



Cite this: *RSC Adv.*, 2019, 9, 626

Photo-enhanced gas sensing of SnS₂ with nanoscale defects†

Wen-Jie Yan,^a Deng-Yun Chen,^a Huei-Ru Fuh,^{*bc} Ying-Lan Li,^a Duan Zhang,^a Huajun Liu,^d Gang Wu,^a Lei Zhang,^{id e} Xiangkui Ren,^{id e} Jiung Cho,^f Miri Choi,^g Byong Sun Chun,^h Cormac Ó. Coileáin,^a Hong-Jun Xu,^a Zhi Wang,^a Zhaotan Jiang,^a Ching-Ray Chang^{*b} and Han-Chun Wu^{id *a}

Recently a SnS₂ based NO₂ gas sensor with a 30 ppb detection limit was demonstrated but this required high operation temperatures. Concurrently, SnS₂ grown by chemical vapor deposition is known to naturally contain nanoscale defects, which could be exploited. Here, we significantly enhance the performance of a NO₂ gas sensor based on SnS₂ with nanoscale defects by photon illumination, and a detection limit of 2.5 ppb is achieved at room temperature. Using a classical Langmuir model and density functional theory simulations, we show S vacancies work as additional adsorption sites with fast adsorption times, higher adsorption energies, and an order of magnitude higher resistance change compared with pristine SnS₂. More interestingly, when electron–hole pairs are excited by photon illumination, the average adsorption time first increases and then decreases with NO₂ concentration, while the average desorption time always decreases with NO₂ concentration. Our results give a deep understanding of photo-enhanced gas sensing of SnS₂ with nanoscale defects, and thus open an interesting window for the design of high performance gas sensing devices based on 2D materials.

Received 25th October 2018
 Accepted 20th December 2018

DOI: 10.1039/c8ra08857h

rsc.li/rsc-advances

Nitrogen dioxide (NO₂) is one of the most common hazardous gas pollutants produced as a by-product of industry.^{1–4} Notably, NO₂ is also a product yielded from the thermal decomposition of organic nitrates, such as TNT, which is a useful fingerprint for controlling these substances and combating international terrorism.⁵ An issue of global concern, serious respiratory diseases can be caused by chronic environmental exposure to as little as 53 ppb NO₂.^{1,4,6} Therefore, the research and development of gas sensors with high sensitivities is always of high priority.^{2,7–19} Recently, 2D materials have been used for gas sensing owing to their large surface-to-volume ratios.^{13,20} A wide variety of 2D materials, such as graphene,^{9,10,14} black

phosphorus,^{2,21} metal dichalcogenides,^{4,12,15,18,19,22–24} have been studied as high performance gas sensors.^{16,20,23,25,26}

As a member of the wide class 2D materials, tin disulfide (SnS₂) displays many intriguing properties in electronics,²⁷ photonics and optoelectronics,^{27–30} as well as gas monitoring.^{24,31–42} It was reported recently by Kalantar-zadeh *et al.* that a SnS₂-based NO₂ gas sensor with a sensitivity of 0.5 ppm could be improved to 30 ppb when heated to 120 °C.²⁴ However, an elevated operating temperature typically increases power consumption and thus a more intricate setup is required.^{3,10,16,20,24,25,43} Furthermore, the high operating temperature might pose the risk of ignition when detecting in flammable or explosive atmospheres which would limit sensor's application.^{44–46} Concurrently, it is known that 2D materials produced by chemical vapor deposition (CVD) usually contain nanoscale defects, such as vacancies, grain boundaries, *etc.* and in particular CVD grown SnS₂ is reported to naturally contain S vacancies.⁴⁷ It is well known that nanoscale defects can significantly affect gas sensing performance.⁴⁸ Thus it is important to investigate how S vacancies in SnS₂ affect the gas sensing performance. Furthermore, it should be also noted that compared with thermally assisted gas sensing, a photon assisted arrangement can not only improve the detection limit of gas sensor but also can accelerate the desorption ratio.^{15,19,49–51} It is reported that SnS₂ nanosheets exhibit excellent optoelectronics properties,^{27–30} making it feasible to improve the gas sensing performance of SnS₂ with photon illumination.

^aSchool of Physics, Beijing Institute of Technology, Beijing 100081, P. R. China. E-mail: wuhc@bit.edu.cn

^bDepartment of Physics, National Taiwan University, Taipei 106, Taiwan. E-mail: crchang@phys.ntu.edu.tw

^cDepartment of Chemical Engineering & Materials Science, Yuan Ze University, Taoyuan City 320, Taiwan. E-mail: hrfuh@saturn.yzu.edu.tw

^dInstitute of Plasma Physics, Chinese Academy of Sciences, Hefei 230031, P. R. China

^eSchool of Chemical Engineering and Technology, Tianjin University, Tianjin 300072, P. R. China

^fWestern Seoul Center, Korea Basic Science Institute, Seoul 03579, Republic of Korea

^gChuncheon Center, Korea Basic Science Institute, Chuncheon 24341, Republic of Korea

^hDivision of Industrial Metrology, Korea Research Institute of Standards and Science, Daejeon 305-340, Republic of Korea

† Electronic supplementary information (ESI) available. See DOI: 10.1039/c8ra08857h



In this work, we demonstrate that photon illumination can significantly enhance the performance of NO₂ gas sensors based on SnS₂, and an ultra-low concentration sensitivity of 2.5 ppb is achieved at room temperature. The results also show that gas sensitivity increases linearly with decreasing SnS₂ nanosheet thickness. In combination with a classical Langmuir model and density functional theory simulations, we show that S vacancies work as additional adsorption sites with fast adsorption times, higher adsorption energies, and greater resistance change compared with pristine SnS₂. Moreover, when electron–hole pairs are excited by photon illumination, the average adsorption time first increases and then decreases with NO₂ concentration. Considering safety, energy efficiency and sensitivity, SnS₂-based photo-illuminated gas sensors provide a possible way to realize the measurement of ultra-low concentrations of NO₂, or other hazardous gases, for environmental, industrial and domestic monitoring.

Fig. 1(a) shows a schematic of the chemical vapor deposition (CVD) arrangement for the SnS₂ nanosheet growth. A sulfur powder source was placed upstream in the furnace. Sn granules were smashed into pieces and were placed at the center of the furnace. Several pieces of a diced 300 nm SiO₂/Si (1 cm × 1 cm) wafer were placed downstream as substrates for SnS₂ growth. Further details of the SnS₂ growth can be found in the Experimental section. After the reaction, SnS₂ nanosheets were formed on the SiO₂/Si substrates. Fig. 1(b) shows a scanning electron microscopy (SEM) image of a SnS₂ flake grown on a SiO₂/Si substrate. The flakes typically have a semi-hexagonal structure with lateral dimensions ranging from 10 μm to 228 μm (Fig. S1†). Fig. 1(c) gives the structure of SnS₂ with the purple and yellow spheres representing Sn and S atoms, respectively.

Atomic force microscopy (AFM) was used to determine the thicknesses of the nanosheets. Fig. 2(a) shows the height profile of an ultrathin SnS₂ nanosheet with a thickness ≈ 6 nm (≈ 10 layers, 0.6 nm per layer),³⁰ as measured along the red dotted line in the inset image, (more examples in Fig. S2†). The SnS₂ was also characterized by Raman spectroscopy, as shown in

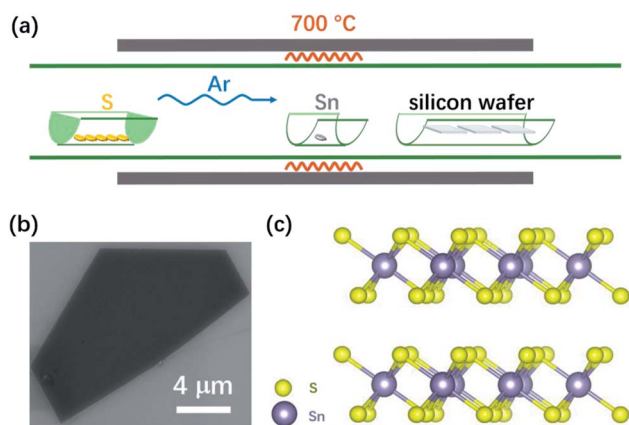


Fig. 1 (a) A schematic of CVD growth setup. (b) A SEM image of a semi-hexagonal nanosheet of SnS₂. (c) Schematic of SnS₂ structure with purple and yellow spheres representing Sn and S atoms, respectively.

