



Cite this: *RSC Adv.*, 2019, 9, 15900

An amorphous MoS_x modified g-C₃N₄ composite for efficient photocatalytic hydrogen evolution under visible light†

Xia Li,^a Bo Wang,^a Xia Shu,^a Dongmei Wang,^a Guangqing Xu,^a Jun Lv^{b*} and Yucheng Wu^{c*}

In this work, an MoS_x/g-C₃N₄ composite photocatalyst was successfully fabricated by a sonochemical approach, where amorphous MoS_x was synthesized using a hydrothermal method with Na₂MoO₄·H₂O, H₄SiO₄(W₃O₉)₄ and CH₃CSNH₂ as precursors, and g-C₃N₄ nanosheets were produced using a two-step thermal polycondensation method. The hydrogen-evolution performance of the MoS_x/g-C₃N₄ composite was tested under visible light. The results show that the H₂-evolution rate of the MoS_x/g-C₃N₄ (7 wt%) photocatalyst reaches a maximum value of 1586 μmol g⁻¹ h⁻¹, which is about 70 times that of pure g-C₃N₄ nanosheets. The main reason is that amorphous MoS_x forms intimate heterojunctions with g-C₃N₄ nanosheets, and the introduction of MoS_x into g-C₃N₄ nanosheets not only enhances the ability to convert H⁺ into H₂, but also promotes the separation of photoinduced electron-hole pairs for the photocatalyst. BET analysis shows that the specific surface area and pore volume of g-C₃N₄ are decreased in the presence of MoS_x. XPS analysis manifests that MoS_x provides a number of active sites. Mott-Schottky plots show that the conduction band of MoS_x (−0.18 V vs. E_{Ag/AgCl}, pH = 7) is more negative than that of g-C₃N₄ nanosheets.

Received 29th November 2018

Accepted 17th April 2019

DOI: 10.1039/c8ra09806a

rsc.li/rsc-advances

1. Introduction

With the intensification of the energy crisis and environmental pollution, the human need for green and renewable energy has become more urgent. Recently, research into and applications of solar energy have attracted great attention.^{1,2} There is little doubt that once solar energy is fully utilized, mankind can solve the energy crisis and alleviate environmental pollution. Hydrogen, a clean fuel with its combustion product being environmentally friendly water, is a suitable substitute for petroleum fuels.³ Furthermore, hydrogen can be obtained by water splitting in large quantities, so how to utilize solar energy to produce hydrogen by water splitting has become one of the key issues in achieving sustainable development.⁴ In recent years, the hydrogen evolution reaction (HER) by semiconductors has increasingly become the focus of research. Scientists have conducted extensive and in-depth studies on solar-driven water splitting to obtain hydrogen using

semiconductors such as TiO₂,⁵ CdS,^{6,7} and graphene.⁸ However, these materials have many different drawbacks, including the narrow range of response to visible light, environmental hazards and poor stability, which severely restrict their practical applications. For these reasons, the development of novel visible light responsive photocatalysts with high efficiency and good stability has become an important research direction.

Graphitic carbon nitride (g-C₃N₄), a non-metallic semiconductor with a two-dimensional structure like graphene, has received great attention in the field of photocatalysis, due to its earth-abundance, high physicochemical stability, suitable band structure ($E_g = 2.7$ eV) and low cost.^{9,10} Since it was first reported as a metal-free semiconductor photocatalyst for H₂ production in 2009,¹¹ g-C₃N₄ has become a potential novel photocatalytic material for hydrogen production. Nevertheless, the photocatalytic performance of g-C₃N₄ is typically limited due to its high recombination of photoexcited electron-hole pairs, small surface area and low quantum efficiency.^{12,13} Different methods have been used to improve its photocatalytic properties by nanostructure modulation,^{14–16} ion doping,^{17,18} loading with co-catalysts,¹⁹ as well as compounding with other semiconductors.²⁰

Molybdenum sulfide is a layered transition-metal dichalcogenide, which has been considered a promising candidate to replace Pt as a co-catalyst for HER. Molybdenum sulfide has superior properties, such as low cost, relative abundance, tunable band structure and absorption of visible light.

^aSchool of Material Sciences and Engineering, Hefei University of Technology, Hefei 230009, China. E-mail: ycwu@hfut.edu.cn; lvjun117@126.com

^bGuangxi Key Laboratory for Electrochemical Energy Materials, Guangxi University, Nanning, 530004, China

^cAnhui Provincial Key Laboratory of Advanced Functional Materials and Devices, Hefei University of Technology, Hefei 230009, China

† Electronic supplementary information (ESI) available. See DOI: 10.1039/c8ra09806a



Crystalline MoS₂ has been reported to be an excellent catalyst for both photocatalytic and electrocatalytic HER due to the existence of active sites derived from the sulfur edges of MoS₂ crystal layers.^{21,22} Reports have always focused on MoS₂ as a co-catalyst to enhance the photocatalytic activity of HER under visible light: for example, MoS₂/C₃N₄,^{23,24} MoS₂/CdS,²⁵ MoS₂/graphene²⁶ and MoS₂/TiO₂.²⁷ Crystalline MoS₂ has two distinct phases: the 2H-phase and the 1T-phase. 2H-MoS₂ exists in the natural environment and is more stable than 1T-MoS₂, but its poor electrical conductivity limits its performance for HER. Whereas 1T-MoS₂ has high electron transfer capability and has been a vital research object.²⁸ In addition, amorphous molybdenum sulfide (MoS_x) has also been found to be highly active for HER because of its ample unsaturated atoms.²⁹ In fact, amorphous MoS_x has recently been demonstrated to be an excellent co-catalyst because of its active centers for HER. Huogen Yu and colleagues declared that an amorphous molybdenum sulfide modified g-C₃N₄ composite exhibits an obviously higher hydrogen-evolution activity than that of a crystalline molybdenum sulfide modified g-C₃N₄ composite.³⁰ More recently, Xiao Hai and colleagues further demonstrated that the co-catalytic performance of monolayer MoS₂ could be gradually enhanced by reducing its crystallinity, and a poorly crystalline MoS₂ monolayer shows even higher co-catalytic performance than that of Pt.³¹

Herein, a novel kind of MoS_x/g-C₃N₄ composite has been prepared by a sonochemical approach. The g-C₃N₄ nanosheets were synthesized by a thermal polycondensation method. The amorphous MoS_x was synthesized by a hydrothermal method. The photocatalytic HER of MoS_x/g-C₃N₄ was investigated and the mechanism of MoS_x/g-C₃N₄ in improving the photocatalytic activity is discussed in detail. This work supplies a novel strategy for using amorphous MoS_x as a co-catalyst to enhance the photocatalytic HER of g-C₃N₄ under visible light.

2. Experimental

2.1 Preparation

2.1.1 Preparation of an ultrathin g-C₃N₄ nanosheet photocatalyst. The g-C₃N₄ nanosheets were prepared by a two-step thermal polycondensation method using melamine powder as the precursor.^{9,32} In detail, 10 g of melamine was calcined in air at 550 °C for 4 h at a heating rate of 8 °C min⁻¹, then cooled down to room temperature. The obtained bulk g-C₃N₄ was milled into powder and then 0.4 g of the powder was heated at 500 °C for 2 h at a heating rate of 5 °C min⁻¹ to prepare the g-C₃N₄ nanosheets. Finally, 50 mg of the g-C₃N₄ nanosheets was dispersed in 80 mL of deionized water at room temperature and exfoliated by ultrasonication for 2 h. The final product was centrifuged at 12 000 rpm and dried at 60 °C for 12 h and was named UCN.

