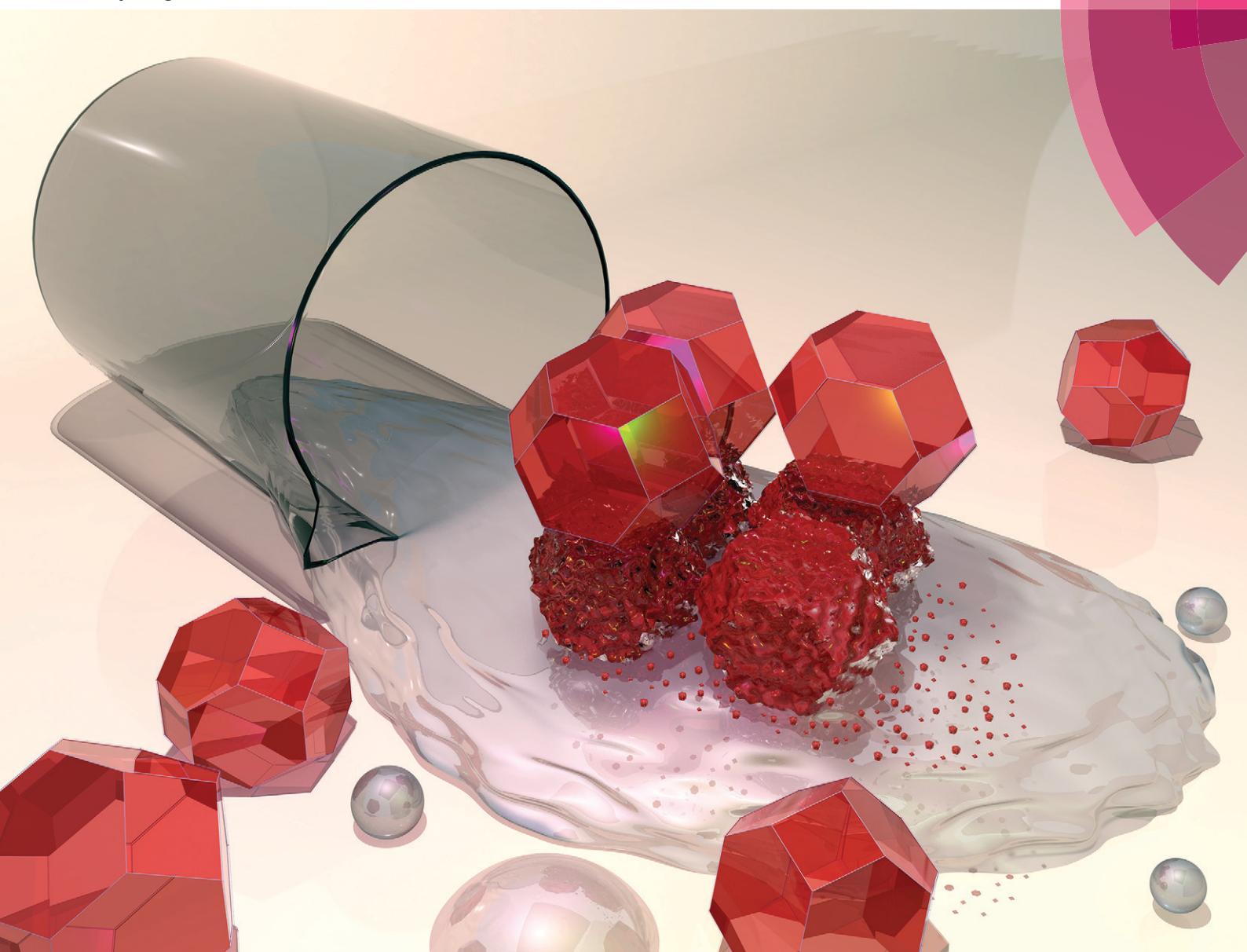


# CrystEngComm

[rsc.li/crystengcomm](http://rsc.li/crystengcomm)



ISSN 1466-8033



ROYAL SOCIETY  
OF CHEMISTRY

Celebrating  
IYPT 2019

PAPER

Christian J. Doonan, Paolo Falcaro *et al.*  
Degradation of ZIF-8 in phosphate buffered saline media



Cite this: *CrystEngComm*, 2019, **21**, 4538

## Degradation of ZIF-8 in phosphate buffered saline media†

Miriam de J. Velásquez-Hernández, <sup>a</sup> Raffaele Ricco, <sup>a</sup> Francesco Carraro, <sup>a</sup> F. Ted Limpoco, <sup>f</sup> Mercedes Linares-Moreau, <sup>a</sup> Erich Leitner, <sup>c</sup> Helmar Wiltsche, <sup>c</sup> Johannes Rattenberger, <sup>d</sup> Hartmuth Schrötter, <sup>d</sup> Philipp Frühwirt, <sup>a</sup> Eduard M. Stadler, <sup>a</sup> Georg Gescheidt, <sup>a</sup> Heinz Amenitsch, <sup>e</sup> Christian J. Doonan <sup>\*b</sup> and Paolo Falcaro <sup>\*ab</sup>

Received 17th May 2019,  
Accepted 7th June 2019

DOI: 10.1039/c9ce00757a

[rsc.li/crystengcomm](http://rsc.li/crystengcomm)

Understanding the stability of zeolitic imidazolate framework-8 (ZIF-8) under physiological conditions is critical in biotechnology and biomedicine for biosensing, biocatalysis, and drug delivery. The use of ZIF-8 has shown that this metal organic framework (MOF) and its derived bio-composites can degrade in presence of buffer solutions. Here we present an in-depth analysis of the structural and chemical changes of pure ZIF-8 particles exposed to phosphate buffered saline (PBS) media. Two different particle sizes (2  $\mu$ m and 250 nm) were selected and the decomposition operated by 10 mM PBS (aka 1X) was studied using powder X-ray diffraction (PXRD), Fourier transformed infrared spectroscopy (FTIR), time resolved atomic force microscopy (AFM), *in situ* small angle X-ray scattering (SAXS), and  $^{31}\text{P}$  NMR.

