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# Hydrogen generation from a sodium borohydride–nickel core@shell structure under hydrolytic conditions†

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Sodium borohydride (NaBH<sub>4</sub>) is an attractive hydrogen carrier owing to its reactivity with water: it can generate 4 equivalents of H<sub>2</sub> by hydrolysis (NaBH<sub>4</sub> + 4H<sub>2</sub>O → NaB(OH)<sub>4</sub> + 4H<sub>2</sub>). Since using NaBH<sub>4</sub> in the solid state is the most favorable way to achieve a high gravimetric hydrogen storage capacity (theoretical maximum of 7.3 wt%), we have investigated the possibility of developing a core@shell nanocomposite (NaBH<sub>4</sub>@Ni) where a metallic nickel catalyst facilitating the hydrolysis is directly supported onto NaBH<sub>4</sub> nanoparticles. Following our initial work on core–shell hydrides, the successful preparation of NaBH<sub>4</sub>@Ni has been confirmed by TEM, EDS, IR, XRD and XPS. During hydrolysis, the intimately combined Ni<sup>0</sup> and NaBH<sub>4</sub> allow the production of H<sub>2</sub> at high rates (e.g. 6.1 L min<sup>-1</sup> g<sup>-1</sup> at 39 °C) when water is used in excess. After H<sub>2</sub> generation, the spent fuel is composed of an aqueous solution of NaB(OH)<sub>4</sub> and a nickel-based agglomerated material in the form of Ni(OH)<sub>2</sub> as evidenced by TEM, XPS and XRD. The effective gravimetric hydrogen storage capacity of nanosized NaBH<sub>4</sub>@Ni has been optimized by adjusting the required amount of water for hydrolysis and an effective hydrogen capacity of 4.4 wt% has been achieved. This is among the best reported values.

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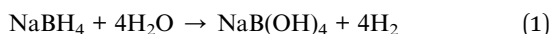
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## Introduction

Sodium borohydride NaBH<sub>4</sub> re-emerged about 20 years ago as an attractive hydrogen carrier. Nothing new had been discovered in comparison with the findings made by Schlesinger and co-workers during the World War II years,<sup>1</sup> but the current energy context has opened a new window of opportunity for a carrier that is able to readily generate H<sub>2</sub> under ambient conditions.<sup>2</sup>

Sodium borohydride offers several advantages. The four hydrides H<sup>δ-</sup> it carries readily react with protic hydrogen H<sup>δ+</sup> coming from water (or an alcohol like methanol). During hydrolysis,<sup>1</sup> the reaction is exothermic (Δ*H* = 210–270 kJ mol<sup>-1</sup>) and follows the common path:<sup>3</sup>



Interestingly, water contributes to generating half of the four molecules of H<sub>2</sub>. According to the stoichiometry shown by eqn (1), the theoretical gravimetric hydrogen storage capacity of the

couple NaBH<sub>4</sub>–H<sub>2</sub>O is as high as 7.3 wt%. The release of H<sub>2</sub> can be accelerated by an acid (e.g. formic and acetic acid) or a metal (e.g. cobalt and nickel)-based catalyst.<sup>4,5</sup> The only by-product that forms upon hydrolysis is sodium tetrahydroxyborate NaB(OH)<sub>4</sub> (also depicted as NaBO<sub>2</sub>·2H<sub>2</sub>O), a water-soluble compound (solubility of 3.5 mol L<sup>-1</sup>).<sup>6</sup> The released H<sub>2</sub> is relatively pure, albeit the presence of a small amount of borates in the steam simultaneously generated because of the reaction exothermicity.<sup>7</sup> For reaching high levels of H<sub>2</sub> purity, the evolving borates have to be trapped (with a “wash tank”, consisting of a flask filled with water) or filtered with the aid of a membrane.<sup>7–10</sup> Sodium borohydride has also few drawbacks that are related to: (i) the precipitation of borates, resulting in clogging of the piping, (ii) the exothermicity of the reaction requiring heat management, and (iii) the deactivation of metal-based catalysts after cyclic use.<sup>4,11–13</sup>

Generation of H<sub>2</sub> based on the hydrolysis of NaBH<sub>4</sub> can be designed in three distinct ways, depending on how the hydride is handled.<sup>14</sup> The most investigated design is based on the use of an alkaline aqueous solution of NaBH<sub>4</sub>. The hydrolysis reaction is started by putting the solution in contact with a heterogeneous metal-based catalyst. At the lab scale, the catalyst is generally introduced into the flask containing the NaBH<sub>4</sub> fuel.<sup>5,15</sup> At the demonstrator scale, the aqueous solution is stored in a tank and pumped into a catalytic reactor when H<sub>2</sub> is needed.<sup>8–11,16</sup> Considering this design, two important comments should be made. First, the large number of articles

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about  $\text{NaBH}_4$  hydrolysis available in the open literature has mainly focused on the synthesis, characterization and assessment of catalysts.<sup>5,17</sup> Second, because of the low solubility of  $\text{NaB(OH)}_4$  ( $3.5 \text{ mol L}^{-1}$ ) in water, the concentration of  $\text{NaBH}_4$  has to be kept at a maximum of  $3.5 \text{ mol L}^{-1}$  to avoid the precipitation of the by-product. This means that an excess of water has to be used in the order of  $15.9 \text{ mol per mol NaBH}_4$ , and thus a theoretical gravimetric hydrogen storage capacity of  $2.5 \text{ wt\%}$  can at best be achieved (the catalyst weight here is not considered).<sup>14</sup> In an alternative design, solid-state  $\text{NaBH}_4$  is put in contact with “catalytic” water (*e.g.* acetic acid solution or containing metal cations like  $\text{Co}^{2+}$ ) to initiate the hydrolysis reaction.<sup>4,18</sup> Solid-state  $\text{NaBH}_4$  can also be premixed with a solid-state metal-based catalyst and  $\text{H}_2$  generation is started by injecting water into the mixture.<sup>19</sup> Each of these three designs presents advantages and drawbacks,<sup>14</sup> but the direct reaction of solid-state  $\text{NaBH}_4$  with water in a stoichiometric amount (eqn (1)) is the best path to reach the theoretical gravimetric hydrogen storage capacity of  $7.3 \text{ wt\%}$  and we recently reported on a demonstrator to evaluate the potential of the approach.<sup>20</sup>

Taking into account that the hydride and catalyst should be intimately mixed in order to reach fast reaction rates whilst optimizing the amount of hydrogen generated, we investigated the potential of the reported  $\text{NaBH}_4$ @Ni core@shell approach in leading to improved  $\text{H}_2$  generation by hydrolysis. Herein, we thus report on the synthesis and characterization of the  $\text{NaBH}_4$ @Ni nanocomposite and its performance during hydrolysis. The post-hydrolysis state of the nickel catalyst and the by-product are also reported in the context of the possible regeneration of the hydride.

## Experimental

### Synthetic procedure

The synthesis of  $\text{NaBH}_4$ @Ni was performed in two stages as previously reported.<sup>21,22</sup> Here, the first step was modified to simplify the synthesis process of nanosized  $\text{NaBH}_4$  particles.

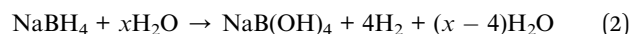
In the first step, nanoparticles of  $\text{NaBH}_4$  were prepared by dissolving  $200 \text{ mg}$  of  $\text{NaBH}_4$  (98%; Sigma-Aldrich) in  $10 \text{ mL}$  of anhydrous diethylene glycol dimethyl ether (diglyme; 99.5%; Sigma-Aldrich). Then,  $10 \text{ mg}$  ( $0.002 \text{ M}$ ) of tetraoctylammonium bromide (TOAB; 98%; Sigma-Aldrich), acting as a stabilizing agent, was dissolved under  $500 \text{ rpm}$  magnetic stirring at room temperature. The solution was aged for  $2 \text{ h}$  and placed under vacuum to remove the solvent. A white solid was obtained and characterized without further purification. The sample is denoted as **1**.

In the second step,  $200 \text{ mg}$  of **1** was suspended in  $20 \text{ mL}$  of anhydrous tetrahydrofuran (THF; HPLC grade; Fisher Scientific; dried using a LC Technology SP-1 Solvent Purification System), and  $18 \text{ mg}$  of tetrabutylammonium bromide (TBAB; 98%; Sigma-Aldrich) was added to stabilize the nanoparticles during coating. In parallel,  $100 \text{ mg}$  of nickel chloride  $\text{NiCl}_2$  (99.99%; Sigma-Aldrich) was suspended in  $10 \text{ mL}$  of THF. Both suspensions were placed under magnetic stirring at  $500 \text{ rpm}$  at room temperature. The  $\text{NiCl}_2$  solution appeared cloudy with a light yellow color. It was then slowly added to the suspension of **1** (at

a rate of  $500 \mu\text{L min}^{-1}$ ). The mixed solution turned grey after a few minutes, and was aged for approximately  $15 \text{ h}$ . The suspension was then centrifuged at  $13\,500 \text{ rpm}$  for  $30 \text{ min}$ . The resulting wet blackish solid was washed with THF and vacuum dried for at least  $5 \text{ h}$ . The as-obtained solid, *i.e.* the  $\text{NaBH}_4$ -supported Ni core@shell nanocomposite, was finally characterized without further purification. It is denoted as **2**.