2.1.2 Preparation of amorphous MoS_x and crystalline MoS₂. The amorphous MoS_x was prepared by a hydrothermal method. 0.2419 g of Na₂MoO₄·H₂O, 2.878 g of H₄SiO₄(W₃O₉)₄ (silicotungstic acid) and 0.4508 g of CH₃CSNH₂ were dissolved in 45 mL of deionized water and stirred for 1 h. The solution was then transferred to a 50 mL Teflon-lined stainless-steel autoclave. The autoclave was sealed tightly and maintained at 240 °C

for 24 h. After naturally cooling down to room temperature, the obtained black solution was centrifuged at 10 000 rpm and the obtained precipitates were washed with deionized water and ethanol several times. The product was dried in a vacuum at 60 °C for 8 h and milled into powder, and denoted as primitive MoS_x. 50 mg of the primitive MoS_x was dispersed in 80 mL of deionized water at room temperature and exfoliated by ultrasonication for 2 h. The final product was centrifuged at 12 000 rpm, dried at 60 °C for 12 h and named MoS_x. Crystalline MoS₂ was obtained by heating the primitive MoS_x at 360 °C in ambient N₂ for 2 h.

2.1.3 Fabrication of MoS_x/UCN and MoS₂/UCN composite photocatalyst. The MoS_x/UCN photocatalyst was fabricated by a green environmental sonochemical approach. For the typical procedure, 3.5 mg of primitive MoS_x and 50 mg of g-C₃N₄ nanosheets were dissolved in 80 mL of deionized water. The suspension was ultrasonicated for 2 h and then dried at 60 °C for 12 h. With this method, the amorphous MoS_x/UCN composite photocatalyst was synthesized, denoted as MoS_x/UCN (7 wt%). By changing the amount of added MoS_x (1.5–5.5 mg), MoS_x/UCN (3–11 wt%), composite photocatalysts with different mass ratios of MoS_x can be obtained. MoS₂/UCN (7 wt%) was prepared by heating MoS_x/UCN (7 wt%) in ambient N₂ at 360 °C for 2 h.

2.2 Characterization

The power X-ray diffraction (XRD) patterns of the samples were recorded by a Rigaku D/MAX2500V X-ray diffractometer (Cu Kα1 irradiation, λ = 0.15406 Å). Transmission electron microscopy (TEM) was performed with an electron microscope (JEM-2100F, Japan) operating at 200 kV. The BET specific surface and pore volume of the samples were determined by a surface area and pore size analyzer (Quadrachrome evo, USA). X-ray photoelectron spectroscopy (XPS) analysis of the samples was performed using an ESCALAB250Xi photoelectron spectrometer (Thermo) with monochromatic Al Kα (1486.60 eV). The UV-vis spectra of the samples were measured on a UV-vis spectrophotometer (UV-3600, Shimadzu Corp, Japan). The photoluminescence (PL) spectra of the samples were measured using a fluorescence spectrophotometer (F-4500) with an excitation wavelength of 315 nm.

2.3 Photocatalytic hydrogen production

Photocatalytic H₂ production experiments were performed in a 300 mL sealed quartz reactor at room temperature. A 300 W Xe lamp with a UV-cutoff filter (λ > 400 nm) was used as the light source. In a typical photocatalytic experiment, 20 mg of photocatalyst powder and 10 mL of triethanolamine were added into the quartz reactor, which contained 90 mL of deionized water, and were sonicated for 1 min. Before the photocatalytic experiments, the quartz reactor was degassed and maintained under vacuum conditions for 30 min by a vacuum pump, to remove the dissolved oxygen and ensure anaerobic conditions. Then, the suspensions were stirred and irradiated for 5 h. The amount of H₂ released was measured by gas chromatography (FULI-9790II, with high-purity argon as the carrier gas).



2.4 Photoelectrochemical measurements

The electrochemical tests were carried out on a CHI660C electrochemical analyzer (Chenhua Instrument, Inc.) in a standard three-electrode configuration with a Pt sheet (1 cm² surface area) used as the counter electrode, Ag/AgCl (in saturated KCl) as the reference electrode, fluorine-doped tin oxide (FTO, 2 cm² surface area) deposited with photocatalysts serving as a working electrode and 0.5 M Na₂SO₄ aqueous solution as the electrolyte. The working electrode was prepared as follows: 5 mg of photocatalyst powder was mixed with 0.5 mL of ethyl alcohol, then 0.3 mL of Nafion solution (5 wt%) was added into the mixture to make a slurry by ultrasonic dispersal for 10 min. The slurry was uniformly dropped onto the working electrode. After drying at room temperature, the working electrode was transferred to a laboratory tube furnace and kept at 100 °C for 30 min in ambient N₂. The photocurrent measurements were performed under the visible light irradiation of a 300 W Xe lamp with a UV-cutoff filter ($\lambda > 400$ nm) and a potential of -0.1 V vs. Ag/AgCl. The EIS experiments were conducted in the frequency range from 100 kHz to 0.01 Hz, with an amplitude voltage of 5 mV at the open circuit voltage. Mott-Schottky analysis was carried out with potentials ranging from -0.9 to 0.1 V vs. Ag/AgCl for UCN and -0.12 to 0.06 V vs. Ag/AgCl for MoS_x with an amplitude voltage of 5 mV.

3. Results and discussion

3.1 Characterization

The MoS_x/UCN composite was synthesized by a sonochemical method, as schematically illustrated in Fig. 1. The XRD patterns of primitive MoS_x and MoS₂ are shown in Fig. 2(a). Amorphous MoS_x has only one weak and broad distinct diffraction peak at 14.4°. This demonstrates that the MoS_x synthesized in this study has a very low crystallinity/amorphous structure, which is different from other MoS₂ synthesized by a hydrothermal method.^{33,34} This is probably because H₄SiMo₁₂O₄₀ is involved in the reaction.³⁵ After calcination, amorphous MoS_x can transform into crystallized MoS₂. The XRD pattern of MoS₂ has three distinct diffraction peaks at 14.4°, 32.9° and 58.2° corresponding to the (002), (100) and (110) planes of 2H-MoS₂ (JCPDS no. 87-1526, molybdenum disulfide 2H), respectively. Fig. 2(b) shows XRD patterns of g-C₃N₄ nanosheets and ultrathin g-C₃N₄ with different contents of MoS_x. For all samples, two diffraction peaks occur at 13.1° and 27.8°, and are indexed to be the (100) and (002) planes of g-C₃N₄ (JCPDS 87-1526). The strongest peak of (002) at about 27.8° is attributed to the inter-planar stacking

of aromatic systems. The weak peak of (100) at 13.1° is induced by the in-plane structural stacking motif. The slight decrease in diffraction peak in the MoS_x/UCN samples indicates that ultrathin g-C₃N₄ with less stacking is produced due to the exfoliation of g-C₃N₄ nanosheets in ultrasonic processing. Nevertheless, diffraction peaks corresponding to MoS_x cannot be found in MoS_x/UCN samples. This may be due to the amorphous structure of MoS_x, and at the same time the broad diffraction peak of amorphous MoS_x is too weak compared with the diffraction peak of the UCN crystal. The existence of MoS_x in the MoS_x/UCN composite will be confirmed further by our other tests.

