



Cite this: *RSC Adv.*, 2019, 9, 11634

Charge balance control of quantum dot light emitting diodes with atomic layer deposited aluminum oxide interlayers†

Hoseok Jin,^a Hyungseok Moon,^a Woosuk Lee,^a Hyeok Hwangbo,^a Sang Heon Yong,^a Ho Kyoong Chung^b and Heeyeop Chae^b *^{ab}

We developed a 1.0 nm thick aluminum oxide (Al₂O₃) interlayer as an electron blocking layer to reduce leakage current and suppress exciton quenching induced by charge imbalance in inverted quantum dot light emitting diodes (QLEDs). The Al₂O₃ interlayer was deposited by an atomic layer deposition (ALD) process that allows precise thickness control. The Al₂O₃ interlayer lowers the mobility of electrons and reduces Auger recombination which causes the degradation of device performance. A maximum current efficiency of 51.2 cd A⁻¹ and an external quantum efficiency (EQE) of 12.2% were achieved in the inverted QLEDs with the Al₂O₃ interlayer. The Al₂O₃ interlayer increased device efficiency by 1.1 times, increased device lifetime by 6 times, and contributed to reducing efficiency roll-off from 38.6% to 19.6% at a current density up to 150 mA cm⁻². The improvement of device performance by the Al₂O₃ interlayer is attributed to the reduction of electron injection and exciton quenching induced by zinc oxide (ZnO) nanoparticles (NPs). This work demonstrates that the Al₂O₃ interlayer is a promising solution for charge control in QLEDs and that the ALD process is a reliable approach for atomic scale thickness control for QLEDs.

Received 7th January 2019
 Accepted 3rd April 2019

DOI: 10.1039/c9ra00145j

rsc.li/rsc-advances

Introduction

Colloidal quantum dots (QDs) are attracting attention as a promising light emitting material for optoelectronic devices due to their high color purity, high quantum yield and low cost solution processing.^{1–3} QDs have been widely applied in many applications such as solar cells, light emitting diodes and biological applications.^{4–8} QDs are particularly applicable to display and lighting devices because the emission wavelength can be easily shifted depending on the size, shape, and composition of the QDs.^{9–13} The device performance of the state-of-the-art QLEDs is comparable to that of commercially available organic light emitting diodes (OLEDs).^{14–17} However, the stability of QLEDs lags behind that of OLEDs and it needs to be overcome for commercialization.

The efficiency and stability of QLEDs are known to be strongly dependent on Auger recombination.^{18–20} Auger recombination, a non-radiative decay process, is an exciton quenching mechanism in which extra carriers absorb exciton energy. The exciton quenching reduces device efficiency, induces efficiency

roll-off and shortens device lifetime. Auger recombination is caused by the imbalance of the charge mobility in transport layers.^{11–14} Electron mobility of ZnO NPs electron transport layers (ETL) is known to be faster than the hole mobility of hole transport layers (HTL) in QLEDs.¹⁴

Several approaches are proposed to enhance the efficiency and stability of QLEDs by controlling electron injection. Adopting composition of QDs,²¹ modifying surface ligands²² or introducing extra shell of the core/shell/shell structure of QDs effectively suppress efficiency roll-off behavior.¹⁹ Various thin interlayers were introduced to improve the charge balance by suppressing electron injection with poly(methyl methacrylate),¹⁴ polyethylenimine ethoxylated (PEIE),¹⁵ polyethylenimine (PEI),²³ polyfluorene derivatives,²⁴ and soluble Al₂O₃.²² Organic materials are sensitive to oxygen and moisture than inorganic materials and decrease stability of QLEDs. All organic and inorganic interlayers have been formed with solution processes so far.^{14,15,23,25} Thickness control in atomic-scale of interlayer is absolutely important because it forms tunneling barrier in the device. However, it is challenging to control the atomic scale thickness with the solution processes.

In this study, we introduced the inorganic Al₂O₃ interlayer between ETL and emitting layer (EML) as the electron blocking layer by a spatially-resolved ALD process that allows for a precisely controlled thickness. ALD processes can form thin conformal films and provide good uniformity and precise thickness control.^{26,27} The thickness effect of Al₂O₃ interlayer on

^aSchool of Chemical Engineering, Sungkyunkwan University, Suwon, 16419, Republic of Korea. E-mail: hchae@skku.edu

^bSKKU Advanced Institute of Nanotechnology, Sungkyunkwan University, Suwon, 16419, Republic of Korea

† Electronic supplementary information (ESI) available. See DOI: 10.1039/c9ra00145j



device efficiency and lifetime was investigated in inverted QLEDs.

