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Fabrication of PP hollow fiber membrane *via* TIPS using environmentally friendly diluents and its CO₂ degassing performance†

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CO₂ removal is an essential water purification process in many fields, such as petrochemical production and thermal power generation. It is challenging to remove low concentrations of CO₂ from RO effluent water. The core component of the membrane degassing technique is a high-performance hydrophobic microporous membrane. Polypropylene (PP) membranes were prepared with environmentally friendly binary diluents *via* thermally induced phase separation. Firstly, the effects of PP concentration on the phase diagram, flat sheet membrane structure and mechanical properties were studied to optimize the PP content for a hollow fiber membrane (HFM). The PP HFM showed a sponge-like cross-sectional structure without any dense skin layer, a large loading force and breaking elongation, and a narrow pore size distribution with a mean pore size of 0.16 μm. The as-fabricated PP HFM module was applied for CO₂ removal from RO effluent water from a petrochemical plant. A higher water temperature and vacuum degree facilitated CO₂ removal. Increasing the effective membrane length enhanced degassing efficiency. Increasing the water flow rate increased CO₂ degassing flux, but simultaneously decreased degassing efficiency. When the water flow rate increased from 20 mL min⁻¹ to 63 mL min⁻¹, although the effective membrane length increased from 3 m to 4.8 m for the best degassing efficiency of 88%, the amount of treated water increased by 3.15 times. The declined CO₂ concentration in the outlet water was 1.6 mg L⁻¹.

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1. Introduction

CO₂ removal from water is an essential water purification process in many fields: for example, petrochemical production, thermal power generation and fabrication of microelectronic devices.¹ Typically, boiler water systems in petrochemical plants require large amounts of high-quality water.² CO₂ in the water needs to be removed before the ion exchange bed, as it adds an excessive load to the ion exchange bed, resulting in frequent chemical regeneration.³ In reverse osmosis (RO) and electro-deionization (EDI) systems, the CO₂ content must also be controlled.⁴ In an RO system, the removal CO₂ can be achieved by adding an alkali such as sodium hydroxide. However, this may lead to more chemical pollutants in RO concentrated water.⁵ Traditional industrial degassing technologies, including packed towers, spray towers and bubble columns, have many

shortcomings such as large floor areas and operational problems like entrainment, flooding, and emulsion foaming.⁶

A hollow fiber membrane (HFM) contactor, as a promising method for CO₂ absorption and desorption, was first introduced in 1985 by Qi and Cussler.⁷ Within this technique, a liquid phase and a gas phase can make non-dispersive contact *via* a microporous membrane that functions as the interface between liquid and gas.^{8,9} Hence, liquid flows inside HFMs, while outside the HFMs, stripping gas or a vacuum were employed. The dissolved gas leaves the liquid phase, penetrates into the membrane pores and then enters the gas phase.¹⁰

Membrane technology has unique advantages, including large interfacial area, independent control of the fluid flow rates, small size, modularity and ease of scaling up or down, as well as operational flexibility.¹¹ Despite the fact that membrane absorption and membrane degassing are two applications for membrane contactors, most research has emphasized CO₂ absorption using gas-liquid membrane contactors. Indeed, investigations into CO₂ degassing through membrane contactors are very rare in the current literature. A few studies reported CO₂ degassing using different HFM modules. Mansourizadeh and Ismail¹² employed porous polyvinylidene fluoride (PVDF) HFMs to evaluate CO₂ stripping performance from water. The results indicated that the liquid phase temperature, gas-liquid contact area and contact time are the

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key parameters for enhancing CO₂ stripping efficiency. Rahbari-Sisakht *et al.*¹³ used a polysulfone HFM contactor for CO₂ stripping from water. The results showed that the gas velocity had a minor effect on the CO₂ desorption flux. Khaisri *et al.*¹⁴ employed polytetrafluoroethylene (PTFE) hollow fiber membranes to investigate the CO₂ desorption performance from CO₂-loaded monoethanolamine (MEA) solutions. The high membrane porosity showed superior desorption performance, but the long-term performance underwent severe degradation due to membrane wetting. Under high temperature conditions, Koonaphapdeelert *et al.*¹⁵ used costly ceramic hollow fiber membrane contactors for CO₂ stripping from MEA solution.

