Electrodeposition of indium from the ionic liquid triethyl(tetradecyl)phosphonium chloride

Clio Deferm, João C. Malaquias, Bieke Onghena, Dipanjan Banerjee, Jan Luyten, Harald Oosterhof, Jan Fransaer and Koen Binnemans

The electrochemical behavior of indium in the ionic liquid triethyl(tetradecyl)phosphonium chloride (Cyphos IL 101) was studied. Cyphos IL 101 first had to be purified, as the impurities present in commercial Cyphos IL 101 interfered with the electrochemical measurements. Electrochemical deposition of indium metal from this electrolyte occurs without hydrogen evolution, increasing the cathodic current efficiency compared to deposition from water and avoiding porosity within the deposited metal. Indium(III) is the most stable oxidation state in the ionic liquid. This ion is reduced in two steps, first from indium(III) to indium(I) and subsequently to indium(0). The high thermal stability of Cyphos IL 101 allowed the electrodeposition of indium at 120 °C and 180 °C. At 180 °C indium was deposited as liquid indium which allows for the easy separation of the indium and the possibility to design a continuous electrowinning process. On molybdenum, indium deposits as liquid droplets even below the melting point of indium. This was explained by the combination of melting point depression and undercooling. The possibility to separate indium from iron and zinc by electrodeposition was tested. It is possible to separate indium from zinc by electrodeposition, but iron deposits together with indium.

Introduction

Indium is important for the electronics industry. Indium-containing semiconductors, such as indium phosphate (InP), indium selenide (InSe), copper indium selenide (CIS), copper indium gallium selenide (CIGS), indium arsenide (InAs), indium gallium phosphide (InGaP), and indium tin oxide (ITO), are essential for components of many electronic and optoelectronic devices (e.g. detectors, lasers, photovoltaic devices and light-emitting diodes). Indium also finds application in low-temperature alloy solders. Electrochemical deposition of indium is one of many different fabrication techniques for these materials, next to closed-space sublimation, metal organic chemical vapor deposition, co-evaporation and molecular beam epitaxy. The advantages of electrodeposition over the other techniques are: lower required temperatures, the possibility to work at ambient pressure with a high deposition rate, easily controllable and low-cost equipment and inexpensive precursors. Electrochemical deposition of indium is also of importance for the purification of indium, as high-purity indium is often required as a starting material for the above-mentioned applications. Electrowinning of indium can be an essential step in the recovery of indium from primary and secondary sources.

Electrochemical deposition of indium from aqueous electrolytes has been extensively described in the literature. It is often complicated due to the simultaneous formation of hydrogen gas, which decreases the electrodeposition efficiency and can cause dendritic growth, pinholes and hydrogen embrittlement. The use of aprotic ionic liquid electrolytes can circumvent the complications associated with hydrogen evolution. Moreover, ionic liquids exhibit a good electrical conductivity, low vapor pressure, good thermal stability and a wide electrochemical window, allowing deposition of metals that cannot be deposited from water, e.g. aluminum.

Ionic liquids are solvents that consist entirely of ions. Two main families of ionic liquids are commonly used for electrochemical deposition purposes: first-generation chloroaluminate ionic liquids, and second-generation “air- and water-stable” ionic liquids with hydrophobic anions, such as tetrafluoroborate, hexafluorophosphate, trifluoromethanesulfonate or bis(trifluoromethylsulfonyl)imide. Until now, numerous
examples of the use of first-generation ionic liquids for the electrochemical deposition of single metals and alloys, including indium, have been reported.\textsuperscript{22–24} The disadvantage of the chloroaluminate ionic liquids is that they are very prone to hydrolysis, so that they must be handled in a glovebox. The replacement of the moisture-sensitive anions in the first-generation ionic liquids has led to the design of air- and water-stable second-generation ionic liquids.\textsuperscript{25} Moreover, a new class of ionic liquid analogues, deep-eutectic solvents, are emerging as electrolytes for metal electrodeposition.\textsuperscript{26} Until now, only a small number of papers report on the electrodeposition of indium or indium alloys from second-generation ionic liquids or deep-eutectic solvents.\textsuperscript{27–38}

High-temperature electrometallurgy using a liquid cathode is used in industry to recover or refine metals, such as aluminium via the Hall–Héroult process.\textsuperscript{39} Electrodeposition using a liquid cathode has also been studied for the electrowinning and electrorefining of the rare earths, actinides and titanium.\textsuperscript{40–45} The use of a liquid cathode offers several advantages: (1) easy separation of electrolysis products due to the large density difference between the electrolyte and the liquid cathode, (2) a constant electrode surface area that permits stable values for the process parameters, (3) irregular crystal growth is avoided, so lower operating voltages are required due to the shorter distance between cathode and anode, (4) easier coalescence of microdrops and metallic fog and (5) the possibility to design a continuous process.\textsuperscript{40,41,46} High-temperature electrowinning of indium has been reported for molten salts electrolytes, but to the best of our knowledge, there is no study regarding indium electrowinning from ionic liquids at elevated temperatures.\textsuperscript{47–50} The use of ionic liquids for electrowinning at temperatures above the melting point of indium can circumvent the complications associated with hydrogen evolution and can offer at the same time the advantages accompanied with the use of a liquid cathode.

In this paper, the electrochemical behavior of indium in the ionic liquid trihexyl(tetradecyl)phosphonium chloride (Cyphos IL 101) is described. The electrodeposition of indium above its melting point is reported. The morphology of the indium deposits was examined alongside the separation of indium from zinc and iron via electrodeposition. The high-temperature electrowinning of liquid indium allows for an easy separation of the indium and the possibility to design a continuous electrowinning process.

Results and discussion

Electrochemical study in commercial Cyphos IL 101

In a first step, the electrochemical window (EW) of commercial trihexyl(tetradecyl)phosphonium chloride, Cyphos IL 101, was determined. Fig. 1 shows the cyclic voltammograms (CVs) of commercial Cyphos IL 101 recorded on a Pt or Mo electrode at 120 °C and 180 °C at a scan rate of 20 mV s\textsuperscript{-1}. The limiting potentials and the corresponding EW for different experimental conditions are shown in Table 1. The EW are narrow: between 0.73 V and 1.31 V vs. Fc\textsuperscript{+}/Fc. Ionic liquids generally have wide electrochemical windows, often exceeding 4.00 V.\textsuperscript{51–53} Furthermore, the trihexyl(tetradecyl)phosphonium cation is reported to have a very high resistance to reduction.\textsuperscript{51,54} Commercial Cyphos IL 101 typically assays >97% trihexyl(tetradecyl)phosphonium chloride. However, the product also contains small amounts of tetrade-cane isomers, dihexyl phosphine (PR\textsubscript{2}H), HCl and trihexylphosphonium hydrochloride ([PR\textsubscript{3}H]Cl).\textsuperscript{55–58} It has been shown that the presence of electroactive impurities in ionic liquids can cause an apparent narrowing of the true EW.\textsuperscript{59,60} The first reductive wave in the forward cycle is possibly a consequence of the reduction of impurities. The narrowing of the EW makes the use of high-purity ionic liquids in electrochemical applications necessary.\textsuperscript{53,61} Therefore, the commercial