Experimental

Inverted QLEDs fabrication

The device with a conventional structure is not suitable in our work due to QDs degradation by plasma in the spatially-resolved ALD process (Fig. S1†). Inverted QLEDs made in this work consists of indium tin oxide (ITO)/ZnO/Al₂O₃/QDs/PEIE/poly(*N,N'*-bis(4-butylphenyl)-*N,N'*-bis(phenyl)benzidine (poly-TPD)/MoO_x/Al as shown in Fig. 1a^{15,33} and the band energy levels of each materials are shown in Fig. 1b. UV-visible absorption spectra and ultraviolet photoelectron spectra of QDs, poly-TPD and ZnO NPs are illustrated in Fig. S2 and S3.† A patterned ITO on glass substrate was cleaned sequentially with acetone and methanol. A 70 nm-thick ZnO NPs (80 mg ml⁻¹, in ethanol) ETL was spin-coated at 1500 rpm for 30 s and baked at 150 °C. The Al₂O₃ interlayers from 0 to 3.0 nm were deposited on ZnO NPs layers by a spatially-resolved ALD process. A 25 nm-thick CdSe@ZnS/ZnS QDs (in hexane/octane, OD = 1.5), a 7.5 nm thick-PEIE (in 2-methoxyethanol, 0.5 wt%), a 40 nm-thick poly-TPD HTL (in chlorobenzene, 10 mg ml⁻¹) and a 10 nm-thick MoO_x hole injection layer (in acetonitrile, 10 mg ml⁻¹) were spin-coated at 3000 rpm for 30 s and baked at 150 °C sequentially. (MoO₃)₁₂·H₃PO₄·(H₂O)_x (PMAH) solution (in acetonitrile, 10 mg ml⁻¹) was stirred for 10 min at 60 °C and filtered through polyvinylidene fluoride (PVDF) membrane before the spin-coating. All layers were deposited and baked in a nitrogen-filled glove box (H₂O < 1 ppm). Finally, Al electrodes (100 nm) were deposited by a thermal evaporation process under high vacuum of ~6 × 10⁻⁶ torr.

Deposition of Al₂O₃ interlayer by spatially-resolved ALD process

The Al₂O₃ interlayer was formed by trimethylaluminum (TMA) as an aluminum precursor and N₂O plasma as an oxygen radical precursor. TMA with Ar carrier gas and N₂O plasma are supplied to source and plasma zone. The sequential procedure of spatially-resolved ALD process are shown Fig. 1c. Each zone is separated by adjacent purge zones to prevent gas mixing and the gap between moving substrate and gas supplied head is less than 2 mm. The gas flowrate of TMA precursor with Ar carrier gas and N₂O precursor are 50 and 20 sccm. The Al₂O₃ interlayer was deposited on the devices at the substrate temperature of 80 °C with the plasma power of 150 W. Substrate moving speed is 125 mm s⁻¹ and the growth per cycle of spatially-resolved ALD process is 1.0 Å per cycle. Binding energy and atomic concentration of Al₂O₃ deposited by spatially-resolved ALD process are shown in Fig. S4.†

Characterizations

Absorption spectra was taken with a UV/visible spectrophotometer (JASCO, V630). PL emission spectra was obtained by a Cary Eclipse fluorescence spectrophotometer (Agilent Technologies, G9803AA). The diameters of QDs and ZnO NPs were

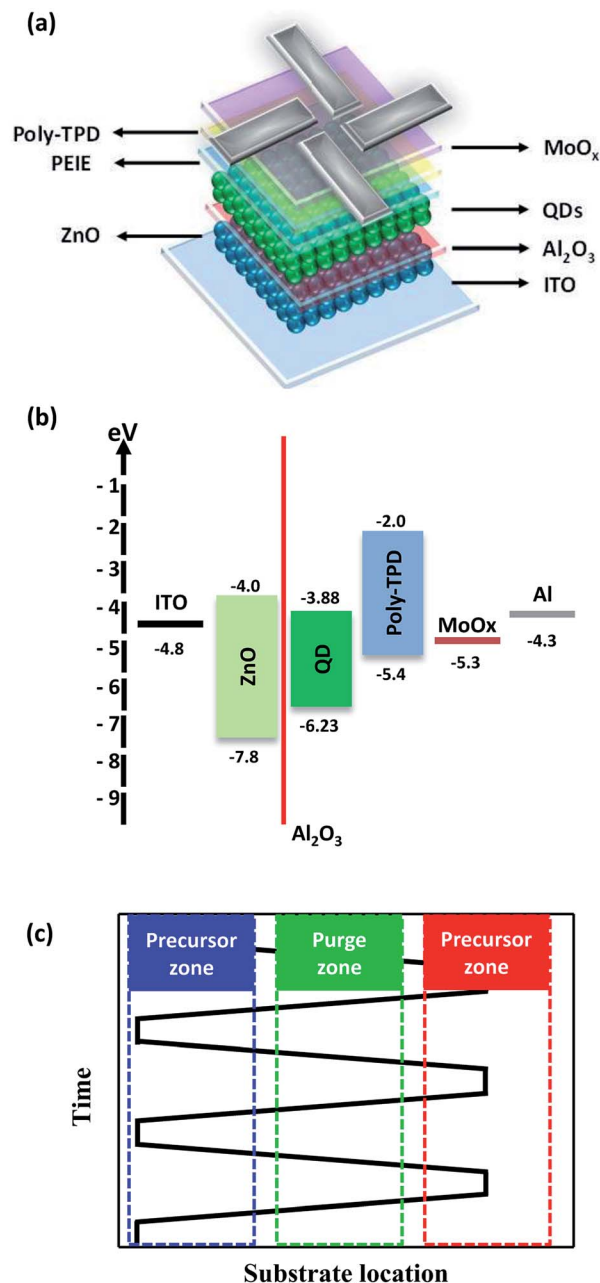


Fig. 1 (a) The schematic structure of the QLED with Al₂O₃ interlayer between QDs and ZnO, (b) band energy level diagram of each materials, (c) the schematic view of the sequential procedure of spatially-resolved ALD process.

