



Molecular spectroscopy in a solid-state device†

Ainhoa Atxabal,^a Thorsten Arnold,^b Subir Parui,^c Elisabetta Zuccatti,^a
Mirko Cinchetti,^d Fèlix Casanova,^{ae} Frank Ortmann^b and
Luis E. Hueso^{id} *^{ae}

Cite this: *Mater. Horiz.*, 2019, 6, 1663

Received 10th February 2019,
Accepted 18th March 2019

DOI: 10.1039/c9mh00218a

rsc.li/materials-horizons

The quantification of the electronic transport energy gap of a molecular semiconductor is essential for pursuing any challenge in molecular optoelectronics. However, this remains largely elusive because of the difficulties in its determination by conventional spectroscopic methods. This communication presents an in-device molecular spectroscopy (i-MOS) technique, which permits measuring this gap seamlessly, in real device operative conditions, at room temperature and without any previous knowledge of the material's parameters. This result is achieved by determining the occupied and unoccupied molecular orbitals of an organic semiconductor thin-film by using a single three terminal solid-state device.

The engineering of molecular semiconductor-based technologies requires precise knowledge of key parameters, such as the relative energetic position of the molecular levels and proper quantification of the transport energy gap.^{1–6} However, the lack of a reliable and straightforward technique for the determination of the transport energy gap in the bulk of an organic semiconductor often results in the adoption of inadequate substitutes, such as the single-molecular fundamental, Kohn–Sham, electrochemical, optical or interface gaps.⁷ The single-molecular fundamental gap, obtained by single-molecule scanning tunneling spectroscopy,^{8,9} cannot be interpreted as a transport gap, because the strong polarization coming from the π -conjugated molecules makes the transport gap different from the energy gap at the single molecule level.^{7,10–12} The use of the Kohn–Sham gap from density functional theory on the other hand has no formal justification and is thus dependent on cancelation of intrinsic errors and inadequacies present in this method.⁷ On the other hand, the electrochemical gap is principally related to reduction–oxidation potentials,

New concepts

In this manuscript we are able to determine the transport energy gap of a molecular semiconductor under *in-operando* conditions and without any prior knowledge of material parameters. The measurements have been performed at room temperature with a three terminal vertical solid-state device, which makes this technique convenient for and related to opto-electronic device applications. In spite of its relevance, the reality is that so far there is not an experimental or computational technique capable of delivering such value without approximations involved. The vast community working in organic electronics has simply adjusted to work with substitutes, such as the energy gap values obtained by UV-VIS or photoemission spectroscopy techniques. The value of the transport gap of the organic semiconductor determines, for example and in a first approximation, the conductivity of a material and its possible integration in commercial devices. Here we demonstrate a novel, simple and accurate way for obtaining the transport gap; a value that cannot be obtained with the current experimental or theoretical approaches.

which can be sensitive to solvation and Coulombic effects.^{13,14} In addition, the optical gap, commonly measured by ultraviolet-visible spectroscopy (UV-VIS), is different from the transport gap due to the strong exciton binding energy of molecules.^{7,13,15–17} Besides, in the case of some organic semiconductors such as C₆₀ fullerenes, the optical transition associated with the transport states may be symmetry forbidden, which leads to UV-VIS possibly measuring higher onset energies. Finally, the interface gap, obtained by a combination of different surface-sensitive techniques, typically ultraviolet photoemission spectroscopy (UPS) and inverse photoemission spectroscopy (IPES),^{18–20} can be used to estimate the transport gap, but under the careful consideration of different limiting factors, such as the low resolution of IPES (0.4 eV), the probing depth at the organic-vacuum interface and other surface dependent parameters like roughness.²¹

In this work, thanks to an in-device molecular spectroscopy (i-MOS) technique, we are able to determine directly in-operando both the energetic positions of the highest occupied molecular orbital (HOMO) states and the lowest unoccupied molecular orbital (LUMO) states and the transport energy gap of a molecular semiconductor. This has been accomplished by using a three-terminal

^a CIC nanoGUNE, 20018 Donostia-San Sebastian, Basque Country, Spain.
E-mail: lhueso@nanogune.eu

^b Center for Advancing Electronics Dresden, Technische Universität Dresden,
01062 Dresden, Germany

^c IMEC and K. U. Leuven, 3001 Leuven, Belgium

^d Experimentelle Physik VI, Technische Universität Dortmund, Dortmund 44221, Germany