<table>
<thead>
<tr>
<th>WE</th>
<th>Temperature (°C)</th>
<th>Cathodic limit vs. Fc\textsuperscript{+}/Fc (V)</th>
<th>Anodic limit vs. Fc\textsuperscript{−}/Fc (V)</th>
<th>EW (V)</th>
</tr>
</thead>
<tbody>
<tr>
<td>Pt</td>
<td>120</td>
<td>−0.90</td>
<td>0.41</td>
<td>1.31</td>
</tr>
<tr>
<td></td>
<td>180</td>
<td>−0.73</td>
<td>0.31</td>
<td>1.04</td>
</tr>
<tr>
<td>Mo</td>
<td>120</td>
<td>−0.91</td>
<td>0.03</td>
<td>0.94</td>
</tr>
<tr>
<td></td>
<td>180</td>
<td>−0.90</td>
<td>−0.17</td>
<td>0.73</td>
</tr>
</tbody>
</table>
Cyphos IL 101 was thus purified prior to determining its true EW according to a literature method as described in the ESI.†

**Electrochemical stability of purified Cyphos IL 101**

Fig. 2 shows the CV of purified Cyphos IL 101 recorded on a Pt or Mo electrode at 120 °C and 180 °C at a scan rate of 20 mV s⁻¹. The limiting potentials and the corresponding EW according to the experimental conditions are shown in Table 2. By removing the impurities present in the commercial Cyphos IL 101, the true EW of the electrolyte becomes observable. This true EW is about 1.5 times wider than the EW observed in the as-received commercial product. It is suggested that the limiting potentials correspond to the cathodic reduction of the phosphonium cation and the anodic oxidation of the chloride anion. When comparing the CV of the commercial Cyphos IL 101 (Fig. 1) to the purified Cyphos IL 101 (Fig. 2), a clear improvement was observed: for the purified Cyphos IL 101, the cathodic part of the window shows no reactions besides the decomposition of Cyphos IL 101 itself. Also, the decomposition of commercial Cyphos IL 101 is starting at a more negative potential than that of purified Cyphos IL 101. The impurities that are reduced during the first reductive wave are probably deposited on the electrode surface, thereby decreasing its active surface area.

In order to find an explanation for the narrowing of the electrochemical window of commercial Cyphos IL 101 by the reductive wave (Fig. 1), several known impurities of commercial Cyphos IL 101 were added separately to the purified Cyphos IL 101: 3 wt% of Cyanex 923 (mixture of four trialkylphosphine oxides)⁶² or 3 wt% of Cytop 360 (trihexylphosphine). Fig. S3 in the ESI† compares the CVs of purified Cyphos IL 101, commercial Cyphos IL 101 and purified Cyphos IL 101 containing 3 wt% of Cyanex 923 or 3 wt% of Cytop 360, recorded on a platinum electrode at 120 °C and 180 °C and a scan rate of 20 mV s⁻¹. When comparing the CVs, it is clear that the reduction wave in the CV of commercial Cyphos IL 101 cannot be attributed to phosphine oxide or trihexylphosphine impurities. No additional reduction wave, besides the decomposition of the phosphonium cation, was observed in the CV of purified Cyphos IL 101 containing 3 wt% of Cyanex 923 or Cytop 360. Vaughan and Dreisinger postulated that the additional reduction wave is attributed to the evolution of hydrogen from the [PR₃H]Cl and HCl impurities.⁶³ Since a difference in hydrogen overpotential is expected between molybdenum and platinum, the potential at which hydrogen gas becomes possible, depends on the metal used as working electrode. Cyphos IL 101 is prepared in industry via a quaternization reaction of trihexylphosphine (Cytop 360) with 1-chlorotetradecane (nucleophilic substitution S N₂).⁵⁵,⁵⁷ Several impurities present in Cytop 360 can be quaternized as well, ending up as impurities in Cyphos IL 101. The reduction of these quaternized impurities will only be seen in the CVs of commercial Cyphos IL 101 and are probably causing the additional reduction wave.

**Electrochemical study of indium in purified Cyphos IL 101**

The electrochemical behavior of indium in the purified Cyphos IL 101 was studied. Fig. 3 shows the CVs of the purified ionic liquid loaded with 10 mM of indium(In) recorded on a platinum electrode at 120 °C and 180 °C at a scan rate of 20 mV s⁻¹. The CVs show a cathodic wave in the forward cycle starting at $E = -2.15$ V vs. FeCl³⁻/Fe at 120 °C and $E = -1.80$ V vs. FeCl³⁻/Fe at 180 °C, corresponding to the reduction of indium(In). There are only small differences (about 0.1 V) in the standard reduction potentials ($E_0$) of the redox couple In³⁺/In, In⁵⁺/In⁴⁺ and In⁷⁺/In⁶⁺.⁶⁴ It is possible that the cathodic wave includes the following reactions.

![Image of CVs](image-url)

**Table 2 Limiting potentials and corresponding electrochemical window (EW) of purified Cyphos IL 101 recorded on different working electrodes (WE) and temperatures**

<table>
<thead>
<tr>
<th>WE</th>
<th>Temperature (°C)</th>
<th>Cathodic limit vs. Fe⁺³/Fe (V)</th>
<th>Anodic limit vs. Fe⁺³/Fe (V)</th>
<th>EW (V)</th>
</tr>
</thead>
<tbody>
<tr>
<td>Pt</td>
<td>120</td>
<td>-2.25</td>
<td>0.43</td>
<td>2.68</td>
</tr>
<tr>
<td></td>
<td>180</td>
<td>-2.19</td>
<td>0.38</td>
<td>2.57</td>
</tr>
<tr>
<td>Mo</td>
<td>120</td>
<td>-2.72</td>
<td>0.53</td>
<td>3.23</td>
</tr>
<tr>
<td></td>
<td>180</td>
<td>-2.38</td>
<td>0.49</td>
<td>3.07</td>
</tr>
</tbody>
</table>
The existence of indium(I) species in chloroindate(III) ionic liquid system, including the InCl$_3$–Cyphos IL 101 system, has already been demonstrated.$^{33,65}$ While the existence of indium(II) species in ionic liquids is still unclear. Based on this information, it is assumed that the In$^{3+}$/In$^+$ reduction takes place, although we do not exclude that both the In$^{3+}$/In$^+$ reduction and In$^{3+}$/In$^{2+}$ reduction are at play. A nucleation loop was observed, indicating a hindered growth on the electrode surface. This behavior is typical for processes which require a large overpotential to form a deposit. Also, a cathodic peak was observed in the backward scan, possibly resulting from the reduction of indium(I) to indium metal. At approximately $−1.11$ V vs. Fe$^+/Fc$ at 120 °C and $−1.26$ V vs. Fe$^+/Fc$ at 180 °C, an anodic peak was observed in the CVs which implies the stripping of the metallic indium deposited in the forward scan. The second oxidation peak in the CV starting at $−0.75$ V vs. Fe$^+/Fc$ at 120 °C and $−0.85$ V vs. Fe$^+/Fc$ at 180 °C is attributed to the stripping of various indium–platinum alloys.$^{66}$ At 180 °C, the broadness of this peak is likely due to the formation of more than one type of intermetallic compound. The fact that the stripping peak of the alloy is higher than that of pure indium indicates that the formation of intermetallics occurs rapidly and extensively, possibly due to the elevated temperature. The formation of indium–platinum alloys makes platinum less suitable as a WE for the recovery of pure indium.

