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Ceria/cobalt borate hybrids as efficient electrocatalysts for water oxidation under neutral conditions†

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Oxygen evolution reaction (OER) catalysts are of importance for electrochemical water splitting and fuel generation. Despite enormous efforts, the design and development of OER catalysts with high catalytic activities under neutral conditions are highly desired but still remain a great challenge. Herein, we report a room temperature chemical route to prepare xceria/cobalt borate ($x\text{CeO}_2/\text{Co-Bi}$) hybrids as efficient OER catalysts by tuning the molar ratio of Ce/Co (x represents the amount of CeO_2). The optimised catalyst ($20\text{CeO}_2/\text{Co-Bi}$ hybrid) was found to exhibit remarkable OER catalytic activity with an overpotential of 453 mV at a current density of 10 mA cm^{-2} , Tafel slope of 120 mV dec^{-1} and long-term stability in neutral medium due to its good conductivity, mass transportation and strong synergetic coupling effects, showing the potential of Co-based electrochemical materials for practical application in energy storage devices.

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1. Introduction

Many efforts have been made to explore alternative energy storage and conversion systems to solve the increasing environmental problems and energy crisis.^{1–3} Hydrogen has been considered as a clean alternative energy due to its high energy output and carbon-neutral combustion products,^{2,4} and electrochemical water splitting is thought to be a promising technology to convert electrical energy into chemical fuels for hydrogen storage.^{5–7} However, the oxygen evolution reaction (OER) significantly affects the overall water-splitting efficiency due to its sluggish kinetics with an energy-intensive reaction and high overpotential.^{8–11} Although many electrocatalysts have been explored to achieve high current density at low overpotentials and expedite the OER reaction kinetics process,^{8,9,12,13} noble metal oxides (*e.g.* RuO_2 or IrO_2), operating in acidic or alkaline solutions, are the most active OER electrocatalysts.

However, their high cost, scarcity, and the requirement for extreme pH conditions have greatly hindered the large-scale application of these OER catalysts.^{12,14–16} Therefore, it is highly desirable to develop effective non-noble metal OER catalytic alternatives to improve the efficiency of water oxidation in a near-neutral environment.

Oxygen-evolving complexes composed of manganese and calcium are well known as natural water-splitting complexes.^{17,18} Inspired by this thought, a new category of artificial water oxidation electrocatalysts with amorphous features consisting of metals (such as nickel and cobalt), oxygen, and inorganic borate, namely nickel–borate (Ni–Bi) or cobalt–borate (Co–Bi), has attracted significant attention due to their low cost, high intrinsic activity, and superior stability.^{13,14,19–22} For example, Nocera *et al.* reported that a nickel–borate (Ni–Bi) film, electro-deposited on indium tin oxide (ITO), had high catalytic activity towards electrochemical water oxidation under near-neutral conditions, achieving a current density of 1 mA cm^{-2} in 0.1 M borate buffer at an overpotential of 425 mV.¹⁹ A nickel–borate nanoarray supported by carbon cloth (Ni–Bi/CC) yielded a geometrical catalytic current density of 10 mA cm^{-2} at an overpotential of 470 mV.¹⁴ Recently, amorphous Co–Bi ultrathin nanosheets were designed for electrochemical water oxidation under neutral conditions.^{13,23} However, the application of Co–Bi as a catalyst is limited by its poor OER kinetics and mass transfer ability, which can be solved by adding OER co-catalysts.

As one of the most important rare earth oxides, ceria (CeO_2) has been usually used in the electrochemical field due to the following advantages:^{7,24–27} (i) flexible transition between the Ce^{3+} and Ce^{4+} oxidation states, which can provide the opportunity to generate strong electronic interactions with other

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† Electronic supplementary information (ESI) available: SEM, TEM, XRD and SAED of Co–Bi nanosheets, EDS mapping images, EEIS spectrum of $20\text{CeO}_2/\text{Co-Bi}$, nitrogen adsorption/desorption isotherms of as-grown products (Co–Bi, $10\text{CeO}_2/\text{Co-Bi}$, $20\text{CeO}_2/\text{Co-Bi}$ and $30\text{CeO}_2/\text{Co-Bi}$ composites), CVs of all the samples (Co–Bi, $10\text{CeO}_2/\text{Co-Bi}$, $20\text{CeO}_2/\text{Co-Bi}$ and $30\text{CeO}_2/\text{Co-Bi}$ composites), impedance Nyquist plots of Co–Bi, $10\text{CeO}_2/\text{Co-Bi}$, $20\text{CeO}_2/\text{Co-Bi}$ and $30\text{CeO}_2/\text{Co-Bi}$ composite electrodes. CVs of the $20\text{CeO}_2/\text{Co-Bi}$ composite electrode, comparison of the OER activities of cobalt-based catalysts in neutral medium listed in Table S1. See DOI: 10.1039/c9na00356h



matrices and therefore probably improves the catalytic performance; (ii) good electronic/ionic conductivity, which can improve charge transfer and promote the reaction process; (iii) reversible surface oxygen ion exchange, which will absorb O_2 produced during the OER and enhance the OER activity; (iv) large oxygen-storage capacity, which is helpful to form and repair oxygen vacancies on the CeO_2 surface, binding adsorbates much more strongly than normal oxide sites.²⁴ Therefore, the introduction of CeO_2 onto Co–Bi nanosheets will be beneficial for the formation of hydroperoxy species (OOH_{ad}) on the surface of the hybrids owing to the high mobility of oxygen vacancies. Furthermore, the chemical synergistic effect between Co–Bi nanosheets and CeO_2 would also improve the OER catalytic performance. Thus, the CeO_2 /Co–Bi hybrids should be efficient candidates to improve the electrocatalytic activity for OER in neutral medium.