Fig. 2(b). Only a single Raman peak (A_{1g} mode at 314 cm⁻¹) can be clearly observed for an 8 nm thick SnS₂ nanosheet, and an additional Raman peak, the E_g mode at 205 cm⁻¹, appears for a thicker 60 nm SnS₂ nanosheet, which is consistent with literature.^{28,52} The E_g mode is due to the scattering centers for in-plane scattering. When the thickness drops below 8 nm the intensity decreases and effectively becomes undetectable.³⁰ The crystal structure of few-layer SnS₂ was further characterized by high-resolution transmission electronic microscopy (HRTEM), indicating the single-crystalline structure of the SnS₂ (Fig. 2(c)). The measured width of the lattice fringes is ≈ 0.32 nm which corresponds to the (100) plane of SnS₂. Fig. 2(d) shows the selected-area-electron-diffraction (SAED) pattern, which displays six-fold symmetry diffraction spots, confirming the hexagonal single-crystalline nature of the flake. Additionally, Sn and S elemental mapping using Electron Energy Loss Spectroscopy (EELS), in Fig. 2(f), of the selected area shown in Fig. 2(e), indicates the atomic ratio of Sn : S ≈ 1 : 1.86, suggesting the presence of S vacancies in the SnS₂, which is consistent with earlier reports.⁴⁷

Before studying the photo-enhanced gas sensing properties of SnS₂, we first explored its optoelectronic properties. Fig. 3(a) is an optical image of a typical device prepared from a semi-hexagonal SnS₂ nanosheet on a SiO₂/Si substrate with 300 nm thick SiO₂. Fig. 3(b) shows the current *versus* voltage I_{DS}–V_{DS} (DS, Drain–Source) characteristics of the device with the gate voltage (V_G) varied from –20 V to 20 V in steps of 5 V and Fig. 3(c) plots the relationship between I_{DS} and V_G, with V_{DS} ranging from –10 V to +10 V in steps of 2.5 V indicating transfer characteristics of the device. From Fig. 3(b and c) it is apparent that our CVD grown SnS₂ nanosheets are n-type and the mobility is around 9.7 cm² V⁻¹ s⁻¹ which further supports the presence of the S vacancies.⁴⁷ Fig. 3(d and e) shows the I–V characteristics of the device at V_G = 0 V in the dark and under illumination by 532 nm and 650 nm lasers, in ambient conditions. It is found there is a notable change in the I_{DS} response going from dark to illuminated, and the enhancement of I_{DS} for the 532 nm laser is relatively larger than that observed for the 650 nm laser. Fig. 3(f) further shows the time resolved photo response of another device at V_{DS} = 5 V, where the light intensity of the 532 nm laser was 3.0 mW cm⁻². The responsivity (R_i) and external quantum efficiency (EQE) are calculated to be 31 A W⁻¹ and 7.7 × 10³%, respectively (details of this calculation can be found in ESI Formula (2)†).

Fig. 4(a) is an optical image of a typical SnS₂-based gas sensor fabricated by conventional UV-lithography. The distance between each of the contacts is approximately 5 μm. Fig. 4(b and c) show the dynamic-sensing responses without and with 405 nm laser irradiation, respectively, where the NO₂ gas concentration was varied from 6.5 ppb to 130 ppb. The sensor response is defined as $\Delta R/R_0 = (R_{\text{NO}_2} - R_{\text{N}_2})/R_{\text{N}_2}$, where R_{N₂} and R_{NO₂} are the resistances of the device when exposed to pure N₂ and a NO₂ gas mix, respectively. In the dark environment, the sensor has the first observable response at 78 ppb NO₂. Remarkably, by illuminating the sensor with a 405 nm laser a clear response is observed at 13 ppb NO₂. To confirm this, Fig. 4(d) plotted I_{DS}–V_{DS} curves of the same device under



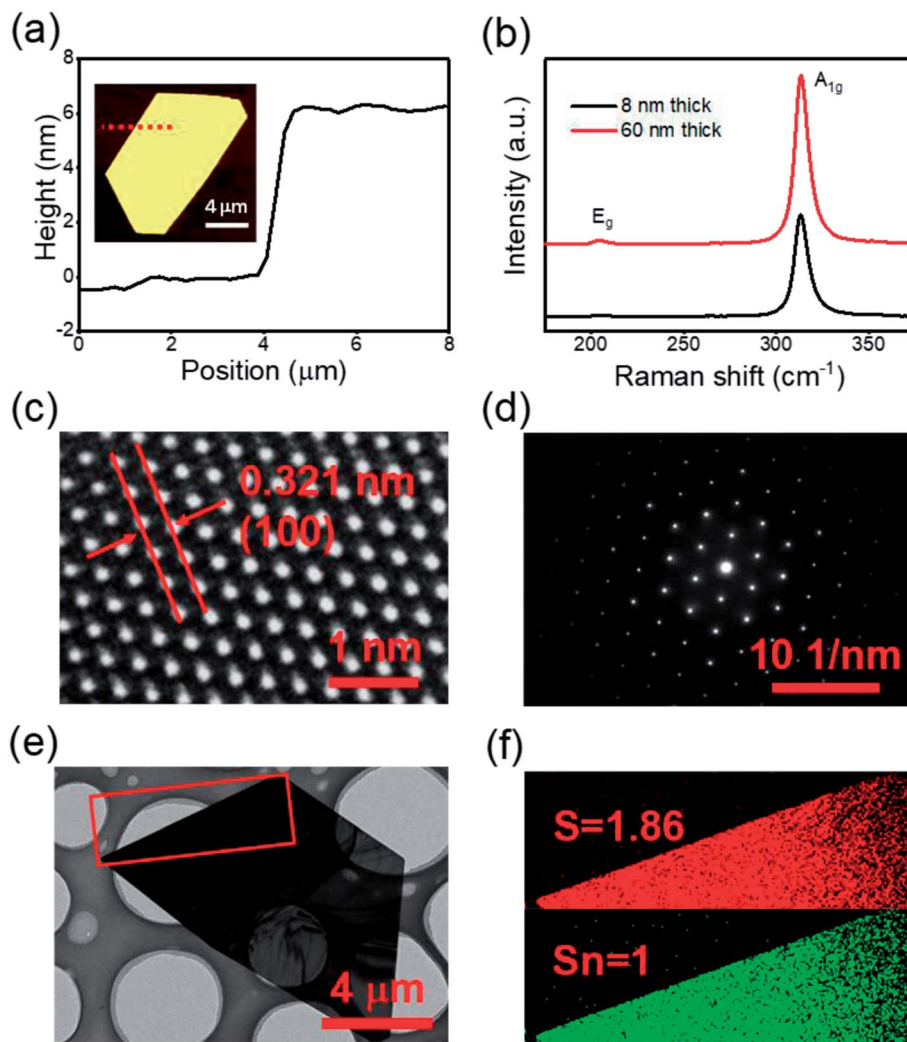


Fig. 2 Characterization of CVD produced single crystal SnS_2 nanosheet. (a) Height profiles of the dashed line shown in the inset AFM image. (b) Raman spectra of SnS_2 nanosheet. (c) High-resolution TEM image and corresponding (d) selected-area-electron-diffraction (SAED) pattern of a typical grown SnS_2 flake. (e) Low magnification bright field TEM image of a semi-hexagonal SnS_2 nanosheet. (f) EELS elemental mapping of the selected area in (e) with the result $\text{Sn} : \text{S} = 1 : 1.86$.

different NO_2 concentrations. It is apparent that in the dark environment, the $I_{\text{DS}}-V_{\text{DS}}$ curves for 0 ppb, 26 ppb and 52 ppb NO_2 merge together. In contrast with this, clear distinctions can be observed between 0 ppb, 26 ppb and 52 ppb NO_2 with 405 nm laser illumination. A similar effect was observed with 650 nm (26 ppb) and 785 nm (52 ppb) lasers (Fig. S3†). Thus, laser illumination can improve the gas sensing capabilities of such devices.

