

Cite this: *RSC Adv.*, 2019, 9, 17653

Dual-functional persistent luminescent nanoparticles with enhanced persistent luminescence and photocatalytic activity†

Ailijiang Tuerdi and Abdukader Abd McKayum *

Dual-functional nanoparticles with near-infrared (NIR) persistent luminescence and sun-light photocatalytic activity are highly desired for medical diagnosis and environmental protection. Here, we report a facile one-step method for simultaneous enhancement of persistent luminescence and photocatalytic activity of the dual-functional persistent luminescent nanoparticles (PLNPs). The Bi³⁺, Cr³⁺ co-doped ZnGa₂O₄ PLNPs, which were less than 10 nm in size, were synthesized by an ethylene glycol-assisted hydrothermal method. The persistent luminescence and the photocatalytic activity of the PLNPs were significantly and simultaneously improved *via* additional doping of Bi³⁺ in ZnGa₂O₄:Cr. The prepared dual-functional PLNPs have great potential in pollutant photo-degradation and long-term imaging *in vivo*.

Received 23rd March 2019
Accepted 28th May 2019

DOI: 10.1039/c9ra02235j

rsc.li/rsc-advances

1. Introduction

The long-lasting phosphorescence (or afterglow) of persistent luminescent materials is an optical phenomenon whereby luminescence can remain for a long time (hours to days) after the excitation source is removed.^{1–3} In recent years, persistent luminescent materials have been drawing increasing attention in many areas, such as emergency signage, traffic signs, lighting, photocatalysis, and medicine.^{4–7} Molecular probes based on persistent luminescent nanoparticles (PLNPs) exhibit a super-long afterglow time and do not need *in situ* excitation during biomedical imaging, so they completely avoid tissue auto-fluorescence, light scattering, and photo-toxicity originating from the excitation light.^{8–10} PLNPs, alone or in combination with other photocatalysts such as TiO₂, have great potential in photocatalysis owing to their potentially increased solar light absorption during day time and subsequent increased solar energy release at night or after the removal of illumination source.^{3,11} Therefore, dual-functional PLNPs with near-infrared (NIR) persistent luminescence and photocatalytic activity have great potential in medical diagnosis and environmental protection.

Zinc gallate (ZnGa₂O₄) is a cubic normal AB₂O₄ spinel crystal semiconductor with the *Fd3m* space group.¹² It is one of the excellent blue-emitting phosphors with wide band gaps (4.4 eV for ZnGa₂O₄) and has great potential applications in displays,

owing to its strong blue emission, better chemical and thermal stability, and excellent cathodoluminescence properties.¹³ ZnGa₂O₄ also acts as an excellent photo-catalyst due to its hybridized orbitals of Zn4s4p, Ga4s4p, and band gap, which can improve the mobility of photo-generated electrons and the absorption efficiency of ultraviolet (UV) light.^{5,14}

ZnGa₂O₄ is also an excellent photoluminescence host material when doped with transition metals, such as Mn-doped ZnGa₂O₄ for green emission, and Cr³⁺-doped ZnGa₂O₄ (ZnGa₂O₄:Cr) for red or near-infrared (NIR) emission.¹⁵ In particular, ZnGa₂O₄:Cr is a bright NIR-emission persistent luminescence phosphor, which is considered a promising optical nanoprobe for *in vivo* bio-imaging.^{16–18} Co-doping of ZnGa₂O₄ or zinc gallogermanate systems with Cr³⁺ and rare earth ions, such as Pr³⁺ and Dy³⁺, can significantly improve the persistent luminescence intensity and afterglow time of PLNPs, because trivalent lanthanide ions with special 4f electron configurations can provide more efficient trap density and depth.^{9,19,20} However, afterglow time of the Cr³⁺-doped ZnGa₂O₄ is still not long enough for long-term *in vivo* imaging. Furthermore, the Cr³⁺-doped ZnGa₂O₄ is hardly used on photocatalytic degradation owing to its lower photocatalytic activity, compared to un-doped ZnGa₂O₄. Therefore, the development of an effective one-step methodology to simultaneously improve the persistent luminescence and the photocatalytic activity of dual-functional PLNPs are highly desired for pollutant photo-degradation and long-term imaging *in vivo*. Nevertheless, to the best of our knowledge, no work on simultaneous enhancement of persistent luminescence and photocatalytic activity for Cr³⁺ co-doped ZnGa₂O₄ have been reported so far.

