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Assignment of the solid state spectra of the group VI hexacarbonyls by inelastic neutron scattering spectroscopy*

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The solid state vibrational spectra of $M(CO)_6$, (M = Cr, Mo, W) in the region below 800 cm⁻¹ have been assigned by a combination of infrared, Raman and the first reported inelastic neutron scattering (INS) spectra from homoleptic metal carbonyls. This region comprises of the lattice modes, the OC-M-CO deformations, the M-C \equiv O bends and the M-C stretches. Three modes that are forbidden in both the infrared and Raman spectra of the parent O_h symmetry gas phase molecule occur in this region. The absence of selection rules for INS spectroscopy means that all three modes are clearly seen for the first time, all previous work has relied on overtone and combination modes. Periodic density functional theory calculations of the complete orthorhombic structure support the assignments.

Introduction

The group VI metal hexacarbonyls $M(CO)_6$, (M = Cr, Mo, W)have been extensively studied over the years.1 Their uses are as diverse as precursors for organometallic reagents in synthetic chemistry² and as metal sources in chemical vapour deposition processes.3-5 The vibrational spectroscopy of the compounds in the gas phase has been comprehensively investigated⁶⁻⁹ and empirical^{6,7,9} and ab initio¹⁰⁻¹² force fields derived. In contrast, the solid state spectra have been much less investigated, the most complete study was by Adams and Taylor¹³ who recorded polarised infrared and Raman spectra from single crystals at 77 K. The spectra in the carbonyl stretch region have also been interpreted using a novel approach.14

However, there are still uncertainties in some of the assignments and these relate to modes that are both infrared and Raman forbidden under the gas phase molecule's O_b symmetry. Knowledge of these modes is largely based on overtone and combination modes and also that they become weakly allowed in the solid state. In the present work we use a combination of inelastic neutron scattering (INS) spectroscopy¹⁵ and periodic density functional theory 16,17 (periodic-DFT) to provide complete assignments of the solid state spectra.

Experimental

The INS spectra were recorded using TOSCA.¹⁸ 10-15 g of the compound (Cr(CO)₆ 98%, Mo(CO)₆ 99.9%, W(CO)₆ 97%, all from Aldrich and used as received) was loaded into a sealed can in an argon-filled dry box. The sample was then cooled to ~ 20 K and the spectrum recorded for ~12 hours. Raman spectra were recorded at 7 K using an upgraded version of a Renishaw InVia spectrometer, that has been previously described. 19 The upgrade consists of an additional laser operating at 532 nm and improved laser rejection filters with a low energy cut-off of $\sim 40 \text{ cm}^{-1}$. Room temperature infrared spectra were recorded using a Bruker Vertex70 FTIR spectrometer, over the range 20 to 4000 cm⁻¹ at 4 cm⁻¹ resolution with a DLaTGS detector using 64 scans and the Bruker Diamond ATR. The use of the ultra-wide range beamsplitter enabled the entire spectral range to be recorded without the need to change beamsplitters. Spectra at 150 K were recorded using the same spectrometer with a SpecAc Golden Gate low temperature ATR accessory. The accessory limits the attainable, lowest energy to 350 cm⁻¹. As the low temperature data is significantly sharper, this is shown where possible, unfortunately, only room temperature data is available for the region below 150 cm⁻¹. The spectra have been corrected for the wavelength-dependent variation in path length using the Bruker software.