Since membrane fouling and wetting will reduce the degassing efficiency, there are several requirements for ideal membrane materials:¹⁶ (1) outstanding chemical and thermal stability; (2) excellent hydrophobicity; (3) high mechanical strength; (4) feasibility for commercial application. Pore size and pore size distribution also have large effects on degassing performance.

Degassing membranes are mainly produced from hydrophobic polymers, particularly polypropylene (PP), PTFE, polyethylene, and poly(vinylidene fluoride).¹⁷ PP is commonly used for membrane preparation because of its excellent properties. PP membranes are usually produced by stretching or thermal methods, on account of their insolubility in common solvents.¹⁸ In fact, several papers have reported the preparation of hydrophobic microporous PP HFM *via* a thermally induced phase separation (TIPS) process, and membrane structure and morphology were optimized by adjusting various preparation conditions. Lloyd *et al.* studied the solid-liquid (S-L) and liquid-liquid (L-L) phase separation of porous PP membranes prepared *via* TIPS.¹⁹ The parameters of diluent type and its content,²⁰ polymer concentration²¹ *etc.* could influence the compatibility between polymer and diluents, and lead to different phase separation types, which would further determine the membrane morphology, pore size and membrane performance.

Nowadays, most PP membranes prepared *via* TIPS use various single and binary diluents, such as diphenyl ether (DPE),²² soybean oil and dibutyl phthalate (DBP),²³ *etc.* Binary diluents can greatly influence the compatibility between polymer and diluents, and can significantly change the phase separation behavior and membrane morphology. However, some diluents mentioned above may be toxic and harmful to human health. In previous studies, we chose castor oil (CO) and soybean oil (SO) as environmentally friendly binary diluents and optimized the appropriate binary diluents mass ratio.

As mentioned above, the studies of membrane degassing reported so far focus on absorption of high-concentration CO₂ solutions. It is challenging to remove a low concentration of CO₂ from RO effluent water. Herein, microporous PP HFM was prepared *via* TIPS for CO₂ degassing from RO effluent water from a petrochemical enterprise. Firstly, the effects of PP concentration on the phase diagram, membrane structure and mechanical properties were studied during preparation of PP flat-sheet membranes in order to optimize the PP content for

HFM. PP HFM was then fabricated *via* TIPS with the optimal casting solution composition. Furthermore, the CO₂ degassing performance of the PP membrane module was studied with RO effluent water. The effects of feed flow rate, vacuum degree, temperature, and effective membrane length on CO₂ degassing performance were investigated.

2. Experimental and materials

2.1. Materials

SO and CO were bought from COFCO, Co. China. The PP was provided by Sinopec, F280M, MI = 2.8 g/10 min (190 °C, 2.16 kg), $M_w = 400\ 000$, $M_w/M_n = 7.7$. Sodium hydroxide was purchased from Beijing Chemical Works, China. Phenolphthalein and potassium hydrogen phthalate were purchased from Shanghai Macklin Biochemical, China. Ethanol and *n*-hexane of analytical grade were purchased from China Sino-pharm Chemical Reagent Co., China.

2.2. Phase diagram and membrane preparation

Homogeneous PP-diluents solid samples were prepared by a previously described method.²⁴ The sample preparation conditions are listed in detail in Table S1 (in ESI[†]). A small piece of solid sample was sandwiched between two glass coverslips, sealed around with vacuum grease to limit evaporation of diluents. The sample was heated on a hot stage (Linkam THMS 600, UK) up to 200 °C at a heating rate of 50 °C min⁻¹, maintained for 5 min to assure homogeneity, then cooled to about 50 °C at a cooling rate of 10 °C min⁻¹. The cloud point temperature of the sample was determined visually by noting the appearance of turbidity under a phase contrast mode microscope (BX51M, Olympus, JAPAN). The crystallization temperature of the sample was determined visually by noting the appearance of nucleation under a polarized microscope (BX51M, Olympus, JAPAN). The temperature values were obtained based on an average of three parallel measurements.