Here, we report a facile one-step method for simultaneous enhancement of persistent luminescence and photocatalytic

Xinjiang Laboratory of Native Medicinal and Edible Plant Resources Chemistry, College of Chemistry and Environmental Sciences, Kashgar University, Kashgar 844007, China. E-mail: abdukadera@sina.com

† Electronic supplementary information (ESI) available. See DOI: 10.1039/c9ra02235j



activity of the dual-functional PLNPs. The Bi^{3+} , Cr^{3+} co-doped ZnGa_2O_4 PLNPs, which were less than 10 nm in size, were synthesized by an ethylene glycol-assisted hydrothermal method. The NIR persistent luminescence and photocatalytic activity of Cr^{3+} co-doped ZnGa_2O_4 are significantly, and simultaneously, improved by additional doping with Bi^{3+} . The prepared dual-functional PLNPs are promising for *in vivo* bio-imaging and photocatalytic application.

2. Experimental

2.1 Sample preparation

Bi^{3+} , Cr^{3+} co-doped ZnGa_2O_4 PLNPs were prepared by an ethylene glycol assisted hydrothermal method. To prepare the $\text{ZnGa}_{1.97}\text{O}_4:\text{Cr}_{0.01}/\text{Bi}_{0.02}$, 10.0 mL of ethylene glycol, 3.94 mL of Ga^{3+} (0.2 mol L^{-1}), 1 mL of Zn^{2+} (0.4 mol L^{-1}), 0.4 mL of Cr^{3+} (0.01 mol L^{-1}), and 0.8 mL of Bi^{3+} (0.01 mol L^{-1}) were mixed and stirred at room temperature for 30 min. The pH of the mixture solution was then adjusted to 9 with aqueous ammonia (15 wt%) and stirred at room temperature for 1 h. The resulting mixture was transferred to a 20 mL Teflon-lined stainless-steel autoclave in ambient environment and heated at 170 °C for 24 h. The autoclave was then allowed to cool to room temperature. The purified products were separated by centrifugation and the NPs were precipitated and washed with distilled water and ethanol and dried in a vacuum oven at 60 °C for 12 h. Finally, the PLNPs were annealed in air at 1000 °C for 3 h.

2.2 Characterization

X-ray diffraction (XRD) patterns were recorded on a D/max-2500 diffractometer (Rigaku, Japan) using $\text{Cu K}\alpha$ radiation ($\lambda = 1.5418 \text{ \AA}$). Transmission electron microscopy (TEM) and high-resolution TEM images were obtained on JEOL-100CX II and JEM-2100F field emission transmission electron microscopes (JEOL, Japan), respectively. The elemental composition of samples was analyzed with an EDAX Genesis XM2 attached to the S-4800 SEM (Hitachi, Japan). X-ray photoelectron spectra (XPS) were recorded on Escalab 250Xi (Thermo Fisher Scientific, USA) X-ray photoelectron spectrometer. The excitation and emission spectra and the long-lasting luminescence decay curves were obtained at room temperature using a F-4500 fluorescence spectrophotometer (Hitachi, Japan). The UV-Vis diffuse reflectance spectra were obtained using a UV-2600 UV-Vis Spectrophotometer (Shimadzu, Japan).

3. Results and discussion

3.1 Structure and morphology

Fig. 1 shows the XRD patterns of ZnGa_2O_4 , $\text{ZnGa}_2\text{O}_4:\text{Cr}$, and $\text{ZnGa}_2\text{O}_4:\text{Cr}/\text{Bi}$. All the peaks are assigned to the ZnGa_2O_4 sample with 2θ at 18.45, 30.28, 35.7, 37.4, 43.58, 57.56, 63.19, 65.82, and 74.72, which were indexed to the (111), (220), (311), (222), (400), (422), (511), (440), and (531) crystal planes, respectively.²⁰ All diffraction peaks are consistent with the ZnGa_2O_4 spinel phase (JCPDS no. 38-1240). When the contents of Bi are 0.01 and 0.02, no characteristic peaks of other phases, such as ZnO , Ga_2O_3 and

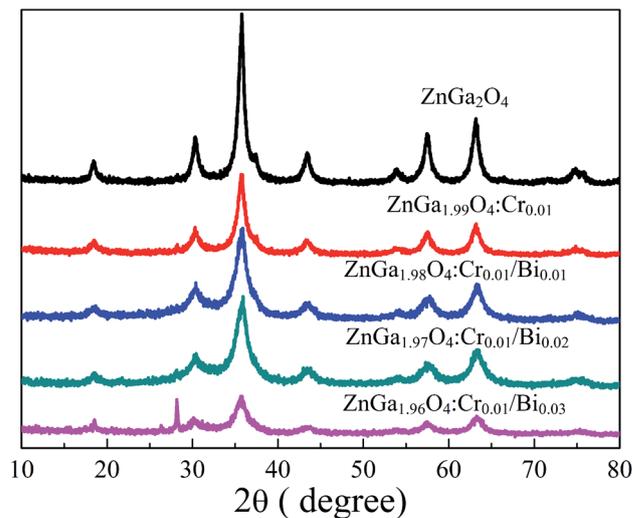


Fig. 1 XRD patterns of ZnGa_2O_4 , $\text{ZnGa}_2\text{O}_4:\text{Cr}$ and $\text{ZnGa}_2\text{O}_4:\text{Cr}/\text{Bi}$.

