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Advances in crosslinking strategies of biomedical hydrogels

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Biomedical hydrogels as sole repair matrices or combined with pre-seeded cells and bioactive growth factors are extensively applied in tissue engineering and regenerative medicine. Hydrogels normally provide three dimensional structures for cell adhesion and proliferation or the controlled release of the loading of drugs or proteins. Various physiochemical properties of hydrogels endow them with distinct applications. In this review, we present the commonly used crosslinking method for hydrogel synthesis involving physical and chemical crosslinks and summarize their current progress and future perspectives.

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Introduction

Extracellular matrix (ECM) plays a pivotal role in manipulating multiple cellular fates and largely influences the repair and regeneration of injured organs and tissues.^{1,2} Up to now, the structures, components and properties of natural ECMs have been clearly illustrated, and most of natural ECMs exhibit porous network matrices that normally consist of some nano-fiber materials like collagen, fibronectin and laminin.³ Like natural ECMs, hydrogels as artificial ECMs with 3D network structures are fabricated to provide microenvironments⁴ for cell adhesion, proliferation and migration, and promote the exchange of nutrients and signalling molecules.⁵ Hydrogels are hydrophilic polymers with high affinities towards water that are prevented from dissolution, owing to their chemical or physical crosslinking networks.⁶ Mechanical and biochemical properties of hydrogels are highly associated with their crosslinking methods, and even the hydrogels with the same constituents but different crosslinking structures can present various functions.⁷

Hydrogels are mainly fabricated by the crosslinking formation of the stable polymeric networks. Among various crosslinking methods, physical and chemical crosslinks are two basic strategies.^{8–10} Physical crosslinks include ionic/electrostatic interaction, hydrogen bonding, crystallization/stereo-complex

and hydrophobic interactions of thermal induction based on LCST (Lower Critical Solution Temperature)^{11–13}/UCST (Upper Critical Solution Temperature)^{14,15} and ultrasonication^{16–18} mediated sol-to-gel phase transition. Chemical crosslinks comprise photo-polymerization,^{19,20} enzyme-induced crosslink,^{21–24} and “click” chemistry^{25–27} including Michael type-addition,^{28–30} Diels–Alder “click” reaction,^{31–33} oxime formation,^{34–38} and Schiff base formation.^{39–41} In this review, we summarize current physical and chemical crosslinking strategies of biomedical hydrogels (Fig. 1). These crosslinking methods may inspire us to design and fabricate novel hydrogels with superior structures and desirable properties.

Physically crosslinked hydrogels

Physically crosslinked hydrogels are usually created by inter-molecular reversible interactions,⁴² such as ionic/electrostatic interaction,^{43,44} hydrogen bonds, polymerized entanglements, hydrophobic/hydrophilic interactions, crystallization/stereo-complex formation, metal coordination and π - π stacking. The prominent advantage of a physical crosslink is biomedical safety owing to the absence of chemical crosslinking agents, thus, avoiding potential cytotoxicity from unreacted chemical crosslinkers.⁴⁵ More importantly, physically crosslinked hydrogels are stimuli-responsive with self-healing and injectable properties under room temperature. These hydrogels can be designed as bioactive hydrogels for the encapsulation of living cells and drug delivery of therapeutic molecules.⁴⁶

Hydrogel crosslinking by ionic/electrostatic interactions

The ionic/electrostatic interaction that has been extensively applied to the construction of hydrogels is the basis of a routine physical crosslink with 2 molecules of opposite electric

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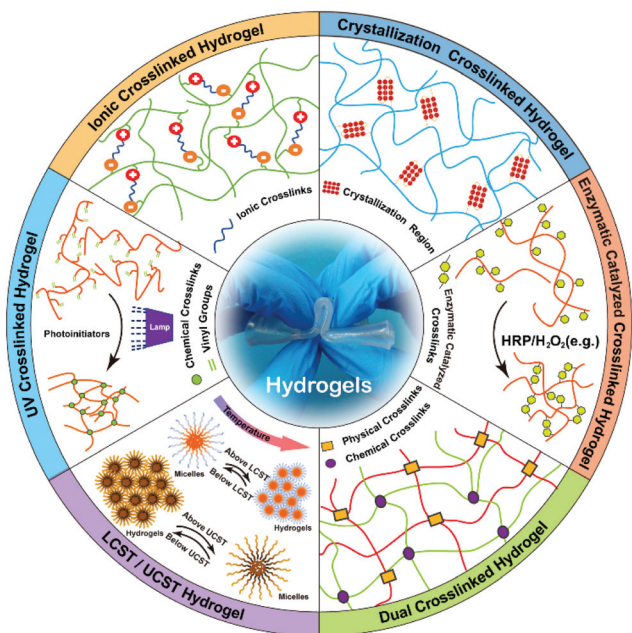


Fig. 1 Crosslinking strategies for hydrogel construction. In this diagram, the most widely used crosslinking methods are presented, including physical crosslinking, such as ionic crosslink, LCST/UCST induced hydrogel formation, crystallization induced hydrogel formation and chemical crosslinking, such as UV-induced crosslink and Enzyme catalyzed crosslink.

charges. For example, alginate, a naturally derived polysaccharide with mannuronic and glucuronic acid residues, can be crosslinked by divalent cations, such as calcium (Ca^{2+}), barium (Ba^{2+}) and magnesium (Mg^{2+}).⁴⁷ Divalent cations can solely bind to guluronate blocks from the alginate chains with a high degree of coordination of the divalent ions. The guluronate blocks of one polymer then form junctions with the guluronate blocks of the adjacent polymer chains, resulting in a gel structure. So far, alginate hydrogels have been explored in wound healing, drug delivery, and tissue engineering.^{48,49}

Electrostatic interactions occur between the opposite charged macromolecules and those macromolecules interact with each other to yield polyelectrolyte complexes.⁵⁰ Chitosan is a natural polycationic biopolymer that consists of β -[1-4]-linked 2-acetamido-2-deoxy-D-glucopyranose and 2-amino-2-deoxy-D-glucopyranose. Thus, chitosan easily forms polyelectrolyte complexes (PECs) *via* electrostatic interactions between its cationic amino groups and anionic groups from various anionic polyelectrolytes in nature, such as pectin, chondroitin sulfate and alginate.^{45,51} Also, chitosan can similarly interact with some synthetic polymers, like polylactic acid, polyacrylic acid and polyphosphoric acid.⁵² The hydrogels generated from those polyelectrolyte complexes can be modulated with a number of factors, including the charge density of the polymers, the mixed ratio and the amount of each polymer, as well as the soluble microenvironment of polymer. If the net charge of the formed complex is zero, it will influence the solubility and the complex will precipitate.⁵³ Ye *et al.*⁵⁴ prepared a

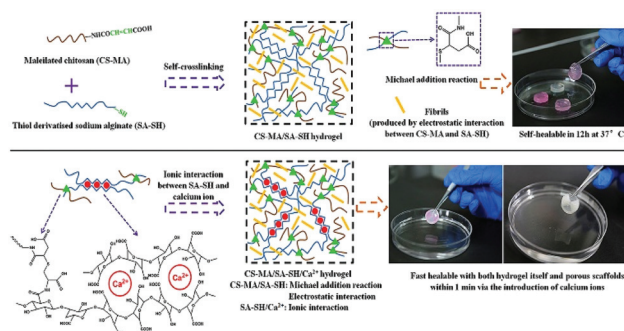


Fig. 2 Schematic illustration of the fibrils-reinforced polysaccharide-based composite hydrogels. The hydrogels were generated by self-crosslinking of CS-MA and SA-SH. The ionic interaction between CS-MA and SA-SH was able to improve mechanical properties of hydrogels. Reproduced with permission from *Compos. Sci. Technol.*, 2018, 156, 238–246. Copyright 2018 Elsevier.⁵⁴

fibrils-reinforced polysaccharide composite hydrogel that was formable *in situ* (Fig. 2). A maleilated chitosan (CS-MA) and a thiol derivatized sodium alginate (SA-SH) were separately synthesized, and the hydrogel was subsequently self-crosslinked *via* Michael addition and ionic interaction between CS-MA and SA-SH. The resultant hydrogels exhibited dual healing ability with good cytocompatibility as well as excellent mechanical properties.

The specific advantage of an ionic/electrostatic interaction is its self-healing ability as a result of which the physical network of hydrogels can be broken at high stress and reform once the stress is removed. However, the mechanical strength of hydrogel is extremely limited owing to the crosslinking strategy of the ionic/electrostatic interaction.

Hydrogel crosslinking by hydrophobic interactions

Hydrophobic interactions for hydrogel crosslinking exist in water-soluble polymers with hydrophobic end groups, side chains or monomers. Two methods create hydrophobic interaction. One is thermal induction based on LCST (Lower Critical Solution Temperature) or UCST (Upper Critical Solution Temperature), the other is the ultrasonic treatment. Generally, hydrogels can be produced using both methods by promoting sol-to-gel transition under certain conditions.

Thermal induction based on LCST/UCST. For a thermally induced phase transition, hydrogels can be fabricated when they are thermally treated at a critical temperature for sol-gel phase transition. The amphiphilic block or graft copolymers are sometimes termed as surfactants, which can self-assemble into an organized structure (*e.g.* micelle) in aqueous solutions with a hydrophobic core at lower concentrations. The hydrophilic parts of the polymers form loops and hydrophobic groups are constrained in the same micellar core. With a thermally induced phase transition, the bridges between micelles are generated *via* hydrophobic interactions and eventually lead to the hydrogel formation. Some polymers are usually soluble below the LCST (lower critical solution temperature). When

the solution temperature is above the LCST, polymers can become hydrophobic and insoluble, and the nearby micelles re-aggregate *via* hydrophobic interactions, resulting in gel formation.⁵⁵ Poly(*N*-isopropylacrylamide) (PNIPAM) and its derivatives can form such a thermo-responsive hydrogel.⁵⁶ For some other examples, the amphiphilic copolymers combining hydrophilic poly(ethylene oxide) with hydrophobic poly(propylene oxide) or poly(glycolide), poly(lactide) and poly(ϵ -caprolactone) segments are the most commonly used polymers. They have been extensively utilized as drug delivery systems, which are injectable solutions below the LCST and switch to solid drug containers at body temperature.⁵⁷ Park *et al.*⁵⁸ developed a thermo-sensitive methylcellulose (MC) hydrogel containing CaP nanoparticles using one-pot reaction. The LCST gelation behaviour of MC solution was affected by the polymer and salt concentrations (Fig. 3). For example, when 0.1 M CaCl₂ and 0.1 M Na₂HPO₄ were added to the MC solution, the gelation temperature dropped from 32.0 °C to 29.1 °C, the corresponding gelation time dropped from 54 s to 14 s.