### Catalytic tests

The  $\text{H}_2$  evolution (hydrolysis) experiments were performed as follows. In an argon-filled glove box (MBraun M200B;  $\text{O}_2 < 0.1 \text{ ppm}$  and  $\text{H}_2\text{O} < 0.1 \text{ ppm}$ ),  $35 \text{ mg}$  of **2** was weighed and transferred into a tubular Schlenk flask. The flask was connected to an inverted burette (filled with blue-colored water) *via* a cold trap (maintained at  $0 \text{ }^\circ\text{C}$ ), and immersed in a water bath (at  $10$  or  $17 \text{ }^\circ\text{C}$ ) or an oil bath (at  $25$ ,  $39$ ,  $53$  or  $60 \text{ }^\circ\text{C}$ ). The hydrolysis reaction was started by injecting onto **2** a given volume of water. The volume of water was determined according to the value of  $x$  as:



For the calculation of the water volume from  $x$ , it was assumed that the  $35 \text{ mg}$  of **2** were composed of  $23.5 \text{ mg}$  of  $\text{NaBH}_4$  and  $11.5 \text{ mg}$  of  $\text{NiCl}_2$  (in agreement with the proportions of each reactant during the synthesis). The catalytic tests were mainly performed with an excess of water (*i.e.*  $x = 64$ ), which implies a volume of  $720 \mu\text{L}$ . The experiments aiming at optimizing the effective gravimetric hydrogen storage capacity were carried out with a lower amount of water, namely  $x = 32$ ,  $16$ ,  $8$  and  $4$ , implying volumes of  $360$ ,  $180$ ,  $90$  and  $45 \mu\text{L}$ , respectively. Upon the injection of water,  $\text{H}_2$  rapidly evolved. The inverted burette was video-recorded during the catalytic test. At the end of the experiment, the video was computationally scrutinized and exploited to plot the volume of  $\text{H}_2$  as a function of time.

After the reaction, the nickel-based material and the spent fuel (aqueous solution of borates) were separated by centrifugation ( $18\,000 \text{ rpm}$ ;  $5 \text{ min}$ ). The former was subsequently washed two times with water and one time with ethanol (with intermediate centrifugation steps) and dried at  $60 \text{ }^\circ\text{C}$  before analysis. The as-obtained dry nickel-based material is denoted as **3**. Half of the spent fuel was also characterized in the solution-state. It is denoted as **4a**. The other half was subjected to water removal and drying at  $60 \text{ }^\circ\text{C}$  to lead to solid-state borate (denoted as **4b**).

### Characterization

**1** was analyzed by transmission electron microscopy (TEM; Philips CM200 operated at  $200 \text{ kV}$ ), infrared (IR) spectroscopy (Bruker Vertex 70V equipped with a Harrick diffuse reflectance infrared Fourier-transform spectroscope and a Praying Mantis accessory), and powder X-ray diffraction (XRD; PANalytical X'pert Multipurpose;  $\text{CuK}\alpha$  radiation with  $\lambda = 1.5406 \text{ \AA}$ ). For all of these analyses, the sample was protected from air and moisture. For the IR analysis, loading in an air-tight chamber was done in an argon-filled glove box (LC Technology Solutions



Inc.;  $O_2 < 1$  ppm,  $H_2O < 1$  ppm). For the XRD experiment, Kapton foil was used. The grid (carbon-coated copper) for TEM was prepared as follows. The sample was dispersed in cyclohexane (HPLC grade; Fisher Scientific; dried using a LC Technology SP-1 Solvent Purification System), sonicated, dropped onto the grid, and dried in a glove box. The transfer to the microscope was fast to minimize air exposure.

2 and 3 were analyzed by the aforementioned techniques as well as by energy dispersive X-ray spectroscopy (EDS; Philips CM200 operated at 200 kV) and X-ray photoelectron spectroscopy (XPS; ESCALAB250Xi from Thermo Scientific; monochromatic Al K $\alpha$  with 1486.7 keV, an X-ray source at 150 W

power and a spot size of 50  $\mu$ m; data analyzed using Advantage software). For all of these analyses, 2 was protected from air and moisture. The grid for EDS (and TEM) was realized as described in the previous paragraph. For XPS, a pellet was prepared inside the glove box, put in an argon-filled vial, and rapidly transferred into the instrument.

4a was analyzed by inductively coupled plasma optical emission spectrometry (ICP-OES; Perkin Elmer OPTIMA 7300) to detect whether nickel cations were present in the spent fuel. It was also analyzed by  $^{11}B$  nuclear magnetic resonance spectroscopy (NMR; Bruker AVANCE-300; probe head BBO10, 96.29 MHz, 30  $^{\circ}C$ ,  $D_2O$ ).

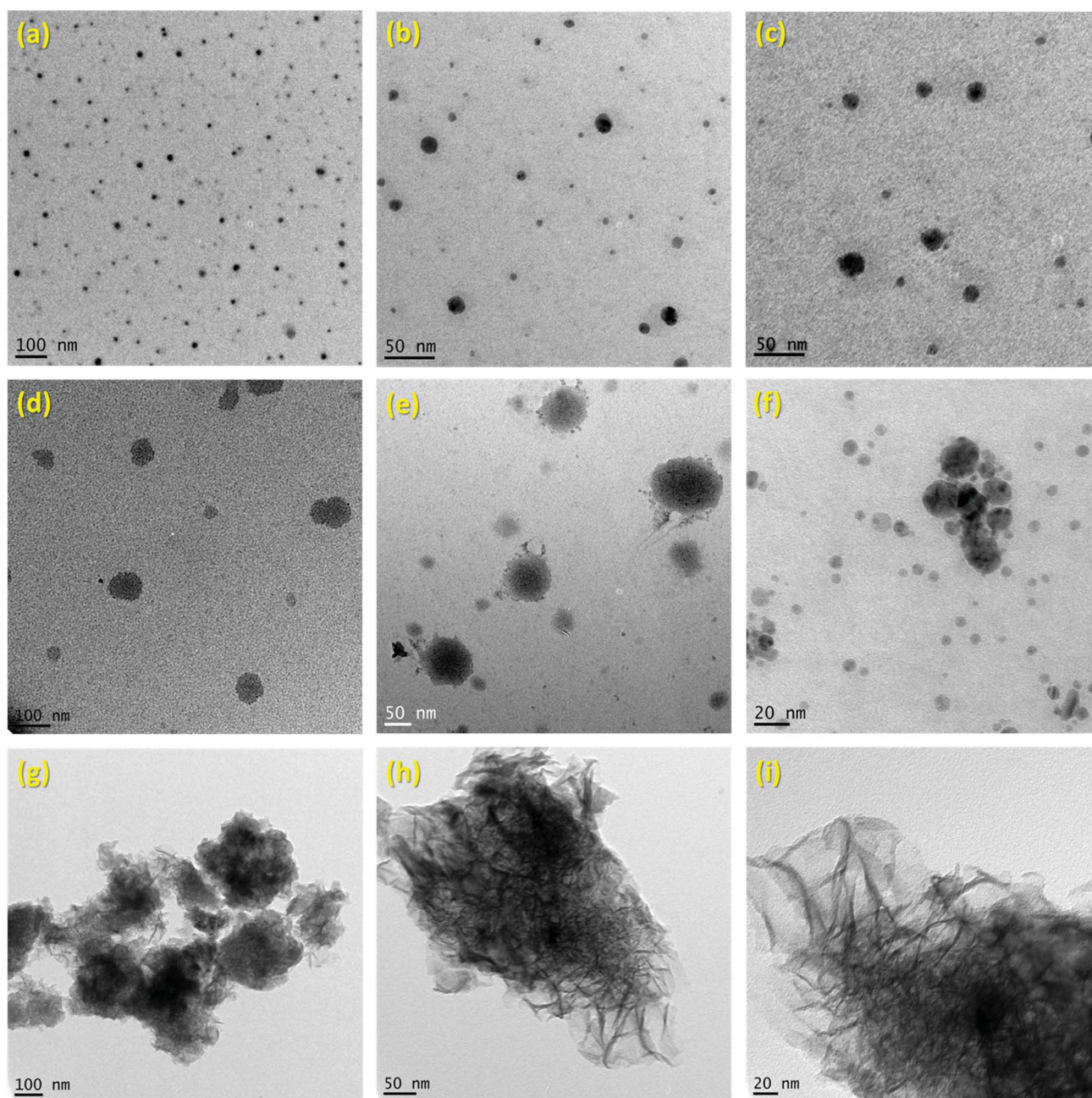


Fig. 1 TEM images of: (a–c) 1 *i.e.* the NaBH<sub>4</sub> nanoparticles; (d–f) 2 *i.e.* NaBH<sub>4</sub>@Ni; (g–i) 3 *i.e.* the nickel-based catalyst recovered after hydrolysis.



