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Enhancing the stability and porosity of penetrated metal–organic frameworks through the insertion of coordination sites†

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Guided by the insertion of coordination sites within ligands, an interpenetrated metal–organic framework NKU-112 and a self-penetrated framework NKU-113 were obtained. The two MOFs have similar cage-based framework structures, while NKU-113 reveals enhanced porosity and stability compared with NKU-112, owing to the self-penetrated structure induced by the additional chelating bipyridine moiety in the ligand. To the best of our knowledge, this is the first study that attempts to shift the structure topology of a MOF from interpenetrated to self-penetrated while achieving a delicate modulation of the location and distances within the penetrated structure by inserting new coordination sites.

Introduction

Metal–organic frameworks (MOFs) composed of metal ions/clusters connected by organic linkers have emerged as a class of attractive porous materials.¹ Owing to their tailorable porous structures, MOFs have shown great potential in various applications such as gas storage and/or separation, catalysis, sensing, *etc.*² Though significant progress has been made in the structure and property modulation of MOFs, the practical applications of MOFs have been limited by their relatively low stability.³ Although several strategies have been proposed to enhance the stability of MOFs, including ligand or metal ion changes, surface modification, interpenetration and the construction of multi-walls,⁴ the development of a facile and straightforward design and construction strategy for stable MOFs is still a desired research goal.⁵

Framework interpenetration frequently occurs in MOF structures, particularly when extended organic ligands are used for MOF construction. In spite of the reduction of the pore volume of the framework, framework interpenetration in MOFs has been observed to not only enhance MOF stability but also regulate the pore size, thus augmenting their gas sorption

properties originating from the promoted interactions between the individual networks. For example, Zhou *et al.* have compared the H₂ sorption performances of non-interpenetrated and two-fold interpenetrated MOFs to find that framework interpenetration could benefit the stability of the framework and gas sorption at low pressure.⁶ However, the enhancement of framework stability by interpenetration is somewhat unmanageable, since the van der Waals interactions between the organic ligands of the individual networks may not be strong enough to prevent structure deformation and framework slippage in response to the removal of guest species in the framework. On the other hand, the coordination geometry of secondary building units (SBUs) composed of metal centers with coordinated solvent molecules may vary in response to the removal of solvents, which could also affect the stability of the MOFs.

Focusing on the stability enhancement of MOFs, the combination of framework penetration and the stabilization of SBUs could be a rational strategy. In principle, if the coordination sphere of the metal center is broadened and additional metal–ligand bonds are introduced to increase the liganacy of the metal cluster and to enhance the strength of the interactions between the individual networks, the resulting self-penetrated framework may become more stable than the original interpenetrated framework. However, realizing the above-mentioned scenario remains to be a challenging task, mainly because of the mismatched distance between frameworks as well as unfavorable coordination environments for additional metal ions.

In this study, one such rare example is reported. By inserting additional coordinating sites (here through the introduction of bipyridine groups) in the backbone of the organic ligand, a two-fold interpenetrated framework [Ni₂L1(μ₂-H₂O)(H₂O)₂(DMF)₂](solvents)_n (denoted as NKU-112, NKU-Nankai University,

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DMF = *N,N'*-dimethylformamide, $H_4L1 = 5,5'$ -((thiophene-2,5-dicarbonyl)bis(azanediyl))diisophthalic acid, shown in Fig. S1†) can be turned into a self-interpenetrated framework $[Co_2L2(\mu_2-H_2O)(H_2O)_2] \cdot (\text{solvents})_n$ (denoted as NKU-113, $H_4L2 = 5,5'$ -([2,2'-bipyridine]-5,5'-dicarbonyl)bis(azanediyl) diisophthalic acid, Fig. S1†), in which the extent of the mutual framework interaction is significantly increased in comparison with the two-fold interpenetrated MOF (Fig. 1). As a result, NKU-113 displays enhanced porosity and stability with respect to NKU-112, although they have similar cage-based packing structures (Scheme 1). This work provides a unique strategy for the enhancement of coordination interactions for increasing the stability and porosity of penetrated MOFs.

Results and discussion

Structures of NKU-112 and NKU-113

A solvothermal reaction of $Ni(NO_3)_2 \cdot 6H_2O$ and H_4L1 in DMF and acetonitrile affords crystals of NKU-112, whereas crystals of NKU-113 are obtained from the solvothermal reaction of $Co(NO_3)_2 \cdot 6H_2O$ and H_4L2 in DMF, acetonitrile and H_2O . The structures of NKU-112 and NKU-113 were determined using single-crystal X-ray diffractometry. The bulk samples of NKU-112 and NKU-113 were characterized using IR (Fig. S2†), and their phase purities were verified by the well matched powder X-ray diffraction (PXRD) patterns of the as-synthesized samples and the simulated ones (Fig. 2). Thermogravimetric (TG)