In contrast, the UCST (upper critical solution temperature) induced hydrogel is prepared during the cooling of a polymer solution, and the cooling temperature is termed as UCST.⁵⁹ Hydrogels form by the micelle aggregation below UCST and disintegrate when the temperature restores UCST because the hydrophobic micelle cores become water-soluble.⁶⁰ Fu *et al.*⁶¹ prepared a thermo-reversible physically crosslinked hydrogel from UCST-type thermosensitive ABA linear triblock

copolymers, which were composed of hydrophilic poly(poly(ethylene glycol) methyl ether methacrylate) (PEGMMA) middle block and UCST-type thermosensitive poly(acrylamide-co-acrylonitrile) (P(AAm-co-AN)) outer block. Those triblock copolymers exhibited cooling-induced, reversible sol-gel transitions at concentrations of 3% and 5%. The transition temperature of sol-gel elevated along with the increased content of P(AAm-co-AN) segment as well as polymer concentration.

Ultrasonic induction. With the ultrasonically induced cross-linking there is always phase variation. Natural proteins or polymers, for example, silk fibroin, possess complex secondary structures (*e.g.* α -helix, β -sheet).⁶² The highly repetitive sequence GAGAGS of silk fibroin can generate the antiparallel β -sheet crystalline region that is able to self-assemble into hydrogel when exposed to heat, physical shear, or some organic solvents.^{16,17} The sol-to-gel transition is driven by physically crosslinked β -sheet crystals that are dependent on protein concentration, temperature, presence of metal ions, and pH. The ultrasonic treatment of the aqueous solution of silk fibroin can promote the formation of β -sheet crystals *via* hydrophobically hydrated alteration, thus, accelerating the formation of physically crosslinked silk fibroin hydrogels.¹⁷ Recently, a hybrid cell-encapsulating hydrogel of silk fibroin/collagen protein was fabricated by the ultrasonically induced gelation. The silk fibroin/collagen hydrogel exhibits physically crosslinked interpenetrating networks (IPNs) and tunable gelation and improved physical properties⁶³ (Fig. 4). Moreover, the hydrogel can easily encapsulate cells to provide a better 3D culture microenvironment.

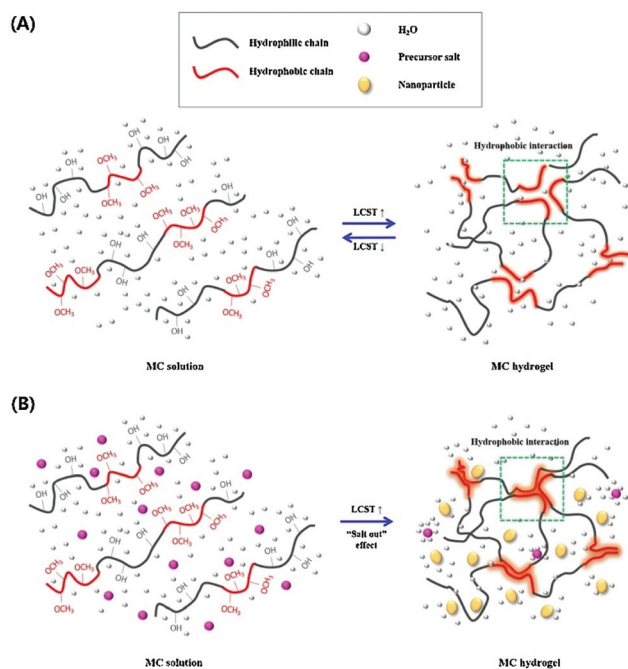


Fig. 3 The sol-to-gel transition process of MC, (A) pure MC solution and (B) MC solution with salt. As the temperature increases, hydrogen bonds are broken, which results in hydrogel formation. Addition of salts to the MC solution destroys hydrogen bonds and reduces the sol-gel transition temperature and time. Reproduced with permission from *Carbohydr. Polym.*, 2017, 15, 775–783. Copyright 2017 Elsevier.⁵⁸

Hydrogel crosslinking by crystallization

The crystallites of polymer chains act as physically crosslinking sites in the network, resulting in the hydrogel formation. For instance, the hydrogel will form when the aqueous solution of PVA repeatedly undergoes freeze-thawing. The properties of the above hydrogel depend on molecular weight, aqueous solution concentration, freezing temperature and time, and the number of freeze-thawing cycles.⁶⁴ A poly(vinyl alcohol) (PVA)/sodium alginate (SA) hydrogel was fabricated by the freeze-

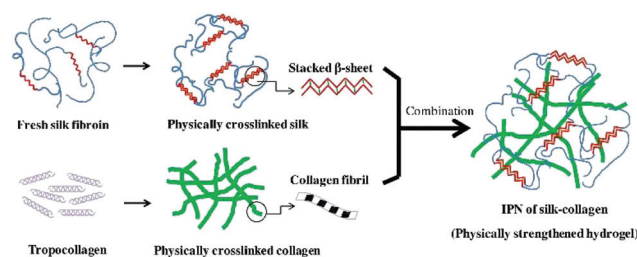


Fig. 4 The schematic illustration of ultrasonic induced gelation of fibroin/collagen IPNs. Silk fibroin solution was placed in an ultrasound device for inducing β -sheet formation, and the ultrasonicated solution was added to the collagen solution. The mixed solution was aliquoted into a mold and incubated at 37 °C for complete gelation. Reproduced with permission from *Acta Biomater.*, 2018, 69, 218–233. Copyright 2018 Elsevier.⁶³

thawing technique, and the saturated NaCl solution was used to enhance the gel strength and conductivity.⁶⁵ Zhang *et al.*⁶⁶ prepared a physical double network (PDN) hydrogel which was composed of a physically crosslinked PVA and a hydrophobically associated polyacrylamide (HAPAM) by one-pot *in situ* polymerization and subsequent freeze-thawing cycling (Fig. 5). Mechanical strength of PDN gel was improved owing to strong crystallization of PVA and the presence of hydrogen bonds between PVA and PAM chains.

The stereo-complex interaction can form racemic crystallization between 2 enantiomeric polymers, which can be used for the hydrogel preparation. Compared with homopolymer crystallite, the racemic crystallite exhibits more compact side by side crystallization.⁶⁷ Poly(L-lactic acid) (PLLA) and poly(D-lactic acid) (PDLA) are enantiomeric polymers, which can form a stereo-complex based on racemic crystallite. In a recent study,⁶⁸ an enantiomeric mixture of PDLA/poly(ethylene glycol) (PEG) di-block copolymer and PLLA/PEG triblock copolymer was used as a novel stereo-complex system. Hydrogel formed *via* unique gel-sol-gel multiple transitions upon heating, and the stereo-complexation of PLLA/PDLA segment played an important role in the phase transition of gel-sol-gel.

Hydrogel crosslinking by hydrogen bonding

Hydrogen bond is one of the most important noncovalent interactions. For example, hydrogen bond could stabilize a secondary structure during a peptide or agarose based hydrogel formation. For hydrogel formation, amide, urea, carboxylic acid, pyrrole, carbazole and hydroxyl groups could form hydrogen bonds among themselves or interact with electron donor groups, such as pyridine and imidazole groups. However, a single hydrogen bond is generally not strong enough to support hydrogel formation. By creating multiple multivalent hydrogen bonds, for example, a strong network can be formed using ureidopyrimidinone (UPy).⁶⁹

For example, a novel self-healing hydrogel was fabricated based on UPy-UPy interactions.⁷⁰ The DMAEMA (2-(dimethylamino)ethyl methacrylate) was copolymerized with 2-(3-(6-methyl-4-oxo-1,4-dihydropyrimidin-2-yl)ureido)ethyl methacrylate (SCMHBMA) containing UPy to produce DMAEMA-SCMHBMA copolymer. The polymer formed hydrogels above pH 8, and the hydrogels exhibited dynamic assembly and disassembly in response to damage along the stretching direction.

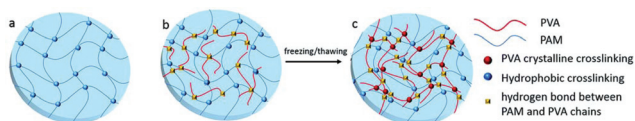


Fig. 5 Schematic illustration of hydrogel network structure: (a) HAPAM gel, (b) semi-IPN HAPAM/PVA gel, (c) PDN gel after freezing/thawing. The Semi-IPN HAPAM/PVA hydrogels were prepared by one-pot *in situ* polymerization in the PVA solution, and then the PDN hydrogel was fabricated *via* the freezing/thawing treatment. Reproduced with permission from *RSC Adv.*, 2016, 6, 112468–112476. Copyright 2016 RSC.⁶⁶

A multiblock copolymer with PEG and UPy was synthesized, and the polymer adopted nanoscopic physical cross-links between UPy-UPy dimers embedded in hydrophobic domains within the PEG matrix.⁷¹ The UPy units with self-complementary properties could assemble into dimers *via* 4-fold hydrogen bonds (H-bonding), which reinforced networks. The formed hydrogels exhibited high strength and resilience upon deformation as well as shape memory behaviour (Fig. 6).

A mechanically strong supramolecular hydrogel with self-healable property was created and the poly(*N*-acryloyl glycinamide) (PNAGA) containing glycinamide-conjugated monomer with dual-amide in one side group was able to reconstruct and amplify the hydrogen bonding interactions between amino acid residues in a polymer hydrogel.⁷² Among the dual amide motifs, the self-recognizing H-bonded supramolecular interaction occurred, and the hydrogel was rewarded with high mechanical performance, thermoplasticity and self-healing ability.