## Introduction

Metal-organic frameworks (MOFs) are constructed *via* connecting metal ions or clusters coordinated with organic ligands into an extended network.<sup>1,2</sup> The chemical mutability of the constituent building blocks allows for the properties of MOFs to be tailored for a variety of different applications including gas storage, catalysis, sensing, and optics.<sup>3–5</sup> Recently, MOFs have been explored for their potential use in drug delivery, bio-storage and bio-catalysis.<sup>6–10</sup> For example, MOFs can act as protective hosts to preserve enzymes and active pharmaceutical ingredients (APIs) from biological degradation induced by temperature, solvents and proteolytic agents.<sup>11–15</sup> Zeolitic imidazolate framework-8 (ZIF-8), composed of tetrahedral  $\text{Zn}^{2+}$  ions linked by 2-methylimidazolate (mIM) ligands, is one of the most studied MOF carriers.<sup>16–20</sup> The widespread interest in ZIF-8 can be attributed the following

reasons: 1) encapsulation of APIs and biomolecules within ZIF-8 can be carried out in aqueous media,<sup>16,20</sup> 2) the loading and release efficiency can reach 100%,<sup>20</sup> 3) the protection of biomolecules in ZIF-8 can be extended to conditions that would typically result in their degradation (*e.g.* high temperature),<sup>21</sup> and 4) the release of APIs and biomolecules can be controlled either by lowering the pH below 6.5 or adding chelating agents (*e.g.* ethylenediaminetetraacetic acid, EDTA).<sup>20</sup>

These properties encouraged the investigation of ZIF-8 as a drug delivery system from small to large therapeutics (*e.g.* doxorubicin and insulin) for treatment of diseases ranging from cancer to diabetes.<sup>22,23</sup> The use of ZIF-8 was extended to bio-catalytic application with ZIF particles encapsulating and protecting enzymes such as horseradish peroxidase (HRP), catalase, lipase from inhospitable environments.<sup>9</sup> Then, protective properties of ZIF-8 were tested for the bio-preservation of viruses and yeast cells showing that the ZIF-8 coatings can be controllably assembled and dissolved.<sup>24,25</sup> In all these examples, the use of ZIF-8 with APIs, enzymes, viruses and cells, involved MOF crystals ranging from hundreds of nanometers to micrometers.<sup>9,16,24–29</sup> For example, enzyme@MOF particles produced *via* biomimetic mineralization are in the micrometer range (*e.g.* 2  $\mu$ m).<sup>9,16,26,27</sup>

However, for drug delivery applications, particles around 250 nm are preferable as they present long blood-circulation time, and high cellular uptake, which help with the drug delivery efficiency in intravenous drug administration.<sup>10</sup> Another important property that needs to be considered for the drug release performance of MOFs is how the material

<sup>a</sup> Institute of Physical and Theoretical Chemistry, Graz University of Technology, 8010 Graz, Austria. E-mail: paolo.falcaro@tugraz.at

<sup>b</sup> Department of Chemistry, The University of Adelaide, 5005 Adelaide, South Australia, Australia. E-mail: christian.doonan@adelaide.edu.au

<sup>c</sup> Institute of Analytical Chemistry and Food Chemistry, Graz University of Technology, 8010, Graz, Austria

<sup>d</sup> Graz Centre for Electron Microscopy (ZFE), 8010 Graz, Austria

<sup>e</sup> Institute of Inorganic Chemistry, Graz University of Technology, 8010 Graz, Austria

<sup>f</sup> Oxford Instruments GmbH Asylum Research, 65205 Wiesbaden, Germany

† Electronic supplementary information (ESI) available: Experimental details, additional figures and *in situ* real-time AFM movie. See DOI: 10.1039/c9ce00757a



degrades under physiological conditions.<sup>30–33</sup> Ruyra *et al.* demonstrated that several nanoMOFs (e.g. UiO-67, ZIF-7, HKUST-1) become amorphous or decompose releasing cations when exposed to the cell culture medium (Dulbecco's modified Eagle's medium).<sup>31</sup> Li *et al.* showed that the particle size of MIL-100(Fe), and MIL-101(Fe) determined the stability and biodegradability of the MOF when exposed to phosphate buffered saline (PBS).<sup>32</sup> The mechanism of the MOF degradation was attributed to the affinity of phosphate groups for polyvalent cations. This hypothesis was supported by the detection of insoluble metal phosphates.<sup>32</sup> Additionally, Liang *et al.* proposed the degradation of ZIF-8 in PBS to explain the enzymatic activity of solutions containing catalase@ZIF-8 micrometric particles.<sup>9</sup> In another study, Luzuriaga *et al.* showed that ZIF-8 can degrade in several buffer solutions suggesting that this aspect should be taken into account for biotechnological applications.<sup>33</sup> However, the mechanism of degradation of ZIF-8 particles in PBS has not been fully elucidated and the related kinetics of decomposition remain unknown.

To shed light on the kinetics of biodegradation of ZIF-8 we present a comparative study on the stability of pure micro- and nano-ZIF-8 particles (2  $\mu$ m and 250 nm) using a PBS (10 mM, aka 1X) medium under physiological pH (7.4) and temperature (37 °C). We focus on PBS as it is one of the most frequently used media to maintain physiological pH and osmolarity for biomedical dilutions in cell culture and protein chemistry.<sup>34–37</sup> We first examined the chemical and structural behavior of pure micro-ZIF-8 with an average size of *ca.* 2  $\mu$ m, as this is the typical size of ZIF-8-based biocatalysts.<sup>9,16,26–28</sup> The stability of these particles was evaluated in terms of crystallinity, morphology, and chemical composition, using powder X-ray diffraction (PXRD), infrared spectroscopy (IR), and scanning electron microscopy with energy dispersive X-ray spectroscopy (SEM/EDS). Further, the decomposition of micro-ZIF-8 particles was monitored by *in situ* atomic force microscopy (AFM) experiments, <sup>31</sup>P NMR, and quantitatively determined by monitoring the concentration of 2-methylimidazole released over time, using gas chromatography-mass spectrometry (GC-MS) as analytical technique. We then focused on examining how the nano-ZIF-8 particles (average size of 250 nm) behave when exposed to PBS. Our results indicate that the decomposition of ZIF-8 particles is size-dependent (degradation was found to be faster for these smaller particles) and thus suggests that the particle dimensions must be carefully considered when designing ZIF-8-based composites for applications where controlled release is required.

