


 Cite this: *RSC Adv.*, 2019, **9**, 18599

 Received 14th May 2019
 Accepted 7th June 2019

 DOI: 10.1039/c9ra03636a
rsc.li/rsc-advances

Polyurethane ionomers based on amino ethers of *ortho*-phosphoric acid

 I. M. Davletbaeva,^a O. O. Sazonov,^a A. R. Fazlyev,^a R. S. Davletbaev,^b S. V. Efimov^c and V. V. Klochkov^c

The etherification of *ortho*-phosphoric acid with triethanolamine and polyoxypropylene glycol is studied. The reaction process is accompanied by the formation of hyperbranched amino ethers of *ortho*-phosphoric acid terminated by hydroxyl groups. A specific feature of the chemical structure of the compounds obtained is the existence of ion pairs in their structure separated in space. The reaction of the etherification of *ortho*-phosphoric acid with glycols becomes possible through the use of tertiary amines. The amino ethers of *ortho*-phosphoric acid are investigated as a polyol component for the synthesis of polyurethanes with high adhesion characteristics and strength properties. The experimental results presented allow us to relate polyurethanes obtained on the basis of *ortho*-phosphoric acid amino ethers to polymers of ionomeric nature.

1 Introduction

Ionomers are polymeric materials having a small number of ionic side groups in hydrophobic main chains. Ionic fragments are included in these polymers in the form of cation, anion or zwitterion particles in various concentrations. Depending on the type of polymer, ionic groups can be distributed along the main chain periodically, randomly or as end groups.^{1–10} The concentration of ionic groups usually varies in the range of 1–15 mol% and they can be completely or partially neutralized. Intensified interaction between ionic pairs and their clustering increases the density of the nodes of their spatial polymer grid. This feature leads to a high probability of self-organization with formation of various types of nano-and microstructures containing ionic clusters and ionconductive channels.^{11–29} The resulting polymer network also causes strong changes in the mechanical properties of ionomers compared to the original non-ionic polymers. To a certain critical degree, this effect may lead to an increase in abrasion resistance, optical transparency, antistatic properties, interaction with various types of fillers, heat sealability and adhesive properties to various types of materials, such as polymers, ceramics, and metals.

Polyurethanes are a class of highly versatile polymers and their range of use can be further broadened by the introduction of ionic groups into their structures. Therefore, growing

attention has been paid to the synthesis of polyurethane ionomers.^{30–44}

Polyols bearing ionic or ionizable functions are particularly useful as reactive components for the synthesis of self emulsifying functionalized soft polyurethane segments.^{45,46} On the basis of triethanolamine, boric acid and polyoxyethyleneglycols, hyperbranched ethers of boric acid used as the basic compounds for the preparation of polyurethanes of ionomeric nature were obtained.^{47,48} There exist also significant opportunities to study the properties of new polyurethane ionomers containing mixed soft segments involving ionic groups.⁴⁹

Despite a variety of studies of polyurethane ionomers, however, there are relatively few publications related to the synthesis of polyurethane ionomers containing phosphoric acid units.^{50–54}

Promising polyol components for the production of polyurethane ionomers are *ortho*-phosphoric ethers. There exist many ways to synthesize them.^{55–57} The most widely used phosphorylating agent for alcohols is phosphorus oxychloride, a characteristic feature of which is high toxicity. The direct use of affordable and low-toxic H_3PO_4 (OPA) for the production of *ortho*-phosphoric acid ethers is limited due to its low reactivity with respect to hydroxyl-containing compounds. As a rule, the etherification reaction takes place in this case at high temperatures. The condensation of OPA and hydroxyl-containing compounds at relatively low temperatures was first carried out by means of catalytic exposure to a given reaction system of equimolar OPA amounts of trialkylamines.^{58–60} In those works the reaction processes leading to obtaining OPA mono-phosphate were studied. The etherification was carried out in an organic solvent medium. To achieve high conversions, catalysts were additionally introduced, among which there were

^aKazan National Research Technological University, 68 Karl Marx Str., Kazan, Republic of Tatarstan 420015, Russian Federation. E-mail: davletbaeva09@mail.ru; sazonov.oleg2010@gmail.com

^bKazan National Research Technical University named after A. N. Tupolev-KAI, 10 Karl Marx Str., Kazan, Republic of Tatarstan 420111, Russian Federation

^cKazan Federal University, 18 Kremlyovskaya Str., Kazan, Republic of Tatarstan 420008, Russian Federation



imidazoles and derivatives of rhenium oxides (VII). At the same time, the complete etherification of OPA was not considered there.

Our paper demonstrates an evidently simple strategy to synthesis of amino ethers of *ortho*-phosphoric acid (AEPA) by etherification of OPA with triethanolamine and polyoxypropylene glycol. Due to the branched topological structure containing terminal hydroxyl groups, the incorporation of the polyether component in the AEPA composition and the presence of PO^- anions with adjustable content in them, these compounds turned out to be an advanced polyol basis for the synthesis of new polyurethane ionomers. The described new strategy of synthesis may allow industrial production of such polymers in the future.

2 Experimental

2.1 Materials

Polypropylene glycol (PPG), (Wanol 2310) was purchased from Wanhua Chemical (Beijing, China). Triethanolamine (TEOA), triethylamine (TEA), toluene were obtained from Ltd. "Component-reakтив" (Moscow, Russia). 85% aqueous solution of *ortho*-phosphoric acid was purchased from Ltd "MCD-Chemicals" (Moscow, Russia). Polyisocyanate "Wannate PM-200" (PIC) was purchased from Kumho Mitsui Chemicals, Inc. (China).

2.2 Synthetic procedures

General procedure for synthesis of amino ethers of *ortho*-phosphoric acid (AEPA). The etherification reaction was carried out in one stage by reacting OPA with TEOA and PPG. To obtain AEPA, triethanolamine, *ortho*-phosphoric acid and PPG were used at their molar ratios [TEOA] : [H₃PO₄] : [PPG] = 1 : 3 : 6, 1 : 6 : 6, 1 : 9 : 6 (AEPA-3, AEPA-6 and AEPA-9, respectively). The calculated amount of OPA and PPG was placed in a round-bottom flask, mixed for two minutes, then TEOA was added to the reaction system. Within two hours, the reaction mass was stirred at $T = 80^\circ\text{C}$ and a residual pressure of 0.7 kPa. The synthesized liquid AEPA-PPG was collected into a stoppered flask. The amount of residual water did not exceed 0.3 wt%.

The etherification of OPA with TEA and PPG was carried out similarly to the synthesis of AEPA. Compounds were obtained at molar ratios [TEA] : [H₃PO₄] : [PPG] = 1 : 3 : 6 (EPA-3), 1 : 3 : 7 (EPA-3*), 1 : 6 : 6 (EPA-6) and 1 : 9 : 6 (EPA-9). The reaction was quenched after the desired amount of hydroxylation toward the target product. Reaction progress was monitored by titration to determine a hydroxyl groups concentration.