Bi_2O_3 , were observed, indicating the formation of a pure spinel phase zinc gallate solid solution. When the content of Bi^{3+} in $\text{ZnGa}_2\text{O}_4:\text{Cr}/\text{Bi}$ was increased from 0.02 to 0.03, we observed a strong peak at 28.01° , which was indexed to the rhombohedral Bi structure. The result indicates that the increasing content of Bi^{3+} influences the crystal structure of ZnGa_2O_4 .

Fig. 2a shows TEM image of the as-synthesized $\text{ZnGa}_{1.97}\text{O}_4:\text{Cr}_{0.01}/\text{Bi}_{0.02}$ PLNPs and reveals that the PLNPs are less than 10 nm in size. The high-resolution TEM analysis shown in Fig. 2b indicates that the particles are single crystalline and the distance between the lattice fringes is 4.81 Å, which corresponds to the d-spacing of the spinel ZnGa_2O_4 (111) lattice planes. The selected-area electron diffraction (SAED) pattern further confirmed that the crystal of as-synthesized $\text{ZnGa}_{1.97}\text{O}_4:\text{Cr}_{0.01}/\text{Bi}_{0.02}$ PLNPs is a pure spinel single crystal (Fig. 2c). The results of EDX analysis are demonstrated that the $\text{ZnGa}_{1.97}\text{O}_4:\text{Cr}_{0.01}/\text{Bi}_{0.02}$ PLNPs were composed of only Zn, Ga, Cr, Bi, and O (Fig. 2d). These results revealed that the ZnGa_2O_4 crystal structure was successfully doped with Bi and Cr.

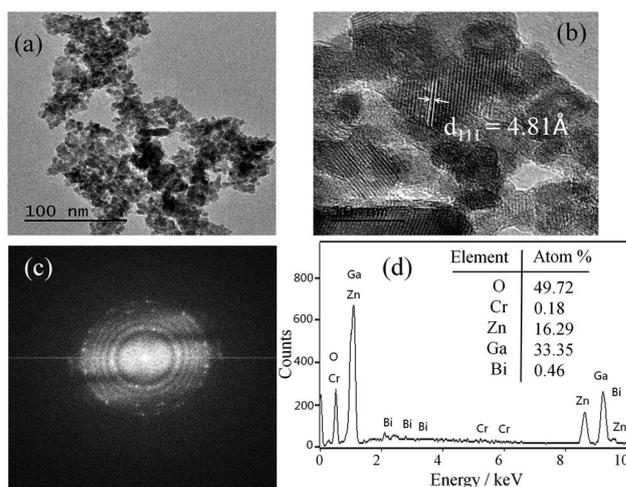


Fig. 2 (a–c) TEM, HR-TEM, SAED images and (d) EDX patterns of $\text{ZnGa}_{1.97}\text{O}_4:\text{Cr}_{0.01}/\text{Bi}_{0.02}$.



3.2 Persistent luminescence of PLNPs

Fig. 3 shows the phosphorescence emission spectra of $\text{ZnGa}_2\text{O}_4:\text{Cr}$ and $\text{ZnGa}_2\text{O}_4:\text{Cr}/\text{Bi}$ PLNPs at room temperature at an excitation wavelength of 254 nm. $\text{ZnGa}_2\text{O}_4:\text{Cr}$ produced an intensive NIR emission band peaking at 698 nm (assigned to ${}^2\text{E} \rightarrow {}^4\text{A}_2$ transition of Cr^{3+} ions) that superimposes on a broad emission band at 600–800 nm (assigned to ${}^4\text{T}_2 \rightarrow {}^4\text{A}_2$ transition of disordered Cr^{3+} ions).² The emission spectra of $\text{ZnGa}_2\text{O}_4:\text{Cr}/\text{Bi}$ are similar to $\text{ZnGa}_2\text{O}_4:\text{Cr}$, but the emission peaks of $\text{ZnGa}_2\text{O}_4:\text{Cr}/\text{Bi}$ were slightly red-shifted due to the crystal field strength of Cr^{3+} centres decreased by additional doping of Bi^{3+} in $\text{ZnGa}_2\text{O}_4:\text{Cr}$.²¹ Fig. S1 (ESI[†]) shows the phosphorescence excitation spectra of $\text{ZnGa}_2\text{O}_4:\text{Cr}$ and $\text{ZnGa}_2\text{O}_4:\text{Cr}/\text{Bi}$. There are four main absorption bands. The strong band peak around 260 nm is probably the combination of the ZnGa_2O_4 host excitation band and the O–Cr charge transfer band.⁹ The bands peaked at approximately 411 nm, 471 nm, and 554 nm and originated from the ${}^4\text{A}_2 \rightarrow {}^4\text{T}_1(\text{te}^-)$ transition, the ${}^4\text{A}_2 \rightarrow {}^4\text{T}_1(\text{t}^2\text{e}^-)$ transition, and the ${}^4\text{A}_2 \rightarrow {}^4\text{T}_2$ transition originated from the 3d intrashell transitions of Cr^{3+} , respectively.²