^e IKERBASQUE, Basque Foundation for Science, 48013 Bilbao, Basque Country, Spain

† Electronic supplementary information (ESI) available. See DOI: 10.1039/c9mh00218a



solid-state device and without any pre-knowledge of the material parameters. As proof of principle we have chosen C_{60} and C_{70} , two well established molecular semiconductors in the field of electronics and optoelectronics.^{22–27} Moreover, the similar chemical structure and the consequent close molecular orbital energies of both materials permit us to test the sensitivity of i-MOS.

i-MOS is performed in a three-terminal vertical device, which is composed of an aluminium/aluminium oxide emitter (E), a gold (Au) base (B) and an n-type organic semiconductor (C_{60} and C_{70}) collector with an aluminium top contact (C) (see Fig. 1).²¹ Details on the device fabrication and thin film characterization can be found in the Methods section [device cross section and top view are shown in ESI,† Fig. S1. Atomic force microscopy and X-ray diffraction measurements are shown in ESI,† Fig. S2 and S3, respectively, while a basic electrical characterization of the device is available in ESI,† Fig. S4 and S5]. We should highlight the simple fabrication method, which should be available to many groups working in the field of organic optoelectronics. Fig. 1 shows the working principle of i-MOS. When a negative emitter base bias, V_{EB} , is applied, a current I_E is injected from the emitter

into the base by tunnelling through the Al_2O_3 barrier. These electrons are “hot” in the base as their energy is well above ($\gg k_B T$) the Fermi level of the metal, and a small fraction of them crosses the thin metallic base ballistically without any significant energy attenuation.^{11,28–30} For the case in which the applied bias $-eV_{EB} \geq \Delta$ (Δ is the metal–semiconductor energy barrier), some of the injected hot electrons enter into the LUMO states, while the remaining ones flow back into the base. At higher energies, electrons can also enter into higher energy conductive states, such as LUMO+1 states. Since C_{60} and C_{70} are n-type semiconductors and they are sandwiched between two metallic contacts with different work functions, the emerging built-in potential enables the detection of the electron current, I_C , without any applied collector–base bias, V_{CB} (Fig. 1a).^{31,32} Under a negative applied V_{CB} (Fig. 1c), the electric field in the base–collector junction repulses hot electrons and leads to the detection of holes injected into the HOMO level of the n-type semiconductor, thus enabling the determination of the relative energy of that molecular level with respect to E_F of the base metal. These holes are created by the inelastic scattering of the incident hot electrons in the gold metal base with cold electrons (below the Fermi level).^{33–35}

A similar mechanism operates for positive V_{EB} , but in this case holes are injected into the base, instead of electrons. Fig. 1b represents an Auger-like scattering process in which incident holes excite secondary electron–hole pairs in the gold metal base and thus hot electrons are created.^{36–38} Fig. 1d shows the direct process of hole-injection and hole-detection, respectively.

Fig. 2 reports the experimental values of the collector current, I_C , with V_{EB} of C_{60} -based i-MOS, measured under the operations described in Fig. 1. Fig. 2a shows the direct electron process, *i.e.* electron injection and electron detection, when $V_{EB} < 0$ V and $V_{CB} = 0$ V is applied. Fig. 2a also shows the derivative of the I_C , $|dI_C/dV|$. Two onsets are observed in the $|dI_C/dV|$ curve, which corresponds to the selection of the molecular orbital levels, in this case LUMO at 0.8 ± 0.1 eV and LUMO+1 at 1.7 ± 0.1 eV. The energies of the molecular orbital levels extracted with i-MOS are relative to the Fermi energy of the gold base. The relative energy of the LUMO level is consistent with the energy value acquired for the same molecular level by secondary hole–electron processes when $V_{EB} > 0$ V and $V_{CB} = 0$ V (see Fig. 2b). The $|dI_C/dV|$ curve of this plot shows the onset of the LUMO at 0.9 ± 0.1 eV. Fig. 2c and d show the I_C and $|dI_C/dV|$ versus V_{EB} for secondary electron–hole and direct hole processes. In both figures an onset is observed at the $|dI_C/dV|$ curves, which corresponds to the relative energy of the HOMO level at 1.6 ± 0.1 eV. The direct hole process has been achieved by $V_{EB} > 0$ V and $V_{CB} = -0.5$ V, while the secondary electron–hole process by $V_{EB} < 0$ V and $V_{CB} = -0.5$ V. Playing with the freedom that the collector electrode offers us, we tune the built-in potential of the organic semiconductor by the V_{CB} , which permits choice of the carrier type (electrons or holes) responsible for the transport in the organic semiconductor-based device. ESI,† Fig. S6–S9 show a detailed V_{CB} dependence of C_{60} i-MOS.