The electrochemical behavior of indium in the purified Cyphos IL 101 was also studied on a molybdenum working electrode, as it does not alloy with indium.$^{67}$ Fig. 4 shows the CVs of the purified ionic liquid loaded with changing amounts of indium(III) (10, 20, 30, 40 and 50 mM) recorded at 120 °C and 180 °C and at a scan rate of 20 mV s$^{-1}$. Because a clear difference was observed between the first and second cycle at 120 °C, the first cycle was shown instead of the second one. In the previous CVs, the second cycle was identical to the first one. At 120 °C, the forward scan shows a cathodic peak between $−1.35$ V vs. Fe$^+/Fc$ and $−1.75$ V vs. Fe$^+/Fc$ probably corresponding to the reduction of indium(III) to indium(I). This peak is most outspoken at an indium concentration of 30 mM.

At higher or lower indium concentrations, it is difficult to distinguish this peak. At $−2.00$ V vs. Fe$^+/Fc$, indium(I) is

\begin{align}
\text{In}^{3+} + 3\text{e}^- & \rightarrow \text{In} \quad E_0 = −0.3382\text{ V} \\
\text{In}^{3+} + 2\text{e}^- & \rightarrow \text{In}^+ \quad E_0 = −0.443\text{ V} \\
\text{In}^{3+} + \text{e}^- & \rightarrow \text{In}^{2+} \quad E_0 = −0.49\text{ V} \\
\text{In}^{2+} + \text{e}^- & \rightarrow \text{In}^+ \quad E_0 = −0.40\text{ V}
\end{align}

The existence of indium(II) species in chloroindate(III) ionic liquid system, including the InCl$_3$–Cyphos IL 101 system, has already been demonstrated.$^{33,65}$ While the existence of indium(II) species in ionic liquids is still unclear. Based on this information, it is assumed that the In$^{3+}$/In$^+$ reduction takes place, although we do not exclude that both the In$^{3+}$/In$^+$ reduction and In$^{3+}$/In$^{2+}$ reduction are at play. A nucleation loop was observed, indicating a hindered growth on the electrode surface. This behavior is typical for processes which require a large overpotential to form a deposit. Also, a cathodic peak was observed in the backward scan, possibly resulting from the reduction of indium(I), created in the forward scan, to indium metal. At approximately $−1.11$ V vs. Fe$^+/Fc$ at 120 °C and $−1.26$ V vs. Fe$^+/Fc$ at 180 °C, an anodic peak was observed in the CVs which implies the stripping of the metallic indium deposited in the forward scan. The second oxidation peak in the CV starting at $−0.75$ V vs. Fe$^+/Fc$ at 120 °C and $−0.85$ V vs. Fe$^+/Fc$ at 180 °C is attributed to the stripping of various indium–platinum alloys.$^{66}$ At 180 °C, the broadness of this peak is likely due to the formation of more than one type of intermetallic compound. The fact that the stripping peak of the alloy is higher than that of pure indium indicates that the formation of intermetallics occurs rapidly and extensively, possibly due to the elevated temperature. The formation of indium–platinum alloys makes platinum less suitable as a WE for the recovery of pure indium.

The electrochemical behavior of indium in the purified Cyphos IL 101 was also studied on a molybdenum working electrode, as it does not alloy with indium.$^{67}$ Fig. 4 shows the CVs of the purified ionic liquid loaded with changing amounts of indium(III) (10, 20, 30, 40 and 50 mM) recorded at 120 °C and 180 °C and at a scan rate of 20 mV s$^{-1}$. Because a clear difference was observed between the first and second cycle at 120 °C, the first cycle was shown instead of the second one. In the previous CVs, the second cycle was identical to the first one. At 120 °C, the forward scan shows a cathodic peak between $−1.35$ V vs. Fe$^+/Fc$ and $−1.75$ V vs. Fe$^+/Fc$ probably corresponding to the reduction of indium(III) to indium(I). This peak is most outspoken at an indium concentration of 30 mM.

At higher or lower indium concentrations, it is difficult to distinguish this peak. At $−2.00$ V vs. Fe$^+/Fc$, indium(I) is
characteristic for an irreversible or quasi-reversible system.68 cathodic peaks, \( \Delta \) scan rate increases, the separation between the anodic and  above the two-step reduction process. peak is related to the oxidation of indium(I) to indium(III). fied Cyphos IL 101 loaded with 30 mM of indium(III) a t 120 °C of the electrochemical processes. Fig. S6 lyte, the scan rate was also varied to investigate the reversibility the electrode due to an incomplete stripping, facilitating the disproportionation reaction.

Besides varying the indium(III) concentration in the electrolyte, the scan rate was also varied to investigate the reversibility of the electrochemical processes. Fig. S6† shows the CVs of purified Cyphos IL 101 loaded with 30 mM of indium(III) at 120 °C with three consecutive cycles. The forward scan of the first cycle clearly shows a two-step reduction process, which seems to evolve into a one-step reduction process with increasing consecutive cycle. After each experiment, a grey indium metal layer was still present on the molybdenum working electrode, indicating incomplete indium stripping. The nucleation loop in the first cycle is also considerably bigger than in the second and third cycle. Since it is thermodynamically more favorable to deposit indium on indium(III)-containing electrolyte, starting between the two-step reduction process.

To investigate the involvement of indium(i) or even indium(ii) in the reduction process of indium(III) and oxidation process of indium metal in Cyphos IL 101, 30 mM of indium(i) chloride or 30 mM of indium(ii) chloride was added to the purified electrolyte. After addition of indium(i) chloride or indium(ii) chloride to the electrolyte, an immediate color change to silvery-grey was observed followed by particle coalescence resulting in the formation of a solid grey-silvery material, probably indium metal. The ionic liquid containing 30 mM of indium(II) also turned cloudy (Fig. S7†). It is hence likely that neither In(i) nor In(ii) are stable in Cyphos IL 101 and that (complete or partial) disproportionation of indium(i) (eqn (5)) and indium(ii) (eqn (6)) takes place:

\[
\begin{align*}
3\text{In}(i) &\rightleftharpoons 2\text{In}(0) + \text{In}(iii) \quad (5) \\
3\text{In}(ii) &\rightleftharpoons \text{In}(0) + 2\text{In}(iii) \quad (6)
\end{align*}
\]

As stated before, we do not exclude the existence of indium(i), but the formation of indium(i) in Cyphos IL 101 has already been demonstrated.65 The disproportionation behavior of indium(i) can explain the appearance of a two-step reduction process in the first cycle on molybdenum evolving into a one-step reduction process with an increasing number of cycles. It was already established that the disproportionation of indium(i) is very fast in Cyphos IL 101.65 It is possible that the disproportionation is hindered in the first cycle because a second phase (indium metal) must be formed in the electrolyte. In the second cycle, indium metal is already present on the electrode due to an incomplete stripping, facilitating the disproportionation reaction.