**4b** was analyzed by Fourier transform IR spectroscopy (Nicolet 710; 128 scans) and powder XRD (PANalytical X'Pert diffractometer equipped with an X'Celerator detector; CuK $\alpha$  radiation with  $\lambda = 1.5406 \text{ \AA}$ ).

## Results and discussion

### Syntheses

The successful preparation of **1** (*i.e.* the NaBH<sub>4</sub> nanoparticles) was first verified by TEM (Fig. 1). Isolated spherical nanoparticles with a size between 5 and 40 nm were obtained (Fig. S1<sup>†</sup>). This particle size distribution is narrower than the previously reported one (2–60 nm).<sup>21,22</sup> The molecular structure of **1** (Fig. 2) was verified by IR spectroscopy and was found to be consistent with that of bulk NaBH<sub>4</sub>. Vibration bands between 3000 and 2800 cm<sup>-1</sup> (C–H bonds) were also seen, which are due to the stabilizing agent TOAB. The crystal structure of **1** (Fig. 3) was favorably compared to that of cubic NaBH<sub>4</sub> ( $\alpha$  phase; ref. code 00-009-0386). A Scherrer analysis suggested crystallites with an average size of  $38 \pm 2 \text{ nm}$  and this may be associated

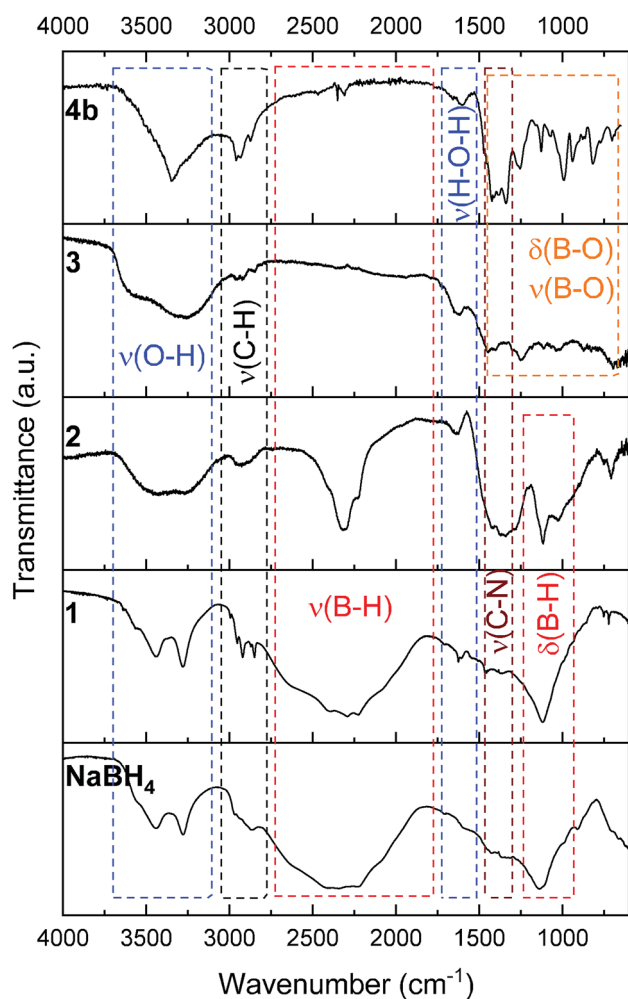


Fig. 2 IR spectra of (from bottom to top): bulk NaBH<sub>4</sub>; **1** (the NaBH<sub>4</sub> nanoparticles); **2** (NaBH<sub>4</sub>@Ni); **3** (the nickel-based catalyst recovered after hydrolysis); **4b** (the solid-state spent fuel).

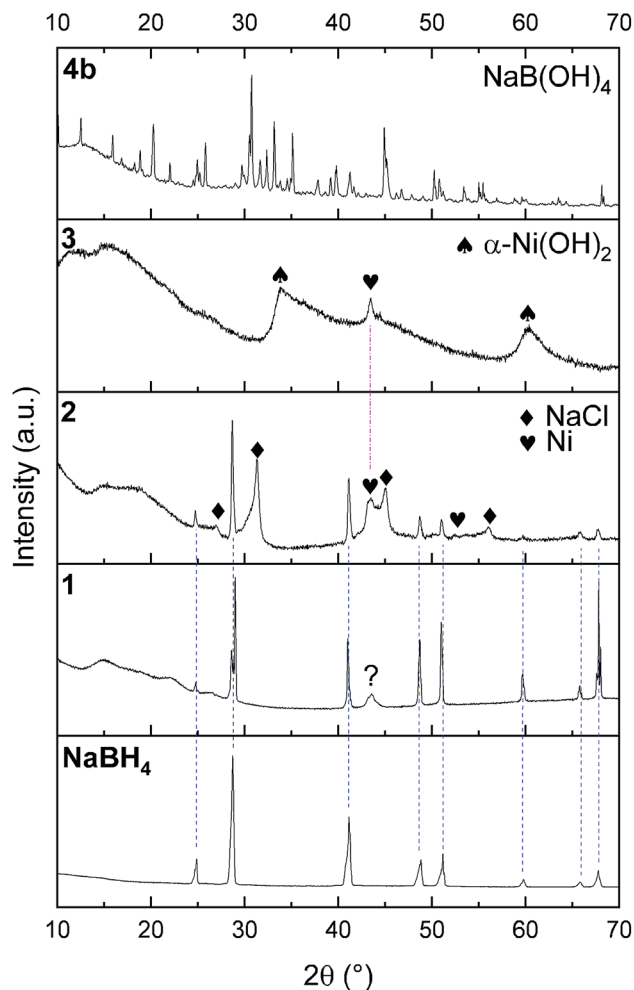


Fig. 3 XRD patterns of (from bottom to top): bulk NaBH<sub>4</sub>; **1** (the NaBH<sub>4</sub> nanoparticles); **2** (NaBH<sub>4</sub>@Ni); **3** (the nickel-based catalyst recovered after hydrolysis); **4b** (the solid-state spent fuel). The blue dashed lines indicate the diffraction peaks belonging to NaBH<sub>4</sub>. The pink dashed lines indicate the diffraction peaks belonging to Ni(OH)<sub>2</sub>. The other peaks (shown by symbols) are ascribed to Ni, NaCl and Ni(OH)<sub>2</sub>. The peak marked by “?” is unidentified.

with the larger NaBH<sub>4</sub> nanoparticles in the sample (Fig. S2<sup>†</sup>). As for the micro-sized NaBH<sub>4</sub> particles (*i.e.* bulk NaBH<sub>4</sub>),<sup>1–3</sup> **1** solubilizes in water (pH  $\geq 7$ ) and slowly evolves by spontaneous hydrolysis (eqn (1)). For having potential for hydrogen storage and generation, **1** has to be catalysed.<sup>4,5</sup>

**1** was thus used to prepare **2** (*i.e.* the NaBH<sub>4</sub>-supported Ni core@shell nanocomposite) by transmetalation, which is a sacrificial method by which the surface of the nanoparticles of **1** is coated with a nickel shell *via* partial consumption of the surface NaBH<sub>4</sub> molecules.<sup>21,22</sup> This is an effective strategy to support the catalyst onto the nanosized hydrogen carrier while putting in intimate contact both components. After the transmetalation operation, **2** showed a blackish color (*vs.* white for **1**) which is a visual indication of the reduction of Ni<sup>2+</sup> of NiCl<sub>2</sub> to Ni<sup>0</sup>. The successful preparation of **1** was scrutinized by TEM (Fig. 1). Spherical isolated nanoparticles (light grey) with a size between 11 and 119 nm were obtained (Fig. S3<sup>†</sup>). The size range



is larger than for **1**, implying coalescence of  $\text{NaBH}_4$  nanoparticles during the coating process.<sup>21</sup> This is in agreement with our previous report, but herein the particle size distribution is narrower (11–119 nm vs. 10–200 nm).<sup>22</sup> Darker small nanoparticles due to an element with higher electron density like Ni were observed while recovering the aforementioned spherical nanoparticles. By EDS (Fig. 4), the coexistence of both Na and Ni

within each of the composite nanoparticle was confirmed; the two elements were found to be homogeneously distributed, which indicates the possible formation of a Ni shell on the surface of the  $\text{NaBH}_4$  nanoparticles. The close vicinity of Na and Ni was also confirmed by XPS. Within the 10 nm penetration depth, which allowed analyzing the surface and subsurface of the nanoparticles of **2**, the elements Ni, B and Na were all detected. Even Br of TOAB and/or TBAB was detected (Fig. S4†). The binding energy (BE) of Ni 2p<sub>3/2</sub> is 852.4 eV (Fig. 5); this is typical of the metal  $\text{Ni}^0$  (confirming the reduction of  $\text{Ni}^{2+}$  by  $\text{NaBH}_4$ ).<sup>23</sup> For B 1s, two peaks were observed, at 188 and 192.5 eV (Fig. 6). The former BE is much likely due to  $\text{NaBH}_4$ ,<sup>24</sup> although nickel boride  $\text{Ni}_2\text{B}$  is at the same binding energy.<sup>23</sup> With respect