In this work, we report a room temperature chemical route to prepare $xCeO_2$ /Co–Bi hybrids ($10CeO_2$ /Co–Bi, $20CeO_2$ /Co–Bi, $30CeO_2$ /Co–Bi; x represents the amount of CeO_2) through tuning the molar ratio of Ce/Co, and their OER catalytic activities were compared. It was found that $20CeO_2$ /Co–Bi exhibited the highest OER catalytic activity with an overpotential of 453 mV at a current density of 10 mA cm^{-2} , Tafel slope of around 120 mV dec^{-1} and long-term stability in phosphate-buffered saline (PBS) solution because of the good conductivity, mass transportation and strong synergetic coupling effects.

2. Experimental

All chemicals were commercially available and used without further purification. Cobalt nitrate hexahydrate ($Co(NO_3)_2 \cdot 6H_2O$), cerium nitrate hexahydrate ($Ce(NO_3)_3 \cdot 6H_2O$) and sodium borohydride ($NaBH_4$) were purchased from Sigma. Milli-Q water ($18.2 \text{ M}\Omega \text{ cm}$) was used in this work.

2.1 Synthesis of CeO_2 /Co–Bi composites

The CeO_2 /Co–Bi samples were prepared by a room temperature chemical synthesis method as illustrated in Fig. 1a. In particular, $n \text{ mmol}$ ($n = 0.1, 0.2$ and 0.3) $Ce(NO_3)_3 \cdot 6H_2O$ and $(1 - n) \text{ mmol}$ $Co(NO_3)_2 \cdot 6H_2O$ were added into a flask with 100 ml deionized water under continuous and vigorous stirring for complete dissolution (marked as A). Then, 5 mL $NaBH_4$ solution (0.5 mol L^{-1}) was added to A, and the mixed solution was kept at room temperature for 40 min under strong stirring; the solution gradually turned into black suspended solids (marked as B). The suspended products were collected by centrifugation and laved with deionized water and ethanol several times. The as-prepared products were finally dried in a vacuum overnight for further characterization. Co–Bi hierarchical nanosheets were prepared by the same procedure but without using $Ce(NO_3)_3 \cdot 6H_2O$.

2.2 Characterization

The surface area was measured by nitrogen physisorption (Micromeritics, ASAP 2020 HD88) based on the Brunauer–

Emmett–Teller (BET) method. The phase evolution of the as-synthesized nanostructures was monitored by powder X-ray diffraction (XRD). The XRD patterns with diffraction intensity *versus* 2θ were recorded with a Shimadzu X-ray diffractometer (Model 6000) using $Cu \text{ K}\alpha$ radiation. X-ray photoelectron spectra (XPS) were acquired on a Thermo Scientific Model K-Alpha with $Al \text{ K}\alpha$ as the excitation source. Raman spectra were recorded at room temperature using an XY Dilor spectrograph equipped with a Spex CCD detector (2000×800 pixels). Spectra were recorded in microconfiguration with a laser impact of *ca.* $1 \mu\text{m}$ diameter. The excitation source was the 633 nm line of an argon laser at the power level of 5.4 mW. Scanning electron microscopy (SEM) was performed on a Hitachi Su8010 scanning electron microscope. Transmission electron microscopy (TEM) studies were conducted on a Hitachi HT-7700 transmission electron microscope with an accelerating voltage of 120 kV. High-resolution TEM, selected area electron diffraction (SAED), electron energy loss spectroscopy (EELS), and energy-dispersive X-ray spectroscopy (EDS) were conducted on a JEOL JEM 2100F transmission electron microscope with an accelerating voltage of 200 kV.

2.3 Electrochemical measurements

Electrocatalytic activities assessed by linear sweep voltammetry (LSV) and chronoamperometry were measured on a CHI 660D electrochemical workstation (CH Instruments, Shanghai, China) with a standard three-electrode system in 0.1 M PBS aqueous solution ($\text{pH} = 7.4$). The counter and reference electrodes were platinum wire and standard $Ag/AgCl$ (3 M KCl), respectively. The catalyst with a loading of 0.56 mg cm^{-2} on carbon fiber paper (CFP) was used as the working electrode. The scanning rate for LSV measurements was 5 mV s^{-1} . Rotating-disk electrode voltammograms were obtained at a scan rate of 10 mV s^{-1} . Electrochemical impedance spectroscopy (EIS) was performed on the AUTOLAB PGSTAT204 electrochemical workstation in the frequency range from 0.01 Hz to 100 kHz at an open circuit potential, with 10 mV as the amplitude potential.