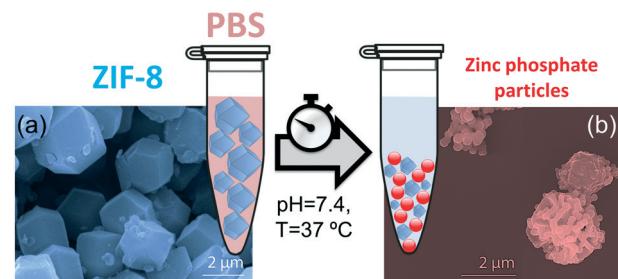
## Results and discussion

### Synthesis of micro- and nano-ZIF-8 particles

Micro-ZIF-8 particles, *ca.* 2  $\mu$ m, were prepared by mixing aqueous solutions of Zn(OAc)<sub>2</sub>·2H<sub>2</sub>O (15 mL, 120 mM) and 2-methylimidazole (HmIM) (30 mL, 960 mM) (see ESI† for details). Based on our previous study on how precursor stoichiometry influences the ZIF structure,<sup>38</sup> we employed a HmIM: Zn<sup>2+</sup> molar ratio of 16:1 to yield pure sod-ZIF-8. Furthermore, the crystal size was controlled by modifying the concentration of the MOF precursors in the reaction mixture. Thus, nano-ZIF-8 particles, *ca.* 250 nm, were achieved by increasing the concentration of Zn(OAc)<sub>2</sub>·2H<sub>2</sub>O (10 mL, 240 mM) and HmIM (20 mL, 1920 mM). Close analysis of the SEM reveals that both micro- and nano-ZIF-8 crystals present a rhombic dodecahedron morphology typical of sod-ZIF-8 (Fig. S3a and b, ESI†). The phase purity of both samples was confirmed by powder X-ray diffraction (PXRD) analysis (Fig. S3c and d, ESI†).

**Analysis of the stability of micro-ZIF-8 and nano-ZIF-8 particles**

PBS (1X) closely mimics the pH, osmolarity, and ion concentrations of the human body. Therefore, this buffer is widely used as a medium in biological research.<sup>34–37</sup> In this realm, PBS has been widely employed in therapeutic *in vitro* trials to evaluate the suitability of MOFs as drug carriers,<sup>22,39–41</sup> and has also been used for standard enzymatic tests applied to active MOF biocomposites.<sup>42–49</sup> Accordingly, a thorough understanding of how PBS buffer affects the stability of ZIF-8, including the kinetics of decomposition and resulting by-products, is important not only for the design of drug carriers with controlled release of their corresponding cargo, but also for the interpretation of bio-catalytic activity data for enzyme@MOF composites. To this end, micro-ZIF-8 (0.5 mg mL<sup>-1</sup>) particles were incubated in PBS (1X) buffer media, under physiological conditions of pH (7.4) and temperature (37 °C), for 1 h, 3 h, 6 h, and 24 h (Fig. 1). The concentration of the micro-ZIF-8 particles used in this study was based on: (1) the cytotoxicity threshold concentration of ZIF-8 (30  $\mu$ g mL<sup>-1</sup>),<sup>50</sup> (2) the minimum amount required to enable the recollection of the remaining ZIF-8 particles upon 24 h of treatment with PBS (see ESI† for details). The effect of the PBS was monitored by recovering the solid, *via* filtration, at different incubation time points (see ESI† for details). The solid collected after the incubation of micro-ZIF-8 particles for 24 hours in PBS was analyzed by SEM. The data revealed that



**Fig. 1** Incubation process of ZIF-8 particles in phosphate buffered saline media (PBS) under physiological conditions of pH (7.4) and temperature (37 °C). (a) SEM image of as-synthesized micro-ZIF-8 particles. (b) SEM image of micro-ZIF-8 particles after being soaked in PBS for 24 h.



the initial rhombic dodecahedron morphology was transformed into clusters of spherical shaped nanoparticles (Fig. 1a and b).

Then, PXRD technique was employed to evaluate the relative crystallinity of the micro-ZIF-8 particles upon their treatment in PBS. The measurements were performed adding a known quantity (20% wt) of commercial  $\text{TiO}_2$  (anatase) as internal standard (Fig. 2a). This method allows us to exclude the possibility that the decrease in intensity of the diffraction peaks resulted from fluctuations in the mass of the crystalline material from sample to sample. The diffraction patterns were baseline corrected. Then, the peak corresponding to the (101) anatase ( $2\theta = 25.3^\circ$ )<sup>51</sup> was fitted with a Gaussian curve, and the intensity normalized (see ESI† for details).

We found that the normalized intensity of the peak arising from the (011) plane of sod-micro-ZIF-8 crystal decreased with incubation time in PBS (Fig. 2). Specifically, after 1 h the intensity drops to 26% of its original value, and only 11% after 24 h (Table S2, ESI†). This loss in peak intensity suggests that, under these conditions, the long range order is lost as MOF crystals are decomposing.<sup>52</sup> The PXRD patterns collected after the incubation process do not present new diffraction peaks suggesting that by-products are amorphous.

To monitor the morphological changes of micro-ZIF-8 particles during immersion in PBS, a polycrystalline ZIF-8 coating was fluxed with 50  $\mu\text{L min}^{-1}$  of buffer solution within a fluidic cell. An *in situ* real-time AFM study was performed to examine the changes in morphology (ESI† Movie.mp4). Selected 3D AFM images are shown in Fig. 2b where a single micro-ZIF-8 particle is tracked over time. A change in morphology is clearly evident within the first 3 minutes of incubation; the initial sharp edges of the rhombic dodecahedron crystal became progressively rounded. Within 9 minutes, a 20% reduc-

tion in the original height of the ZIF-8 crystal can be observed, and in less than 15 minutes the crystal is almost completely dissolved (Fig. 2c). This qualitative measurement demonstrates the rapid decomposition of micro-ZIF-8 particles once exposed to 10 mM PBS media.

