Enhanced photoluminescence of LEuH nanosheets: 2D photonic crystals self-assembled by core–shell SiO$_2$@LEuH spheres†

Xinying Wang, Pingping Feng, Baiyi Shao, Fangming Cui and Xiaojing Yang

The enhancement in photoluminescence (PL) is a challenge for layered rare-earth hydroxides, which usually have weak PL due to the quenching effect of hydroxyls. In this work, we provide a strategy to enhance the PL behavior by constructing two-dimensional (2D) photonic crystals. The core–shell structured SiO$_2$@LEuH spheres were prepared by attaching the positively charged layered europium hydroxide (LEuH) nanosheets onto the negatively charged surfaces of the SiO$_2$ spheres; the core–shell spheres further formed a monolayered 2D colloidal crystal with the hexagonal lattice on a quartz substrate through an evaporation-induced assembly process. The 2D colloidal crystals exhibited a significantly enhanced photoluminescence at 611 nm related to the $^5$D$_0$ → $^7$F$_2$ transition of Eu$^{3+}$ compared with the SiO$_2$@LEuH spheres and the LEuH nanosheets dispersed in deionized water. The emission band of Eu$^{3+}$ hardly changed in the three samples; therefore, the PL enhancement can be attributed to the emission band located at the short edge of the photonic band-gap of the 2D crystals.

Introduction

The synthesis, structure and exfoliation$^3$ of layered rare-earth hydroxide (LRH) nanosheets have been widely studied in the past decades due to their potential applications in absorption,$^4$ magnetism,$^5$ photoluminescence$^6$ (PL), etc. However, the PL intensity of LRHs is relatively weak compared with that of the rare-earth-doped ceramic materials due to the hydroxyl groups of LRHs. For the layered europium hydroxides (LEuHs), the $^5$D$_0$ → $^7$F$_2$ transition at 615 nm of Eu$^{3+}$ is largely derived from the $C_4$-site, while the $^5$D$_0$ → $^7$F$_4$ transition at 698 nm is associated with the $C_{4v}$-site. In general, the luminescence enhancement of Eu$^{3+}$ is mainly concentrated on the regulation of $^5$D$_0$ → $^7$F$_2.$ In order to improve the PL performance of LRHs, various organic anions have been intercalated into the interlayers of LRHs based on the sensitization enhancement effect.$^8$–$^{10}$ Other studies achieved the PL enhancement by converting LRHs into rare-earth oxides$^{11}$ via calcination or a topological reaction.$^{12}$ Byeon$^+^+$ et al. prepared multi-colored LRH films via alternately stacking the LEuHs and LTbH nanosheets.$^{13}$ Hu et al. fabricated LRH films at the hexane/water interface.$^{14}$ All these works achieved enhanced PL performance through annealing at high temperatures to remove the hydroxyl groups. These works have focused more on adjusting the structure of LRHs instead of modifying their surrounding environment.

It is noticed that the two-dimensional (2D) monolayers of a polystyrene sphere on the phosphor film surface can improve the PL efficiency of the phosphor films.$^{15}$ Colloidal crystals have attracted intensive research attention owing to their high reproducibility, low cost and wide applications in electronics and optics.$^{16}$ The relationship between the photonic band-gap of the colloidal crystals and the PL of the rare-earth materials has been researched due to the narrow PL emission lines of rare earth elements compared with that of the photonic band-gap of the colloidal crystals.$^{17}$ Inverse opal structural materials of Eu$^{3+}$-doped Gd$_2$O$_3$,$^{18}$ Lu$_2$O$_3$,$^{19}$ YVO$_4$ (ref. 19) and NaY(MoO$_4$)$_2$ (ref. 20) have been prepared and their PL performance results show that the PL emission can be significantly suppressed when the emission bands overlap with the photonic band gap, whereas the enhancement in the emission occurs when the emission bands appear at the long edge of the photonic band gap. Furthermore, these results also showed that the $^5$D$_0$ → $^7$F$_2$ radiative lifetimes of Eu$^{3+}$ in the inverse opal structure prolonged and the concentration quenching of Eu$^{3+}$ was suppressed. However, the influence of the photonic band-gap on opal structural materials is rarely reported; furthermore, the influence on the PL performance of LRHs has not been reported.

Herein, we attached LEuH nanosheets onto the surface of silica micro-spheres by phosphate modification, forming core–
shell-structured SiO2@LEuH micro-spheres, which were further self-assembled into a mono-layered 2D colloidal crystal with a hexagonal lattice on a quartz substrate through an evaporation-induced assembly process. The mono-layered 2D colloidal crystal showed improved PL performance due to the effects of the photonic band gap on the PL emission of LEuHs.