Dispersion corrected periodic DFT calculations of the crystalline structures were carried out using the plane-wave pseudopotential method implemented in the CASTEP code. 16,17 Exchange and correlation were approximated using the Perdew-Burke-Ernzerhof (PBE) functional, with the Tkatchenko-Scheffler (TS) dispersion correction scheme²¹ within the generalized gradient

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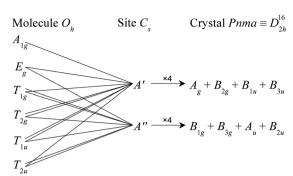


Fig. 1 Correlation diagram for the metal hexacarbonyls from the gas phase to the solid state. All gerade modes are Raman active, $B_{1\mu}$, $B_{2\mu}$, $B_{3\mu}$, are infrared active and A_{u} is inactive in both forms of spectroscopy. Note that all modes are allowed in INS spectroscopy.

approximation (GGA). Norm-conserving pseudopotentials were generated using the kinetic-energy optimised method²² with a plane-wave cut-off energy of 830 (Cr, Mo) or 900 eV (W). Brillouin-zone sampling of electronic states was performed on a $6 \times 6 \times 8$ Monkhorst–Pack grid (36 k-points). The equilibrium structure, an essential prerequisite for lattice dynamics calculations was obtained by Broyden–Fletcher–Goldfarb–Shanno (BFGS)

179.38

179.19

179.45

90.36

90.09

90.03

89.53

89.54

89.79

179,20

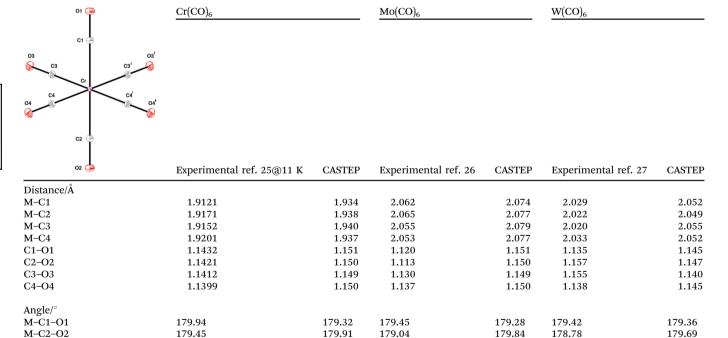
90.88

geometry optimization after which the residual forces were converged to zero within ± 0.0017 eV \mathring{A}^{-1} . Phonon frequencies were obtained by diagonalization of dynamical matrices computed using density-functional perturbation theory²³ (DFPT). An analysis of the resulting eigenvectors was used to map the computed modes to the corresponding irreducible representations of the point group and assign IUPAC symmetry labels. DFPT was also used to compute the dielectric response and the Born effective charges, and from these the mode oscillator strength tensor and infrared absorptivity were calculated. In addition to the direct evaluation of frequencies and intensities at zero wavevector, phonon dispersion was also calculated along high symmetry directions throughout the Brillouin zone. For this purpose, dynamical matrices were computed on a regular grid of wavevectors throughout the Brillouin zone and Fourier interpolation was used to extend the computed grid to the desired fine set of points along the high-symmetry paths.²⁴ The INS spectra were generated from the CASTEP output using ACLIMAX.²⁵

Results and discussion

All three metal hexacarbonyls crystallise in space group Pnma (D_{2h}^{16} number 62) with four molecules in the unit cell, each

Table 1 Comparison of experimental and calculated bond distances and angles in $M(CO)_6$ (M = Cr, Mo, W). The calculated values are from the lattice and geometry optimised calculations with inclusion of the TS dispersion correction



179.53

179.30

179.34

90.14

89.57

90.33

89.97

89.60

89.77

179.30

90.85

179.27

179.43

179.81

90.48

89.92

89.65

89.95

90.67

89.90

179.30

89.53

179.20.

178.71

179.56

89.34

89.44

90.97

90.26

89.75

89.98

178,74

90.27

179.40

179.50

179.21

90.17

89.65

90.39

89.81

89.24

90.00

90.75

179,22

179.51

179.44

179.47

90.11

89.55

90.26

90.08

89.87

89.84

179.55

90.46

M-C3-O3

M-C4-O4

C1-M-C2

C1-M-C3

C1-M-C4

C2-M-C3

C2-M-C4

C3-M-C3i

C3-M-C4

C3-M-C4i

C4-M-C4i

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occupying a site of C_s symmetry, $(Cr(CO)_6,^{26} Mo(CO)_6,^{27} W(CO)_6^{28})$.