The morphology and microstructure of UCN, MoS_x and MoS_x/UCN samples were investigated by TEM characterization, and the results are shown in Fig. 3. Fig. 3(a) shows the two-dimensional morphology of the synthesized MoS_x. Fig. 3(b) presents the typical TEM image of MoS_x/UCN, which has a two-dimensional morphology similar to that of pure g-C₃N₄ nanosheets.³⁶ Fig. 3(c) is an HRTEM image of MoS_x, a disordered structure in the plane is observed, which indicates that MoS_x is amorphous. The lattice fringe of the MoS_x nanosheet edge confirms its layered structure with an interlayer distance of 0.62 nm. Fig. 3(d) shows HRTEM images for MoS_x/UCN (7 wt%). The layered structure of MoS_x can also be found, which demonstrates that the layered MoS_x co-catalysts are successfully loaded onto the surface of UCN. Fig. 3(e) is the HRTEM image of another area in MoS_x/UCN (7 wt%). The lattice spacing in Fig. 3(e) is 0.27 nm, which corresponds to the (100) facets of 2H-MoS₂. The lattice fringe of 2H-MoS₂ can be found because of the presence of partially crystalline MoS₂ in MoS_x.²⁹ The co-existence of MoS₂ and MoS_x is shown in Fig. S1.† The EDX pattern (Fig. 3(f)) of area I in Fig. 3(b) and elemental mapping (Fig. S2†) fully prove the combination of UCN and MoS_x from another perspective. The BET specific surface area and pore volume of the samples are listed in Table S2.† The BET specific surface area of UCN and MoS_x/UCN are 89.29 and 72.56 m² g⁻¹, respectively. The pore volumes of UCN and MoS_x/UCN are 0.31 and 0.18 m³ g⁻¹. After loading with MoS_x, the specific surface area and pore volume of MoS_x/UCN decreased, which may be caused by the MoS_x covering the pores of UCN. The results indicate that a MoS_x/UCN photocatalyst with a two-dimensional structure has been successfully prepared in this study.

X-ray photoelectron spectroscopy was carried out to check the chemical states of the elements in MoS_x/UCN photocatalysts. Fig. 4(a) indicates the presence of S, Mo, C, N and O in MoS_x/UCN samples. Fig. 4(b–e) show the high-resolution XPS spectra of C 1s, N 1s, Mo 3d and S 2p in the MoS_x/UCN samples. The C 1s spectrum shown in Fig. 4(b) can be deconvoluted into three peaks at 284.6, 285.9 and 288.6 eV, which are ascribed to the sp²-bonded aromatic structure (N–C=N), graphitic carbon (C–C), and residual C–O, respectively.^{37,38} As shown in Fig. 4(c), N 1s can be deconvoluted into three peaks at around 398.7 eV, 400.2 eV and 401.4 eV. The main peak at 398.7 eV can be assigned to sp² hybridized aromatic N bonded to carbon atoms (C=N–C).³⁹ In addition, the peaks at 400.2 eV and 401.4 eV are ascribed to the tertiary N bonded to carbon atoms in the form of N–(C)₃ and amino functional groups carrying

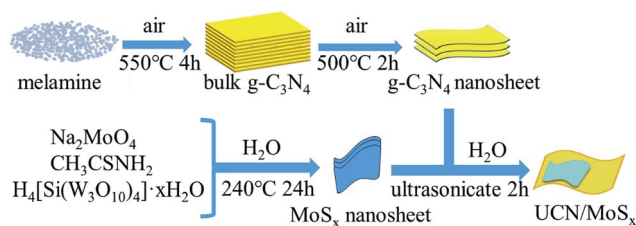


Fig. 1 Schematic diagram for the synthesis of MoS_x/UCN composite.



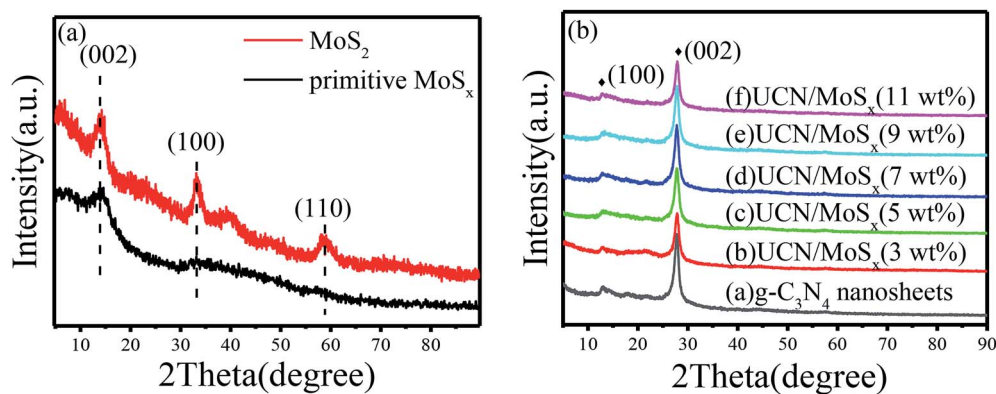


Fig. 2 XRD patterns of (a) primitive MoS_x and MoS₂, (b) g-C₃N₄ nanosheets and MoS_x/UCN composite.

hydrogen (C–N–H), respectively.⁴⁰ From Fig. 4(d), we can confirm that, except for two S 2s peaks at binding energies of 227.3 and 226.1 eV, other peaks can be readily assigned to the Mo 3d spectrum.⁴¹ The peaks at 232.6 and 229.5 eV are attributed to Mo 3d_{3/2} and Mo 3d_{5/2}, which are binding energies for

Mo⁴⁺ species in amorphous molybdenum sulfide.⁴² The Mo 3d doublet at 229.8 and 233.3 eV may be attributed to Mo⁴⁺ or Mo⁵⁺ chemical state as in molybdenum oxysulfide MoS_xO_y, which can be surmised to be the presence of Mo=O defects in MoS_x/UCN.^{43,44} Recently, Tran P. D. and colleagues found that Mo=O