determined by a *C_s*-corrected high-resolution transmission electron microscopy (HRTEM) with an accelerating voltage 200 kV (JEOL, JEM-ARM 200F). X-ray diffraction pattern was taken with an X-ray diffractometer (Bruker Corporation, D8 ADVANCE). Surface roughness was measured by high-resolution atomic force microscope (SII Nanotechnology, SPA-300HV). Band energy levels of each materials were determined using X-ray monochromator (Thermo-Scientific, ESCALAB 250Xi) and UV/visible spectrophotometer. Electrical and optical properties of inverted QLEDs were taken with



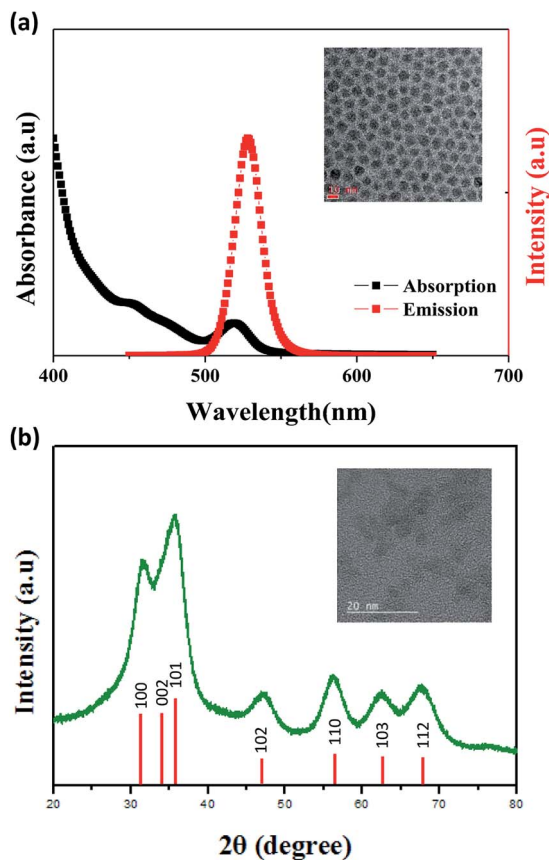


Fig. 2 (a) The absorption and PL spectra of green QDs in hexane/octane. (inset) TEM image of QDs, (b) X-ray diffraction pattern of ZnO NPs. (inset) TEM image of ZnO NPs.

spectroradiometer (Konica-Minolta, CS-2000) coupled with a voltage–current source unit (Tektronix, Keithley 2400). Impedance spectra depending on the thickness of the Al_2O_3 interlayer was taken with a solar simulator (ABET Technologies, Sun 2000).

Results and discussion

The device performances of QLEDs are strongly dependent on the QDs qualities, such as quantum yield and size distribution. QDs used in our work are an alloyed core/shell structure QDs with a thick shell for stability²⁸ and the absorption and PL spectra of the green QDs used in our work are shown in Fig. 2a. The QDs have an average diameter of 11 nm (inset Fig. 2a) and a high quantum yield ($\sim 90\%$), which contributes to the fabrication of efficient devices. The uniform distribution of QDs contributes to narrow emission bandwidth which a full width at half maximum (FWHM) is 20 nm with a peak emission wavelength of QDs at 528 nm in hexane/octane. The high degree of crystallinity is required for good device performance because the crystallinity of ZnO NPs is related to electronic properties.⁴ ZnO NPs were synthesized for ETL with the diameter of 3 nm as shown in inset Fig. 2b, and a well crystallized wurtzite structure