The stability of a sensor is critical for practical applications. Multi-cycle dynamic sensing responses of the SnS_2 -based sensor against to 13, 39 and 78 ppb NO_2 are used here to demonstrate the good repeatability and reproducibility of such SnS_2 -based sensors (Fig. 4(e)). Moreover, the response to varied concentrations of NO_2 with a relative humidity 64% is also showed in Fig. S5(d)†. These indicate the reliability of the illuminated SnS_2 -based sensor and that the presence of water does not degrade performance.

To probe the function of sensitivity *versus* the thickness of SnS_2 , we also tested more than 10 samples with different

thicknesses, ranging from 17 nm to 370 nm (Fig. S4–S9†). It is found that the sensitivity increases linearly as the thickness of the SnS_2 -nanosheets is reduced, as shown in Fig. 4(f). Remarkably, a significant response is unmistakably observed at 2.5 ppb NO_2 for a 22 nm thick SnS_2 based gas sensor (Fig. S8(d)†), suggesting that SnS_2 based gas sensor can achieve a NO_2 detection limit better than 2.5 ppb. To the best of our knowledge, this is the lowest detection limit for NO_2 achieved using SnS_2 devices to date, which is an improvement by a factor greater than ten on previous work.²⁴ Table 1 summarizes and compares the important features of SnS_2 -based NO_2 gas sensors from literature. Kalantar-zadeh *et al.* heated their SnS_2 flake sensor to 120 °C to achieve a 30 ppb detection limit.²⁴ Hao *et al.* heated a SnS_2 flower interspersed with SnO_2 to 100 °C to reach a 125 ppb detection limit.³¹ Ahn *et al.* built utilized vertical SnS_2 flake structures to detect NO_2 at 300 ppb, probably under its optimum conditions at 160 °C.³⁹ Li *et al.* heated SnS_2 their flake devices to 120 °C to attain a 600 ppb detection limit.⁴⁰ Wang *et al.* heated a SnS_2 - SnO_2 composite flake device to 80 °C for



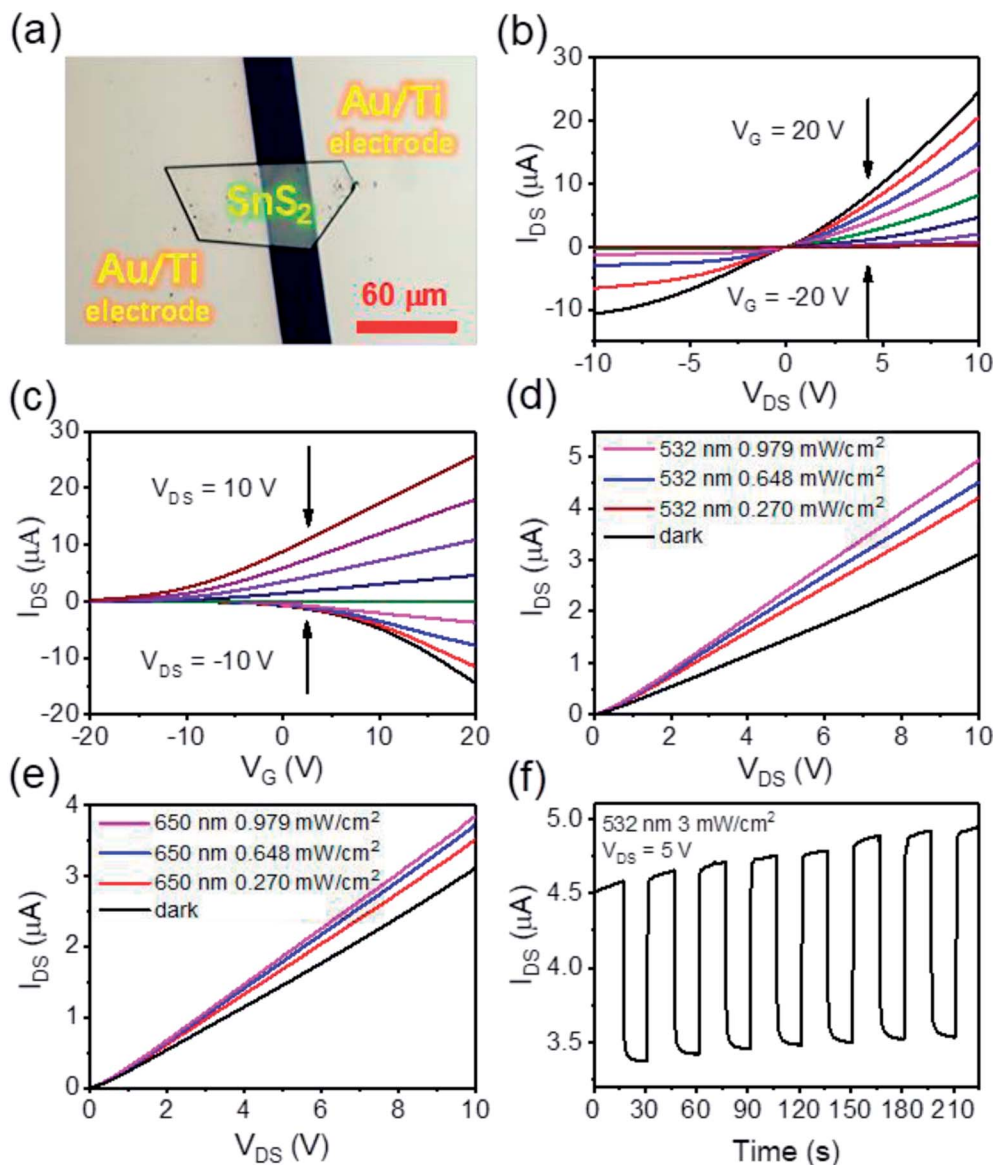


Fig. 3 (a) Optical microscope image of a SnS₂ device for the following measurement. (b) I_{DS} - V_{DS} curves for different V_G from -20 to 20 V in steps of 5 V. (c) I_{DS} - V_G curves for different V_{DS} from -10 to 10 V in steps of 2.5 V. (d) I_{DS} - V_{DS} curves of device under 532 nm laser illumination by different energy densities. (e) I_{DS} - V_{DS} curves of device under 650 nm laser illumination by different energy densities. (f) Time-resolved photon response of another SnS₂ device measured in vacuum ($V_{DS} = 5$ V, 532 nm laser 0.979 mW cm⁻²).

a 1000 ppb detection limit.⁴¹ While Xue *et al.* used SnS_{1.5}/Se_{0.5} flake devices to detect 1000 ppb NO₂ at room temperature.⁴²

To better illustrate the photo-enhanced gas sensing properties, we summarized the signal/noise (S/N) ratios of the $\Delta R/R_0$ from response in Fig. 4(b and c) into Fig. 5(a).⁵³⁻⁵⁷ It is clear that there are 3 distinct regions. At high concentrations of NO₂ the S/N is almost constant (dark and 785 nm). This means the surface adsorption is almost saturated and has reached its upper limit. Increasing gas concentration does not increase the signal. In the medium concentration range (classic region), S/N increases linearly with gas concentration. In the low NO₂ concentration regime (<52 ppb) no effective signal is detected for dark conditions. But a signal is detected with illumination, which is especially prominent for the short wavelength light, indicating extra energy is required to desorb NO₂ from surface of SnS₂.