Fig. 6 shows the CVs of purified Cyphos IL 101 containing 30 mM of indium(i), 30 mM of indium(ii) or 30 mM of indium(III) recorded on a molybdenum electrode at 120 °C and 180 °C at a scan rate of 20 mV s\(^{-1}\). When comparing the CVs at 120 °C, it was noticed that the potential where the reduction current for indium(i) to indium(0) starts, in the indium(III)-containing electrolyte, coincides with the potential where the reduction starts of the indium present in the initial indium(i) and indium(ii)-containing electrolytes. Although disproportionation of indium(i) or indium(ii) takes place in these electrolytes forming indium(III) and indium metal, the CVs suggest that a certain concentration of indium(i) is in equilibrium with an indium(III) solution. The reduction wave starting at approximately −1.00 V vs. Fe\(^{3+}/Fe\(^{2+}\)) in the forward cycle of the CVs is attributed to the reduction of indium(i) to indium(0). It is to be expected based on the standard reduction potentials of indium, that this reaction will take place at less negative potentials (eqn (7)) than the reduction of indium(III) to indium(0) (eqn (1)) or of indium(III) to indium(i) (eqn (2)).64

\[
\begin{align*}
\text{In}\(^{+}\) + e\(^{-}\) &\rightleftharpoons \text{In} \quad E_0 = -0.14 \text{ V} \quad (7) \\
\text{In}\(^{2+}\) + e\(^{-}\) &\rightleftharpoons \text{In}\(^{+}\) \quad E_0 = -0.40 \text{ V} \quad (8)
\end{align*}
\]

At 180 °C, a clear difference in the CVs was observed. The reduction wave in the indium(III)-containing electrolyte no longer coincides with the reduction wave in the initial indium(i)- and indium(ii)-containing electrolytes. The cathodic peak, in the indium(III)-containing electrolyte, starting between

---

**Fig. 5** CVs (three consecutive cycles) of purified Cyphos IL 101 loaded with 30 mM of indium(III) on a Mo working electrode (WE) at 120 °C. Scans were recorded at \( v_{\text{scan}} = 20 \text{ mV s}^{-1} \).
−1.35 V vs. Fe⁺/Fe and −1.50 V vs. Fe⁺/Fe, probably involves the reduction of indium(III) to indium(II) and subsequently to indium metal while the cathodic peak starting at −1.00 V vs. Fe⁺/Fe, in the indium(II)- and indium(III)-containing electrolytes, can be attributed to the reduction of indium(II) or indium(III) to indium(0). It is possible that the exchange current density for the indium(II) to indium(0) reduction is not high enough at 120 °C to observe a reduction wave in the CV of the initial indium(II)- and indium(III)-containing electrolytes. At 180 °C, the exchange current density is high enough to create a visible reduction wave.

To get a better insight in the cathodic reactions that take place, the potentiostat was coupled to an electrochemical quartz crystal microbalance (EQCM). A CV of the blank electrolyte with EQCM analysis was measured first as a reference situation and to determine if the reduction or oxidation of impurities leads to a change in mass on the electrode. The mass change of the quartz crystals is correlated to the frequency change using the Sauerbrey equation:

\[
\Delta m = -C_f \Delta f
\]

(9)

where \(\Delta f\) (Hz) is the change in the resonance frequency of the crystal (working electrode) due to the change in mass \(\Delta m\) (g cm\(^{-2}\)) of the crystal; \(C_f\) (g Hz\(^{-1}\) cm\(^{-2}\)) is the calibration constant. Fig. S9† shows the CV of purified Cyphos IL 101 with EQCM analysis. The CV was recorded at 120 °C on a Pt electrode with a scan rate of 2 mV s\(^{-1}\). Since no significant shift in frequency was observed from 0.50 V vs. Fe⁺/Fe to −1.50 V vs. Fe⁺/Fe in the forward or backward cycle, it can be assumed that the reduction and oxidation of impurities is not accompanied by a mass change. Also no significant mass change was observed between −1.50 V vs. Fe⁺/Fe to −3.00 V vs. Fe⁺/Fe together with the cathodic decomposition of the ionic liquid, indicating that it is not accompanied with the deposition of cathodic decomposition products or the formation of a solid compound on the WE. In the backward cycle, a high loss of mass was observed at the potential where anodic decomposition of the ionic liquid takes place. The formed chlorine gas reacted with the platinum WE forming platinum chloride that dissolves in the ionic liquid.

Fig. 7 shows the CV of purified Cyphos IL 101 loaded with 50 mM of indium(III) with EQCM analysis. The CV was recorded at 120 °C on a Pt electrode with a scan rate of 2 mV s\(^{-1}\). A significant decrease in frequency was observed starting from −1.53 V vs. Fe⁺/Fe in the forward cycle to −0.81 V vs. Fe⁺/Fe in the backward cycle, indicating that deposition of indium(0) takes place. Although a noteworthy negative current starting from −0.77 V vs. Fe⁺/Fe is observed in the forward cycle of the CV, it is not accompanied by a significant frequency change. This indicates the presence of a second reduction process without deposition starting at −0.77 V vs. Fe⁺/Fe, most probably the reduction of indium(III) to indium(II). The reduction of indium(II) to indium(0) is presumably also included in the cathodic wave starting from −1.53 V vs. Fe⁺/Fe. From −0.81 V vs. Fe⁺/Fe to 0.83 V vs. Fe⁺/Fe in the backward cycle an increase in frequency was observed arising from the stripping of indium(0) and indium–platinum alloys. Between 0.83 V vs. Fe⁺/Fe in the backward cycle and 0.71 V vs. Fe⁺/Fe in the forward cycle an increase of frequency is observed due to the anodic decomposition of the ionic liquid. The theoretical amount of indium deposited from −0.77 V vs. Fe⁺/Fe in the forward cycle to −0.81 V vs. Fe⁺/Fe in the backward cycle is calculated and compared to the actual mass deposited. A current efficiency of 70% was obtained, leading to the conclusion that
some of the current is used for the reduction of indium(m) to indium(i) and/or the reduction of impurities.

To study the indium electrodeposition mechanism further, a curve of mass/charge (m/z) as a function of the potential was constructed based on Faraday’s law and the Sauerbrey equation:

$$\frac{m}{z} = -\frac{F_i \Delta f}{\Delta Q}$$

(10)

where $M$ (g mol$^{-1}$) is the molar mass of the reducing/oxidizing agent; $z$ is the number of electrons involved in the electrochemical reaction; $F = 96 485.34$ (C mol$^{-1}$) is the Faraday constant; $\Delta Q$ (C cm$^{-2}$) is the change in charged consumed during the electrochemical reaction. Eqn (10) can be used to determine $m/z$ as a function of applied potential, which allows to examine changes in reaction mechanisms along a CV. However, when the current in CV goes through zero, experimentally determined $m/z$ values become uncertain. Therefore, these $m/z$ values are removed from the graph. In this paper, we use the absolute negative value of charge change $\Delta Q$ in eqn (10), so that $m/z$ has an opposite sign as $\Delta m$, i.e. negative for deposition and positive for a stripping process.