Fig. 4 shows the persistent luminescence emission decay curves of $\text{ZnGa}_2\text{O}_4:\text{Cr}$ and $\text{ZnGa}_2\text{O}_4:\text{Cr}/\text{Bi}$ PLNPs, monitored at 698 nm after being excited by 254 nm UV light for 5 min. The data represent the persistent luminescence intensity (I) as a function of time (t) and the data recording lasted for 20 min. The persistent luminescence intensity increased as the content of Bi increased from 0 to 0.02, but decreased with a further increase from 0.02 to 0.03. The most intense persistent luminescence was obtained for $\text{ZnGa}_{1.97}\text{O}_4:\text{Cr}_{0.01}/\text{Bi}_{0.02}$. The decay curves of the persistent luminescence can be well fitted by the three-exponential eqn (1) as follows:⁹

$$I(t) = I_0 + A_1 \exp\left(-\frac{t}{\tau_1}\right) + A_2 \exp\left(-\frac{t}{\tau_2}\right) + A_3 \exp\left(-\frac{t}{\tau_3}\right) \quad (1)$$

where $I(t)$ is the intensity of persistent luminescence at time t ; I_0 is the initial intensity of persistent luminescence; A_1 , A_2 , and A_3 are constants; τ_1 , τ_2 , and τ_3 are derived lifetimes for the exponential components indicating the three different decay processes. The fitting results according to the above formula are

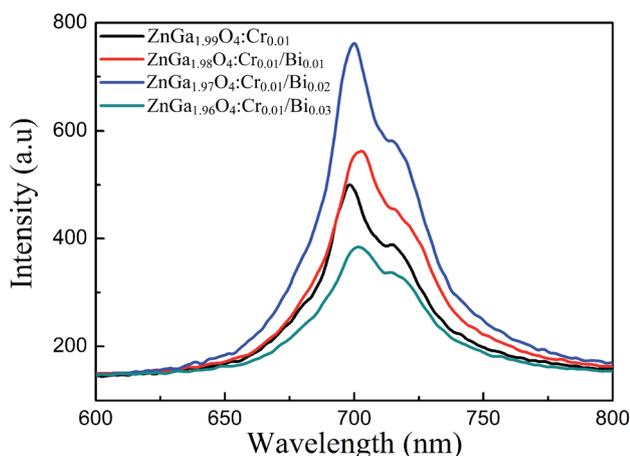


Fig. 3 Phosphorescence emission spectra of $\text{ZnGa}_2\text{O}_4:\text{Cr}$ and $\text{ZnGa}_2\text{O}_4:\text{Cr}/\text{Bi}$ at an excitation wavelength of 254 nm.

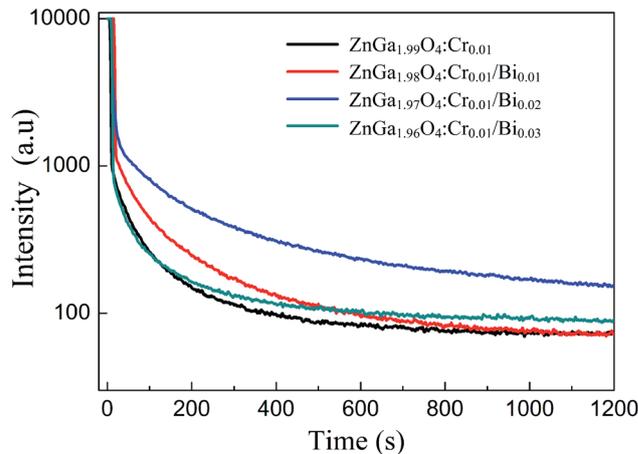


Fig. 4 Afterglow decay curve of $\text{ZnGa}_2\text{O}_4:\text{Cr}$ and $\text{ZnGa}_2\text{O}_4:\text{Cr}/\text{Bi}$ after irradiation with a 254 nm UV lamp for 5 min.

