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Highly efficient and flexible photodetector based on MoS₂–ZnO heterostructures†

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Two-dimensional (2D) transition metal dichalcogenides (TMDs) such as molybdenum disulfide (MoS₂) and tungsten diselenide (WSe₂), have recently attracted attention for their applicability as building blocks for fabricating advanced functional materials. In this study, a high quality hybrid material based on 2D TMD nanosheets and ZnO nanopatches was demonstrated. An organic promoter layer was employed for the large-scale growth of the TMD sheet, and atomic layer deposition (ALD) was utilized for the growth of ZnO nanopatches. Photodetectors based on 2D TMD nanosheets and ZnO nanopatches were successfully fabricated and investigated, which showed a high photoresponsivity of 2.7 A/W. Our novel approach is a promising and effective method for the fabrication of photodetectors with a new structure for application in TMD-based transparent and flexible optoelectronic devices.

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Introduction

Interest in two-dimensional (2D) transition-metal dichalcogenides (TMDs) (MS₂, M = Ti, W, Mo, *etc.*) has been increasing recently thanks to their unique structural, physical, and chemical properties.^{1,2} TMD semiconductors, such as molybdenum disulfide (MoS₂), show great potential because their band-gap can be tuned by adjusting the layer thickness. Therefore, MoS₂-based optoelectronic devices have been studied extensively owing to their unique properties.^{3–6} In order to manipulate their opto-electrical properties, many approaches for the functionalization of MoS₂ nanosheets have been reported, using metallic nanoparticles (NPs),^{7–9} and aromatic molecules.^{10,11} Moreover, studies have also been reported on the application of high-performance photodetectors using photo-reactive metal nanoislands coated on sensing materials.^{12,13} However, the synthesis method of large-area-based high quality MoS₂ nanosheets and their reliable functionalization techniques for enhancing the electrical and chemical properties are still required for the application in MoS₂-based nano-electronics. In this study, we developed a novel method for the functionalization of a large-scale MoS₂ layer, by growing ZnO on the MoS₂ surface *via* an atomic layer deposition (ALD) process. Here, (NH₄)₂MoS₄ solution as the growth source of MoS₂ nanosheets, was spin-coated on a solid substrate such as

SiO₂ or quartz film. Subsequently, thermal-annealing for the MoS₂ growth was performed by a similar method to our previous work.¹⁴ ZnO for the functionalization of MoS₂ nanosheets were then grown by diethylzinc (DEZ) precursors.^{15,16} For the successful demonstration of large-scale MoS₂–ZnO based photodetectors, the electrical and optical characteristics of the resulting hybrid photodetectors were examined. The synthesis method is an efficient approach to fabricate TMD-based hybrid nanosheets for future photoelectronic applications, including flexible photodetectors and multi-functional sensors.

Experimental

The MoS₂ film was synthesized through a hydrothermal route with (NH₄)₂MoS₄ as a single source precursor. Following the typical procedure, 1.25 wt% (NH₄)₂MoS₄ was dispersed in ethylene glycol. The dispersed solution was stirred with a magnetic stirrer for 24 hours. The mixed solution was then spin-coated onto the cleaned SiO₂ substrate and heated at 100 °C for 2 min to evaporate the solvent. As-coated (NH₄)₂MoS₄ layer was placed at the center of the CVD chamber and pre-annealed at 280 °C under an inert 100 sccm argon gas flow. In this process, MoO₃ film was formed *via* thermal decomposition of a (NH₄)₂MoS₄ layer and annealed at 450 °C in order to obtain a large-scale production of high quality MoS₂ nanosheets sequentially. After synthesis of MoS₂ nanosheets, ZnO was physically deposited onto the MoS₂ nanosheets. All pulse and purge steps were carried out using air-operated valves and manipulated by a computer. The substrate temperature was set to room temperature and the working pressure was 3 Torr for all optimized conditions. DEZ and H₂O were employed as the Zn

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precursor and oxidant gas, respectively. DEZ and H₂O were kept at room temperature and delivered to the reactor. Herein, one ZnO growth cycle includes the following procedure steps: DEZ pulse (0.2 s), purge (10 s), H₂O pulse (5 s), and purge (10 s). These process steps are carried out at 10, 20, 30, and 40 cycles.