Detail of the preparation of PP flat-sheet membranes and PP HFMs *via* TIPS can be found in ESI.[†]

2.3. Membrane characterization

2.3.1. Scanning electron microscopy (SEM). The morphologies of the flat-sheet membranes and HFM were photographed on an SEM microscope (Hitachi S-4800, Japan) under an accelerating voltage of 1.0 kV. The surface and cross-section of the membrane were sputtered with Au in a vacuum for 50 s.

2.3.2. Pore size and pore size distribution. The pore size and pore size distribution of the PP HFMs were measured by a fully-automatic bubble pressure method filter membrane pore size analyzer (3H-2000PB, BeiShiDe Instrument Technology Co., Ltd., China). The membrane samples consisted of three HFMs with a length of 20 cm. Before measurement, the membranes were fully infiltrated with Porofil liquid (surface tension 16 dyn cm⁻¹).

2.3.3. Mechanical properties test. The mechanical properties of the PP flat-sheet membranes and PP HFMs were measured with a universal machine (3342 Universal Material



Testing Machine, Instron, USA). The PP flat-sheet membranes were cut into 10 mm wide strips and the PP HFMs were cut into lengths of about 60 mm. The gap between clamps was 40 mm, and the test was conducted with a strain rate of 40 mm min⁻¹ at room temperature. The thickness of the PP flat-sheet membranes and the diameter of the PP HFMs were measured using an optical microscope (Olympus CX31, Japan). The mechanical properties were examined according to the obtained strength-strain curve. The average of three parallel results was reported.

2.3.4. Water contact angle and liquid entry pressure test.

The details of testing the water contact angle (WCA) and liquid entry pressure (LEP) were shown in our previous report.²⁴

2.3.5. Porosity determination. The porosity (ϵ) of the membrane was determined according to a method based on density measurements.¹²

2.3.6. Gas permeability test. The PP HFM sample was fixed in a self-made test tube by connecting with a needle pressurized at 0.010 MPa by a nitrogen bottle. Then, the gas flow rate for a certain period of time was measured by a soap film flowmeter connected to the test tube until the rate was stable, and the gas permeability was calculated based on the membrane area.

2.4. CO₂ degassing experiment

A total of 10 PP HFMs were packed randomly in a polymethyl methacrylate tube with the two ends sealed by epoxy resin for fabrication into a membrane module. The specifications of the membrane module are given in Table 1.

The RO effluent water with dissolved CO₂ flowed through the lumen side of the membrane module and a vacuum was introduced on the shell side. This effluent water was collected from the RO system at the Luoyang Petrochemical Engineering Corporation Ltd/SINOPEC. The water temperature was kept constant with a water bath and the flow rate was controlled by a peristaltic pump. The vacuum degree was adjusted by a vacuum regulating valve. A flow diagram of the experimental setup is shown in detail in Fig. S1 (in ESI†). The CO₂ concentrations of the entry and outlet water of the module were measured by the chemical titration method.¹³

The CO₂ degassing efficiency was calculated as:

$$\eta(\%) = \left(1 - \frac{C_{o,l}}{C_{i,l}}\right) \times 100\% \quad (1)$$

where $C_{o,l}$ and $C_{i,l}$ are the liquid phase CO₂ concentrations (mg L⁻¹) in the outlet and inlet of the membrane contactor, respectively.

Table 1 Specifications of the PP HFM module

Properties	Values
Length of module/cm	70
Effective length/cm	60
Module I.D/O.D/cm	4.3/5.0
Fiber I.D/O.D/ μ m	1161/1822
No. of fibers	10

The CO₂ degassing flux was calculated as:

$$J_{CO_2} = \frac{(C_{i,l} - C_{o,l}) \times Q_1}{A_i} \quad (2)$$

where J_{CO_2} is the CO₂ degassing flux (mol m⁻² s⁻¹); Q_1 is the feed flow rate (m³ s⁻¹); and A_i is the inner surface area of the PP HFMs (m²).