Then, we performed FTIR spectroscopy to assess changes in the atomic connectivity upon exposure to PBS. Fig. 2d shows that significant changes in the vibrational bands occur when the as-synthesized micro-ZIF-8 particles are immersed in PBS. For example, a progressive decrease in the peak intensity of vibration modes related to HmIM ( $\nu_{\text{C}=\text{N}}$ , 1584  $\text{cm}^{-1}$ ;  $\nu_{\text{ring}}$ , 1500–1350  $\text{cm}^{-1}$ ; and  $\delta_{\text{out-of-plane}}$  bending, 800–650  $\text{cm}^{-1}$ ) is evident. Furthermore, the spectra show a progressive reduction in the intensity of the band attributed to the Zn–N stretching mode (421  $\text{cm}^{-1}$ ). These data suggest that the degradation process of ZIF-8 involves the release of the ligand with a change in the coordination environment of the  $\text{Zn}^{2+}$  ions. In addition, new vibrational modes were detected at 1160–900  $\text{cm}^{-1}$  and 660–530  $\text{cm}^{-1}$  with increasing intensity over time. Given the composition of the solution, and previously reported infrared studies of amorphous zinc phosphate nanospheres,<sup>53</sup> the broad bands observed at 1160–900  $\text{cm}^{-1}$  could be ascribed to the antisymmetric stretching mode of  $\text{PO}_4^{3-}$  with the shoulder peak at 956  $\text{cm}^{-1}$  originating from the P–O bending mode, and the broad band at 660–530  $\text{cm}^{-1}$  is engendered by bending of  $\text{PO}_4^{3-}$  groups. Considering the high affinity of phosphate groups for polyvalent cations (e.g.  $\text{Zn}^{2+}$ ),<sup>54</sup> it may be proposed that the new bands in the FTIR spectra are the result of zinc phosphates formed as degradation by-products of the micro-ZIF-8 crystals.

To ascertain the presence of zinc phosphates, we performed EDX analysis (Fig. 3a and b). The elemental maps of the samples before and after exposure to PBS show a

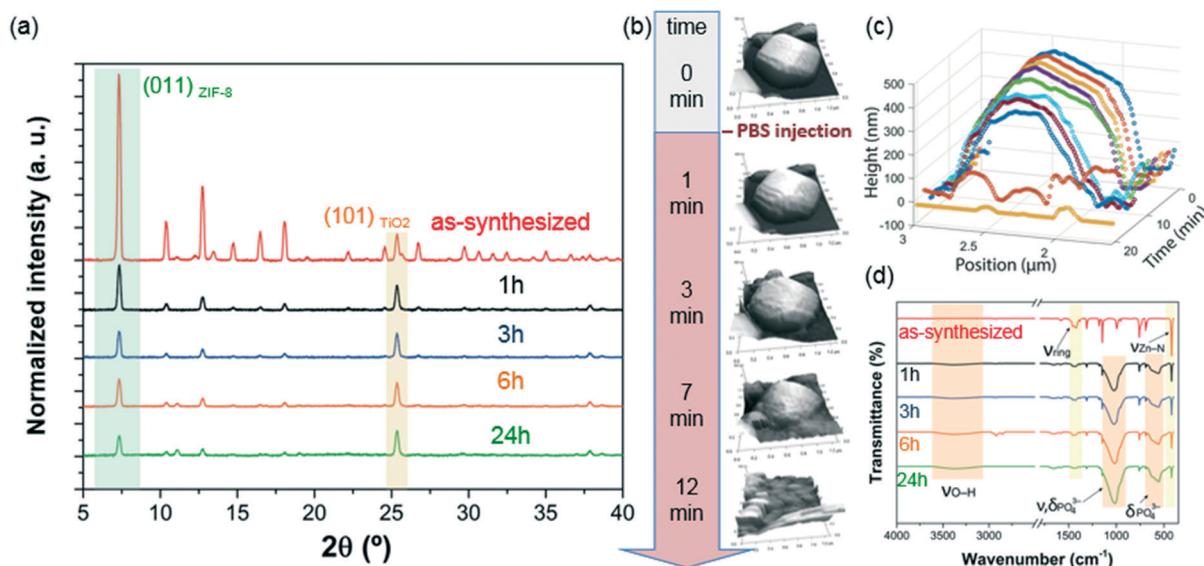
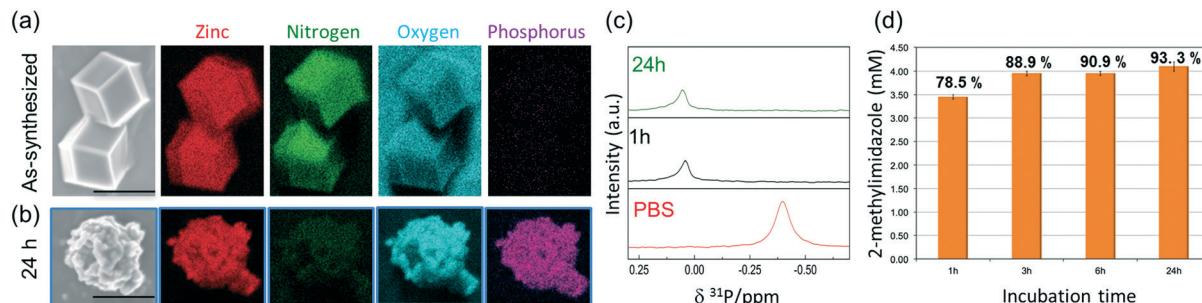


Fig. 2 (a) PXRD patterns illustrating the structural evolution of micro-ZIF-8 particles before and after the incubation process in PBS. (b) Evaluation of the degradation process followed by *in situ* AFM experiments. (c) Height profiles extracted from (b), showing the morphological evolution of a single micrometric ZIF-8 particle upon the incubation in PBS (the first 6 min the measurements were performed in deionized (DI) water as incubation media). (d) Comparative FTIR spectroscopic analysis of the micro-ZIF-8 particles before and after the treatment with PBS.