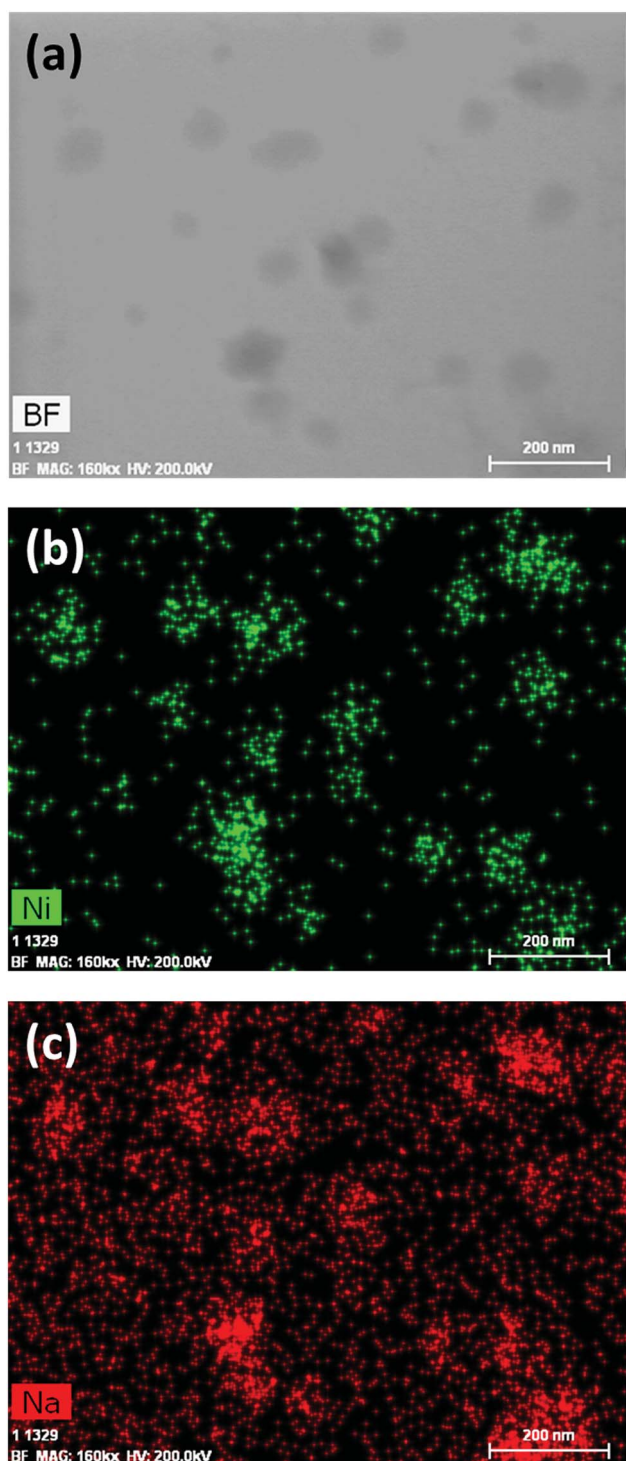


Fig. 4 EDS results of **2** ( $\text{NaBH}_4$ @Ni): (a) bright field (BF) image; (b) mapping of nickel (Ni); (c) mapping of sodium (Na).

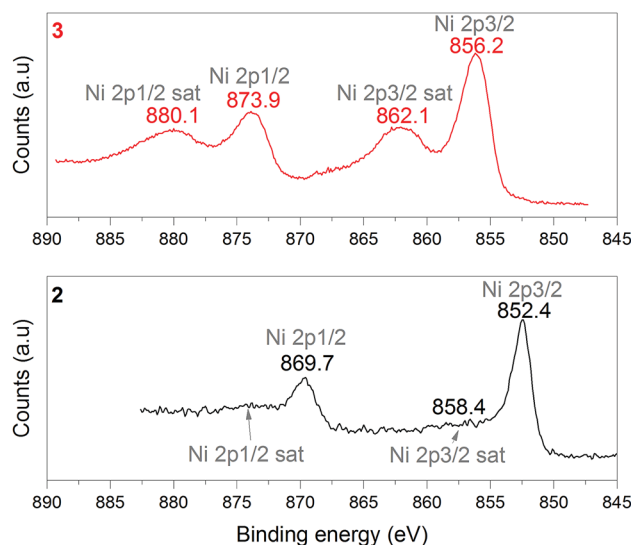


Fig. 5 XPS analyses of **2** ( $\text{NaBH}_4$ @Ni) and **3** (the nickel-based catalyst recovered after hydrolysis): focus on the Ni 2p spectra.

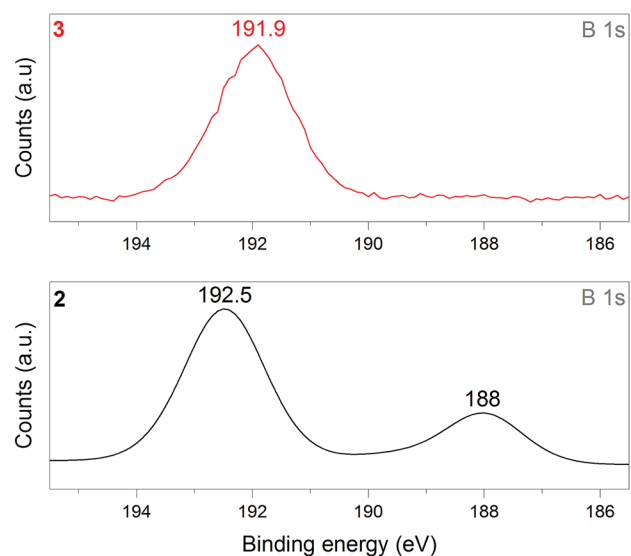


Fig. 6 XPS analyses of **2** ( $\text{NaBH}_4$ @Ni) and **3** (the nickel-based catalyst recovered after hydrolysis): focus on the B 1s spectra.



to the BE of 192.5 eV, it is generally attributed to B–O bonds,<sup>25</sup> but the formation of borates under our experimental conditions is unlikely (which is besides supported by the absence of oxidized nickel). Such a BE is also typical of B bound to Cl.<sup>24</sup> The formation of B–Cl bonds most likely occurs during the reduction of Ni<sup>2+</sup> of NiCl<sub>2</sub> by BH<sub>4</sub><sup>−</sup> of NaBH<sub>4</sub>.<sup>26</sup> With respect to Na 1s (Fig. 7), the BE of 1072.5 eV can be ascribed to Na<sup>+</sup> in NaBH<sub>4</sub> and NaCl.<sup>24</sup> Indeed, the XRD pattern of **2** (Fig. 3) shows diffraction peaks ascribed to NaBH<sub>4</sub> (ref. code 00-009-0386), NaCl (ref.

pattern 04-002-1178) and cubic Ni (ref. code 03-065-0380). The fingerprint of NaBH<sub>4</sub> was confirmed by IR spectroscopy (Fig. 2), discarding the presence of borate species. It is worth mentioning that the stretching bands of the B–H bonds are narrower than that for bulk NaBH<sub>4</sub>, which may be due to a lower degree of freedom because of the metallic shell (assimilated to a confinement effect).<sup>27</sup> Consistent with our previous achievement,<sup>21,22</sup> all of these results (*i.e.* small Ni<sup>0</sup> nanoparticles coating larger NaBH<sub>4</sub> spherical nanoparticles and close vicinity of Ni and Na as confirmed by both EDS and XPS) are in good agreement with the formation of **2**, a nanocomposite mainly consisting of a core of NaBH<sub>4</sub> and a shell of Ni<sup>0</sup>.