3. Results and discussion

3.1. Phase diagram

The type of phase separation is directly determined from the phase diagram.^{20,25} Fig. 1 is the phase diagram of the PP/binary diluents system on various PP concentrations with 16 wt% CO in the diluents. It presents cloud point and dynamic crystallization temperatures. The cloud point decreased greatly with an increase in PP concentration. The cloud point reached its highest value of 176 °C at 10 wt% PP, and dropped to 136 °C at 40 wt% PP. The phenomena can be analysed in terms of the interaction between PP and the binary diluents. The Flory-Huggins interaction parameter (χ) is typically used to interpret the interaction between the polymer and the diluent. When the PP concentration was increased, the χ became smaller, which resulted in the decrease in cloud point.²⁰ This indicated that when the PP concentration ranged from 10 wt% to 40 wt%, the PP/binary diluents system showed a typical L-L phase separation. However, a high viscosity of casting solution is unfavorable for membrane preparation due to the high PP concentration. Thus, the PP content was controlled at 40 wt%.

Moreover, Fig. 1 shows that the crystallization temperature of PP remains almost constant within the range of PP concentrations studied, which is characteristic of the crystal phase formed in L-L phase separation and is consistent with the results reported previously.²⁰ The crystallization temperature of the original PP in this study was about 127 °C, the addition of binary diluents contributed to this slight decrease.

3.2. Effect of PP concentration on the flat-sheet membrane properties

3.2.1. Membrane morphology. The effect of PP content on membrane morphology was investigated by fixing a binary

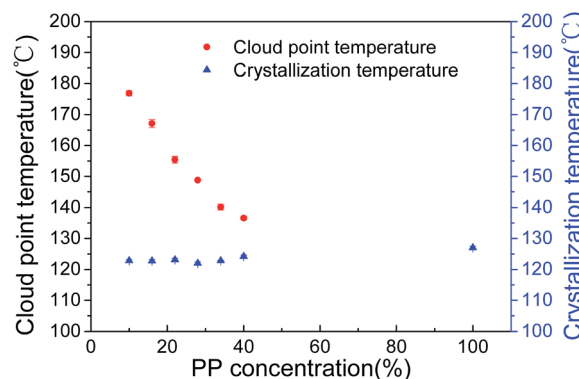


Fig. 1 Phase diagram for the PP/binary diluents system with various PP contents: diluents content (SO/CO): 84/16.



diluents containing 16 wt% CO. The surface and cross-sectional morphologies of the membranes were observed by SEM. As shown in Fig. 2, the cross-section of the flat-sheet membranes with different PP concentrations remains an interconnected sponge-like structure, which confirms the occurrence of L-L phase separation. Moreover, the cross-sectional structure becomes highly compact, and the pores are almost closed when the PP concentrations are 34 wt% and 40 wt%. The surface of all the flat-sheet membranes shows a non-porous structure due to the insolubility between the casting solution and the water coagulation bath. On the other hand, the fast solidification caused by rapid cooling when the homogeneous casting solution was quenched from 195 °C to the 15 °C also caused a dense surface.²⁶

The SEM images also show that the pore size of the flat-sheet membranes decreased slightly with an increase in PP concentration. Phase separation was induced by lowering the temperature in the TIPS process. The PP phase formed the membrane skeleton and the diluent phase became the membrane pores upon diluent extraction. Therefore, the increase in PP concentration decreased membrane pore size.²⁷ On the other hand, during the L-L phase separation process, when the PP concentration increased, the L-L phase separation region became smaller, and then the lean-polymer phase grew for a shorter time, so the pore size decreased.¹⁹

3.2.2. Mechanical properties. Fig. 3 shows the mechanical properties of PP flat-sheet membranes with different PP concentrations. The results indicated that the loading force became higher with PP concentration. The breaking elongation also increased with PP concentration, but when the PP