3. Results and discussion

The morphology and composition of the as-prepared products were characterized by SEM, TEM and XRD, as shown in Fig. 1. The SEM (Fig. 1b) and TEM (Fig. 1c) images of the as-prepared Co–Bi nanosheets show a hierarchical structure comprising aggregated ultrathin nanosheets. XRD patterns (Fig. 1c) show that Co–Bi ultrathin nanosheets have no diffraction peaks, implying the presence of amorphous structures, which is in agreement with a previous report.¹³ The amorphous structure was further confirmed by selected area electron diffraction (SAED) as shown in Fig. 1e. Some new nanoparticles were observed to be uniformly coated on the surface of Co–Bi nanosheets after introducing the source of Ce (Fig. 2a–c). In addition, the amount of the particles increases with the incremental amount of Ce source, but excess particles would destroy the structure of the Co–Bi nanosheets (Fig. 2a–c). A new



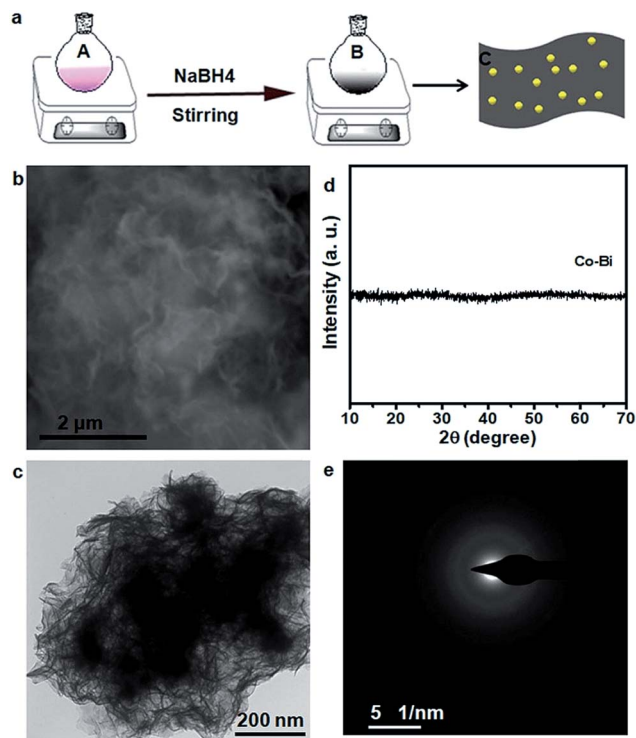


Fig. 1 (a) Schematic illustration of the synthesis of the $\text{CeO}_2/\text{Co-Bi}$ hybrids by a room temperature chemical method. A refers to the solution consisting of $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ and $\text{Ce}(\text{NO}_3)_3 \cdot 6\text{H}_2\text{O}$. B refers to the suspended products of $\text{CeO}_2/\text{Co-Bi}$. C refers to the structure of $\text{CeO}_2/\text{Co-Bi}$ hybrids. (b) SEM and (c) TEM images, (d) XRD patterns, and (e) SAED pattern of pure Co-Bi nanosheets.

diffraction peak at 33° (Fig. 2d, marked by *), indexed to the (200) of CeO_2 ^{24,26} was detected from $\text{CeO}_2/\text{Co-Bi}$ composites compared to pure Co-Bi nanosheets, and its intensity increased with the increase in the Ce source (Fig. 1d). No more XRD peaks of CeO_2 (■ represents the peaks of pure CeO_2 , JCPDS no. 34-0394) were detected due to the small size of CeO_2 particles (<5 nm), their good dispersibility on the nanosheets and the effect of the amorphous structure of Co-Bi nanosheets. Additionally, EDS mapping and EELS spectra show the presence and the homogeneous distribution of the elements Co, B, O, and Ce in the as-prepared $20\text{CeO}_2/\text{Co-Bi}$ composite, as shown in Fig. S1 and S2† which depicts the K-edge of B at 188 eV,^{23,28} L-edges of Co at 780 eV and M-edges of Ce at 890 eV.²⁹⁻³¹ In order to further analyse the compositions, we chose $20\text{CeO}_2/\text{Co-Bi}$ as a representative to be studied by HRTEM (Fig. 2e). The nanoparticles on the surface of the Co-Bi nanosheets show a lattice fringe with a distance of 0.31 nm, which was assigned to (111) of CeO_2 . The results demonstrate that the $\text{CeO}_2/\text{Co-Bi}$ composites have been successfully prepared.

Raman spectroscopy was applied to further confirm the presence of CeO_2 in the as-prepared composites because the F_{2g} mode of CeO_2 of the fluorite structure is Raman active at 465 cm^{-1} .^{32,33} Fig. 2f shows the Raman spectra collected from the composites (black line indicates pure Co-Bi nanosheets), and the intensity increases with the increase in the amount of