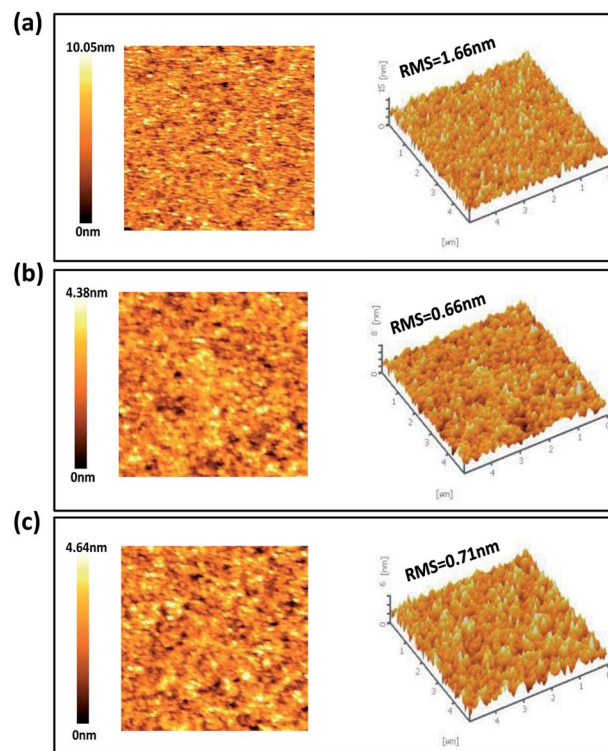


Fig. 3 Surface roughness of (a) ZnO, (b) ZnO/ Al_2O_3 (1 nm), and (c) ZnO/ Al_2O_3 (2 nm).

of ZnO NPs was identified by X-ray diffraction pattern as shown in Fig. 2b.

The smooth underlying layer is required for the uniform QDs layers because the efficiency and stability are strongly dependent on the QDs thickness uniformity.³⁰ The surface roughness of ZnO and ZnO/ Al_2O_3 films is shown in Fig. 3. The deposition of the Al_2O_3 layer on ZnO NPs *via* the ALD process reduced the root-mean squared (RMS) surface roughness to 0.66 nm of 1 nm Al_2O_3 layer and 0.71 nm of 2 nm Al_2O_3 layer from 1.66 nm without the Al_2O_3 layer. It has been demonstrated that Al_2O_3 could be deposited well on the ZnO NP surface and the reduction of RMS roughness is attributed to the filling of a valley in the ZnO thin film by the Al_2O_3 interlayer.²⁹ Smooth surface morphology of the ETL by the Al_2O_3 interlayer enables uniform QD thin film deposition and low leakage current in QLEDs, which is critical to improve device performance.²³ We also measured PL spectra to investigate the effect of Al_2O_3 interlayer on non-radiative electron transfer induced by ZnO NPs as shown in Fig. S5.†^{14,35,36} The Al_2O_3 interlayer contributed to preserve fluorescence of QDs by preventing non-radiative electron transfer. The Al_2O_3 interlayer serves not only as the electron blocking layer suppressing electron injection but also as a buffer layer preventing electron transfer.

The electrical and optical properties of inverted QLEDs depending on the thickness of the Al_2O_3 interlayer (from 0 to 3.0 nm) are shown in Fig. 4. The current density of QLEDs was reduced as the thickness of the Al_2O_3 interlayer increased, and the reduction of the current density was particularly prominent



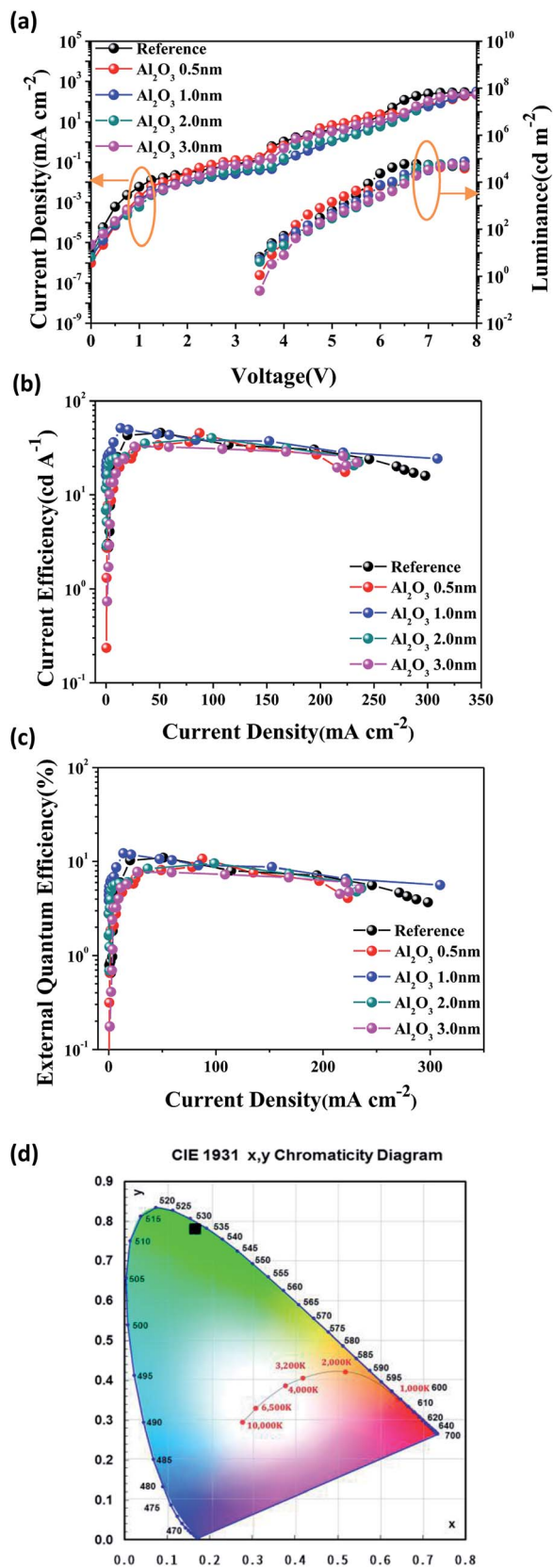


Fig. 4 Optical and electrical properties of QLEDs (a) current density–voltage and luminance–voltage, (b) current efficiency–current density, (c) external quantum efficiency–current density and (d) CIE coordinate of the QLEDs with the 1.0 nm Al_2O_3 interlayer.