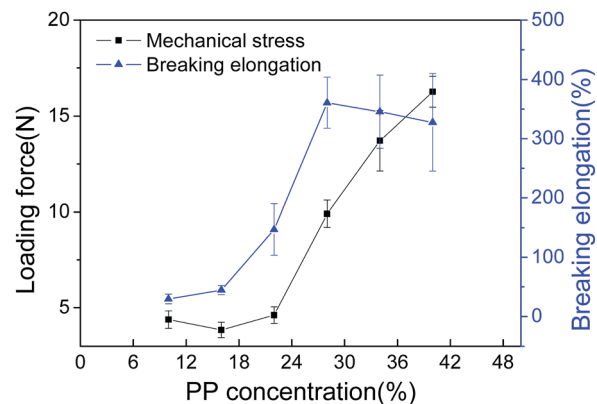


Fig. 3 Mechanical properties of PP membranes prepared with different PP concentrations.

concentration increased to 40 wt%, the breaking elongation decreased slightly. It was observed that the membranes prepared with more than 28 wt% PP became both crispy and hard, and, as also shown in Fig. 2, the pores in cross-section became compact, which resulted in the decline in breaking elongation. A similar phenomenon was also found in our previous study, where PP flat-sheet membranes were prepared using carnauba wax and SO as binary diluents.²⁴

The results in Fig. 1 indicated that the PP/SO/CO systems with PP concentrations ranging from 10 wt% to 40 wt% all underwent L-L phase separation. If the PP concentration was higher than 28 wt%, it was difficult to scrape the membranes due to the high viscosity of the casting solution. On the other hand, when the PP concentration was less than 20 wt%, the loading force of the resulting membranes was poor. In addition, the pore size of the membranes decreased with increasing PP concentration. Furthermore, the relatively small pore size was favorable for resistance to water penetration. Therefore, considering the smaller pore size and high mechanical properties, 28 wt% PP was appropriate for this PP/diluents system.

3.3. Characterizations of PP HFM

3.3.1. Membrane morphology. Based on the effects of PP concentration on the phase diagram, loading force and breaking elongation of flat-sheet membrane, the selected compositions were 28 wt% PP with a mass ratio CO/SO of 16/84. Fig. 4 shows the structure of the outer surface, inner surface and cross-sectional structure of PP HFM. Prepared at a relatively slow cooling rate, the membrane showed discernable spherulites in its overall look, indicating that L-L phase separation mainly occurred. However, the membrane structure in the interspherulitic space and intra-spherulitic space showed an interconnected sponge-like structure.

As also shown in Fig. 4, the inner and outer surfaces have no dense skin layer but a loose porous morphology. In PP HFM preparation, diluent-like solvents with a 90/10 mass ratio of SO/CO were applied in both the bore liquid and the outer coagulation baths. The diluent-like solvents had good compatibility with the binary diluents, and could exchange with the casting

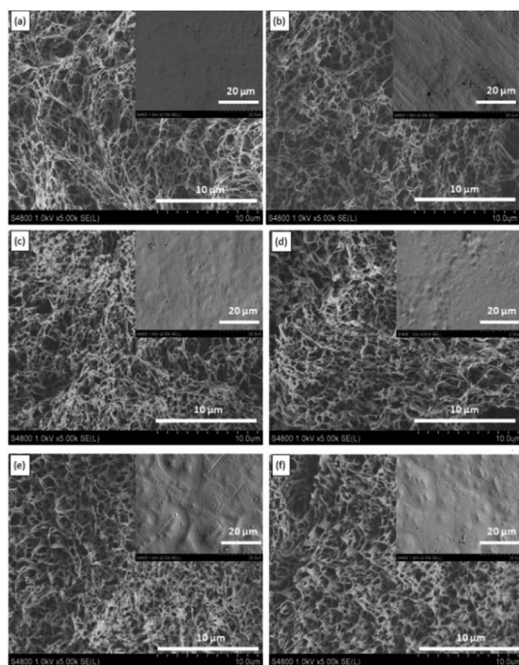


Fig. 2 SEM images of PP flat-sheet membranes using various PP concentrations. (PP concentration: (a) 10 wt%; (b) 16 wt%; (c) 22 wt%; (d) 28 wt%; (e) 34 wt%; (f) 40 wt%. Cross-section at magnification 5k and inserts showing the surface at magnification 2k).