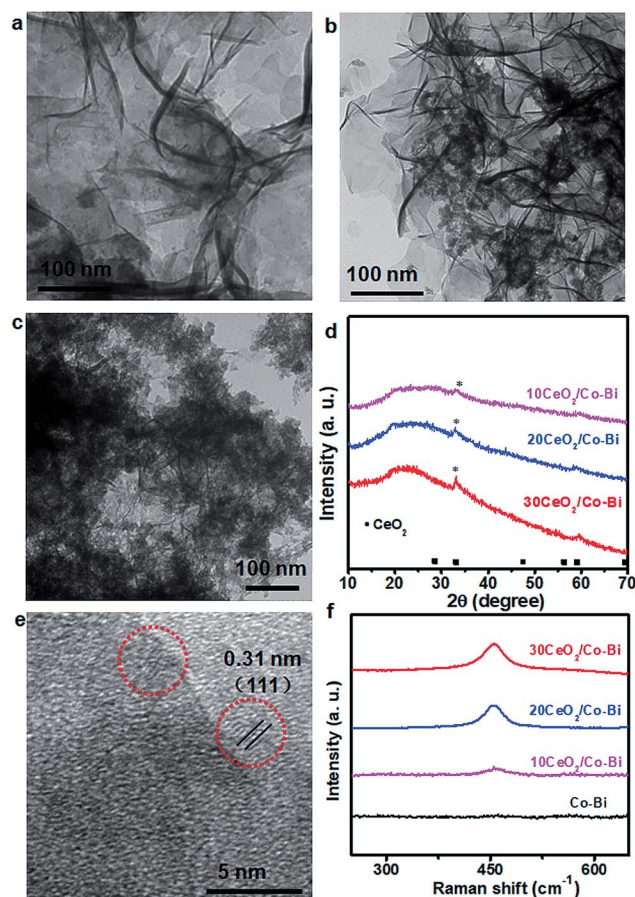


Fig. 2 Characterization of $\text{CeO}_2/\text{Co-Bi}$ composites. TEM images (a–c) and (d) XRD patterns of $10\text{CeO}_2/\text{Co-Bi}$, $20\text{CeO}_2/\text{Co-Bi}$ and $30\text{CeO}_2/\text{Co-Bi}$, respectively. (e) HRTEM images of $20\text{CeO}_2/\text{Co-Bi}$ and (f) Raman spectra for Co-Bi, $10\text{CeO}_2/\text{Co-Bi}$, $20\text{CeO}_2/\text{Co-Bi}$ and $30\text{CeO}_2/\text{Co-Bi}$.

CeO_2 . The $\sim 10\text{ cm}^{-1}$ shift of Raman frequency between our samples and literature reports³³⁻³⁵ is attributed to the size effects, which is consistent with a previous report.³⁶

The specific surface area of the as-grown samples was measured from the standard BET procedure (Fig. S3†). The BET surface areas of $10\text{CeO}_2/\text{Co-Bi}$, $20\text{CeO}_2/\text{Co-Bi}$ and $30\text{CeO}_2/\text{Co-Bi}$ were $249.8\text{ m}^2\text{ g}^{-1}$, $290.2\text{ m}^2\text{ g}^{-1}$ and $330.1\text{ m}^2\text{ g}^{-1}$, respectively, much higher than that of pure Co-Bi nanosheets ($134.0\text{ m}^2\text{ g}^{-1}$). It is generally accepted that a larger surface area means better catalytic activity.

To evaluate the OER activities of the $\text{CeO}_2/\text{Co-Bi}$ composites, the catalysts were loaded onto the CFP electrode with a density of 0.56 mg cm^{-2} by drop casting. The potentials reported in this work were calibrated to the RHE using the following equation: $E(\text{RHE}) = E(\text{Ag}/\text{AgCl}) + (0.197 + 0.0591 \times \text{pH})\text{ V}$.¹³ Fig. 3a shows the LSV curves of a bare CFP substrate, Co-Bi, $10\text{CeO}_2/\text{Co-Bi}$, $20\text{CeO}_2/\text{Co-Bi}$, $30\text{CeO}_2/\text{Co-Bi}$ and commercial RuO_2/C electrodes in 0.1 M PBS. Compared to the strong catalytic current from electrodes with catalysts, almost no current density from the bare CFP substrate was detected at 1.8 V (vs. RHE), indicating that CFP is inactive towards O_2 evolution and



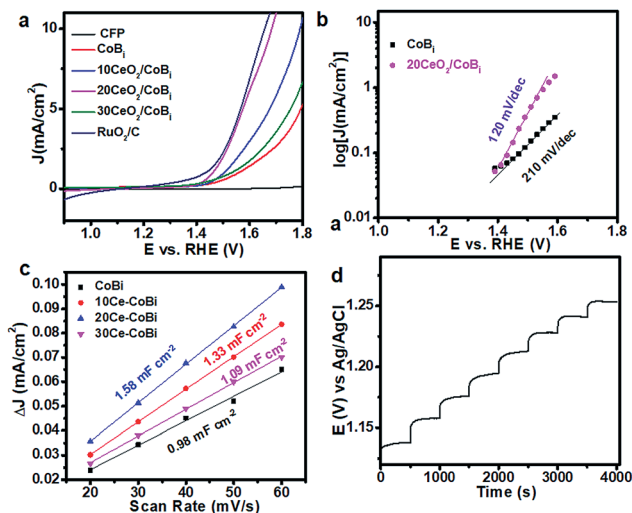


Fig. 3 (a) Linear sweep voltammograms of all catalytic (CFP, Co-Bi, 10CeO₂/Co-Bi, 20CeO₂/Co-Bi, 30CeO₂/Co-Bi and commercial RuO₂/C) electrodes in 0.1 M PBS without *iR* correction; (b) Tafel slope for Co-Bi and 20CeO₂/Co-Bi; (c) multi-current process of 20CeO₂/Co-Bi in 0.1 M PBS. The current density started at 2 mA cm⁻² and ended at 9 mA cm⁻², with an increment of about 1 mA cm⁻² per 500 s, without *iR* correction and (d) charging current density differences plotted versus scan rate. The linear slope, equivalent to twice the double-layer capacitance C_{dl} , was used to represent the ECSA.

a background correction for the CFP support is unnecessary for all catalytic electrodes. The overpotentials at the specified current density of 5 mA cm⁻², the current densities at 1.8 V (*vs.* RHE) and the Tafel slopes of all catalysts, obtained in 0.1 M PBS, are summarized in Table 1. The 20CeO₂/Co-Bi electrode demands an overpotential of only 346 mV to deliver a catalytic current density of 5 mA cm⁻², much lower than those of 10CeO₂/Co-Bi (446 mV) and 30CeO₂/Co-Bi (525 mV) under the same catalytic current density, respectively. For comparison, the electrocatalytic OER activity of Co-Bi nanosheets was measured as well, obtaining an overpotential of 560 mV at a current density of 5 mA cm⁻², much higher than those of CeO₂/Co-Bi composite electrodes. In addition, we obtained an overpotential of 453 mV from the 20CeO₂/Co-Bi electrode at a current density of 10 mA cm⁻², which is close to that of RuO₂/C (430 mV) and also comparable to the values reported previously under neutral conditions (Table S1†).^{13,14,23,37–44} The current density of the 20CeO₂/Co-Bi electrode at 1.8 V (*vs.* RHE) was 19.2 mA cm⁻², not only much larger than 11.4, 7.2, and 5.7 mA cm⁻² obtained