Table 1 Summarized device performances of QLEDs depending on the thickness of Al_2O_3 interlayer

	Max. luminance (cd m^{-2})	Max. current efficiency (cd A^{-1})	Max. EQE (%)
Reference	59 000	45.8	10.9
Al_2O_3 (0.5 nm)	52 800	45.3	10.7
Al_2O_3 (1.0 nm)	75 100	51.2	12.2
Al_2O_3 (2.0 nm)	57 600	40.2	9.5
Al_2O_3 (3.0 nm)	57 100	32.6	7.8

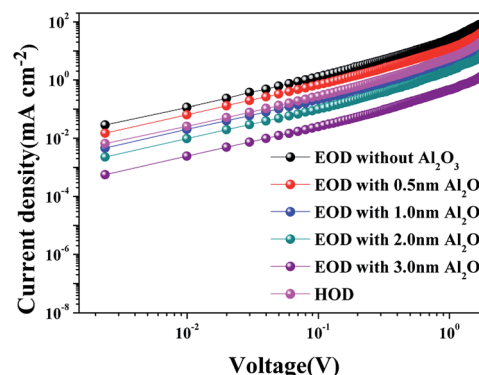


Fig. 5 Current densities of electron only device (EOD, ITO/ZnO/QD/ZnO/Al) without and with Al_2O_3 interlayers, and current density of hole only device (HOD, ITO/poly-TPD/QD/poly-TPD/MoO_x/Al).

before a turn-on voltage (<3.5 V). The turn-on voltage and luminance for the device without the Al_2O_3 interlayer are virtually the same as for the device with the Al_2O_3 interlayer, whereas the current density is much higher as shown in Fig. 4a. The current efficiency and EQE with the 1.0 nm Al_2O_3 interlayer show the highest enhancement values compared to those without Al_2O_3 interlayer as shown in Fig. 4b and c and the device performances summarized in Table 1. It demonstrates that the Al_2O_3 interlayer suppress excessive electron injection due to its insulating property and that charge injection is balanced in the device with the 1.0 nm Al_2O_3 interlayer. On the other hand, the performance with a 0.5 nm thick Al_2O_3 interlayer is nearly identical to that without the Al_2O_3 interlayer. It demonstrates that the very thin interlayer hardly interferes with the injection of electrons, even though it is an insulating materials. We are also worth noting that the device performances drastically decreases and the turn-on voltage is delayed when the thickness of the Al_2O_3 interlayer is above 3.0 nm (Fig. S6[†]). The reduction of device performance and the delay of the turn-on voltage are evident when Al_2O_3 interlayer is thicker than 5.0 nm (Fig. S7[†]) and this phenomena can be explained with tunneling effect.³¹ The Al_2O_3 interlayer acts as a tunneling barrier, and the tunneling current decreases exponentially as the thickness of the barrier layer increases.³² However, in our results, the thickness dependence of the current does not seem to be exponential and this might because the Al_2O_3 film on the ZnO NP film was not continuous film. The optimum thickness of Al_2O_3 interlayer is found to be 1.0 nm and the 1.1-fold



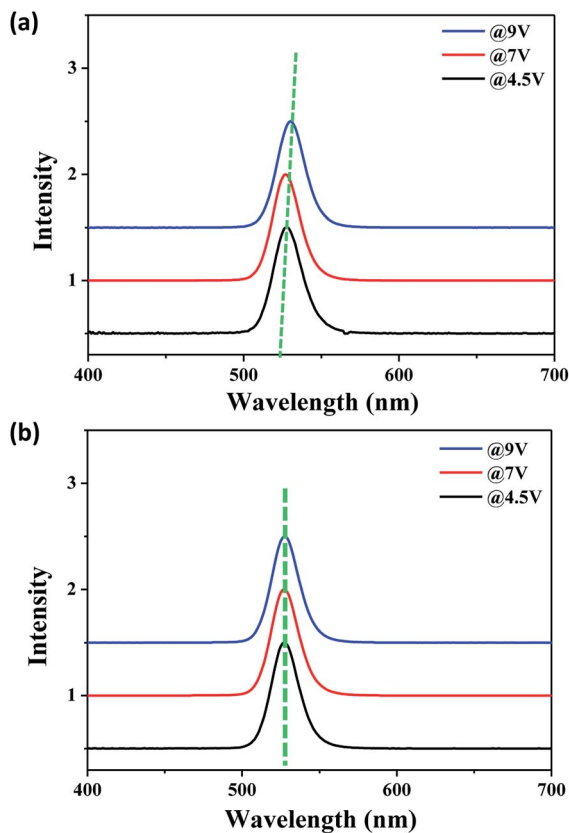


Fig. 6 EL spectra of the devices (a) without and (b) with the Al_2O_3 interlayer depending on applied voltages.