The motive to study such a material is that we consider that it may have three merits: first, it would provide a new method to “solidify” the nanosheets of LRHs; second, we can provide a strategy to enhance the PL intensity of LRHs without any further chemical treatments, such as annealing at high temperatures or topological reactions; third, this study would enrich the work on the influence of the photonic band gap on the optical structural materials.

Experimental

Materials

Eu2O3 (99.9%), KOH and SiO2 spheres (1 µm) were purchased from Shanghai Aladdin Biochem Technology Co., Ltd; TSPMP was purchased from HEOOWNS Biochem Technologies, LLC.

Preparation of LEuH nanosheets

The Eu(NO3)3·6H2O powder was first prepared as follows: 3 g Eu2O3 (99.9%) was added into aqueous HNO3 solution and then, the solution was heated in an oil bath at 80 °C for 3 h under stirring. After that, the Eu(NO3)3·6H2O powder was obtained by filtration and drying.

The LEuH nanosheets were prepared according to a reported method.21 Typically, 50 mL 0.10 M aqueous KOH solution was added dropwise to 50 mL 0.05 M aqueous Eu(NO3)3·6H2O solution under vigorous stirring at room temperature and kept for 12 h. The precipitate was collected by centrifugation and washed with deionized water several times. A slurry of the LEuH nanosheets was redispersed into deionized water by sonication to prepare a 0.2 g L⁻¹ aqueous LEuH suspension.

Preparation of core–shell structured SiO2@LEuH spheres

First, the SiO2 spheres were modified with phosphate through mixing and reaction of 0.025 g SiO2 spheres and 0.5 mL sodium(3-(trihydroxysilyl)propyl)methylphosphonate (TPMP) in 40 mL water and 10 mL ethanol mixture for 5 hours. Second, the phosphate-modified SiO2 spheres were collected by centrifugation and washing with deionized water several times. A slurry of the LEuH nanosheets was redispersed into deionized water by sonication to prepare a 0.2 g L⁻¹ aqueous LEuH suspension.

Preparation of 2D colloidal crystals of SiO2@LEuHs

The 2D colloidal crystals of SiO2@LEuHs were prepared by an evaporation-induced assembly process. First, 3 mL aqueous SiO2@LEuH solutions with different SiO2@LEuH contents, i.e., 0.0002, 0.002, 0.004, 0.006, 0.008, 0.01 and 0.012 wt% and 3 mL water–ethanol mixtures with different ethanol volume fractions for tuning the surface tension were prepared, as seen from the data in Table 1. Second, the clean quartz substrates were immersed vertically into the prepared solutions and the mixtures were kept at 40 °C for solvent evaporation. Finally, the substrates were taken out after the solvent evaporated and dried at an ambient temperature. The 2D arrays and 2D colloidal crystals of SiO2@LEuHs were then deposited on the substrates.

Table 1 Features of different solutions for the preparation of the 2D colloidal crystals of SiO2@LEuHs

<table>
<thead>
<tr>
<th>Solutions</th>
<th>V (%)</th>
<th>σ (N m⁻¹)</th>
</tr>
</thead>
<tbody>
<tr>
<td>Sol. 1</td>
<td>0</td>
<td>71.5</td>
</tr>
<tr>
<td>Sol. 2</td>
<td>20</td>
<td>41.8</td>
</tr>
<tr>
<td>Sol. 3</td>
<td>33</td>
<td>36.0</td>
</tr>
<tr>
<td>Sol. 4</td>
<td>50</td>
<td>30.7</td>
</tr>
<tr>
<td>Sol. 5</td>
<td>67</td>
<td>27.3</td>
</tr>
<tr>
<td>Sol. 6</td>
<td>80</td>
<td>25.5</td>
</tr>
</tbody>
</table>

Characterization

X-ray diffraction (XRD) measurements were carried out using a Philips X’Pert Pro MPD diffractometer with Cu-Kα radiation (λ = 0.1541 nm) at room temperature (40 kV, 40 mA, 10 s per step) and a step size of 0.01°. Scanning electron microscopy (SEM) images were obtained using a model S-4800 microscope (Hitachi, Ltd.) with an accelerating voltage of 5.0 kV. The surface tension of the solutions was measured by BCZ-600 with the resolution of 0.1 mN m⁻¹ at room temperature. A transmission spectrum was measured under the ‘Transmittance’ mode by a SHIMADZU UV-2600 spectrophotometer. The slit width was set as 2 nm; the sweeping was set under a low speed and the sampling interval was set as 1 nm s⁻¹. Excitation and emission spectra of the photoluminescence were measured by an FS5 fluorescence spectrophotometer with slit width of 5 nm and the incident light vertical to the substrates. The elemental mapping was carried out by energy dispersive X-ray spectroscopy under a TEM scanning model.

