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Introduction

Energy is an important need for the growth of human civilization. Currently, the world's energy demand is basically accomplished using fossil fuels (coal, petroleum and natural gases). Fossil fuels have the main problem that they are non-renewable and generate $CO₂$ and other greenhouse gases. There are other renewable energy sources like wind and hydropower, but they have high organization and maintenance costs.¹⁻⁷ Solar energy is a continuous and renewable source of energy. At present, silicon based solar cells are considered the main commercialized technology by which we generate electricity, but the fabrication cost of silicon solar cells is high.⁸ Dye sensitized solar cells (DSSCs) have attracted much attention because of their low cost, environmental friendliness, high energy conversion efficiency, and simple fabrication procedure.⁹⁻¹⁷ A DSSC is constituted of a counter electrode which is generally made of Pt, a photoanode which is sensitized with a dye, and an electrolyte containing a redox couple (I_3^-/I^-) . Platinum is a costly, low abundance noble metal and it suffers from slow dissolution because of the corrosive redox electrolyte, which decays its longterm stability. These are the drawbacks that prevent the largescale application of the Pt electrode in DSSCs.¹⁸⁻²⁰ At present,

Fabrication of a counter electrode for dyesensitized solar cells (DSSCs) using a carbon material produced with the organic ligand 2 methyl-8-hydroxyquinolinol (Mq)

Rahul Kumar[,](http://orcid.org/0000-0003-3212-561X) $\mathbf{D}^{\star ab}$ Veena Sahajwallab and Parag Bhargava^a

Dye sensitized solar cells (DSSCs) are low cost solar cells and their fabrication process is easy relative to silicon based solar cells. Platinum can be replaced with carbon materials as counter electrodes in DSSCs because of their good catalytic properties and low cost. A carbon material was produced by carbonization of an organic ligand (2 methyl 8-hydroxy quinolinol (Mq)) at high temperature in flowing argon gas. Polyvinylpyrrolidone (PVP) was used as a surfactant for making carbon slurry from carbon produced using Mq. For the fabrication of the counter electrode, a carbon coating was prepared by using the doctor blading technique and the carbon slurry was coated on the FTO substrate. DSSCs based on the carbon counter electrode exhibit a higher $V_{\rm oc}$ of 0.75 V than that of the Pt counter electrode (0.69 V). DSSCs based on the carbon material showed a power conversion efficiency (PCE) of 4.25% and fill factor (FF) of 0.51 which are slightly lower than those of the platinum (Pt) based counter electrode which showed a PCE of 5.86% and FF of 0.68. PAPER

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a vital problem is a search for economical and highly catalytic materials to supersede Pt for DSSC development.²¹–²⁴ Some other low cost materials, such as carbon materials,²⁵⁻³⁴ alloys,³⁵ conductive polymers,³⁶⁻³⁹ metal carbides,⁴⁰ metal oxides,^{41,42} and transition metal based materials including metal nitrides,⁴³⁻⁴⁶ metal sulphides and metal alloys,⁴⁷⁻⁴⁹ have been utilized as counter electrodes and shown very good catalytic activity for the triiodide reduction. Carbon materials are very attractive materials to supersede the platinum electrode in DSSCs because of their high electronic conductivity, large surface area, low cost, environmental friendliness, corrosion resistance towards iodine, high reactivity for triiodide reduction, and availability. Several carbon materials such as graphite, graphene, carbon black, glassy carbon, carbon nanotubes, activated carbon, nanocarbon, carbon fibers, well-ordered mesoporous carbon, carbon produced from discarded toners of printer, carbon quantum dot, fullerene, conductive carbon paste, carbon derived from human hair, graphene composites, N and P doped graphene, 3D SWCNTs/graphene aerogel carbon derived from saccharide and sugar-free, carbon derived from organic ligands, carbon derived from Triton-X, etc. have been successfully employed as counter electrodes.²⁵⁻³⁴ Organic compounds, which have carbon (C) hydrogen (H), and nitrogen (N) atoms, are converted into graphitized carbons after carbonization at high temperature.³⁴ Nitrogen allows one to obtain strong and thermally stable carbon materials and may participate in the final formation of oriented graphite compounds.⁵⁰ It is well known that carbons derived from N containing compounds normally

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a Department of Metallurgical Engineering and Materials Science, Indian Institute of Technology Bombay, Mumbai, India 400076. E-mail: kumarrahul003@gmail.com b Centre for Sustainable Materials Research and Technology, School of Materials Science and Engineering, University of New South Wales, Sydney, NSW, 2052,

have abundant porosity, tunable surface properties and high stability.⁵¹ In this work, the preparation of the counter electrode was done by using a carbon material produced by carbonization of an organic ligand (2-methyl-8-hydroxy quinolinol (Mq)) at 1200 \degree C in flowing argon gas. It was expected that from the carbonization of the organic ligand, a good quality carbon material would be obtained.