General procedure for synthesis of polyurethanes based on amino ethers of *ortho*-phosphoric acid. The synthesized AEPA (1 g) was mixed with PIC (1 g), then 4.6 mL of toluene was added, stirred for 5 minutes at room temperature, and cast onto prepared surfaces. The curing of polyurethanes was also carried out at room temperature. 24 hours after the final curing, the samples were heated for 10 minutes at 100 °C to remove residual solvent.

2.3 Light-scattering of AEPA solution

Dynamic light scattering experiments were carried out on Zetasizer Nano ZS (Malvern, Great Britain). This instrument has 4 mV He-Ne laser, which works on the 632.8 nm wavelength. Measurements were carried out at the 173° detection angle. The experiments were carried out at 25 °C in the disposable plastic cuvettes of 1 cm path length.

2.4 Fourier transform infrared spectroscopy analysis (FTIR)

The FTIR spectra of the products were recorded on an InfraLUM FT 08 Fourier transform spectrometer (Lumex, St. Petersburg, Russia) using the attenuated total reflection technique. The spectral resolution was 2 cm⁻¹, and the number of scans was 60.

2.5 Measurements of critical micelle concentration (CMC)

The droplet counting method was used to determine the surface tension (σ). The basis of the calculations is the law, where the weight of the drop that comes off the pipette is proportional to the surface tension of the fluid and the radius of the pipette (R): $m = 2\pi R\sigma/g$, where: g is acceleration of gravity; m is the drop mass of the test liquid.

2.6 NMR spectroscopy

¹H and ³¹P NMR spectra were obtained on a BrukerAvance II 500 spectrometer (500.13 MHz for ¹H and 202.46 MHz for ³¹P) using an inverse TBI probehead (1H/31P-BB-2D) or a direct BBO probehead (BB-1H-2D). The samples contained 120 μL of the investigated polymer mix and 480 μL of toluene-d₈. The spectra were recorded at 25 °C; separate sets of temperature measurements were carried out in the range from 6 °C to 32 °C. The proton chemical shift scale is referenced with respect to the quintet of the methyl group in toluene at 2.09 ppm; for ³¹P, the external standard signal of 85% H₃PO₄ ($\delta_P = 0$) poured in a capillary and placed into the polymer solution was used (it allowed us to assign the more high-field signal to phosphoric acid).

2.7 Water concentration measurement

Water concentration was measured on a Mettler Toledo V20 volumetric titrator according to Karl Fischer.

2.8 Viscosity measurements

The dynamic viscosity of samples was determined at the temperature 296 K, at atmospheric pressure and using a SVM 3000 Stabinger Viscometer (Anton Paar, Austria) with an error of 0.00005 g cm⁻³.

2.9 Tensile stress-strain measurements

Tensile stress-strain measurements were obtained from the film samples of size 40 mm × 15 mm with Universal Testing Machine Inspekt mini (Hegewald & PeschkeMeß- und Prüftechnik GmbH, Nossen, Germany) at 293 ± 2 K, 1 kN. The crosshead speed was set at 50 mm min⁻¹ and the test continued



until sample failure. Minima of five tests were analyzed for each sample and the average values were reported.

2.10 The coating adhesion on the breakaway

The value of adhesion of the coating to tearing was measured using a PSO-10MG4S (Ltd. Stroibpibor, Chelyabinsk, Russia) device designed to control the adhesion strength of protective and paint coatings with the base using the method of normal tearing of steel disks.

2.11 Thermomechanical analysis (TMA)

The thermomechanical curves of polymer samples were obtained using TMA 402 F (Netzsch, Selb, Germany) thermomechanical analyzer in the compression mode. The sample thickness was 2 mm, and the rate of heating was $3\text{ }^{\circ}\text{C min}^{-1}$ from $-50\text{ }^{\circ}\text{C}$ to $350\text{ }^{\circ}\text{C}$ in the static mode. The load was 2 N.

2.12 Mechanical loss tangent measurements (MLT)

The MLT curves of polymer samples were taken using the dynamic mechanical analyzer DMA 242 (Netzsch, Selb, Germany) in the mode of oscillating load. Force and stress-stain correspondence were calibrated using a standard mass. The thickness of the sample was 2 mm. Viscoelastic properties were measured under nitrogen. The samples were heated from $-50\text{ }^{\circ}\text{C}$ to $350\text{ }^{\circ}\text{C}$ at the rate of $3\text{ }^{\circ}\text{C min}^{-1}$ and frequency of 1 Hz. The mechanical loss tangent was defined as the ratio of the viscosity modulus G'' to the elasticity modulus G' .

3 Results and discussion

3.1 Synthesis and characterization of amino ethers of *ortho*-phosphoric acid

The interaction between TEOA, OPA and PPG was studied at such a stoichiometric ratio, at which under the conditions of formation of total phosphates all three hydroxyl groups of triethanolamine and one hydroxyl group of PPG could potentially be consumed for the etherification reaction. At the same time, the formation of branched amino ethers of *ortho*-phosphoric acid (AEPA-3) was expected, where the tertiary amine is the main branch center and the phosphates play the function of the subsequent branch centers.

To analyze the size of the AEPA-3 obtained, the initial PPG was first investigated. The average hydrodynamic diameter of the PPG particles in toluene was 396 nm (Fig. 1). A relatively

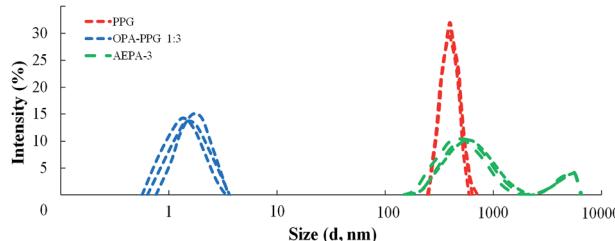


Fig. 1 Particle size distribution in the environment of toluene, determined by dynamic light scattering.

wide particle size distribution indicates the formation of PPG associates in toluene.

In order to determine the role of tertiary amines in the etherification reaction of OPA and PPG, the interaction of 1 mole of OPA and 3 moles of PPG at $80\text{ }^{\circ}\text{C}$ in the absence of a tertiary amine was investigated. For the OPA-PPG system (Fig. 1), the average hydrodynamic particle diameter is 1.7 nm. At the same time, particles with an average size of 396 nm are not revealed. The results obtained can be explained by the fact that OPA, due to the formation of hydrogen bonds involving R-OH and hydroxyl groups in the PPG, destroys the associative OH and hydroxyl groups in the PPG, destroys the associative interactions responsible for the formation of large PPG aggregates, and converts the oligoetherdiol to the monomeric form.

For the AEPA-3 solution, the average hydrodynamic particle diameter is 542 nm with a wide interval of their distribution (from 110 nm to 1100 nm) (Fig. 1). In addition, larger particles are formed with sizes up to 4500 nm. The most likely reason for the formation of such large-sized particles and a wide distribution with relatively small sizes of AEPA-3 may be associated with micelle formation processes. The diagram does not observe particles with small sizes. The fact that the PPG has entered into interaction with the OPA, is evidenced by the fact that the sizes and distribution of particles here significantly exceed the size distribution region for PPG. The absence of free OPA was judged by the absence of particle distribution in the region of small sizes.