We fitted the desorption (adsorption) sections of the dynamic-sensing response curves under different laser illuminations with the expression $y = y_0 + Ae^{-\frac{x}{\tau_D}}$ ($y = y_0 + Ae^{\frac{x}{\tau_A}}$), where τ_D (τ_A) is the average desorption (adsorption) time. The average desorption times and average adsorption times under various laser illuminations are summarized in Fig. 5(b) and (c) respectively. It is clear illumination not only enables ultra-high sensitivity but also results in faster desorption and better S/N ratios. Interestingly, with the 405 nm laser illumination, the adsorption time first increases and then decreases with gas concentration when the concentration reaches ~ 78 ppb. While for dark and 785 nm laser illumination, both the desorption and adsorption times decrease with increasing gas concentration.



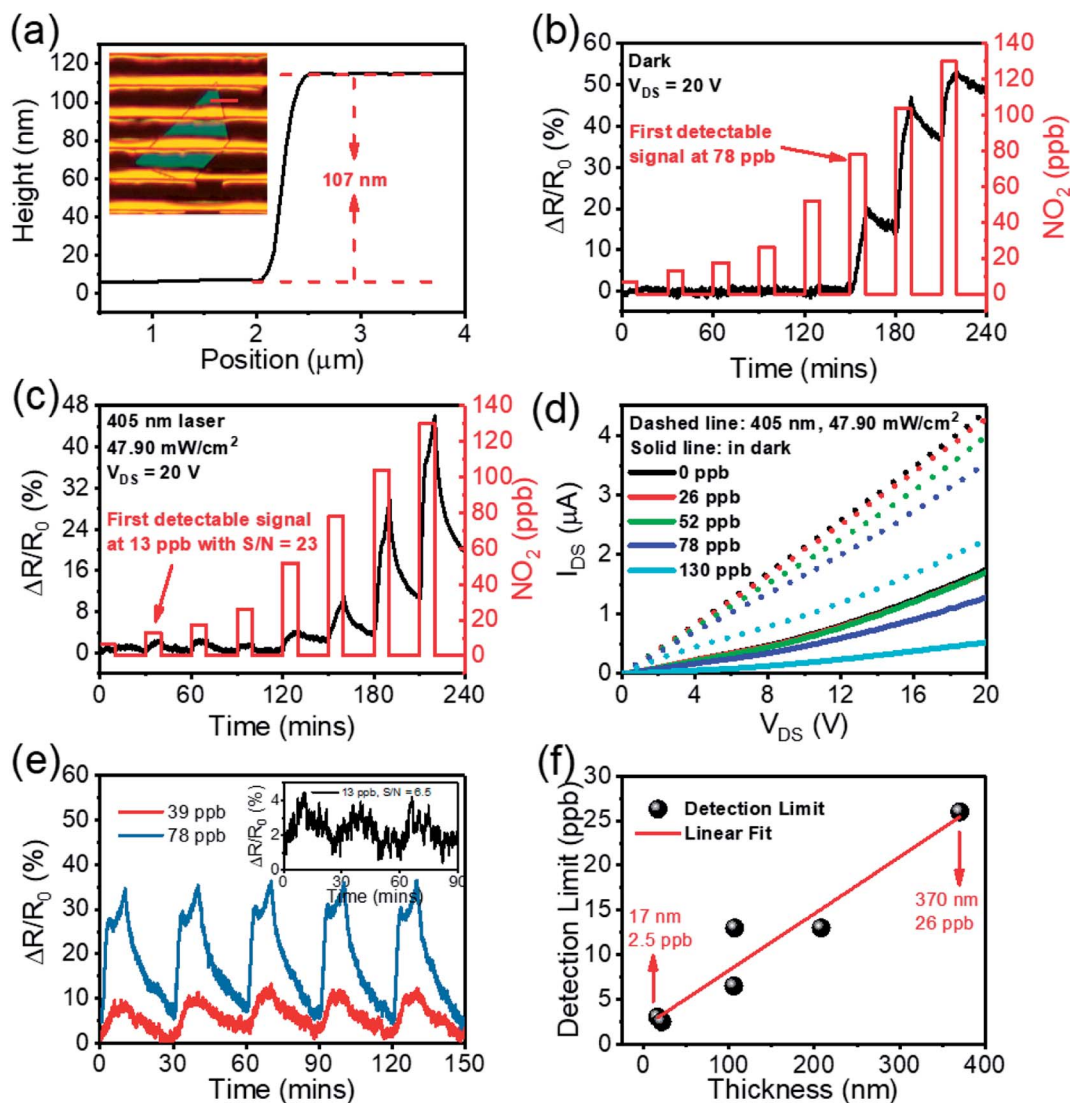


Fig. 4 (a) Optical microscope image of a SnS₂ device for gas sensor measurements. Dynamic sensing response of $\Delta R/R_0$ change versus time for the SnS₂-based sensor upon exposure to NO₂ gas with concentrations ranging from 6.5 ppb to 130 ppb under (b) dark environment and (c) 405 nm laser irradiation. (d) Comparison I_{DS} – V_{DS} curves of the device under different light illuminations and NO₂ concentrations. (e) Multi-cycle responses of sensor when exposed to concentrations of 13, 39, and 78 ppb NO₂. (f) Sensitivity as a function of SnS₂ thickness.

We replotted the dynamic-sensing response curves under 405 nm laser illumination based on the Langmuir model (Fig. 6).⁵⁸ There are two classes of adsorption sites. Similar behavior has been observed under 785 nm and 650 nm laser illumination (Fig. S10†). For the adsorption process, NO₂ can

adsorb on both sites but prefer sites with fast adsorption times. With increasing gas concentration, more NO₂ molecules will be adsorbed on sites with slower adsorption times which increases the average adsorption time. At yet higher gas concentrations all the fast adsorption time sites will be occupied. The NO₂

Table 1 Comparison of NO₂ gas sensors performance for SnS₂-based sensors. Our SnS₂ gas sensor achieves a remarkably detection limit more than 10 times better than values reported in previous works

No.	Material	Limit of detection	Special method	Reference
1	SnS ₂ flake	2.5 ppb	UV illumination	This work
2	SnS ₂ flake	30 ppb	Heater	24
3	SnS ₂ /SnO ₂ flakes	125 ppb	Heater	31
4	SnS ₂ flakes	300 ppb	Heater	39
5	SnS ₂ flakes	600 ppb	Heater	40
6	SnS ₂ /SnO ₂	1000 ppb	Heater	41
7	SnS _{1.5} /Se _{0.5}	1000 ppb	Room temperature	42