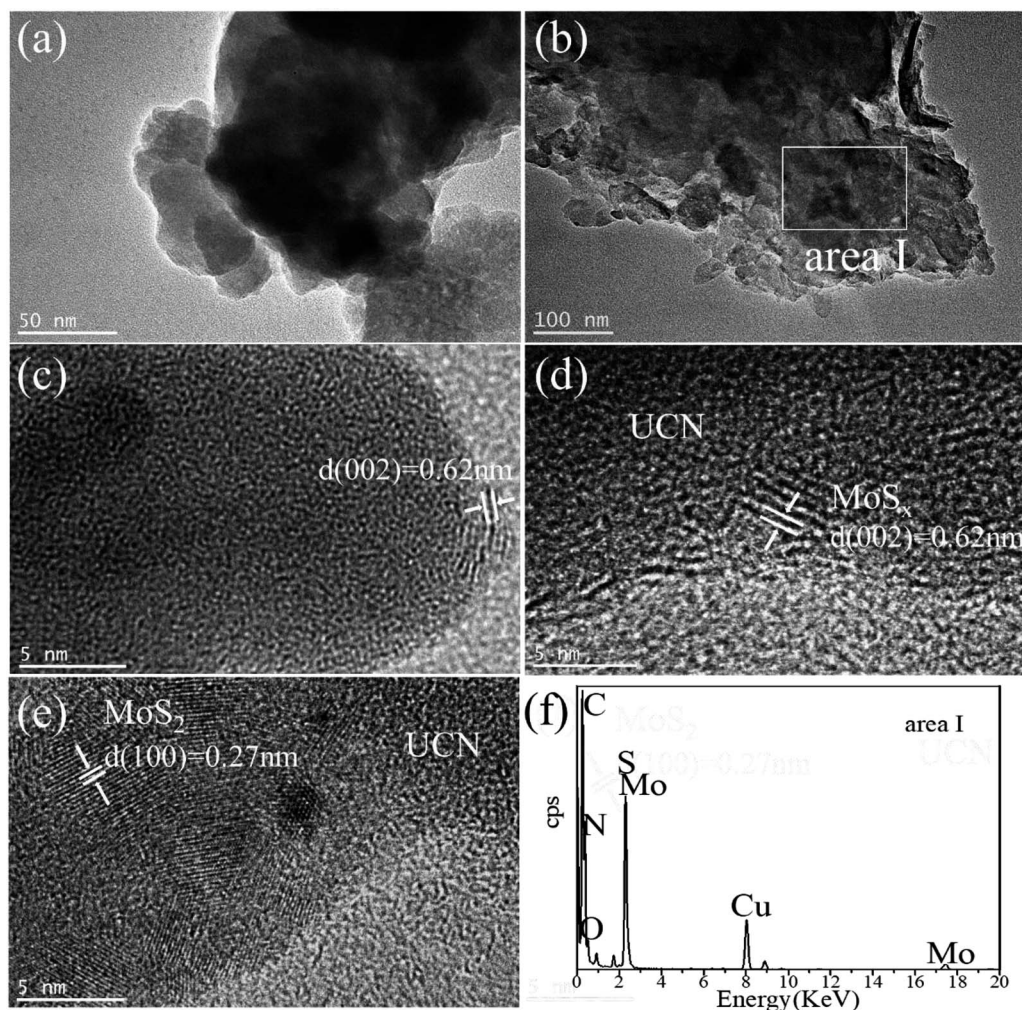


Fig. 3 TEM images of (a) MoS_x and (b) UCN; HRTEM images of (c) MoS_x and (d and e) MoS_x/UCN; EDX pattern of (f) area I.



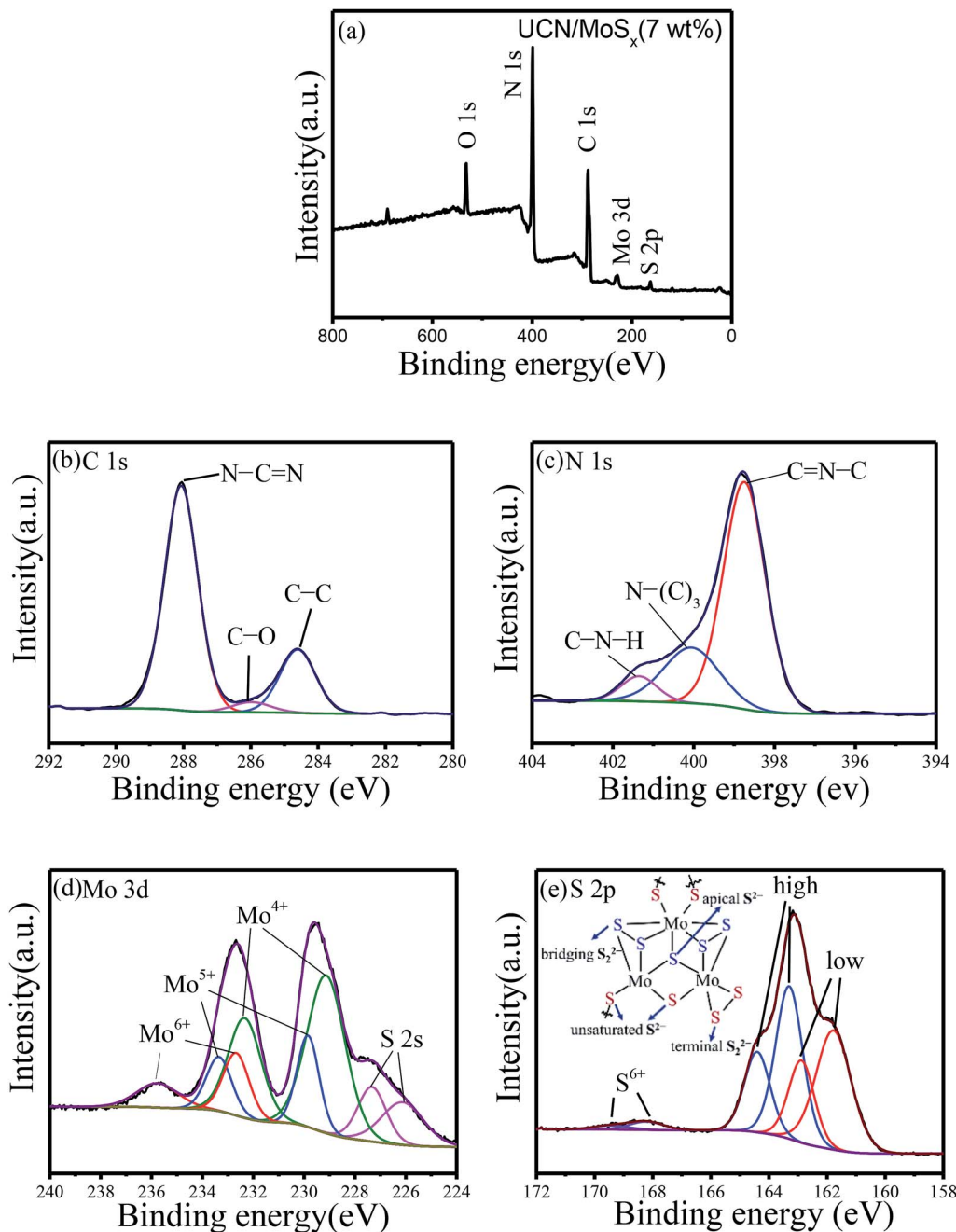


Fig. 4 Typical (a) XPS survey spectra and high-resolution XPS spectra of (b) C 1s, (c) N 1s, (d) Mo 3d and (e) S 2p.

defects will translate into Mo-bound hydride species which can act as the active species for H_2 -evolution.⁴⁵ The doublet for Mo 3d at 233.1 and 236.1 eV suggests the presence of Mo^{6+} , indicating the formation of small amounts of surface oxide species.²³ The high-resolution XPS spectra of S 2p suggest the existence of both disulfide S_2^{2-} and monosulfide S^{2-} , as shown in Fig. 4(e).^{46,47} And the higher binding energy signals (at 163.1 and 164.2 eV) are assigned to apical S^{2-} or bridging S_2^{2-} , while the lower higher binding energy signals (at 161.3 and 162.6 eV) are assigned to unsaturated S^{2-} or terminal S_2^{2-} .^{31,48} The S atoms with high binding energy have better catalytic performance for the catalyst than those with low binding energy.⁴⁹ The

percentage of S atoms with high binding energy in MoS_x is about 50%, which is conducive to the MoS_x co-catalyst displaying a superior catalytic performance to crystalline MoS_2 . The peaks at 168.2 and 169.4 eV could be attributed to S^{6+} or S^{4+} in surface oxide species. High-resolution XPS spectra of Mo 3d and S 2p in MoS_2 are shown in Fig. S3,† which shows a marked difference from MoS_x . Furthermore, the atomic ratio of S/Mo was estimated as 2.7 according to XPS, which is higher than that of highly crystalline MoS_2 .⁵⁰

The UV-vis diffuse reflectance spectra of UCN and MoS_x/UCN are shown in Fig. 5(a). Pure UCN shows an absorption edge at 415 nm, which is consistent with previous studies.^{50,51}