Experimental

Materials

An organic ligand (2-methyl-8-hydroxy quinolinol, Sigma Aldrich 98%, chemical formula $C_{10}H_9NO$ was used as the carbon material. Fluorine doped tin oxide (FTO) (TEC8, sheet resistance 8-9 Ω \square^{-1} , Pilkington) was used as the substrate to prepare the photoanodes and counter electrodes. Polyvinylpyrrolidone (PVP), (Sigma Aldrich, K-32) was used as the surfactant and ethanol (Changshu Yangyuan Chemical, China AR Grade) was used as the solvent for the preparation of the carbon-based counter electrodes. TiO2 nanopowder (P25, Degussa) was used as the photocatalyst, while polyethylene glycol $(M_w = 600)$ PEG-600 (Thomas Baker) was used as the dispersant to prepare the photoanodes. $RuL_2(NCS)_2$ (L $= 2,2$ -bipyridyl-4,4-dicarboxylic acid) (known as N₃ dye) (Dyesol) was used as the photosensitizer for the fabrication of the cells. For the preparation of the electrolyte, guanidinium thiocyanate (0.10 M), 4-tert-butylpyridine (0.50 M) [TBP] (96% Aldrich), 1-methyl-3 propylimidazolium iodide [PMII] (\geq 98% Aldrich) (0.60 M) and iodine (LR, Thomas Baker) $[I_2]$ (0.03 M) were used.³¹⁻³⁴ Paper

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Synthesis of the carbon material and fabrication of the counter electrode

For the fabrication of the counter electrode for DSSCs, the carbon material was obtained by carbonization of the organic ligand (2 methyl-8-hydroxy quinolinol (Mq)) at 1200 \degree C in flowing argon gas. The carbon material appeared flaky in nature. To make the carbon material a finer powder it was crushed in a mortar and pestle. For the preparation of carbon slurry, initially, PVP was added to ethanol and kept on a stirrer for stirring for half an hour to make a homogeneous solution. After making the PVP solution, the carbon powder was mixed with the PVP solution and the carbon mixture was kept on a pot mill for 48 hours to reduce the deagglomeration of the carbon powder and make a homogeneous slurry. The PVP to carbon weight ratio was 1 : 8. Counter electrodes were prepared on an FTO substrate by coating the carbon slurry using the doctor blade technique. Scotch tape (3M) was used to coat the carbon slurry on the required area. After this, the carbon counter electrodes were dried in an oven at 50 \degree C and then placed in a furnace at 450 \degree C for 1 hour in flowing argon gas in order to burn out the binder. A sputter deposited platinum counter electrode was also fabricated for comparison with the carbon counter electrode.³¹⁻³⁴

Fabrication of photoanodes and dye loading

To fabricate the photoanodes, titania slurry was fabricated by roller milling of TiO₂ nanopowder, PEG 600 (dispersant) and ethanol (solvent) for 24 h. The prepared slurry has been used to

fabricate the photoanodes. The prepared slurry was coated onto the cleaned FTO substrates and sintering was done at $450\degree C$ for 1 h in flowing argon gas. After this, the electrodes were cooled to 80 °C and dipped into a 0.3 mM solution of N_3 dye in ethanol. The electrodes were removed from the dye solution after 24 h and then rinsed with ethanol and dried. It is expected that the amount of dye adsorbed by the photoanodes for both the devices would be the same as the photoanodes were prepared at the same time and using the same procedure.³¹⁻³⁴

Preparation of electrolyte

The electrolyte used in the DSSCs was fabricated by mixing of guanidinium thiocyanate (0.10 M), iodine (LR, Thomas Baker) $[I_2]$ (0.03 M), 4-tert-butylpyridine (0.50 M) [TBP] (96% Aldrich), 1methyl-3-propylimidazolium and iodide $[PMII]$ (\geq 98% Aldrich) (0.60 M), in the mixture of valeronitrile and acetonitrile (volume ratio: 15 : 85). The solution was kept on a magnetic stirrer for stirring for around 30 minutes.³¹⁻³⁴

Fabrication of DSSCs

For the preparation of DSSCs, counter electrodes and photoanodes were prepared separately as explained above. DSSCs were prepared using the counter electrode, electrolyte, photoanode, and spacer. A 25 µm spacer was used to combine both the electrodes together. The spacer was kept on the photoanode and heated to bond with it. The electrolyte was filled in the gap defined by the spacer and the carbon or platinum coated FTO substrate (counter electrode) was kept over it. Binder clips were used to hold the assembly together. The active area of the DSSCs was 0.25 cm^2 .³¹⁻³⁴

Characterization

The carbon material was produced by carbonization of the organic ligand (2-methyl-8-hydroxy quinolinol (Mq)) at 1200 °C. The produced material was characterized by using different techniques such as, Brunauer–Emmett–Teller (BET) (Smart Sorb 92/93, Smart Instruments Co.), X-ray powder diffraction (XRD) (Expert Pro, 40 kV, 30 mA, Panalytical), Raman spectroscopy (Jobin-Yvon, France Ramnor HG-2S Spectrometer), field emission gun electron transmission microscopy (FEG-TEM) (JEOL, JEM-2100F) and field emission gun electron scanning microscopy (FEG-SEM) (JEOL, JSM-7600F) independently.

Cyclic voltammetry (CV) of both the counter electrodes (Pt and carbon-based counter electrode) was performed using a three-electrode assembly at a scan rate of 50 mV $\rm s^{-1}$ from -0.6 to +1.2 V where the platinum or carbon coated FTO-substrate, Ag/AgCl and platinum wire worked as the working electrode, reference electrode and counter electrode respectively. The electrolyte solution consisted of 0.5 mM iodine, 0.1 M lithium perchlorate and 5 mM LiI in acetonitrile.³¹⁻³⁴

The electrochemical impedance spectra (EIS) of the DSSCs were recorded in a frequency range of 100 kHz to 0.1 Hz under AM 1.5G simulated solar illumination at 100 mW $\rm cm^{-2}$. The ac amplitude and applied bias voltage were fixed at open circuit voltage (V_{oc}) at 5 mV and 0 mV respectively. The EIS spectra of DSSCs were recorded using Zsimpwin software using two different equivalent circuit models. The thickness of the platinum films and $TiO₂$ film was measured using a profilometer (Dektak XT profilometer, Icon Analytical Equipment Pvt. Ltd.).