improved current efficiency of 51.2 cd A^{-1} and EQE of 12.2% (average: 9.6 ± 1.3) were achieved by suppressing excessive electron injection. The EQEs of the 20 unencapsulated QLEDs with the 1.0 nm Al_2O_3 interlayer are summarized in histogram (Fig. S8†). The device with Al_2O_3 interlayer emit deep green light with the Commission International de l'Éclairage (CIE) coordinates of (0.16, 0.78) as shown in Fig. 4d.

The charge balance depending on the thickness of the Al_2O_3 interlayer was investigated by electron only (ITO/ZnO/QD/ZnO/Al) and hole only (ITO/poly-TPD/QD/poly-TPD/MoO_x/Al) devices as shown in Fig. 5. The significantly higher current density is observed in the electron only device without the Al_2O_3 interlayer than that of hole only device and this indicates the significant imbalance of charge injection.³⁴ The current density of the electron only device with the 0.5 nm Al_2O_3 interlayer is similar to that of the electron only device without the Al_2O_3 interlayer. The current density decreases with the increase as Al_2O_3 interlayer thickness increases, and the current density of the electron only device is similar to that of hole only device with the 1.0 nm Al_2O_3 interlayer. Charge injection with the 1.0 nm Al_2O_3 interlayer is well balanced and thinner Al_2O_3 interlayer hardly affects with electron injection. We also analyzed impedance to elucidate thickness effect of the Al_2O_3 interlayer further from electrochemical impedance spectroscopy (EIS). The amplitude of impedance increases drastically and phase angles are in the range of 80–90° when the thickness

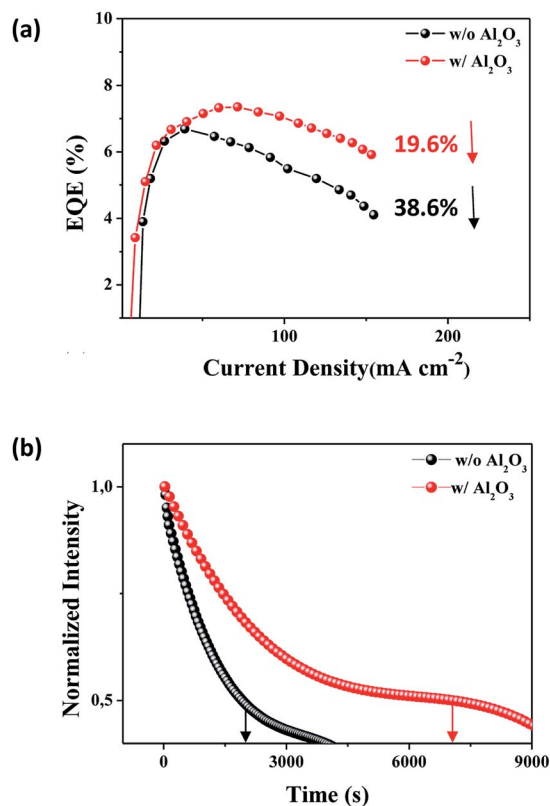


Fig. 7 (a) Efficiency roll-off and (b) device lifetime of QLEDs without and with Al_2O_3 interlayer.

of the Al_2O_3 interlayer is above 2.0 nm (Fig. S9†). Both the impedance and the phase angle responses indicate the drastic increase in resistance and capacitance with the Al_2O_3 interlayer thicker than 1.0 nm resulting in the significant suppression of charge injection.

The effect of applied voltage on EL spectra was investigated with and without the Al_2O_3 interlayer as shown in Fig. 6. The EL emission of the device with the Al_2O_3 interlayer remains unchanged with different applied voltages, whereas the EL emission of the device without the Al_2O_3 interlayer shifts to lower energy. The shift of EL emission peak at different applied voltages reflects the operational stability of QLEDs, and this is known to be related to QDs degradation caused by joule heating originated from Auger recombination in QLEDs.^{19,37} The peak shift of EL emission indicates that the insulating Al_2O_3 interlayer also improves the stability of EL emission by suppressing the joule heating.