Results and discussion

For the synthesis of the large-area MoS₂ nanosheets, a thermal decomposition process was utilized using a solution-processed (NH₄)₂MoS₄ thin films (Fig. 1). First, 1.25 wt% (NH₄)₂MoS₄, as a single source precursor in ethylene glycol, was spin-coated onto a SiO₂ substrate. The ethylene glycol was then removed immediately by heating the sample to 100 °C for 2 min (Fig. 1(a)). The (NH₄)₂MoS₄ layer was annealed at 280 °C under argon carrier gas flowing at 100 sccm under a 1.8 Torr pressure for 30 min to form MoO₃ *via* thermal decomposition. Subsequently, the MoO₃ film with amorphous phase was annealed at 450 °C to achieve high quality and continuous MoS₂ nanosheets. Here, the additional sulfurization process was not carried out to form the MoS₂ nanosheets as previously reported work¹² (Fig. 1(b)). After the MoS₂ growth, ZnO nanopatches were formed onto MoS₂ nanosheets by introducing DEZ. A purge process was conducted with liquid nitrogen gas flowing, and the covering area of ZnO nanopatches was controlled by adjusting the process cycles. Finally, Cr (5 nm)/Au (60 nm) electrodes were deposited onto the MoS₂-ZnO hybrid nanosheets for the fabrication of MoS₂-based electrical devices. It should be noted that this method allows the synthesis of large-scale functionalized MoS₂ nanosheets by combing the large-scale MoS₂ synthetic method with the ALD technique for nanoscale ZnO patches.

In order to observe the surface morphology of MoS₂ nanosheets with ZnO nanopatches, scanning electron microscopy (SEM) was employed with regard to the process cycles of ZnO (Fig. 2(a)). MoS₂ nanosheets produced by solution-based synthesis exhibited a uniform surface coverage and a continuous film on the SiO₂ substrate. The surface morphologies of MoS₂ nanosheets with ZnO were then investigated depending on the process cycles. At first, the partially aggregated grains

and rough surface were observed, indicating that clearly displays 10-cycle ZnO nanopatches on MoS₂ nanosheets over the entire area (Fig. 2(a)(ii)). Formation of granular surface is common for epitaxial ZnO films produced by the ALD process.^{17,18} When increasing the ZnO film thickness by the process cycles, ZnO nanopatches with island-shape disappear, and flat thin films were formed in the MoS₂ nanosheets (Fig. S1†). Moreover, atomic force microscopy (AFM) analysis was carried out to confirm the surface roughness of MoS₂ nanosheets with ZnO nanopatches (Fig. 2(b)). These results were similar to surface morphologies identified by SEM analysis. Interestingly, surface roughness of MoS₂ nanosheets with ZnO nanopatches at 10 cycles was the highest (~0.57 nm) due to the presence of numerous ZnO grains at the initial deposition stage (Fig. 2(c)). When the process cycles for ZnO growth were increased, the RMS roughness decreased. This phenomenon could be understood by the formation of ZnO film at the increased process cycles. In order to provide the exact height profiles of as-deposited ZnO, XRR (X-ray specular reflectivity) was employed to confirm the thickness of thin film more accurately (Fig. S2†). In the case of 10 cycles-ZnO, the thickness cannot be measured because the thin film was not formed. This result indirectly predicts that 10 cycles-ZnO is formed with nanopatches. Furthermore, we could identify that the thickness becomes increases gradually from several nm to ~10 nm with increasing the number of cycles and then the ZnO thin film is formed eventually. Fig. 2(d) is a photograph of the large area-based MoS₂ nanosheets with ZnO on an 8 inch SiO₂ substrate. Notably, the hybrid film fabricated using this approach was uniform over the whole area. Furthermore, MoS₂ nanosheets with ZnO were successfully transferred onto the quartz substrate (2 cm × 2 cm) in order to confirm the applicability to the transparent electronic devices (Fig. 2(e)). Optical transmittance characterization was carried out using UV-vis spectroscopy (Fig. 2(f)). Herein, there was a difference in the optical transmittance with and without MoS₂. In general, the absorption spectra of ZnO nanoparticles depend on the method of fabrication, shape, and particle size. In this study, the optical transmittance spectra based on ZnO-MoS₂ hybrid films showed absorption peaks at 335 nm and 345 nm, and the two absorption peaks gradually increased as the number of ZnO cycles increased. The decrease in transmittance was related to an increase in the thickness of the films and it reveals a relatively low thickness of the ZnO films. Since the relatively thin ZnO was formed using the ALD process and act as a nanopatch, it only contributes to the healing of numerous defect sites, such as imperfections and impurities, in as-synthesized MoS₂ film.