The photo–current–voltage $(I-V)$ characteristics of the DSSCs were measured by using a Keithley model 2420 source measure unit. The irradiation source was a 150 W xenon lamp on a Newport solar simulator with AM 1.5G. The active area of the DSSCs was 0.25 cm^2 .³¹⁻³⁴

Results and discussion

The carbon material produced using Mq was examined by using different techniques such as BET, XRD, Raman spectroscopy, FEG-TEM, FEG-SEM and cyclic voltammetry (CV). The carbon material produced using Mq was characterized by the BET method and the surface area of the carbon material was obtained as 17 m 2 g $^{-1}$. XRD measurement was done with a scan rate of 2 degree per min with a Cu-Ka source. The XRD pattern of the carbon material is shown in Fig. 1. Two peaks were obtained, which are commensurate with the crystalline reflections from the (101) and (002) planes respectively. The maximum intensity at which 2θ values are 43.63/26.40 was obtained for the $(101)/(002)$ planes. The produced carbon material was characterized by Raman spectroscopy as shown in Fig. 2. Raman spectroscopy measurement was performed with a 514 nm excitation source with a scan rate of 2000 cm^{-1} min^{-1} . Two prominent bands were found in Raman spectra. One band was found at 1363 $\mathrm{cm}^{-1}\mathrm{,}$ and this band is known as the D band and the D band is a feature of disorder such as defects, vacancies, bond disruption, etc. made in the carbon lattice. We have observed one another band at 1581 cm^{-1} , and this band is known as the G band and the G band is a signature of the graphitic structure and is produced because of the in-plane vibrational modes of the $sp²$ bonded carbon atoms. Both the bands appear in the carbon-based materials.^{52–58} The I_D/I_G ratio of the carbon material was found to be 0.98. The crystallite size (La) was measured by using the wavelength power law formula of the produced carbon material as given below using eqn (1) .⁵⁹ Nanoscale Advances Awares Article. Published on 27 June 2019. The content of the published on 27 Hundred Unit (1974) and 27 June 2019. The published on 27 June 2019. The published on 27 June 2019. The published on 27 June

Fig. 1 XRD pattern of carbon produced from Mq.

Fig. 2 Raman spectra of carbon produced from Mq.

$$
La (nm) = (2.4 \times 10^{-10}) \lambda_{laser}^4 (I_D/I_G)^{-1}
$$
 (1)

where λ is the excitation laser wavelength in nm units, and I_D/I_G is the ratio of the D and G band intensity.

The crystallite size of the carbon was measured to be 19.49 nm. FEG-TEM was utilized to characterize the morphology and nature of the carbon material. Fig. 3(a and b) demonstrate that the carbon particles are spherical in shape as shown in the FEG-TEM micrographs. Fig. 3(c) shows the crystalline nature of the carbon material. The diffraction pattern (Fig. 3(d)) of the carbon material indicates polycrystalline nature.

The surface morphology of the carbon material and counter electrode after heat treatment at 450 °C (1 hour) in flowing argon gas was characterized by FEG-SEM as demonstrated in Fig. 4. The morphology of the produced carbon material can be seen in (Fig. 4(a and b)). Carbon particles produced using Mq seem to be a combination of spherical and elliptical particles. It was

Fig. 3 FEG-TEM micrographs (a 6 b) at low magnification and (c) at high magnification, and (d) the diffraction pattern of carbon produced from Mq.

Fig. 5 Cyclic voltammetry curves of the carbon counter electrode at a scan rate of 50 mV s⁻¹ and Pt counter electrodes at a scan rate of 50 mV s⁻¹ from -0.6 to +1.2 V in 5 mM Lil, 0.5 mM I₂ and 0.1 M LiClO₄ acetonitrile solution. Reference electrode: Ag/Ag⁺ in acetonitrile. Auxiliary electrode: Pt/FTO.

$$
I_3^- + 2e^- = 3I^- \tag{2}
$$

$$
3I_2 + 2e^- = 2I_3 \tag{3}
$$

Fig. 4 FEG-SEM micrographs (a) at low magnification and (b) at high magnification of carbon produced from Mq, (c) the upper surface of the carbon film, (d) side view of the carbon film sintered at 450 \degree C, and (e) & (f) the upper surface of the sputtered platinum film.

expected that the surfactant (PVP) is transformed into carbon after heat treatment. The film seems to be porous in character and pores were present nonuniformly within the film as demonstrated in Fig. $4(c)$. The thickness of the film was examined by FEG-SEM as demonstrated in Fig. 4(d) and found to be 4.79 µm. The thickness of the film was chosen on the basis of our previous study, resulting in a high efficiency of DSSCs.⁶⁰ The sputtered platinum film seems to be dense in nature as demonstrated in Fig. 4(e and f). The thickness of the platinum film was examined using a profilometer and it was found to be 40 nm.

The cyclic voltammograms (CV) of platinum and carbon electrodes are shown in Fig. 5. The CV of the counter electrodes was carried out using a three-electrode assembly at a scan rate of 50 mV s⁻¹ from -0.6 to +1.2 V where the platinum or carboncoated FTO substrate, Ag/AgCl, and platinum wire, worked as the working electrode, reference electrode and counter electrode respectively. The electrolyte solution was prepared from 5 mM LiI, 0.5 mM iodine, and 0.1 M lithium perchlorate in acetonitrile. Fig. 5 shows the CV of the I_2/I^- system for carbon and platinum electrodes. In CV curves we obtained two pairs of redox peaks and the peaks obtained on the positive side are considered as anodic peaks and the negative peaks are considered as cathodic peaks. In the CV curves, the anodic peaks correspond to the oxidation of iodide and triiodide and the cathodic peaks refer to the reduction of triiodide.