Results and discussion

Scheme 1 shows a schematic for the formation process of the 2D colloidal crystals of the core–shell-structured SiO2@LEuHs. First, the surfaces of SiO2 spheres were modified with negatively charged phosphate ions by TSPMP. Then, the LEuH nanosheets were absorbed and adhered on the surface of the modified SiO2 spheres through the electrostatic interaction between the positive LEuH nanosheets and the negative phosphate ions on the SiO2 sphere surfaces; this is certified by the results of the zeta-potentials of the samples in Fig. S1. Finally, the 2D colloidal crystals of the core–shell-structured SiO2@LEuHs were prepared and deposited on the quartz substrates via an evaporation-induced assembly process under the action of capillary forces.

The XRD patterns of the samples are shown in Fig. 1. The original LEuH nanosheets present two sharp diffraction peaks at 2θ = 9.65 and 19.46°, corresponding to the (002) and (004)
planes of the LEuH nanosheets and indicating the well-ordered hydroxide layers, as seen in curve (a) of Fig. 1.22,23 The plane distances are 9.1 and 4.54 Å, which are consistent with our previous results.24 The wide diffraction peak around 2θ/C14 in curve (b) of Fig. 1 can be attributed to the diffraction of amorphous silica of SiO2 spheres. The core–shell-structured SiO2@LEuHs show two weak diffraction peaks at 2θ/C14 = 28 and 30°, as seen in curve (c) of Fig. 1, which can be indexed as the (400) and (220) intra-layer planes of the LEuH nanosheets, respectively, and are consistent with the diffractions of the original LEuH nanosheets, as indicated by two vertical dot lines. Meanwhile, the core–shell-structured SiO2@LEuHs do not show the diffraction peaks of the (002) and (004) planes of the LEuH nanosheets again, which is due to the dramatic reduction in the number of layers of the LEuH nanosheets along the c-axis, indicating the monolayer or few-layer attachment of LEuH nanosheets onto the surface of SiO2 spheres. Furthermore, the thickness of the nanosheets after ultra-sonication was measured as 5.45 nm (Fig. S2†), indicating that the nanosheets comprise few layers.

The TEM images of the prepared LEuH nanosheets are shown in Fig. 2, and they present the layer structured nanosheets with 2D size at ~200 nm; europium homogeneously disperses in the nanosheet area, which indicates the good integrity of the LEuH nanosheets and homogeneous composition, as seen in Fig. 2(C). The aqueous LEuH solution presents an obvious Tyndall effect under red laser irradiation, indicating the good dispersion of the LEuH nanosheets in water, as seen in Fig. 2(D).

The microstructure and elemental mapping of the core–shell-structured SiO2@LEuH spheres were characterized by SEM and TEM, as shown in Fig. 3. The SiO2 spheres present a smooth surface and uniform diameter of ~1 μm, as seen in Fig. 3(A). The phosphate-modified SiO2 spheres pack more closely on the SEM sample support maybe due to the surface functional molecules and show the same diameter of ~1 μm as the unmodified SiO2 spheres, as seen in Fig. 3(B). The core–shell-structured SiO2@LEuH spheres present rough surfaces compared with the two former, as seen in Fig. 3(C), which are due to the attached LEuH nanosheets on the surface of the SiO2 spheres. Furthermore, the surface-attached LEuH nanosheets can be clearly seen from the TEM image of a single core–shell-
structured SiO$_2$@LEuH sphere in Fig. 3(D). The enlarged TEM image at an arbitrary edge and the surface of a single SiO$_2$@LEuH sphere, indicated by the red rectangle in Fig. 3(D), clearly show the laminar lattice fringes of the LEuH nanosheets with a spacing of 0.32 nm, which is consistent with the distance of the (400) intra-layer planes of the LEuH nanosheets characterized by above XRD, as seen in Fig. 3(E) and (F). The TEM elemental mapping results of a single SiO$_2$@LEuH sphere, shown in Fig. 3(G)-(I), give more evidence of the core–shell structure of the SiO$_2$@LEuH sphere. The homogeneous distribution of phosphorous and europium on/around the surface of the SiO$_2$ sphere indicates the complete modification of phosphate and the coating of the LEuH nanosheets on the surface of the SiO$_2$ sphere.