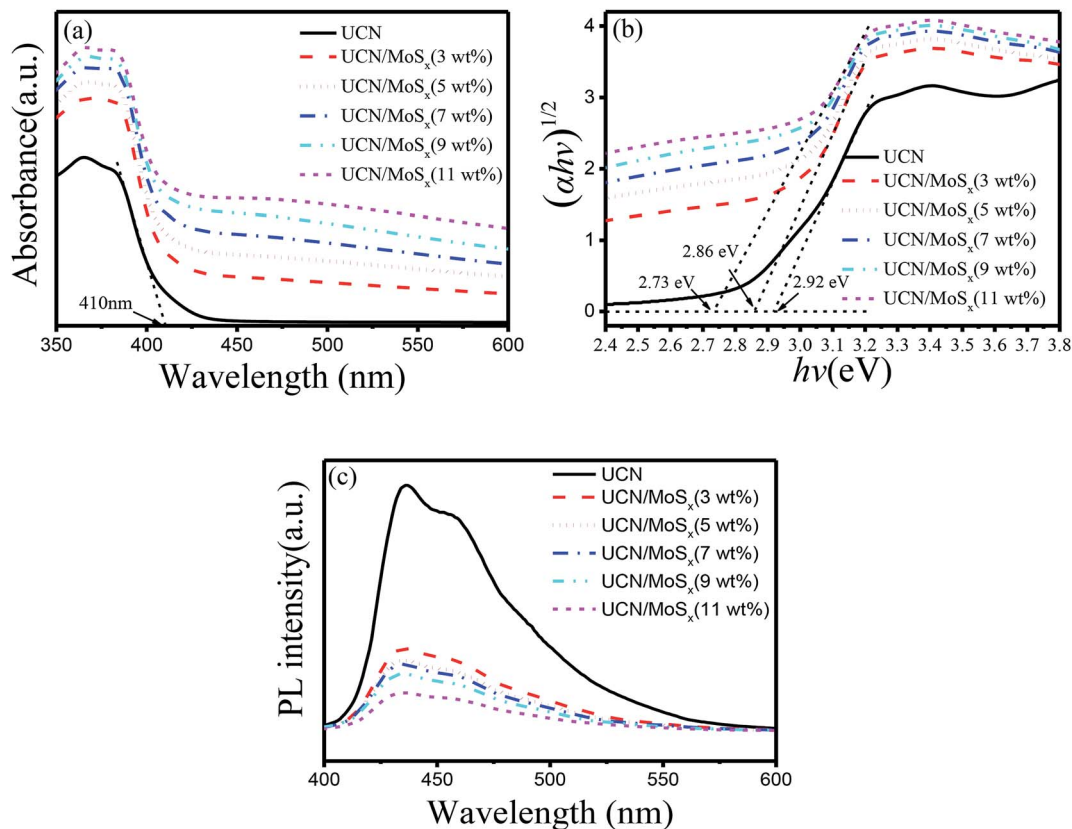


Fig. 5 Photoresponse properties of samples: (a) UV-vis diffuse reflectance spectra, (b) Tauc plots and E_g values and (c) photoluminescence spectra.

Compared with UCN, the MoS_x/UCN composite exhibits a redshift of the absorption edge and obviously enhanced light absorption in the visible region. The magnitude of the redshift and the enhancement in light absorption gradually increase with increasing MoS_x content. At the same time, the color of the prepared samples also changes from light yellow to dark gray. Fig. 5(b) shows that the E_g value of UCN is 2.92 eV and the E_g value of MoS_x/UCN (3–11 wt%) is 2.86–2.73 eV. In short, MoS_x expands the visible light-harvesting ability and narrows the E_g value of the UCN photocatalyst. This lets the photocatalyst absorb more photons, thereby enhancing its photocatalytic properties. In addition, as shown in Fig. 5(c), the photo-luminescence emission of UCN is remarkably quenched after MoS_x has been loaded onto it. And the intensity of the PL emission peaks for the MoS_x/UCN composite gradually decreases with an increase in MoS_x content, which means that MoS_x loading on the UCN surface can greatly suppress the rapid recombination of photogenerated electron-hole pairs in UCN. The results of UV-vis and PL measurements further prove that MoS_x has been successfully deposited onto the surface of UCN, and 2D MoS_x/UCN heterojunctions have been constructed.

3.2 Photocatalytic H_2 -evolution activity

The photocatalytic H_2 -evolution performance of pure UCN, MoS_2/UCN (7 wt%) and UCN/MoS_x with different percentages

of MoS_x were evaluated under visible light irradiation ($\lambda > 400$ nm) using triethanolamine as a sacrificial reagent. Fig. 6(a) shows the H_2 -evolution rate by pure UCN and the MoS_x/UCN composite. The linearly increasing rate of H_2 -evolution clearly shows that the MoS_x/UCN composite photocatalysts exhibit very good photostability within five hours. Further comparison of the average photocatalytic H_2 -evolution rates of various photocatalysts is shown in Fig. 6(b). It can be seen that the H_2 -evolution rate of pure UCN is only $23 \mu\text{mol g}^{-1} \text{h}^{-1}$, due to its fast recombination of electron-hole pairs. After 3–7 wt% MoS_x have combined with UCN, the average hydrogen-evolution rates of MoS_x/UCN increase to an optimal value of $1586 \mu\text{mol g}^{-1} \text{h}^{-1}$. With a further increase in MoS_x from 7 wt% to 11 wt%, the average H_2 -evolution rate begins to decrease from $1586 \mu\text{mol g}^{-1} \text{h}^{-1}$ to $994 \mu\text{mol g}^{-1} \text{h}^{-1}$. We also found that the photocatalytic activity of MoS_2/UCN is much lower than that of MoS_x/UCN , which indicates that as a co-catalyst MoS_x is much better than crystal MoS_2 . It is obvious that the photocatalytic activity of the binary composite photocatalysts is much higher than that of the pure UCN photocatalyst. The most outstanding H_2 -evolution rate of the MoS_x/UCN (7 wt%) photocatalyst is about 70 times that of pure UCN. In addition, MoS_x itself does not show such high photocatalytic activity for hydrogen evolution under visible light irradiation in the test, which confirms that MoS_x only plays the role of co-catalyst in MoS_x/UCN . In order to investigate the photocatalytic stability