The relative negative pair is defined by redox reaction (2) and the positive one is defined by redox reaction (3) .

The carbon material produced using Mq exhibited both oxidation and reduction peaks which are comparable with those of the Pt electrode. The same behaviour of the carbon material signifies that the catalytic behavior of the carbon material is comparable to that of Pt. The carbon film made using carbon slurry has a porous structure made using the doctor blade technique and each carbon particle has catalytic sites which facilitate I_3^- reduction. The redox reaction reaction (eqn 2) which occurs at the counter electrode of a DSSC and examine the catalytic capability and performance of the counter electrode. The redox reaction (eqn 3) which occurs in CV because of the reduction of I_2 molecules.^{61,62}

The peak current density (I_p) and peak-to-peak separation (ΔE_p) are two important parameters for estimating the catalytic activities of the electrodes. A high I_p value represents the high catalytic activity of the CE and a low $\Delta E_{\rm p}$ represents a smooth redox reaction on the CE surface because ΔE_p is inversely proportional to the charge transfer rate; thus, the catalytic activity is enhanced with decreasing $\Delta E_{\rm p}$.⁶³⁻⁶⁶ It was observed that the Ip value of the redox peak for the carbon produced from Mq ($I_p = -2.90$ mA cm⁻²) is higher than that of the Pt CE ($I_p =$ -0.40 mA cm⁻²), and the ΔE_p value of the carbon produced from Mq ($\Delta E_p = 0.31$ V) is lower than that of the Pt CE ($\Delta E_p =$ 0.37 V) as demonstrated in Fig. 5. Both the values (ΔE_p and I_p) for the carbon produced from Mq indicate that the carbon material shows high redox activity towards I_3^- and I_2 even higher than that of Pt.⁶⁷ The comparison of $\Delta E_{\rm p}$ and $I_{\rm p}$ values for various carbon-based materials is given in Table 1.

To investigate the catalytic activity of the carbon material and Pt counter electrodes (CEs), we carried out electrochemical impedance spectra (EIS) measurements. Nyquist plots for the carbon and Pt are shown in Fig. 6. As can be observed from

Table 1 ΔE_p and I_p values for various carbon-based materials^a

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		Table 1 ΔE_p and I_p values for various carbon-based materials ^a		for EIS inserted in Fig. 6 and are listed in Table 2. The sum of R_s		
Carbon-based				R_{ct} , and R_{ct} is the total internal series resistance (R_{total}) . ⁷⁸ The		
material	$\Delta E_{\rm p}$ (V)	$I_{\rm p}$ (mA cm ⁻²)	Reference	total internal series resistance for DSSCs fabricated using		
				carbon and Pt is 54.5 Ω and 24.97 Ω respectively. It can be		
A-CNT	0.34	-0.95	68	observed that the Pt CE exhibits better performance. In both the		
F-MC	0.42	-2.82	69	experiments, the photoanode electrodes are the same. It was		
O-NCI	0.57	-1.70	70	observed that R_{ct_1} is affected by the counter electrodes, so the		
MWCNTs	0.77		71			
CВ	0.64		72	fitting value R_{ct} of the carbon-based CE is higher than that of Pt		
APNT	0.49		73	The smaller R_{ct} , (7.07 Ω) of the Pt CE as compared to that of the		
N-CCS	0.35		74	carbon CE (17.3 Ω) suggests that Pt has more capability for		
BCM-PW	0.58	-1.08	75	electrocatalytic reduction of I_3 ⁻ to I ⁻ ions in an electrolyte as		
BCM-FT	0.53	-1.57	75	compared to the carbon CE.		
BCM-FP	0.54	-1.49	75	The Tafel-polarization test was done to further study the		
BCM-PL	0.58	-1.19	75	electrocatalytic activities of platinum and carbon CEs. Fig. 7		
SDC	0.70		60			
Mq	0.31	-2.90	Present work	shows the Tafel curves of the carbon and platinum CEs, which were derived from symmetrical cells. These cells were prepared		
carbon (SDC).		nanotubes (MWCNTs), carbon black (CB), activated polypyrrole nanotubes (APNT), N-doped core-shell carbon spheres (N-CCS), biowaste derived carbon materials (BCM) (PW: Phoenix wood, FT: facial tissue, FP: filter paper, PL: palm leaf), and sucrose derived		thick Surlyn film. The symmetrical cells were prepared with the same electrolyte, which was used to fabricate DSSCs. The exchange current density (J_0) of the electron transfer kinetics at the electrolyte/CE interface under equilibrium conditions car be directly observed from the Tafel polarization curve. It can be obtained when the overpotential is zero from the intercept of		
		Fig. 6, the Nyquist plots have two dominant semicircles. ⁷⁶ The		the extrapolated linear region of the anodic or cathodic branch. ⁵⁴ The equation between J_0 and $R_{\rm ct}$ can be written as		
		circuit model, which was used to fit the experimental data is				
		shown in the inset of Fig. 6. The first semicircle, which can be		demonstrated in eqn (4):		
		seen in the high-frequency region (100-0.1 kHz), represents the				
		impedance of charge-transfer $(R_{ct}$ at the counter electrode and		$J_0 = RT/nFR_{\rm ct}$ $\left(4\right)$		
the first semicircle also represents the sheet resistance (R_s) , which is governed by the FTO/glass substrate and the thickness of CEs. The second semicircle, which can be seen in the middle- frequency range (100-1 Hz), is mainly contributed by the charge-transfer (R_{ct_2}) process at the TiO ₂ /dye/electrolyte inter- face. ⁷⁷ The constant phase element (CPE) is used to model the capacitance that originates at the electrode/electrolyte interface				where J_0 is the exchange current density on the electrode surface, R is the gas constant, T is the absolute temperature, R_c is the charge-transfer resistance, F is Faraday's constant and n is the number of electrons involved in the reduction of I_3 ⁻⁷⁹⁻⁸¹ β better catalytic activity is observed for a higher J_0 value. It can be observed that carbon CEs show a higher value than platinum		