The formation of P-O-C bonds in the composition of AEPA-3 is evidenced by the presence of a band at 980 cm^{-1} in the FTIR spectra corresponding to the stretching vibrations of the P-O bond in P-O-C (Fig. 2). The intensity of this band is increasing in the series AEPA-3, AEPA-6 and AEPA-9. The high-intensity bands due to the stretching vibrations of the P-O bond in the structure of the P-OH groups of OPA at 875 cm^{-1} and 950 cm^{-1} practically disappear.

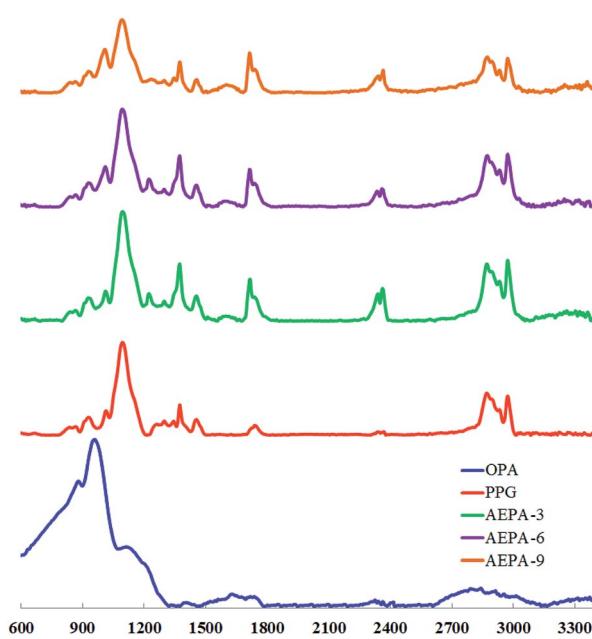


Fig. 2 FTIR spectra.



On the FTIR spectra, bands appear in the region of 2370 cm^{-1} and 2430 cm^{-1} , which correspond to the N-H bond in the composition of the tertiary ammonium. In the spectra of AEPA-3 and AEPA-6, new absorption bands appear in the region of 1720 cm^{-1} and 1745 cm^{-1} , which can reflect the formation of phosphate of the PO^- anion.

To establish the completeness of the reaction by acid groups, their content in AEPA-3 was determined by titration. As a result, 21 mg of KOH was consumed to neutralize free P-OH groups in 1 g of AEPA-3. The consumed amount of KOH corresponds to the equivalent of one P-OH group per each reacted molecule of OPA. The obtained data allow to judge about the existence in the composition of AEPA-3 of secondary acid phosphates of polyoxypropylene glycol.

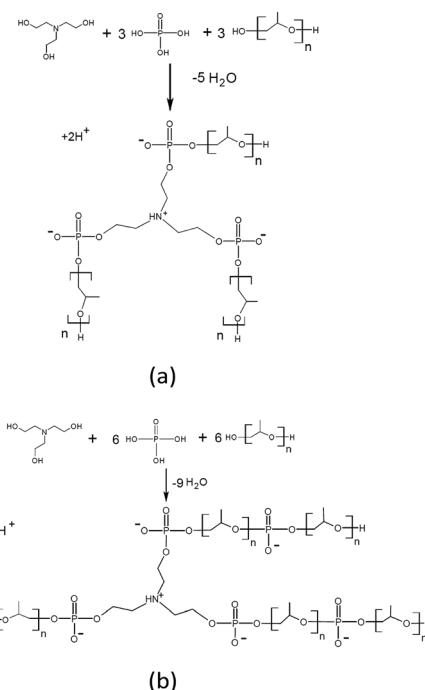
As a result, almost half of the used PPG in the composition of the product of the interaction of 1 mole of TEOA, 3 moles of OPA and 6 moles of PPG remains in its original, unreacted state. This circumstance explains the wide particle size distribution for AEPA-3 (Fig. 1).

Thus, the analysis of the results obtained suggests that the presence of triethanolamine is a condition for the etherification reaction of one P-OH group of *ortho*-phosphoric acid with glycols at $T = 75\text{--}80\text{ }^\circ\text{C}$ without the need for additional use of cocatalysts.

In order to increase the content of acid phosphates in AEPA, the products of interaction of TEOA, OPA and PPG were obtained, during the synthesis of which the molar ratio of TEOA to PPG remained constant, and the molar excess of OPA relative to TEOA increased (AEPA-6). In these cases, it was assumed that both hydroxyl groups located at the ends of the PPG can enter into the etherification reaction. 50 mg of KOH were consumed to neutralize free P-OH groups in 1 g of AEPA-6. The amount of KOH consumed corresponds to the equivalent of one P-OH group per each reacted molecule of OPA, that is, the formation of secondary acid phosphates of PPG.

The solution of AEPA-6 in toluene is opalescent due to the formation of micellar particles. When measuring their size using the method of light scattering, the sizes of these particles exceed 5000 nm. In this regard, the measurements were performed in acetone. For AEPA-6 (Fig. 2), the average hydrodynamic particle diameter in acetone is 1030 nm, and the size distribution itself turned out to be relatively narrow – from 850 to 1100 nm. The formation of particles of relatively large sizes can be explained by the fact that for a given molar ratio of the reactants, the PPG reacts along both terminal hydroxyl groups. The observed narrow particle size distribution may be due to the homogeneity of the composition of the interaction products in the reaction system under consideration and the fact that the amount of PPG used completely entered the AEPA-6 (Scheme 1b and Fig. 3).

Thus, as a result of the interaction between TEOA, OPA and PPG, secondary acid phosphates are predominantly formed. In this regard, it should be noted that OPA is a tribasic acid, which is characterized by a large difference in the dissociation constants of each P-OH group. So, *ortho*-phosphoric acid is strong in the first stage ($K_{\text{I}} = 7.52 \times 10^{-3}$), moderate in the second stage ($K_{\text{II}} = 6.31 \times 10^{-8}$) and very weak in the third stage



Scheme 1 Formation of AEPA-3 (a); AEPA-6 (b).

($K_{\text{III}} = 1.26 \times 10^{-12}$).⁶¹ Such a high dissociation constant of the third P-OH group does not allow it to enter into the etherification reaction. Remaining as part of the acidic secondary and primary phosphates of PPG, the third P-OH group, as judged by the data of FTIR-spectroscopy, dissociates, existing in AEPA in the form of the PO^- anion.

In order to confirm the formation of spatially separated ion pairs in the AEPA structure, the concentration dependences of the surface tension in aqueous solutions were measured for these compounds (Fig. 4). Since polyoxypropylene glycol exhibits the properties of surfactants, surface-active properties were also measured for PPG.

To confirm the proposed variants of the AEPA structure, ^1H NMR and ^{31}P NMR spectroscopy were used. In connection with the established structure of AEPA, the main issue is related to the existence in the resulting compounds of $\text{CH}_2\text{--OH}$ and OH groups with associated protons, PO^- and P-O-C groups.