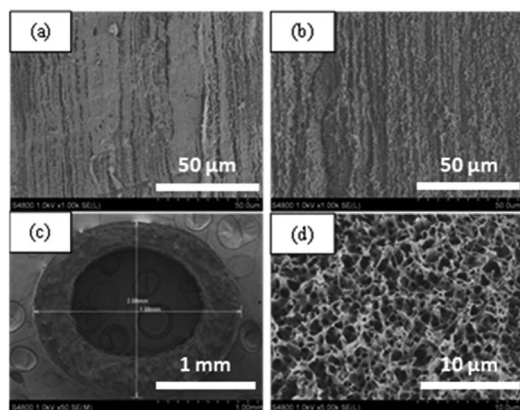


Fig. 4 SEM images of PP HFM: (a) inner surface; (b) outer surface; (c) cross-section with magnification 50; (d) cross-section with magnification 5k.

solution on the inner and outer surfaces to result in a porous structure.²⁶

3.3.2. Mechanical properties. Fig. 5 shows the strength-strain curves for three parallel tests of PP HFMs. According to the strength-strain curve analyses of the PP HFMs during the measurements, the instrument automatically calculated a loading force of 6.78 ± 0.23 N, a tensile strength of 4.38 ± 0.15 MPa, a Young's modulus of 123.7 ± 19.7 MPa and a breaking elongation of $27.87 \pm 3.9\%$. It should be noticed that the breaking elongation of HFM was weaker than that of the flat-sheet membrane with the same composition of casting solution, which was mainly due to the relatively slow cooling rate. The PP HFM has excellent breaking elongation and strength. A comparison of the mechanical properties with the literature data is shown in Table S2 (in ESI†).

3.3.3. Pore size and pore size distribution. The pore size and pore size distribution of the PP HFMs prepared with 16 wt% CO in the diluents are shown in Fig. 6. The results indicate that the PP HFMs presented a narrow and uniform pore size distribution with a mean pore size of $0.16 \mu\text{m}$.

This narrow pore size structure is helpful for preventing the membrane pore from being wetted and penetrated by water, for a long-term stable performance.

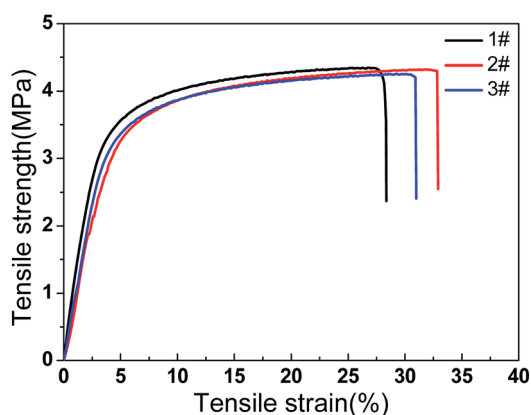


Fig. 5 Strength versus strain curves of three parallel tests of PP HFMs.

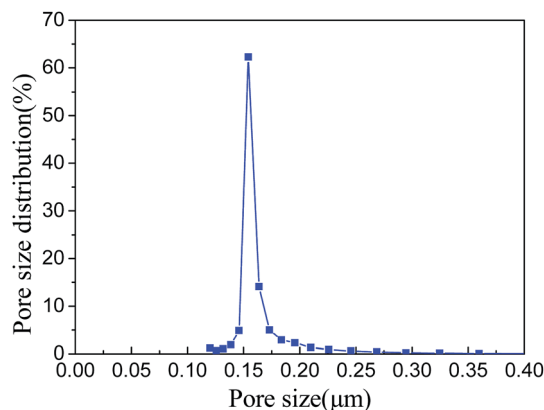


Fig. 6 Pore size and pore size distribution of PP HFM.

The other properties of PP HFM, including water contact angle (WCA), liquid entering pressure (LEP), *etc.* are summarized in Table S3 (ESI†).

