

Cite this: *RSC Adv.*, 2019, 9, 16790

Ultrafast synthesis of AFX-Type zeolite with enhanced activity in the selective catalytic reduction of NO_x and hydrothermal stability†

Anand Chokkalingam,^a Watcharop Chaikittisilp,^{‡a} Kenta Iyoki,^a Sye Hoe Keoh,^a Yutaka Yanaba,^b Takeshi Yoshikawa,^b Tetsuro Kusamoto,^c Tatsuya Okubo^a and Toru Wakihara^{*,a}

Shortening the synthesis time of SSZ-16 (AFX type) zeolite from several days to 2 h has been achieved using an ultrafast synthesis route involving *N,N,N',N'*-tetraethylbicyclo[2.2.2]oct-7-ene-2,3,5,6-dipyrrolidinium (TEBOP) as an organic structure-directing agent (OSDA) in a tubular reactor assisted by seed crystals. Recently, copper exchanged SSZ-16 has been looked upon as one of the few equivalents to SSZ-13 for the selective catalytic reduction of NO_x with ammonia (NH₃-SCR) from automobile exhausts. Hydrothermal stability is one of the crucial properties for any zeolites that compete for automobile applications. All the samples prepared were analyzed using sophisticated physio-chemical techniques and those prepared from TEBOP were subjected to SCR of NO_x reactions. The rapid crystal growth induced by high synthesis temperature bestowed the ultrafast prepared SSZ-16 with high crystallinity and hydrothermal stability as well as enhanced SCR of NO_x activity even when aged at 800 °C. Compared to 1,1'-(1,4-butanediyl)bis-4-aza-1-azoniabicyclo[2.2.2]octane dibromide (DABCO), TEBOP was found to be desirable as an OSDA for high crystallinity and hydrothermal stability.

Received 13th April 2019
Accepted 22nd May 2019

DOI: 10.1039/c9ra02787d

rsc.li/rsc-advances

1. Introduction

Zeolites have been studied as an alternative for the traditional mixed oxide catalysts based on vanadia for treating nitrogen oxides (NO_x) emitted from combustion of fossil fuels. NO_x is one of the prominent ingredients polluting global air quality and affecting human health for decades. Selective catalytic reduction of NO_x with ammonia (NH₃-SCR) using on-board converters is recognized as one of the most advanced emission control technologies in meeting the emission standards set by environmental policies.^{1–10} During the primordial infusion of the SCR technology for the conventional treatment of industrial emissions, mixed oxides of vanadium, tungsten and titanium were used as catalysts. However, their efficiency is limited to a narrow temperature window (*i.e.* 300–400 °C), above which they volatilize and also succumb to toxins from secondary

emissions.^{8–10} To be a better SCR catalyst, the material should possess high hydrothermal stability.

Since the pioneering work on the SCR activity of Cu-ZSM-5,¹¹ numerous zeolites have been studied and mostly found to be unstable at high operating temperatures or lacking low-temperature SCR activity^{12–14} until the discovery of Cu-SSZ-13, which exhibited high hydrothermal stability as well as low-temperature activity.^{15–18} Despite its success in SCR market, the flaws related to cold start and high temperature exhaust conditions seem to affect the SCR activity and structural stability of Cu-SSZ-13 as well, which has aroused interest on small pore zeolites with inherent large cavities capable of retaining their structural integrity despite repeated exposure to severe hydrothermal conditions.^{10,15,19} Even, some of them (SSZ-16, SSZ-39, STA-7 and SSZ-52) are considered alternatives to SSZ-13, particularly, SSZ-16 (AFX topology, see Fig. S1†) made up of elongated larger *afu* cages (0.55 × 1.35 nm) and smaller *gme* cages (0.33 × 0.74 nm) has drawn interest due to its promising catalytic performances in SCR of NO_x.^{20–25} Recently, the SSZ-16 has been projected as a remarkable NH₃-SCR catalyst achieving ~100% NO_x conversion owing to ease of pore accessibility and generates research interests on developing its physicochemical aspects from an SCR performance perspective.^{26,27} However, reports dedicated to reducing its synthesis time from an industrial view point (*i.e.* production cost) are scanty. For many years zeolites have been overlooked as potential applicants in number of industrial applications

^aDepartment of Chemical System Engineering, The University of Tokyo, Tokyo, 113-8656, Japan. E-mail: wakihara@chemsys.t.u-tokyo.ac.jp

^bInstitute of Industrial Science, The University of Tokyo, Tokyo, 153-8505, Japan

^cDepartment of Chemistry, Graduate School of Science, The University of Tokyo, Bunkyo-ku, Tokyo 113-0033, Japan

† Electronic supplementary information (ESI) available. See DOI: 10.1039/c9ra02787d

‡ Present address: Research and Services Division of Materials Data and Integrated System (MaDIS), National Institute for Materials Science (NIMS), 1-2-1 Sengen, Tsukuba, Ibaraki 305-0047, Japan.



particularly because of the assumption that their syntheses are time elongated, and thus involve high production cost. This has led to innovative studies that report efficient methods to produce zeolites in short period of time.^{28–31}

Temperature, a significant factor in zeolite syntheses that govern the rate of nucleation and crystal growth is crucial for shortening the synthesis time of zeolites, which would otherwise take several days to even weeks. Hitherto, the complexity surrounding the zeolite formation mechanism remains unexplained and many studies have suggested that in fact, it is the nucleation that consumes much time rather than the crystal growth.^{30–34} Recently, the ultrafast synthesis techniques that involves tubular reactor with the assistance of seed that reduces the synthesis time of zeo-type and zeolites *viz.*, AFI, CHA, MFI, ERI, and MOR down to several minutes were reported by us.^{30,35–42} Table S1† compares the experimental conditions of these zeolites with the current study. Tubular reactor has proven effective to overcome the thermal lag faced in conventional autoclave (Fig. S2†), which allows rapid heat transfer, capable of producing zeolites in the order of minute rendering the synthesis of zeolites high quantitative efficiency in terms of time, energy and cost.