For this purpose, ^1H NMR spectra of AEPA-3 and AEPA-9 were obtained at different temperatures (Fig. 5a). The existence of $-\text{OH} + \text{H}^+ \leftrightarrow -\text{OH}_2^+$ exchange interactions causes a blurred signal of 5.6–5.4 ppm at $+6\text{ }^\circ\text{C}$, which with increasing

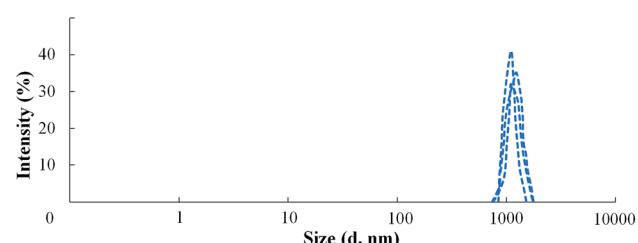


Fig. 3 Particle size distribution for AEPA-6 in the environment of acetone, determined by dynamic light scattering.



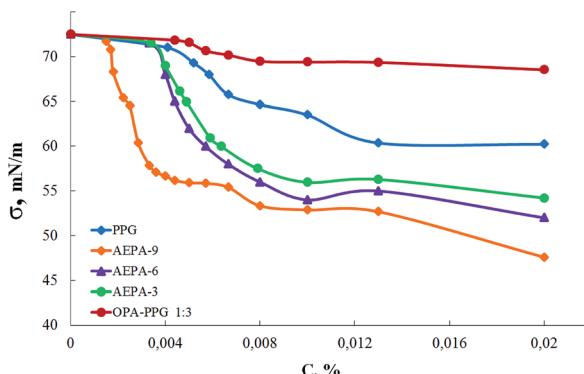


Fig. 4 Surface tension isotherms.

temperature to $+14\text{ }^{\circ}\text{C}$ turns into a wide signal in the 5.4 ppm area, and at $+20\text{ }^{\circ}\text{C}$ and $+30\text{ }^{\circ}\text{C}$ turns into a singlet at $\delta = 4.8\text{ ppm}$.

For AEPA-6, a similar shift of signals with increasing temperature is observed in the range $6.5\text{--}6.0\text{ ppm}$ (Fig. 5b).

For PPG under conditions of a low content of OH groups and the absence of their exchange interactions with protons, the corresponding signals on the ^1H NMR spectra do not manifest themselves in the entire temperature range (Fig. 5c).

On the ^1H NMR spectrum of TEOA, an intense signal is observed at $\delta = 5.4\text{ ppm}$, which corresponds to a proton in the structure of hydroxyl groups. The absence of this signal in the structure of AEPA is a consequence of the involvement of all three hydroxyl groups of TEOA in the interaction with OPA.

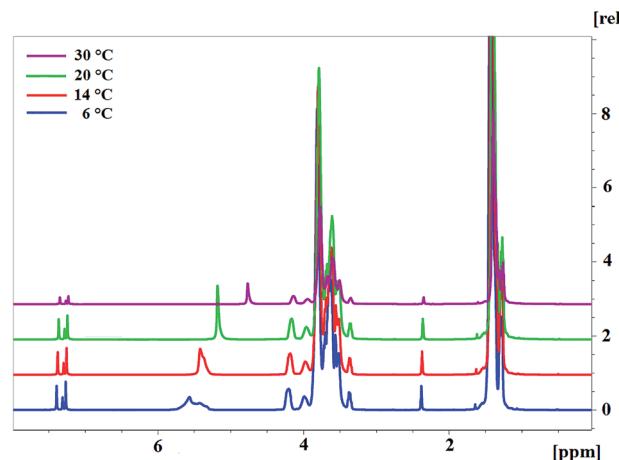
Since AEPA-3 contains the least amount of free protons, a very weak signal, corresponding to $-\text{OH} + \text{H}^+ \leftrightarrow -\text{OH}_2^+$ exchange interactions, appears at $\delta = 4.5\text{ ppm}$ (Fig. 6). In all ^1H NMR spectra of the AEPA, the signal at $\delta = 5.0\text{ ppm}$, characteristic of OPA, does not appear.

The ^{31}P NMR spectra of AEPA (Fig. 7) can be divided into two regions. At $\delta = 1.3\text{--}0.9\text{ ppm}$, the first region corresponds to PO^- anions, the position and manifestation of which undergoes noticeable changes during the transition from AEPA-3 to AEPA-6. The second region at $\delta = 3.0\text{--}3.1\text{ ppm}$ corresponds to the phosphorus bound by the ether linkage. There is a regular increase in signal intensity at $\delta = 3.0\text{--}3.1\text{ ppm}$ from AEPA-3 to AEPA-9.

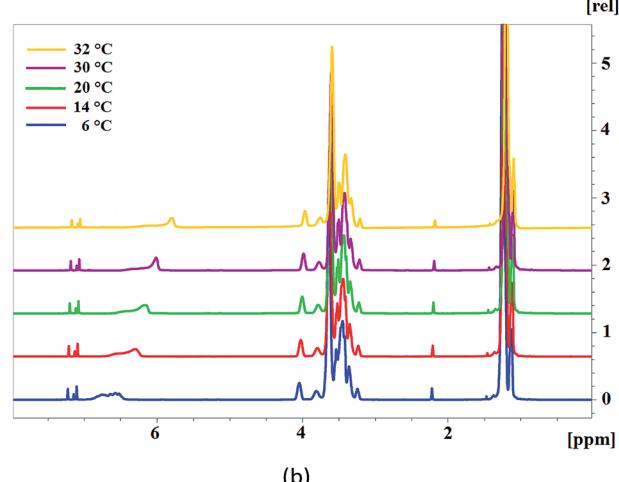
The ^{31}P NMR spectrum was also measured for AEPA-9, AEPA-15 and OPA (Fig. 7). Synthesis of AEPA-9 and AEPA-15 assumed the initial excess of the content of OPA from its theoretically possible entry into the etherification reaction. Despite the large excess of OPA in the composition of AEPA-15, on its ^{31}P NMR spectrum there is no narrow signal at $\delta = 0\text{ ppm}$, which characterizes OPA in the initial state.

The results obtained allow us to conclude that, due to the presence of PO^- anions in the AEPA, these branched compounds are involved in the formation of clusters. As a result of a cooperative nature, unreacted OPA molecules also enter into these interactions. This is also indicated by the expansion and complication of the spectra from AEPA-9 to AEPA-15 in the range $\delta = 0.4\text{--}1.6\text{ ppm}$.