We investigated efficiency roll-off that can be an indicator of the exciton quenching by Auger recombination on the QLEDs as shown in Fig. 7a.¹⁹ The efficiency roll-off of 38.6% at 150 mA cm^{-2} was observed without the Al_2O_3 interlayer and 19.6% observed with the Al_2O_3 interlayer. The EQE with the Al_2O_3 interlayer was also maintained higher than that without Al_2O_3 interlayer throughout the entire experimental current density range. Device lifetime was monitored as shown in Fig. 7b and also improved by the Al_2O_3 interlayer since the exciton quenching, such as Auger recombination, was suppressed by



interfering with excess electron injection. The devices were tested with a constant driving current density (10 mA cm^{-2}) without encapsulation for the lifetime measurement. The lifetime is defined here with the time for the initial luminance to reduce to half. Initial luminance was 2600 and 1750 cd m^{-2} with and without the Al_2O_3 interlayer, respectively. The device with the Al_2O_3 interlayer shows the device half lifetime of 7070 s , which improved by 3-fold as compared to 2030 s for the device without the Al_2O_3 interlayer. The half lifetime can be converted into the lifetime at 100 cd m^{-2} with the following relation.³⁸

$$L_0^n T_{50} = \text{constant} \quad (n = 1.5) \quad (1)$$

At luminance of 100 cd m^{-2} , the device lifetime is about 42 h and 260 h for the devices without and with Al_2O_3 interlayer, respectively. The device lifetime with the Al_2O_3 interlayer is improved about 6 times than the device without the Al_2O_3 interlayer.

Conclusions

We deposited a 1.0 nm thick Al_2O_3 interlayer by an ALD process between EML and ETL of inverted QLED. The ALD process enables precise thickness control in the atomic scale. The 1.0 nm thick Al_2O_3 interlayer improves the charge balance in the device and the device with the interlayer exhibited device performances of 51.2 cd A^{-1} in current efficiency and 12.2% in EQE. The efficiency roll-off was suppressed from 38.6% to 19.6% at current density up to 150 mA cm^{-2} and the device lifetime increased about 6 times than the device without Al_2O_3 interlayer. The Al_2O_3 interlayer as a tunneling barrier reduces leakage current of the device and improves radiative recombination in the EML by suppressing Auger recombination and non-radiative electron transfer induced by ZnO NPs. This work demonstrates for the first time within our knowledge in the inverted QLEDs that the Al_2O_3 interlayer is a promising candidate for charge control and paves a reliable way for fabricating the efficient and stable devices.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This research was supported by the Global Frontier R&D Program on Center for Multiscale Energy System funded by the National Research Foundation under the Ministry of Science, ICT & Future Planning, Korea (2012M3A6A7054855). This work was supported by the National Research Foundation of Korea (NRF) grant funded by the Korea government (MSIT) (2018R1A2A3074950).

Notes and references

- P. O. Anikeeva, J. E. Halpert, M. G. Bawendi and V. Bulovic, *Nano Lett.*, 2009, **9**, 2532.
- J. Zhao, J. A. Bardecker, A. M. Munro, M. S. Liu, Y. Niu, I. K. Ding, J. Luo, B. Chen, A. K. Y. Jen and D. S. Ginger, *Nano Lett.*, 2006, **6**, 463–467.
- D. V. Talapin, J. Lee, M. V. Kovalenko and E. V. Shevchenko, *Chem. Rev.*, 2010, **110**, 389–458.
- L. Qian, Y. Zheng, J. Xue and P. H. Holloway, *Nat. Photonics*, 2011, **5**, 543–548.
- H. J. Yun, T. Paik, M. E. Edley, J. B. Baxter and C. B. Murray, *ACS Appl. Mater. Interfaces*, 2014, **6**, 3721–3728.
- H. Y. Yang, Y. Fu, M. S. Jang, Y. Li, J. H. Lee, H. Chae and D. S. Lee, *ACS Appl. Mater. Interfaces*, 2016, **8**, 35021–35032.
- J. W. Stouwdam and R. A. J. Janssen, *J. Mater. Chem.*, 2008, **18**, 1889–1894.
- N. Kim, W. Na, W. Yin, H. Jin, T. K. Ahn, S. M. Cho and H. Chae, *J. Mater. Chem. C*, 2016, **4**, 2457–2462.
- B. Mashford, M. Stevenson, Z. Popovic, C. Hamilton, Z. Zhou, C. Breen, J. Steckel, V. Bulovic, M. Bawendi, S. Coe-Sullivan and P. T. Kazlas, *Nat. Photonics*, 2013, **7**, 407–412.
- Y. Fu, D. Kim, H. Moon, H. Yang and H. Chae, *J. Mater. Chem. C*, 2017, **5**, 522.
- J. Pan, J. Chen, Q. Huang, Q. Khan, X. Liu, Z. Tao, Z. Zhang, W. Lei and A. Nathan, *ACS Photonics*, 2016, **3**, 215–222.
- H. M. Kim, D. Geng, J. Kim, E. Hwang and J. Jang, *ACS Appl. Mater. Interfaces*, 2016, **8**, 28727–28736.
- G. Zaiats, S. Ikeda, S. Kinge and P. V. Kamat, *ACS Appl. Mater. Interfaces*, 2017, **9**, 30741–30745.
- X. Dai, Z. Zhang, Y. Jin, Y. Niu, H. Cao, X. Liang, X. Chen, J. Wang and X. Peng, *Nature*, 2014, **515**, 96–99.
- D. Kim, Y. Fu, S. Kim, W. Lee, K. H. Lee, H. K. Chung, H. J. Lee, H. Yang and H. Chae, *ACS Nano*, 2017, **11**, 1982–1990.
- K. H. Lee, J. H. Lee, W. S. Song, H. Ko, C. Lee, J. H. Lee and H. Yang, *ACS Nano*, 2013, **7**, 7295–7302.
- Q. Lin, L. Wang, Z. Li, H. Shen, L. Guo, Y. Kuang, H. Wang and L. S. Li, *ACS Photonics*, 2018, **5**, 939–946.
- V. I. Klimov, A. A. Mikhailovsky, D. W. McBranch, C. A. Leatherdale and M. G. Bawendi, *Science*, 2000, **287**, 1011–1013.
- W. K. Bae, Y. S. Park, J. Lim, D. Lee, L. A. Padilha, H. McDaniel, I. Robel, C. Lee, J. M. Pietryga and V. I. Klimov, *Nat. Commun.*, 2013, **7**, 9019–9026.
- W. K. Bae, S. Brovelli and V. I. Klimov, *MRS Bull.*, 2013, **38**, 721–730.
- J. Kwak, J. Lim, M. Park, S. Lee, K. Char and C. Lee, *Nano Lett.*, 2015, **15**, 3793–3799.
- P. Rastogi, F. Palazon, M. Prato, F. Di Stasio and R. Krahne, *ACS Appl. Mater. Interfaces*, 2018, **10**, 5665–5672.
- A. Castelli, F. Meinardi, M. Pasini, F. Galeotti, V. Pinchetti, M. Lorenzon, L. Manna, I. Moreels, U. Giovanella and S. Brovelli, *Nano Lett.*, 2015, **15**, 5455.