^a Activated carbon nanotubes (A-CNT), functionalized mesoporous carbon (F-MC), organic nanocarbon ink (O-NCI), multiwalled carbon nanotubes (MWCNTs), carbon black (CB), activated polypyrrole nanotubes (APNT), N-doped core–shell carbon spheres (N-CCS), biowaste derived carbon materials (BCM) (PW: Phoenix wood, FT: facial tissue, FP: filter paper, PL: palm leaf), and sucrose derived carbon (SDC).

Fig. 6, the Nyquist plots have two dominant semicircles.⁷⁶ The circuit model, which was used to fit the experimental data is shown in the inset of Fig. 6. The first semicircle, which can be seen in the high-frequency region (100–0.1 kHz), represents the impedance of charge-transfer (R_{ct_1}) at the counter electrode and the first semicircle also represents the sheet resistance (R_s) , which is governed by the FTO/glass substrate and the thickness of CEs. The second semicircle, which can be seen in the middlefrequency range (100–1 Hz), is mainly contributed by the charge-transfer (R_{ct_2}) process at the TiO₂/dye/electrolyte interface.⁷⁷ The constant phase element (CPE) is used to model the capacitance that originates at the electrode/electrolyte interface because of the accumulation of ions at the electrode surface. R_s , R_{ct_1} , and R_{ct_2} represent the fitted values of the equivalent circuit

Fig. 6 Nyquist plots of electrochemical impedance spectra of DSSCs made using carbon produced from Mq and Pt measured at open circuit voltage (V_{oc}) from 100 kHz to 0.1 Hz under 1 sun illumination $(AM 1.5, 100 mW cm^{-2}).$

$$
J_0 = RT/nFR_{\rm ct} \tag{4}
$$

where J_0 is the exchange current density on the electrode surface, R is the gas constant, T is the absolute temperature, R_{ct} is the charge-transfer resistance, F is Faraday's constant and n is the number of electrons involved in the reduction of I_3 ⁻.⁷⁹⁻⁸¹ A better catalytic activity is observed for a higher J_0 value. It can be observed that carbon CEs show a higher value than platinumbased CEs. This implies that carbon CEs exhibit better catalytic activity. The CEs prepared using the carbon material show a higher J_0 value, and this may be due to a result of the higher active surface area of the carbon material. It is expected that CEs prepared using the carbon material produced from Mq will exhibit higher stability than the Pt CEs. It is already reported by many research groups that carbon-based materials are more stable than Pt as CEs in DSSC applications.⁸²⁻⁸⁵

The J–V characteristic curves of DSSCs using the carbon material and Pt counter electrodes are shown in Fig. 8. The photovoltaic performance parameters of the DSSCs are given in Table 3.

The power conversion efficiency (η) and fill factor (FF) of the DSSCs were calculated using eqn (5) and (6):

Table 2 The fitted EIS parameters of DSSCs with different counter electrodes fabricated using carbon and Pt

Device	$R_{\rm s}(\Omega)$	$R_{\rm ct.}(\Omega)$	$R_{\rm ct}(\Omega)$	$R_{\text{total}}(\Omega)$
Carbon	13.9	17.3	23.3	54.5
Platinum	9.2	7.07	8.7	24.97

Fig. 7 Tafel polarization curves of carbon and Pt counter electrodes obtained for symmetrical cells.

Fig. 8 Photocurrent–voltage characteristics of the cells made using the carbon and Pt counter electrodes with N_3 dye and TiO₂ paste.

$$
\eta\text{ }^{(\%)} = \frac{V_{\text{max}} \times J_{\text{max}} \times 100\%}{P_{\text{in}}} \tag{5}
$$

$$
FF = \frac{V_{\text{max}} \times J_{\text{max}}}{V_{\text{oc}} \times J_{\text{sc}}} \tag{6}
$$

where $J_{\rm max} \, (\rm{mA \ cm^{-2}})$ and $V_{\rm max} \, (\rm{V})$ are the current density and voltage at the point of maximum power output in the J–V curves, P_{in} is the light intensity of the incident light, V_{oc} is the opencircuit voltage (V) and $J_{\rm sc}$ is the short-circuit current density (mA $\rm cm^{-2})$ respectively.⁸⁶

The DSSCs prepared using the carbon material show a short circuit current density J_sc of 11.00 mA cm $^{-2}$, fill factor (FF) of 0.51, open circuit voltage V_{oc} of 0.75 V and conversion efficiency (η) of 4.25%. The DSSCs fabricated using platinum as the counter electrode showed a short circuit current density $J_{\rm sc}$ of 12.40 mA $\rm cm^{-2}$, fill factor (FF) of 0.68, open circuit voltage $V_{\rm oc}$ of 0.69 V and conversion efficiency (η) of 5.86% as demonstrated in Table 3. The carbon material exhibits a lower conversion efficiency than the Pt electrode although the carbon material exhibits higher redox reactivity. DSSCs fabricated from the carbon material also show a lower fill factor compared to the platinum-based cell

Table 3 Photovoltaic parameters of DSSCs with different counter electrodes fabricated using the carbon and Pt