In order to clarify the characteristics of chemical bonding in the MoS₂ nanosheets with ZnO nanopatch, Raman spectroscopy is employed. There are two representative Raman modes in MoS₂.¹⁹ The A_{1g} mode is originated from the out-of-plane vibrations of molybdenum atoms, and the E_{2g}¹ mode is responsible for the in-plane vibrations of molybdenum and sulfur atoms. Remarkably, the Raman shift was almost constant regardless of the process cycle of ZnO film (Fig. 3(a)). Furthermore, the frequency difference of A_{1g} mode and E_{2g}¹ mode was approximately 21.6 cm⁻¹ for the entire process condition

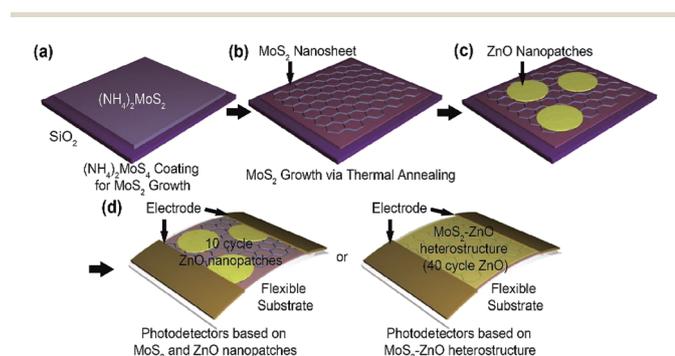


Fig. 1 Schematic diagram of the MoS₂ nanosheets with ZnO nanopatches synthesis. (a) Spin-coating of (NH₄)₂MoS₄ as a precursor for MoS₂ growth on a SiO₂ substrate. (b) MoS₂ growth *via* thermal annealing. (c) Growth of ZnO nanopatches on top of MoS₂ nanosheets by ALD process. (d) Fabrication of photodetectors based on MoS₂-ZnO nanopatches or heterostructure.