In the gas phase, the O_h symmetry results in 13 modes: three $C \equiv O$ stretches (A_{1g}, E_g, T_{1u}) , three M–CO stretches (A_{1g}, E_g, T_{1u}) , four M–C $\equiv O$ bends $(T_{1g}, T_{1u}, T_{2g}, T_{2u})$ and three OC–M–CO deformations (T_{1u}, T_{2g}, T_{2u}) . The C $\equiv O$ stretch modes occur at ~ 2000 –2150 cm⁻¹, the M–CO stretches and M–C $\equiv O$ bends at ~ 300 –700 cm⁻¹, and the OC–M–CO deformations below 150 cm⁻¹. The A_{1g} , E_g and T_{2g} modes are Raman active, the T_{1u} are infrared active and the T_{1g} and T_{2u} are inactive in both forms of spectroscopy. The effect of the C_s site symmetry is to formally remove all the degeneracies and the factor group splitting results in a quadrupling of the modes, as shown in Fig. 1. However, in all three cases, the geometry is that of a near-regular octahedron, Table 1, and for most of the modes both the site group and factor group splitting is small. ¹³

In the solid state, all the modes have components that are infrared or Raman allowed, in reality, the forbidden modes give, at best, very weak transitions and they have usually been obtained from overtone or combination bands. ²⁹ INS spectroscopy has no selection rules, thus all modes are allowed. The resolution of INS spectrometers in the C \equiv O stretch region is insufficient to resolve the modes, however, in the region below 800 cm $^{-1}$, the modes are more widely separated and the spectral resolution is much better. Fortunately, this is also the region where the forbidden modes occur and these are the focus of this paper.

The INS, Raman and infrared spectra of $Cr(CO)_6$, $Mo(CO)_6$ and $W(CO)_6$ in the 0-800 cm⁻¹ region are shown in Fig. 2-4 respectively. In the 300-800 cm⁻¹ region, seven modes are expected: three Raman active, two infrared active and two forbidden (in O_h symmetry) modes. Comparison of the spectra immediately identifies the forbidden modes at 365 and 511 cm⁻¹ (Cr), 343 and 510 cm⁻¹ (Mo) and 362 and 523 cm⁻¹ (W).

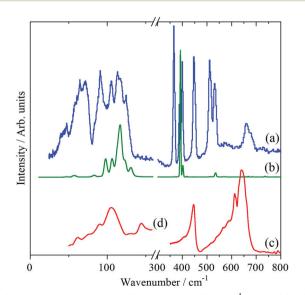


Fig. 2 Vibrational spectra of $Cr(CO)_6$ in the $0-800~cm^{-1}$ region: (a) INS at 10 K, (b) Raman at 7 K and infrared at (c) 150 K (400–800 cm⁻¹) and (d) room temperature, (<200 cm⁻¹). (d) is $\times 5$ ordinate expanded relative to (c).

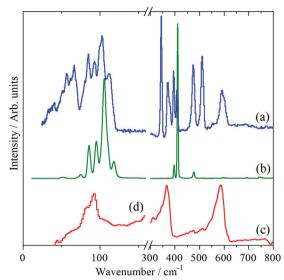


Fig. 3 Vibrational spectra of $Mo(CO)_6$ in the 0–800 cm $^{-1}$ region: (a) INS at 10 K, (b) Raman at 7 K and infrared at (c) 150 K (400–800 cm $^{-1}$) and (d) room temperature, (<200 cm $^{-1}$). (d) is $\times 5$ ordinate expanded relative to (c).

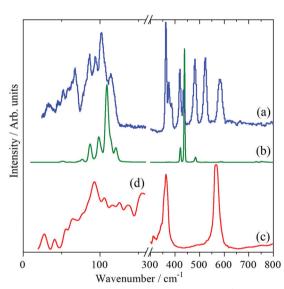


Fig. 4 Vibrational spectra of W(CO)₆ in the 0-800 cm⁻¹ region: (a) INS at 10 K, (b) Raman at 7 K and infrared at (c) 150 K ($400-800 \text{ cm}^{-1}$) and (d) room temperature, ($<200 \text{ cm}^{-1}$). (d) is $\times 5$ ordinate expanded relative to (c).