In order to extend our approach to another compound, in Fig. 3 we study the i-MOS operations described in Fig. 1 for C_{70} -based i-MOS devices. C_{70} is a larger n-type buckyball than C_{60} , which permitted us to test the sensitivity and energy

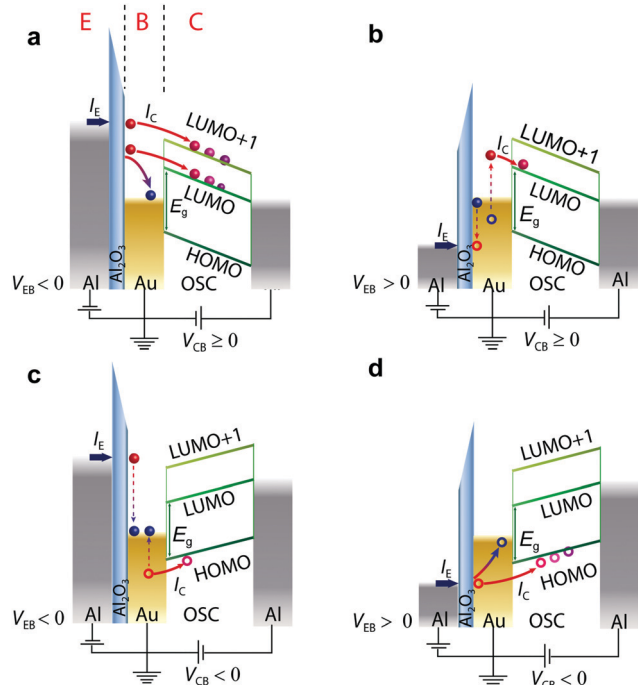


Fig. 1 In-device molecular spectroscopy (i-MOS) operation under different emitter–base bias V_{EB} polarities at room temperature. (a) Schematic representation of the direct process in which electrons are injected by applying a negative emitter–base bias, $V_{EB} < 0$ V, and detected in the molecular semiconductor. The measurements can be performed either without any externally applied base collector bias, V_{CB} , or by $V_{CB} \geq 0$ V. (b) Schematic representation of the secondary process at i-MOS when holes are injected into the device, $V_{EB} > 0$ V, but hot electrons coming from a similar effect to Auger scattering are detected with $V_{CB} \geq 0$ V. (c) Schematic representation of the secondary process at i-MOS when electrons are injected, $V_{EB} < 0$ V, into the device but holes are detected. $V_{CB} < 0$ V is required in order to cancel the built-in potential and help hole injection into and transport through the organic semiconductor. (d) Schematic representation of the direct process at i-MOS when holes are injected, $V_{EB} > 0$ V, and detected with $V_{CB} < 0$ V.