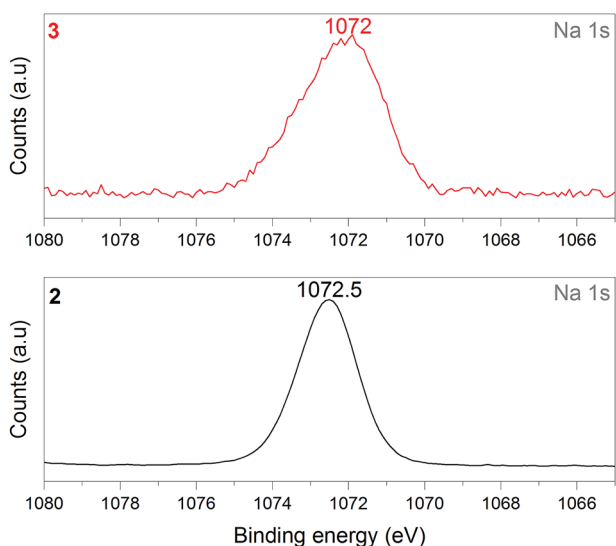


Fig. 7 XPS analyses of **2** (NaBH<sub>4</sub>@Ni) and **3** (the nickel-based catalyst recovered after hydrolysis): focus on the Na 1s spectra.

### Catalytic study

The preliminary H<sub>2</sub> evolution experiment was performed under conditions that are favorable for hydrolysis, namely at 39 °C and in the presence of an excess of water ( $x = 64$ ). As the 35 mg of **2** are theoretically composed of 23.5 mg of NaBH<sub>4</sub> and 5.3 mg of nickel (theoretical molar ratio NaBH<sub>4</sub>/Ni of 6.9), a total conversion of NaBH<sub>4</sub> (eqn (1)) should have produced 59.7 mL of H<sub>2</sub> (with  $V_m = 24 \text{ L mol}^{-1}$ ). The H<sub>2</sub> evolution curve (Fig. 8) however shows a final volume of 42 mL, which is equivalent to 70.4% of the theoretical target of 59.7 mL. The measured volume of 42 mL suggests therefore a molar ratio NaBH<sub>4</sub>/Ni of 4.9 (*i.e.* 70.4% of the aforementioned theoretical ratio of 6.9). In other words, 2 moles of NaBH<sub>4</sub> per mole of Ni are “missing” for H<sub>2</sub> generation. The value of 2 moles of NaBH<sub>4</sub> is in fact typical of the amount of NaBH<sub>4</sub> required to reduce a metal(II) cation like Ni<sup>2+</sup>.<sup>23,28</sup> Under our conditions, the synthesis of **2** implies thus the consumption of two equivalents of the starting NaBH<sub>4</sub> for

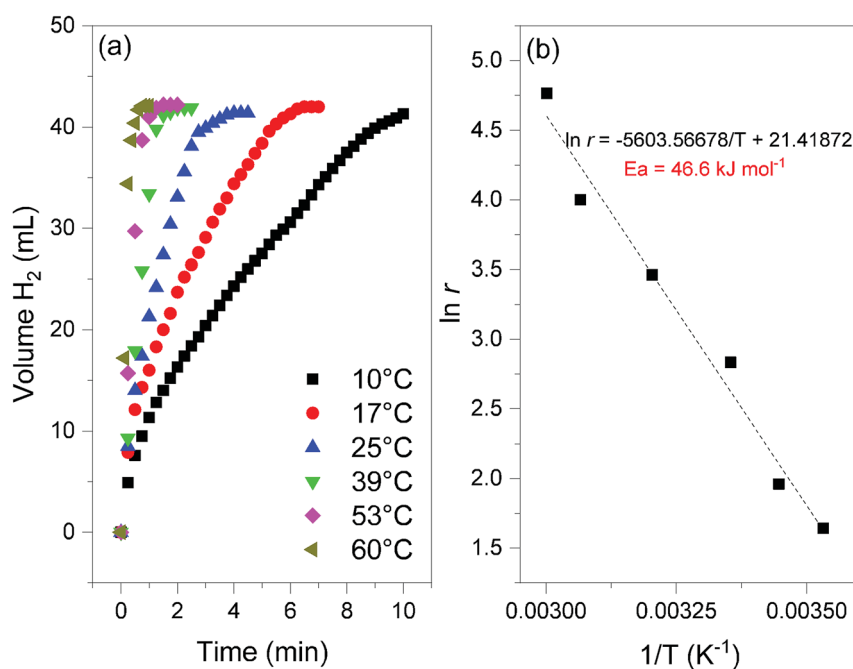


Fig. 8 (a) Hydrogen evolution curves of **2** (NaBH<sub>4</sub>@Ni; 35 mg) obtained at different temperatures (from 10 to 60 °C) and for  $x = 64$  (*cf.* eqn (2)). (b) Determination of the apparent activation energy ( $E_a$ , in  $\text{kJ mol}^{-1}$ ) from the slope of the linear evolution of the natural logarithm of the H<sub>2</sub> generation rate ( $\ln r$ , with  $r$  in  $\text{mL min}^{-1}$ ) as a function of the inverse of the temperature ( $T$  in Kelvin).



reducing one equivalent of  $\text{NiCl}_2$ . Accordingly one may assume that the content of unreacted  $\text{NaBH}_4$  in **2** is effectively 47.3 wt% (*i.e.* 70.4% of 23.5 mg  $\text{NaBH}_4$  in the weighed 35 mg of **2**). In view of this assumption, the weight of unreacted  $\text{NaBH}_4$  in our 35 mg of **2** would be 16.55 mg and a total conversion would result in the evolution of 42 mL (Fig. 8). The spent fuel **4a** (which is in the liquid state) was analyzed by  $^{11}\text{B}$  NMR to ensure the total conversion of  $\text{NaBH}_4$  (Fig. 9). The spectrum features two signals with positive chemical shifts belonging to  $\text{B}(\text{OH})_3$  under alkaline conditions ( $\delta = 11.9$  ppm) and  $\text{B}(\text{OH})_4^-$  ( $\delta = 4.2$  ppm).<sup>29</sup> These results confirm a total conversion of  $\text{NaBH}_4$  under our hydrolytic conditions. It is thus reasonable to conclude that each  $\text{Ni}^{2+}$  cation is reduced by two equivalents of  $\text{NaBH}_4$  for forming **2**, and that **2** is composed of 47.3 wt%  $\text{NaBH}_4$ . As mentioned above, this implies, for **2**, a  $\text{NaBH}_4 : \text{Ni}^0$  stoichiometry of 4.9 : 1 (which takes into account the amount of  $\text{NaBH}_4$  consumed for reducing  $\text{Ni}^{2+}$  to  $\text{Ni}^0$ ).

At 39 °C and in an excess of water ( $x = 64$ ), the hydrogen generation rate, calculated from the linear part of the curve, was determined to be  $31.9 \text{ mL min}^{-1}$ . Expressed per gram of Ni, the rate is  $6.1 \text{ L min}^{-1} \text{ g}^{-1}$ . This is one of the highest rates ever reported for a nickel catalyst and it is comparable to the rates reported for the best unsupported cobalt-based catalysts ( $5\text{--}10 \text{ L min}^{-1} \text{ g}^{-1}$ ).<sup>5,30,31</sup> The temperature was varied to tune the hydrogen generation rate (Fig. 8). It was found to be as high as  $22.5 \text{ L min}^{-1} \text{ g}^{-1}$  at 60 °C, and as low as  $1 \text{ L min}^{-1} \text{ g}^{-1}$  at 10 °C. In addition, rates of about 1.4, 3.3 and  $10.6 \text{ L min}^{-1} \text{ g}^{-1}$  were measured at 17, 25 and 53 °C respectively. These rates confirm the good to very good performance of **2**, over a wide range of temperature (10–60 °C).<sup>5,30,31</sup> With the help of the Arrhenius equation, the apparent activation energy was determined to be  $46.6 \text{ kJ mol}^{-1}$  (Fig. 8). This is consistent with the energies reported so far for the catalytic hydrolysis of  $\text{NaBH}_4$ .<sup>32</sup> The results discussed above show that **2** is an attractive and performing nanocomposite to be used as a catalyzed hydrogen carrier in a demonstrator

designed using solid-state  $\text{NaBH}_4$ . The total conversion of the borohydride and high hydrogen generation rates can be achieved.

### Post-hydrolysis analyses

After hydrolysis, the nickel-based solid **3** was separated from the liquid-state spent fuel **4a** by centrifugation. **3** was washed

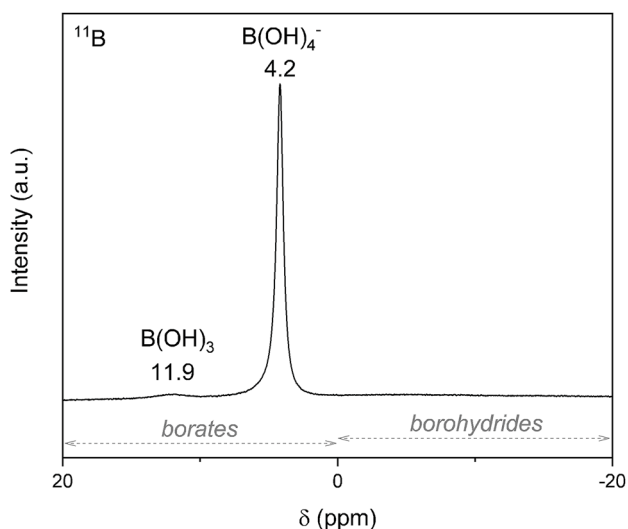


Fig. 9  $^{11}\text{B}$  NMR spectrum of **4a** (the liquid-state spent fuel). The typical chemical shift ranges for borohydrides and borates are indicated by the arrows.