Counter electrode	$J_{\rm sc}$ (mA $\rm cm^{-2})$	$V_{\rm oc}$ (V)	FF	η (%)
Platinum	12.40	0.69	0.68	5.86
Carbon derived from Mq	11.00	0.75	0.51	4.25

while the open circuit voltage V_{oc} is higher compared to that of the platinum-based cell. The higher V_{oc} in carbon based DSSC is probably due to the low charge recombination rate compared to that of the Pt based DSSC while the low $J_{\rm sc}$ and fill factor may be due to the short electron lifetime and high internal resistance from ion diffusion. The low short-circuit current $(I_{\rm sc})$ and higher $V_{\rm oc}$ correlate with the charge of the carbon atom. This may be attributed to electrostatic interactions between the carbon atom and I^- or I_3^- , and a higher concentration of mediator anions in proximity to the carbon surface, resulting in the increase of the regeneration and recombination rates.^{87,88} The lower fill factor and $J_{\rm sc}$ are responsible for the poor efficiency. Even though a higher catalytic activity was observed in the electrode fabricated using the produced carbon due to the highly porous film offering a higher effective surface area for the reduction of triiodide species, the resistance of the film or higher overpotential towards the redox reaction appears to be the dominating factor in deciding the cell performance. The efficiency of carbon-based DSSCs can be improved by increasing the thickness of the counter electrodes because a higher catalytic activity may be observed in the carbon-based electrode due to the highly porous film. The porosity of the carbon film will increase with increasing thickness and it will offer a higher effective surface area for the reduction of tri-iodide species.⁶⁰ Paper Maximum Maximum

Conclusions

A carbon material was produced by carbonization of an organic ligand (Mq) at high temperature in flowing argon gas. The carbon material showed a graphitic nature and exhibited catalytic properties. The carbon material was utilized to prepare the counter electrode in DSSCs. The cell prepared with the carbon material exhibited an open circuit voltage V_{oc} of 0.75 V, short circuit current density $J_{\rm sc}$ of 11.00 mA cm⁻², fill factor (FF) of 0.51 and conversion efficiency (η) of 4.25%. The DSSCs made using the carbon showed higher V_{oc} than Pt based DSSCs, and this may be due to the low recombination rate. The low $J_{\rm sc}$ and FF may be due to the higher overpotential towards the redox reaction or resistance of the carbon film. These factors seem to be the dominating factors in deciding cell performance. The efficiency of carbon based DSSCs can be enhanced with increasing the thickness of the counter electrodes.

Conflicts of interest

There are no conflicts to declare.