3.4. Effects of operational parameters on degassing performance

3.4.1. Feed flow rate. The CO_2 concentration of the employed RO effluent water was 12.56 mg L^{-1} . As shown in Fig. 7(a), the degassing efficiency decreased gradually with an increase in water flow rate. The largest CO_2 degassing efficiency was 57.4% when the flow rate was 20 mL min^{-1} . Although the reduced liquid phase boundary thickness enhanced the mass transfer when the flow rate increased, the residence time of water in the membrane decreased sharply. This result demonstrated that the gas-liquid contact time played an important role in CO_2 degassing. Therefore, a higher degassing efficiency should be achieved at a longer residence time.

However, the degassing flux increased gradually with water flow rate, indicating that the increase in flux was mainly caused by increasing flow rate, although the CO_2 concentration change was smaller.²⁸ Hence, in terms of CO_2 degassing flux, a higher flow rate was considered to be better. At a higher water flow rate, the mass transfer resistance of the liquid phase became smaller, but the operating energy consumption might increase. However, the flow state was still in the laminar flow region (Reynolds number 126) at a flow rate of 63 mL min^{-1} . This is acceptable considering the fluid resistance energy consumption.

3.4.2. Vacuum degree. Fig. 7(b) shows that the degassing efficiency and degassing flux both increased slightly when the vacuum degree increased, mainly due to the enhancement in mass transfer driving force. When the vacuum degree increased, the mass transfer driving force apparently became larger.

3.4.3. Feed water temperature. As shown in Fig. 7(c), the degassing efficiency increased slightly within a narrower elevated temperature range. This result could be attributed to the influence of two variables: solubility and diffusion coefficient. The CO_2 solubility decreased in water with an elevated temperature. Moreover, the CO_2 diffusion coefficient



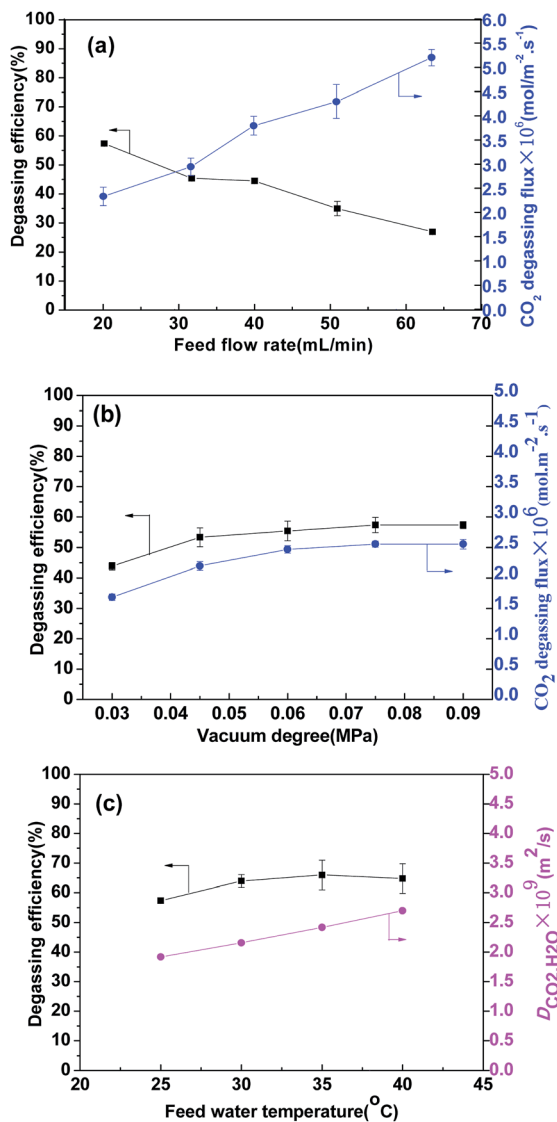


Fig. 7 Effects of operational parameters on degassing performance. (a) Feed flow rate; (b) vacuum degree; (c) feed water temperature. (Operating conditions: (a) water temperature of 298 K; vacuum degree of 0.095 MPa. (b) water temperature of 298 K; flow rate of 20 mL min⁻¹ (c) CO₂ concentration of 16.5 mg L⁻¹; vacuum degree of 0.095 MPa; flow rate of 20 mL min⁻¹).

increased.²⁹ These changes led to the improvement in degassing performance. However, after 30 °C, the degassing performance was basically stable.