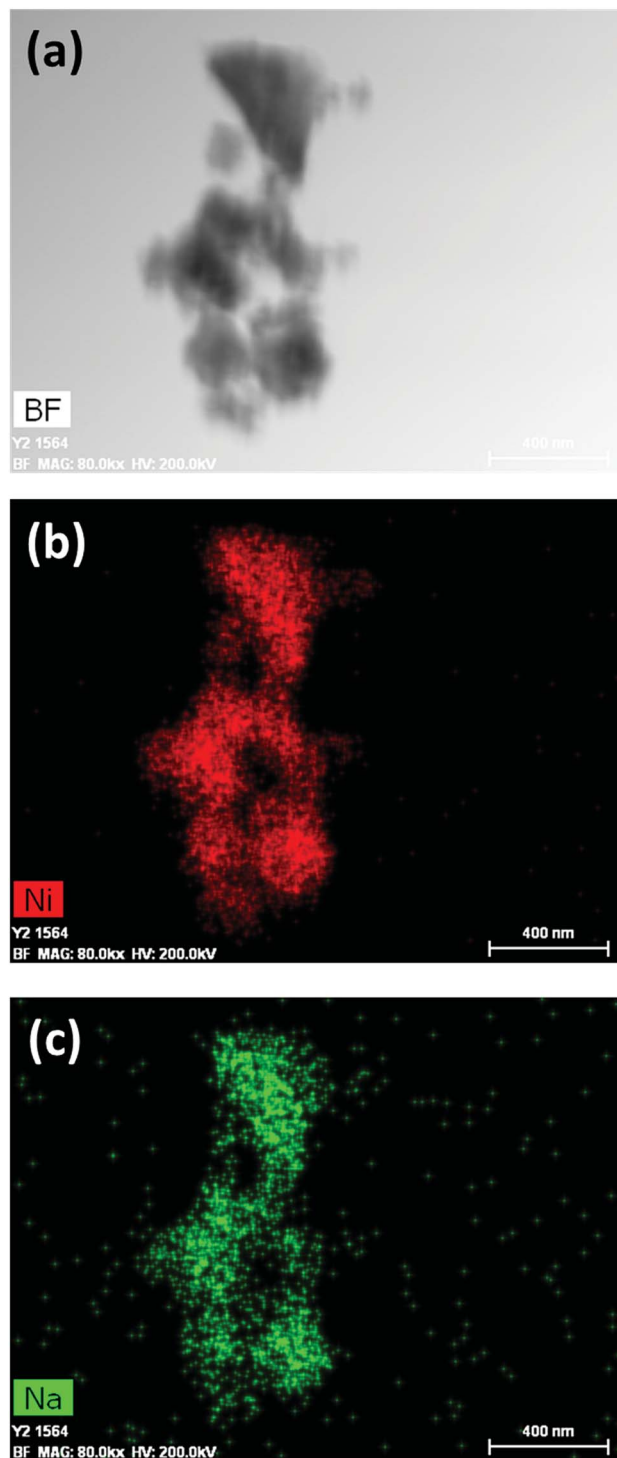


Fig. 10 EDS results of **3** (the nickel-based catalyst recovered after hydrolysis): (a) bright field (BF) image; (b) mapping of nickel (Ni); (c) mapping of sodium (Na).



several times and dried before analysis by TEM, EDS, XPS, IR and XRD. **3** consisted of big agglomerates (hundreds of nm) where the initial nanoparticles cannot be distinguished anymore (Fig. 1). The images show a difference of brightness suggesting a light-grey matrix of lighter elements (*i.e.* B and Na) surrounding the metal-based agglomerates. This behavior is typical of nickel- and cobalt-based catalysts upon hydrolysis.<sup>33,34</sup> Although initially they were in the form of nanoparticles, they generally form agglomerates of metal oxide/hydroxide surrounded by a matrix mainly containing hydrated sodium (poly-) borates ( $\text{Na}_n\text{B}_m\text{O}_p\text{H}_p \cdot q\text{H}_2\text{O}$ ).<sup>25</sup> By EDS (Fig. 10), the presence of nickel and sodium in **3** was confirmed. According to the XPS results (Fig. 5–7),<sup>24</sup> these elements are oxidized ( $\text{Ni}^{\text{II}}$  and  $\text{Na}^+$ ). The Ni 2p BEs (*i.e.* 2p<sub>3/2</sub>, 2p<sub>3/2</sub> sat, 2p<sub>1/2</sub> and 2p<sub>1/2</sub> sat) all suggest  $\text{Ni}(\text{OH})_2$  (Fig. 5), which is in agreement with the post-hydrolysis oxidation of nickel ( $\text{Ni}^0 \rightarrow \text{Ni}^{\text{II}}$ ) as reported elsewhere.<sup>35</sup> The Na 1s BE is typical of  $\text{Na}^+$  (Fig. 7), compensating the negative charge of  $\text{Cl}^-$  and/or that of borates. With respect to  $\text{Cl}^-$ , it (Cl 2p<sub>3/2</sub> BE of 198.3 eV) was detected by XPS (Fig. S5†). The borates were revealed by XPS (Fig. 6, with a B 1s BE of 191.9 eV)<sup>25</sup> as well as by IR analysis (Fig. 2) which shows vibration bands ascribed to B–O and O–H bonds in the hydrated sodium polyborates ( $\text{Na}_n\text{B}_m\text{O}_p\text{H}_p \cdot q\text{H}_2\text{O}$ ).<sup>36</sup> **3** was found to be mainly amorphous (Fig. 3),<sup>33,34</sup> with a few peaks belonging to the  $\alpha$  phase of  $\text{Ni}(\text{OH})_2$ ,<sup>37,38</sup> in agreement with the XPS analysis. It can thus be concluded that **3** consists of agglomerated  $\text{Ni}(\text{OH})_2$  particles surrounded by a matrix made of  $\text{Na}_n\text{B}_m\text{O}_p\text{H}_p \cdot q\text{H}_2\text{O}$  and NaCl.

As discussed above, the liquid-state spent fuel **4a** (Fig. 9) is an alkaline aqueous solution of  $\text{B}(\text{OH})_4^-$  in equilibrium with a small fraction of its acidic form  $\text{B}(\text{OH})_3$  ( $\text{pK}_a = 9.3$ ). The

solution was also subjected to chemical analysis by ICP-OES to check for the presence of  $\text{Ni}^{2+}$ . No trace of this cation was found (within the detection limit of the spectrometer). In other words, leaching of Ni of **2** did not take place under our hydrolysis conditions. Water of **4a** was then extracted. A white solid, **4b**, was obtained (*i.e.* the solid-state spent fuel) and analyzed. The IR spectrum (Fig. 2) was found to be typical of  $\text{NaB}(\text{OH})_4$ .<sup>36</sup> The vibration bands due to C–H stretching between 3000 and 2800  $\text{cm}^{-1}$  may be ascribed to TOAB and/or TBAB. The XRD pattern is comparable to the reference pattern of  $\text{NaB}(\text{OH})_4$  (ref. code. 04-011-2875). This is in good agreement with the occurrence of the reaction depicted by eqn (1).

### Hydrogen storage capacities

Additional  $\text{H}_2$  evolution experiments were performed with **2** to optimize the gravimetric hydrogen storage capacity of the couple  $\text{NaBH}_4\text{--H}_2\text{O}$ . To this end, the amount of  $\text{H}_2\text{O}$  (characterized by  $x$  in eqn (2)) was decreased from  $x = 64$  to  $x = 4$  while the temperature was 39 °C (Fig. 11).