So far, several flexible OSDAs have been employed for preparing SSZ-16,^{43–50} but recently, a bulky OSDA with a rigid structure named TEBOP, often involved in the synthesis of MSE-type zeolites, was introduced for the synthesis of SSZ-16.^{43,44} However, it required at least 24 h for preparing the SSZ-16 using TEBOP in an autoclave and achieving it at a faster pace remained challenging because the degree of synthesis freedom offered by such small pore zeolites are quite narrow.^{43,44} Extending the ultrafast synthesis approach, we were able to synthesize SSZ-16 (AFX-type) zeolite in 2 hours using sealed tubular reactor which would otherwise require several days.^{45–51} The SSZ-16 from TEBOP bettered their DABCO counterpart with physio-chemical traits that are promising for enhanced SCR of NO_x and in particular they were bestowed with excellent hydrothermal stability. Especially, the SSZ-16 prepared in a tubular reactor endured under the harsh hydrothermal treatment conditions (*i.e.* 800 °C × 5 h with continuous streaming of humid air with 10 volume% H₂O) by retaining ~70% of its crystallinity and exhibiting NO_x conversion over 80%.

2. Experimental section

2.1. Synthesis of SSZ-16 with DABCO

Based on a previous report, DABCO (for molecular structure see Fig. S3†) was prepared and used as OSDA for SSZ-16 synthesis.⁵¹ In a typical synthesis, a reactant with a molar composition of 30.0SiO₂–1.0Al₂O₃–12.0Na₂O–3.0DABCO–520H₂O involving FAU type zeolite (HSZ-373HUA, Tosoh, Si/Al = 15, H⁺ type) as both Si and Al sources was prepared by applying centrifugal force in a mixer (THINKY MIXER ARE-310). Then, the reactant transferred to a conventional Teflon lined stainless steel autoclave (AC) was treated at 140 °C for 2 days using a rotary oven under 20 rpm. Prior to calcination at 550 °C for 10 h, the products in the AC were recovered by centrifugation, washed

thrice with de-ionized water and dried overnight at 80 °C. The calcined SSZ-16 prepared from DABCO is designated as AC-DABCO.

2.2. Synthesis of SSZ-16 seed with Et6-Diquat-5

The OSDA, *N,N'*-bis-triethylpentanediyldiammonium (Et6-diquat-5, for molecular structure see Fig. S3†) was prepared by slightly modifying the recipe from a previous report.⁵² In a typical SSZ-16 synthesis, a reactant with a molar composition of 30.0SiO₂–0.5Al₂O₃–30.0NaOH–3.0OSDA–1200.0H₂O involving fumed silica (Aerosil 200, Degussa) as Si source and Al(NO₃)₃·9H₂O (Wako, Japan) as Al source was prepared by stirring at 90 °C for 72 h under 500 rpm. Then, it was treated at 150 °C for 6 days under 22 rpm after contained in AC. The SSZ-16 thus produced was used as a seed.

2.3. Synthesis of SSZ-16 with TEBOP

The 20 wt% solution of TEBOP (for molecular structure see Fig. S3†) was prepared by a slightly modified recipe reported elsewhere⁴³. In a typical SSZ-16 synthesis, a reactant with a molar composition of 1.0SiO₂–0.092Al₂O₃–0.1NaOH–0.1TEBOP–20.0H₂O involving FAU type zeolite (HSZ-350HUA, Tosoh, SiO₂/Al₂O₃ = 10, H⁺ type) as both Si and Al sources was prepared by stirring at room temperature for 2 h under 400 rpm. Ten minutes prior to the hydrothermal treatment, the seed (10 wt% w.r.t. SiO₂ content) was added and dispersed well into this reactant mixture. The mixture was treated static at 160 °C for 4 days after contained in AC. The SSZ-16 thus produced is designated as AC-TEBOP.

2.4. Faster synthesis of SSZ-16 with TEBOP

The synthesis of SSZ-16 at a faster pace was achieved by modifying the molar composition of the reactant slightly (1.0SiO₂–0.092Al₂O₃–0.15NaOH–0.15TEBOP–30.0H₂O), replacing the AC with tubular reactors (TRs) and increasing the temperature from 160 to 210 °C. Prior to the fast synthesis, TRs were packed with the reactant (up to 75% of its total volume), sealed tightly using stainless steel caps (Swagelok, 1/4 inch, SS-400-C, at both ends) and immersed within a preheated oil bath connected with temperature controller and sensors. After 2 h, the TRs were quenched in a cold water bath followed by sample recovery. The material thus obtained is designated as TR-TEBOP.

The yield of the samples were calculated with respect to the amount of SiO₂ added to the initial gel. After applying the same recovery and calcination treatment conditions similar to AC-DABCO, all the samples (except seed) were ion-exchanged with 1 M solution (1 g SSZ-16/40 mL) of NH₄NO₃ (Wako, Japan) at 80 °C for 4 h. After subjected to the ion-exchange treatment twice, the samples were dried overnight at 80 °C.