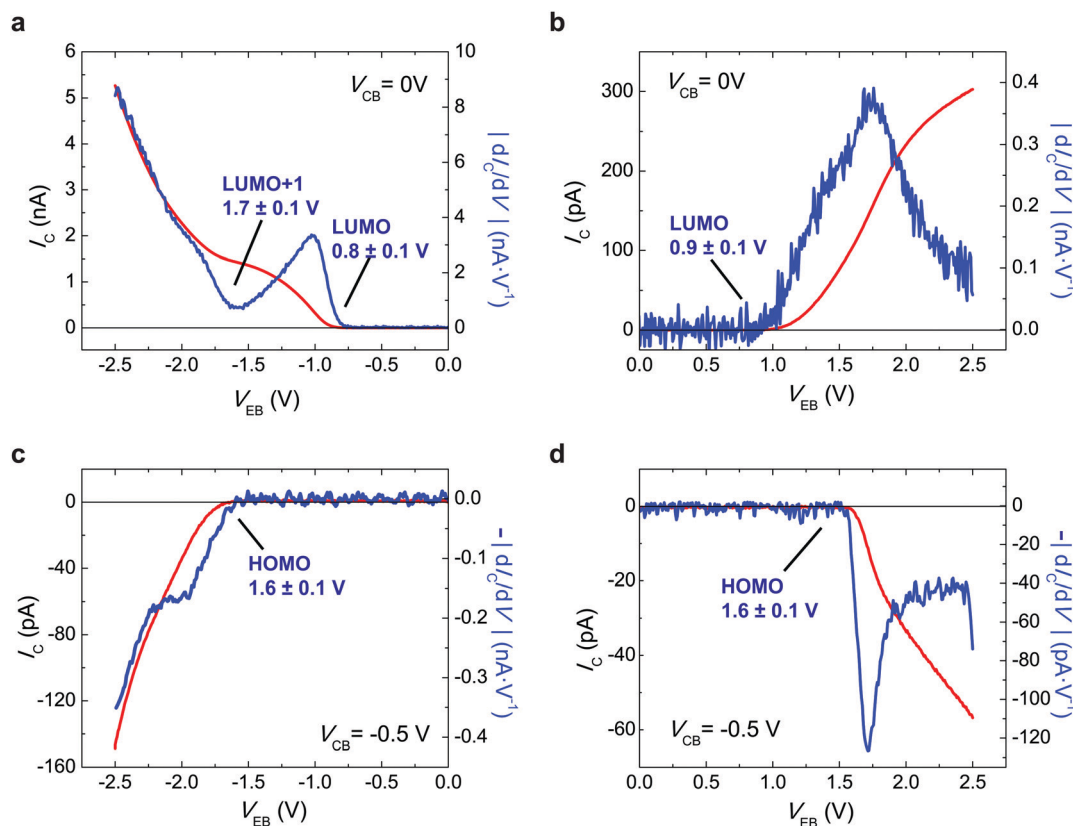


Fig. 2 In-device molecular spectroscopy (i-MOS) of C_{60} fullerene performed at different emitter-base bias V_{EB} polarities. (a) Collector current I_C for negative emitter base bias $V_{EB} < 0$ V (red solid line) and its derivative with respect to V_{EB} (blue solid line) at 300 K. Energetic electrons are injected with $V_{CB} = 0$ V both into the lowest molecular orbital (LUMO) and into the next higher excited conductive molecular level (LUMO+1) states of C_{60} . Note that in the current configuration, positive I_C indicates electron current detection whereas negative I_C corresponds to hole current detection in C_{60} . (b) Electron current, I_C , and LUMO states detected with $V_{EB} > 0$ V and $V_{CB} = 0$ V (red solid line) together with the derivative of the I_C with respect to V_{EB} (blue solid line) at 300 K. (c) Detected I_C for $V_{EB} < 0$ V and $V_{CB} = -0.5$ V (red solid line) and the derivative of the I_C with respect to V_{EB} (blue solid line) at room temperature, 300 K. Injection of a hole current into C_{60} leads to the detection of its highest occupied molecular orbital (HOMO) states. (d) I_C versus V_{EB} when $V_{EB} > 0$ V and $V_{CB} = -0.5$ V (red solid line), and the derivative of the I_C with respect to V_{EB} (blue solid line). Injection of hot holes permits the detection of the HOMO states of C_{60} in a direct process.

resolution of i-MOS for systems with similar energetic structure. In the case of C_{70} -based i-MOS, we could determine that the relative energy of the LUMO with respect to the Fermi level of gold is 0.7 ± 0.1 eV, the one of LUMO+1 is 1.6 ± 0.1 eV (Fig. 3a and b) and the HOMO level of C_{70} is at 1.5 ± 0.1 eV below the Fermi level of the base metal (Fig. 3c and d). The values extracted from direct processes and secondary processes are in agreement. ESI,† Fig. S10–S13 show a detailed V_{CB} dependence of C_{70} i-MOS. All these energy levels are summarized in Fig. 4.