Prior to the discussion about the storage capacities, two preliminary observations have to be reported. First, the final volume of  $\text{H}_2$  detected was 42 mL, and this suggested a total conversion of  $\text{NaBH}_4$  amounting to 47.3 wt% **2**. Second, the kinetics of  $\text{H}_2$  release was impacted by the water amount, especially for  $x < 16$ . The lower the value  $x$ , the longer the reaction. For example, the experiment was completed after 6 min for  $x = 4$  whereas it took  $\sim 2$  min for  $x \geq 16$ . The differences are much less important for  $x = 16, 32$  and 64. Hydrogen generation rates of 8.1, 17.1, 26.5, 27 and 31.9  $\text{mL min}^{-1}$  (or 1.6, 3.3, 5.1, 5.2 and 6.1  $\text{L min}^{-1} \text{g}^{-1}$ ) were determined for  $x = 4, 8, 16, 32$  and 64, respectively. By plotting the natural logarithm of

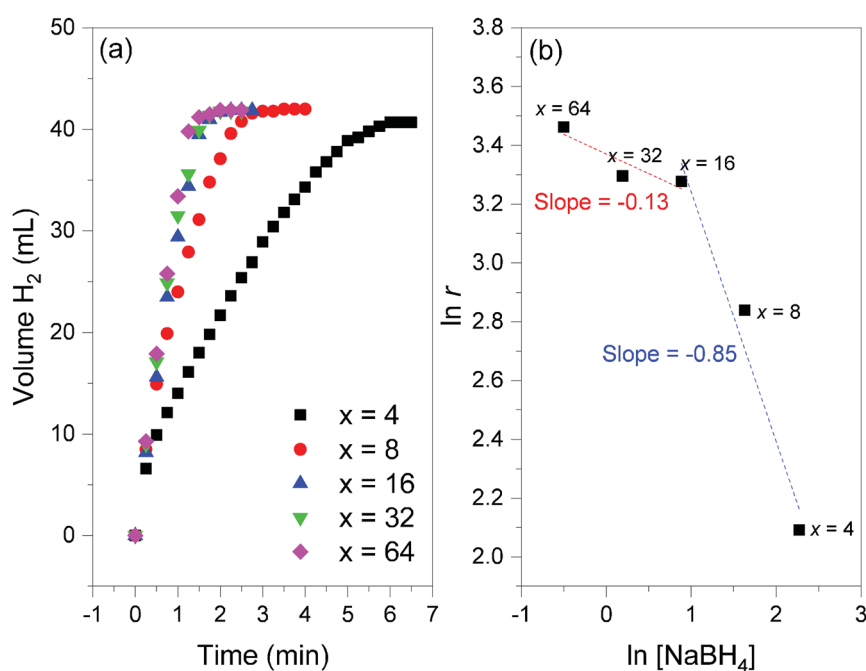


Fig. 11 (a) Hydrogen evolution curves of **2** ( $\text{NaBH}_4\text{@Ni}$ ; 35 mg) obtained at 39 °C and for different  $x$  values (from 4 to 64; *cf.* eqn (2)). (b) Determination of the reaction order *versus* the concentration of  $\text{NaBH}_4$  ( $[\text{NaBH}_4]$ ) from the slope of the linear evolution of natural logarithm of  $[\text{NaBH}_4]$  as a function of the natural logarithm of the  $\text{H}_2$  generation rates ( $\ln r$ ); the slopes are indicated.



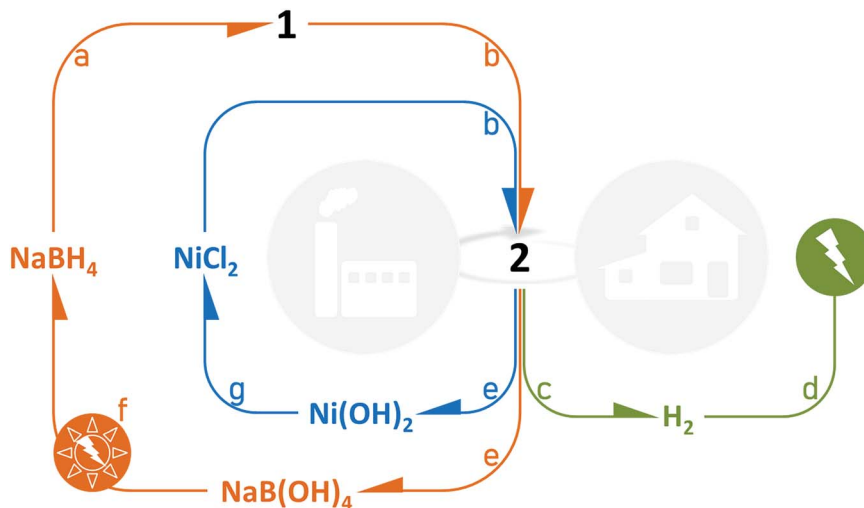


Fig. 12 Cycles with 2 ( $\text{NaBH}_4\text{@Ni}$ ): (a) industrial preparation of 1 from bulk  $\text{NaBH}_4$ ; (b) industrial preparation of 2 from 1 and  $\text{NiCl}_2$ ; (c) customer use of a cartridge/reactor of 2 for generating  $\text{H}_2$  by hydrolysis; (d) electricity generation by feeding a fuel cell with the as-produced  $\text{H}_2$ ; (e) industrial recovery and separation of the spent fuel (4a and 4b;  $\text{NaB(OH)}_4$ ) and the nickel-based solid (3;  $\text{Ni(OH)}_2$ ); (f) industrial recycling of  $\text{NaB(OH)}_4$  and regeneration of  $\text{NaBH}_4$  using renewable energy; (g) industrial formation of  $\text{NiCl}_2$  from  $\text{Ni(OH)}_2$ .

the rate as a function of the logarithm of the concentration of  $\text{NaBH}_4$  (Fig. 11), the reaction order *versus* the concentration of  $\text{NaBH}_4$  was found to be  $-0.13$  for  $x \geq 16$  ( $[\text{BH}_4^-] \leq 2.4 \text{ mol L}^{-1}$ ) and  $-0.85$  for  $x \leq 16$  ( $[\text{BH}_4^-] \geq 2.4 \text{ mol L}^{-1}$ ). A negative reaction order is generally explained by hardly desorbing borate by-products,<sup>39</sup> because of water deficit and the relatively low solubility of  $\text{NaB(OH)}_4$  ( $3.5 \text{ mol L}^{-1}$ ). This is in line with our results for  $x \leq 16$ . The near-zero order found for  $x \geq 16$  is quite typical of diluted  $\text{NaBH}_4$  solutions.<sup>32,40</sup> The catalytic behavior of  $\text{Ni}^0$  of 2 is thus comparable to that of previously reported metal-based catalysts.<sup>5,30,31,39–41</sup>

The best gravimetric hydrogen storage capacity was expected for the reaction involving the lower amount of  $\text{H}_2\text{O}$  ( $x = 4$ ). By taking into account the weight of 2, that of  $\text{H}_2\text{O}$ , and the content of  $\text{NaBH}_4$  in 2, an effective gravimetric hydrogen storage capacity of 4.4 wt% can be proposed. This is a high effective capacity.<sup>14</sup> As a comparison one may cite a commercial device based on an alkaline solution of 20 wt%  $\text{NaBH}_4$ : the theoretical gravimetric hydrogen storage capacity of the fuel is 4.2 wt%, whereas the weight of the catalyst is not taken into consideration in the calculation of the capacity.<sup>11</sup> It is worth mentioning that, under our conditions, when  $x = 8$ , the effective gravimetric hydrogen storage capacity is about 2.8 wt%, which is quite attractive when the hydrogen generation rate ( $3.3 \text{ L min}^{-1} \text{ g}^{-1}$ ) is also regarded.

### Prospects

We have shown that 2 is easily prepared and is an attractive all-in-one solution to be used as a solid-state catalyzed hydrogen-rich nanocomposite based on both  $\text{NaBH}_4$  and the cheap metal  $\text{Ni}^0$ . Indeed,  $\text{NaBH}_4$  of 2 is totally converted into  $\text{H}_2$  and  $\text{NaB(OH)}_4$  over the temperature range  $10\text{--}60 \text{ }^\circ\text{C}$ , and the hydrogen generation rates are high. From a technological point of view, the sample offers several other advantages (Fig. 12). In

2, the hydrogen carrier and the catalyst are combined, which avoids the user handling two separate chemicals. 2 can be easily loaded into a cartridge/reactor and the hydrolysis reaction can be triggered simply by adding the required amount of water.<sup>14</sup>

After the reaction, the cartridge/reactor is filled with  $\text{Ni(OH)}_2$  and borates. An additional amount of water can be added to the mixture to allow separation by centrifugation. The former solid can be industrially/chemically valorized under acidic conditions to reform  $\text{NiCl}_2$ .<sup>42</sup> The aqueous solution can be handled for isolating  $\text{NaB(OH)}_4$  and recycling it following one of the processes under investigation.<sup>43,44</sup> The viability of the regeneration of  $\text{NaBH}_4$  from  $\text{NaB(OH)}_4$  strongly depends on how the energy cost can be lowered; this clearly implies the use of renewable energy sources.<sup>45</sup> 2 is thus a candidate for relevantly closing the  $\text{H}_2$  cycle with  $\text{NaBH}_4$  and control the life cycle of the Ni-based catalytic material.

Irrespective of the hydrogen generation rates, quite high effective gravimetric hydrogen storage capacities (material-based) can be achieved. Under our conditions, the best performance was found to be 4.4 wt%, which may be seen as being outstanding.<sup>46</sup> Installed on a device for which the weight of 2 would represent 50% of the weight of the storage system (taken as a whole, *i.e.* including all of the components), one may expect a neat (*i.e.* system-based) gravimetric hydrogen storage capacity of 2.2 wt%. This should open implementation prospects for portable and mobile fuel cell-powered devices like laptop computers, power generators, and electric bicycles.