2.5. Copper ion-exchange

The ammonia forms of the samples were subjected to copper ion-exchange by stirring with 0.08 M solution (1 g sample/50 mL) of Cu(CH₃COO)₂ (Wako, Japan) at 100 °C under 400 rpm for 4 h (thrice). The samples were recovered by centrifugation,



dried and applied as catalysts in the SCR of NO_x reactions. All the copper exchanged samples are designated with a “Cu” prefixed to their actual name and are collectively termed as “fresh”.

2.6. Hydrothermal stability test

The copper ion-exchanged samples (~0.5 g) were packed within the mid zone of a quartz tube contained by quartz wool at the bottom. A controlled heating of the tube to desired temperatures (every 50 °C from 700 to 800 °C) was performed at a rate of 10 °C min⁻¹ under a constant flow of dry air (100 cm³ min⁻¹). Once the temperature attains the target, a stream of humid air with 10 vol% H₂O is passed continuously for 5 h. Further, the samples treated under hydrothermal stability test will be termed as “aged” and designated with a number (respective to their hydrothermal aging temperature) suffixed to their actual name.

2.7. Characterization

Structural elucidation of the samples by powder X-ray diffraction technique (XRD) were collected on Rigaku Ultima IV diffractometers with CuK α radiation (λ = 1.5406 Å, V = 40 kV, I = 40 mA) with 4° min⁻¹ scan rate. The crystallinity of the samples after hydrothermal treatment (HT) was obtained by peak fitting technique (Igor software) that calculates and compares the areas under the XRD peaks between the 2θ values 15° to 35° obtained before and after HT. Nitrogen (N₂) adsorption–desorption measurements were performed in a Quantachrome Autosorb-iQ2 instrument at liquid N₂ temperature. Prior to the measurements, the samples were outgassed at 350 °C for 6 h under vacuum. The surface areas of the samples were calculated using a multipoint BET (Brunauer–Emmett–Teller) method (with the recommended relative pressure range 0.05–0.2) and their micropore volumes were assessed using t -plot method. Morphology and particle size data were obtained from a field emission scanning electron microscope (FE-SEM, JSM-7000F JEOL, Japan).

A Thermo iCAP 6300 inductively coupled plasma-atomic emission spectrometer (ICP-AES) was used to analyze the elemental composition of the samples. Electron Spin Resonance (ESR) spectroscopic data of selective samples were obtained using JEOL-JES-FA200 spectrometer on the X-Band frequency at room temperature. The ESR signals were recorded in the field region of 160 to 460 mT with a microwave power of 2 mW, a modulation frequency of 100 kHz, a central field of 310 mT and a sweep time of 30 s. Solid-state magic angle spinning (MAS) nuclear magnetic resonance (NMR) spectroscopy measurements were conducted on a JNM-ECA 500 (JEOL) instrument. ²⁹Si MAS NMR spectra were recorded at 99.37 MHz with a $\pi/2$ pulse length of 5.0 μ s, a recycle delay of 60 s and a spinning frequency of 10 kHz. ²⁷Al MAS NMR spectra were recorded on a JEOL ECA 800 (18.8 T) spectrometer using a Larmor frequency of 208.49 MHz. All the measurements involved a JEOL 3.2 mm MAS probe and ZrO₂ rotor (spun at 20 kHz). To ensure full relaxation of the ²⁷Al spins, the spectra were obtained by collecting 256 scans with a pulse of ≤ 0.27 μ s and its

repetition time of 1 s. The ²⁷Al chemical shifts were calibrated with a 1 M aqueous solution of Al(NO₃)₃ as 0 ppm.

2.8. SCR reactor system

The selective catalytic reduction of NO_x experiments with NH₃ were performed in a quartz fixed bed reactor measuring 8 mm in height. The quartz reactor was stuffed (maximum packing limit was 6 mm) with 20 mg of catalyst together with 280 mg SiC (silicon carbide). A pre-treatment of the catalyst was done by passing a mixture of gases (O₂: 5%, H₂O: 3% and N₂: 92%) using a mass flow controller. The experiments were conducted at a temperature ranging from 150 to 600 °C using a mixture of NO: 300 ppm and NH₃: 300 ppm with a flow rate of 200 cm³ min⁻¹ and GHSV is 50 000 h⁻¹. The conversion readings were obtained for every 50 °C rise and analyzed by a quadrupole mass spectrometer (Q-MS).

3. Results and discussion

The structure of all the samples (before and after ion-exchange with copper) were highly crystalline and matched well with pure AFX phase as determined by XRD patterns (Fig. S4†). The influence of reactant ratio and OSDA over the physicochemical properties of these samples are shown in Table 1. The molar compositions, surface area and porosity of TR-TEBOP showed marginal difference as compared to AC-TEBOP, however the increase in yield per batch was more than 30% (Table 1). Other than shortening the synthesis time from 96 h (at 160 °C) to 2 h (at 210 °C) the effect of replacing AC by TR on the physicochemical traits of TR-TEBOP remains unclear. Fig. S5† and Table 2 also display impetus information on the results of hydrothermal aging of the SSZ-16 samples (based on their crystallinity factor), which could very well clarify the uncertainty surrounding the advantages of involving TR. The intensity of the XRD peaks (Fig. S5†) are intact until an hydrothermal aging temperature of 750 °C above which it deteriorates for Cu-AC-DABCO and Cu-AC-TEBOP. Cu-AC-DABCO undergoes complete degradation at 800 °C and Cu-AC-TEBOP despite retaining AFX phase became highly amorphous. Table 2 compares the percentage of relative crystallinity (calculated from the XRD patterns) of these samples. Only Cu-TR-TEBOP managed to retain the AFX phase throughout the hydrothermal aging process (Fig. S5†). The essence of Fig. S5† and