The 2D colloidal crystals of SiO$_2$@LEuHs were prepared by an evaporation-induced assembly process, and the preparation process was not as easy as that of smooth SiO$_2$ spheres because of the rough surface of the core–shell-structured SiO$_2$@LEuH spheres. Here, both the SiO$_2$@LEuH content in the water–ethanol solutions and the surface tension of the solutions were important. We first prepared the arrays of SiO$_2$@LEuHs on quartz substrates from a series of solutions with different SiO$_2$@LEuH contents and found out an optimal and appropriate SiO$_2$@LEuH content of 0.008 wt% for the periodic 2D colloidal crystal preparation, as seen in the SEM images in Fig. S3.† Second, the surface tension of the SiO$_2$@LEuH solution was tuned by changing the ethanol volume fraction of the solution, as shown in Table 1. The surface tension ($\sigma$) decreases as $V$ increases due to the smaller surface tension of ethanol than that of deionized water, and $\sigma$ falls sharply before 20%, which is consistent with previous results.  

Fig. 4 shows the SEM images of the 2D colloidal crystals of the core–shell-structured SiO$_2$@LEuHs prepared from aqueous solutions with different ethanol volume fractions at 0.008 wt% SiO$_2$@LEuH content. The 2D colloidal crystals of SiO$_2$@LEuHs prepared from Sol. 3 presented the best 2D periodic morphology with uniform monolayers and few vacancies compared with others, as seen in Fig. 4(C).

Whether the spheres arrange closely and uniformly depends on the infiltration degree of the liquid to the quartz surface and the matching degree of the surface energies between the liquid and the spheres. When the volume fraction of ethanol is 20%, $\sigma$ of the liquid drops rapidly and then slowly with continuous increments in the volume fraction. This could be because the liquid can completely infiltrate the quartz surface when the volume fraction is 20%. As the liquid is water, $\sigma$ of the liquid is equal to that of water, and the liquid cannot completely infiltrate the surface of quartz. At the same time, the surface energy of the liquid is higher than that of the spheres, and the above two factors both result in the presence of obvious local 3D packing of the spheres, as seen in Fig. 4(A). As the volume fraction of ethanol increases, the surface energy of the liquid gradually approaches that of the spheres, and the agglomeration phenomenon is weakened. When the volume fraction reaches 33%, the surface energy of the liquid is closest to that of the spheres, leading to monolayer close packing (Fig. 4(C)). On further increasing the volume fraction of ethanol, the surface energy of the liquid becomes lower than that of the spheres, leading to the presence of vacancies and agglomeration, as seen in Fig. 4(D)-(F). Meanwhile, the faster evaporation rate of the liquid leads to significant agglomeration (Fig. 4(F)) compared with the observations obtained from Fig. 4(D) and (E).

Generally, in photonic crystals, the change in the PL behavior of Eu$^{3+}$ could be explained by a variety of reasons. First, as the emission band lies in the range of the band gap, the change in the PL behavior is mainly ascribed to the photonic band-gap effects: (1) significant suppression of the emission will be detected if the photonic band-gap overlaps with the rare-earth emission bands;  

**Table 2** Summary about the effects of photonic band-gap on spontaneous emissions

<table>
<thead>
<tr>
<th>Position of emission bands</th>
<th>Inverse opal</th>
<th>Opal</th>
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<tbody>
<tr>
<td>Center</td>
<td>Suppression</td>
<td>Suppression</td>
</tr>
<tr>
<td>Long-wavelength edge</td>
<td>Enhancement$^a$</td>
<td>Suppression$^b$</td>
</tr>
<tr>
<td>Short-wavelength edge</td>
<td>Suppression$^a$</td>
<td>Enhancement$^a$</td>
</tr>
</tbody>
</table>

$^a$ The standing wave lies at the pole. $^b$ The standing wave lies at the equator.

...
gap, (1) the suppression of the emission is ascribed to the higher site symmetry of Eu\(^{3+}\) ions compared with that of the precursor\(^{17}\) and (2) the enhancement of the PL properties is ascribed to the higher effective refractive indices that minimize the surface scattering of both the excitation and emission lights after close packing of the core–shell structures.\(^{8}\)

In this experiment, we synthesized a periodic hexagonal close-packed single layer parallel to the laminate, and it was considered that a photonic band gap would be present in the direction parallel to the laminate rather than in the direction perpendicular to the laminate. Fig. 5(A) shows the transmission spectrum with light parallel to the direction of the laminate. This stop-band is centred at 660 nm.