The transport gap of both C_{60} and C_{70} in real device operative conditions can be simply calculated taking into account the energetic difference between the HOMO and LUMO levels, leading to a value of 2.4 ± 0.2 eV in the case of C_{60} (Fig. 4a) and 2.2 ± 0.2 eV in the case of C_{70} (Fig. 4b), when averaged over dozens of devices (ESI,† Table S1). This quantity is obtained without any assumption regarding the nature of the molecular material, and hence the method presented is easily transferred to other systems under study. The difference detected in the transport gap between C_{60} and C_{70} proves the sensitivity of i-MOS. The carrier transport gap is a key parameter for many (opto)electronic applications^{1–6,39} and there are currently several other techniques

routinely used to approach this quantity indirectly,⁷ including electrochemical spectroscopy,^{13,14} ultraviolet-visible spectroscopy (UV-VIS),^{15–17} ultraviolet photoemission spectroscopy (UPS), inverse photoemission spectroscopy (IPES),^{18–20,40} and single-molecule scanning tunnelling spectroscopy.^{8,10} ESI,† Table S2 compares the E_G of C_{60} on gold extracted by several different methods. In our case, we can determine directly and without any prior assumptions, both the relative energetic positions of the HOMO and LUMO states, and the transport energy gap of a molecular semiconductor in real operative conditions and at room temperature.

Conclusions

In conclusion, by making use of the i-MOS technique we report an unprecedented full energy characterization of both hole and electron transport levels of molecular semiconductors under in-operando conditions. This information cannot be obtained by any other experimental technique, while it is paramount for the understanding and development of any future complex molecule-based (opto)-electronic devices. Moreover, we have shown that i-MOS is able