from 10CeO₂/Co-Bi, 30CeO₂/Co-Bi and Co-Bi electrodes, respectively, but also superior to those of A-CoS_{4.6}O_{0.6} PNCs (4.59 mA cm⁻²) and Co-Bi NS/G (14.4 mA cm⁻²) measured at the same potential (1.8 V *vs.* RHE).^{13,45} All these results indicate the high OER catalytic activity of the 20CeO₂/Co-Bi hybrid under neutral conditions.

We also fitted the polarization curves (Fig. 3a) using the Tafel equation $\eta = b \log(j/j_0)$, where η represents the overpotential, b is the Tafel slope, j refers to the current density and j_0 is the exchange current density.⁹ Accordingly, the derived Tafel slope of 120 mV dec⁻¹ from 20CeO₂/Co-Bi was much lower than the slope of 210 mV dec⁻¹ from Co-Bi (Fig. 3b) and close to or even lower than the values measured from other reported catalysts, including 3D Ni-Bi nanowire arrays (107 mV dec⁻¹),¹⁴ Co-Bi/graphene composites (160 mV dec⁻¹),¹³ Ni-Bi/RGO (176 mV dec⁻¹)⁴⁶ and 3D Co-Pi nanowires (187 mV dec⁻¹),¹² suggesting the high OER activity of 20CeO₂/Co-Bi. Additionally, Fig. 3c shows a multi-step chronopotentiometric curve for the 20CeO₂/Co-Bi composite electrode at current densities from 2 to 9 mA cm⁻² (*ca.* 1 mA cm⁻² per 500 s). The potential immediately levels off at 1.13 V (*vs.* Ag/AgCl) at the initial current value and remains unchanged for 500 s, and the other steps also show similar results, implying good conductivity, mass transportation, and mechanical robustness of this electrode.^{47,48} These results further demonstrate that 20CeO₂/Co-Bi composite electrodes have excellent OER catalytic performance.

The reasons for the enhancement of electrochemical catalytic OER activity of CeO₂/Co-Bi compared to that of Co-Bi are summarised as follows. First, CeO₂/Co-Bi composites have a much larger surface area than Co-Bi and thus more catalytically active sites, which can significantly improve the OER process (Table 1). Although the surface area of the 30CeO₂/Co-Bi composite (330.1 m² g⁻¹) is larger than that of the 20CeO₂/Co-Bi composite (290.2 m² g⁻¹), 30CeO₂/Co-Bi shows worse OER performance than 20CeO₂/Co-Bi because the nanosheet structure of 30CeO₂/Co-Bi has been destroyed (Fig. 2c). Electrochemical impedance spectroscopy (EIS) was used to further investigate the detailed characteristics of Co-Bi, 10CeO₂/Co-Bi, 20CeO₂/Co-Bi, and 30CeO₂/Co-Bi as capacitive electrodes (see ESI Fig. S4†). The semicircular characteristic of EIS curves suggests the smallest charge transfer and the best charge conductivity of 20CeO₂/Co-Bi compared to those of other catalysts (Co-Bi, 10CeO₂/Co-Bi and 30CeO₂/Co-Bi) and thus the highest OER catalytic activity among the samples.

Table 1 Summary of the OER electrocatalytic activities of the samples in phosphate-buffered saline solution (pH = 7.4)

Sample	η^a at 5 (mA cm ⁻²)	j^b at 1.8 V (<i>vs.</i> RHE)	Tafel slope (mV dec ⁻¹)	BET surface area (m ² g ⁻¹)	Capacitance (mF cm ⁻²)
Co-Bi	560	5.7	210	134.0	0.98
10CeO ₂ /Co-Bi	446	11.4	160	249.8	1.33
20CeO ₂ /Co-Bi	346	19.2	120	290.2	1.58
30CeO ₂ /Co-Bi	525	7.2	198	330.1	1.09

^a η refers to the overpotential (mV) of the samples at a current density of 5 mA cm⁻². ^b j refers to the current densities of the catalysts at a specific overpotential.

Second, $\text{CeO}_2/\text{Co-Bi}$ composites have a larger electrochemical surface area (ECSA) compared to Co-Bi . A larger ECSA means more accessible surface permeation and thus higher catalytic activity. According to the CV curves of Co-Bi , $10\text{CeO}_2/\text{Co-Bi}$, $20\text{CeO}_2/\text{Co-Bi}$, and $30\text{CeO}_2/\text{Co-Bi}$ (see ESI Fig. S5[†]), capacitive currents as a function of the scan rate of the samples are plotted in Fig. 3c. The double-layer capacitance (C_{dl}) of each sample was estimated from the slope of the corresponding linear fitting, and therefore, one can obtain the value of ECSA, which is double the C_{dl} value.⁴⁵ Obviously, $20\text{CeO}_2/\text{Co-Bi}$ has the highest C_{dl} among the four samples (Fig. 3d), and thus, the largest ECSA contributes to the best catalytic performance.