Hydrogel crosslinking by metal coordination

Metal coordination (metal-ligand interaction) between metal ions and functional groups in polymer chains is frequently used as a physical crosslinking method to fabricate hydrogels. The metal-ligand interaction can be regarded as a special Lewis acid-base interaction, which is stronger than most non-covalent interactions, but weaker than typical covalent bonding interactions.⁷³ Therefore, metal-ligand interactions can dynamically occur or disappear owing to their moderate bond energy that is responsible for the self-healing property.^{74–77}

For example, hydrogels derived from side-chain bipyridyl-functionalized poly(2-oxazoline)s could be crosslinked with Fe(II), Ru(II), Ni(II) or Co(III).^{78,79} The extent of hydrogel swelling is dependent on the degree of functionalization of a polymer. The metal-ligand interactions of Fe(II) and Co(III) were relatively stable at room temperature, but the interactions would disintegrate when the temperature reached 30 °C. That was caused by the exchange from intermolecular crosslinks to

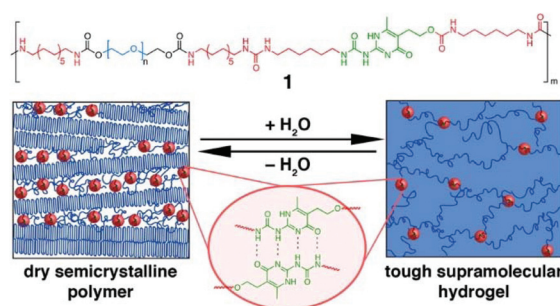


Fig. 6 The PEG-UPy copolymers with multiblock architectures and self-complementary quadruple H-bonding interaction between 2 UPy segments. The descriptive morphology of dry semicrystalline polymer and a reversible transition to hydrogel. Reproduced with permission from *J. Am. Chem. Soc.*, 2014, 136, 6969–6977. Copyright 2014 ACS.⁷¹

intramolecular crosslinks. In contrast, hydrogels crosslinked by Ru(II) were quite stable even in boiling water. When the liquid was evaporated from hydrogels, all dry gels could recover gelation in water.

In addition, Harrington *et al.*⁸⁰ discovered that mussel byssal threads originated from coordination between ferric ions and catechol ligands. The extent of catecholato-iron chelation influenced the stiffness, toughness, and self-healing capacity of the mussel byssal threads. Following the ferric ions and catechol interaction, a mussel-based hydrogel was fabricated using linear and branched PEG, which was end-functionalized with 1 to 4 3,4-dihydroxyphenylalanine (dopa) groups.^{81,82} Hydrogels were formed upon mixing with oxidizing agents. The 4-armed PEG was end-functionalized with dopa and presented a similar structure as catechol that could bind to Fe(III) ions to form catechol-Fe³⁺ complexes. The complex structure (mono-, bis-, or tris-) is controlled by pH *via* the deprotonation of the catechol hydroxyls. In detail, the metal–ligand interaction formed mono-complexes at pH below 5, bis-complexes about pH 8 and tris-complexes above pH 8. These results indicated that it was possible to control the gelation and crosslink density by pH variation.

Zheng's group synthesized a series of poly(acrylamide-*co*-acrylic acid) (P(AAm-*co*-AAc)) copolymers.⁸³ A specifically tough hydrogel was subsequently prepared by swelling a cast film of poly(acrylamide-*co*-acrylic acid) (P(AAm-*co*-AAc)) in FeCl₃ aqueous solution. The supramolecular networks were cross-linked by carboxyl-Fe³⁺ coordination bonds and the hydrogels possessed high stiffness and toughness, fatigue resistance, and stimulation-triggered healing along with shape memory and processing abilities. Furthermore, the strength of the coordination bond was controlled by varying pH, and the mechanical properties as well as shape memory abilities of hydrogels could be tuned (Fig. 7).

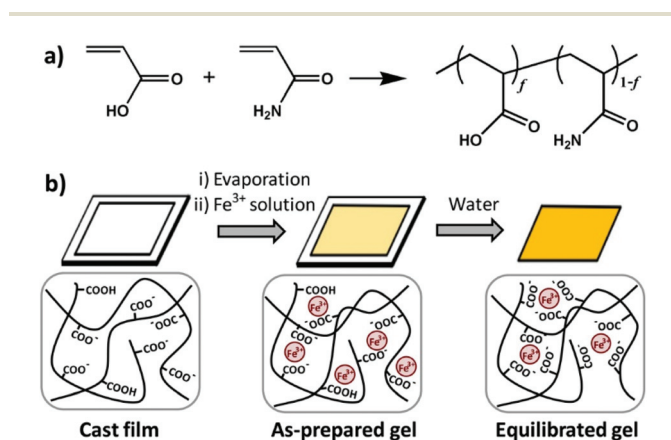


Fig. 7 Diagrams of synthesis of copolymers (a) and physical crosslinked hydrogels (b). Hydrogels were formed by carboxyl-Fe³⁺ coordination. The dynamic nature of coordination bonds was rate dependent and the mechanical performance of hydrogels could be tuned by controlling the composition of copolymers, concentration of metallic ions, and pH values of solutions. Reproduced with permission from *Macromolecules*, 2016, **49**, 9637–9646. Copyright 2016 ACS.⁸³

Hydrogel crosslinking by host–guest interactions

Among various noncovalent interactions for hydrogel fabrication, the host–guest interaction is quite an important crosslinking strategy. By host–guest inclusion, polymers can be integrated in a facile and reversible way, promoting construction of hydrogels.⁸⁴ Usually, a host is a molecule with a large cavity volume, such as cyclodextrins (CDs), cucurbiturils (CBs), calixarenes (CAs) and crown ethers, while guests with complementary shapes can interact with the hosts. Various noncovalent interactions can facilitate the host–guest inclusion, such as hydrogen bonding, electrostatics, and van der Waals, hydrophobic interactions; molecular shape is also important. Furthermore, the host–guest interaction is reversible and could be selective between one host and one guest.⁸⁵

In aqueous solutions, host–guest interactions usually occur by encapsulating hydrophobic guest molecules into hydrophobic cavities. This host–guest inclusion exhibits strong binding, and, more importantly, fixed geometry and directionality of the host–guest inclusion can respond to pH⁸⁶ or other physical stimuli.^{87,88} Therefore, it can be useful for drug delivery. Among various host molecules, cyclodextrins (CD) are widely used because of the hydrophilic surface outside and the hydrophobic cavity inside.

For example, PEG can form host–guest complexes with α -cyclodextrin (α -CD), and PEG can crosslink to form larger aggregates or thread among a series of CD molecules. When PEG is mixed with α -CD, the inclusion complexes of PEG chains in α -CD cavities are formed after several days. The time and concentration of PEG required to form α -CD inclusion complexes were dependent on the molecular weight of PEG.^{89–91}

PEG-CD inclusion complexes are always designed as injectable hydrogels and used for a controlled drug release. A hydrogel was prepared by mixing PEG-*b*-poly[(*R*)-3-hydroxybutyrate]-*b*-PEG with α -CD.⁹² The inclusion of PEG segments with α -CD, as well as the hydrophobic interactions between poly[(*R*)-3-hydroxybutyrate] blocks, resulted in the formation of a strong macromolecular network, which could be used for a long-term sustained controlled release of macromolecular drugs. Wang *et al.*⁹³ prepared a branched polyrotaxane hydrogel based on α -CD inclusion with low-molecular-weight four-arm PEG. The hydrogel was injectable and exhibited shear-thinning and thixotropic properties. Drug release was controlled by shear stress.

Recently, to improve the mechanical properties the PEG-CD based hydrogels, Liu *et al.*⁹⁴ synthesized nucleobase guanine/cytosine (G/C)-terminated PEGs (G-PEG-G and C-PEG-C), where the base-pairing interaction between G and C enhanced the storage moduli of the PEG-CD inclusion complex. The hydrogels exhibited better mechanical properties, because the G–C base pair formation acted as an additional network junction (Fig. 8). The hydrogels were thermo-responsive and could be used as thermo-controlled drug delivery systems.

Chemically crosslinked hydrogels

Covalent bonds are normally formed among polymer chains in chemically crosslinked hydrogels, and most of their linkages

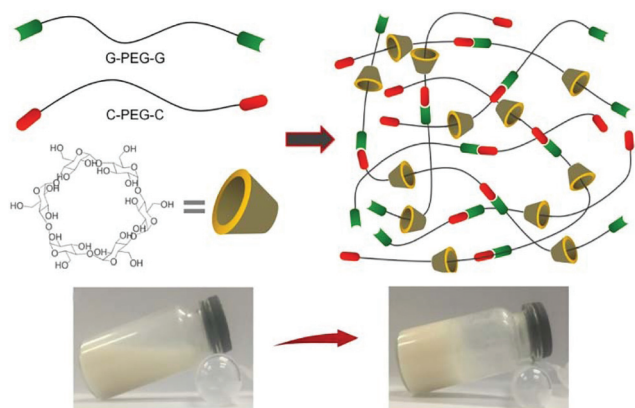


Fig. 8 Synthesis of supramolecular hydrogels by host–guest inclusion between guanine/cytosine (G/C)-terminated PEG and α -cyclodextrin. The G–C base pairing acts as additional network junctions and enhances the hydrogel mechanical properties. These hydrogels also exhibit excellent cytocompatibility and temperature-responsivity. Reproduced with permission from *Mater. Sci. Eng., C*, 2018, **82**, 25–28. Copyright 2016 Elsevier.⁹⁴

are strong and permanent when they are compared with those of the physically crosslinked hydrogels. Up to now, several crosslinking methods have been reported,⁹⁵ and they typically involve free radical polymerization induced crosslink, enzymatic induced crosslink, Diels–Alder “click” reaction, Schiff base formation, oxime formation and Michael type-addition. Compared with physically crosslinked hydrogels, chemically crosslinked hydrogels usually exhibit enhanced stability under physiological conditions and excellent mechanical properties as well as tuneable degradation behaviour.