Fig. 2 shows the CV and EQCM data analyzed for m/z of purified Cyphos IL 101 loaded with 50 mM of indium(i). The value of m/z when indium is being deposited with 100% efficiency, i.e., without any other parallel process on the surface of the electrode is $38.27$ g mol$^{-1}$. From Fig. S10, it was observed that experimental m/z values obtained during indium reduction start at quite low m/z values. At this stage, indium(i) is mainly reduced to indium(i). Indium(i) is not reduced further to indium(0) and is lost in solution due to disproportionation. As the potential is shifted to more negative values, the reaction rate for the reduction of indium(i) to indium(0) is increased and the m/z values moved toward a value of $-38.27$ g mol$^{-1}$ as the deposition advanced and stays constant for a while. From $-2.47$ V vs. Fc$^+/Fc$ in the forward cycle an $-2.45$ V vs. Fc$^+/Fc$ in the backward cycle, decomposition of the ionic liquid takes place, leading to more positive m/z values. The more positive m/z values that are obtained outside of the decomposition potentials of the ionic liquid, but still within the cathodic current range, are due to the simultaneous electrochemical reduction of indium(iii) to indium(i) and subsequently disproportionation in combination with the reduction of indium(i) to indium(0).

The stripping of indium(0) and indium–platinum alloys takes place from $-0.81$ V vs. Fc$^+/Fc$ to $0.83$ V vs. Fc$^+/Fc$ in the backward cycle. From $-0.43$ V vs. Fc$^+/Fc$ to $0.23$ V vs. Fc$^+/Fc$ in the backward cycle, theoretical m/z values more positive than $38.27$ g mol$^{-1}$ are obtained indicating that the main electrochemical process taken place is the oxidation of indium(0) to indium(i). Starting from $0.23$ V vs. Fc$^+/Fc$ to $0.83$ V vs. Fc$^+/Fc$ in the backward cycle the theoretical m/z values vary between $38.27$ g mol$^{-1}$ and $31.30$ g mol$^{-1}$, illustrating that the simultaneous oxidation of indium(0) to indium(i) followed by a disproportionation reaction of indium(i) is accompanied with oxidation of indium(i) to indium(m). Between $0.83$ V vs. Fc$^+/Fc$ in the backward cycle and $0.71$ V vs. Fc$^+/Fc$ in the forward cycle the anodic decomposition of the ionic liquid takes place accompanied with chlorine gas evolution. Due to the dissolution of Pt in the ionic liquid as platinum chloride the m/z values become uncertain.

This hypothesis is confirmed by using a rotating ring disk electrode (RRDE). Cyclic voltammetry was applied on the disk, while the ring was kept at a potential of $0.5$ V vs. Fc$^+/Fc$ (Fig. 8). During the cathodic scan, indium(iii) was reduced to indium(i), but when the indium(i) ions reached the ring, they were re-oxidized to indium(iii). In the anodic scan, indium(0) is oxidized to indium(i) and transported to the ring where it is further oxidized to indium(iii). Indium(m) is being reduced to indium(0) by a two-step process involving indium(i) and vice versa re-oxidized. A current efficiency of 59% was obtained under the assumption that only indium(i) is produced on the disk and that every indium(i) ion produced also reaches the ring.

The two-step deposition reaction of indium(iii) to indium(i) and subsequently to indium(0) has already been reported for several chloroindate(iii) ionic liquids including Cyphos IL 101. Estager et al. described a method for establishing the formation of indium(i). By holding the working electrode (platinum) at a potential between that for the reduction to indium(i), and that for electrodeposition of indium(0), electroplating of indium(0) was prevented. The produced indium(i)-containing anions were immediately dispersed in the electrolyte by vigorous stirring. Subsequently, indium(i) disproportionated into indium(0) and indium(iii) in the bulk electrolyte leading to the formation of relatively large (>1000 nm), irregular ‘chunks’ of indium in the case of Cyphos IL 101 due to poor capping agent abilities of the bulky phosphonium cation. In this work, electrodeposition of indium(iii) to indium(i) on a platinum working electrode was performed at constant
potential \( E = -1.65 \text{ V vs. } \text{Fc}^+/\text{Fc} \) from purified Cyphos IL 101 with 400 mM indium(III) at 120 °C for 18 h. The experiment was performed in a small porcelain crucible \((V = 3 \text{ mL})\) under continuous stirring \((1000 \text{ rpm})\) with a magnetic stirring bar. After 18 h, irregular ‘chunks’ of indium were observed inside the electrolyte and a small amount of indium was found on the platinum electrode (Fig. S11). The scanning electron microscopy (SEM) images of these ‘chunks’ are shown in Fig. S12. The ‘chunks’ seem to be composed of a great number of smaller pieces sticking together. These ‘chunks’ did not just fall off the working electrode as a consequence of large amounts of indium being deposited. Indium forms alloys with platinum, so it is expected that a thin film would be formed covering the entire surface. This was not the case, not even when, the surface of the platinum working electrode was abraded with sand paper to improve the adhesion. From these results it can be concluded that indium(I) is indeed involved in the reduction of indium(III).

The morphology of indium deposits on a molybdenum electrode were examined by SEM. The primary concern regarding the indium deposition from chloride medium is the release of chlorine gas at the anode. Fortunately, it has been recently proven that phosphonium ionic liquids are stable in the presence of chlorine gas.69 Electrodeposition of indium on molybdenum was performed at constant potential \( E = -2.00 \text{ V vs. } \text{Fc}^+/\text{Fc} \) from purified Cyphos IL 101 with 400 mM indium(III) at 100 °C, 120 °C, 140 °C, 160 °C and 180 °C for 2 h. The temperature of the electrolyte was measured by placing a thermocouple imbedded in glass in the ionic liquid. The experiments were performed in a small porcelain crucible \((V = 3 \text{ mL})\). At all temperatures silvery white deposits were observed on the molybdenum electrode (Fig. 9, 10 and Fig. S13, S14†). From the SEM images, it can be seen that at all temperatures (so even at 100 °C, which is far below the melting point of indium of 156.6 °C) indium deposited as spherical droplets. Below the melting point of indium, coarse indium particles were occasionally observed between the droplets. At temperatures above the melting point of indium, only droplets were observed and some of the spherical particles have a vein-like surface. This is due to solidification of the liquid indium droplets after deposition. If the place of last solidification occurs on the surface, the surface shrinks due to the increase in indium density causing the appearance of vein-like structures. As the temperature increases, so does the packing of droplets. A close packing of droplets, showing droplets on top of other droplets, implying droplet-on-droplet deposition without coalescence was observed starting from 120 °C. Since a continuous indium layer was not obtained, it can be concluded that indium metal does not wet molybdenum. Care had to be taken to avoid removal of the deposits when washing the electrode due to bad adhesion.