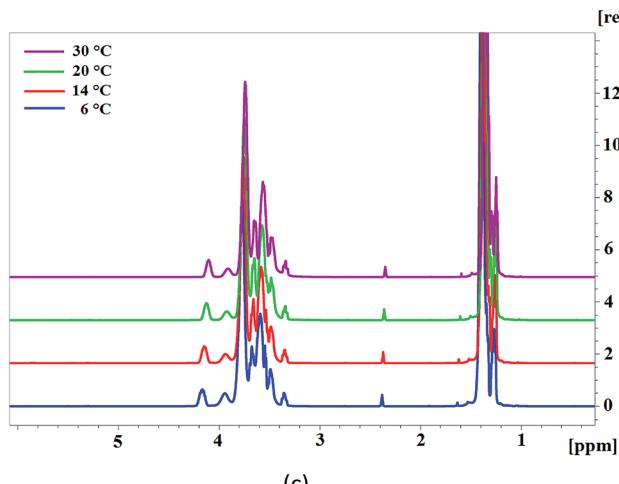
To study the effect of trifunctional TEOA on the topological structure of AEPA, and the effect of tertiary amines on the



(a)



(b)

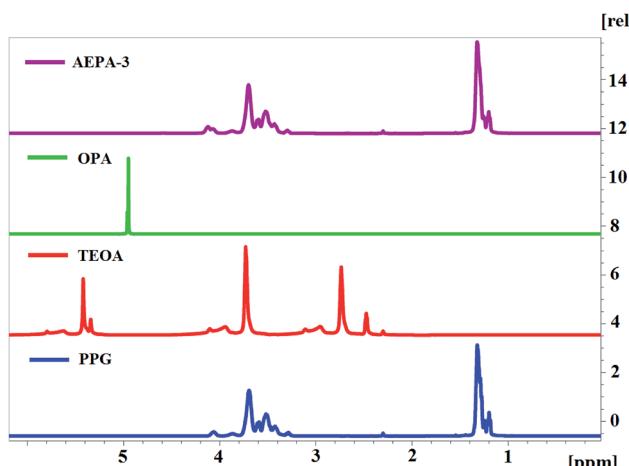
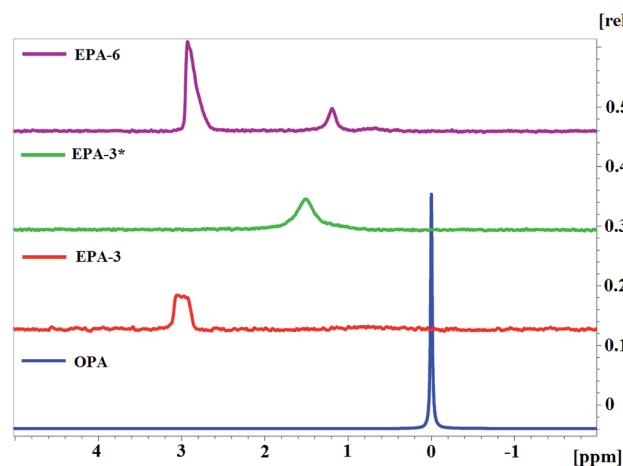
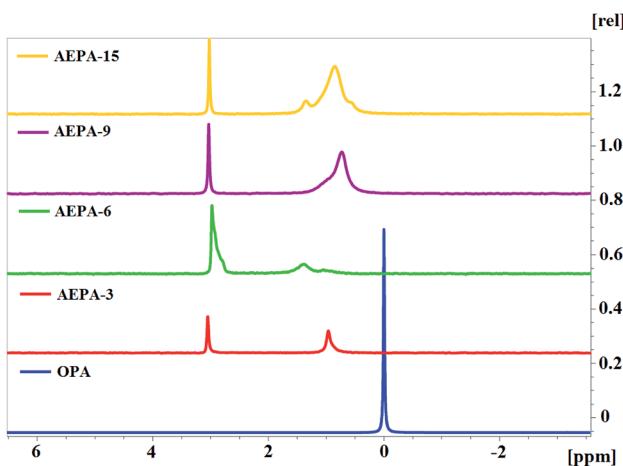


(c)

Fig. 5 ^1H NMR spectra of AEPA-9 (a), AEPA-6 (b), PPG (c) at different temperatures.

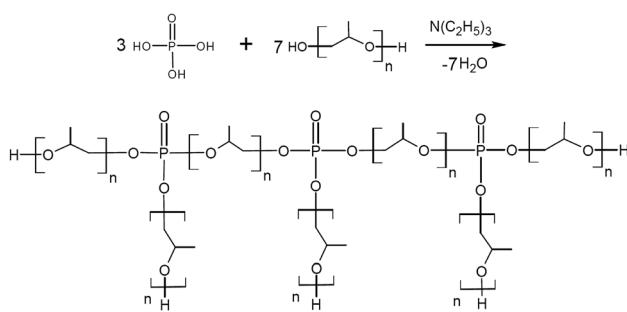
etherification reactions of OPA, triethanolamine was replaced with triethylamine in order to obtain *ortho*-phosphoric acid ethers (EPA).

For EPA-6, two signals are observed on the ^{31}P NMR spectrum, similar to AEPA-6 (Fig. 7). For EPA-3*, obtained at a molar

Fig. 6 ^1H NMR spectra.Fig. 8 ^{31}P NMR spectra.Fig. 7 ^{31}P NMR spectra.

ratio $[\text{TEA}]:[\text{H}_3\text{PO}_4]:[\text{PPG}] = 1:3:7$ (Scheme 2), only one signal is observed in the ^{31}P NMR spectrum at $\delta = 1.5$ ppm. For EPA-3, obtained at the molar ratio $[\text{TEA}]:[\text{H}_3\text{PO}_4]:[\text{PPG}] = 1:3:6$, only one signal is also observed in the region $\delta = 3.2\text{--}2.8$ ppm.

On the basis of the spectra obtained, it can be judged that all phosphorus atoms in EPA-3* and EPA-3 are in the form of complete OPA ethers and therefore cannot participate in the clustering processes. In this case, EPA should exhibit

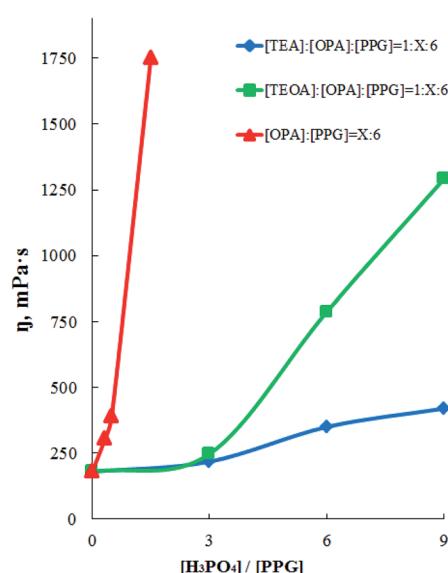


Scheme 2 Formation of EPA-3*.

a topological structure with a large number of branches in comparison with EPA (Fig. 8).

According to the literature,^{62,63} hyperbranched polymers and oligomers are characterized by low viscosity compared with linear analogues. Moreover, their viscosity is much lower than viscosity of linear analogues both in solution and in the molten state. The higher the degree of branching, the lower the viscosity of the branched polymer. In this regard, measurements of the dynamic viscosity of the OPA-PPG, AEPA and EPA system, obtained with a different molar excess of OPA relative to the PPG, were carried out (Fig. 9). For the OPA-PPG system (without using tertiary amines), the viscosity increases significantly even at a relatively low content of *ortho*-phosphoric acid; with a further increase in the molar fraction of OPA, an increase in its viscosity becomes non-additive.