Table 1 Comparison of physicochemical data of SSZ-16 zeolites before and after copper exchange

Sample name	ICP molar ratio		BET surface area (m ² g ⁻¹)	Micropore volume (cc g ⁻¹)	Yield (%)
	Si/Al	Cu/Al			
AC-DABCO	3.18	0.00	621	0.23	33
AC-TEBOP	4.63	0.00	687	0.26	51
TR-TEBOP	4.66	0.00	679	0.26	82
Cu-AC-DABCO	3.17	0.33	627	0.23	
Cu-AC-TEBOP	4.60	0.36	594	0.20	
Cu-TR-TEBOP	4.77	0.33	620	0.22	

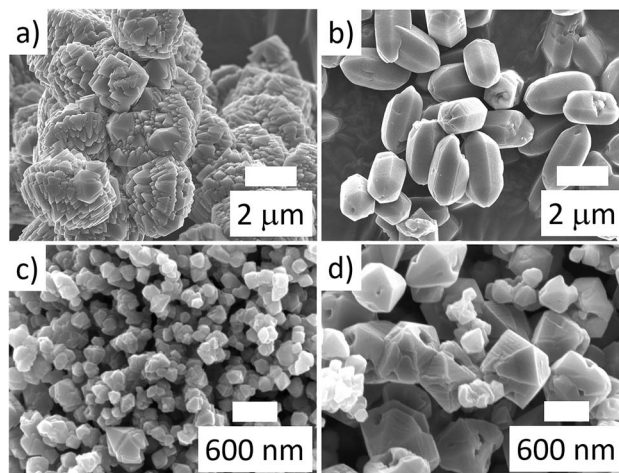


Table 2 Comparison of physicochemical data of the copper ion-exchanged samples before and after hydrothermal aging

Sample name	BET surface area (m ² g ⁻¹)	Micropore volume (cc g ⁻¹)	Crystallinity (%)
Cu-AC-DABCO	627	0.23	100
Cu-AC-DABCO-700	617	0.23	86
Cu-AC-DABCO-750	616	0.22	75
Cu-AC-DABCO-800	4	0	0
Cu-AC-TEBOP	594	0.20	100
Cu-AC-TEBOP-700	656	0.24	88
Cu-AC-TEBOP-750	679	0.25	86
Cu-AC-TEBOP-800	127	0.04	12
Cu-TR-TEBOP	620	0.22	100
Cu-TR-TEBOP-700	616	0.21	89
Cu-TR-TEBOP-750	579	0.19	88
Cu-TR-TEBOP-800	477	0.16	69

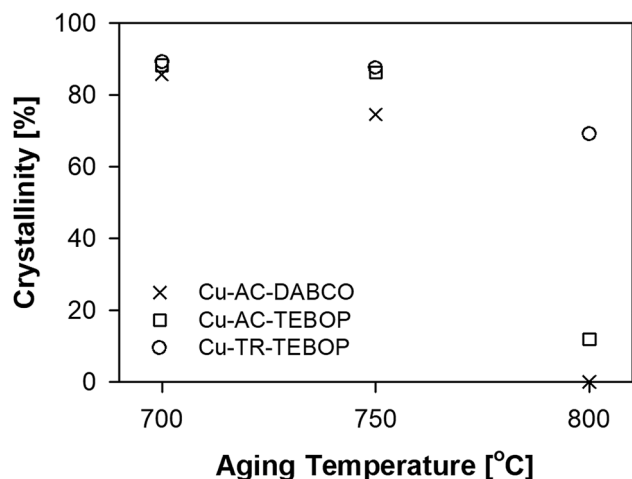
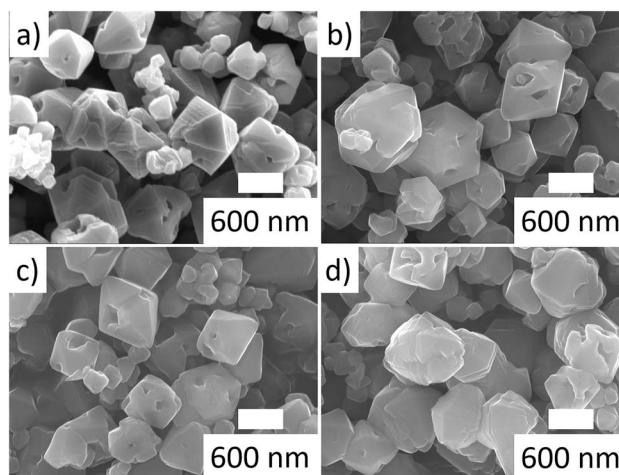
Table 2 in terms of hydrothermal stability depicted as crystallinity curves is shown in Fig. 1. Plotted as a function of hydrothermal aging temperature, these crystallinity curves support the fact that compared to DABCO, TEBOP produces SSZ-16 zeolite with high hydrothermal stability especially when tubular reactor was employed.