3.4.4. Effective membrane length. In order to enhance the degassing efficiency, one efficient approach is to increase the effective membrane length, thereby prolonging the water retention time in the membrane. Generally, the membrane modules could be connected in series. However, due to restrictions on the number of modules, this experiment was carried out by increasing the operating circle times. Herein, one degassing operation means that the effective membrane length is 0.6 m; for two operations the effective membrane length is 1.2 m, and so on.

The effect of effective membrane length under two flow rates was investigated and the results are presented in Fig. 8. In

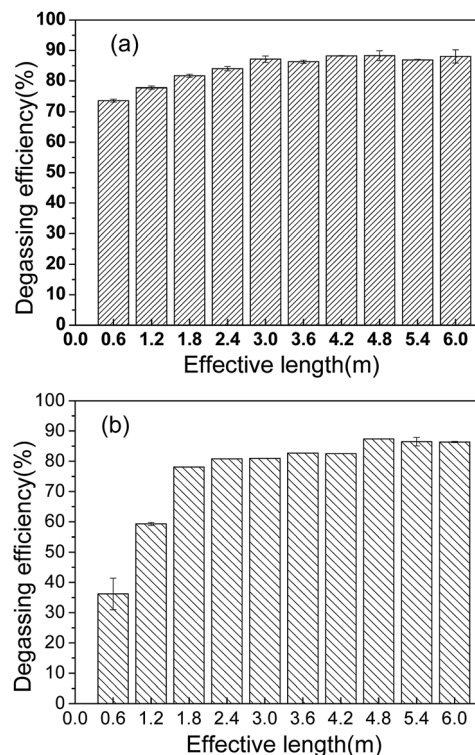


Fig. 8 Effect of effective length on degassing efficiency. (Operating conditions: initial CO₂ concentration of 16.5 mg L⁻¹; vacuum degree of 0.095 MPa; flow rate of (a) 20 mL min⁻¹ and (b) 63 mL min⁻¹).

Fig. 8(a), when the effective length increased from 0.6 m to 3 m, the CO₂ removal efficiency increased. When the length reached 3 m, it was basically stable at about 88%, and the CO₂ content of the outlet water declined to 1.6 mg L⁻¹. From Fig. 8(b) for 63 mL min⁻¹, the degassing efficiency of the first circle got smaller. But it increased rapidly in the next two circles. After that, the increase trend slowed down. The final degassing efficiency reached about 87% at 4.8 m. The Reynolds number in the fiber lumen increased from 40 to 126 when the flow rate varied from 20 mL min⁻¹ to 63 mL min⁻¹. The mass transfer at a higher flow rate was enhanced. It was concluded that the degassing performance could be improved with a longer membrane length or at a higher flow rate. For a single membrane module, it seems that only when the membrane is long enough can a desired degassing efficiency be achieved. For a large processing capacity in industrial applications, a higher water flow rate was considered to be better. As for the CO₂ removal efficiency, it can be ensured by increasing operation circle times or connecting the modules in series.

4. Conclusions

Hydrophobic PP HFMs prepared *via* TIPS with environmentally friendly binary diluents were fabricated into a membrane contactor for CO₂ degassing from RO effluent water from petrochemical production. The following conclusions were drawn:

(1) The phase diagram indicated that the cloud point temperature decreased with PP concentrations from 10 wt% to 40 wt%, but the crystallization temperature was almost stable.



The cross-sections of the PP flat-sheet membranes displayed a sponge-like structure. The largest breaking elongation was achieved at 28 wt% PP.

(2) PP HFM showed a sponge-like cross-sectional structure without any dense skin layer, a large loading force and breaking elongation, and a narrow pore size distribution with a mean pore size of 0.16 μm .

(3) When the flow rate increased from 20 mL min^{-1} to 63 mL min^{-1} , the effective membrane length increased from 3 m to 4.8 m for the best degassing efficiency of about 88%; the amount of treated water or CO_2 removed increased by 3.15 times.

Conflicts of interest

There are no conflicts to declare.

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