**Fig. 3** (a) EDX elemental maps of the as-synthesized ZIF-8 microcrystalline powder. (b) EDX elemental maps of the powder recovered after the incubation process in PBS for 24 h. (c)  $^{31}\text{P}$  NMR of PBS prepared in  $\text{D}_2\text{O}$  before (lowest trace) and after adding micro-ZIF-8 particles ( $0.5 \text{ mg mL}^{-1}$ , 1 and 24 h, middle and upper trace, resp.). (d) Quantitative determination of 2-methylimidazole released after the incubation process in PBS (1 h, 3 h, 6 h and 24 h).

significant change in the elemental composition and distribution. The as-synthesized micro-ZIF-8 particles present a uniform distribution of Zn and N within the micro-sized particles and no evidence of P (Fig. 3a). However, after the incubation in PBS, we found that while the N signal was negligible, Zn was present as a constituent of solid aggregates. Furthermore, these aggregates showed a local concentration of O co-localized with P (Fig. 3b).

To gain an insight into the degradation process we monitored the  $^{31}\text{P}$  NMR resonance of a PBS solution (prepared with  $\text{D}_2\text{O}$ ) after adding ZIF-8 (Fig. 3c). The intensity of the signal associated to phosphate in PBS decreases upon the addition of ZIF-8, accompanied by a small downfield shift of 0.44 ppm. Importantly, no other  $^{31}\text{P}$  peaks appeared in spectra recorded after 1 and 24 h (Fig. 3c). This observation can be explained by the formation of insoluble zinc phosphates, which escape from NMR detection (see Fig. S1†).<sup>54</sup> Insoluble zinc phosphates, that are formed during the degradation, can be removed by filtration, as confirmed by the low concentration of  $\text{Zn}^{2+}$  in the supernatant determined by ICP-OES (see Table S1†).

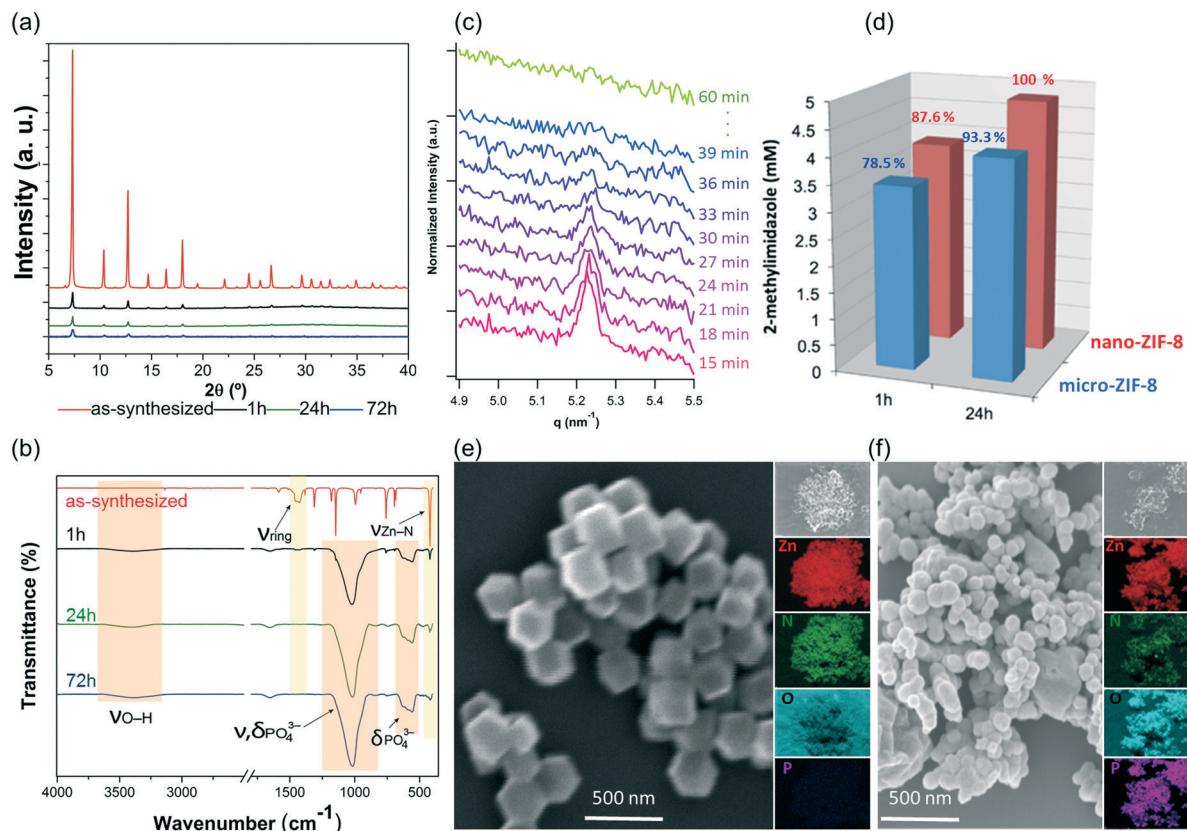
These findings clarify the mechanism of the degradation process, in which the coordination equilibrium, between  $\text{Zn}^{2+}$  ions and HmIM in solution, is altered by the presence of phosphate species. Such phosphates have a high affinity for Lewis metal centers shifting the equilibrium towards the formation of insoluble inorganic by-products. Our data suggest that the competition of phosphate species for the metallic centers progressively releases HmIM in solution. This process might be also favored by the pH conditions imposed by the buffer media ( $\text{pH} = 7.4$ ). Indeed, under these conditions, the ligand exists in solution as a protonated species ( $\text{p}K_{\text{a}1} = 7.85$ ;  $\text{p}K_{\text{a}2} = 15.1$ ),<sup>55</sup> which reduces the complexing power of the linker toward the cation.