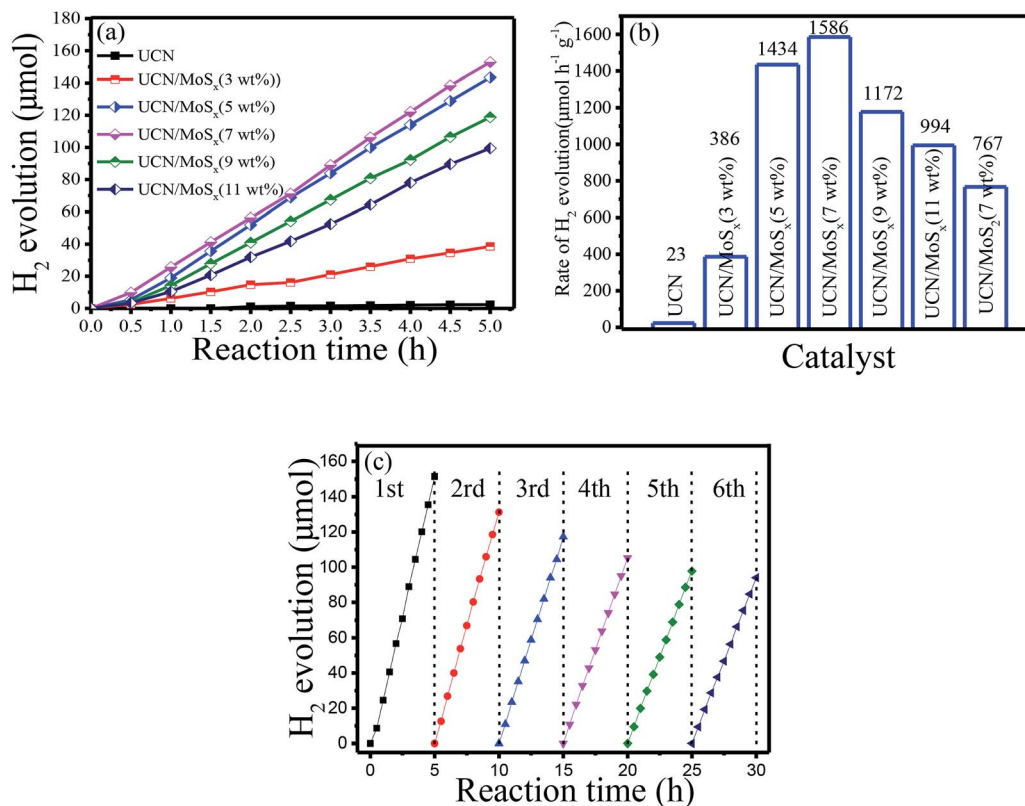


Fig. 6 Photocatalytic H₂-evolution properties of samples: (a) the photocatalytic H₂-evolution activities, (b) the photocatalytic H₂-evolution rate and (c) cycle run tests for photocatalytic H₂-evolution on MoS_x/UCN (7 wt%).

of the MoS_x/UCN composite, cycle running tests were conducted under visible light irradiation and are shown in Fig. 6(c). The activity of the photocatalyst decreases quickly with the first three cycle tests. This phenomenon could be induced by the degradation of MoS_x. After the fourth cycle, the H₂-evolution volume remains at about 62% of the H₂ volume produced with the first run test. From these results, deposition of MoS_x on the surface of UCN plays a decisive role in enhancing the photocatalytic activity of H₂-evolution. As mentioned, on the one hand, MoS_x can improve the electron-hole separation efficiency as well as the visible light response of UCN; on the other hand, MoS_x can provide a large number

of active sites for H₂-evolution. However, excessive MoS_x above 7 wt% may lead to a lower H₂-evolution rate, which can be caused by the shielding effects of MoS_x for both the surface active sites and the visible light response of UCN.^{52,53} For comparison, the results of our research work and other reports are listed in Table S1.† The MoS_x/UCN samples in our research work show much better photocatalytic activity for H₂-evolution compared with most of the other research.

3.3 Photoelectrochemical measurements

To further confirm the elevated separation efficiency of photo-generated carriers and the transfer rate of charge carriers for

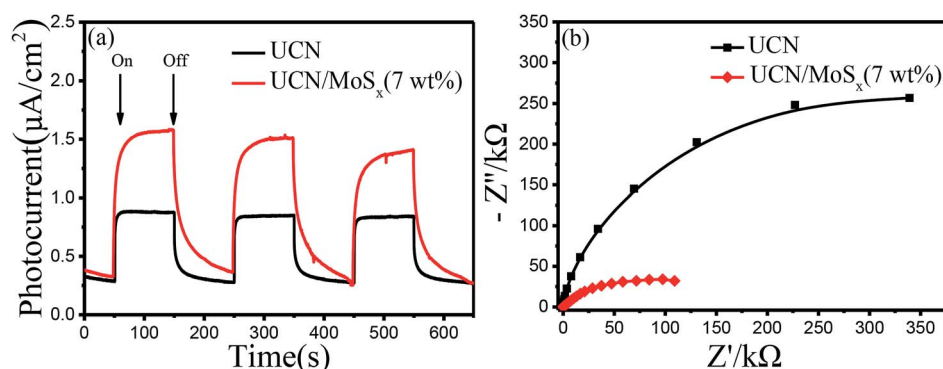


Fig. 7 (a) Photocurrent responses and (b) electrochemical impedance spectroscopy of samples.



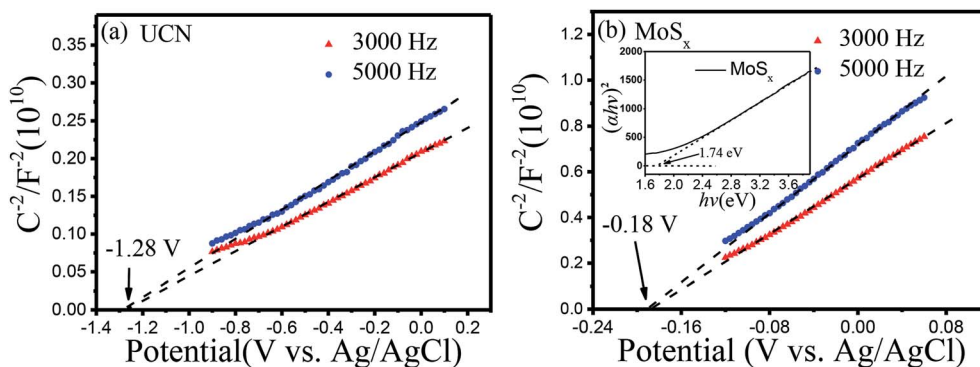


Fig. 8 Mott-Schottky plots of samples: (a) UCN, (b) MoS_x.

photocatalysts, the transient photocurrent response (*I*-*t* curves) and the electrochemical impedance spectroscopy (EIS) of pure UCN and the MoS_x/UCN (7 wt%) composite photocatalysts performed by a typical three-electrode system are shown in Fig. 7. The photocurrent of the MoS_x/UCN (7 wt%) under visible light is almost twice that of UCN, as shown in Fig. 7(a). The higher photocurrent of MoS_x/UCN (7 wt%) can be ascribed to the introduction of MoS_x, which increases the separation of electron-hole pairs. As shown in Fig. 7(b), it is obvious that MoS_x/UCN (7 wt%) exhibits a smaller arc size compared to that of pure UCN, suggesting that the addition of MoS_x can accelerate the electron transfer through the intimate interfaces between UCN and MoS_x.

In order to study the energy band structure of the semiconductor photocatalysts, a Mott-Schottky study was carried out. From the Mott-Schottky plots (Fig. 8), the conduction band potential of UCN and MoS_x can be estimated.⁵⁴ The conduction bands (CBs) are determined to be -1.28 V and -0.18 V (*vs.* E_{Ag/AgCl}, pH = 7) for UCN and MoS_x, respectively. The potentials are converted to the NHE scale according to the equation:^{55,56}

$$E_{\text{NHE}} = E_{\text{Ag/AgCl}} + E_{\text{Ag/AgCl}}^0 (0.198 \text{ V})$$

The recalculated valence band (VB) potentials of UCN and MoS_x are -1.08 and 0.02 V (*vs.* E_{NHE}, pH = 7), respectively.