The micropore volumes of the SSZ-16 samples prepared from TEBOP is apparently higher than the one prepared from DABCO (Tables 1 and 2). Crystals of Cu-AC-DABCO were spindled hexagonal while samples from TEBOP show di-hexahedron morphology (Fig. 2). As for the particle size, Cu-AC-DABCO is the largest (2 μm) with Cu-TR-TEBOP being larger (with a broad size distribution) than Cu-AC-TEBOP. Though, the influence of increasing hydrothermal aging temperature over the crystallinity of Cu-TR-TEBOP sample series was clearly evident from XRD (crystallinity) data, any changes in their particle size and morphology were not observed from their SEM images (Fig. 3). The ESR spectra of the fresh and aged samples are shown in Fig. 4a and b respectively. Only the paramagnetic copper cations (Cu²⁺) are ESR active while Cu⁺ and Cu⁰ remain anonymous to ESR. The peaks at lower field confirmed the presence of Cu²⁺

**Fig. 2** SEM images of SSZ-16 samples (a) seed; (b) Cu-AC-DABCO; (c) Cu-AC-TEBOP; (d) Cu-TR-TEBOP.

species in fresh samples (Fig. 4a). At 700 °C hydrothermal aging, the signal intensity of Cu-TR-TEBOP decreased, however, it remained almost unchanged when the hydrothermal aging temperature was increased further till 800 °C (Fig. 4b) and resolving their hyperfine structures at room temperature ESR measurements is unlikely, probably due to the interference from the ionic mobility and dipolar interactions of the Cu²⁺ species.^{53,54} As such, the ESR results revealed that the hydrothermal aging at varying temperatures did not alter the state of the copper species in Cu-TR-TEBOP sample series, which is of critical importance for their SCR of NO_x activity.

The ²⁹Si MAS NMR spectrum of all the samples except Cu-AC-DABCO shows two peaks at −116.8 ppm and −111.0 ppm (Fig. 4c and d) corresponding to Q⁴ (Si(OSi)₄) and Q⁴ (Si(OAl)(OSi)₃) species respectively. Cu-AC-DABCO exhibited an additional shoulder peak around −105 ppm that corresponds to Q⁴ (Si(OAl)₂(OSi)₂) species (Fig. 4c). The intensity of the NMR signals for Q⁴ (Si(OAl)(OSi)₃) in Cu-TR-TEBOP sample weakened with increasing the hydrothermal aging temperature from 700 to 800 °C (Fig. 4d) which might be due to partial structural

**Fig. 1** Comparison of crystallinity curves of the fresh copper-ion exchanged SSZ-16 samples plotted as a function of hydrothermal aging temperature.**Fig. 3** SEM images of Cu-TR-TEBOP samples (a) Cu-TR-TEBOP; (b) Cu-TR-TEBOP-700; (c) Cu-TR-TEBOP-750; (d) Cu-TR-TEBOP-800.