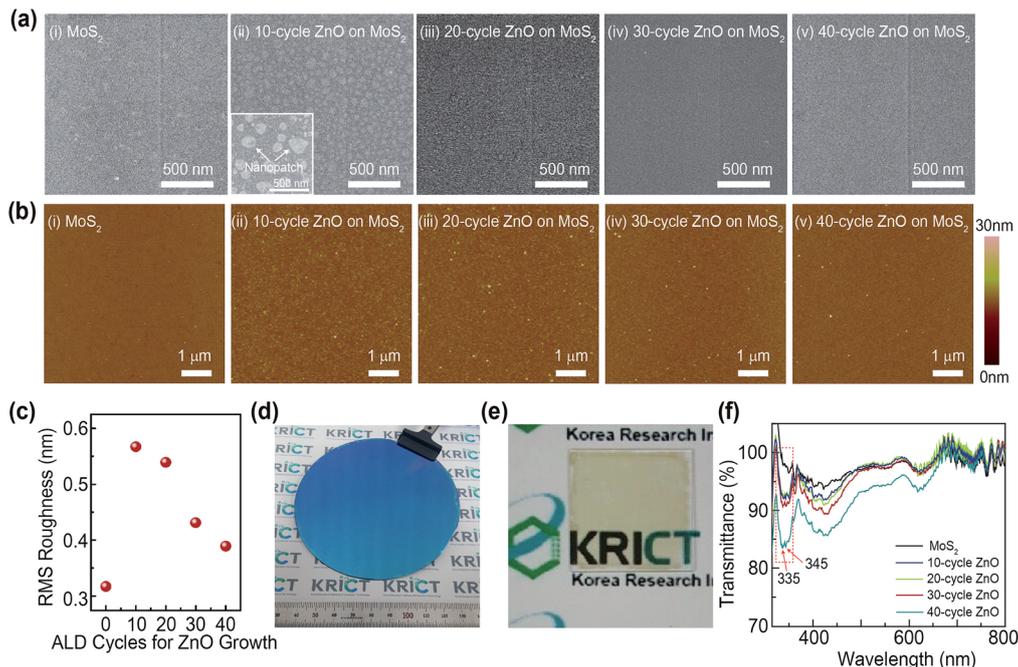


Fig. 2 (a) Scanning electron microscopy (SEM) and (b) atomic force microscopy (AFM) images, and (c) RMS roughness of MoS₂ nanosheets with ZnO nanopatches as a function of the process cycles (0, 10, 20, 30, and 40 cycles) during ALD process. Picture of large-area MoS₂ and ZnO hybrid film on the 8 inch SiO₂ substrate (d) and the transferred MoS₂ nanosheets with ZnO nanopatches (40 cycles) (e). (f) Optical transmittance of ZnO–MoS₂ hybrid film on a PET (polyethylene terephthalate) substrate for diverse ALD cycles (0, 10, 20, 30, and 40 cycles).

(Fig. 3(b)). These results indicate that the structure of as-synthesized MoS₂ nanosheets were well-maintained without any structural deformation during the deposition process of ZnO film. X-ray photoelectron spectroscopy (XPS) was also carried out to examine the chemical composition of the MoS₂ nanosheets with ZnO nanopatch (Fig. 3(c)). Herein, the Mo 3d, S 2p core level spectra for the MoS₂ nanosheets and the Zn 2p, O 1s core level spectra related to the deposited ZnO were explored for the structural analysis of the MoS₂–ZnO hybrid film. For the pristine MoS₂ nanosheets, the Mo 3d_{3/2} and 3d_{5/2} peaks were located at binding energy (E_B) of 232.5 eV and 229.3 eV, respectively. S 2p_{1/2} and 2p_{3/2} peaks at E_B of 163.3 eV and 162 eV, respectively, were obtained in pristine MoS₂ nanosheets, revealing that MoS₂ was synthesized successfully using the solution process.¹⁴ Furthermore, the intensity of Mo 3d and S 2p core level spectra decreased as the process cycle of ZnO increased. In the Zn 2p core level spectra, Zn 2p_{1/2} at the higher binding energy and Zn 2p_{3/2} at the lower binding energy was observed. The difference in binding energy between the Zn 2p_{3/2} and Zn 2p_{1/2} was about 23 eV. This value, originated from the spin-orbit splitting of energy level, is consistent with previous reports that demonstrated the existence of ZnO.²⁰ Moreover, the intensity of Zn 2p core level spectra increased as the deposition cycle of ZnO film increased. Typically, XPS is known as a powerful surface analysis tool with surface-sensitive technique. As increasing the number of ZnO cycles, the coverage region of ZnO on the MoS₂ surface is increased. So, when the incident X-ray beam is injected on the MoS₂–ZnO hybrid structures, the peak intensity and composition ratio could be

altered. In other words, the Mo 3d and S 2p peaks become decreased as increasing the number of ZnO cycles. Whereas, the Zn 2p peaks related to ZnO become increased. Atomic concentration of MoS₂–ZnO hybrid structure as a function of the number of ZnO cycles was represented in the Fig. S3.† In the O 1s core level spectra, O 1s core level spectrum was located at E_B of 529.9 eV, indicating the presence of oxygen atoms in the crystal lattice without any oxygen vacancies. With increasing process cycles of ZnO film, the prominent extra peaks at E_B of 531.6 eV increased. This phenomenon indicates the formation of oxygen atoms in the ZnO lattice with oxygen vacancies.

Fig. 4(a–i) is an optical image of an electronic device based on a MoS₂ nanosheet with a ZnO nanopatch. Here, the width and length of the channel were 40 μm, 200 μm, respectively. In order to clarify the electrical properties, output curves were examined depending on the number of ZnO cycles. Thereby, we could confirm the improved electrical properties with increasing the number of ZnO (Fig. S4†). This behavior could be understood by the fact that as-deposited ZnO contribute to heal the numerous defect sites in MoS₂. Furthermore, transfer characteristics with back gate configuration were investigated for MoS₂ nanosheets with a ZnO nanopatch (process cycle: 10, 20, 30, 40 cycles) under cyclic UV light irradiation ($\lambda = 254$ nm) (Fig. 4(a)(ii–vi)). Pristine MoS₂ shows p-type characteristics due to the adsorbed oxygen atoms on the thin MoS₂ surface in an arbitrary environment. With increasing the process cycles of ZnO films, a lot of photo-excited electrons were transferred to the ZnO film with a higher work-function than MoS₂.²¹ Thus, the threshold voltage of MoS₂ nanosheets was shifted to a more