Two possible phenomena can be responsible for the morphologies of the indium that is electrodeposited below the melting point of indium: (1) melting point depression in combination with undercooling or (2) dewetting. Size-dependent melting point depression is an important phenomenon for small particles of nanometer size. These particles have a lower melting point than the bulk material. This is due to the increased influence of the surface atoms as the ratio of surface to bulk atoms increases. As the size of the particles decreases, an increased proportion of atoms occupy surface or interfacial sites. These atoms are more loosely bound than bulk atoms, which facilitates the melting.70 The size-dependent melting point depression of indium nanoparticles has been studied.70,71 According to these literature reports, indium particles are liquid at 100 °C when their radius is 3 to 4 nm. If the indium particles being deposited have a radius of 3 to 4 nm, they are deposited in the liquid form. These particles could coalesce without solidifying (undercooling), increasing in size. The presence of coarse indium particles in between the droplets at temperatures below the melting point of indium is an indication that melting point depression could take place. As

![Fig. 9](image1.png)

**Fig. 9** SEM picture of indium deposited on a molybdenum working electrode at \(-2.00 \text{ V vs. } \text{Fc}^+/\text{Fc} \) at 100 °C for 2 h from purified Cyphos IL 101 loaded with 400 mM of indium(III).

![Fig. 10](image2.png)

**Fig. 10** SEM picture of indium deposited on a molybdenum working electrode at \(-2.00 \text{ V vs. } \text{Fc}^+/\text{Fc} \) at 160 °C for 2 h from purified Cyphos IL 101 loaded with 400 mM of indium(III).
can be seen in Fig. 9, all small particles are spherical while the largest particles are non-spherical. This is an indication that the spherical particles are liquid during deposition and consist of undercooled indium. As the particle grow in size by coalescence, the probability for nucleation increases and they become solid and non-spherical.

During solid-state dewetting, thin films, metastable in the as-deposited state, dewet to form isolated islands while the material remains in the solid state. The driving force is minimization of the total energy of the free surfaces of the film and substrate, and of the film-substrate interface.\(^72\) Dewetting can occur via surface self-diffusion. Eckert and Drickamer studied the bulk self-diffusion coefficient of indium near the melting point of indium. At 100 °C, the bulk self-diffusion coefficient of indium is approximately \(5 \times 10^{-14}\) m\(^2\) s\(^{-1}\).\(^73\) The surface diffusion coefficient of metals can be 10 to 100 times higher than the bulk diffusion coefficient.\(^74\)–\(^76\) Considering a surface self-diffusion coefficient equal to, 10 or 100 times larger than the bulk self-diffusion coefficient, the distance over which deposited indium can diffuse to form spherical particles at 100 °C over a time span of 2 h can be estimated from eqn (11) (Table 3):

\[
x = \sqrt{qDt} = \sqrt{2Dt}
\]

in which \(x\) is the mean displacement, \(D\) is the surface self-diffusion coefficient and \(t\) is the time. \(q\) is a constant depending on the dimensions of the displacement: 2 (1D), 4 (2D), 6 (3D).

The average distance between two single spherical indium particles deposited at 100 °C on the molybdenum electrode is 6.7 µm, so that solid-state dewetting is possible. Liquid state dewetting can explain the droplet deposition above the melting point of indium. The bulk self-diffusion coefficient of indium slightly above the melting point of indium, at 157.30 °C, is around \(1 \times 10^{-9}\) m\(^2\) s\(^{-1}\), a factor \(2 \times 10^4\) higher than at 100 °C.\(^73\) Considering that the surface self-diffusion coefficient above the melting point of indium is even higher, liquid state dewetting is possible.

Droplet-on-droplet deposition could be explained by the formation of an a surface oxidation layer that is slowly, step by step encapsulating the droplet. Coalescence is prevented as soon as the oxidation layer completely covers the droplet. The \(\text{In}_2\text{O}_3\) layer is more conducting than indium metal and indium will preferentially be deposited on top of the indium droplets covered with \(\text{In}_2\text{O}_3\). Although the electrodeposition was executed in an oxygen-free environment (oxygen content glovebox < 1 ppm), oxygen can be released by the reduction of oxide-containing impurities or by the reduction of amounts of water still present inside the ionic liquid. Although the reduction of phosphine oxide impurities did not occur, other oxide impurities can be present in Cyphos IL 101.\(^75\) Cyphos IL 101 is synthesized through a quaternization reaction of trihexylphosphine with 1 chlorotetradecane (nucleophilic substitution \(\text{SN}_2\)).\(^55\)–\(^57\) When phosphines are oxidized in air, especially trialkyl phosphines, a multitude of oxygen containing byproducts are possible formed.\(^77\)–\(^78\) Another possibility is that due to the bad wetting of indium metal on molybdenum deposited indium droplets flock together under the influence of capillary forces during drying. In that case, the flocks of indium droplets stacked on top of each other do not originate during deposition but form after deposition.

In order to get more inside in the deposition mechanism and exclude several hypotheses, a dewetting experiment was

**Table 3** Mean diffusion distance \((x)\) of indium atoms at 100 °C during a time span of 2 h as a function of the surface diffusion coefficient

<table>
<thead>
<tr>
<th>Surface diffusion coefficient (m(^2) s(^{-1}))</th>
<th>(x) (µm)</th>
</tr>
</thead>
<tbody>
<tr>
<td>(5 \times 10^{-14})</td>
<td>27</td>
</tr>
<tr>
<td>(5 \times 10^{-13})</td>
<td>32</td>
</tr>
<tr>
<td>(5 \times 10^{-12})</td>
<td>100</td>
</tr>
</tbody>
</table>
Electrochemical study of zinc and iron on Mo in purified Cyphos IL 101