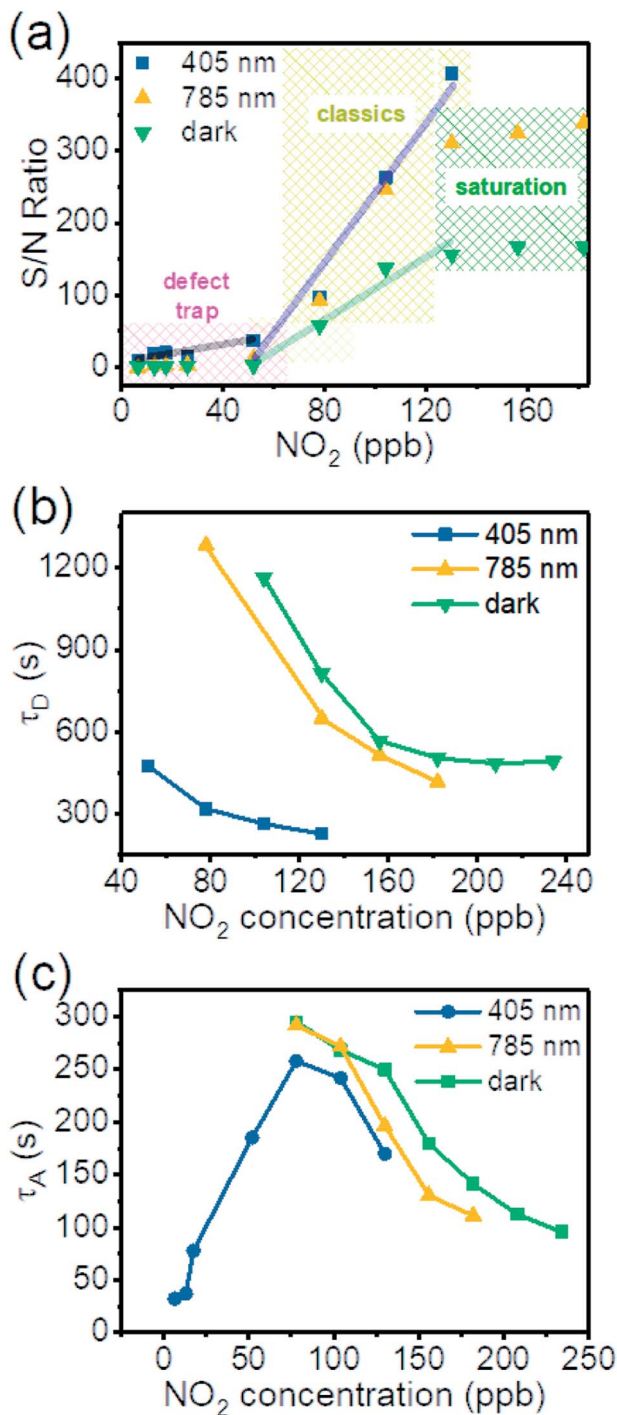


Fig. 5 (a) Comparison of S/N ratio of the dynamic response of the SnS₂ sensor for dark and illuminated environments. Green line is a linear fitting of dark environment data between 52 to 130 ppb NO₂. Black line and blue line are linear fitting of 405 nm illumination data between 6.5 to 52 ppb and 52 to 130 ppb NO₂. (b) Desorption time as a function of gas concentration. (c) Adsorption time as a function of gas concentration.

molecules will be adsorbed on the sites with slow adsorption times, which can be treated as one site adsorption. According to the Langmuir model considering one adsorption site, $\tau_D = \frac{1}{aN_c}$, where N_c is proportional to the gas concentration.

This is why the average adsorption time first increases and then decreases with gas concentration. On the other hand, for the desorption process at low concentrations it is the sites with slow desorption times that are the limiting factor. With increasing gas concentration, more molecules will be desorbed from the sites with faster desorption times, thus decreasing the average desorption time. Moreover, NO₂ molecules adsorb on sites with faster adsorption times will significantly decrease the overall conductance compared with NO₂ molecules bound to sites with slow adsorption.

To investigate whether S vacancies could give such an effect, we calculated the adsorption energy, charge density distribution, and conductance of NO₂ molecules on a SnS₂ monolayer with and without S vacancies. Details of the calculations can be found in the Methods section. Fig. 7(a) and (b) show the optimized atomic structures of a NO₂ molecule adsorbed on pristine SnS₂ and SnS₂ with a surface S vacancy respectively. For a NO₂ molecule adsorbed on pristine SnS₂, the perpendicular distance between the O atom of a NO₂ molecule and the nearest S atom of the SnS₂ (R) is about 2.95 Å and it is decreased significantly to 0.5 Å for a NO₂ molecule on SnS₂ with a S vacancy, indicating the S vacancy induces strong adsorption between SnS₂ and NO₂. To understand the adsorption mechanism, the adsorption energy as a function of distance R is plotted in Fig. 7(c). The adsorption energy is 1.011 eV when a NO₂ molecule adsorbed on SnS₂ with a S vacancy. While for a NO₂ molecule adsorbed on pristine SnS₂, the corresponding adsorption energy is only 116 meV. Thus, NO₂ molecules adsorbed on the SnS₂ monolayer with S vacancies have a larger adsorption energy compared with a pristine SnS₂ monolayer, indicating that extra energy is required to desorb NO₂ molecule from the S vacancy site (Fig. 7(d)). The adsorption energy for chemisorption is usually an energy of 1–10 eV, but for physisorption the energy is smaller than this. Therefore, the adsorption mechanism may change from physisorption to chemisorption when a NO₂ molecule adsorbed on SnS₂ with S vacancy. Moreover, there is a large partial charge transfer between S and the O atoms of a NO₂ molecule adsorbed on SnS₂ with S vacancy in Fig. 7(e) and (f).

The electrical conductivity of an NO₂ molecule adsorbed on pristine SnS₂ and on SnS₂ with a single S vacancy at 300 K are shown in Fig. 8. We found that monolayer SnS₂ contains a narrow electrical conductivity tail near $\mu = 0$ resulting from electrons crossing the Fermi level from thermal agitation at $T = 300$ K, shown in Fig. 8(a). When this pristine SnS₂ adsorbs an NO₂ molecule there is an additional conductivity peak region near $\mu = 0$ with small values of 1.1×10^3 and 3.0×10^2 ($1/\Omega \text{ m}^{-1}$) for σ_{xx} and σ_{yy} , respectively, as shown in Fig. 8(b). However for SnS₂ with a S vacancy adsorbing a NO₂ molecule, there is a broad additional conductivity region around $\mu = 0$ with values about 1.5×10^4 and 1.9×10^4 ($1/\Omega \text{ m}^{-1}$) for σ_{xx} and σ_{yy} , respectively, shown in Fig. 8(c). These results from transport and DFT analysis are consistent with the experimental conclusions on the resistance changes, and also indicate charge transfer between NO₂ and SnS₂ is critical. It also suggests that the sites with fast adsorption times are S vacancies and the sites with slow adsorption times are the areas of pristine SnS₂. To



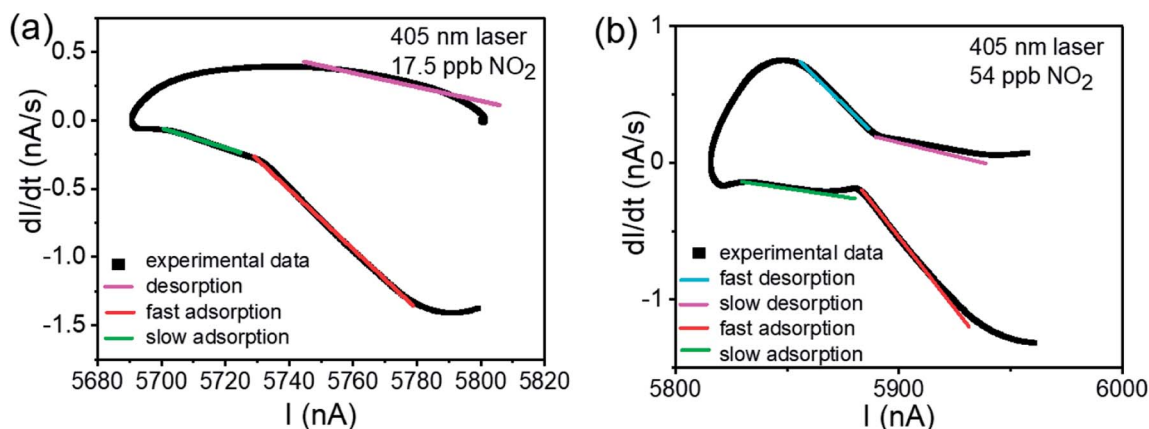


Fig. 6 The $dI/dt-I$ relationship of Fig. 4(c). (a) The $dI/dt-I$ curve of 17.5 ppb NO_2 test. (b) The $dI/dt-I$ curve of 54 ppb NO_2 test. In both of two picture, the fitting lines are shown, with blue line depicting the fast desorption process, violet line depicting slow desorption process, red line depicting fast adsorption process and green line depicting slow adsorption process.