- 24 K. Ding, H. Chen, L. Fan, B. Wang, Z. Huang, S. Zhuang, B. Hu and L. Wang, *ACS Appl. Mater. Interfaces*, 2017, **9**, 20231–20238.
- 25 H. Zhang, N. Sui, X. Chi, Y. Wang, Q. Liu, H. Zhang and W. Ji, *ACS Appl. Mater. Interfaces*, 2016, **8**, 31385–31391.
- 26 Y. Q. Yang, Y. Duan, P. Chen, F. B. Sun, Y. H. Duan, X. Wang and D. J. Yang, *J. Phys. Chem. C*, 2013, **117**, 20308–20312.
- 27 Y. Duan, F. Sun, Y. Yang, P. Chen, D. Yang, Y. Duan and X. Wang, *ACS Appl. Mater. Interfaces*, 2014, **6**, 3799–3804.
- 28 Y. Fu, D. Kim, W. Jiang, W. Yin, T. K. Ahn and H. Chae, *RSC Adv.*, 2017, **7**, 40866–40872.
- 29 J.-H. Choi, J. Kim, S. J. Oh, D. Kim, Y.-H. Kim, H. Chae and H. Kim, *Met. Mater. Int.*, 2016, **22**, 723–729.
- 30 H. M. Kim, J. H. Youn, G. J. Seo and J. Jang, *J. Mater. Chem. C*, 2013, **1**, 1567–1573.
- 31 H. A. Al Attar and A. P. Monkman, *Adv. Funct. Mater.*, 2006, **16**, 2231–2242.
- 32 Y. E. Kim, H. Park and J. J. Kim, *Appl. Phys. Lett.*, 1996, **69**, 599–601.
- 33 Y. Zhou, C. Fuentes-Hernandez, J. Shim, J. Meyer, A. J. Giordano, H. Li, P. Winget, T. Papadopoulos, H. Cheun, J. Kim, M. Fenoll, A. Dindar, W. Haske, E. Najafabadi, T. M. Khan, H. Sojoudi, S. Barlow, S. Graham, J. L. Bredas, S. R. Marder, A. Kahn and B. Kippelen, *Science*, 2012, **336**, 327–332.
- 34 J. Li, Z. Liang, Q. Su, H. Jin, K. Wang, G. Xu and X. Xu, *ACS Appl. Mater. Interfaces*, 2018, **10**, 3865–3873.
- 35 K. Tvrđy, P. A. Frantsuzov and P. V. Kamat, *Proc. Natl. Acad. Sci. U. S. A.*, 2011, **108**, 29–34.
- 36 V. Wood, M. J. Panzer, L. E. Halpert, J. M. Caruge, M. G. Bawendi and V. Bulovic, *ACS Nano*, 2009, **3**, 3581–3586.
- 37 S. Shendre, V. K. Sharma, C. Dang and H. V. Demir, *ACS Photonics*, 2018, **5**, 480–486.
- 38 P. Wellmann, M. Hofmann, O. Zeika, A. Werner, J. Birnstock, R. Meerheim, G. He, K. Walzer, M. Pfeiffer and K. Leo, *J. Soc. Inf. Disp.*, 2005, **13**, 393–397.