A quantitative determination of the degradation process was achieved by measuring the concentration of HmIM released in the mother liquors by gas chromatography-mass spectrometry (GC-MS) (Fig. 3d) after 1, 3, 6 and 24 h exposure of ZIF-8 to PBS. The collected data show that the most significant changes occur within the first hour of treatment with PBS, as around 79% of the HmIM is released into the incubation

media. The release of HmIM after 3 h and 6 h, respectively, of treatment in PBS is comparatively less significant. After 24 h incubation, *ca.* 93% of HmIM is present in the mother liquors thus supporting the interpretation of the PXRD and  $^{31}\text{P}$  NMR data. Given the relevance of nanoMOFs for biomedical applications,<sup>10</sup> we extended the analysis to evaluate the stability of the nano-ZIF-8 particles with average size of 250 nm, incubated in PBS, following a procedure analogous to that described for micro-ZIF-8 samples (see ESI† for details). The degradation process was monitored up to 72 h, which is a suitable period to assess the feasibility of these systems for long-term drug delivery applications.<sup>10</sup>

The PXRD patterns and FTIR data indicate that the nano-ZIF-8 particles experienced a faster degradation than micro-ZIF-8 particles. This is evidenced by the rapid decrease in the intensity of both the Bragg peaks of sod-ZIF-8, and the bands associated to Zn-N stretch mode ( $421 \text{ cm}^{-1}$ ) (Fig. 4a and b). Furthermore, FTIR confirms the formation of zinc phosphate species for nano-ZIF-8 exposed to PBS. We performed time-resolved SAXS experiments to follow the rapid degradation of the nano-sized particles, by monitoring a nano-ZIF-8 dispersion in PBS (10 mM) and in DI water as a control. The evolution of the diffraction peak attributed to the plane (011) sod-ZIF-8 was recorded with a 3 min time resolution (Fig. 4c). These data, together with the simultaneous decrease of the scattering intensity at low angles, confirm the instability of nano-ZIF-8 particles in the PBS solution: the crystallinity is completely lost within 40 min (see ESI† for details). Conversely, the intensity of the (011) diffraction peak of sod-ZIF-8 particles dispersed in DI water is almost unchanged after a 5 h exposure (Fig. S2†). These observations were quantitatively corroborated by GC-MS to determine the concentration of HmIM in the mother liquors (Fig. 4d). Importantly, the comparative results between the micro- and nano-ZIF-8 particles show that the concentration of ligand in solution for the nano-ZIF-8 is 10% larger for 1 h and 24 h incubation. This is supported by the AFM images, micro-ZIF-8 particles that show etching of the external surface after only one minute. Thus, the faster degradation of nano-ZIF-8 particles can be explained by their larger surface area exposed to the PBS media. We also investigated the morphology of nano-ZIF-8 before





**Fig. 4** (a) PXRD diffractograms of nano-ZIF-8 particles before and after the incubation process in PBS (1 h, 24 h, and 72 h). (b) FTIR spectroscopy of the micro-ZIF-8 particles before and after the treatment with PBS. (c) Time-resolved SAXS of nano-ZIF-8 particles dispersed in PBS (1X) solution. The reported  $q$  range ( $4.9\text{--}5.5\text{ nm}^{-1}$ ) highlights the time-evolution of the intensity of the (001) diffraction peak of sod-ZIF-8. (d) Quantitative comparison of the degradation process suffered by nano- and micro-ZIF-8 particles by the determination of the concentration of 2-methylimidazole in the mother liquors. (e) SEM micrographs, and EDX elemental mapping (Zn, N, O, and P) images of the as-synthesized nano-ZIF-8 particles. (f) SEM and EDX images of the powder recovered upon the treatment with PBS for 24 h.

and after 24 h of treatment in PBS. SEM micrographs show a transformation from original rhombic dodecahedron morphology to clusters of spherical nanoparticles (Fig. 4e and f). The EDX elemental maps analysis confirmed the presence of Zn, O, and P in such clusters; however, N was not detected. These experiments evidenced that nano-ZIF-8 degrades into zinc phosphate species as in the case of micro-ZIF-8. However, we note that the kinetics of degradation was slower for large particles and faster for smaller ZIF-8 crystals.

## Conclusions

In this study, we demonstrated that ZIF-8 particles can rapidly decompose when immersed in PBS (10 mM). The decomposition of the sod-Zn(mIM)<sub>2</sub> crystalline network leads to the formation of zinc phosphate insoluble particles. The kinetic of decomposition depends on ZIF-8 particles size and is faster for smaller particles. The current elucidation of the decomposition mechanism and by-products can be relevant for biochemical processes. In case of bio-composites for biocatalysis (e.g. enzyme@MOF), the potential release of enzymes in buffer solution should be taken into consideration, especially when the precise quantification of the enzymatic

activity is needed (e.g. recyclability of the biocatalyst). In case of drug release, phosphates are essential components in intravascular, interstitial and intracellular fluids.<sup>34,56,57</sup> Therefore, it is possible to take advantage of the biodegradability of ZIF-8, induced by the presence of phosphates, to stimulate the release of APIs incorporated within the porous network.

## Conflicts of interest

The authors confirm that there are no conflicts to declare.

## Acknowledgements

The research leading to these results has received funding from the European Research Council under the European Union's Horizon 2020 Programme (FP/2014-2020)/ERC Grant Agreement no. 771834 – POPCRYSTAL. The authors acknowledge support from the European Union's Horizon 2020 FETOPEN-1-2016-2017 research and innovation program under grant agreement 801464. M. J. V. H. acknowledges The National Council of Science and Technology (CONACyT, México) for the postdoctoral scholarship (CVU 419210). R. R. acknowledges the European Union's Horizon 2020 research



and innovation programme under the Marie Skłodowska-Curie grant agreement #748649 (project “MNEMONIC”). P. F. acknowledges TU Graz for the Lead Project (LP-03). Dr. Florian Johann (Oxford Instruments GmbH) is acknowledged for his technical assistance with the AFM measurements. The authors acknowledge the CERIC-ERIC Consortium for the access to experimental facilities and financial support. P. F., E. M. S., G. G. acknowledge NAWI Graz for support.