Lastly, in order to investigate the strong electronic interactions between Co-Bi and CeO_2 , XPS was applied to characterize pure CeO_2 , Co-Bi , $20\text{CeO}_2/\text{Co-Bi}$, $20\text{CeO}_2/\text{Co-Bi}$ and $30\text{CeO}_2/\text{Co-Bi}$ hybrids, and the results are plotted in Figs 4, S6 and S7.[†] All the spectra were referenced to the aliphatic carbon at a binding energy (BE) of 284.5 eV. The survey spectra of $20\text{CeO}_2/\text{Co-Bi}$ showed the presence of Co, B, O and Ce, but no Ce was detected from Co-Bi nanosheets (Fig. 4c), consistent with EDS and EELS results (see ESI Fig. S1 and S2[†]).

As shown in Fig. 4b, the peak at 192.5 eV is assigned to the core level of B^{3+} in borate species. The O 1s spectra of pure CeO_2 , Co-Bi , $10\text{CeO}_2/\text{Co-Bi}$, $20\text{CeO}_2/\text{Co-Bi}$ and $20\text{CeO}_2/\text{Co-Bi}$ are shown in Fig. S8.[†] The peak at 529.4 eV is ascribed to the O 1s of Ce-O and that at 531.9 eV corresponds to O 1s of the central oxygen atoms in borate.^{25,49} For high-resolution Ce 3d spectra of pure CeO_2 (Fig. 4c), the peaks located at 896–926 eV correspond to $\text{Ce } 3d_{3/2}$, and the peaks located at 880–892 eV are consistent with $\text{Ce } 3d_{5/2}$, which demonstrated the coexistence of Ce^{3+} and Ce^{4+} in CeO_2 . However, after introducing CeO_2 NPs onto the Co-Bi nanosheets, the ratio of $\text{Ce}^{3+} : \text{Ce}^{4+}$ in the $\text{CeO}_2/\text{Co-Bi}$ hybrid changed compared with that of pure CeO_2 , suggesting that the valence states of Ce in the $\text{CeO}_2/\text{Co-Bi}$ hybrid have rearranged.^{25,26} As shown in Fig. 4d, the binding energy of Co 2p from $20\text{CeO}_2/\text{Co-Bi}$ showed an ~ 0.5 eV negative shift

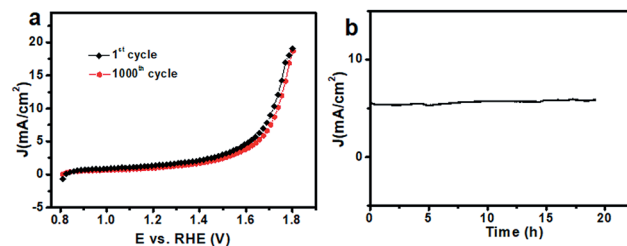


Fig. 5 (a) Polarization curves of $20\text{CeO}_2/\text{Co-Bi}$ before and after 1000 CV cycles in PBS; (b) chronoamperometric response of the as-prepared $20\text{CeO}_2/\text{Co-Bi}$ composite recorded at a constant potential of 1.65 V vs. RHE.

compared to that of pure Co-Bi nanosheets, presumably caused by the electron transfer from CeO_2 to Co-Bi , and it is also found that there is a more negative shift of the $\text{CeO}_2/\text{Co-Bi}$ hybrid with the increase in the amount of CeO_2 (Fig. S7[†]). This phenomenon has been demonstrated in previous reports.^{24,26,50} Moreover, this modification in the electronic structure makes CeO_2 more acidic (Lewis acid) and thus facilitates the activation of H_2O molecules (Lewis base), benefiting the formation of OOH_{ad} on the surface of the composites and facilitating the OER process and thus enhancement of the efficiency of water oxidation.²⁴ Additionally, the increase in the charge conductivity of $\text{CeO}_2/\text{Co-Bi}$ after introducing CeO_2 indicates faster charge transfer during the catalytic reaction, improving the catalytic efficiency of water oxidation (see ESI Fig. S6[†]). After introducing the small CeO_2 nanoparticles, the hybrids display a larger surface area, smaller charge transfer, better charge conductivity and stronger synergistic effects. All these advantages of the hybrids together are responsible for the good OER catalytic activity of the new $\text{CeO}_2/\text{Co-Bi}$ catalysts.

Besides the catalytic activity, the stability of the catalyst, another major concern in designing cost-effective OER catalysts, was investigated as well. After 1000 CV cycles in 0.1 M PBS (Fig. S8[†]), there was almost no obvious loss in catalytic current (Fig. 5a), suggesting its superior stability. Additionally, the stability of the $20\text{CeO}_2/\text{Co-Bi}$ composite was determined by measuring chronoamperometric responses ($i-t$) at 1.65 V (vs. RHE) in 0.1 M PBS. During the measurements, the working electrode was continuously rotating at 1600 rpm to remove the generated O_2 bubbles. As shown in Fig. 5b, the stabilized current density indicates no obvious deactivation of the $20\text{CeO}_2/\text{Co-Bi}$ composite over 19 h. The durability test of the $20\text{CeO}_2/\text{Co-Bi}$ electrode under higher current density was also performed, as shown in Fig. S9.[†] The current density of the $20\text{CeO}_2/\text{Co-Bi}$ composite over 30 h was only slightly attenuated. The results suggest that the $20\text{CeO}_2/\text{Co-Bi}$ composite exhibits good catalytic stability under neutral conditions.

