K power factor becomes 0.4042 × 10^{-4} W Ω^{-1} m⁻² K⁻¹ and finally at 800 K, the power factor of CoP₃ then became 10.5941×10^{-4} W Ω^{-1} m⁻² K⁻¹.

The calculations for CoAs₃ at 50 K gave us a power factor value of $2.66 \times 10^{-11} \,\mathrm{W}\,\Omega^{-1}\,\mathrm{m}^{-2}\,\mathrm{K}^{-1}$. Then, with an increase in the temperature to 300 K, the power factor increased to a value of $0.68351 \times 10^{-4} \text{ W } \Omega^{-1} \text{ m}^{-2} \text{ K}^{-1}$. Finally, at 800 K, the power factor became 16.6251 \times 10⁻⁴ W Ω^{-1} m⁻² K⁻¹. The estimated value of the power factor for CoSb₃ at 50 K was 0.3863×10^{-4} W $\Omega^{-1}~\text{m}^{-2}~\text{K}^{-1}$ and increased with temperature to the values $9.5366\times 10^{-4}~W~\Omega^{-1}~m^{-2}~K^{-1}$ and $72.9146\times 10^{-4}~W~\Omega^{-1}~m^{-2}$ K⁻¹ at 300 and 800 K, respectively. The results of the power factor for IrP3 at the temperatures 50, 300 and 800 K are 1.2846 $\times~10^{-4}~W~\Omega^{-1}~m^{-2}~K^{-1},~15.3747~\times~10^{-4}~W~\Omega^{-1}~m^{-2}~K^{-1}$ and $59.5952 \times 10^{-4} \, \mathrm{W} \, \Omega^{-1} \, \mathrm{m}^{-2} \, \mathrm{K}^{-1}$, respectively. The IrAs₃ values of the power factor showed an usual increase with temperature, and the values obtained at the temperatures 50, 300 and 800 K are $0.3672 \times 10^{-4} \, \text{W} \, \Omega^{-1} \, \text{m}^{-2} \, \text{K}^{-1}$, $7.3490 \times ^{-4} \, \text{W} \, \Omega^{-1} \, \text{m}^{-2} \, \text{K}^{-1}$ and 35.7767 \times 10⁻⁴ W Ω^{-1} m⁻² K⁻¹, respectively.

Our calculations for IrSb₃ gave a value of 0.0011 \times 10⁻⁴ W Ω^{-1} m⁻² K⁻¹, 12.2335 \times 10⁻⁴ W Ω^{-1} m⁻² K⁻¹ and 58.8840 \times 10⁻⁴ W Ω^{-1} m⁻² K⁻¹ for the power factor at 50, 300 and 800 K, respectively. The calculations performed for RhAs₃ showed an increase with temperature, and the results obtained at 50, 300 and 800 K are 0.0250×10^{-4} W Ω^{-1} m⁻² K⁻¹, 18.9600 \times 10⁻⁴ W Ω^{-1} m⁻² K⁻¹, respectively. The power factor for RhSb₃ at the temperatures 50, 300 and 800 K was 0.18076×10^{-4} W Ω^{-1} m⁻² K⁻¹, 19.1806 \times 10⁻⁴ W Ω^{-1} m⁻² K⁻¹ and 77.7349 \times 10⁻⁴ W Ω^{-1} m⁻² K⁻¹, respectively.

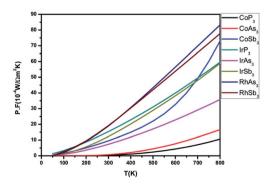


Fig. 5 Power factor at different temperatures for the binary skutterudites.

Conclusions

In conclusion, we theoretically explored the behaviour of the Seebeck coefficient, electrical conductivity, thermal conductivity and power factor of the binary semiconducting skutterdites (CoP₃, CoAs₃, CoSb₃, IrP₃, IrAs₃, IrSb₃, RhAs₃, and RhSb₃), with respect to the temperature using DFT and post-DFT Boltzmann's transport theory. The Seebeck value for CoSb₃ as investigated here was the highest (215 µV K⁻¹) at room temperature among all the compounds under study and this result was in close agreement with the experimental findings; however, the electrical conductivity results for RhAs3 showed the highest electrical conductance of 3.736149 \times 10⁵ Ω^{-1} m⁻¹ among all the compounds. However, the maximum Power factor was obtained for RhSb₃ with a value of 19.1806 \times 10⁻⁴ W Ω^{-1} m⁻² K⁻¹ at room temperature. We calculated the roomtemperature properties as well as the thermoelectric properties in the temperature range from 50 to 800 K for these compounds and compared their results.

Conflicts of interest

There are no conflicts to declare.

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