Hydrogel crosslinking by photopolymerization

Photo-activated crosslinking has been widely used for hydrogel formation in the field of therapeutics or cytokines encapsulation.^{96–99} The advantage of this method is the rapid formation of hydrogel networks at ambient temperature under mild conditions, and the mechanical properties of hydrogels can be tuned by controlling the crosslinking reactions.¹⁰⁰ The crosslinked site is also ready to be accurately selected, because the photo-initiated polymerization takes place under light exposure and only the irradiated areas are involved in hydrogel crosslinking.¹⁰¹

Photo-initiated polymerization is related to the presence of unsaturated groups, in most situations, the (meth)acrylates. The double bonded carbons in these groups are highly reactive and promote a free radical chain-growth polymerization when they are exposed to photo irradiation. Conventionally, water-soluble polymers with hydroxyl, carboxyl and amino groups can react with acryloyl chloride, glycidylmethacrylate (GMA) and *N*-(3-aminopropyl) methacrylamide to introduce vinyl groups.¹⁰² Formation of biomedical hydrogels usually requires the presence of cytocompatible photoinitiators, such as Irgacure 2959,¹⁰³ Irgacure 1173,¹⁰⁴ Irgacure 819,¹⁰⁵ Irgacure 651,¹⁰⁶ riboflavin phosphate,¹⁰⁷ camphorquinone,¹⁰⁸ eosin

Y,¹⁰⁹ and so on. Those photo-initiators can absorb specific light at different wavelengths, including UV (250 nm–370 nm), visible blue & purple (405 nm–550 nm) and red light (750 nm–810 nm) and either decompose (Type I) or abstract hydrogen from a donor molecule (Type II) to form radical initiating species.¹⁰⁰ However, type II initiators always require a co-initiator. For example, camphorquinone (CQ) requires ethyl 4-*N,N*-dimethylaminobenzoate (4EDMAB), triethanolamine (TEA) and the photosensitizer iso-propyl thioxanthone when the hydrogel is crosslinked by visible light.¹¹⁰ Another example is eosin Y, where the hydrogel crosslinking relies on triethanolamine and *N*-vinyl pyrrolidone (NVP) to promote the visible light activated photopolymerization.¹⁰⁹

For all photo-initiators, cytocompatibility of light-activated hydrogel systems should be initially considered when they are designed to encapsulate cells and drugs. Bryant group¹¹⁰ systematically investigated the cytocompatibility of various photo-initiators with different photo-initiator concentrations and intensities. The 3T3 cells were exposed to photo-initiators under varying concentrations from 0.01% (w/w) to 0.1% (w/w). At low photo-initiator concentrations ($\leq 0.01\%$ (w/w)), most of the initiator molecules showed excellent biocompatibility, while CQ, Irgacure 651 and 4EDMAB exhibited a relatively low survival. In the presence of low intensity initiating light (365 nm UV light about 6 mW cm^{-2} , and about 470–490 nm visible light about 60 mW cm^{-2}), the Irgacure 2959 at concentrations $\sim 0.05\%$ (w/w) and CQ at concentrations $\sim 0.01\%$ (w/w) were the most promising cytocompatible UV and visible light initiating systems.

For ultraviolet light (UV-light) induced polymerization, the initiator, such as Irgacure 2959,¹⁰³ is prevailing to prepare the functional hydrogels with well-designed patterned structure *in situ*. More importantly, cytotoxicity of UV-induced free radicals is relatively low. In general, the photo-initiator Irgacure 2959 is well tolerated by many cell types and it is the least cytotoxic.¹⁰³ The long-wave UV light centred around 365 nm is mostly applied, because this long-wave UV exposure at intensities less than 10 mW cm^{-2} is well tolerated by most cell types for exposure times less than 3–5 min, providing cell viability following photo encapsulation typically greater than 90%. For example, a methacrylated γ -PGA (mPGA) was prepared¹¹¹ *via* crosslinking between γ -PGA and glycidyl methacrylate (GMA). The hydrogel synthesized by UV-light radiation exhibited ionic- and pH-sensitive properties and low cytotoxicity towards bovine chondrocytes.¹¹² A chitosan hydrogel with a cell-loading pattern was fabricated by incorporating UV crosslinking method.¹¹³ This hydrogel exhibited low cytotoxicity based on the acute inflammatory response (Fig. 9).

However, there are still some concerns about DNA damage caused by UV radiation.¹¹⁴ Some studies claimed that UV radiation posed a potential risk of accelerating organ/tissue aging¹¹⁵ and cancer onset.¹¹⁶ Thus, visible light photoinitiation might be an alternative candidate. Blue light (visible) photo-initiators have been used, such as camphorquinone¹⁰⁸ and eosin Y,¹¹⁷ lithium phenyl-2,4,6-trimethylbenzoylphosphinate (LAP),¹¹⁸ riboflavin¹¹⁹ and ruthenium.¹²⁰

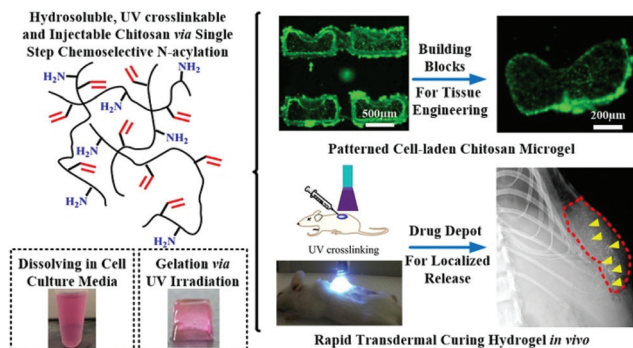


Fig. 9 UV irradiation induced radical polymerization for hydrogel synthesis. Water-soluble *N*-methacryloyl chitosan (N-MAC) was synthesized and able to fabricate cell-loading microgels with the desired patterns. The hydrogels could be utilized in tissue engineering as a platform for sustained protein delivery. Reproduced with permission from *Acta Biomater.*, 2015, 22, 59–69. Copyright 2015 Elsevier.¹¹³

These photo-initiators were able to effectively photo encapsulate various cells with high viability. Besides, visible light more readily penetrates tissue, which creates a pre-polymer solution forming hydrogel after subcutaneous injection. For example, Shih *et al.*¹¹⁷ reported a visible-light-mediated thiol-ene hydrogel using eosin-Y as the only photo-initiator. The hydrogel exhibited a rapid and tunable step-growth gelation by PEG-norbornene crosslinking with dithiothreitol under visible light exposure. Recently, chemical modification of photo-initiator appears to be more exciting as it could improve the photo-initiation effect. A carboxylated camphorquinone was synthesized to increase water solubility and photo reactivity of camphorquinone. The resultant hydrogel based on carboxylated-camphorquinone showed a significantly higher photoactivity and crosslink density as well as better mechanical properties.¹²¹

Light attenuation by the initiator restricts the maximum attainable cure depth to a few millimetres.¹²² To overcome this issue, polymerization of co-monomers containing complementary reactive groups, such as thiol-acrylates/enes, offers a better choice. The advantage of this strategy is the presence of a more homogeneous structure, greater tensile strength and increased strain at break. For thiol-acrylate/ene polymerization, the photo-initiator radical abstracts a hydrogen atom from a thiol and forms a thiyl radical. Polymerization was caused by the presence of the thiyl radicals and the corresponding thiol-acrylate/ene photopolymerization can rapidly occur with a low light intensity.¹²³ More interesting, this approach can be employed in the absence of a photo-initiator, thereby, allowing curing to greater depths even in the lack of light intensity due to the photo-initiator attenuation.

In the process of thiol-acrylate photopolymerization, the competing Michael-type addition reaction between the thiol and acrylates takes place at the same time, resulting in a mixed-mode polymerization. Such mixed-mode polymerizations can lead to tuneable hydrogel properties, such as network structure and degradation by adjusting the ratio of

thiol and acrylate. Furthermore, the residual thiol groups remain to be utilized to functionalize post-polymerization of the hydrogel. One elegant approach for the fast polymer post-functionalization and step-growth polymerization has been suggested, and it is based on the visible light photocatalytic thiol-ene “click” reaction.¹²⁴ Two model polymers, polybutadiene and poly(allyl methacrylates), were modified with a large range of functional thiols. Subsequently, photoredox thiol-ene was successfully used to prepare linear polymers by step-growth addition reactions, which was highly efficient under low-energy (blue LED, 4.8 W) and eco-friendly visible light in the presence of air.

Hydrogel crosslinking by enzyme catalyzed reactions

Enzymatic crosslinking is an attractive method, as it offers the possibility for kinetic manipulation of gel formation *in situ* by controlling the enzyme concentration.¹²⁵ The advantage of an enzymatic crosslink is the strongly covalent bonding as well as rapid gelation (always no more than 10 min) under physiological conditions. So far, there are many kinds of enzymatically catalyzed crosslinked methods for hydrogel formation *in situ*. For instance, the transglutaminase (TG) along with calcium ions as cofactors can promote formation of amide linkages between carboxamide and amine groups.¹²⁶ Another crosslinking method is the horseradish peroxidase (HRP)-catalyzed crosslinking reaction. HRP catalyzes the coupling of aniline,²¹ phenol¹²⁷ and their derivative tyramine^{22,128–130} in the presence of hydrogen peroxide (H₂O₂). Kuo group constructed a novel vascularized tissue using an injectable cell-loaded enzymatically crosslinked collagen hydrogel. Collagen was conjugated with tyramine (collagen-Ph) and the hydrogel was formed in the presence of HRP and H₂O₂. The human blood-derived endothelial colony-forming cells (ECFCs) and bone marrow-derived mesenchymal stem cells (MSCs) were incorporated in the hydrogels. The results of cell evaluation indicated that human ECFCs-lined vascular networks were able to be generated within 7 days. The animal evaluation further proved that cell-combined collagen-Ph hydrogel could improve long-term differentiation of MSCs into osteoblasts as well as the increase the number of adipocytes in the mouse model after 1 month of implantation¹³¹ (Fig. 10).

Hydrogel crosslinking by “click chemistry”

Utilization of “click chemistry” for design and fabrication of functional hydrogels grew rapidly since 2001.²⁵ Chemical reactions that are termed click reactions possess advantages such as high yields under mild conditions, fewer by-products, high specificity and selectivity. A wide variety of functional groups can act as attractive candidates for the fabrication of complex polymeric materials.¹³² Here, we illustrate several classical crosslinking methods, including (1) Diels–Alder, (2) Schiff base, (3) oxime and (4) Michael-type addition.

Hydrogel crosslinking by Diels–Alder reaction. Diels–Alder (DA) reaction is a highly selective [4 + 2] cycloaddition between a diene (furan) and a dienophile (maleimide), which is free from side reactions and byproducts. The advantage of this reaction

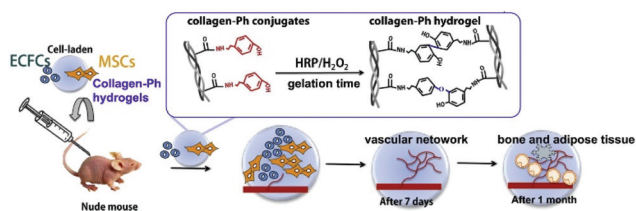


Fig. 10 Schematic illustration of fabricating collagen-Ph hydrogel and *in vivo* experiment. Collagen was conjugated with tyramine and the injectable hydrogel containing ECFs and MSCs formed in the presence of HRP/H₂O₂. Cell-loaded hydrogels could form vascular networks after 7 days and promote adipose and bone-like tissue formation. Reproduced with permission from *Acta Biomater.*, 2015, 27, 151–168. Copyright 2015 Elsevier.¹³¹

is one-step crosslinking without the use of any initiators, catalysts and coupling agents.³¹ Polymers were usually modified with furan or furan derivatives to react with poly(ethylene glycol) dimaleimide for hydrogel formation.¹³³ In recent years, methylfuran was used to replace furan, because it provided more electron-rich diene, thus, accelerating the DA reaction at pH 7.4¹³⁴ (Fig. 11).