## Conclusions

The potential of a hybrid structure where a Ni catalyst is supported on the surface of  $\text{NaBH}_4$  nanoparticles, *i.e.* core@shell nanocomposite, was investigated for hydrogen generation through hydrolysis of  $\text{NaBH}_4$ . By hydrolysis at temperatures



varying from 10 to 60 °C, H<sub>2</sub> was successfully generated from this core-shell structure but with some notable features. In particular, high hydrogen generation rates were recorded in the presence of an excess of water (such as H<sub>2</sub>O/NaBH<sub>4</sub> = 64). For example, the rates were found to be 6.1 and 22.5 L min<sup>-1</sup> g<sup>-1</sup> at 39 and 60 °C, respectively, and these values are among the fastest hydrogen generation rates reported so far. Upon the completion of the H<sub>2</sub> generation, the hydrolysis by-product was identified as NaB(OH)<sub>4</sub> and the nickel-based catalyst evolved into agglomerates of Ni(OH)<sub>2</sub>. In other words, recycling the end-product to regenerate the starting materials, NaBH<sub>4</sub> and NiCl<sub>2</sub>, would be possible through conventional chemical routes. Besides the attractive catalytic performance of the proposed core-shell approach for hydrogen generation, another advantage of our NaBH<sub>4</sub>@Ni<sup>0</sup> nanocomposite is the possibility to get high effective gravimetric hydrogen storage capacities by decreasing the amount of water necessary to the hydrolysis reaction. We optimized the conditions to achieve an effective capacity of 4.4 wt%, which is a very attractive value that may open technological application prospects for portable and mobile fuel cell-powered devices, for example.

## Conflicts of interest

There are no conflicts to declare.

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## Notes and references

- H. I. Schlesinger, H. C. Brown, A. E. Finholt, J. R. Gilbreath, H. R. Hoekstra and E. K. Hyde, *J. Am. Chem. Soc.*, 1953, **75**, 215.
- Q. L. Zhu and Q. Xu, *Energy Environ. Sci.*, 2015, **8**, 478; M. Yadav and Q. Xu, *Energy Environ. Sci.*, 2012, **5**, 9698.
- J. Zhang, T. S. Fisher, J. P. Gore, D. Hazra and P. V. Ramachandran, *Int. J. Hydrogen Energy*, 2006, **31**, 2292.
- H. J. Kim, K. J. Shin, H. J. Kim, M. K. Han, H. Kim, Y. G. Shul and K. T. Jung, *Int. J. Hydrogen Energy*, 2010, **35**, 12239.
- P. Brack, S. E. Dann and K. G. U. Wijayantha, *Energy Sci. Eng.*, 2015, **3**, 174.
- E. Y. Marrero-Alfonso, J. R. Gray, T. A. Davis and M. A. Matthews, *Int. J. Hydrogen Energy*, 2007, **32**, 4723.
- E. Petit, P. Miele and U. B. Demirci, *ChemSusChem*, 2016, **9**, 1777.
- J. H. Kim, K. H. Choi and Y. S. Choi, *Int. J. Hydrogen Energy*, 2010, **35**, 4015.
- J. Kim and T. Kim, *Appl. Energy*, 2015, **160**, 945.
- H. Kim, T. H. Oh and S. Kwon, *Int. J. Hydrogen Energy*, 2016, **41**, 1018.
- N. Lapeña-Rey, J. A. Blanco, E. Ferreyra, J. L. Lemus, S. Pereira and E. Serrot, *Int. J. Hydrogen Energy*, 2017, **42**, 6926.
- E. Okumus, F. G. B. San, O. Okur, B. E. Turk, E. Cengelci, M. Kilic, C. Karadag, M. Cavdar, A. Turkmen and M. S. Yazici, *Int. J. Hydrogen Energy*, 2017, **42**, 2691.
- F. C. Wang and W. H. Fang, *Int. J. Hydrogen Energy*, 2017, **42**, 10376.
- U. B. Demirci, *Energy Technol.*, 2018, **6**, 470.
- G. M. Arzac, M. Paladini, V. Godinho, A. M. Beltrán, M. C. Jiménez de Haro and A. Fernández, *Sci. Rep.*, 2018, **8**, 9755.
- E. S. Jung, H. Kim, S. Kwon and T. H. Oh, *Int. J. Green Energy*, 2018, **15**, 385.
- U. B. Demirci, *Int. J. Hydrogen Energy*, 2015, **40**, 2673.
- O. V. Netskina, O. V. Komova and V. I. Simagina, *Catal. Ind.*, 2018, **10**, 166.
- P. P. Prosini and P. Gislou, *J. Power Sources*, 2006, **161**, 290.
- Project in progress at the University of Montpellier*, Institut Européen des Membranes, IEM, not yet published.
- M. L. Christian and K. F. Aguey-Zinsou, *ACS Nano*, 2012, **6**, 7739.
- M. Christian and K. F. Aguey-Zinsou, *Chem. Commun.*, 2013, **49**, 6794.
- J. Legrand, S. Gota, M. J. Guittet and C. Petit, *Langmuir*, 2002, **18**, 4131.
- A. V. Naumkin, A. Kraut-Vass, S. W. Gaarenstroom, and C. J. Powell, *NIST Standard Reference Database 20, Version 4.1*, web version, <http://srdata.nist.gov/xps/>, 2012.
- U. B. Demirci and P. Miele, *Phys. Chem. Chem. Phys.*, 2014, **16**, 6872.
- S. B. Kalidindi, J. Joseph and B. R. Jagirdar, *Energy Environ. Sci.*, 2009, **2**, 1274.
- A. V. Iogansen, *Spectrochim. Acta, Part A*, 1999, **55**, 1585.
- G. N. Glavee, K. J. Klabunde, C. M. Sorensen and G. C. Hadjipanayis, *Langmuir*, 1994, **10**, 4726.
- K. Ishihara, A. Nagasawa, K. Umamoto, H. Ito and K. Saito, *Inorg. Chem.*, 1994, **33**, 3811.
- U. B. Demirci, O. Akdim, J. Andrieux, J. Hannauer, R. Chamoun and P. Miele, *Fuel Cells*, 2010, **3**, 335.
- S. S. Muir and X. Yao, *Int. J. Hydrogen Energy*, 2011, **36**, 5983.
- R. Retnamma, A. Q. Novais and C. M. Rangel, *Int. J. Hydrogen Energy*, 2011, **36**, 9772.
- J. Andrieux, D. Swierczynski, L. Laversenne, A. Garron, S. Bennici, C. Goutaudier, P. Miele, A. Auroux and B. Bonnetot, *Int. J. Hydrogen Energy*, 2009, **34**, 938.
- Z. Wu, X. Mao, Q. Zi, R. Zhang, T. Dou and A. C. K. Yip, *J. Power Sources*, 2014, **268**, 596.
- J. H. Kim, K. T. Kim, Y. M. Kang, H. S. Kim, M. S. Song, Y. J. Lee, P. S. Lee and J. Y. Lee, *J. Alloys Compd.*, 2004, **379**, 222.
- L. Jun, X. Shuping and G. Shiyang, *Spectrochim. Acta, Part A*, 1995, **51**, 519.
- H. Y. Wu, Y. L. Xie and Z. A. Hu, *Int. J. Electrochem. Sci.*, 2013, **8**, 1839.



- 38 D. S. Hall, D. J. Lockwood, C. A. Bock and B. R. MacDougall, *Proc. R. Soc. A*, 2014, **471**, 20140792.
- 39 F. Baydaroglu, E. Özdemir and A. Hasimoglu, *Int. J. Hydrogen Energy*, 2014, **39**, 1516.
- 40 N. Patel, R. Fernandes and A. Miotello, *J. Power Sources*, 2009, **188**, 411.
- 41 U. B. Demirci and P. Miele, *C. R. Chim.*, 2014, **17**, 707.
- 42 B. Beverskog and I. Puigdomenech, *Corros. Sci.*, 1997, **39**, 969.
- 43 M. Rivarolo, O. Improta, L. Magistri, M. Panizza and A. Barbucci, *Int. J. Hydrogen Energy*, 2018, **43**, 1606.
- 44 H. Zhong, L. Z. Ouyang, J. S. Ye, J. W. Liu, H. Wang, X. D. Yao and M. Zhu, *Energy Storage Materials*, 2017, **7**, 222.
- 45 M. Monteverde and L. Magistri, *Int. J. Hydrogen Energy*, 2012, **37**, 5452.
- 46 J. Ren, N. M. Musyoka, H. W. Langmi, M. Mathe and S. Liao, *Int. J. Hydrogen Energy*, 2017, **42**, 289.