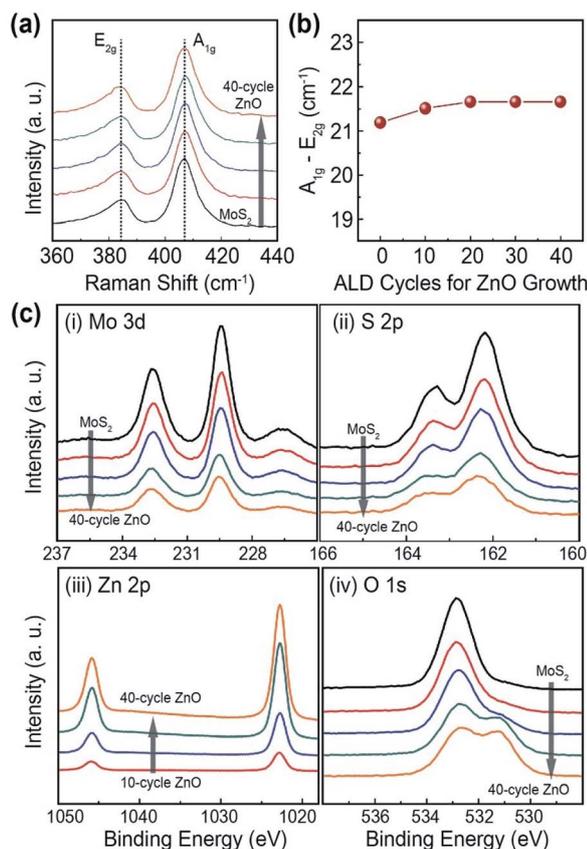


Fig. 3 (a) Raman spectra and (b) frequency difference between A_{1g} mode and E_{2g} mode of MoS_2 nanosheets with ZnO nanopatches (process cycle: 10, 20, 30, and 40 cycles). (c) X-ray photoelectron spectroscopy (XPS) of MoS_2 nanosheets with ZnO nanopatches. (i) Mo 3d, (ii) S 2p, (iii) Zn 2p, and (iv) O 1s core level spectra of MoS_2 nanosheets with ZnO nanopatches.

positive voltage, and their photocurrent significantly improved due to the high sensitivity of the ZnO film under UV exposure. The dependence of the photocurrent response in the UV region on the process cycle of ZnO at bias voltage of 10 V (Fig. 4(b)) was investigated. The photocurrent increased considerably with increasing process cycles. MoS_2 nanosheets with 40 cycle-ZnO experienced the highest photocurrent owing to the unique properties of ZnO.²² The rise and decay time for photodetectors based on MoS_2 -ZnO heterostructure was calculated in Table S1.† With increasing the number of ZnO cycles, the deep level defect states (DLDS) originated from the sulfur vacancies in MoS_2 crystal become disappeared and the recombination rate of photo-excited electron-hole pairs was decreased also. Our photodetectors based on MoS_2 -ZnO hybrid structures could improve the photoresponsivity as well as heal the several defect sites in MoS_2 crystals by introducing ZnO. The relevant working principle of photodetector based on MoS_2 -ZnO hybrid film was depicted in Fig. S5.† The photocurrent response of MoS_2 nanosheets with 40 cycle-ZnO was identified depending on the bias voltage (Fig. 4(c)). At increasing bias voltage, the photocurrent response of the MoS_2 -ZnO hybrid film increased dramatically and the highest photoresponsivity was estimated to 2.7 A/W at a bias voltage of 40 V. Table S2 in ESI† exhibits the parameters of MoS_2 based ultraviolet photodetectors, indicating that performance for our photodetectors based on ZnO- MoS_2 hybrid film is more competitive than previous reports. Furthermore, the variation of photocurrent was evaluated to confirm the applicability to flexible optoelectronic devices based on MoS_2 nanosheets with ZnO nanopatch (Fig. 4(d)). PMMA-assisted wet transfer method was employed to transfer as-fabricated film to the flexible PET substrates. Fig. 4(d) shows the bending process of the MoS_2 -ZnO hybrid film. The variations of the photocurrent response of pristine MoS_2 and MoS_2 -ZnO film (bending radius = 3 mm) as a function of the number