Hydrogel crosslinking by Schiff base formation. The formation of Schiff base conventionally occurs between amino and aldehyde groups to generate an imine linkage under physiological conditions. It can be easily applied to construction of an injectable and *in situ* forming hydrogel, because the aldehyde group can strongly adhere to tissues or organs.¹³⁵ The Schiff base takes advantage of dynamic equilibrium between the Schiff base linkages and the aldehyde and amine reactants; these linkages can be viewed as pseudo-covalent bonds. The uncoupling and recoupling of imine linkages take place in hydrogel networks, resulting in self-healing capability of the hydrogel.¹³⁶ For example, a self-healable polymeric hydrogel could be fabricated *via* self-crosslinking between the amino group from acrylamide-modified chitin (AMC) and dialdehyde group (ADA) from oxidized alginate. The self-healing capability is also dependent on the molar ratio of AMC and ADA as well as the surrounding pH¹³⁷ (Fig. 12).

In another case, the reaction of Schiff base can also occur between aldehyde and hydrazide groups. Ma *et al.*¹³⁸ developed a novel injectable hydrogel for protein delivery by self-

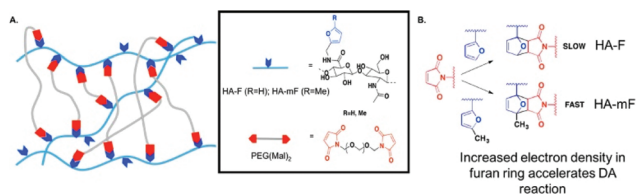


Fig. 11 Schematic illustration of Diels–Alder (DA) reaction. When the furan group is replaced by methylfuran, DA reaction accelerates and hydrogels can be synthesized efficiently at pH 7.4 since the methylfuran group provided a more electron-rich diene. Reproduced with permission from *Biomacromolecules*, 2018, 19, 926–935. Copyright 2018 ACS.¹³⁴

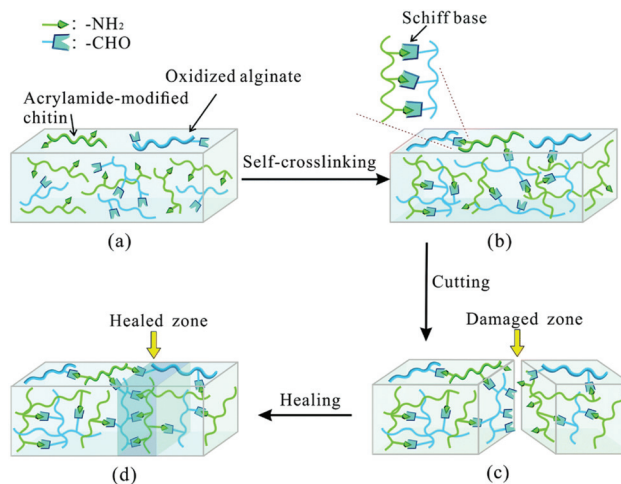


Fig. 12 Schematic illustration of Schiff base formation and self-healing process. The self-crosslinked hydrogel was synthesized *via* Schiff base linkage between the amino group of acrylamide-modified chitin (AMC) and the dialdehyde group (ADA) of oxidized alginate. The self-healing capability relied on the molar ratio of AMC and ADA as well as pH. Reproduced with permission from *Soft Matter*, 2015, 11, 3971–3976. Copyright 2015 RSC.¹³⁷

crosslinking of aldehyde hyaluronic acid (HA-CHO) and hydrazide-modified poly (γ -glutamic acid) (γ -PGA-ADH). The gelation time of hydrogel was as fast as 9s with high swelling ratios. Simultaneously, the hydrogel exhibited favorable mechanical properties and biocompatibility.

Hydrogel crosslinking by oxime crosslink. Oxime crosslink is a reaction between an aminoxy/hydroxylamine group and an aldehyde or ketone. The oxime bond exhibits enhanced hydrolytic stability as the equilibrium lies far toward the oxime.³⁵ Like DA (Diels–Alder) reaction, it is a chemo-specific “click” reaction as 2 reactive agents react efficiently and specifically with each other, even in the presence of other functional groups. Besides, the oxime formation can be catalyzed under acidic conditions and the only byproduct is water.³⁶ Compared with a hydrazine, an oxime bond is more stable. But it is reversible in some biologically relevant environments.³⁷ Mukherjee *et al.*³⁴ developed self-healing oxime-functional hydrogels, which could undergo a reversible gel-to-sol transition *via* oxime exchange under acidic conditions. They synthesized keto-functional P(DMA-*stat*-DAA), and the hydrogel was prepared by adding difunctional alkoxyamines into the P(DMA-*stat*-DAA) polymer solutions. In the presence of excess of monofunctional alkoxyamine and an acid catalyst, the reversible sol-to-gel transition could be realized and awarded hydrogels with self-healing ability (Fig. 13).

Hydrogel crosslinking by Michael addition. Michael type-addition or Michael addition is a facile reaction between nucleophiles (Michael donors) and activated electrophilic olefins or alkynes (Michael acceptors) whereby a nucleophile is added across a carbon–carbon multiple bond. On the one hand, the donor has sufficient nucleophilicity. Donors include not only enolate nucleophiles but also non-enolate nucleo-

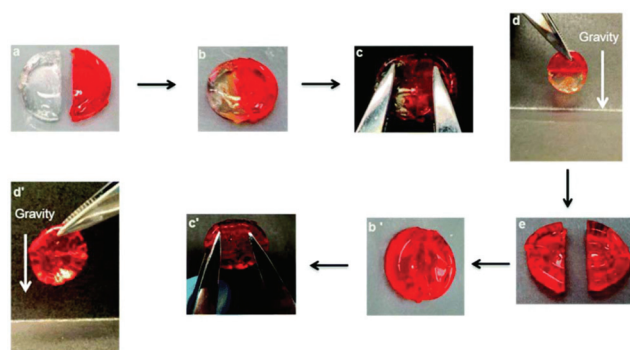


Fig. 13 Schematic illustration of self-healing behaviour. Two previously cut pieces of hydrogels (a) were placed in contact for 2 h (b) and the repaired gels were stretched with tweezers (c and c'); the healed gels were suspended under gravity (d and d'). When the gel was self-healed for 24 h, the hydrogel was cut again (e). Reproduced with permission from *Soft Matter*, 2015, 11, 6152–6161. Copyright 2015 RSC.³⁴

philes such as amines, thiols, and phosphines. On the other hand, the acceptor with electron withdrawing and resonance ability can stabilize the anionic intermediate. In detail, Michael addition acceptors include acrylate esters, acrylonitrile, acrylamides, maleimides, alkyl methacrylates, cyanoacrylates and vinyl sulfones. Advantages of Michael addition include mild reaction conditions, highly regioselective and efficient click chemistry, and favourable reaction rates.¹³⁹ Moreover, coupling of polymers is always facilitated by the Michael reaction, which makes the reaction well-suited for gene transfection, cellular scaffold and tissue replacement.²⁸ For construction of hydrogels, Michael addition often refers to thiol containing polymers added to the activated α,β -unsaturated carbonyl polymers under basic conditions.¹⁴⁰ As thiols are present in proteins in cysteine residues, hydrogels can be easily synthesized between proteins and vinyl containing polymers. However, there are side reactions in thiol Michael additions, because disulfide bonds could form during the addition process.

A chitosan-based hydrogel with high mechanical strength was prepared *via* Michael addition.¹⁴¹ Hydrogels were fabricated from poly(ethylene glycol) diacrylate (PEGDA) and 3 thiolated natural polymers (chitosan, gelatin and heparin) *via* Michael addition under physiological conditions. Poly(ethylene glycol-*b*-caprolactone-*b*-ethylene glycol) (PECL) micelles with double bonds (bi-acrylated PECL micelles) were also incorporated to form mechanically reinforced chitosan-based hydrogels. Drugs were trapped inside the PECL micelles and hydrogels containing these drug-loading micelles exhibited excellent delivery performance (Fig. 14).

Another *in situ*-forming Michael addition hydrogel was synthesized by mixing dithiothreitol (DTT) and glycidyl methacrylate derivatized dextran (Dex-GMA) in phosphate buffered saline (PBS). By changing the pH of PBS, the mechanical properties, gelation and the degree of swelling of the hydrogels could be easily tuned. The cell-loaded hydrogel exhibited good viability and cytocompatibility.¹⁴²

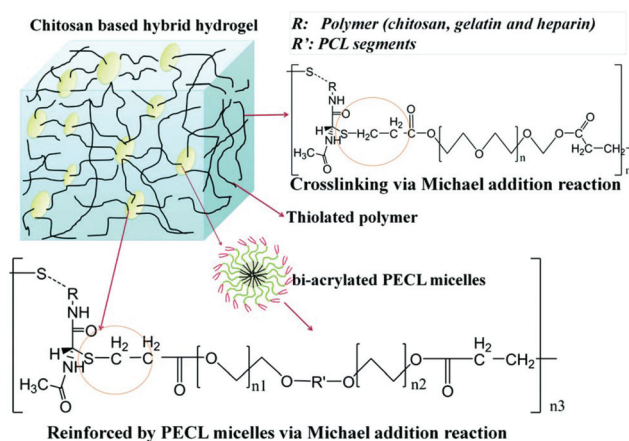


Fig. 14 Schematic illustration of Michael addition of chitosan-based hydrogels cross-linked with PECL micelles. Chitosan, gelatin and heparin were thiolated and crosslinked with PEGDA and bi-acrylated PECL micelles at different molar ratios. Reproduced with permission from *J. Mater. Chem. B*, 2017, 5, 961–971. Copyright 2017 RSC.¹⁴¹