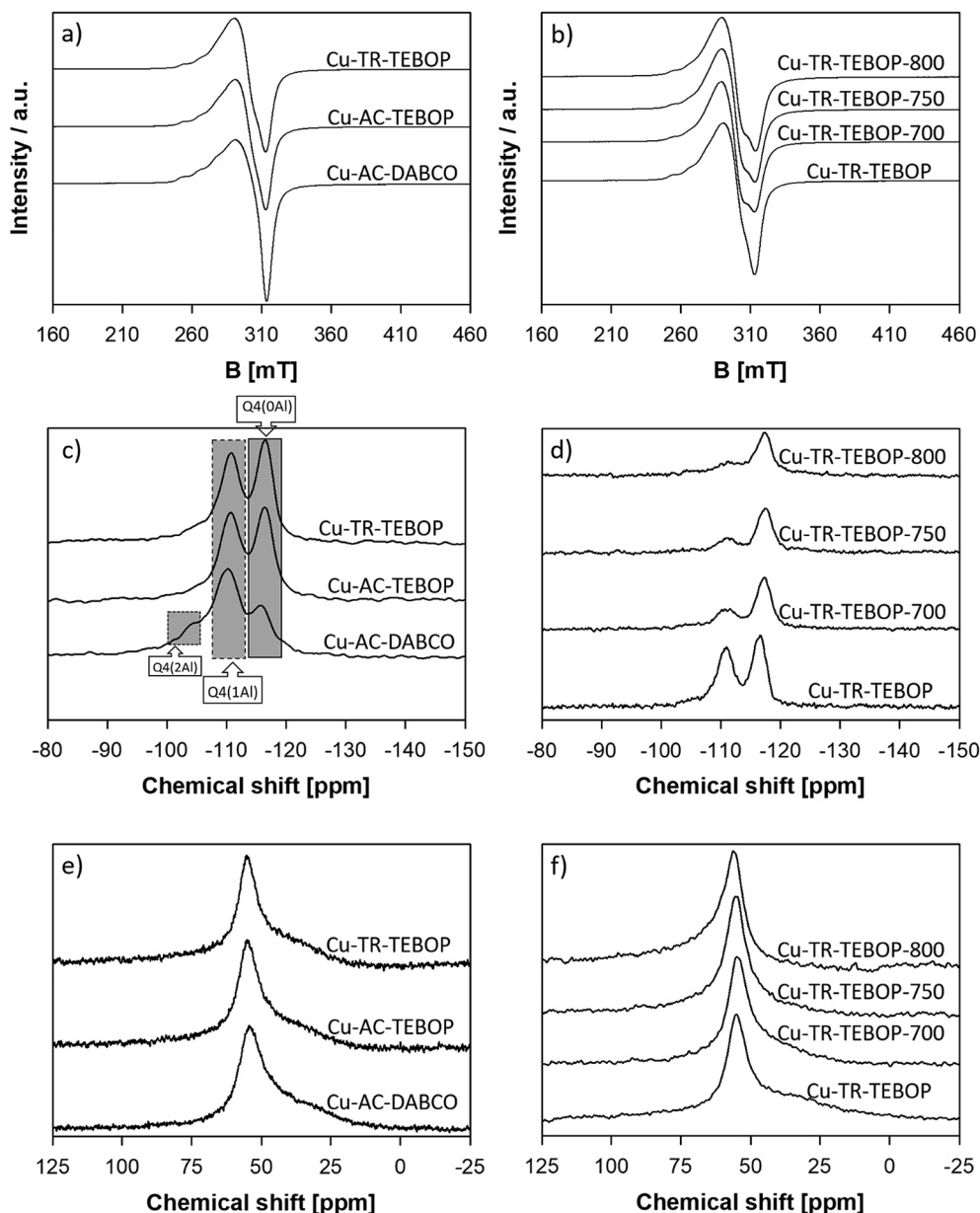


Fig. 4 ESR spectra of SSZ-16 samples (a) fresh and (b) Cu-TR-TEBOP series; ^{29}Si solid MAS NMR of (c) fresh and (d) Cu-TR-TEBOP series and ^{27}Al solid MAS NMR of (e) fresh and (f) Cu-TR-TEBOP series.

collapse and dealumination. The ^{27}Al MAS NMR of all the samples exhibited one peak in the range of 54.4 to 56.9 ppm characteristics of tetrahedral coordinated framework aluminum atoms (Fig. 4e and f). Absence of signal near 0 ppm indicates that all the samples are free from octahedral or extra framework aluminum species. Especially, Fig. 4f confirmed that the aluminum atoms in Cu-TR-TEBOP remained intact with tetrahedral coordination even after undergoing high temperature rigorous hydrothermal aging treatment.

Fig. 5a and b compares the performance of the Cu-AC-TEBOP and Cu-TR-TEBOP samples in the SCR of NO_x as a function of temperature. Their tendency to reduce NO_x over the entire SCR temperature range (*i.e.* 150 to 600 °C) remained practically the same (*i.e.* 100%) until the hydrothermal aging temperature of 700 °C. However, the activity of Cu-AC-TEBOP-800 dropped

drastically while Cu-TR-TEBOP-800 managed to achieve well above 80% NO_x conversion exposing the fact that how the crystallinity of the latter has been enhanced to withstand rigorous hydrothermal aging conditions.

Though we have previously accounted the role of TRs at higher synthesis temperature as a means to shorten the synthesis time of several zeolites, the high hydrothermal stability and SCR performance of Cu-TR-TEBOP-800 despite having its physio-chemical data comparatively similar (except particle size) to Cu-AC-TEBOP is quite intriguing. In accordance with our previous studies,^{35,42} we conclusively highlight that the degree of healing structural defects under high temperature ultrafast synthesis as a most probable cause for the high hydrothermal stability and therefore SCR activity of Cu-TR-TEBOP. Still it is tedious to evaluate the exact amount of defects, we used hydrothermal aging and SCR



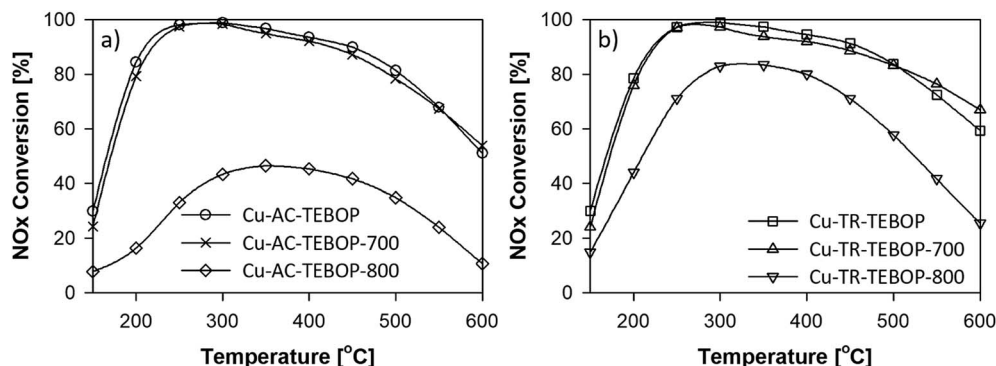


Fig. 5 NH_3 SCR of NO_x performance of (a) Cu-AC-TEBOP series; (b) Cu-TR-TEBOP series.

assessment supported by the SEM images of both fresh and aged Cu-TR-TEBOP samples (Fig. 3a–d) to apparently understand the influence of defects in Cu-AC-TEBOP and the role of high temperature ultrafast synthesis in healing them in Cu-TR-TEBOP.

4. Conclusion

The synthesis of SSZ-16 zeolites using TEBOP (a recently reported OSDA for AFX synthesis) has been achieved in just 2 h for the first time following a novel ultrafast synthesis route involving tubular reactors. The TR combined with high synthesis temperature and seed assistance plays a pivotal role for synthesizing SSZ-16 in 2 h (instead of 4 days in autoclaves (AC)). After copper exchange and hydrothermal aging at 700 °C, both the Cu-TR-TEBOP and Cu-AC-TEBOP samples exhibited physio-chemical properties and SCR of NO_x activity with marginal difference. However, Cu-TR-TEBOP revealed higher hydrothermal stability by retaining its NH_3 -SCR of NO_x activity even after hydrothermal aging at 800 °C. It was found that the particles of Cu-AC-TEBOP are majorly defective but the high temperature synthesis medium in Cu-TR-TEBOP initiates a rapid healing process to reduce the structural defects. Therefore, the high crystallinity, hydrothermal stability and SCR activity of Cu-TR-TEBOP specific to 800 °C hydrothermal aging is due to structurally healed crystals.

Conflicts of interest

There are no conflicts to declare.