## References

- O. M. Yaghi, M. O’Keeffe, N. W. Ockwig, H. K. Chae, M. Eddaoudi and J. Kim, *Nature*, 2003, **423**, 705–714.
- C. S. Diercks, M. J. Kalmutzki, N. J. Diercks and O. M. Yaghi, *ACS Cent. Sci.*, 2018, **4**, 1457–1464.
- H. Furukawa, K. E. Cordova, M. O’Keeffe and O. M. Yaghi, *Science*, 2013, **341**, 1230444.
- C. A. Trickett, A. Helal, B. A. Al-Maythalony, Z. H. Yamani, K. E. Cordova and O. M. Yaghi, *Nat. Rev. Mater.*, 2017, **2**, 17045.
- C. S. Diercks, Y. Liu, K. E. Cordova and O. M. Yaghi, *Nat. Mater.*, 2018, **17**, 301–307.
- A. C. McKinlay, R. E. Morris, P. Horcajada, G. Férey, R. Gref, P. Couvreur and C. Serre, *Angew. Chem., Int. Ed.*, 2010, **49**, 6260–6266.
- W. Chen and C. Wu, *Dalton Trans.*, 2018, **47**, 2114–2133.
- Z. Zhao, J. Pang, W. Liu, T. Lin, F. Ye and S. Zhao, *Microchim. Acta*, 2019, **186**, 295.
- W. Liang, H. Xu, F. Carraro, N. K. Maddigan, Q. Li, S. G. Bell, D. M. Huang, A. Tarzia, M. B. Solomon, H. Amenitsch, L. Vaccari, C. J. Sumby, P. Falcaro and C. J. Doonan, *J. Am. Chem. Soc.*, 2019, **141**, 2348–2355.
- T. Simon-Yarza, A. Mielcarek, P. Couvreur and C. Serre, *Adv. Mater.*, 2018, **30**, 1707365.
- Y. Chen, P. Li, J. A. Modica, R. J. Drouet and O. K. Farha, *J. Am. Chem. Soc.*, 2018, **140**, 5678–5681.
- C. Doonan, R. Riccò, K. Liang, D. Bradshaw and P. Falcaro, *Acc. Chem. Res.*, 2017, **50**, 1423–1432.
- M. Giménez-Marqués, T. Hidalgo, C. Serre and P. Horcajada, *Coord. Chem. Rev.*, 2016, **307**, 342–360.
- P. Horcajada, T. Chalati, C. Serre, B. Gillet, C. Sebrie, T. Baati, J. F. Eubank, D. Heurtaux, P. Clayette, C. Kreuz, J.-S. Chang, Y. K. Hwang, V. Marsaud, P.-N. Bories, L. Cynober, S. Gil, G. Férey, P. Couvreur and R. Gref, *Nat. Mater.*, 2010, **9**, 172–178.
- M. H. Teplensky, M. Fantham, P. Li, T. C. Wang, J. P. Mehta, L. J. Young, P. Z. Moghadam, J. T. Hupp, O. K. Farha, C. F. Kaminski and D. Fairen-Jimenez, *J. Am. Chem. Soc.*, 2017, **139**, 7522–7532.
- K. Liang, R. Ricco, C. M. Doherty, M. J. Styles, S. Bell, N. Kirby, S. Mudie, D. Haylock, A. J. Hill, C. J. Doonan and P. Falcaro, *Nat. Commun.*, 2015, **6**, 7240.
- C. Wang, G. Sudlow, Z. Wang, S. Cao, Q. Jiang, A. Neiner, J. J. Morrissey, E. D. Kharasch, S. Achilefu and S. Singamaneni, *Adv. Healthcare Mater.*, 2018, **7**, 1800950.
- F. Lyu, Y. Zhang, R. N. Zare, J. Ge and Z. Liu, *Nano Lett.*, 2014, **14**, 5761–5765.
- F.-S. Liao, W.-S. Lo, Y.-S. Hsu, C.-C. Wu, S.-C. Wang, F.-K. Shieh, J. V. Morabito, L.-Y. Chou, K. C.-W. Wu and C.-K. Tsung, *J. Am. Chem. Soc.*, 2017, **139**, 6530–6533.
- E. Astria, M. Thonhofer, R. Ricco, W. Liang, A. Chemelli, A. Tarzia, K. Alt, C. E. Hagemeyer, J. Rattenberger, H. Schroettner, T. Wrodnigg, H. Amenitsch, D. M. Huang, C. J. Doonan and P. Falcaro, *Mater. Horiz.*, 2019, **6**, 969–977.
- K. S. Park, Z. Ni, A. P. Cote, J. Y. Choi, R. Huang, F. J. Uribe-Romo, H. K. Chae, M. O’Keeffe and O. M. Yaghi, *Proc. Natl. Acad. Sci. U. S. A.*, 2006, **103**, 10186–10191.
- H. Zheng, Y. Zhang, L. Liu, W. Wan, P. Guo, A. M. Nyström and X. Zou, *J. Am. Chem. Soc.*, 2016, **138**, 962–968.
- W.-H. Chen, G.-F. Luo, M. Vázquez-González, R. Cazelles, Y. S. Sohn, R. Nechushtai, Y. Mandel and I. Willner, *ACS Nano*, 2018, **12**, 7538–7545.
- S. Li, M. Dharmawardana, R. P. Welch, C. E. Benjamin, A. M. Shamir, S. O. Nielsen and J. J. Gassensmith, *ACS Appl. Mater. Interfaces*, 2018, **10**, 18161–18169.
- R. Riccò, W. Liang, S. Li, J. J. Gassensmith, F. Caruso, C. Doonan and P. Falcaro, *ACS Nano*, 2018, **12**, 13–23.
- Q. Wang, X. Zhang, L. Huang, Z. Zhang and S. Dong, *Angew. Chem., Int. Ed.*, 2017, **56**, 16082–16085.
- F. Pitzalis, C. Carucci, M. Naseri, L. Fotouhi, E. Magner and A. Salis, *ChemCatChem*, 2018, **10**, 1578–1585.
- W. Liang, R. Ricco, N. K. Maddigan, R. P. Dickinson, H. Xu, Q. Li, C. J. Sumby, S. G. Bell, P. Falcaro and C. J. Doonan, *Chem. Mater.*, 2018, **30**, 1069–1077.
- M. A. Luzuriaga, R. P. Welch, M. Dharmawardana, C. E. Benjamin, S. Li, A. Shahriarkevishahi, S. Popal, L. H. Tuong, C. T. Creswell and J. J. Gassensmith, *ACS Appl. Mater. Interfaces*, 2019, **11**, 9740–9746.
- E. Bellido, M. Guillevic, T. Hidalgo, M. J. Santander-Ortega, C. Serre and P. Horcajada, *Langmuir*, 2014, **30**, 5911–5920.
- À. Ruyra, A. Yazdi, J. Espín, A. Carné-Sánchez, N. Roher, J. Lorenzo, I. Imaz and D. Maspoch, *Chem. – Eur. J.*, 2015, **21**, 2508–2518.
- X. Li, L. Lachmansi, S. Safi, S. Sene, C. Serre, J. M. Grenèche, J. Zhang and R. Gref, *Sci. Rep.*, 2017, **7**, 13142.
- M. A. Luzuriaga, C. E. Benjamin, M. W. Gaertner, H. Lee, F. C. Herbert, S. Mallick and J. J. Gassensmith, *Supramol. Chem.*, 2019, **0**, 1–6.
- D. G. Fasman, *Handbook of biochemistry and molecular biology. Physical and Chemical Data*, CRC Press, Boca Raton, Florida, 3rd edn, 1975, vol. 1.
- Cell biology: a laboratory handbook*, ed. J. E. Celis, Academic Press, San Diego, 2nd edn, 1998.
- M. N. Gupta, *Methods for affinity-based separations of enzymes and proteins*, 2002.
- D. C. White, *Proteins, Peptides, and Amino Acids SourceBook*, Springer, 2002.
- W. Liang, R. Ricco, N. K. Maddigan, R. P. Dickinson, H. Xu, Q. Li, C. J. Sumby, S. G. Bell, P. Falcaro and C. J. Doonan, *Chem. Mater.*, 2018, **30**, 1069–1077.