Hydrogel crosslinking by dynamic covalent chemistry

The reversibility in hydrogels can be achieved *via* both physical crosslinks and reversible covalent bonds. The advantage of a reversible crosslink is its ability to make the hydrogels not only maintain the robust integrity of covalently cross-linked materials, but also employ the intrinsic reversibility of physical crosslinking.¹⁴³ Boronic ester, as an example of a molecule containing dynamic covalent bonds, has a great potential for preparation of self-healing hydrogels. Recently, reversible boronate esters prepared from boronic acids and 1,2- or 1,3-diols have been explored to create self-healing hydrogels.¹⁴⁴ In aqueous media, the strength and reversibility of a boronate ester are largely dependent on the pH value of solution and the pK_a of the boronic acid component. Formation of the boronate ester bond is favoured at pH values above the pK_a of the boronic acid, whereas the free boronic acid and diol are favoured at pH values below the pK_a . This dynamic equilibrium between diol/boronic acid and boronate ester governs bond rearrangement to exhibit a self-healing behaviour. The newly formed bonds could span the damaged interface between 2 boronate ester-cross-linked materials. Smithmyer's group¹⁴⁵ developed new boronic acid-based hydrogels that exhibited good viability in MDA-MB-231 breast cancer cells and CCL151 pulmonary fibroblasts. Huang *et al.* synthesized injectable hydrogels based on boronated PEG coupling with various linkers to form dynamic boronate ester bonds, such as plant derived polyphenols ellagic acid (EA), epigallocatechin gallate (EGCG), tannic acid (TA), nordihydroguaiaretic acid (NDGA), rutin trihydrate (RT), rosmarinic acid (RA) and carminic acid (CA).¹⁴⁶ Among all polyphenol linkers, the ellagic acid (EA), epigallocatechin gallate (EGCG) and tannic acid (TA) were proved to form stable hydrogels under physiological conditions. Chen *et al.*¹⁴⁷ prepared 2 kinds of cell-membrane mimicking copolymers based on 2-methacryloyloxyethyl phosphorylcholine (MPC), which were modified with benzoxaborole

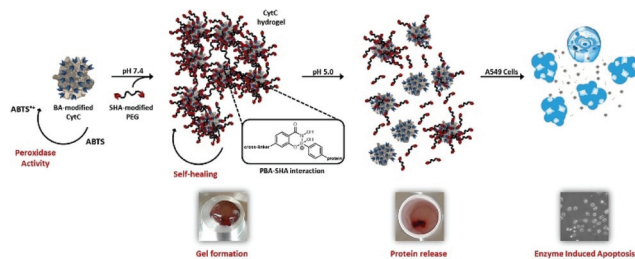


Fig. 15 Schematic illustration of a smart hydrogel with dynamic covalent bonds based on boronic acid/salicyl hydroxamate, which incorporated an apoptosis enzyme, cytochrome c. The hydrogel is bioactive and responsive to controlled administration. Reproduced with permission from *Tetrahedron*, 2017, 73, 499–4987. Copyright 2017 Elsevier.¹⁴⁹

and catechol pendant groups. Hydrogel gelation occurred rapidly at pH 7.4 and the hydrogels exhibited self-healing ability as well as pH/sugar dual responses. Also, hydrogels containing dynamic bonds between boronic acid and alginate were prepared,¹⁴⁸ the carboxyl groups of alginate were conjugated with the amine group of boronic acid (alginate-BA) and the reversible boronate-*cis*-diol complexation took place between boronated groups and the intrinsic *cis*-diol along alginate backbones. Seidler *et al.* applied a molecular recognition strategy based on boronic acid/salicyl hydroxamate, where an apoptosis enzyme, cytochrome c, was encapsulated into a hydrogel *via* dynamic covalent interactions (Fig. 15).¹⁴⁹ This strategy facilitated the structural integration of native enzymes into the hydrogel scaffold and provided a stimulus-controlled enzyme release.

Perspectives

Comprehensive hydrogel crosslinking strategies

Hydrogel properties are highly associated with their crosslinking methods, so functional improvement of hydrogels is extremely dependent on the innovation of fabrication strategies. Normally, chemical methods are apt to achieve a stable hydrogel with suitable mechanical properties, whereas a physical method takes advantage of biocompatibility owing to the absence of chemical crosslinking agents. Consequently, a combination of physical and chemical approaches would be beneficial for attaining balanced properties of hydrogels. Actually, several pioneering studies have reported applying the combined crosslinking strategy to produce self-healing hydrogels.¹⁵⁰ For instance, a double crosslinked hydrogel was prepared *via* physical crosslinking of hydrogen bonds and chemical crosslinking of imine bonds.¹⁵¹ This type of hydrogel exhibited an excellent self-healing capability. Besides, the physical crosslinking can act as the secondary crosslinking for enhancing the mechanical strength of a hydrogel. Consequently, a comprehensive crosslinking strategy of hydrogel generation paves a new way to construct self-healing smart hydrogels.

Hierarchical structures of hydrogels

Ideally, bio-inspired hydrogels should have anisotropic and highly ordered architecture because natural soft tissues or organs consist of multiscale hierarchical structures.¹⁵² Hydrogel crosslinking methods play a pivotal role on determining the inner structure of hydrogels. So far, a bi-layered composite hydrogel with a well-designed hierarchical structure was successfully developed to promote osteochondral tissue regeneration.^{153,154} Thus, the hierarchically biomimetic structure of hydrogels should be considered as a crucial factor prior to designing and constructing new functional hydrogels. For example, the double crosslinking and interpenetrating polymer network (IPN) hydrogels usually have hierarchical structures as they are prepared using step by step crosslinking procedures.

Intelligent stimuli-responsibility of hydrogels

Hydrogels, particularly injectable ones, have been widely applied in drug delivery systems. However, designing an intelligent hydrogel for drug delivery that can be modulated by physical and chemical stimulation is still a major challenge in the field of hydrogel-based drug delivery. This kind of drug delivery system is capable of releasing an appropriate amount of drug whenever required by the inflammatory tissue. To realize the on-demand drug release therapeutics, hydrogels should be synthesized *via* physical or chemical stimuli-responsive interactions.¹⁵⁵ The method of drug-loaded hydrogel preparation influences the stimuli-responsive drug delivery behaviour and more innovative crosslinking methods should be proposed.

Conclusions

In this review, we summarized the typical crosslinking methods of hydrogel synthesis involving physical and chemical strategies and illustrated their specific mechanisms and conditions of hydrogel formation. We also suggested potential prospects of hydrogel development based on a balanced combination between physical and chemical approaches, construction of hierarchical hydrogels and stimuli-responsive hydrogels for drug delivery systems. In the future, hydrogels with desirable mechanics and favourable biocompatibility could be useful for tissue engineering and regenerative medicine.

Conflicts of interest

There are no conflicts to declare.

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References

- F. Gattazzo, A. Urciuolo and P. Bonaldo, *Biochim. Biophys. Acta*, 2014, **1840**, 2506–2519.
- F. M. Watt and W. T. Huck, *Nat. Rev. Mol. Cell Biol.*, 2013, **14**, 467–473.
- J. K. Mouw, G. Ou and V. M. Weaver, *Nat. Rev. Mol. Cell Biol.*, 2014, **15**, 771–785.
- M. W. Tibbitt and K. S. Anseth, *Biotechnol. Bioeng.*, 2009, **103**, 655–663.
- K. Y. Lee and D. J. Mooney, *Chem. Rev.*, 2001, **101**, 1869–1879.
- N. A. Peppas, J. Z. Hilt, A. Khademhosseini and R. Langer, *Adv. Mater.*, 2006, **18**, 1345–1360.
- B. P. Hung, J. N. Harvestine, A. M. Saiz, T. Gonzalez-Fernandez, D. E. Sahar, M. L. Weiss and J. K. Leach, *Biomaterials*, 2019, **189**, 1–10.
- M. Liu, X. Zeng, C. Ma, H. Yi, Z. Ali, X. Mou, S. Li, Y. Deng and N. He, *Bone Res.*, 2017, **5**, 17014.
- S. Q. Liu, R. Tay, M. Khan, P. L. R. Ee, J. L. Hedrick and Y. Y. Yang, *Soft Matter*, 2010, **6**, 67–81.
- W. E. Hennink and C. F. van Nostrum, *Adv. Drug Delivery Rev.*, 2012, **64**, 223–236.
- A. S. Hoffman, *J. Controlled Release*, 1987, **6**, 297–305.
- X. S. Wu, A. S. Hoffman and P. Yager, *J. Polym. Sci., Polym. Chem.*, 1992, **30**, 2121–2129.
- L. C. Dong and A. S. Hoffman, *J. Controlled Release*, 1986, **4**, 223–227.
- M. Boustta, P. E. Colombo, S. Lenglet, S. Poujol and M. Vert, *J. Controlled Release*, 2014, **174**, 1–6.
- H. Y. Ye, C. Owh and X. J. Loh, *RSC Adv.*, 2015, **5**, 48720–48728.
- X. Hu, Q. Lu, L. Sun, P. Cebe, X. Wang, X. Zhang and D. L. Kaplan, *Biomacromolecules*, 2010, **11**, 3178–3188.
- X. Wang, J. A. Kluge, G. G. Leisk and D. L. Kaplan, *Biomaterials*, 2008, **29**, 1054–1064.
- W. Zhang, X. Wang, S. Wang, J. Zhao, L. Xu, C. Zhu, D. Zeng, J. Chen, Z. Zhang, D. L. Kaplan and X. Jiang, *Biomaterials*, 2011, **32**, 9415–9424.
- S. L. Bourke, M. Al-Khalili, T. Briggs, B. B. Michniak, J. Kohn and L. A. Poole-Warren, *AAPS PharmSci.*, 2003, **5**, E33.
- I. Mironi-Harpaz, D. Y. Wang, S. Venkatraman and D. Seliktar, *Acta Biomater.*, 2012, **8**, 1838–1848.
- S. Kobayashi, H. Uyama and S. Kimura, *Chem. Rev.*, 2001, **101**, 3793–3818.
- K. S. Kim, S. J. Park, J. A. Yang, J. H. Jeon, S. H. Bhang, B. S. Kim and S. K. Hahn, *Acta Biomater.*, 2011, **7**, 666–674.
- R. Jin, L. S. Teixeira, P. J. Dijkstra, C. A. van Blitterswijk, M. Karperien and J. Feijen, *Biomaterials*, 2010, **31**, 3103–3113.
- L. S. Teixeira, J. Feijen, C. A. van Blitterswijk, P. J. Dijkstra and M. Karperien, *Biomaterials*, 2012, **33**, 1281–1290.
- H. C. Kolb, M. G. Finn and K. B. Sharpless, *Angew. Chem.*, 2001, **40**, 2004–2021.
- M. van Dijk, D. T. Rijkers, R. M. Liskamp, C. F. van Nostrum and W. E. Hennink, *Bioconjugate Chem.*, 2009, **20**, 2001–2016.
- C. E. Hoyle and C. N. Bowman, *Angew. Chem.*, 2010, **49**, 1540–1573.
- B. D. Mather, K. Viswanathan, K. M. Miller and T. E. Long, *Prog. Polym. Sci.*, 2006, **31**, 487–531.
- M. P. Lutolf and J. A. Hubbell, *Biomacromolecules*, 2003, **4**, 713–722.
- D. P. Nair, M. Podgorski, S. Chatani, T. Gong, W. X. Xi, C. R. Fenoli and C. N. Bowman, *Chem. Mater.*, 2014, **26**, 724–744.
- C. M. Nimmo, S. C. Owen and M. S. Shoichet, *Biomacromolecules*, 2011, **12**, 824–830.
- A. Gandini, *Prog. Polym. Sci.*, 2013, **38**, 1–29.
- Z. Wei, J. H. Yang, X. J. Du, F. Xu, M. Zrinyi, Y. Osada, F. Li and Y. M. Chen, *Macromol. Rapid Commun.*, 2013, **34**, 1464–1470.
- S. Mukherjee, M. R. Hill and B. S. Sumerlin, *Soft Matter*, 2015, **11**, 6152–6161.
- J. Kalia and R. T. Raines, *Angew. Chem.*, 2008, **47**, 7523–7526.
- G. N. Grover, J. Lam, T. H. Nguyen, T. Segura and H. D. Maynard, *Biomacromolecules*, 2012, **13**, 3013–3017.
- F. Lin, J. Yu, W. Tang, J. Zheng, A. Defante, K. Guo, C. Wesdemiotis and M. L. Becker, *Biomacromolecules*, 2013, **14**, 3749–3758.
- J. Collins, Z. Y. Xiao, M. Mullner and L. A. Connal, *Polym. Chem.*, 2016, **7**, 3812–3826.
- W. Tao, T. Mahir and G. Sundaram, *Polym. Int.*, 2004, **53**, 911–918.
- Y. Xin and J. Y. Yuan, *Polym. Chem.*, 2012, **3**, 3045–3055.
- Y. Jia and J. Li, *Chem. Rev.*, 2015, **115**, 1597–1621.
- L. Voorhaar and R. Hoogenboom, *Chem. Soc. Rev.*, 2016, **45**, 4013–4031.
- J. Berger, M. Reist, J. M. Mayer, O. Felt, N. A. Peppas and R. Gurny, *Eur. J. Pharm. Biopharm.*, 2004, **57**, 19–34.
- C. K. Kuo and P. X. Ma, *Biomaterials*, 2001, **22**, 511–521.
- J. Berger, M. Reist, J. M. Mayer, O. Felt and R. Gurny, *Eur. J. Pharm. Biopharm.*, 2004, **57**, 35–52.
- H. J. Chung, Y. Lee and T. G. Park, *J. Controlled Release*, 2008, **127**, 22–30.
- P. Gacesa, *Carbohydr. Polym.*, 1988, **8**, 161–182.
- K. Y. Lee and D. J. Mooney, *Prog. Polym. Sci.*, 2012, **37**, 106–126.
- H. Lin, Q. Li, Q. Du, O. Wang, Z. Wang, L. Akert, M. A. Carlson, C. Zhang, A. Subramanian, C. Zhang, M. Lunning, M. Li and Y. Lei, *Biomaterials*, 2019, **189**, 23–36.
- X. Wang, F. Liu, X. Zheng and J. Sun, *Angew. Chem.*, 2011, **50**, 11378–11381.
- J. Ostrowska-Czubenko and M. Gierszewska-Druzynska, *Carbohydr. Polym.*, 2009, **77**, 590–598.