For AEPA, viscosity increases with an increase in the molar excess of OPA relative to PPG up to 3 only insignificantly. With a further increase in the molar excess of OPA, the contribution

Fig. 9 Dependence of dynamic viscosity on molar excess (X) of H_3PO_4 . $T = 20^\circ\text{C}$.

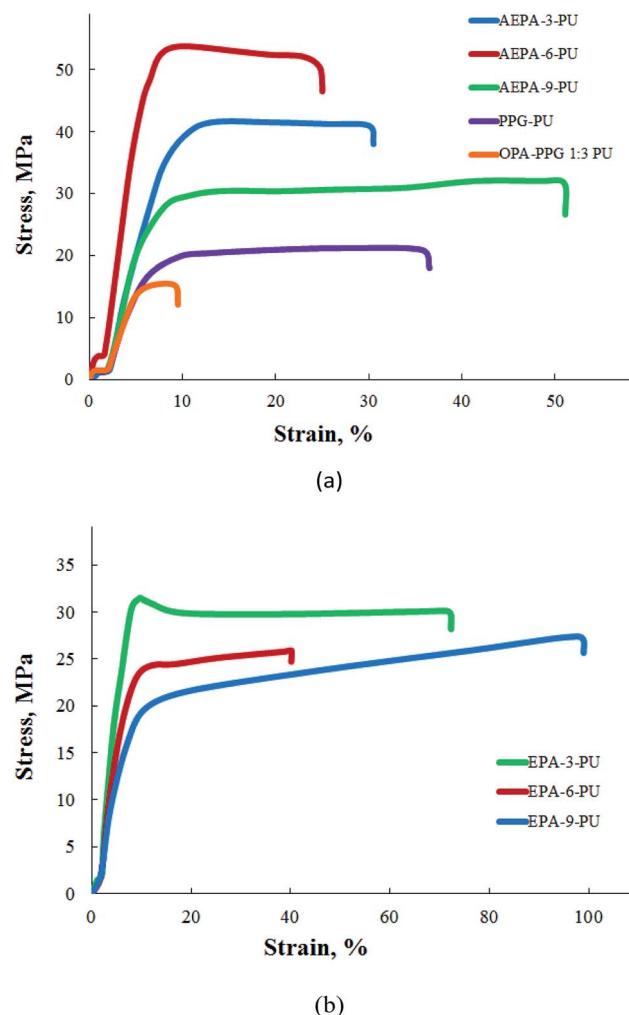


Fig. 10 Tensile tests of polymers.

to the AEPA structure of unreacted P-OH groups of *ortho*-phosphoric acid increases. The increase in the content of the acid ethers of OPA is the cause of the formation of viscous associates. As a result, an increase in the relative content of OPA leads to an intensive increase in the dynamic viscosity values, due to the fact that AEPA-6 and AEPA-9 contain unreacted P-OH groups, whose content increases from AEPA-3 to AEPA-9.

Table 1 Mechanical properties of polyurethanes

Polyol component	Adhesive strength to steel, MPa	Elastic modulus, MPa	Ultimate tensile strength, MPa
AEPA-3	4.2	1050	40
AEPA-6	6.0	1100	55
AEPA-9	3.5	850	30
EPA-3	1.0	380	30
EPA-6	1.2	310	25
EPA-9	0.5	275	25
PPG	5.0	205	20
$[H_3PO_4] : [PPG] = 1 : 3$	0.4	145	16

For EPA, viscosity with an increase in molar fraction of OPA increases with a smaller intensity, indicating a greater degree of branching of EPA compared to AEPA and that AEPA, unlike EPA, exhibits ionomeric nature (Scheme 2).

Thus, when replacing TEA with TEOA, the hydroxyl groups of triethanolamine are fully involved in etherification with OPA, but the catalytic activity of the tertiary amine weakens as a result of a decrease in its availability in the branched structure of AEPA.

The obtained AEPA were used as a polyol component of ionomeric nature for the synthesis of polyurethanes (AEPA-PU). The isocyanate component served as a polyisocyanate based on 4,4'-diphenylmethane diisocyanate.

According to the table and Fig. 10a, polyurethanes exhibit high adhesive characteristics and strength properties. For comparison purposes, stress-strain relations for polyurethanes are given, in which the AEPA was replaced with PPG (PPG-PU) and the product of the interaction of PPG was replaced with OPA (PPG-OPA-PU). The observed significant differences in the mechanical behaviour of AEPA-PU and PPG-PU are a consequence of the ionomeric nature of AEPA-PU. The most optimal components from the point of view of a combination of adhesive characteristics, modulus of elasticity and strength are AEPA-3-PU and AEPA-6-PU. An increase in the fraction of OPA in the composition of AEPA leads to an increase in the content of PO^- anions, but at the same time the number of urethane groups in the composition of AEPA-PU and the predicted density of the nodes of its spatial polymer network decrease. The high content of PO^- anions in AEPA-6-PU eliminates the reduction of cross-linking sites in comparison with AEPA-3-PU. As a result, the mechanical properties of AEPA-6-PU turned out to be higher in comparison with AEPA-3-PU. A further increase in the proportion of OPA in the composition of AEPA-9 leads to an increase in the content of associatively bound molecules of unreacted OPA. As a result, adhesion, elastic modulus, strength decreased, and plastic deformation values of AEPA-9-PU increased.

The mechanical behavior of polyurethanes synthesized using EPA (EPA-PU) was also investigated (Table 1 and Fig. 10b). The decrease in both strength and adhesive characteristics for polyurethanes obtained on the basis of EPA is a consequence of the fact that, unlike AEPA, these compounds lack PO^- anions.

Polyurethanes obtained on the basis of AEPA exhibit high heat resistance. The fact that the highest contribution to the heat resistance causes the ionomer component is indicated by

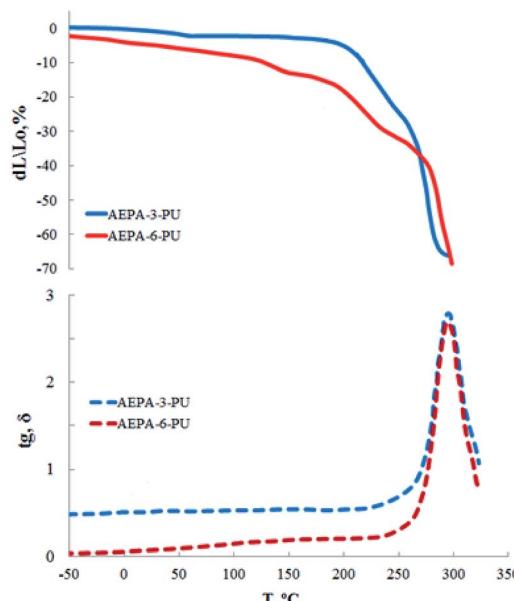


Fig. 11 TMA curves and temperature dependence of the mechanical loss tangent.

the TMA and DMA analysis curves (Fig. 11). So, if deformation processes for AEPA-3-PU begin with a temperature of 140 °C, then for AEPA-6-PU, which is characterized by a higher content of PO^- anions, the beginning of deformation corresponds to a temperature of 210 °C. The thermo-destructive flow of AEPA-6-PU starts at 250 °C with a maximum response to the curves of the tangent of the angle of mechanical losses at 290 °C for both AEPA-3-PU and AEPA-6-PU.