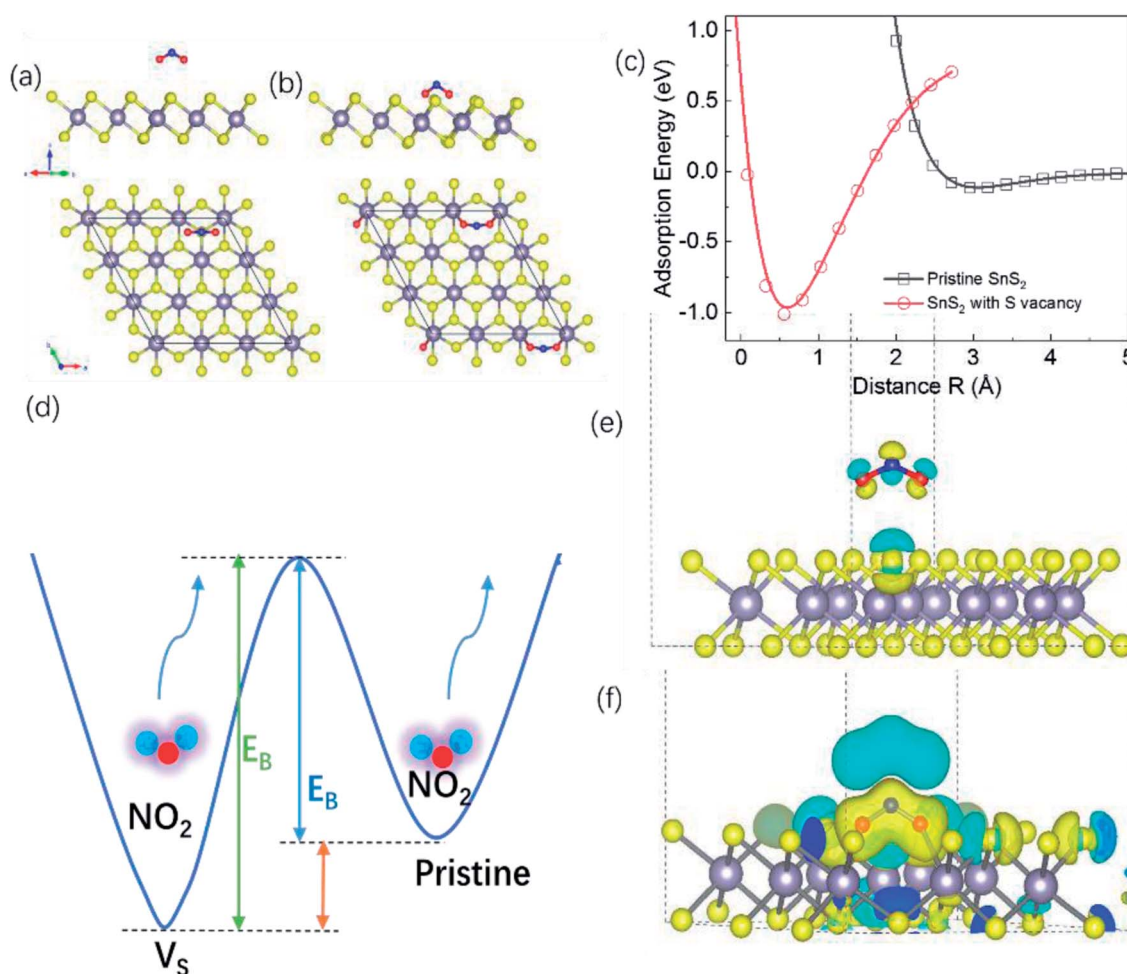


Fig. 7 (a) and (b) Side view and top view of the atomic structure of NO_2 molecule adsorbed on Sn_9S_{18} and Sn_9S_{17} surfaces, respectively. The large dark spheres represents Sn atoms, while the yellow, blue and red spheres represent the S, N and O atoms, respectively. (c) Adsorption energy as a function of distance R (defined as the perpendicular distance between the O atom of NO_2 molecule and the nearest S atom of the SnS_2) for NO_2 molecule adsorbed on Sn_9S_{18} and Sn_9S_{17} surfaces. (d) Schematic of adsorption energies for an NO_2 molecule adsorbed on pristine SnS_2 and on SnS_2 with a S vacancy. (e) and (f) Charge density difference distribution for NO_2 molecule adsorbed on Sn_9S_{18} and Sn_9S_{17} surfaces, respectively. The charge density difference in yellow indicate increases in charge density, while blue indicates a decreases in charge density (isosurface = $0.0015 \text{ eV } \text{\AA}^{-3}$).



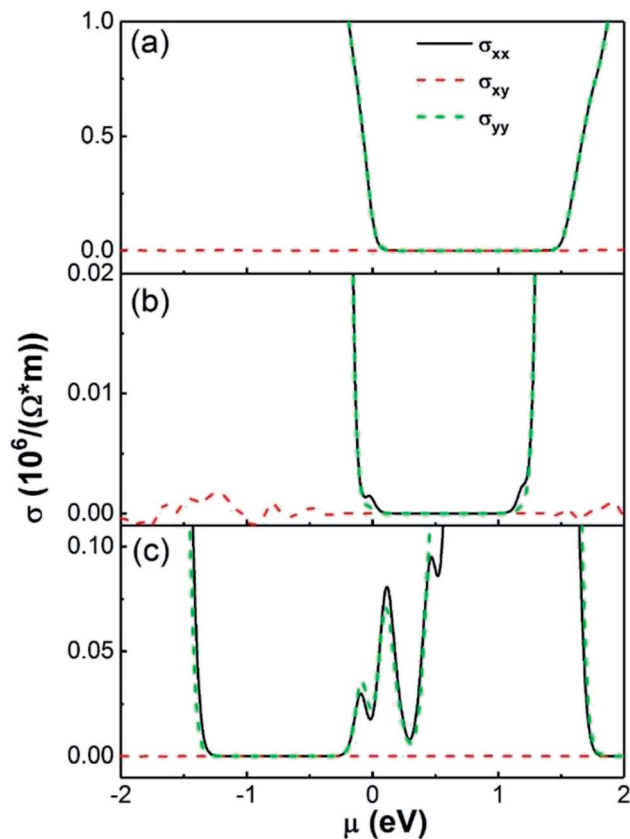


Fig. 8 Electron conductivity of (a) SnS₂ monolayer (b) SnS₂ with adsorbed NO₂ molecule and (c) SnS₂ with S vacancy adsorbed NO₂ molecule which S vacancy concentration is 3.70% at 300 K.

further highlight the importance of S vacancies for the gas sensor performance, we measured the gas performance of gas sensors based on same thick SnS₂ but with different S vacancy density, where the density of S vacancies density was determined by mapping the whole sensor using the EDS equipped in the SEM chamber. A relatively low detection limit or high gas performance was achieved for the sample with higher S vacancies density. For example, for gas sensors based one 106 nm thick SnS₂, the detection limit for sensor with an atomic ratio of Sn : S ≈ 1 : 1.72 is 6.5 ppb and that for sensor with an atomic ratio of Sn : S ≈ 1 : 1.84 is 13 ppb. We would like to stress that the nanoscale defects are naturally occurring defects forming during chemical vapor deposition. Therefore, they are quite stable when interacting with NO₂ molecules.