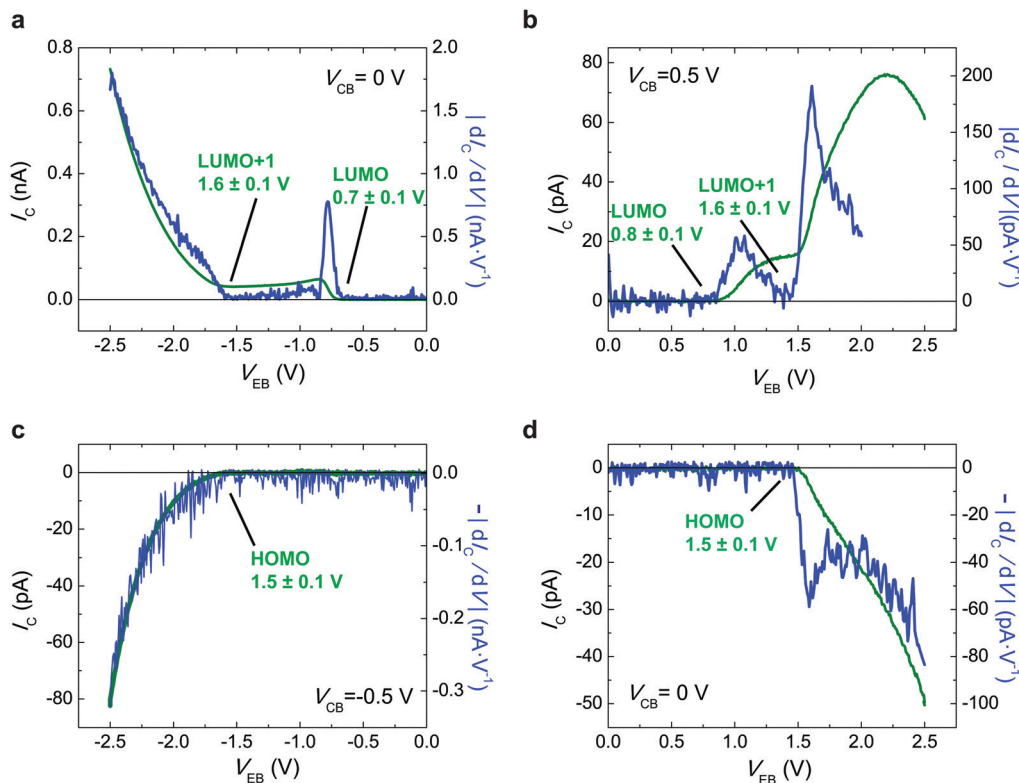


Fig. 3 In-device molecular spectroscopy (i-MOS) operation for C_{70} fullerene under different emitter-base bias V_{EB} polarities. (a) Collector current I_C for negative emitter base bias $V_{EB} < 0$ V (green solid line) and its derivative with respect to V_{EB} (blue solid line) at 200 K. Energetic electrons are injected with $V_{CB} = 0$ V both into the lowest molecular orbital (LUMO) and into the next higher excited conductive molecular level (LUMO+1) states of C_{70} . Note that in the current configuration, positive I_C indicates electron current detection whereas negative I_C corresponds to hole current detection in C_{70} . (b) Electron current, I_C , and LUMO and LUMO+1 states detected with $V_{EB} > 0$ V and $V_{CB} = 0.5$ V (green solid line) together with the derivative of I_C with respect to V_{EB} (blue solid line) at 200 K. (c) Detected I_C for $V_{EB} < 0$ V and $V_{CB} = -0.5$ V (green solid line) and the derivative of the I_C with respect to V_{EB} (blue solid line) at 200 K. Injection of a hole current into C_{70} leads to the detection of its highest occupied molecular orbital (HOMO) states. (d) I_C versus V_{EB} when $V_{EB} > 0$ V and $V_{CB} = -0.5$ V (green solid line), and the derivative of the I_C with respect to V_{EB} (blue solid line). Injection of energetic holes permits the detection of the HOMO states of C_{70} in a direct process.

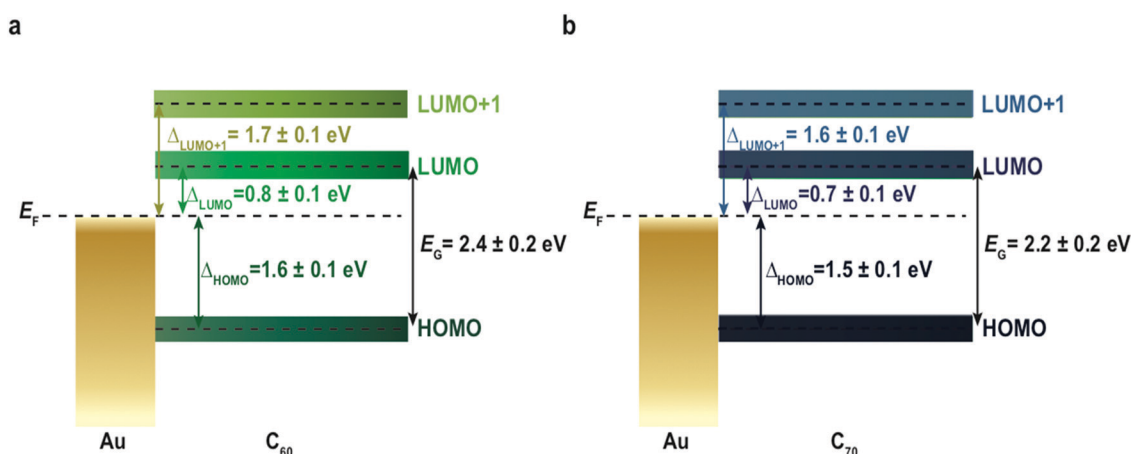


Fig. 4 Au/ C_{60} and Au/ C_{70} energy level alignment. E_F is the Fermi energy of gold. The LUMO corresponds to the lowest unoccupied molecular orbital and LUMO+1 to the next higher excited conductive molecular orbitals, respectively, and HOMO to the highest occupied molecular orbital. E_G is the energy gap between HOMO and LUMO of (a) C_{60} and (b) C_{70} . Δ corresponds to the energy barrier between the Fermi energy of gold and the molecular level devoted to charge transport. The sublabel of each Δ refers to its corresponding molecular orbital.

to detect small energy differences in the energy level alignment of similar molecules such as C_{60} and C_{70} . The results reported in this work present a reliable, direct and easily applicable

device-spectroscopic method. This makes it a promising technique for studies on the injection properties of metal-organic interfaces as well as the manipulation of the electronic transport states of



organic semiconductors, for instance by doping. Since the knowledge of the electronic levels and its optimization are crucial to improve the efficiency of virtually all optoelectronic devices, we expect i-MOS in the future to provide accurate energies and new insights which are at the root of new developments.

Methods

Device fabrication

All the devices described in this work were fabricated *in situ* in an ultra high vacuum (UHV) multi chamber evaporator (base pressure $<10^{-9}$ mbar) with a shadow mask system. The emitter is a 12 nm-thick aluminium contact, 99.95% purity (Lesker), which was thermally evaporated in an effusion cell with a rate of 0.6 Å s^{-1} . A crucible of pyrolytic boron nitride (PBN) was used. The Al_2O_3 tunnel junction was made by plasma oxidizing the aluminium contact, first for two minutes at low power (1200 V and 10 mA at 0.1 mbar) and then for three minutes at high power (1200 V and 50 mA at 0.1 mbar). A 10 nm-thick gold base (99.95% purity, Lesker) was e-beam evaporated from a vitreous-coated graphite-based crucible, and used as a base contact. The evaporation rate was 1.0 Å s^{-1} . C_{60} (and C_{70}) triple-sublimed quality (99.9%, Sigma Aldrich), was thermally evaporated in a quartz-based crucible with a rate of 0.1 Å s^{-1} . Finally, a 12 nm-thick aluminium top electrode was again thermally evaporated. A two-step deposition (2 nm at 0.1 Å s^{-1} and 10 nm at 0.6 Å s^{-1}) was followed in order to minimize the damage to the organic film.