4 Conclusions

The etherification of *ortho*-phosphoric acid with polyoxypropylene glycol was studied in the presence of tertiary amines. It was shown that when triethylamine is used, complete OPA etherification occurs, followed by the formation of branched OPA ethers terminated by hydroxyl groups.

When replacing TEA with TEOA, the hydroxyl groups of triethanolamine are fully involved in etherification with OPA, but the catalytic activity of the tertiary amine weakens due to a decrease in its availability in the branched AEPA structure. As a result, the most acidic P-OH groups remain unreacted in the amino ethers of *ortho*-phosphoric acid.

Due to the presence of PO^- anions in the AEPA, these branched compounds are involved in the formation of clusters. As a result of a cooperative nature, unreacted OPA molecules also enter into these interactions.

Polyurethane ionomers were synthesized on the basis of AEPA. It was shown that AEPA-PU exhibit significantly higher adhesive strength compared with EPA-PU. The decrease in both strength and adhesive characteristics for polyurethanes obtained on the basis of EPA is a consequence of the fact that, unlike AEPA, these compounds lack PO^- anions. With an increase in the content of PO^- anions in the composition of

AEPA, a regular increase in the heat resistance of AEPA-PU occurs.

The described method of synthesis as well as the results of the present study can be used to produce industrially such polymers in the future.

Conflicts of interest

There are no conflicts to declare.

Abbreviation

OPA	<i>ortho</i> -Phosphoric acid
TEOA	Triethanolamine
TEA	Triethylamine
PPG	Polypropylene glycol
PIC	Polyisocyanate "Wannate PM-200"
AEPA	Amino ethers of <i>ortho</i> -phosphoric acid
AEPA-3	Amino ethers of <i>ortho</i> -phosphoric acid synthesized on the basis of [TEOA] : $[\text{H}_3\text{PO}_4]$: [PPG] = 1 : 3 : 6
AEPA-6	Amino ethers of <i>ortho</i> -phosphoric acid synthesized on the basis of [TEOA] : $[\text{H}_3\text{PO}_4]$: [PPG] = 1 : 6 : 6
AEPA-9	Amino ethers of <i>ortho</i> -phosphoric acid synthesized on the basis of [TEOA] : $[\text{H}_3\text{PO}_4]$: [PPG] = 1 : 9 : 6
EPA	Ethers of <i>ortho</i> -phosphoric acid
EPA-3	Amino ethers of <i>ortho</i> -phosphoric acid synthesized on the basis of [TEA] : $[\text{H}_3\text{PO}_4]$: [PPG] = 1 : 3 : 6
EPA-3*	Amino ethers of <i>ortho</i> -phosphoric acid synthesized on the basis of [TEA] : $[\text{H}_3\text{PO}_4]$: [PPG] = 1 : 3 : 7
EPA-6	Amino ethers of <i>ortho</i> -phosphoric acid synthesized on the basis of [TEA] : $[\text{H}_3\text{PO}_4]$: [PPG] = 1 : 6 : 6
EPA-9	Amino ethers of <i>ortho</i> -phosphoric acid synthesized on the basis of [TEA] : $[\text{H}_3\text{PO}_4]$: [PPG] = 1 : 9 : 6
PU	Polyurethane
AEPA-PU	Polyurethanes synthesized on the basis of AEPA
AEPA-3-PU	Polyurethanes synthesized on the basis of AEPA-3
AEPA-6-PU	Polyurethanes synthesized on the basis of AEPA-6
AEPA-9-PU	Polyurethanes synthesized on the basis of AEPA-9
EPA-PU	Polyurethanes synthesized on the basis of EPA
EPA-3-PU	Polyurethanes synthesized on the basis of EPA-3
EPA-6-PU	Polyurethanes synthesized on the basis of EPA-6
EPA-9-PU	Polyurethanes synthesized on the basis of EPA-9
PPG-PU	Polyurethanes synthesized on the basis of PPG and PIC

Acknowledgements

This work was supported by the Russian Science Foundation (grant no. 19-19-00136). NMR measurements were performed at the Center of Shared Facilities of Kazan Federal University.



References

1 I. Capek, *Adv. Colloid Interface Sci.*, 2004, **112**, 1.

2 A. Eisenberg and M. King, *Ion-Containing Polymers: Physical Properties and Structure*, Academic Press, New York, 1977, vol. 7, p. 3.

3 J. Ro, S. J. Huang and R. A. Weiss, *J. Biomed. Mater. Res., Part A*, 2008, **49**, 422.

4 H. Tachino, H. Hara, E. Hirasawa, S. Kutsumizu and S. Yano, *Polymer*, 1994, **26**, 1170.

5 S. Wu, Q. Guo, M. Kraska, B. Stuhn and Y. W. Mai, *Macromolecules*, 2013, **46**, 8190.

6 R. Dolog and R. A. Weiss, *Macromolecules*, 2013, **46**, 7845.

7 M. Grande, L. Castelnovo, L. D. Landro, C. Giacomuzzo, A. Francesconi and M. A. Rahman, *J. Appl. Polym. Sci.*, 2013, **130**, 1949.

8 F. Pierre, B. Commarieu, A. C. Tavares and J. Claverie, *Polymer*, 2016, **86**, 91.

9 G. Polizos, A. Kyritsis, P. Pissis, V. V. Shilov and V. V. Shevchenko, *Solid State Ionics*, 2000, **136**, 1139.

10 C.-L. Xu, J.-B. Zeng and Y.-Z. Wang, *Compos. Sci. Technol.*, 2014, **96**, 109.

11 G. C. Bazuin and A. Eisenberg, *Ind. Eng. Chem. Prod. Res. Dev.*, 1981, **20**, 271.

12 M. Carmo, D. L. Fritz, J. Mergel and D. Stolten, *Int. J. Hydrogen Energy*, 2013, **38**, 4901.

13 H. Zhang and P. K. Shen, *Chem. Rev.*, 2012, **112**, 2780.

14 T. J. Peckham and S. Holdcroft, *Adv. Mater.*, 2010, **22**, 4667.

15 K. D. Kreuer, *J. Membr. Sci.*, 2001, **185**, 29.

16 T. J. Peckham, J. Schmeisser, M. Rodgers and S. Holdcroft, *J. Mater. Chem.*, 2007, **17**, 3255.

17 T. Weissbach, E. M. W. Tsang, A. C. C. Yang, R. Narimani, B. J. Frisken and S. Holdcroft, *J. Mater. Chem.*, 2012, **22**, 24348.