References

- 1 S. E. Lehman and S. C. Larsen, *Environ. Sci.: Nano*, 2014, **1**, 200–213.
- 2 P. A. Jacobs, M. Dusselier and B. F. Sels, *Angew. Chem., Int. Ed.*, 2014, **53**, 8621–8626.
- 3 E. Khodaverdi, H. A. Soleimani, F. Mohammadpour and F. Hadizadeh, *Chem. Biol. Drug Des.*, 2016, **87**, 849–857.
- 4 T. Stavrou, J. F. Muller, K. F. Boersma, D. Smedt and R. J. van der A, *Geophys. Res. Lett.*, 2008, **35**, 1–5.
- 5 U. Latza, S. Gerdes and X. Baur, *Int. J. Hyg. Environ. Health*, 2009, **212**, 271–287.
- 6 P. Schneider and R. J. van der A, *J. Geophys. Res.*, 2012, **117**, 1–17.
- 7 A. Saiz-Lopez, R. Borge, A. Notario, J. A. Adame, D. de la Paz, X. Querol, B. Artinano, F. J. Gomez-Moreno and C. A. Cuevas, *Sci. Rep.*, 2017, **7**, 45956.
- 8 S. Brandenberger, O. Krocher, A. Tissler and R. Althoff, *Catal. Rev.*, 2008, **50**, 492–531.
- 9 Z. Xie, X. Zhou, H. Wu, L. Chen, H. Zhao, Y. Liu, L. Pan and H. Chen, *Sci. Rep.*, 2016, **6**, 30132.
- 10 S. Vishnu Priya, T. Ohnishi, Y. Shimada, Y. Kubota, T. Masuda, Y. Nakasaka, M. Matsukata, K. Itabashi, T. Okubo, T. Sano, N. Tsunooji, T. Yokoi and M. Ogura, *Bull. Chem. Soc. Jpn.*, 2018, **91**, 355–361.
- 11 M. Iwamoto and H. Hamada, *Catal. Today*, 1991, **10**, 57–71.
- 12 H. Y. Chen and W. M. H. Sachtler, *Catal. Today*, 1998, **42**, 73–83.
- 13 R. Q. Long and R. T. Yang, *J. Am. Chem. Soc.*, 1999, **121**, 5595–5596.
- 14 F. Gao, J. H. Kwak, J. Szanyi and C. H. F. Peden, *Top. Catal.*, 2013, **56**, 1441–1459.
- 15 I. Bull, *et al.*, *U.S. Pat.*, 7601662 B2, 2009.
- 16 I. Bull, *et al.*, *U.S. Pat.*, 7704475 B2, 2010.
- 17 P. J. Andersen, *et al.*, *U.S. Pat.*, 7998443 B2, 2011.
- 18 P. J. Andersen, *et al.*, *U.S. Pat.*, 8182777 B2, 2012.
- 19 P. G. Plakeman, E. M. Burkholder, H. Y. Chen, J. E. Collier, J. M. Fedeyko, H. Jobson and R. R. Rajaram, *Catal. Today*, 2014, **231**, 56–63.
- 20 D. W. Fickel, E. D'Addio, J. A. Lauterbach and R. F. Lobo, *Appl. Catal., B*, 2011, **102**, 441–448.
- 21 M. Moliner, C. Franch, E. Palomares, M. Grill and A. Corma, *Chem. Commun.*, 2012, **48**, 8264–8266.
- 22 A. L. Picone, S. J. Warrender, A. M. Z. Slawin, D. M. Dawson, S. E. Ashbrook, P. A. Wright, S. P. Thompson, L. Gaberova, P. L. Llewellyn, B. Moulin, A. Vimont, M. Daturi, M. B. Park, S. K. Sung, N. In-Sik and S. B. Hong, *Microporous Mesoporous Mater.*, 2011, **146**, 36–47.
- 23 D. Xie, L. B. McCusker, C. Baerlocher, S. I. Zones, W. Wan and X. Zou, *J. Am. Chem. Soc.*, 2013, **135**, 10519–10524.
- 24 T. M. Davis, A. T. Liu, C. M. Lew, D. Xie, A. I. Benin, S. Elamori, S. I. Zones and M. W. Deem, *Chem. Mater.*, 2016, **28**, 708–711.