presented in Table S1 (ESI[†]). These average photoluminescence lifetimes (τ_{av}) indicate that the $\text{ZnGa}_{1.97}\text{O}_4:\text{Cr}_{0.01}/\text{Bi}_{0.02}$ PLNPs produced a much longer photoluminescence lifetime (32.41 s) than the single Cr^{3+} -doped ZnGa_2O_4 PLNPs (9.35 s). The above experimental results indicated that the additional doping of Bi^{3+} cannot cause remarkable changes in the excitation and emission spectra of $\text{ZnGa}_2\text{O}_4:\text{Cr}$ but can significantly improve the intensity and time of persistent luminescence for $\text{ZnGa}_2\text{O}_4:\text{Cr}$ PLNPs. Previous work reported that additional doping with Bi^{3+} in $\text{ZnGa}_2\text{O}_4:\text{Cr}$ can also enhance the persistent luminescence of PLNPs.^{22,23} The Cr^{3+} prefers to substitute for Ga^{3+} (with a similar ionic radius) in distorted octahedral coordination in $\text{ZnGa}_2\text{O}_4:\text{Cr}$. The Bi^{3+} substituting for Ga^{3+} can cause a higher periodic lattice distortion in $\text{ZnGa}_2\text{O}_4:\text{Cr}$, because the radius of Bi^{3+} (103 pm) is bigger than that of Ga^{3+} (62 pm). The higher periodic lattice distortion could generate more defects in the ZnGa_2O_4 host.²⁴ Doping with Bi^{3+} probably increases the afterglow time of $\text{ZnGa}_2\text{O}_4:\text{Cr}$ PLNPs by producing more hole traps.¹⁹

3.3 Photocatalytic activity of PLNPs

Fig. 5 shows the UV-Vis diffuse reflectance spectra of ZnGa_2O_4 , $\text{ZnGa}_2\text{O}_4:\text{Cr}$, and $\text{ZnGa}_{1.97}\text{O}_4:\text{Cr}_{0.01}/\text{Bi}_{0.02}$. The Bi^{3+} , Cr^{3+} co-doped ZnGa_2O_4 PLNPs have a broad absorption spectra from UV to visible light (670 nm), and the UV-Vis light absorption is remarkably enhanced, compared to ZnGa_2O_4 and $\text{ZnGa}_2\text{O}_4:\text{Cr}$. The broad absorption band of the Bi^{3+} , Cr^{3+} co-doped ZnGa_2O_4 PLNPs at 450 nm is associated with the charge transfer band from $\text{Bi}^{3+} 6s^2$ to $\text{Bi}^{5+} 6s^0$.^{25,26} We carried out the XPS experiments to verify the existence of Bi^{5+} . The Bi 4f XPS spectrum shows two distinct peaks which were fitted into three peaks at 158.87 eV, 160.23 eV and 163.92 eV, respectively (Fig. S2 (ESI[†])). The peaks of Bi 4f_{7/2} at 158.87 eV and Bi 4f_{5/2} at 163.92 eV are corresponding to Bi(III). The peak of Bi 4f_{7/2} at 160.23 eV is ascribed to Bi(V), which indicates existence of the Bi^{5+} in the $\text{ZnGa}_{1.97}\text{O}_4:\text{Cr}_{0.01}/\text{Bi}_{0.02}$.^{27,28} Band gaps (E_g) of ZnGa_2O_4 , $\text{ZnGa}_2\text{O}_4:\text{Cr}$, and $\text{ZnGa}_{1.97}\text{O}_4:\text{Cr}_{0.01}/\text{Bi}_{0.02}$ were calculated to be approximately 4.38 eV, 4.45 eV, and 2.75 eV, respectively (Fig. S3 (ESI[†])), according to the following eqn (2):²⁹