Zinc and iron are often associated with indium due to their presence in indium-containing ores or by-products. Therefore, the capability of Cyphos IL 101 as an electrolyte for the separation of indium and zinc or iron and indium was investigated. CVs of purified Cyphos IL 101 containing 30 mM of zinc(II) or iron(III) chloride were recorded on a Mo WE at 120 °C and 180 °C at a scan rate of 20 mV s⁻¹ (Fig. 13). A small reduction peak was observed starting at −0.32 V vs. Fe⁺/Fe in the forward cycle, probably due to the reduction of a small amount of impurities left in the purified Cyphos IL 101. The corresponding oxidation peak started at 0.32 V vs. Fe⁺/Fe in the backward cycle and coincided with the anodic decomposition of the ionic liquid. This indicates that co-deposition of zinc with indium can be avoided. In the CV of purified Cyphos IL 101 containing 30 mM of iron(III) a reduction peak starting at 0.09 V vs. Fe⁺/Fe was observed in the forward cycle. This peak is assigned to the reduction of iron(III) to iron(II) (eqn (14)). The absence of a corresponding oxidation peak can be explained by the comproportionation reaction between iron(II) and iron(III) forming iron(II) (eqn (14)) or by the formation of a passivation layer, FeCl₂. FeCl₃-Cyphos IL 101 is a Lewis-basic ionic liquid with an excess of Cyphos IL 101 (FeCl₃-Cyphos IL 101 mole ratio <0.50) and free chlorides in equilibrium with [FeCl₄]⁻. Upon dissolution of iron, free chlorides are available making the passivation of iron by FeCl₂ very unlikely. The passivation of metals is reported in acidic chlorometallate ionic liquids (MCl₃⁻ ionic liquid mole ratio >0.50). In Lewis-acidic ionic liquids the activity of free chlorides is insignificant. Upon dissolution of the metal, the necessary ligands will have to come from the dimeric [Me₂Cl₄]⁻ anions, decreasing the Lewis acidity of the electrolyte at the interface. The solubility of metal chlorides decreases with increasing Lewis acidity of the ionic liquid, forming a passivation layer upon precipitation. The reduction of iron(III) to iron(II) occurred at more positive potentials than the reduction of indium, so that it is difficult to separate iron from indium by electroreduction.
Electrodeposition of iron and zinc on molybdenum was performed at constant potential from purified Cyphos IL 101 loaded with 400 mM iron(II), 400 mM zinc(II) and 400 mM indium(III) at 120 and 180 °C. A 3 h measurement at −1.80 V vs. Fc+/Fc and a 2 h measurement at −2.00 V vs. Fc+/Fc were performed. The experiments were performed in a small porcelain crucible (V = 3 mL). Grey-black deposits were observed on the molybdenum electrodes. Scanning electron microscopy with energy-dispersive X-ray spectroscopy (SEM-EDS) was executed on the deposits. EDS confirmed the presence of iron and indium, but not zinc (Fig. S19–S22†). The composition of the deposits was determined by total reflection X-ray fluorescence spectroscopy (TXRF) (Table 4). Very little zinc could be detected. Also, the TXRF results state a strong overestimation of the real concentration of iron in the material deposited at 180 °C. Iron has a limited solubility in indium at 180 °C of the real concentration of iron in the material deposited at 180 °C. Indium and iron form two phases. While indium

<table>
<thead>
<tr>
<th>Element</th>
<th>120 °C, 3 h, −1.8 V vs. Fc+/Fc</th>
<th>180 °C, 3 h, −1.8 V vs. Fc+/Fc</th>
<th>120 °C, 2 h, −2.0 V vs. Fc+/Fc</th>
<th>180 °C, 2 h, −2.0 V vs. Fc+/Fc</th>
</tr>
</thead>
<tbody>
<tr>
<td>In</td>
<td>89.9%</td>
<td>25.9%</td>
<td>99.0%</td>
<td>82.8%</td>
</tr>
<tr>
<td>Fe</td>
<td>9.8%</td>
<td>73.6%</td>
<td>0.8%</td>
<td>17.0%</td>
</tr>
<tr>
<td>Zn</td>
<td>0.3%</td>
<td>0.4%</td>
<td>0.2%</td>
<td>0.2%</td>
</tr>
</tbody>
</table>

Table 4 Composition of the deposits on a molybdenum working electrode from purified Cyphos IL 101 loaded with 400 mM iron(II), 400 mM zinc(II) and 400 mM indium(III), determined by TXRF

Fe+/Fc. Electrodeposition of iron and zinc on molybdenum was performed at constant potential from purified Cyphos IL 101 loaded with 400 mM iron(II), 400 mM zinc(II) and 400 mM indium(III) at 120 and 180 °C. A 3 h measurement at −1.80 V vs. Fc+/Fc and a 2 h measurement at −2.00 V vs. Fc+/Fc were performed. The experiments were performed in a small porcelain crucible (V = 3 mL). Grey-black deposits were observed on the molybdenum electrodes. Scanning electron microscopy with energy-dispersive X-ray spectroscopy (SEM-EDS) was executed on the deposits. EDS confirmed the presence of iron and indium, but not zinc (Fig. S19–S22†). The composition of the deposits was determined by total reflection X-ray fluorescence spectroscopy (TXRF) (Table 4). Very little zinc could be detected. Also, the TXRF results state a strong overestimation of the real concentration of iron in the material deposited at 180 °C. Iron has a limited solubility in indium at 180 °C of the real concentration of iron in the material deposited at 180 °C. Indium and iron form two phases. While indium

Hsiu et al. investigated the Lewis acidity dependency of the electrochemical window of zinc chloride–1-ethyl-3-methylimidazolium chloride ionic liquids.95 It was noticed that at a ZnCl4−:ionic liquid molar ratio of 1 : 3, the cathodic limit of the electrochemical window corresponds to the reduction of the imidazolium cation rather than the electrodeposition of zinc metal. Zn(II) is present as anionic ZnCl42−, which is more difficult to reduce than the imidazolium cation. In our system, the ZnCl4− ionic liquid ratio is also less than 1 : 3, explaining the absence of zinc(II) deposition, which is due to the presence of Lewis-basic ZnCl42− anions.93 A ZnCl4−:ionic liquid ratio above 1 : 3 will probably shift the cathodic limit of the electrochemical window to the deposition of zinc(II) due to the presence of Lewis-acidic zinc(II) species of which the number of chloride that complexed to the zinc is less than four.

**Indium speciation in purified Cyphos IL 101**

It has been reported in the literature that the speciation of indium in chloroindate(III) ions based on 1-alkyl-3-methylimidazolium cations ([Cmim]3−, where n = 4 or 8), is dependent of the molar ratio of indium(III) chloride to the organic chloride salt (commonly expressed as the effective mole fraction of ‘monomeric’ indium(III) chloride, xInCl3).94–96 [Cmim]+-based chloroindate(III) ionic liquids with a composition of xInCl3 = 0.25 have been shown to contain the [InCl4]3− anion as the major indium species. As more indium(III) chloride was added to the mixture, the major indium species gradually changed to [InCl4]3− and [InCl5]− at xInCl3 = 0.33 and xInCl3 = 0.50, respectively. To verify if the same indium speciation is also present in Cyphos IL 101, EXAFS spectra were recorded of indium(III) chloride dissolved in dry Cyphos IL 101 in three mole fractions xInCl3 = 0.20, 0.25 and 0.33 (Fig. 14). Reference EXAFS spectra of indium(III) chloride dissolved in [C8mim]Cl at the same mole fractions xInCl3 = 0.20, 0.25 and 0.33 were recorded for comparison. By fixing the amplitude reduction factor at 0.9, the coordination numbers of all three reference samples matched well with the expected main indium(III) species, namely [InCl4]3− for xInCl3 = 0.33 and [InCl5]− for xInCl3 = 0.25 and 0.20. The same value of the amplitude reduction factor was also used in the fitting of the Cyphos IL 101 samples. The figures with the fitted models of all samples can be found in the ESI (Fig. S23 and S24†). Unexpectedly, the fitted coordination number and the bond distance of all three Cyphos IL 101 samples indicates that the main species in all three Cyphos IL 101 samples is [InCl4]3− (Table 5). This can also be observed in the Fourier-transform, where the peak of all three Cyphos IL 101 samples lies close to the peak of the xInCl3 = 0.33 reference sample, containing [InCl5]− as the main species (Fig. 15). The bond distance increased slightly with decreasing amount of