Theoretically, the position of the photonic band-gap in photonic crystals can be estimated by the Bragg’s law of diffraction combined with the Snell’s law.\(^{7}\)

\[
\lambda_{\text{PBG}} = 2d_{\text{hkl}} \sqrt{n_{\text{eff}}^2 - \sin^2 \theta}
\]

(1)

\[
n_{\text{eff}} = n_{\text{sphere}} \phi + n_{\text{air}} (1 - \phi)
\]

(2)

Here, \(\lambda_{\text{PBG}}\) is the centre of the band-gap, \(d_{\text{hkl}}\) is the \(hkl\) plane distance, \(n_{\text{eff}}\) is the average refractive index (\(n_{\text{sphere}} = 1.364\)), \(\theta\) is the angle from the incident light to the normal of the substrate surface, and \(\phi\) is the volume fraction of the spheres (\(\phi = 0.74\)). In this discussion, the monolayer is considered as the (111) plane of the face-centred cubic (fcc) phase due to the higher site symmetry of the Eu\(^{3+}\) ions leads to the suppression of \(5D_0 \rightarrow 7F_2\) transitions of Eu\(^{3+}\) under excitations of 394 nm and 248 nm, respectively. The PL behavior of \(5D_0 \rightarrow 7F_{2,4}\) will be discussed due to the obvious changes. According to a previous report, the theoretical calculation result of the relative photonic band-gap width, \(\Delta/\lambda_0\), for an fcc phase, where \(\Delta\) is the width at half-maximum of the peak and \(\lambda_0\) is the centre wavelength of the peak, is 7%\(^{29}\); we can establish that the transition \(5D_0 \rightarrow 7F_2\) (611 nm) of Eu\(^{3+}\) lies in the short range of the stop-band gap and the transition \(5D_0 \rightarrow 7F_2\) (690 nm) of Eu\(^{3+}\) overlaps with the stop-band gap. As seen from the figures, first, the PL intensity of \(5D_0 \rightarrow 7F_2\) of the mono-layered 2D photonic crystals with the hexagonal lattice is significantly enhanced compared with those of the LEuH nanosheets and the core–shell-structured SiO\(_2@LEuH\) spheres. For this opal system, the \(5D_0 \rightarrow 7F_2\) transition at 611 nm lies in the short range of the stop-band gap, where the standing wave corresponds to a position at the pole, which results in the photoluminescence enhancement. Second, the PL intensity of the \(5D_0 \rightarrow 7F_4\) transition at \(\sim\)690 nm of the periodic monolayer decreases compared with that of the LEuH nanosheets due to the overlap of \(5D_0 \rightarrow 7F_4\) with the stop-band gap. Meanwhile, the \(5D_0 \rightarrow 7F_{2,4}\) transitions of Eu\(^{3+}\) are electric-dipole allowed and their intensities are sensitive to the local structure surrounding the Eu\(^{3+}\) ions. The higher site symmetry of the Eu\(^{3+}\) ions leads to the suppression of \(5D_0 \rightarrow 7F_{2,4}\) of the core–shell-structured SiO\(_2@LEuH\)s compared with that observed for the LEuH nanosheets. Overall, as seen from the enhanced PL of the typical \(5D_0 \rightarrow 7F_2\) transition and the higher PL intensity of the periodic monolayer compared with those of the other samples, it can be considered that the enhancement in the photoluminescence of LEuHs is achieved via the modification of the photonic crystal structures.

It is worth noting that to the best of our knowledge, this would be the first report on the photoluminescence enhancement of LRHs due to the modification of the photonic crystal structures; this would also be the first report of a method of enhancing the photoluminescence of the LRHs without any chemical treatments such as calcination or topology reactions.

**Conclusions**

In this paper, we synthesized the core–shell structures of rare-earth hydroxide nanosheets and further formed a mono-
layered 2D colloidal crystal with a hexagonal lattice on a quartz substrate through an evaporation-induced assembly process. First, the most important aspect is that for the first time, we introduced LRHs into the photonic crystals and achieved enhancement in the photoluminescence of LRHs without any further chemical treatments. This provides a strategy to enhance the photoluminescence of LRHs. Second, obtaining solidified rare-earth hydroxide nanosheets. Third, the modification of the photonic crystals for the spontaneous emission of Eu$^{3+}$ would enrich the study of the effects of the photonic crystal structures on opal structures.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

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References