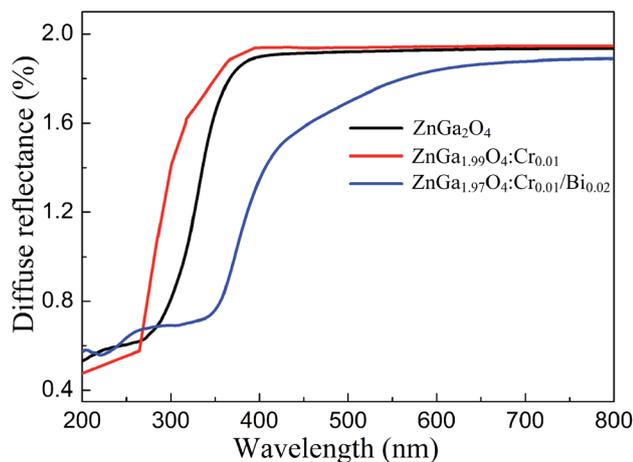


Fig. 5 UV-Vis diffuse reflectance spectra of ZnGa_2O_4 , $\text{ZnGa}_2\text{O}_4:\text{Cr}$ and $\text{ZnGa}_{1.97}\text{O}_4:\text{Cr}_{0.01}/\text{Bi}_{0.02}$.

$$\alpha h\nu = A(h\nu - E_g)^2 \quad (2)$$

where α is the absorption coefficient, $h\nu$ is the photon energy, and A is a constant. The experimental results shows that band gap of the ZnGa_2O_4 was slightly increased by doping Cr^{3+} , probably due to the self-reduction effect on Cr^{3+} ions reduce into lower valence states and the absence of localized states in the band gap.^{30,31} In contrast, the additional doping of $\text{ZnGa}_2\text{O}_4:\text{Cr}$ with Bi^{3+} can significantly decrease the band gap of ZnGa_2O_4 , owing to the electronic structure of Bi with a unique 6s orbital,^{27,32} which has the benefit of increasing solar light absorption in applications of photocatalysis degradation.

Rhodamine B (RhB) was used as the probe molecule to evaluate the photocatalytic performance of the Bi^{3+} , Cr^{3+} co-doped ZnGa_2O_4 PLNPs. The photo-degradations efficiency of ZnGa_2O_4 , $\text{ZnGa}_2\text{O}_4:\text{Cr}$, and $\text{ZnGa}_2\text{O}_4:\text{Cr}/\text{Bi}$ for RhB are shown in Fig. 6. The as-prepared samples were analysed first, to establish an adsorption-desorption equilibrium between the catalyst and the dye molecules. The suspension was stirred in darkness for 40 min, and the samples were then placed under UV light irradiation. The $\text{ZnGa}_{1.97}\text{O}_4:\text{Cr}_{0.01}/\text{Bi}_{0.02}$ PLNPs exhibited an excellent photocatalytic activity with a degradation efficiency of 99.2% after 100 min irradiation with UV light. Experimental results indicate that the photocatalytic activity of the pure ZnGa_2O_4 is lower than that of the Bi^{3+} , Cr^{3+} co-doped ZnGa_2O_4 and higher than that of the single Cr^{3+} doped ZnGa_2O_4 (Fig. 6). The photocatalytic activity of ZnGa_2O_4 was suppressed by the doping with Cr^{3+} , because the Cr^{3+} ions in ZnGa_2O_4 act as recombination centres, which can reduce the number and lifetime of the electron-hole pairs.³³ By contrast, additional doping of Bi^{3+} in $\text{ZnGa}_2\text{O}_4:\text{Cr}$ not only enhanced the UV-Vis light absorption of ZnGa_2O_4 , but also produced more hole trap energy levels, which resulted in more photo-generated electron-holes being captured by traps, thus inhibiting the recombination efficiency of the photo-generated electron-holes.²⁴ As a consequence, the lifetime of the photo-generated electron-holes and the afterglow time were prolonged.

To investigated the role of reactive oxygen species (ROS) in the degradation process, we carried out the ROS scavenging

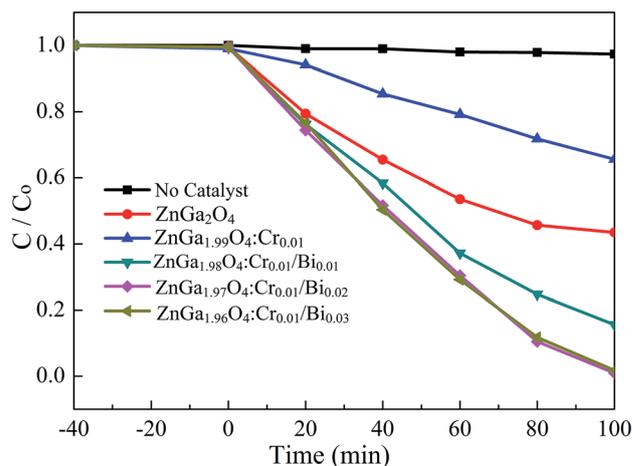
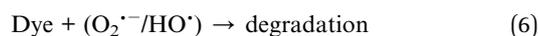
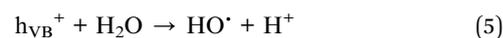
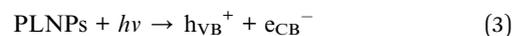