![Fig. 14 EXAFS functions of indium(III) chloride dissolved in [C8mim]Cl: (a) xInCl3 = 0.33, (b) xInCl3 = 0.25 and (c) xInCl3 = 0.20; and in Cyphos IL 101: (d) xInCl3 = 0.33, (e) xInCl3 = 0.25 and (f) xInCl3 = 0.20.](image-url)
indium(III) chloride dissolved in Cyphos IL 101, from 2.430(2) at $\text{xInCl}_3 = 0.33$ to 2.440(2) at $\text{xInCl}_3 = 0.20$, thus indicating that a small, but significant and increasing, fraction of $[\text{InCl}_5]^{3-}$ is present when lowering the indium(III) concentration. It should be stressed that the indium speciation in dried Cyphos IL 101 clearly differs from that in water-saturated Cyphos IL 101. As reported before, in contrast to the indium speciation in dried Cyphos IL 101, the indium speciation in water-saturated Cyphos IL 101 is independent of the indium concentration, i.e. $[\text{InCl}_4]^-$. An indium(III) chloride concentration between 10 and 400 mM in Cyphos IL 101 was used in the previous experiments corresponding to a composition $\text{xInCl}_3 = 0.19$ and 0.006, respectively. Therefore, based on the EXAFS results it can be concluded that the major indium(III) species in solution with composition $\text{xInCl}_3 = 0.19$ corresponds to the $[\text{InCl}_5]^{3-}$ complex. As the indium(III) chloride concentrations decreases, it is expected that an increasing fraction of $[\text{InCl}_6]^{3-}$ will be present. However, the data are inconclusive with respect to the major indium(III) species at very low indium(III) chloride concentrations. It is unclear whether $[\text{InCl}_6]^{3-}$ will become the major indium(III) species in solution with ever decreasing indium(III) chloride concentration. As a consequence, the calculated diffusion coefficient $(7.5 \pm 0.1) \times 10^{-12} \text{ m}^2 \text{s}^{-1}$ at 120 °C (ESI†) could belong to either the $[\text{InCl}_5]^{2-}$ or the $[\text{InCl}_6]^{3-}$ anion in Cyphos IL 101 (10 mM In, 120 °C).

**Table 5** Fitting parameters of the EXAFS spectra of indium(III) chloride dissolved in [C$_{8}$mim]Cl and Cyphos IL 101 at three mole fractions $\text{xInCl}_3 = 0.33, 0.25$ and 0.20. The data of $\text{xInCl}_3 = 0.50$ in Cyphos IL 101 were determined in previous work and added for comparison

<table>
<thead>
<tr>
<th>System</th>
<th>$N$ $(-)^b$</th>
<th>$R'$ (Å)</th>
<th>$2\sigma^d$ (Å$^2$)</th>
<th>Main species</th>
</tr>
</thead>
<tbody>
<tr>
<td>$\text{xInCl}<em>3 = 0.33$ in [C$</em>{8}$mim]Cl (reference)</td>
<td>5.0(1)</td>
<td>2.432(2)</td>
<td>0.008(1)</td>
<td>$[\text{InCl}_5]^{2-}$</td>
</tr>
<tr>
<td>$\text{xInCl}<em>3 = 0.25$ in [C$</em>{8}$mim]Cl (reference)</td>
<td>6.0(1)</td>
<td>2.489(2)</td>
<td>0.008(1)</td>
<td>$[\text{InCl}_6]^{3-}$</td>
</tr>
<tr>
<td>$\text{xInCl}<em>3 = 0.20$ in [C$</em>{8}$mim]Cl (reference)</td>
<td>6.1(1)</td>
<td>2.500(2)</td>
<td>0.008(1)</td>
<td>$[\text{InCl}_6]^{3-}$</td>
</tr>
<tr>
<td>$\text{xInCl}_3 = 0.33$ in Cyphos IL 101</td>
<td>5.1(1)</td>
<td>2.430(2)</td>
<td>0.007(1)</td>
<td>$[\text{InCl}_5]^{2-}$</td>
</tr>
<tr>
<td>$\text{xInCl}_3 = 0.25$ in Cyphos IL 101</td>
<td>5.3(1)</td>
<td>2.441(2)</td>
<td>0.007(1)</td>
<td>$[\text{InCl}_5]^{2-}$</td>
</tr>
<tr>
<td>$\text{xInCl}_3 = 0.20$ in Cyphos IL 101</td>
<td>5.2(1)</td>
<td>2.440(2)</td>
<td>0.007(1)</td>
<td>$[\text{InCl}_5]^{2-}$</td>
</tr>
<tr>
<td>$\text{xInCl}_3 = 0.50$ in Cyphos IL 101 $^a$</td>
<td>4.0(1)</td>
<td>2.364(2)</td>
<td>0.004(1)</td>
<td>$[\text{InCl}_4]^{-}$</td>
</tr>
</tbody>
</table>

$^a$ Data from previous work. $^{14}$ $^b$ $N$ is the number of coordinating chlorine atoms. $^c$ $R'$ is the interatomic distance. $^d$ $\sigma$ is the Debye-Waller factor.

**Conclusions**

The electrochemical behavior of indium in the ionic liquid trihexyl(tetradecyl)phosphonium chloride (Cyphos IL 101) was investigated at 120 °C and 180 °C. Commercial Cyphos IL 101 had to be purified due to interference of impurities in the electrochemical measurements. The electrochemical deposition of indium followed a two-step process, involving the reduction of indium(III) to indium(II), followed by the reduction of indium(II) to indium(0). Indium was deposited at temperatures ranging from 100 to 180 °C. The morphology of the indium deposits was examined by SEM. At all temperatures, droplet-like deposits of indium were observed. The formation of droplets has a two-fold origin: melting-point depression of very small primary indium particles in combination with undercooling. The capability of the electrolyte to eliminate iron(III) and zinc(II) electrochemically was tested by adding indium(III) together with zinc(II) and iron(III) to Cyphos IL 101. Zinc(II) was not co-deposited with indium(III) because the formation of the Lewis-basic ZnCl$_4^{2-}$ anions shifts the deposition potential outside of the cathodic limit of the phosphonium cation. Iron(III) could not be separated from indium(III) in this way. A diffusion coefficient of $(7.5 \pm 0.1) \times 10^{-12} \text{ m}^2 \text{s}^{-1}$ was obtained for the chloroindate(III) anion in Cyphos IL 101 (10 mM In, 120 °C).

**Conflicts of interest**

There are no conflicts to declare.

**Acknowledgements**

This research was supported by the Flemish Institute for the Promotion of Innovation by Science and Technology (IWT Vlaanderen) via a Baekeland PhD fellowship to Clio Deferm (IWT 130305) and by the Umicore Group Research & Development. The research leading to these results also received funding from the European Research Council (ERC) under the European Union’s Horizon 2020 Research and Development Program (ERC grant agreement 306765).
Innovation Programme: Grant Agreement 694078—Solvometallurgy for critical metals (SOLCRIMET). Joop van Deursen made and designed the EQCM cell. The authors also wish to thank FWO for funding the EXAFS measurements, DUBBLE staff for their support, Wouter Monnens for his assistance during measuring shifts at the beamline, Alina Arslanova for providing Matlab support and Frederik Ceyssens for the supercritical drying experiments.

Notes and references