4. Conclusions

In summary, we have successfully synthesised $\text{CeO}_2/\text{Co-Bi}$ composites by facile chemical methods at room temperature. The larger surface area, smaller charge transfer, better charge conductivity and stronger synergistic effects obtained after

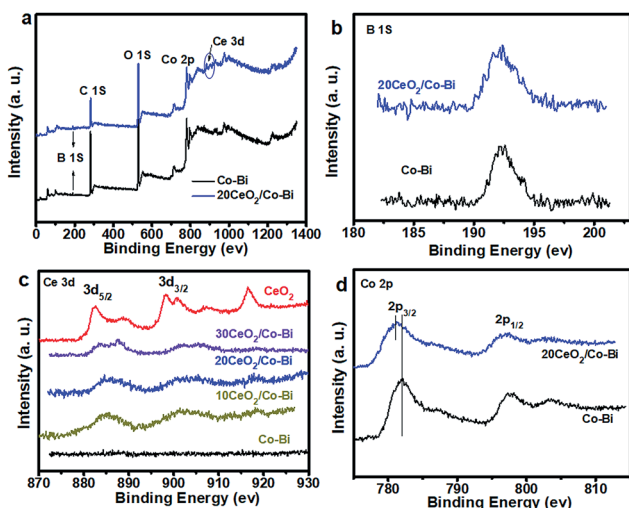


Fig. 4 XPS spectra of Co-Bi and $20\text{CeO}_2/\text{Co-Bi}$ composites (a), and high-resolution XPS spectra for B (b), Ce (c) and Co (d).



introducing CeO₂ nanoparticles are responsible for the good OER catalytic activity of these new CeO₂/Co–Bi composite catalysts. The 20CeO₂/Co–Bi composite exhibits outstanding OER catalytic activity with an overpotential of 453 mV at a current density of 10 mA cm^{−2}, Tafel slope of around 120 mV dec^{−1} and long-term stability in neutral medium, promising for practical application in the electrolysis of water.

Conflicts of interest

There are no conflicts to declare.