39 E. Bellido, T. Hidalgo, M. V. Lozano, M. Guillevic, R. Simón-Vázquez, M. J. Santander-Ortega, Á. González-Fernández, C. Serre, M. J. Alonso and P. Horcajada, *Adv. Healthcare Mater.*, 2015, **4**, 1246–1257.

40 Y. Li, Y. Zheng, X. Lai, Y. Chu and Y. Chen, *RSC Adv.*, 2018, **8**, 23623–23628.

41 F. Shu, D. Lv, X.-L. Song, B. Huang, C. Wang, Y. Yu and S.-C. Zhao, *RSC Adv.*, 2018, **8**, 6581–6589.

42 S. Jung, Y. Kim, S.-J. Kim, T.-H. Kwon, S. Huh and S. Park, *Chem. Commun.*, 2011, **47**, 2904–2906.

43 F.-X. Qin, S.-Y. Jia, F.-F. Wang, S.-H. Wu, J. Song and Y. Liu, *Catal. Sci. Technol.*, 2013, **3**, 2761–2768.

44 S.-L. Cao, H. Xu, X.-H. Li, W.-Y. Lou and M.-H. Zong, *ACS Sustainable Chem. Eng.*, 2015, **3**, 1589–1599.

45 X. Wu, J. Ge, C. Yang, M. Hou and Z. Liu, *Chem. Commun.*, 2015, **51**, 13408–13411.

46 Y. Yin, C. Gao, Q. Xiao, G. Lin, Z. Lin, Z. Cai and H. Yang, *ACS Appl. Mater. Interfaces*, 2016, **8**, 29052–29061.

47 C. Zhang, X. Wang, M. Hou, X. Li, X. Wu and J. Ge, *ACS Appl. Mater. Interfaces*, 2017, **9**, 13831–13836.

48 W.-H. Chen, M. Vázquez-González, A. Zoabi, R. Abu-Reziq and I. Willner, *Nat. Catal.*, 2018, **1**, 689–695.

49 J. Chen, L. Huang, Q. Wang, W. Wu, H. Zhang, Y. Fang and S. Dong, *Nanoscale*, 2019, **11**, 5960–5966.

50 M. Hoop, C. F. Walde, R. Riccò, F. Mushtaq, A. Terzopoulou, X.-Z. Chen, A. J. deMello, C. J. Doonan, P. Falcaro, B. J. Nelson, J. Puigmartí-Luis and S. Pané, *Appl. Mater. Today*, 2018, **11**, 13–21.

51 R. G. Freitas, F. W. S. Lucas, M. A. Santanna, R. A. Mendes, A. J. Terezo, G. L. C. de Souza, L. H. Mascaro and E. C. Pereira, *Phys. Chem. Chem. Phys.*, 2016, **18**, 26885–26893.

52 J. B. DeCoste, M. S. Denny, Jr., G. W. Peterson, J. J. Mahle and S. M. Cohen, *Chem. Sci.*, 2016, **7**, 2711–2716.

53 S.-H. Jung, E. Oh, D. Shim, D.-H. Park, S. Cho, B. R. Lee, Y. U. Jeong, K.-H. Lee and S.-H. Jeong, *Bull. Korean Chem. Soc.*, 2009, **30**, 2280–2282.

54 D. H. Nies, *Metallooms*, 2016, **8**, 481–507.

55 K. Kida, M. Okita, K. Fujita, S. Tanaka and Y. Miyake, *CrystEngComm*, 2013, **15**, 1794.

56 T. Kokubo and H. Takadama, *Biomaterials*, 2006, **27**, 2907–2915.

57 T. Kokubo and H. Takadama, in *Handbook of Biominerization*, ed. E. Buerlein, Wiley-VCH Verlag GmbH, Weinheim, Germany, 2007, pp. 97–109.