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Notes and references

- 1 J. C. Williams, *History of Energy*, http://www.fi.edu/learn/ case-files/energy.html/, accessed 20.04.13.
- 2 Energy Source Comparison, http://www.energy4me.org/ energy-facts/, accessed 05.02.13.
- 3 P. Wurfel, Physics of Solar Cells, Wiley-VCH, Darmstadt, 2005.
- 4 S. G. Awuah, J. Polreis, J. Prakash, Q. Qiao and Y. You, J. Photochem. Photobiol., A, 2011, 224, 116–122.
- 5 P. Joshi, L. Zhang, Q. Chen, D. Galipeau, H. Fong and Q. Qiao, ACS Appl. Mater. Interfaces, 2010, 2, 3572–3577.
- 6 P. Joshi, L. Zhang, D. Davoux, Z. Zhu, D. Galipeau, H. Fong and Q. Qiao, Energy Environ. Sci., 2010, 3, 1507–1510.
- 7 P. Poudel and Q. Qiao, Nanoscale, 2012, 4, 2826–2838.
- 8 M. Green, Prog. Photovoltaics, 2009, 17, 183–189.
- 9 B. O'Regan and M. Grätzel, Nature, 1991, 353, 737-740.
- 10 S. Yun, Y. Qin, A. R. Uhl, N. Vlachopoulos, M. Yin, D. Li, X. Han and A. Hagfeldt, Energy Environ. Sci., 2018, 11, 476– 526.
- 11 S. Yun, Y. Liu, T. Zhang and S. Ahmad, Nanoscale, 2015, 7, 11877–11893.
- 12 S. Yun, A. Hagfeldt and T. Ma, Adv. Mater., 2014, 26, 6210– 6237.
- 13 S. Yun, Y. Zhang, Q. Xu, J. Liu and Y. Qin, Nano Energy, 2019, 60, 600–619.
- 14 S. Yun, J. N. Freitas, A. F. Nogueira, Y. Wang, S. Ahmad and Z. S. Wang, Prog. Polym. Sci., 2016, 59, 1–40.
- 15 M. Freitag, J. Teuscher, Y. Saygili, X. Zhang, F. Giordano, P. Liska, J. Hua, S. M. Zakeeruddin, J. E. Moser, M. Grätzel and A. Hagfeldt, Nat. Photonics, 2017, 11, 372–378.
- 16 L. Zhang, X. Yang, W. Wang, G. G. Gurzadyan, J. Li, X. Li, J. An, Z. Yu, H. Wang, B. Cai, A. Hagfeldt and L. Sun, ACS Energy Lett., 2019, 4, 943–951.
- 17 H. Lindström, A. Holmberg, E. Magnusson, L. Malmqvist and A. Hagfeldt, J. Photochem. Photobiol., A, 2001, 145, 107– 112.
- 18 A. Hagfeldt and M. Grätzel, Acc. Chem. Res., 2000, 33, 269-277.
- 19 M. Grätzel, J. Photochem. Photobiol., C, 2003, 4, 145-153.
- 20 A. Hagfeldt, G. Boschloo, L. Sun, L. Kloo and H. Pettersson, Chem. Rev., 2010, 110, 6595–6663.
- 21 Z. H. Khan and X. Liu, J. Electron. Mater., 2019, 7, 4148-4165.
- 22 P. Poudel and Q. Qiao, Nano Energy, 2014, 4, 157–175.
- 23 A. Kay and M. Grätzel, Sol. Energy Mater. Sol. Cells, 1996, 44, 99–117.
- 24 H. Pettersson, T. Gruszecki, R. Bernhard, L. Haggman, M. Gorlov, G. Boschloo, T. Edvinsson, L. Kloo and A. Hagfeldt, Prog. Photovoltaics, 2007, 15, 113–121.
- 25 T. N. Murakami, S. Ito, Q. Wang, M. K. Nazeeruddin, T. Bessho, I. Cesar, P. Liska, R. Humphry-Baker, P. Comte, P. Péchy and M. Grätzel, J. Electrochem. Soc., 2006, 153(12), A2255–A2261.
- 26 S. Hou, X. Cai, Y. Fu, D. Wang, H. Wu, C. Zhang, Z. Chu and D. Zou, J. Mater. Chem., 2011, 21, 13776–13779.
- 27 W. J. Lee, E. Ramasamy, D. Y. Lee and J. S. Song, ACS Appl. Mater. Interfaces, 2009, 6, 1145–1149.
- 28 M. Wu, X. Lin, T. Wang, J. Qiub and T. Ma, Energy Environ. Sci., 2011, 4, 2308–2315.
- 29 L. Kavan, J. H. Yum, M. K. Nazeeruddin and M. Grattzel, ACS Nano, 2011, 5, 9171–9178.
- 30 C. Bu, Y. Liu, Z. Yu, S. You, N. Huang, L. Liang and X. Z. Zhao, ACS Appl. Mater. Interfaces Sci., 2013, 5, 7432– 7438.
- 31 R. Kumar, V. More, S. P. Mohanty, S. S. Nemala, S. Mallick and P. Bhargava, J. Colloid Interface Sci., 2015, 459, 146–150.
- 32 R. Kumar, P. Bhargava and J. Mat, Mater. Sci. Semicond. Process., 2015, 40, 331–336.
- 33 R. Kumar, S. S. Nemala, S. Mallick and P. Bhargava, Sol. Energy, 2017, 144, 215–220.
- 34 R. Kumar and P. Bhargava, J. Alloys Compd., 2018, 748, 905– 910.
- 35 N. A. M. Barakat, M. Shaheer Akhtar, A. Yousef, M. El-Newehy and H. Y. Kim, Chem. Eng. J., 2012, 9, 211–212.
- 36 Y. F. Lin, C. T. Li and K.-C. Ho, J. Mater. Chem. A, 2016, 4, 384–394.
- 37 C. R. Ke, C. C. Chang and J. M. Ting, J. Power Sources, 2015, 284, 489–496.
- 38 L. Chen, J. Jin, X. Shu and J. Xia, J. Power Sources, 2014, 248, 1234–1240.
- 39 K. H. Park, S. J. Kim, R. Gomes and A. Bhaumik, Chem. Eng. J., 2015, 260, 393–398.
- 40 M. Wu, X. Lin, Y. Wang, L. Wang, W. Guo, D. Qi, X. Peng, A. Hagfeldt, M. Grätzel and T. Ma, J. Am. Chem. Soc., 2012, 134, 3419–3428. Nanoscale Advances
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	- 41 G. H. Guai, M. Y. Leiw, C. M. Ng and C. M. Li, Adv. Energy Mater., 2012, 2, 334–338.
	- 42 M. Wu, X. Lin, A. Hagfeldt and T. Ma, Chem. Commun., 2011, 47, 4535–4537.
	- 43 Q. W. Jiang, G. R. Li, S. Liu and X. P. Gao, J. Phys. Chem. C, 2010, 114, 13397–13401.
	- 44 G. R. Li, J. Song, G. L. Pan and X. P. Gao, Energy Environ. Sci., 2011, 4, 1680–1683.
	- 45 Q. W. Jiang, G. R. Li and X. P. Gao, Chem. Commun., 2009, 6720.
	- 46 G. Oskam, B. V. Bergeron, G. J. Meyer and P. C. Searson, J. Phys. Chem. B, 2001, 105, 6867–6873.
	- 47 W. S. Chi, J. W. Han, S. Yang, D. K. Roh, H. Lee and J. H. Kim, Chem. Commun., 2012, 48, 9501–9503.
	- 48 M. Zheng, J. Huo, Y. Tu, J. Wu, L. Hu and S. Dai, Electrochim. Acta, 2015, 173, 252–259.
	- 49 X. Cui, W. Xu, Z. Xie and Y. Wang, J. Mater. Chem. A, 2016, 4, 1908–1914.
	- 50 J. Kou and L. B. Sun, Ind. Eng. Chem. Res., 2016, 55, 10916– 10925.
	- 51 Yu. N. Sazanov, A. V. Gribanov and V. A. Lysenko, Fibre Chem., 2008, 4, 355–364.
	- 52 C. T. Hsieh, B. H. Yang and J. Y. Lin, Carbon, 2011, 49, 3092– 3097.
	- 53 R. Kumar and P. Bhargava, RSC Adv., 2016, 6, 8705–8713.
- 54 R. Kumar and P. Bhargava, Bull. Mater. Sci., 2017, 40, 1197– 1202.
- 55 R. Kumar, P. Bhargava, G. Chauhan and R. Srivastava, Adv. Sci. Lett., 2014, 20, 1001–1004.
- 56 R. Kumar, R. Srivastava, A. Kumar, M. N. Kamlasanan and K. Singh, Europhys. Lett., 2010, 90, 57004.
- 57 R. Kumar, B. Singh, A. Soam, S. Parida and P. Bhargava J, Diamond Relat. Mater., 2019, 94, 110–117.
- 58 R. Kumar and P. Bhargava, J. Asian Ceram. Soc., 2017, 5, 304– 312.
- 59 M. A. Pimenta, G. Dresselhaus, M. S. Dresselhaus, L. G. Cancado, A. Jorio and R. Saito, Phys. Chem. Chem. Phys., 2007, 9, 1276–1291.
- 60 R. Kumar, S. S. Nemala, S. Mallick and P. Bhargava, Opt. Mater., 2017, 64, 401–405.
- 61 A. I. Popov and D. H. Geske, J. Am. Chem. Soc., 1958, 80, 1340–1352.
- 62 K. Imoto, K. Takahashi, T. Yamaguchi, T. Komur, J. Nakamura and K. Murata, Sol. Energy Mater. Sol. Cells, 2003, 79, 459–469. Paper