Following the equation $E_{\text{VB}} = E_{\text{CB}} + E_{\text{g}}/e$, the valence band potentials for UCN and MoS_x are 1.84 and 1.76 V (*vs.* NHE, pH = 7). The conduction band potential for pure UCN (-1.08 V *vs.* E_{NHE}, pH = 7) is more negative than the H⁺ reduction potential (-0.410 V *vs.* E_{NHE} at pH = 7), which is enough to reduce H⁺ and produce H₂. The conduction band potential of pure MoS_x is more positive than the H⁺ reduction potential when the pH value equals 7, which results in an inertness of HER for MoS_x under visible irradiation. The result is consistent with the studies of previous reports.⁵⁷

3.4 Photocatalytic mechanism

Based on the experimental results, a possible mechanism is proposed to explain the enhanced H₂-production activity of MoS_x/UCN. As shown in Fig. 9, the conduction band position for UCN is higher than that of MoS_x. Under visible light irradiation, UCN can capture the photons, and electrons in the valence band will transfer to the conduction band to form electron-hole pairs. After that, the electrons may move to the surface of UCN to reduce H⁺ and produce H₂ by sufficient overpotential. But pure UCN shows poor H₂-evolution performance due to the lack of active species for H₂ production. After deposition of MoS_x, the electrons excited on the surface of UCN can transfer to MoS_x, because the conduction band of MoS_x is more negative than that of UCN. First, the fast recombination of electron-hole

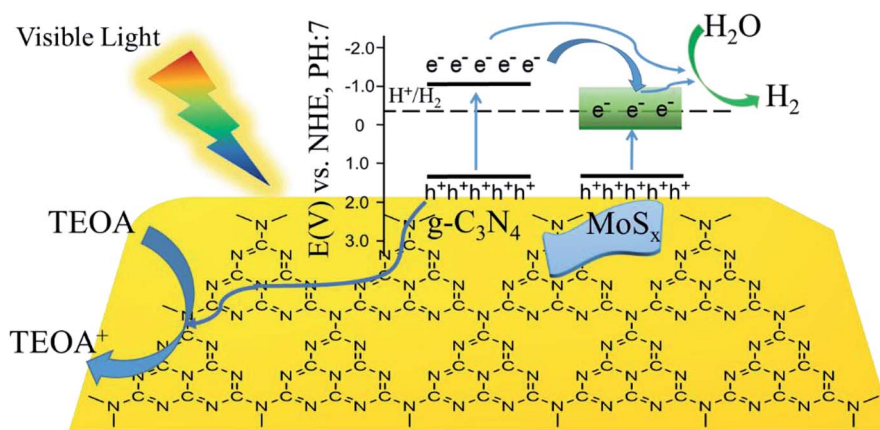


Fig. 9 Schematic illustration of the proposed principle of photocatalytic H₂-evolution by MoS_x/UCN under visible light irradiation.



pairs in UCN can be suppressed, and, second, the rapid consumption of electrons on the MoS_x surface may lead to the production of more H₂. Meanwhile, the holes in the VB of UCN are consumed by the triethanolamine sacrificial reagent. Furthermore, compared to MoS₂, MoS_x has a larger number of unsaturated S-atoms and unsaturated Mo-atoms. Unsaturated S-atoms and unsaturated Mo-atoms can absorb H⁺ to form S-H and Mo-H bonds,^{45,49} which can work as active species for H₂ production, so MoS_x shows better co-catalytic performance than MoS₂.

4. Conclusion

In conclusion, a facile sonochemical approach was used to fabricate an MoS_x/UCN composite photocatalyst for superior photocatalytic H₂ production. The results demonstrate that amorphous MoS_x loaded on UCN plays a co-catalyst role, the range of visible light absorption has been expanded, a large number of active species has been offered, and the separation and transportation of charge carriers have been promoted, leading to a tremendous improvement in the photocatalytic activity of UCN under visible light irradiation. For MoS_x/UCN, the optimal content of MoS_x is 7 wt%, in which the rate of H₂-evolution can reach a maximum value of 1586 μmol g⁻¹ h⁻¹. The present work successfully provides a novel g-C₃N₄ and amorphous MoS_x binary photocatalyst with excellent photocatalytic hydrogen-evolution performance.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work was financially supported by the National Natural Science Foundation of China (51770000), Natural Science Foundation of Anhui Province (1608085QE000), the Fundamental Research Funds for the Central Universities (JZ2017HG7B000), Guangxi Key Laboratory for Electrochemical Energy Materials, and the 111 Project "New Materials and Technology for Clean Energy" (B18018).

References

- L. M. Dai, D. W. Chang, J. B. Baek and W. Lu, *Small*, 2012, **8**, 1130–1166.
- S. Wei, H. He, Y. Cheng, C. Yang, G. Zeng, L. Kang, H. Qian and C. Zhu, *Fuel*, 2017, **200**, 11–21.
- E. Baran and B. Yazici, *Int. J. Hydrogen Energy*, 2016, **41**, 2498–2511.
- K. Tedsree, T. Li, S. Jones, C. W. A. Chan, K. M. K. Yu, P. A. J. Bagot, E. A. Marquis, G. D. W. Smith and S. C. E. Tsang, *Nat. Nanotechnol.*, 2011, **6**, 302–307.
- M. Z. Ge, J. S. Cai, J. Icozzia, C. Y. Cao, J. Y. Huang, X. N. Zhang, J. L. Shen, S. C. Wang, S. N. Zhang, K. Q. Zhang, Y. K. Lai and Z. Q. Lin, *Int. J. Hydrogen Energy*, 2017, **42**, 8418–8449.
- W. Zhang, Y. B. Wang, Z. Wang, Z. Y. Zhong and R. Xu, *Chem. Commun.*, 2010, **46**, 7631–7633.
- T. Simon, N. Bouchonville, M. J. Berr, A. Vaneski, A. Adrovic, D. Volbers, R. Wyrwich, M. Doblinger, A. S. Susha, A. L. Rogach, F. Jackel, J. K. Stolarczyk and J. Feldmann, *Nat. Mater.*, 2014, **13**, 1013–1018.
- Q. Xiang, J. Yu and M. Jaroniec, *Nanoscale*, 2011, **3**, 3670–3678.
- S. C. Yan, Z. S. Li and Z. G. Zou, *Langmuir*, 2009, **25**, 10397–10401.
- S. C. Yan, Z. S. Li and Z. G. Zou, *Langmuir*, 2010, **26**, 3894–3901.
- X. Wang, K. Maeda, A. Thomas, K. Takanabe, G. Xin, J. M. Carlsson, K. Domen and M. Antonietti, *Nat. Mater.*, 2009, **8**, 76–80.
- Y. W. Zhang, J. H. Liu, G. Wu and W. Chen, *Nanoscale*, 2012, **4**, 5300–5303.
- B. Chai, T. Peng, J. Mao, K. Li and L. Zan, *Phys. Chem. Chem. Phys.*, 2012, **14**, 16745–16752.
- Q. Han, B. Wang, J. Gao, Z. Cheng, Y. Zhao, Z. Zhang and L. Qu, *ACS Nano*, 2016, **10**, 2745–2751.
- X. Wang, K. Maeda, X. Chen, K. Takanabe, K. Domen, Y. Hou, X. Fu and M. Antonietti, *J. Am. Chem. Soc.*, 2009, **131**, 1680.
- J. Zhang, M. Zhang, C. Yang and X. Wang, *Adv. Mater.*, 2014, **26**, 4121–4126.
- S. Hu, F. Li, Z. Fan, F. Wang, Y. Zhao and Z. Lv, *Dalton Trans.*, 2015, **44**, 1084–1092.
- G. Liu, P. Niu, C. Sun, S. C. Smith, Z. Chen, G. Q. Lu and H.-M. Cheng, *J. Am. Chem. Soc.*, 2010, **132**, 11642–11648.
- Z. Zhu, Z. Lu, D. Wang, X. Tang, Y. Yan, W. Shi, Y. Wang, N. Gao, X. Yao and H. Dong, *Appl. Catal., B*, 2016, **182**, 115–122.
- O. Elbanna, M. Fujitsuka and T. Majima, *ACS Appl. Mater. Interfaces*, 2017, **9**, 34844–34854.
- J. D. Benck, T. R. Hellstern, J. Kibsgaard, P. Chakthranont and T. F. Jaramillo, *ACS Catal.*, 2014, **4**, 3957–3971.
- S. R. Kadam, D. J. Late, R. P. Panmand, M. V. Kulkarni, L. K. Nikam, S. W. Gosavi, C. J. Park and B. B. Kale, *J. Mater. Chem. A*, 2015, **3**, 21233–21243.
- Y. Hou, A. B. Laursen, J. Zhang, G. Zhang, Y. Zhu, X. Wang, S. Dahl and I. Chorkendorff, *Angew. Chem., Int. Ed.*, 2013, **52**, 3621–3625.
- L. Ge, C. Han, X. Xiao and L. Guo, *Int. J. Hydrogen Energy*, 2013, **38**, 6960–6969.
- B. Han, S. Q. Liu, N. Zhang, Y. J. Xu and Z. R. Tang, *Appl. Catal., B*, 2017, **202**, 298–304.
- M. Latorre-Sanchez, I. Esteve-Adell, A. Primo and H. Garcia, *Carbon*, 2015, **81**, 587–596.
- Y. J. Yuan, Z.-J. Ye, H.-W. Lu, B. Hu, Y. H. Li, D.-Q. Chen, J.-S. Zhong, Z.-T. Yu and Z.-G. Zou, *ACS Catal.*, 2016, **6**, 532–541.
- Q. Liu, Q. Fang, W. S. Chu, Y. Y. Wan, X. L. Li, W. Y. Xu, M. Habib, S. Tao, Y. Zhou, D. B. Liu, T. Xiang, A. Khalil, X. J. Wu, M. Chhowalla, P. M. Ajayan and L. Song, *Chem. Mater.*, 2017, **29**, 4738–4744.