The sample size is $10 \times 10 \text{ mm}^2$, and six devices were produced with every sample.

Electrical characterization

Electrical characterization was performed under high vacuum (base pressure 5×10^{-5} mbar) in a variable-temperature probe-station (Lakeshore). A Keithley 4200 semiconductor analyser system was used to record *I-V* curves.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

The authors acknowledge Prof. Francesca Tinti for her critical opinion about the manuscript. This work is supported by the European Research Council (Grants 257654-SPINTROS and 725767-hyControl), by the Spanish MINECO under Project No. MAT2015-65159-R and Maria de Maeztu Units of Excellence Program – MDM-2016-0618 and by the Basque Government (under Project No. PC2015-1-01). A. A. acknowledges the Basque Government for a PhD fellowship (PRE_2017_2_0052). The work at TU Dresden was supported by the Deutsche Forschungsgemeinschaft (Grant OR 349/1-1). Grants for computing time from the Centre for Information Services and High Performance Computing of TU Dresden are gratefully acknowledged.

References

- 1 M. Schwarze, W. Tress, B. Beyer, F. Gao, R. Scholz, C. Poelking, K. Ortstein, A. A. Günther, D. Kasemann, D. Andrienko and K. Leo, *Science*, 2016, **352**, 1446.
- 2 S. Holliday, R. S. Ashraf, A. Wadsworth, D. Baran, S. A. Yousaf, C. B. Nielsen, C.-H. Tan, S. D. Dimitrov, Z. Shang, N. Gasparini, M. Alamoudi, F. Laquai, C. J. Brabec, A. Salleo, J. R. Durrant and I. McCulloch, *Nat. Commun.*, 2016, **7**, 11585.
- 3 J. Lee, H.-F. Chen, T. Batagoda, C. Coburn, P. I. Djurovich, M. E. Thompson and S. R. Forrest, *Nat. Mater.*, 2015, **15**, 92.
- 4 M. Muccini, *Nat. Mater.*, 2006, **5**, 605.
- 5 S. R. Forrest, *Nature*, 2004, **428**, 911.
- 6 X. Sun, S. Vélez, A. Atxabal, A. Bedoya-Pinto, S. Parui, X. Zhu, R. Llopis, F. Casanova and L. E. Hueso, *Science*, 2017, **357**, 677.
- 7 J. L. Bredas, *Mater. Horiz.*, 2014, **1**, 17.
- 8 A. Nitzan and M. A. Ratner, *Science*, 2003, **300**, 1384.
- 9 A. Bannani, C. Bobisch and R. Moller, *Science*, 2007, **315**, 1824.
- 10 S. F. Alvarado, P. F. Seidler, D. G. Lidzey and D. D. C. Bradley, *Phys. Rev. Lett.*, 1998, **81**, 1082.
- 11 M. Jung, D. Shin, S.-D. Sohn, S.-Y. Kwon, N. Park and H.-J. Shin, *Nanoscale*, 2014, **6**, 11835.
- 12 C. Gaul, S. Hutsch, M. Schwarze, K. S. Schellhammer, F. Bussolotti, S. Kera, G. Cuniberti, K. Leo and F. Ortman, *Nat. Mater.*, 2018, **17**, 439.
- 13 R. Holze, *Organometallics*, 2014, **33**, 503.
- 14 V. Nádazdy, F. Schauer and K. Gmucová, *Appl. Phys. Lett.*, 2014, **105**, 142109.
- 15 D. A. Evans, A. R. Vearey-Roberts, O. R. Roberts, G. T. Williams, S. P. Cooil, D. P. Langstaff, G. Cabailh, I. T. McGovern and J. P. Goss, *J. Appl. Phys.*, 2013, **114**, 123701.
- 16 B. A. Gregg and M. C. Hanna, *J. Appl. Phys.*, 2003, **93**, 3605.
- 17 A. Kahn, *Mater. Horiz.*, 2016, **3**, 7.
- 18 B. S. Braun, W. R. Salaneck and M. Fahlman, *Adv. Mater.*, 2009, **21**, 1450.
- 19 E. L. Shirley and S. G. Louie, *Phys. Rev. Lett.*, 1993, **71**, 133.
- 20 J. Hwang, A. Wan and A. Kahn, *Mater. Sci. Eng., R*, 2009, **64**, 1.
- 21 A. Atxabal, S. Braun, T. Arnold, X. Sun, S. Parui, X. Liu, C. Gozalvez, R. Llopis, A. Mateo-Alonso, F. Casanova, F. Ortman, M. Fahlman and L. E. Hueso, *Adv. Mater.*, 2017, **10**, 1606901.
- 22 X. Wang, E. Perzon, F. Oswald, F. Langa, S. Admassie, M. R. Andersson and O. Inganäs, *Adv. Funct. Mater.*, 2005, **15**, 1665.
- 23 K. Schulze, C. Uhrich, R. Schüppel, K. Leo, M. Pfeiffer, E. Brier, E. Reinold and P. Bäuerle, *Adv. Mater.*, 2006, **18**, 2872.
- 24 D. M. Guldi and N. Martin, *Fullerenes: From Synthesis to Optoelectronic Properties*, Springer, Netherlands, 2002.
- 25 G. A. Salvatore, N. Mu, T. Kinkeldei, L. Petti, C. Zysset, G. Tro, I. Strebel and L. Bu, *Nat. Commun.*, 2014, **5**, 2982.
- 26 M. P. Klinger, A. Fischer, F. Kaschura, R. Scholz, B. Lüssem, B. Kheradmand-Boroujeni, F. Ellinger, D. Kasemann and K. Leo, *Adv. Mater.*, 2015, **27**, 7734.