State Remote Article 2018. Note state Article 2018. At the Sylvester Article 2019. At the S
	- 63 Y. Lu, S. Yun, X. Zhou, Y. Hou, T. Zhang and J. Li, Electrochim. Acta, 2017, 242, 390–399.
	- 64 T. Zhang, S. Yun, X. Li, X. Huang, Y. Hou, Y. Liu, J. Li, X. Zhou and W. Fang, J. Power Sources, 2017, 340, 325–336.
	- 65 S. Yun, X. Zhou, Y. Zhang, C. Wang and Y. Hou, Electrochim. Acta, 2019, 309, 371–381.
	- 66 Y. Zhang, S. Yun, C. Wang, Z. Wang, F. Han and Y. Si, J. Power Sources, 2019, 423, 339–348.
	- 67 M. Durr, A. Yasuda and G. Nelles, Appl. Phys. Lett., 2006, 89, 061110.
	- 68 A. A. Menon, A. A. Arbab, I. A. Sahito, K. C. Sun, N. Mengal and S. H. Jeong, Sol. Energy, 2017, 150, 521–531.
	- 69 A. A. Menon, A. A. Arbab, I. A. Sahito, K. C. Sun, N. Mengal, K. C. Sun, M. B. Qadir, Y. S. Choi and S. H. Jeong, Electrochim. Acta, 2017, 234, 53–62.
	- 70 A. A. Arbab, A. A. Memon, K. C. Sun, J. Y. Choi, N. Mengal, I. A. Sahito and S. H. Jeong, J. Colloid Interface Sci., 2019, 539, 95–106.
- 71 X. Luo, Y. Zhang and S. H. Kim, Nano Lett., 2018, 13, 274– 287.
- 72 J. M. Kim and S. W. Rhee, Electrochim. Acta, 2012, 83, 264– 270.
- 73 M. Baro, J. Harjwani and S. Ramaprabhu, IEEE J. Photovolt., 2019, 9, 700–709.
- 74 W. Wang, J. Yao, X. Zuo and G. Li, Mater. Lett., 2018, 227, 172–175.
- 75 S. Xu, C. Liu and J. Wiezorek, Mater. Chem. Phys., 2018, 204, 294–304.
- 76 H. C. Sun, L. K. Pan, X. Q. Piao and Z. Sun, J. Mater. Chem. A, 2013, 1, 6388–6392.
- 77 S. Yun, M. Wu, Y. Wang, J. Shi, X. Lin, A. Hagfeldt and T. Ma, ChemSusChem, 2013, 6, 411–416.
- 78 A. B. Martinson, M. S. Goes, F. Fabregat Santiago, J. Bisquert, M. J. Pellin and J. T. Hupp, J. Phys. Chem. C, 2009, 113, 4015– 4021.
- 79 J. Wu, Z. Lan, J. Lin, M. Huang, J. Huang, L. Fan, G. Luo, Y. Lin, Y. Xie and Y. Wei, Chem. Soc. Rev., 2017, 46, 5975– 6023.
- 80 J. Ou, C. Gong, M. Wang, J. Xiang and J. Liu, Electrochim. Acta, 2018, 286, 212–218.
- 81 F. Mousavi, M. Shamsipur, A. Taherpour and A. Pashabadi, Electrochim. Acta, 2019, 308, 373–383.
- 82 S. Yun, P. D. Lund and A. Hinsch, Energy Environ. Sci., 2015, 8, 3495–3514.
- 83 S. Yun, M. Xu, Y. Wang, J. Shi, X. Lin, A. Hegfeldt and T. Ma, ChemSusChem, 2013, 6, 411–416.
- 84 S. Yun, H. Pu, J. Chen, A. Hegfeldt and T. Ma, ChemSusChem, 2014, 7, 442–450.
- 85 S. Yun, H. Zhang, H. Pu, J. Chen, A. Hegfeldt and T. Ma, Adv. Eng. Mater., 2013, 3, 1407–1412.
- 86 L. Han, N. Koide, Y. Chiba, A. Islam, R. Komiya, N. Fuke, A. Fukui and R. Yamanaka, Appl. Phys. Lett., 2005, 86, 213501.
- 87 S. Ghu, E. Bi, B. Fu, G. Dodbiba, T. Fujita, D. P. Wilkison, Y. Wei and B. Fang, Chem. Commun., 2017, 53, 5561–5564.
- 88 O. O. Ogunsolu, J. C. Wang and K. Hanson, Inorg. Chem., 2017, 56, 11168–11175.