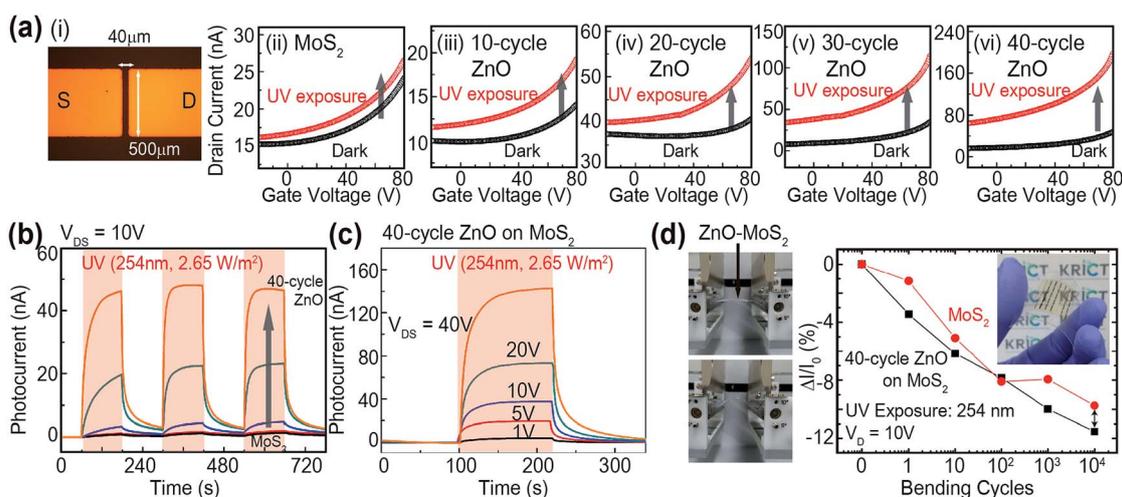


Fig. 4 (a-i) Optical image of an electronic device based on MoS_2 nanosheets with ZnO nanopatches. (a-ii)–(a-vi) Transfer curve of MoS_2 nanosheets with ZnO nanopatches. (b) Time-dependent photocurrent of MoS_2 nanosheets with ZnO nanopatches (process cycle: 10, 20, 30, and 40 cycle) under cyclic UV light irradiation at a bias voltage (V_{DS}) = 10 V. (c) Time-dependent photocurrent of MoS_2 nanosheets with ZnO nanopatches as a function of the bias voltage. (d) Picture of bending test using ZnO- MoS_2 hybrid film on a PET substrate and variation of photocurrent as a function of bending cycles (bending radius = 3 mm).



of bending cycles (up to 10 000 cycles) were examined. Remarkably, pristine MoS₂ and MoS₂-ZnO film do not exhibit dramatic changes upon increase of bending cycles. This reveals that the deposited ZnO film is relatively thin. When only ZnO exists, the variation of electrical properties were less than 3% after the bending process (Fig. S6†). Therefore, it does not influence the large structural deformation under the mechanical stress. We believe that the TMDs-based hybrid film in this study will be a promising candidate for applications requiring high efficient and flexible opto-electrical properties.

Conclusions

We demonstrated the fabrication of flexible photodetectors based on MoS₂ nanosheets and ZnO nanopatches. ZnO with high efficient photocurrent heals the numerous defect sites, such as imperfections and impurities. Therefore, the carrier transport properties of hybrid films could be improved by deposition of ZnO films, and the photocurrent can be significantly increased by reducing the photo-excited electron-hole recombination rate. The MoS₂-ZnO hybrid film for the bending test was transferred onto the flexible PET substrate using a PMMA-assisted wet transfer method. It was found that the degradation of the photocurrent induced by the bending process was not changed dramatically due to the thin thickness of ZnO nanopatches. Our MoS₂-ZnO hybrid film based photodetectors are expected to provide a new approach for future applications of outstanding electronic devices with high efficiency and flexibility.

Conflicts of interest

There are no conflicts to declare.