Fig. 6 The photo-degradation efficiency of ZnGa_2O_4 , $\text{ZnGa}_2\text{O}_4:\text{Cr}$ and $\text{ZnGa}_2\text{O}_4:\text{Cr}/\text{Bi}$ for RhB.

experiments with different scavengers, such as *p*-benzoquinone (BQ) as $\text{O}_2^{\cdot-}$ scavenger, isopropyl alcohol (IPA) as HO^{\cdot} scavenger, and ammonium oxalate (AO) as the hole scavenger.³⁴ The experimental results shows that the photo-degradation of RhB is dramatically decreased by adding AO in the RhB solution which indicates that the holes are the most dominant species in the photocatalytic degradation process (Fig. S4 (ESI⁺)). Compared to AO and BQ, the addition of IPA result in slight decrease for degradation of RhB, which indicates the HO^{\cdot} radicals are not main contributor for degradation process. The electrons and holes are generated on the surface of PLNPs under light UV radiation. The photogenerated electron-hole pairs reacted with oxygen and water to generate peroxide ($\text{O}_2^{\cdot-}$) and hydroxyl radicals (HO^{\cdot}).³⁵ The resulting radicals cause degradation of the organic molecules. This photocatalytic reaction can be represented as follows:



4. Conclusions

The Bi^{3+} , Cr^{3+} co-doped ZnGa_2O_4 PLNPs were synthesized by an ethylene glycol assisted hydrothermal method. The particle size of the as-prepared PLNPs was less than 10 nm. The photo-degradation efficiency of Bi^{3+} , Cr^{3+} co-doped ZnGa_2O_4 PLNPs for RhB was 99.2% after 100 min irradiation with UV light. Compared to $\text{ZnGa}_2\text{O}_4:\text{Cr}$ and un-doped ZnGa_2O_4 , the additional doping of Bi^{3+} in $\text{ZnGa}_2\text{O}_4:\text{Cr}$ not only significantly enhanced the NIR persistence luminescence of the PLNPs, but also remarkably improved the photocatalytic degradation efficiency of RhB. Addition of Bi^{3+} in $\text{ZnGa}_2\text{O}_4:\text{Cr}$ may produce



more trap energy levels, which prolongs the lifetime of the photo-generated electron–holes and the afterglow time. The novel dual-functional PLNPs are promising for *in vivo* bio-imaging and photocatalytic applications.

Conflicts of interest

The authors declare no conflict of interest.

Acknowledgements

This work has been supported by the National Natural Science Foundation of China (Grant 21565017) and the Scientific Research Program of the Higher Education Institution of Xinjiang (XJEDU2018I017, XJEDU2016S073).