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Notes and references

- 1 T. R. Cook, D. K. Dogutan, S. Y. Reece, Y. Surendranath, T. S. Teets and D. G. Nocera, *Chem. Rev.*, 2010, **110**, 6474–6502.
- 2 M. G. Walter, E. L. Warren, J. R. McKone, S. W. Boettcher, Q. Mi, E. A. Santori and N. S. Lewis, *Chem. Rev.*, 2010, **110**, 6446–6473.
- 3 L. Hammarström and S. Hammes-Schiffe, *Acc. Chem. Res.*, 2009, 1859–1860.
- 4 R. F. Service, *Science*, 2009, **324**, 1257.
- 5 J. Li, M. Yan, X. Zhou, Z. Q. Huang, Z. Xia, C. R. Chang, Y. Ma and Y. Qu, *Adv. Funct. Mater.*, 2016, **26**, 6785–6796.
- 6 N. S. Lewis and D. G. Nocera, *Proc. Natl. Acad. Sci. U. S. A.*, 2006, **103**, 15729–15735.
- 7 W. Gao, M. Yan, H.-Y. Cheung, Z. Xia, X. Zhou, Y. Qin, C.-Y. Wong, J. C. Ho, C.-R. Chang and Y. Qu, *Nano Energy*, 2017, **38**, 290–296.
- 8 X. Zhou, Z. Xia, Z. Zhang, Y. Ma and Y. Qu, *J. Mater. Chem. A*, 2014, **2**, 11799–11806.
- 9 X. Zhou, Z. Xia, Z. Tian, Y. Ma and Y. Qu, *J. Mater. Chem. A*, 2015, **3**, 8107–8114.
- 10 R. Ge, M. Ma, X. Ren, F. Qu, Z. Liu, G. Du, A. M. Asiri, L. Chen, B. Zheng and X. Sun, *Chem. Commun.*, 2017, **53**, 7812–7815.
- 11 T. Liu, L. Xie, J. Yang, R. Kong, G. Du, A. M. Asiri, X. Sun and L. Chen, *ChemElectroChem*, 2017, **4**, 1840–1845.
- 12 L. Xie, R. Zhang, L. Cui, D. Liu, S. Hao, Y. Ma, G. Du, A. M. Asiri and X. Sun, *Angew. Chem., Int. Ed.*, 2017, **56**, 1064–1068.
- 13 P. Chen, K. Xu, T. Zhou, Y. Tong, J. Wu, H. Cheng, X. Lu, H. Ding, C. Wu and Y. Xie, *Angew. Chem., Int. Ed.*, 2016, **55**, 2488–2492.
- 14 X. Ji, L. Cui, D. Liu, S. Hao, J. Liu, F. Qu, Y. Ma, G. Du, A. M. Asiri and X. Sun, *Chem. Commun.*, 2017, **53**, 3070–3073.
- 15 M. W. Kanan and D. G. Nocera, *Science*, 2008, **321**, 1072–1075.
- 16 P. Li, R. Zhao, H. Chen, H. Wang, P. Wei, H. Huang, Q. Liu, T. Li, X. Shi, Y. Zhang, M. Liu and X. Sun, *Small*, 2019, **15**, 1805103.
- 17 D. G. Nocera, *Acc. Chem. Res.*, 2012, **45**, 767–776.
- 18 D. G. Nocera, *Acc. Chem. Res.*, 2017, **50**, 616–619.
- 19 M. Dincă, Y. Surendranath and D. G. Nocera, *Proc. Natl. Acad. Sci. U. S. A.*, 2010, **107**, 10337–10341.
- 20 Y. Surendranath, M. W. Kanan and D. G. Nocera, *J. Am. Chem. Soc.*, 2010, **132**, 16501–16509.
- 21 L. Xie, F. Qu, Z. Liu, X. Ren, S. Hao, R. Ge, G. Du, A. M. Asiri, X. Sun and L. Chen, *J. Mater. Chem. A*, 2017, **5**, 7806–7810.
- 22 L. Yang, D. Liu, S. Hao, R. Kong, A. M. Asiri, C. Zhang and X. Sun, *J. Mater. Chem. A*, 2017, **5**, 7305–7308.
- 23 C. Xie, Y. Wang, D. Yan, L. Tao and S. Wang, *Nanoscale*, 2017, **9**, 16059–16065.
- 24 Y. R. Zheng, M. R. Gao, Q. Gao, H. H. Li, J. Xu, Z. Y. Wu and S. H. Yu, *Small*, 2015, **11**, 182–188.
- 25 J. X. Feng, S. H. Ye, H. Xu, Y. X. Tong and G. R. Li, *Adv. Mater.*, 2016, **28**, 4698–4703.
- 26 Z. Liu, N. Li, H. Zhao, Y. Zhang, Y. Huang, Z. Yin and Y. Du, *Chem. Sci.*, 2017, **8**, 3211–3217.
- 27 W. Gao, D. Wen, J. Ho and Y. Qu, *Mater. Today Chem.*, 2019, **12**, 266–281.
- 28 Y. C. Zhu, Y. Bando and R. Z. Ma, *Adv. Mater.*, 2003, **15**, 1377–1379.
- 29 X. Liu, W. Wei, Q. Yuan, X. Zhang, N. Li, Y. Du, G. Ma, C. Yan and D. Ma, *Chem. Commun.*, 2012, **48**, 3155–3157.
- 30 P. Chen, K. Xu, Y. Tong, X. Li, S. Tao, Z. Fang, W. Chu, X. Wu and C. Wu, *Inorg. Chem. Front.*, 2016, **3**, 236–242.
- 31 Z. Zhang, W. Sigle, M. Rühle, E. Jud and L. J. Gauckler, *Acta Mater.*, 2007, **55**, 2907–2917.
- 32 W. Weber, K. Hass and J. McBride, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1993, **48**, 178.
- 33 J. E. Spanier, R. D. Robinson, F. Zhang, S.-W. Chan and I. P. Herman, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2001, **64**, 245407.
- 34 Z. Wu, M. Li, J. Howe, H. M. Meyer III and S. H. Overbury, *Langmuir*, 2010, **26**, 16595–16606.
- 35 F. Zhang, S.-W. Chan, J. E. Spanier, E. Apak, Q. Jin, R. D. Robinson and I. P. Herman, *Appl. Phys. Lett.*, 2002, **80**, 127–129.
- 36 Z. Dohčević-Mitrović, M. Šćepanović, M. Grujić-Brojčin, Z. Popović, S. Bošković, B. Matović, M. Zinkevich and F. Aldinger, *Solid State Commun.*, 2006, **137**, 387–390.
- 37 H. S. Ahn and T. D. Tilley, *Adv. Funct. Mater.*, 2013, **23**, 227–233.
- 38 K. Jin, J. Park, J. Lee, K. D. Yang, G. K. Pradhan, U. Sim, D. Jeong, H. L. Jang, S. Park and D. Kim, *J. Am. Chem. Soc.*, 2014, **136**, 7435–7443.
- 39 J. Park, H. Kim, K. Jin, B. J. Lee, Y.-S. Park, H. Kim, I. Park, K. D. Yang, H.-Y. Jeong and J. Kim, *J. Am. Chem. Soc.*, 2014, **136**, 4201–4211.
- 40 J. Tian, H. Li, A. M. Asiri, A. O. Al-Youbi and X. Sun, *Small*, 2013, **9**, 2709–2714.
- 41 Y. Wu, M. Chen, Y. Han, H. Luo, X. Su, M. T. Zhang, X. Lin, J. Sun, L. Wang and L. Deng, *Angew. Chem., Int. Ed.*, 2015, **54**, 4870–4875.



- 42 Y. Zhang, B. Cui, C. Zhao, H. Lin and J. Li, *Phys. Chem. Chem. Phys.*, 2013, **15**, 7363–7369.
- 43 D. Jeong, K. Jin, S. E. Jerng, H. Seo, D. Kim, S. H. Nahm, S. H. Kim and K. T. Nam, *ACS Catal.*, 2015, **5**, 4624–4628.
- 44 K. Jin, A. Chu, J. Park, D. Jeong, S. E. Jerng, U. Sim, H.-Y. Jeong, C. W. Lee, Y.-S. Park and K. D. Yang, *Sci. Rep.*, 2015, **5**, 10279.
- 45 P. Cai, J. Huang, J. Chen and Z. Wen, *Angew. Chem., Int. Ed.*, 2017, **129**, 4936–4939.
- 46 X. Cao, L. Cui, X. Wang, W. Yang and J. Liu, *ChemCatChem*, 2018, **10**, 2826–2832.
- 47 C. Tang, N. Cheng, Z. Pu, W. Xing and X. Sun, *Angew. Chem., Int. Ed.*, 2015, **127**, 9483–9487.
- 48 X. Lu and C. Zhao, *Nat. Commun.*, 2015, **6**, 6616.
- 49 X. Ren, R. Ge, Y. Zhang, D. Liu, D. Wu, X. Sun, B. Du and Q. Wei, *J. Mater. Chem. A*, 2017, **5**, 7291–7294.
- 50 M.-R. Gao, Y.-F. Xu, J. Jiang, Y.-R. Zheng and S.-H. Yu, *J. Am. Chem. Soc.*, 2012, **134**, 2930–2933.