- 52 J. H. Hamman, *Mar. Drugs*, 2010, **8**, 1305–1322.
- 53 F. Ahmadi, Z. Oveisi, S. M. Samani and Z. Amoozgar, *Res. Pharm. Sci.*, 2015, **10**, 1–16.
- 54 B. H. Ye, S. Y. Zhang, R. W. Li, L. H. Li, L. Lu and C. R. Zhou, *Compos. Sci. Technol.*, 2018, **156**, 238–246.
- 55 S. J. Lue, C. H. Chen and C. M. Shih, *J. Macromol. Sci., Part B: Phys.*, 2011, **50**, 563–579.
- 56 M. A. Haq, Y. Su and D. Wang, *Mater. Sci. Eng., C*, 2017, **70**, 842–855.
- 57 S. Chaterji, I. K. Kwon and K. Park, *Prog. Polym. Sci.*, 2007, **32**, 1083–1122.
- 58 H. Park, M. H. Kim, Y. I. Yoon and W. H. Park, *Carbohydr. Polym.*, 2017, **157**, 775–783.
- 59 L. Klouda and A. G. Mikos, *Eur. J. Pharm. Biopharm.*, 2008, **68**, 34–45.
- 60 H. Zhang, S. W. Guo, S. Y. Fu and Y. Zhao, *Polymers*, 2017, **9**, 238.
- 61 W. X. Fu and B. Zhao, *Polym. Chem.*, 2016, **7**, 6980–6991.
- 62 X. Hu, D. Kaplan and P. Cebe, *Macromolecules*, 2006, **39**, 6161–6170.
- 63 J. O. Buitrago, K. D. Patel, A. El-Fiqi, J. H. Lee, B. Kundu, H. H. Lee and H. W. Kim, *Acta Biomater.*, 2018, **69**, 218–233.
- 64 C. M. Hassan and N. A. Peppas, *Macromolecules*, 2000, **33**, 2472–2479.
- 65 X. Jiang, N. Xiang, H. Zhang, Y. Sun, Z. Lin and L. Hou, *Carbohydr. Polym.*, 2018, **186**, 377–383.
- 66 Y. L. Zhang, M. W. Song, Y. F. Diao, B. W. Li, L. Y. Shi and R. Ran, *RSC Adv.*, 2016, **6**, 112468–112476.
- 67 D. W. Lim, S. H. Choi and T. G. Park, *Macromol. Rapid Commun.*, 2000, **21**, 464–471.
- 68 H. L. Mao, C. Wang, X. H. Chang, H. Q. Cao, G. R. Shan, Y. Z. Bao and P. J. Pan, *Mater. Chem. Front.*, 2018, **2**, 313–322.
- 69 H. Hofmeier, R. Hoogenboom, M. E. Wouters and U. S. Schubert, *J. Am. Chem. Soc.*, 2005, **127**, 2913–2921.
- 70 J. Cui and A. del Campo, *Chem. Commun.*, 2012, **48**, 9302–9304.
- 71 M. Guo, L. M. Pitet, H. M. Wyss, M. Vos, P. Y. Dankers and E. W. Meijer, *J. Am. Chem. Soc.*, 2014, **136**, 6969–6977.
- 72 X. Dai, Y. Zhang, L. Gao, T. Bai, W. Wang, Y. Cui and W. Liu, *Adv. Mater.*, 2015, **27**, 3566–3571.
- 73 H. Li, P. Yang, P. Pageni and C. Tang, *Macromol. Rapid Commun.*, 2017, **38**, 1700109.
- 74 U. S. Schubert and C. Eschbaumer, *Angew. Chem.*, 2002, **41**, 2892–2926.
- 75 C. A. Fustin, P. Guillet, U. S. Schubert and J. F. Gohy, *Adv. Mater.*, 2007, **19**, 1665–1673.
- 76 J. Brassinne, F. D. Jochum, C. A. Fustin and J. F. Gohy, *Int. J. Mol. Sci.*, 2015, **16**, 990–1007.
- 77 J. Yuan, X. Fang, L. Zhang, G. Hong, Y. Lin, Q. Zheng, Y. Xu, Y. Ruan, W. Weng, H. Xia and G. Chen, *J. Mater. Chem.*, 2012, **22**, 11515–11522.
- 78 Y. Chujo, K. Sada and T. Saegusa, *Macromolecules*, 1993, **26**, 6315–6319.
- 79 Y. Chujo, K. Sada and T. Saegusa, *Polym. J.*, 1993, **25**, 599.
- 80 M. J. Harrington, A. Masic, N. Holten-Andersen, J. H. Waite and P. Fratzl, *Science*, 2010, **328**, 216–220.
- 81 B. P. Lee, J. L. Dalsin and P. B. Messersmith, *Biomacromolecules*, 2002, **3**, 1038–1047.
- 82 N. Holten-Andersen, M. J. Harrington, H. Birkedal, B. P. Lee, P. B. Messersmith, K. Y. Lee and J. H. Waite, *Proc. Natl. Acad. Sci. U. S. A.*, 2011, **108**, 2651–2655.
- 83 S. Y. Zheng, H. Ding, J. Qian, J. Yin, Z. L. Wu, Y. Song and Q. Zheng, *Macromolecules*, 2016, **49**, 9637–9646.
- 84 X. Ma and Y. Zhao, *Chem. Rev.*, 2015, **115**, 7794–7839.
- 85 H. Yang, B. Yuan, X. Zhang and O. A. Scherman, *Acc. Chem. Res.*, 2014, **47**, 2106–2115.
- 86 H. Frisch and P. Besenius, *Macromol. Rapid Commun.*, 2015, **36**, 346–363.
- 87 Z. Qi and C. A. Schalley, *Acc. Chem. Res.*, 2014, **47**, 2222–2233.
- 88 X. Ma and H. Tian, *Acc. Chem. Res.*, 2014, **47**, 1971–1981.
- 89 K. M. Huh, T. Ooya, W. K. Lee, S. Sasaki, I. C. Kwon, S. Y. Jeong and N. Yui, *Macromolecules*, 2001, **34**, 8657–8662.
- 90 J. Li, A. Harada and M. Kamachi, *Polym. J.*, 1994, **26**, 1019.
- 91 M. Guo, M. Jiang, S. Pispas, W. Yu and C. Zhou, *Macromolecules*, 2008, **41**, 9744–9749.
- 92 J. Li, X. Li, X. Ni, X. Wang, H. Li and K. W. Leong, *Biomaterials*, 2006, **27**, 4132–4140.
- 93 J. Wang, G. S. Williamson and H. Yang, *Colloids Surf., B*, 2018, **165**, 144–149.
- 94 L. Liu, X. Feng, Y. Pei, J. Wang, J. Ding and L. Chen, *Mater. Sci. Eng., C*, 2018, **82**, 25–28.
- 95 S. Li, Y. Xia, Y. Qiu, X. Chen and S. Shi, *J. Appl. Polym. Sci.*, 2018, **135**, 45761.
- 96 A. S. Sawhney, C. P. Pathak and J. A. Hubbell, *Biomaterials*, 1993, **14**, 1008–1016.
- 97 C. R. Nuttelman, M. C. Tripodi and K. S. Anseth, *Matrix Biol.*, 2005, **24**, 208–218.
- 98 Y. An and J. A. Hubbell, *J. Controlled Release*, 2000, **64**, 205–215.
- 99 O. Jeon, C. Powell, L. D. Solorio, M. D. Krebs and E. Alsberg, *J. Controlled Release*, 2011, **154**, 258–266.
- 100 K. T. Nguyen and J. L. West, *Biomaterials*, 2002, **23**, 4307–4314.
- 101 H. Y. Yao, J. Q. Wang and S. L. Mi, *Polymers*, 2018, **10**, 27.
- 102 W. N. E. van Dijk-Wolthuis, O. Franssen, H. Talsma, M. J. van Steenberg, J. J. Kettenes-van den Bosch and W. E. Hennink, *Macromolecules*, 1995, **28**, 6317–6322.
- 103 C. G. Williams, A. N. Malik, T. K. Kim, P. N. Manson and J. H. Elisseeff, *Biomaterials*, 2005, **26**, 1211–1218.
- 104 M. Guvendiren and J. A. Burdick, *Biomaterials*, 2010, **31**, 6511–6518.
- 105 A. Urrios, C. Parra-Cabrera, N. Bhattacharjee, A. M. Gonzalez-Suarez, L. G. Rigat-Brugarolas, U. Nallapatti, J. Samitier, C. A. DeForest, F. Posas, J. L. Garcia-Cordero and A. Folch, *Lab Chip*, 2016, **16**, 2287–2294.
- 106 M. B. Mellott, K. Searcy and M. V. Pishko, *Biomaterials*, 2001, **22**, 929–941.
- 107 H. J. Lee, G. M. Fernandes-Cunha and D. Myung, *React. Funct. Polym.*, 2018, **131**, 29–35.