Notes and references

- V. D. E. Koen, P. F. Smet and D. Poelman, *Materials*, 2010, **3**, 2536–2566.
- Z. Pan, Y. Y. Lu and F. Liu, *Nat. Mater.*, 2011, **11**, 58–63.
- L. Yang, M. Gecevicius and J. R. Qiu, *Chem. Soc. Rev.*, 2016, **45**, 2090–2136.
- V. D. E. Koen, P. F. Smet and D. Poelman, *Materials*, 2013, **6**, 2789–2818.
- D. R. Li, Y. H. Wang, K. Xu and H. Zhao, *RSC Adv.*, 2015, **46**, 20972–20975.
- S. K. Sun, H. F. Wang and X. P. Yan, *Acc. Chem. Res.*, 2018, **51**, 1131–1143.
- L. Thomas, T. Elliott, R. G. Gonzalo, M. Thomas, V. Bruno, S. Johanne, M. Nathalie, S. Daniel and R. Cyrille, *Theranostics*, 2016, **6**, 2488–2524.
- L. M. C. Quentin, C. Chanéac, J. Seguin, F. Pellé, S. Matrejean, J. P. J. D. Gourier, M. Bessodes and D. Scherman, *Proc. Natl. Acad. Sci. U.S.A.*, 2007, **104**, 9266–9271.
- A. Abdukayum, J. T. Chen, Q. Zhao and X. P. Yan, *J. Am. Chem. Soc.*, 2013, **135**, 14125–14133.
- T. Maldiney, A. Bessière, J. Seguin, E. Teston, S. K. Sharma, B. Viana, A. J. Bos, P. Dorenbos, M. Bessodes and D. Gourier, *Nat. Mater.*, 2014, **13**, 418–426.
- S. Yan, J. Wang, H. Gao, N. Wang, H. Yu, Z. Li, Y. Zhou and Z. Zou, *Adv. Funct. Mater.*, 2013, **23**, 1839–1845.
- Y. E. Lee, D. P. Norton, C. Park and C. M. Rouleau, *J. Appl. Phys.*, 2001, **89**, 1653–1656.
- Z. Gu, F. Liu, X. Li, J. Howe, J. Xu, Y. Zhao and Z. Pan, *J. Phys. Chem. Lett.*, 2010, **1**, 354–357.
- S. Yan, S. Ouyang, J. Gao, M. Yang, J. Y. Feng, X. X. Fan, L. J. Wan, Z. S. Li, J. H. Ye, Y. Zhou and Z. G. Zou, *Angew. Chem., Int. Ed.*, 2010, **49**, 6400–6404.
- L. Zou, X. Xiang, M. Wei, F. Li and D. G. Evans, *Inorg. Chem.*, 2008, **47**, 1361–1369.
- A. Bessiere, S. Jacquart, K. Priolkar, A. Lecointre, B. Viana and D. Gourier, *Opt. Express*, 2011, **19**, 10131–10137.
- J. Liu, T. Lecuyer, J. Seguin, N. Mignet, D. Scherman, B. Viana and C. Richard, *Adv. Drug Delivery Rev.*, 2019, **138**, 193–210.
- A. Abdukayum, C. X. Yang, Q. Zhao, J. T. Chen, L. X. Dong and X. P. Yan, *Anal. Chem.*, 2014, **86**, 4096–4101.
- T. Maldiney, A. I. Lecointre, B. Viana, A. I. Bessière, M. Bessodes, D. Gourier, C. Richard and D. Scherman, *J. Am. Chem. Soc.*, 2011, **133**, 11810–11815.
- A. Abdukayum, A. Tuerdi, R. Abdurahman, M. Tursun and N. Nurmat, *J. Inorg. Mater.*, 2016, **31**, 1363–1369.
- Y. Li, Y. Li, R. Chen, K. Sharafudeen, S. Zhou, M. Gecevicius, H. Wang, G. Dong, Y. Wu, X. Qin and J. Qiu, *NPG Asia Mater.*, 2015, **7**, e180.
- Y. Zhuang, J. Ueda and S. Tanabe, *Appl. Phys. Express*, 2013, **6**, 052602.
- W. B. Dai, Y. F. Lei, S. Ye, E. H. Song, Z. Chen and Q. Y. Zhang, *J. Mater. Chem. B*, 2016, **4**, 1842–1852.
- L. Li, K. Xu, Y. Wang, Z. Hu and H. Zhao, *Opt. Mater. Express*, 2016, **6**, 1022–1030.
- H. Mizoguchi, H. Kawazoe, H. Hosono and S. Fujitsu, *Solid State Commun.*, 1997, **104**, 705–708.
- Y. X. Zhuang, J. Ueda and S. Tanabe, *Opt. Mater. Express*, 2012, **2**, 1378–1383.
- A. Y. Teterin, Y. A. Teterin, K. I. Maslakov, V. G. Yarzhemskii, S. E. Sverchkov, B. I. Denker, B. I. Galagan, L. D. Iskhakova, L. I. Bulatov, V. V. Dvoirin, V. M. Mashinskii, A. A. Umnikov, A. N. Gur'yanov, V. I. Nefedov and E. M. Dianov, *Dokl. Phys.*, 2008, **53**, 566–570.
- S. Bagwasi, Y. Niu, M. Nasir, B. Tian and J. Zhang, *Appl. Surf. Sci.*, 2013, **264**, 139–147.
- Y. B. Cui and J. F. Luan, *J. Environ. Sci.*, 2015, **29**, 51–61.
- S. K. Kushwaha, K. K. Maurya, D. Haranath and G. Bhagavannarayana, *J. Appl. Crystallogr.*, 2011, **44**, 1054–1061.
- Y. Zhuang, J. Ueda, S. Tanabe and P. Dorenbos, *J. Mater. Chem. C*, 2014, **2**, 5502–5509.
- X. C. Meng and Z. S. Zhang, *J. Mol. Catal. A: Chem.*, 2016, **423**, 533–549.
- L. Li, Y. H. Wang, H. Li, H. J. Huang and H. Zhao, *RSC Adv.*, 2015, **5**, 57193–57200.
- B. M. Pirzada, Pushpendra, R. K. Kunchala and B. S. Naidu, *ACS Omega*, 2019, **4**, 2618–2629.
- K. Mageshwari, R. Sathyamoorthy and J. Park, *Powder Technol.*, 2015, **278**, 150–156.