It is reported that Schottky junction^{59,60} and photo-excited electron-hole pairs can contribute the gas sensing performance. Since our gas sensing performance depends crucially on the wavelength of the photons and same bias voltage is used, here we focus our discussion on effect due to photo-excited electron-hole pairs. We measured the gas sensing performance of another sensor under 365, 405, and 650 nm laser illuminations (Fig. S11†). Only 365 and 405 nm lasers can excite electron-hole pairs, because the band gap of SnS₂ is around 2.2 eV. The 365 and 405 nm laser illuminations have similar detection limit, better than 650 nm. More importantly, the average adsorption time first increases and then decreases with

NO₂ concentration under the 365 and 405 nm laser illuminations, indicating the importance of photo-excited electron-hole pairs for the charge transfer process upon NO₂ adsorption. Therefore, the observed distinct behavior of the adsorption time may be due to the photo-excited electron-hole pairs. We would like to stress that the average adsorption time is faster under the 405 nm photo illumination than the 365 nm, indicating that the 405 nm laser illumination is more efficient since the adsorption spectrum of NO₂ peaks at ~400 nm at 293 K.⁶¹

In summary, NO₂ gas sensors were fabricated based on CVD grown SnS₂. An ultra-high sensitivity of 2.5 ppb at room temperature was achieved by utilizing inherent defects under photon illumination. Our results further show S vacancies in SnS₂ work as additional adsorption sites and the detection limit, signal to noise (S/N) value, and desorption time can be significantly improved by photo illumination, thus providing a plausible way to design high performance gas sensors based on 2D materials with nanoscale defects.

Method

The detailed experiments and characterizations are shown in the ESI.†

Author contributions

H. C. W. conceived the study. D. Y. C., and Y. L. L. grew the sample. W. J. Y., H. J. X. and Z. W. fabricated the sample. J. C., M. C., and B. S. C. carried out the TEM measurements. W. J. Y. and C. C. performed the gas sensing measurements. H. R. F. and C. R. C. conducted the DFT simulation. Z. T. J. and H. J. L. gave scientific advice. H. C. W., W. J. Y., D. Y. C., and C. C. wrote the manuscript. All authors discussed the results and commented on the manuscript.

Conflicts of interest

There no conflicts to declare.

Acknowledgements

This work was supported by the National Key R&D Program of China (Grant No. 2017YFE0301404 and 2017YFA0303800), the National Natural Science Foundation of China (No. 61874010, 11804237), and by the Science and Technology Innovation Program for Creative Talents in Beijing Institute of Technology (No. 2017CX01006).

References

- 1 B. Liu, L. Chen, G. Liu, A. N. Abbas, M. Fathi and C. Zhou, *ACS Nano*, 2014, 8(5), 5304–5314.
- 2 A. N. Abbas, B. Liu, L. Chen, Y. Ma, S. Cong, N. Aroonyadet, M. Köpf, T. Nilges and C. Zhou, *ACS Nano*, 2015, 9(5), 5618–5624.