In the very low energy region, 0–150 cm $^{-1}$, the three OC–M–CO deformations occur, one of which is forbidden under $O_{\rm h}$ symmetry. However, the 12 translational (three acoustic and nine optic) and 12 librational modes also occur (three of each type for each of the four molecules in the primitive cell) in this region. This results in the INS spectra being very congested, since all of these will contribute to the INS spectrum. However, it can be seen that in all the INS spectra there are two main areas of intensity: 0–80 and 80–150 cm $^{-1}$.

In the case of $Cr(CO)_6$, there are three clear peaks at 91, 105 and 115 cm⁻¹. The infrared and Raman spectra show that there are (relatively) strong modes at 90–100 and 110–115 cm⁻¹

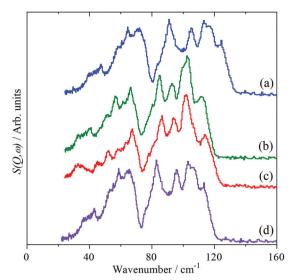


Fig. 5 INS spectra of: (a) Cr(CO)₆, (b) Mo(CO)₆, (c) W(CO)₆ and (d) Cr(CO)₆ scaled by 0.91 in the lattice mode and OC-M-CO bend region.

respectively, which are assigned to the allowed modes, 9,13 accordingly the lowest energy peak at 91 cm⁻¹ is assigned to the forbidden mode. The only previous assignment of this mode was at 68 cm⁻¹ and was based on a very weak combination mode in the infrared spectrum of the gas phase molecule. The $+23~\mathrm{cm}^{-1}$ shift on going from the gas phase to the solid state is consistent with the other two OC-M-CO deformation modes which shift by +8 cm⁻¹ and +24 cm⁻¹. These three modes probably account for most of the intensity in the 80–150 cm⁻¹ range in the INS spectra, suggesting that the lattice modes result in the 0-80 cm⁻¹ massif.

For Mo(CO)₆ and W(CO)₆, the 0-150 cm⁻¹ regions are almost superimposable and strongly resemble a "squashed" version of that of Cr(CO)₆, as shown in Fig. 5. Accordingly, the OC-M-CO deformation modes are as for Cr(CO)6, with the forbidden mode observed at 85 and 86 cm⁻¹ for Mo(CO)₆ and W(CO)₆ respectively. Table 2 lists the assignments.

In order to confirm these assignments, periodic density functional theory (DFT) calculations of the complete unit cell

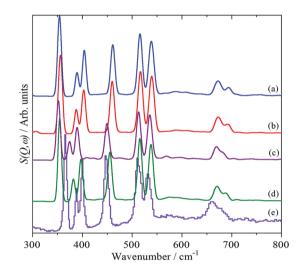
Table 2 Assignments for the $0-800 \text{ cm}^{-1}$ region of M(CO)₆ (M = Cr, Mo, W) in the gas phase^a and the solid state with O_h symmetry

	Cr/cm ⁻¹		Mo/cm ⁻¹		W/cm ⁻¹		
Mode	Solid (INS)	Gas	Solid (INS)	Gas	Solid (INS)	Gas	$Description^b$
T_{2u}	91	67.9	85	60	86	61.4	δ(C-M-C)
T_{1u}	105	97.2	93	81.6	94	82.0	δ (C-M-C)
T_{2g}	117	89.7	103	79.2	102	81.4	δ (C-M-C)
$T_{1g}^{\ \ \ \ \ \ \ \ \ \ \ }$	367	364.1	344	341.6	362	361.6	$\delta(M-C \equiv O)$
A_{1g}	391	379.2	410	391.2	434	426	ν (M–C)
E_g	400	390.6	393	381	419	410	ν (M–C)
T_{1u}	446	440.5	370	367.2	376	374.4	δ (M-C \equiv O)
T_{2u}	511	510.9	512	507.2	523	521.3	$\delta(\mathbf{M}-\mathbf{C} \equiv \mathbf{O})$
T_{2g}	533	532.1	475	477.4	481	482.0	$\delta(\mathbf{M}-\mathbf{C} \equiv \mathbf{O})$
T_{1u}	660	668.1	590	595.6	583	586.6	ν(M -C)