- 29 S. C. Lee, J. D. Benck, C. Tsai, J. Park, A. L. Koh, F. Abild-Pedersen, T. F. Jaramillo and R. Sinclair, *ACS Nano*, 2016, **10**, 624–632.
- 30 H. Yu, P. Xiao, P. Wang and J. Yu, *Appl. Catal., B*, 2016, **193**, 217–225.
- 31 X. Hai, W. Zhou, K. Chang, H. Pang, H. Liu, L. Shi, F. Ichihara and J. Ye, *J. Mater. Chem. A*, 2017, **5**, 8591–8598.
- 32 P. Niu, L. Zhang, G. Liu and H.-M. Cheng, *Adv. Funct. Mater.*, 2012, **22**, 4763–4770.
- 33 Y. Hou, Z. Wen, S. Cui, X. Guo and J. Chen, *Adv. Mater.*, 2013, **25**, 6291–6297.
- 34 W. Zhou, Z. Yin, Y. Du, X. Huang, Z. Zeng, Z. Fan, H. Liu, J. Wang and H. Zhang, *Small*, 2013, **9**, 140–147.
- 35 H. Lin, X. Chen, H. Li, M. Yang and Y. Qi, *Mater. Lett.*, 2010, **64**, 1748–1750.
- 36 Q. Liu, T. Chen, Y. Guo, Z. Zhang and X. Fang, *Appl. Catal., B*, 2016, **193**, 248–258.
- 37 J. Mao, T. Peng, X. Zhang, K. Li, L. Ye and L. Zan, *Catal. Sci. Technol.*, 2013, **3**, 1253–1260.
- 38 Y. P. Liu, P. Chen, Y. Chen, H. D. Lu, J. X. Wang, Z. S. Yang, Z. H. Lu, M. Li and L. Fang, *RSC Adv.*, 2016, **6**, 10802–10809.
- 39 X. D. Zhang, X. Xie, H. Wang, J. J. Zhang, B. C. Pan and Y. Xie, *J. Am. Chem. Soc.*, 2013, **135**, 18–21.
- 40 X. Yuan, C. Zhou, Y. Jin, Q. Jing, Y. Yang, X. Shen, Q. Tang, Y. Mu and A.-K. Du, *J. Colloid Interface Sci.*, 2016, **468**, 211–219.
- 41 H. Vrubel and X. Hu, *ACS Catal.*, 2013, **3**, 2002–2011.
- 42 A. Y. Lu, X. Yang, C. C. Tseng, S. Min, S. H. Lin, C. L. Hsu, H. Li, H. Idriss, J.-L. Kuo, K.-W. Huang and L.-J. Li, *Small*, 2016, **12**, 5530–5537.
- 43 P. Kien-Cuong, D. S. McPhail, A. T. S. Wee and D. H. C. Chua, *RSC Adv.*, 2017, **7**, 6856–6864.
- 44 Y. Chen, P. D. Tran, P. Boix, Y. Ren, S. Y. Chiam, Z. Li, K. Fu, L. H. Wong and J. Barber, *ACS Nano*, 2015, **9**, 3829–3836.
- 45 P. D. Tran, T. V. Tran, M. Orio, S. Torelli, Q. D. Truong, K. Nayuki, Y. Sasaki, S. Y. Chiam, R. Yi, I. Honma, J. Barber and V. Artero, *Nat. Mater.*, 2016, **15**, 640.
- 46 D. Merki, S. Fierro, H. Vrubel and X. Hu, *Chem. Sci.*, 2011, **2**, 1262–1267.
- 47 L. Ye, D. Wang and S. Chen, *ACS Appl. Mater. Interfaces*, 2016, **8**, 5280–5289.
- 48 J. Kibsgaard, T. F. Jaramillo and F. Besenbacher, *Nat. Chem.*, 2014, **6**, 248–253.
- 49 L. R. L. Ting, Y. Deng, L. Ma, Y.-J. Zhang, A. A. Peterson and B. S. Yeo, *ACS Catal.*, 2016, **6**, 861–867.
- 50 K. K. Liu, W. J. Zhang, Y. H. Lee, Y. C. Lin, M. T. Chang, C. Y. Su, C. S. Chang, H. Li, Y. M. Shi, H. Zhang, C. S. Lai and L. J. Li, *Nano Lett.*, 2012, **12**, 1538–1544.
- 51 F. Dong, Y. Li, Z. Wang and W.-K. Ho, *Appl. Surf. Sci.*, 2015, **358**, 393–403.
- 52 Y. Min, G. He, Q. Xu and Y. Chen, *J. Mater. Chem. A*, 2014, **2**, 2578–2584.
- 53 K. Chang, Z. Mei, T. Wang, Q. Kang, S. Ouyang and J. Ye, *ACS Nano*, 2014, **8**, 7078–7087.
- 54 T. Giannakopoulou, I. Papailias, N. Todorova, N. Boukos, Y. Liu, J. Yu and C. Trapalis, *Chem. Eng. J.*, 2017, **310**, 571–580.
- 55 S. Hong, D. P. Kumar, D. A. Reddy, J. Choi and T. K. Kim, *Appl. Surf. Sci.*, 2017, **396**, 421–429.
- 56 M. Li, L. Zhang, X. Fan, M. Wu, Y. Du, M. Wang, Q. Kong, L. Zhang and J. Shi, *Appl. Catal., B*, 2016, **190**, 36–43.
- 57 Q. Xiang, B. Cheng and J. Yu, *Angew. Chem., Int. Ed.*, 2015, **54**, 11350–11366.