- 3 S. S. Shendage, V. L. Patil, S. A. Vanalakar, S. P. Patil, N. S. Harale, J. L. Bhosale, J. H. Kim and P. S. Patil, *Sens. Actuators, B*, 2017, **240**, 426–433.
- 4 T. Xu, Y. Pei, Y. Liu, D. Wu, Z. Shi, J. Xu, Y. Tian and X. Li, *J. Alloys Compd.*, 2017, **725**, 253–259.
- 5 C. Hao, P. B. Shepson, J. W. Drummond and K. Muthuramu, *Anal. Chem.*, 1994, **66**(21), 3737–3743.
- 6 <https://www.epa.gov/criteria-air-pollutants/naaqs-table>.
- 7 J. Kong, N. R. Franklin, C. Zhou, M. G. Chapline, S. Peng, K. Cho and H. Dai, *Science*, 2000, **287**(5453), 622–625.
- 8 E. Rossinyol, A. Prim, E. Pellicer, J. Arbiol, F. Hernández-Ramírez, F. Peiró, A. Cornet, J. R. Morante, L. A. Solovyov, B. Tian, T. Bo and D. Zhao, *Adv. Funct. Mater.*, 2007, **17**(11), 1801–1806.
- 9 F. Schedin, A. K. Geim, S. V. Morozov, E. W. Hill, P. Blake, M. I. Katsnelson and K. S. Novoselov, *Nat. Mater.*, 2007, **6**(9), 652–655.
- 10 J. D. Fowler, M. J. Allen, V. C. Tung, Y. Yang, R. B. Kaner and B. H. Weiller, *ACS Nano*, 2009, **3**(2), 301–306.
- 11 S. Deng, V. Tjoa, H. M. Fan, H. R. Tan, D. C. Sayle, M. Olivo, S. Mhaisalkar, J. Wei and C. H. Sow, *J. Am. Chem. Soc.*, 2012, **134**(10), 4905–4917.
- 12 F. K. Butt, C. Cao, W. S. Khan, Z. Ali, R. Ahmed, F. Idrees, I. Aslam, M. Tanveer, J. Li and S. Zaman, *Mater. Chem. Phys.*, 2012, **137**(137), 565.
- 13 F. K. Perkins, A. L. Friedman, E. Cobas, P. M. Campbell, G. G. Jernigan and B. T. Jonker, *Nano Lett.*, 2013, **13**(2), 668–673.
- 14 W. Yuan, A. Liu, L. Huang, C. Li and G. Shi, *Adv. Mater.*, 2013, **25**(5), 766–771.
- 15 N. Huo, S. Yang, Z. Wei, S.-S. Li, J.-B. Xia and J. Li, *Sci. Rep.*, 2014, **4**, 5209.
- 16 H. Long, A. Harley-Trochimczyk, T. Pham, Z. Tang, T. Shi, A. Zettl, C. Carraro, M. A. Worsley and R. Maboudian, *Adv. Funct. Mater.*, 2016, **26**(28), 5158–5165.
- 17 J. Zhang, X. Liu, G. Neri and N. Pinna, *Adv. Mater.*, 2016, **28**(5), 795–831.
- 18 J. Baek, D. Yin, N. Liu, I. Omkaram, C. Jung, H. Im, S. Hong, S. M. Kim, Y. K. Hong, J. Hur, Y. Yoon and S. Kim, *Nano Res.*, 2017, **10**(8), 2904.
- 19 Z. Feng, Y. Xie, E. Wu, Y. Yu, S. Zheng, R. Zhang, X. Chen, C. Sun, H. Zhang, W. Pang, J. Liu and D. Zhang, *Micromachines*, 2017, **8**(5), 155.
- 20 B. Cho, J. Yoon, S. K. Lim, A. R. Kim, D.-H. Kim, S.-G. Park, J.-D. Kwon, Y.-J. Lee, K.-H. Lee, B. H. Lee, H. C. Ko and M. G. Hahm, *ACS Appl. Mater. Interfaces*, 2015, **7**(30), 16775–16780.
- 21 L. Kou, T. Frauenheim and C. Chen, *J. Phys. Chem. Lett.*, 2014, **5**(15), 2675–2681.
- 22 S. Weidong, H. Lihua, W. Haishui, Z. Hongjie, Y. Jianhui and W. Pinghui, *Nanotechnology*, 2006, **17**(12), 2918.
- 23 D. J. Late, T. Doneux and M. Bougouma, *Appl. Phys. Lett.*, 2014, **105**(23), 233103.
- 24 J. Z. Ou, W. Ge, B. Carey, T. Daeneke, A. Rotbart, W. Shan, Y. Wang, Z. Fu, A. F. Chrimes, W. Wlodarski, S. P. Russo, Y. X. Li and K. Kalantar-zadeh, *ACS Nano*, 2015, **9**(10), 10313–10323.
- 25 Y. Li, S. G. Leonardi, A. Bonavita, G. Neri and W. Wlodarski, *Procedia Eng.*, 2016, **168**, 1102–1105.
- 26 W. Yang, L. Gan, H. Li and T. Zhai, *Inorg. Chem. Front.*, 2016, **3**(4), 433–451.
- 27 Y. Huang, E. Sutter, J. T. Sadowski, M. Cotlet, O. L. A. Monti, D. A. Racke, M. R. Neupane, D. Wickramaratne, R. K. Lake, B. A. Parkinson and P. Sutter, *ACS Nano*, 2014, **8**(10), 10743–10755.
- 28 G. Su, V. G. Hadjiev, P. E. Loya, J. Zhang, S. Lei, S. Maharjan, P. Dong, P. M. Ajayan, J. Lou and H. Peng, *Nano Lett.*, 2015, **15**(1), 506–513.
- 29 J. Xia, D. Zhu, L. Wang, B. Huang, X. Huang and X.-M. Meng, *Adv. Funct. Mater.*, 2015, **25**(27), 4255–4261.
- 30 X. Zhou, Q. Zhang, L. Gan, H. Li and T. Zhai, *Adv. Funct. Mater.*, 2016, **26**(24), 4405–4413.
- 31 J. Hao, D. Zhang, Q. Sun, S. Zheng, J. Sun and Y. Wang, *Nanoscale*, 2018, **10**(15), 7210–7217.
- 32 D. Gu, X. Li, Y. Zhao and J. Wang, *Sens. Actuators, B*, 2017, **244**, 67–76.
- 33 R. Li, K. Jiang, S. Chen, Z. Lou, T. Huang, D. Chen and G. Shen, *RSC Adv.*, 2017, **7**(83), 52503–52509.
- 34 K. Xu, N. Li, D. Zeng, S. Tian, S. Zhang, D. Hu and C. Xie, *ACS Appl. Mater. Interfaces*, 2015, **7**(21), 11359.
- 35 H. Chen, Y. Chen, H. Zhang, D. W. Zhang, P. Zhou and J. Huang, *Adv. Funct. Mater.*, 2018, **28**(20), 1801035.
- 36 M. Ma, H. Khan, W. Shan, Y. Wang, J. Z. Ou, Z. Liu, K. Kalantar-zadeh and Y. Li, *Sens. Actuators, B*, 2017, **239**, 711–717.
- 37 Y. Xiong, W. Xu, D. Ding, W. Lu, L. Zhu, Z. Zhu, Y. Wang and Q. Xue, *J. Hazard. Mater.*, 2018, **341**, 159–167.
- 38 Z. Qin, K. Xu, H. Yue, H. Wang, J. Zhang, C. Ouyang, C. Xie and D. Zeng, *Sens. Actuators, B*, 2018, **262**, 771–779.
- 39 C. Kim, J. C. Park, S. Y. Choi, Y. Kim, S. Y. Seo, T. E. Park, S. H. Kwon, B. Cho and J. H. Ahn, *Small*, 2018, 1704116.
- 40 M. Ma, H. Khan, W. Shan, Y. Wang, J. Z. Ou, Z. Liu, K. Kalantar-zadeh and Y. Li, *Sens. Actuators, B*, 2017, **239**, 711–717.
- 41 D. Gu, X. Li, Y. Zhao and J. Wang, *Sens. Actuators, B*, 2017, **244**, 67–76.
- 42 Z. Yang, H. Liang, X. Wang, X. Ma, T. Zhang, Y. Yang, L. Xie, D. Chen, Y. Long and J. Chen, *ACS Nano*, 2015, **10**(1), 755–762.
- 43 B. Cho, M. G. Hahm, M. Choi, J. Yoon, A. R. Kim, Y.-J. Lee, S.-G. Park, J.-D. Kwon, C. S. Kim, M. Song, Y. Jeong, K.-S. Nam, S. Lee, T. J. Yoo, C. G. Kang, B. H. Lee, H. C. Ko, P. M. Ajayan and D.-H. Kim, *Sci. Rep.*, 2015, **5**, 8052.
- 44 D. Barreca, D. Bekermann, E. Comini, A. Devi, R. A. Fischer, A. Gasparotto, C. Maccato, G. Sberveglieri and E. Tondello, *Sens. Actuators, B*, 2010, **149**(1), 1–7.
- 45 J. Deng, Z. Luo, X. Wu and Y. Hu, *Min. Sci. Technol.*, 2010, **20**(4), 557–562.
- 46 Z. S. Hosseini, A. I. zad and A. Mortezaali, *Sens. Actuators, B*, 2015, **207**, 865–871.
- 47 Y. Gong, H. Yuan, C.-L. Wu, P. Tang, S.-Z. Yang, A. Yang, G. Li, B. Liu, J. van de Groep, M. L. Brongersma, M. F. Chisholm, S.-C. Zhang, W. Zhou and Y. Cui, *Nat. Nanotechnol.*, 2018, **13**(4), 294–299.



- 48 J. Shen, G. Liu, K. Huang, Z. Chu, W. Jin and N. Xu, *ACS Nano*, 2016, **10**(3), 3398–3409.
- 49 G. Chen, T. M. Paronyan, E. M. Pigos and A. R. Harutyunyan, *Sci. Rep.*, 2012, **2**, 343.
- 50 S. Park, S. An, H. Ko, S. Lee and C. Lee, *Sens. Actuators, B*, 2013, **188**, 1270–1276.
- 51 S. Park, S. An, Y. Mun and C. Lee, *ACS Appl. Mater. Interfaces*, 2013, **5**(10), 4285–4292.
- 52 J. H. Ahn, M. J. Lee, H. Heo, H. S. Ji, K. Kim, H. Hwang and M. H. Jo, *Nano Lett.*, 2015, **15**(6), 3703.
- 53 L. A. Currie, Quellenangabe, *Pure Appl. Chem.*, 1995, 1699–1723.
- 54 J. Li, Y. Lu, Q. Ye, M. Cinke, J. Han and M. Meyyappan, *Nano Lett.*, 2003, **3**(7), 929–933.
- 55 K. Lee, R. Gatensby, N. McEvoy, T. Hallam and G. S. Duesberg, *Adv. Mater.*, 2013, **25**(46), 6699–6702.
- 56 Y. Wu, D. Zhang, K. Lee, G. S. Duesberg, A. Syrlybekov, X. Liu, M. Abid, M. Abid, Y. Liu and L. Zhang, *Adv. Mater. Technol.*, 2016, 1600197.
- 57 P. E. Blöchl, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1994, **50**(24), 17953–17979.
- 58 C. Wen, Q. Ye, S.-L. Zhang and D. Wu, *Sens. Actuators, B*, 2016, **223**, 791–798.
- 59 M. Zhu, X. Li, S. Chung, L. Zhao, X. Li, X. Zang, K. Wang, J. Wei, M. Zhong, K. Zhou, D. Xie and H. Zhu, *Carbon*, 2015, **84**, 138–145.
- 60 H.-Y. Kim, K. Lee, N. McEvoy, C. Yim and G. S. Duesberg, *Nano Lett.*, 2013, **13**(5), 2182–2188.
- 61 J. P. Burrows, A. Dehn, B. Deters, S. Himmelmann, A. Richter, S. Voigt and J. Orphal, *J. Quant. Spectrosc. Radiat. Transfer*, 1998, **60**(6), 1025–1031.