^a Ref. 9. ^b δ = bend, ν = stretch.

are required. Comparison of observed and calculated INS spectra is an exacting test¹⁵ of the accuracy of such studies.

The results of a series of calculations for Cr(CO)₆ are shown in Fig. 6 and summarised in Table 3. All of these are generated from calculations across the complete Brillouin zone, so as to include the effects of vibrational dispersion (variation of transition energy with wavevector). As may be seen from Fig. 7, this is significant in the region below 200 cm⁻¹, but less so for the stretch and deformation region. (The dispersion curves for the C=O stretch region and the corresponding sets for Mo(CO)₆ and W(CO)₆ are shown in Fig. S1–S7 of the ESI†). From Fig. 6, it can be seen that the agreement between observed and calculated spectra, is very sensitive to the details of the method used for the lattice mode region, but less so for the stretch and deformation region. The calculations with the experimental



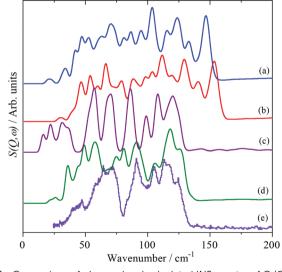


Fig. 6 Comparison of observed and calculated INS spectra of Cr(CO)₆ in the Cr−C stretch and Cr−C≡O bend region (upper panel) and the lattice mode and OC-Cr-CO bend region (lower panel). (a) Calculated at the experimental lattice parameters. (b) calculated at the experimental lattice parameters with the TS correction, (c) calculated with optimised lattice parameters, (d) calculated with optimised lattice parameters and with the TS correction and (e) experimental data

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Table 3 Comparison of relative total energies and lattice parameters in M(CO)₆

		Relative ^a total	Lattice parameters/Å			Cell volume	
Method		energy/eV	a	b	с	$\mathring{\mathbf{A}}^3$	$\Delta^b\%$
Fixed at experimental lattice parameters (no TS correction)		4.3259	11.474	10.894	6.188	773.549	0
,	Mo	6.514	12.019	11.415	6.488	890.133	0
	W	6.579	11.944	11.370	6.459	877.153	0
Lattice optimised c (no TS correction)		3.4015	12.770	11.833	6.868	1037.894	+34.2
Fixed at experimental lattice parameters with the TS correction included		0.0951	11.474	10.894	6.188	773.549	0
		0	12.019	11.415	6.488	890.133	0
	W	0	11.944	11.370	6.459	877.153	0
Lattice optimised with the TS correction included		0	11.741	11.117	6.308	823.284	+6.4
•	Mo	0	11.970	11.374	6.438	876.610	-1.5
	W	0	11.875	11.360	6.417	865.540	-1.3

^a Relative to lowest energy structure. ^b % difference in cell volume. ^c Cell angles fixed at 90° in order to maintain the *Pnma* symmetry, but cell lengths are allowed to optimise.

lattice parameters result in poor agreement, allowing the lattice lengths to optimise (but not the angles, so as to maintain the orthorhombic symmetry), results in much better agreement.