- 27 T. D. Anthopoulos, B. Singh, N. Marjanovic, N. S. Sariciftci, A. Montaigne Ramil, H. Sitter, M. Cölle and D. M. De Leeuw, *Appl. Phys. Lett.*, 2006, **89**, 7.
- 28 M. Gobbi, L. Pietrobon, A. Atxabal, A. Bedoya-Pinto, X. Sun, F. Golmar, R. Llopis, F. Casanova and L. E. Hueso, *Nat. Commun.*, 2014, **5**, 4161.
- 29 N. Balkan, *Hot Electrons Semicond.*, Clarendon Press, Oxford, 1998, pp. 385–427.
- 30 T. Arnold, A. Atxabal, S. Parui, L. E. Hueso and F. Ortmann, *Adv. Funct. Mater.*, 2018, **20**, 1706105.
- 31 W. J. Kaiser and L. D. Bell, *Phys. Rev. Lett.*, 1988, **60**, 1406.
- 32 L. D. Bell and W. J. Kaiser, *Phys. Rev. Lett.*, 1988, **61**, 2368.
- 33 M. Bauer, A. Marienfeld and M. Aeschlimann, *Prog. Surf. Sci.*, 2015, **90**, 319.
- 34 T. Banerjee, E. Haq, M. H. Siekman, J. C. Lodder and R. Jansen, *IEEE Trans. Magn.*, 2005, **41**, 2642.
- 35 B. G. Park, E. Haq, T. Banerjee, B. C. Min, J. C. Lodder and R. Jansen, *J. Appl. Phys.*, 2006, **99**, 08S701.
- 36 W. Yi, V. Narayanamurti, H. Lu, M. A. Scarpulla and A. C. Gossard, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2010, **81**, 235325.
- 37 W. Yi, V. Narayanamurti, H. Lu, M. A. Scarpulla, A. C. Gossard, Y. Huang, J. H. Ryou and R. D. Dupuis, *Appl. Phys. Lett.*, 2009, **95**, 112102.
- 38 L. D. Bell, *J. Vac. Sci. Technol., B: Microelectron. Nanometer Struct.–Process., Meas., Phenom.*, 1991, **9**, 594.
- 39 M. Cinchetti, V. A. Dediu and L. E. Hueso, *Nat. Mater.*, 2017, **16**, 507.
- 40 H. Yoshida, *Chem. Phys. Lett.*, 2012, **53**, 180.