- 25 J. D. Albarracin-Caballero, I. Khurana, J. R. Di Iorio, A. J. Shih, J. E. Schmidt, M. Dusselier, M. E. Davis, A. Yezerets, J. T. Miller, F. H. Ribeiro and R. Gounder, *React. Chem. Eng.*, 2017, **2**, 168–179.
- 26 J. Li, X. Jin, H. Duan, N. Ji, C. Song and Q. Liu, *Procedia Eng.*, 2015, **121**, 967–974.
- 27 P. Hrabanek, A. Zikanova, T. Supinkova, J. Drahokoupil, V. Fila, M. Lhotka, H. Dragounova, F. Laufek, L. Brabec, I. Jirka, B. Bernauer, O. Prokopova, V. Martin-Gil and M. Kocirik, *Microporous Mesoporous Mater.*, 2016, **228**, 107–115.
- 28 Q. Li, D. Creaser and J. Sterte, The Synthesis of Small Colloidal Crystals of TPA-silicalite-1 with Short Synthesis Time and High Yields, in *Porous Materials in Environmentally Friendly Processes*, ed. I. Kiricsi, G. Pál-Borbély, J. B. Nagy and H. G. Karge, Elsevier, Amsterdam, 1999, vol. 125, pp. 133–140.
- 29 P. Pal, J. K. Das, N. Das and S. Bandyopadhyay, *Ultrason. Sonochem.*, 2013, **20**, 314–321.
- 30 Z. Liu, K. Okabe, C. Anand, Y. Yonezawa, J. Zhu, H. Yamada, A. Endo, Y. Yanaba, T. Yoshikawa, K. Ohara, T. Okubo and T. Wakihara, *Proc. Natl. Acad. Sci. U. S. A.*, 2016, **113**, 14267–14271.
- 31 J. Cejka, R. E. Morris and P. Nachtigall, *Zeolites in Catalysis: Properties and Applications*, Royal Society of Chemistry, Croydon, London, 2017.
- 32 *Catalysts for Fine Chemical Synthesis*, ed. E. G. Derouane and S. M. Roberts, John Wiley & Sons, London, 2006.
- 33 U. Schubert and N. Hüsing, *Synthesis of Inorganic Materials*, John Wiley & Sons, Weinheim, Germany, 2012.
- 34 E. Verheyen, L. Joos, K. V. Havenbergh, E. Breynaert, N. Kasian, E. Gobechiya, K. Houthoofd, C. Martineau, M. Hinterstein, F. Taulelle, V. V. Speybroeck, M. Waroquier, S. Bals, G. V. Tendeloo, C. E. A. Kirschhock and J. A. Martens, *Nat. Mater.*, 2012, **11**, 1059–1064.
- 35 Z. Liu, T. Wakihara, D. Nishioka, K. Oshima, T. Takewaki and T. Okubo, *Chem. Commun.*, 2014, **50**, 2526–2528.
- 36 Z. Liu, T. Wakihara, K. Oshima, D. Nishioka, Y. Hotta, S. P. Elangovan, Y. Yanaba, T. Yoshikawa, W. Chaikittisilp, T. Matsuo, T. Takewaki and T. Okubo, *Angew. Chem., Int. Ed.*, 2015, **54**, 5683–5687.
- 37 Z. Liu, N. Nomura, D. Nishioka, Y. Hotta, T. Matsuo, K. Oshima, Y. Yanaba, T. Yoshikawa, K. Ohara, S. Kohara, T. Takewaki, T. Okubo and T. Wakihara, *Chem. Commun.*, 2015, **51**, 12567–12570.
- 38 Z. Liu, T. Wakihara, N. Nomura, T. Matsuo, C. Anand, S. P. Elangovan, Y. Yanaba, T. Yoshikawa and T. Okubo, *Chem. Mater.*, 2016, **28**, 4840–4847.
- 39 Z. Liu, T. Wakihara, C. Anand, S. H. Keoh, D. Nishioka, Y. Hotta, T. Matsuo, T. Takewaki and T. Okubo, *Microporous Mesoporous Mater.*, 2016, **223**, 140–144.
- 40 J. Zhu, Z. Liu, A. Endo, Y. Yanaba, T. Yoshikawa, T. Wakihara and T. Okubo, *CrystEngComm*, 2017, **19**, 632–640.
- 41 J. Zhu, Z. Liu, K. Iyoki, C. Anand, K. Yoshida, Y. Sasaki, S. Sukenaga, M. Ando, H. Shibata, T. Okubo and T. Wakihara, *Chem. Commun.*, 2017, **53**, 6796–6799.
- 42 C. Peng, Z. Liu, A. Horimoto, C. Anand, H. Yamada, K. Ohara, S. Sukenaga, M. Ando, H. Shibata, T. Takewaki, R. R. Mukti, T. Okubo and T. Wakihara, *Microporous Mesoporous Mater.*, 2018, **255**, 192–199.
- 43 N. Nakazawa, S. Inagaki and Y. Kuboto, *Adv. Porous Mater.*, 2016, **4**, 219–229.
- 44 N. Martin, C. Paris, P. N. R. Vennestrom, J. R. Thogersen, M. Moliner and A. Corma, *Appl. Catal., B*, 2017, **217**, 125–136.
- 45 S. I. Zones, *U.S. Pat.*, 4508837, 1985.
- 46 S. I. Zones and R. A. V. Nordstrand, *Zeolites*, 1988, **8**, 409.
- 47 S. I. Zones, *U.S. Pat.*, 5194235, 1993.
- 48 R. F. Lobo, S. I. Zones and R. C. Medrud, *Chem. Mater.*, 1996, **8**, 2409.
- 49 A. Jackowski, S. I. Zones, S.-J. Hwang and A. W. Burton, *J. Am. Chem. Soc.*, 2009, **131**, 1092.
- 50 R. H. Archer, S. I. Zones and M. E. Davis, *Microporous Mesoporous Mater.*, 2010, **130**, 255.
- 51 H. Yamada, T. Iida, Z. Liu, Y. Naraki, K. Ohara, S. Kohara, T. Okubo and T. Wakihara, *Cryst. Growth Des.*, 2016, **16**, 3389–3394.
- 52 S.-H. Lee, C.-H. Shin, G. J. Choi, T.-J. Park, I.-S. Nam, B. Han and S. B. Hong, *Microporous Mesoporous Mater.*, 2003, **60**, 237–247.
- 53 F. Gao, E. D. Walter, E. M. Karp, J. Luo, R. G. Tonkyn, J. H. Kwak, J. Szanyi and C. H. F. Peden, *J. Catal.*, 2013, **300**, 20–29.
- 54 A. Kharchenko, V. Zholobenko, A. Vicente, C. Fernandez, H. Vezin, V. De Waele and S. Mintova, *Phys. Chem. Chem. Phys.*, 2018, **20**, 2880–2889.