As these compounds have essentially zero permanent dipole moment, it follows that the solids are largely stabilised by van der Waals interactions. These are not included in conventional

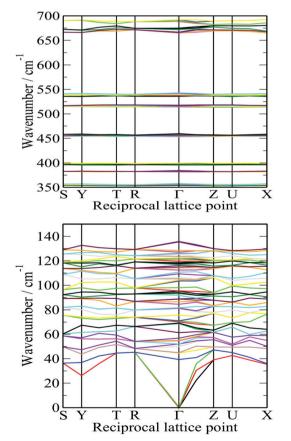


Fig. 7 Calculated dispersion curves of $Cr(CO)_6$ in the Cr-C stretch and $Cr-C \equiv O$ bend region (upper panel) and the lattice mode and OC-Cr-CO bend region (lower panel). The calculations are for the lattice optimised structure with the TS correction and correspond to Fig. 5d.

DFT functionals, so to compensate for this an empirical correction is added to the functional. There are a variety of such methods, the most comprehensive currently available is that of Tkatchenko and Scheffler²¹ (TS). Inclusion of the TS dispersion correction does not greatly change the fixed lattice results, but it does significantly modify the optimised lattice calculations in several respects: the change in cell volume is reduced from +34% to +6.4%, the acoustic modes (the bands below 50 cm⁻¹ in Fig. 6c) harden and the profile of the 80–150 cm⁻¹ feature is in excellent agreement with that found experimentally, Fig. 6e. Table 1 compares selected observed and calculated bond distances and angles for the lattice and geometry optimised structures with inclusion of the TS dispersion correction.

A calculation of an $O_{\rm h}$ symmetry $Cr(CO)_{\rm 6}$ molecule in $10 \times 10 \times 10$ Å cell, (which simulates a gas phase molecule) found that only the lowest three modes underwent a downshift of > 5 cm⁻¹ (solid–gas phase), consistent with our assignment.

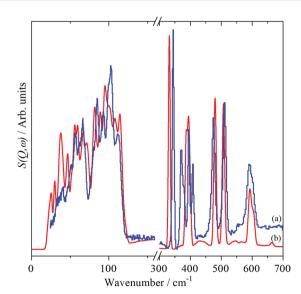


Fig. 8 Comparison of (a) observed and (b) calculated INS spectra of $\mathrm{Mo}(\mathrm{CO})_6.$

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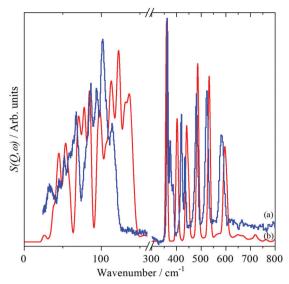


Fig. 9 Comparison of (a) observed and (b) calculated INS spectra of W(CO)₆.

Fig. 8 and 9 show the comparison of the observed and calculated INS spectra for Mo(CO)₆ and W(CO)₆ respectively. Both calculated spectra are from lattice and geometry optimisations with the TS correction. It can be seen that the result for Mo(CO)₆ is in good agreement with the observed spectrum, while that for W(CO)6 is in poor agreement for the region <150 cm⁻¹. We have no explanation for this event. It is noticeable that the lattice optimisation resulted in a small shrinkage of the lattice, Table 2, rather than an expansion as seen for Cr(CO)6. The contraction would harden the modes, but it does not explain why there is a difference between the Mo and W case.

The calculations confirm the assignments derived above. Tables S1-S3 (ESI†) list the calculated modes with their assignments for all three molecules.

Conclusions

The solid state vibrational spectra of $M(CO)_6$, (M = Cr, Mo, W)in the region below 800 cm⁻¹ have been assigned by a combination of infrared, Raman and the first reported inelastic neutron scattering (INS) spectra from homoleptic metal carbonyls. This has enabled the three modes that are forbidden in both the infrared and Raman spectra of the parent O_h symmetry gas phase molecule to be directly observed for the first time. Periodic density functional theory calculations of the complete orthorhombic structure support the assignments, although the agreement is not exact in the region below 100 cm⁻¹, which highlights the difficulty of obtaining reliable assignments for the external modes (translations and librations) even for fairly simple molecules.

Conflicts of interest

There are no conflicts of interest to declare.

Acknowledgements

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