18 N. Li and M. D. Guiver, *Macromolecules*, 2014, **47**, 2175.

19 G. He, Z. Li, J. Zhao, S. Wang, H. Wu, M. D. Guiver and Z. Jiang, *Adv. Mater.*, 2015, **27**, 5280.

20 C. C. Yang, R. Narimani, B. J. Frisken and S. Holdcroft, *J. Membr. Sci.*, 2014, **469**, 251.

21 Y. A. Elabd and M. Hickner, *Macromolecules*, 2011, **44**, 1.

22 J. Ding, C. Chuy and S. Holdcroft, *Chem. Mater.*, 2001, **13**, 2231.

23 J. R. Rowlett, V. Lilavivat, A. T. Shaver, Y. Chen, A. Daryaei, H. Xu, C. Mittelsteadt, S. Shimpalee, J. S. Riffle and J. E. McGrath, *Polymer*, 2017, **122**, 296.

24 Y. A. Elabd, E. Napadensky, C. W. Walker and K. I. Winey, *Macromolecules*, 2006, **39**, 399.

25 J. Peron, A. Mani, X. Zhao, D. Edwards, M. Adachi, T. Soboleva, Z. Shi, Z. Xie, T. Navessin and S. Holdcroft, *J. Membr. Sci.*, 2010, **356**, 44.

26 D. W. Shin, M. D. Guiver and Y. M. Lee, *Chem. Rev.*, 2017, **117**, 4759.

27 U. Plawky and W. Wenig, *J. Mater. Sci.*, 1998, **33**, 1611.

28 D. Basu, A. Das, K. W. Stockelhuber and S. Wiebner, *Designing of Elastomer Nanocomposites*, 2016, vol. 275, p. 235.

29 D. G. Peiffer, B. L. Hager, R. A. Weiss, P. K. Agarwal and R. D. Lundberg, *J. Polym. Sci., Polym. Phys. Ed.*, 1985, **23**, 1869.

30 J. T. Koberstein, A. F. Galambos and L. M. Leung, *Macromolecules*, 1992, **25**, 6195.

31 S. Chen and W. Chan, *J. Polym. Sci., Part B: Polym. Phys.*, 1990, **28**, 1499.

32 G. Nelson, *ACS Symposium Fire and polymers*, ACS Symposium Series, Washington, 1990.

33 R. Lyon, *Polym. Degrad. Stab.*, 1998, **61**, 201.

34 V. D. Krevelen, *Properties of polymers*, Elsevier, New York, ch. 26B, 1976.

35 W. Bassett, *Proc. fire retardant chemicals association meeting*, San Antonio, USA, 1989.

36 C. Tsonos, L. Apekitis, K. Viras, L. Stepanenko, L. Karabanova and L. Sergeeva, *J. Macromol. Sci., Part B: Phys.*, 2000, **39**(2), 155.

37 C. Tsonos, L. Apekitis, K. Viras, L. Stepanenko, L. Karabanova and L. Sergeeva, *Solid State Ionics*, 2001, **143**, 229.

38 C. Tsonos, L. Apekitis, C. Zois and G. Tsonos, *Acta Mater.*, 2004, **52**, 1319.

39 H. S. Egboh, A. Ghaffar, M. H. George, J. A. Barrie and D. J. Walsh, *Polymer*, 1982, **23**, 1167.

40 T. Y. T Chui, A. S Coote, C. Butler, M. H. George and J. A. Barrie, *Polym. Commun.*, 1988, **29**, 40.

41 T. Y. T. Chui, P. K. H. Lam, M. H. George and J. A. Barrie, *Polym. Commun.*, 1998, **29**, 317.

42 P. K. H. Lam, M. H. George and J. A. Barrie, *Polym. Commun.*, 1989, **30**, 2321.

43 S. Shahgaldi, I. Alaefour and X. Li, *Appl. Energy*, 2018, **217**, 295.

44 M. Kuramoto, M. Sakamoto, T. Teshirogi, J. Komiyama and Y. Iijima, *J. Appl. Polym. Sci.*, 1984, **29**, 977.

45 B. K. Kim, J. S. Yang, S. M. Yoo and J. S. Lee, *Colloid Polym. Sci.*, 2003, **281**, 461.

46 X. Wei and X. Yu, *J. Polym. Sci.*, 1997, **35**, 225.

47 M. Davletbaeva, O. Yu. Emelina, I. V. Vorotyntsev, R. S. Davletbaev, E. S. Grebennikova, A. N. Petukhov, A. I. Ahkmetshina, T. S. Sazanova and V. V. Loskutov, *RSC Adv.*, 2015, **5**, 65674.

48 M. Davletbaeva, G. R. Nurgaliyeva, A. I. Ahkmetshina, R. S. Davletbaev, A. A. Atlaskin, T. S. Sazanova, S. V. Efimov, V. V. Klochkov and I. V. Vorotyntsev, *RSC Adv.*, 2016, **6**, 111109.

49 P. Johnston and R. Adhikari, *Eur. Polym. J.*, 2017, **95**, 138.

50 D. K. Kakati and M. H. George, *Polymer*, 1993, **34**, 4319.

51 J.-S. Kim, Y. H. Nah, S. S. Jarng, W. Kim, Y. Lee and Y. W. Kim, *Polymer*, 2000, **41**, 3099.

52 K. Mequanint, R. Sanderson and H. Pasch, *Polym. Degrad. Stab.*, 2002, **77**, 121.

53 Z. Petrovic, Z. Zavargo, J. Flynn and W. MacKnight, *J. Appl. Polym. Sci.*, 1994, **51**, 1087.

54 K. Troev, R. Tsevl, T. Bourova, S. Kobayashi, H. Uayama and D. Roundhill, *J. Polym. Sci., Part A: Polym. Chem.*, 1996, **34**, 621.

55 L. D. Quin, *A Guide to Organophosphorus Chemistry*, Wiley-Interscience, New York, 2000.



56 R. Engel and J. I. Cohen, *Synthesis of Carbon-Phosphorus Bonds*, Florida, CRC Press LLC, 2004.

57 K.-J. Fest and C. Schmidt, *The Chemistry of Organophosphorus*, Springer-Verlag, Berlin, Heidelberg, New York, 1982.

58 A. Sakakura, M. Katsukawa, T. Hayashib and K. Ishihara, *Green Chem.*, 2007, **9**, 1166.

59 A. Sakakura, M. Katsukawa and K. Ishihara, *Org. Lett.*, 2005, **10**, 1999.

60 A. Sakakura, M. Katsukawa and K. Ishihara, *Angew. Chem., Int. Ed.*, 2007, **46**, 1423.

61 K. J. Powell, P. L. Brown, R. H. Byrne, T. Gajda, G. Hefter, S. Sjöberg and H. Wanner, *Pure Appl. Chem.*, 2005, **77**(4), 739.

62 E. Malmstrom and A. Hult, *J. Macromol. Sci., Rev. Macromol. Chem. Phys.*, 1997, **37**, 555.

63 B. Romagnoli and W. Hayes, *J. Mater. Chem.*, 2002, **12**, 767.