- 108 J. Hu, Y. Hou, H. Park, B. Choi, S. Hou, A. Chung and M. Lee, *Acta Biomater.*, 2012, **8**, 1730–1738.
- 109 Y. D. Park, N. Tirelli and J. A. Hubbell, *Biomaterials*, 2003, **24**, 893–900.
- 110 S. J. Bryant, C. R. Nuttelman and K. S. Anseth, *J. Biomater. Sci., Polym. Ed.*, 2000, **11**, 439–457.
- 111 W. Zeng, W.-k. Hu, H. Li, Y.-h. Jing, H. Kang, Q. Jiang and C. Zhang, *Chin. J. Polym. Sci.*, 2014, **32**, 1507–1514.
- 112 W. Hu, X. Feng, X. Liu, S. Dai, W. Zeng, Q. Jiang, B. Chen, C. Quan, K. Sun and C. Zhang, *J. Biomater. Sci., Polym. Ed.*, 2016, **27**, 1–13.
- 113 B. Li, L. Wang, F. Xu, X. Gang, U. Demirci, D. Wei, Y. Li, Y. Feng, D. Jia and Y. Zhou, *Acta Biomater.*, 2015, **22**, 59–69.
- 114 J. H. Hoeijmakers, *N. Engl. J. Med.*, 2009, **361**, 1475–1485.
- 115 U. Panich, G. Sittithumcharee, N. Rathviboon and S. Jirawatnotai, *Stem Cells Int.*, 2016, **2016**, 7370642.
- 116 P. Dakup and S. Gaddameedhi, *Photochem. Photobiol.*, 2017, **93**, 296–303.
- 117 H. Shih and C. C. Lin, *Macromol. Rapid Commun.*, 2013, **34**, 269–273.
- 118 H. Lin, D. Zhang, P. G. Alexander, G. Yang, J. Tan, A. W. Cheng and R. S. Tuan, *Biomaterials*, 2013, **34**, 331–339.
- 119 G. M. Fernandes-Cunha, H. J. Lee, A. Kumar, A. Kreymerman, S. Heilshorn and D. Myung, *Biomacromolecules*, 2017, **18**, 3185–3196.
- 120 J. W. Bjork, S. L. Johnson and R. T. Tranquillo, *Biomaterials*, 2011, **32**, 2479–2488.
- 121 E. A. Kamoun, A. Winkel, M. Eisenburger and H. Menzel, *Arabian J. Chem.*, 2016, **9**, 745–754.
- 122 A. E. Rydholm, C. N. Bowman and K. S. Anseth, *Biomaterials*, 2005, **26**, 4495–4506.
- 123 N. B. Cramer and C. N. Bowman, *J. Polym. Sci., Part A: Polym. Chem.*, 2001, **39**, 3311–3319.
- 124 J. Xu and C. Boyer, *Macromolecules*, 2015, **48**, 520–529.
- 125 J. J. Sperinde and L. G. Griffith, *Macromolecules*, 1997, **30**, 5255–5264.
- 126 M. K. McHale, L. A. Setton and A. Chilkoti, *Tissue Eng.*, 2005, **11**, 1768–1779.
- 127 R. Jin, L. S. Moreira Teixeira, P. J. Dijkstra, M. Karperien, C. A. van Blitterswijk, Z. Y. Zhong and J. Feijen, *Biomaterials*, 2009, **30**, 2544–2551.
- 128 M. Kurisawa, J. E. Chung, Y. Y. Yang, S. J. Gao and H. Uyama, *Chem. Commun.*, 2005, 4312–4314, DOI: 10.1039/b506989k.
- 129 R. Jin, C. Hiemstra, Z. Zhong and J. Feijen, *Biomaterials*, 2007, **28**, 2791–2800.
- 130 R. Jin, L. S. Moreira Teixeira, P. J. Dijkstra, C. A. van Blitterswijk, M. Karperien and J. Feijen, *J. Controlled Release*, 2011, **152**, 186–195.
- 131 K. C. Kuo, R. Z. Lin, H. W. Tien, P. Y. Wu, Y. C. Li, J. M. Melero-Martin and Y. C. Chen, *Acta Biomater.*, 2015, **27**, 151–166.
- 132 S. Yigit, R. Sanyal and A. Sanyal, *Chem. – Asian J.*, 2011, **6**, 2648–2659.
- 133 F. Yu, X. Cao, J. Du, G. Wang and X. Chen, *ACS Appl. Mater. Interfaces*, 2015, **7**, 24023–24031.
- 134 L. J. Smith, S. M. Taimoory, R. Y. Tam, A. E. G. Baker, N. Binth Mohammad, J. F. Trant and M. S. Shoichet, *Biomacromolecules*, 2018, **19**, 926–935.
- 135 N. L. Morozowich, J. L. Nichol and H. R. Allcock, *J. Polym. Sci., Polym. Chem.*, 2016, **54**, 2984–2991.
- 136 Y. Zhang, L. Tao, S. Li and Y. Wei, *Biomacromolecules*, 2011, **12**, 2894–2901.
- 137 F. Ding, S. Wu, S. Wang, Y. Xiong, Y. Li, B. Li, H. Deng, Y. Du, L. Xiao and X. Shi, *Soft Matter*, 2015, **11**, 3971–3976.
- 138 X. Ma, T. Xu, W. Chen, H. Qin, B. Chi and Z. Ye, *Carbohydr. Polym.*, 2018, **179**, 100–109.
- 139 X. Sui, L. van Ingen, M. A. Hempenius and G. J. Vancso, *Macromol. Rapid Commun.*, 2010, **31**, 2059–2063.
- 140 S. C. Rizzi and J. A. Hubbell, *Biomacromolecules*, 2005, **6**, 1226–1238.
- 141 Y. Wen, F. Li, C. G. Li, Y. J. Yin and J. J. Li, *J. Mater. Chem. B*, 2017, **5**, 961–971.
- 142 Z. Q. Liu, Z. Wei, X. L. Zhu, G. Y. Huang, F. Xu, J. H. Yang, Y. Osada, M. Zrinyi, J. H. Li and Y. M. Chen, *Colloids Surf., B*, 2015, **128**, 140–148.
- 143 R. J. Wojtecki, M. A. Meador and S. J. Rowan, *Nat. Mater.*, 2011, **10**, 14–27.
- 144 C. C. Deng, W. L. A. Brooks, K. A. Abboud and B. S. Sumerlin, *ACS Macro Lett.*, 2015, **4**, 220–224.
- 145 M. E. Smithmyer, C. C. Deng, S. E. Cassel, P. J. LeValley, B. S. Sumerlin and A. M. Kloxin, *ACS Macro Lett.*, 2018, **7**, 1105–1110.
- 146 Z. Huang, P. Delparastan, P. Burch, J. Cheng, Y. Cao and P. B. Messersmith, *Biomater. Sci.*, 2018, **6**, 2487–2495.
- 147 Y. J. Chen, D. Diaz-Dussan, D. Wu, W. D. Wang, Y. Y. Peng, A. B. Asha, D. G. Hall, K. Ishihara and R. Narain, *ACS Macro Lett.*, 2018, **7**, 904–908.
- 148 S. H. Hong, S. Kim, J. P. Park, M. Shin, K. Kim, J. H. Ryu and H. Lee, *Biomacromolecules*, 2018, **19**, 2053–2061.
- 149 C. Seidler, D. Y. W. Ng and T. Weil, *Tetrahedron*, 2017, **73**, 4979–4987.
- 150 T. J. Long, Y. X. Li, X. Fang and J. Q. Sun, *Adv. Funct. Mater.*, 2018, **28**, 9.
- 151 R. Wang, Q. Li, B. Chi, X. Wang, Z. Xu, Z. Xu, S. Chen and H. Xu, *Chem. Commun.*, 2017, **53**, 4803–4806.
- 152 M. T. I. Mredha, Y. Z. Guo, T. Nonoyama, T. Nakajima, T. Kurokawa and J. P. Gong, *Adv. Mater.*, 2018, **30**, 1704937.
- 153 K. Kim, J. Lam, S. Lu, P. P. Spicer, A. Lueckgen, Y. Tabata, M. E. Wong, J. A. Jansen, A. G. Mikos and F. K. Kasper, *J. Controlled Release*, 2013, **168**, 166–178.
- 154 S. Lu, J. Lam, J. E. Trachtenberg, E. J. Lee, H. Seyednejad, J. van den Beucken, Y. Tabata, M. E. Wong, J. A. Jansen, A. G. Mikos and F. K. Kasper, *Biomaterials*, 2014, **35**, 8829–8839.
- 155 N. Joshi, J. Yan, S. Levy, S. Bhagchandani, K. V. Slaughter, N. E. Sherman, J. Amirault, Y. Wang, L. Riegel, X. He, T. S. Rui, M. Valic, P. K. Vemula, O. R. Miranda, O. Levy, E. M. Gravallesse, A. O. Aliprantis, J. Ermann and J. M. Karp, *Nat. Commun.*, 2018, **9**, 1275.