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Notes and references

1 H. Wang, H. Yuan, S. S. Hong, Y. Li and Y. Cui, *Chem. Soc. Rev.*, 2015, **44**, 2664–2680.

- 2 R. Lv, J. A. Robinson, R. E. Schaak, D. Sun, Y. Sun, T. E. Mallouk and M. Terrones, *Acc. Chem. Res.*, 2015, **48**, 56–64.
- 3 J. Zhu, J. Wu, Y. Sun, J. Hunag, Y. Xia, H. Wang, H. Wang, Y. Wang, Q. Yi and G. Zou, *RSC Adv.*, 2016, **6**, 110604.
- 4 K. F. Mak, C. Lee, J. Hone, J. Shan and T. F. Heinz, *Phys. Rev. Lett.*, 2010, **105**, 136805.
- 5 H. J. Conley, B. Wang, J. I. Ziegler, R. F. Haglund, S. T. Pantelides and K. I. Bolotin, *Nano Lett.*, 2013, **13**, 3626–3630.
- 6 A. Molina-Sánchez, K. Hummer and L. Wirtz, *Surf. Sci. Rep.*, 2015, **70**, 554–586.
- 7 Z. Cheng, B. He and L. Zhou, *J. Mater. Chem. A*, 2015, **3**, 1042.
- 8 T. S. Sreepasad, P. Nguyen, N. Kim and V. Berry, *Nano Lett.*, 2013, **13**, 4434–4444.
- 9 Y. Shi, J.-K. Huang, L. Jin, Y.-T. Hsu, S. F. Yu, L. Li and H. Y. Yang, *Sci. Rep.*, 2013, **3**, 1839.
- 10 D. Jariwala, S. L. Howell, K.-S. Chen, J. Kang, V. K. Sangwan, S. A. Philippone, R. Turrisi, T. J. Marks, L. J. Lauhon and M. C. Hersam, *Nano Lett.*, 2016, **16**, 497–503.
- 11 S. Andleeb, A. K. Singh and J. Eom, *Sci. Technol. Adv. Mater.*, 2015, **16**, 035009.
- 12 M. A. Nazirzadeh, F. B. Atar, B. B. Turgut and A. K. Okyay, *Sci. Rep.*, 2014, **4**, 7103.
- 13 A. Sharma, R. Kumar, B. Bhattacharyya and S. Husale, *Sci. Rep.*, 2016, **6**, 22939.
- 14 Y. R. Lim, W. Song, J. K. Han, Y. B. Lee, S. J. Kim, S. Myung, S. S. Lee, K.-S. An, C.-J. Choi and J. Lim, *Adv. Mater.*, 2016, **28**, 5025–5030.
- 15 E. Guziewicz, M. Godlewski, T. Krajewski, Ł. Wachnicki, A. Szczepanik, K. Kopalko, A. Wójcik-Głodowska, E. Przeździecka, W. Paszkowicz, E. Łusakowska, P. Kruszewski, N. Huby, G. Tallarida and S. Ferrari, *J. Appl. Phys.*, 2009, **105**, 122413.
- 16 E. Janocha and C. Pettenkofer, *Appl. Surf. Sci.*, 2011, **257**, 10031–10035.
- 17 G. A. Velázquez-Nevárez, J. R. Vargas-García, J. Aguilar-Hernández, O. E. Vega-Becerra, F. Chen, Q. Shen and L. Zhang, *Mater. Res.*, 2016, **19**, 113–117.
- 18 Z.-N. Ng, K.-Y. Chan, C.-Y. Low, S. A. Kamaruddin and M. Z. Sahdan, *Ceram. Int.*, 2015, **41**, 254–258.
- 19 C. Lee, H. Yan, L. E. Brus, T. F. Heinz, J. Hone and S. Ryu, *ACS Nano*, 2010, **4**, 2695–2700.
- 20 A. G. Joshi, S. Sahai, N. Gandhi, Y. G. R. Krishna and D. Haranth, *Appl. Phys. Lett.*, 2010, **96**, 123102.
- 21 G. Nazir, M. F. Khan, I. Akhtar, K. Akbar, P. Gautam, H. Noh, Y. Seo, S.-H. Chun and J. Eom, *RSC Adv.*, 2017, **7**, 16890.
- 22 K. Liu, M. Sakurai and M. Aono, *Sensors*, 2010, **10**, 8604–8634.