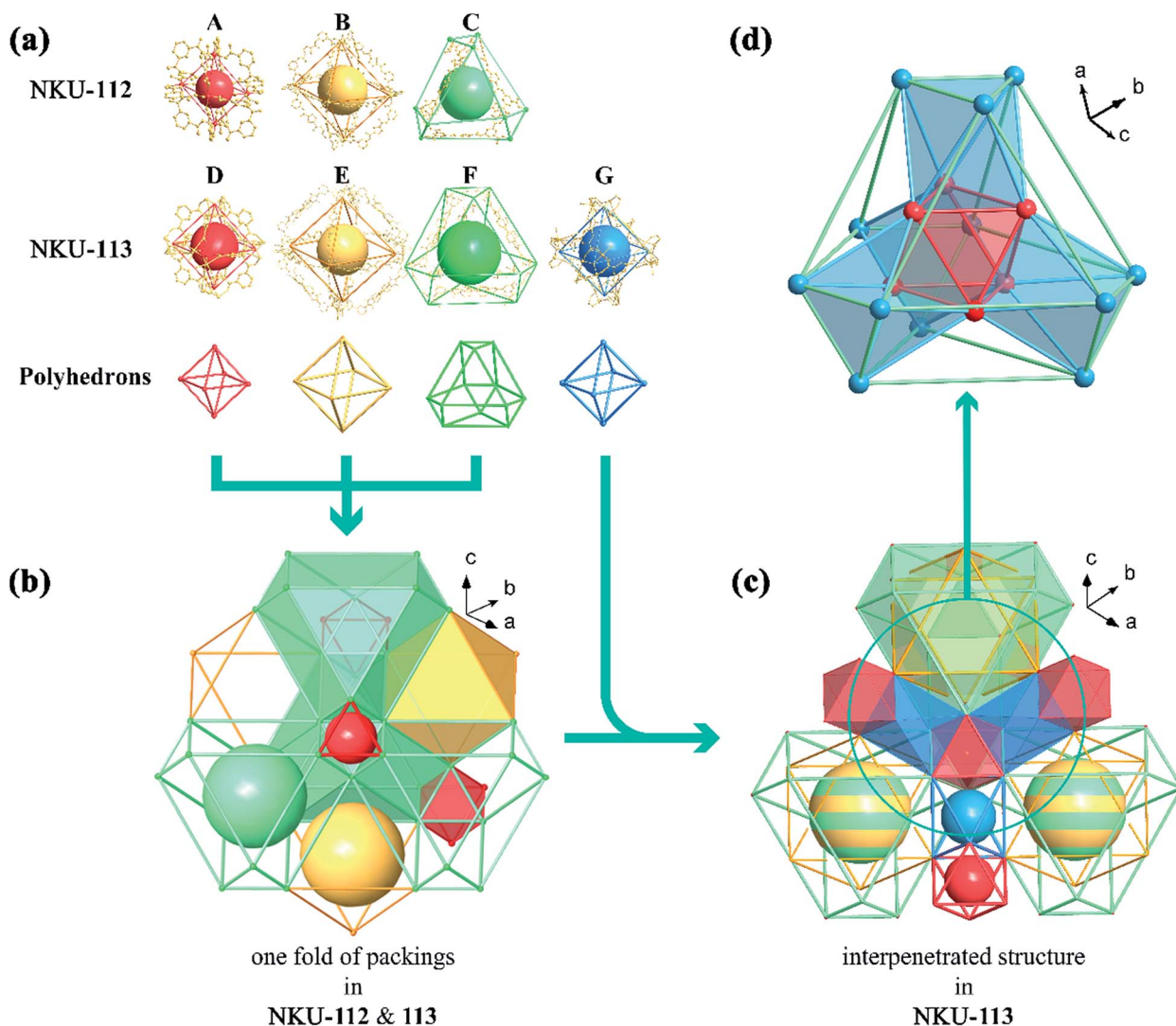
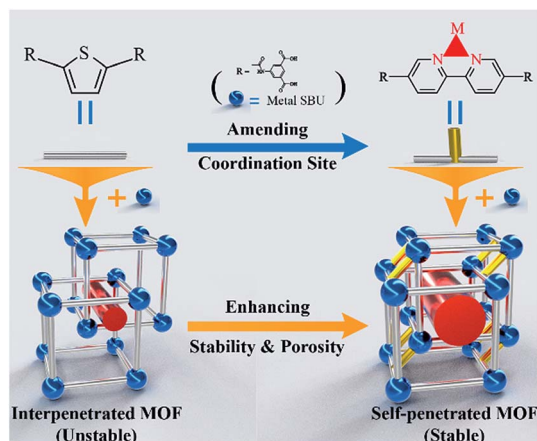


Fig. 1 Structure diagrams of cages (a), one-fold packing cages in NKU-112 and 113 (b), interpenetrated two-fold cages in NKU-113 (c) and nestification relationships in NKU-113 (d). (a) The octahedron combined with six $[Ni_2(CO_2)_4(\mu_2-H_2O)(H_2O)_2DMF_2]$ clusters and twelve isophthalic moieties in NKU-112 (cage A); another octahedron combined with six Ni clusters and twelve $L1^{4-}$ (cage B); a distorted cuboctahedron combined with six $L1^{4-}$ and twelve Ni clusters (cage C); an octahedron combined with six $[Co_2(CO_2)_4(\mu_2-H_2O)(H_2O)_2]$ clusters and twelve isophthalic acids in NKU-113 (cage D); an octahedron combined with six Co clusters and twelve $L2^{4-}$ (cage E); a distorted cuboctahedron combined with six $L2^{4-}$ and twelve Co clusters (cage F); an octahedron combined with six Co clusters and six $L2^{4-}$ (cage G). (b) The packing structure of the three kinds of cages in NKU-112. (c) The packing structure of the four kinds of cages in NKU-113. (d) A diagram of the nestification of cages D, F, and G in NKU-113.





Scheme 1 Framework and modification strategy diagram. The blue-colored balls and the silver-colored sticks represent metal secondary building units and ligands, respectively. The yellow-colored sticks represent inserted coordinate bonds, which covalently connect two sets of frameworks to each other. The red-colored cylinders represent pores, which are enlarged after ligand modification.

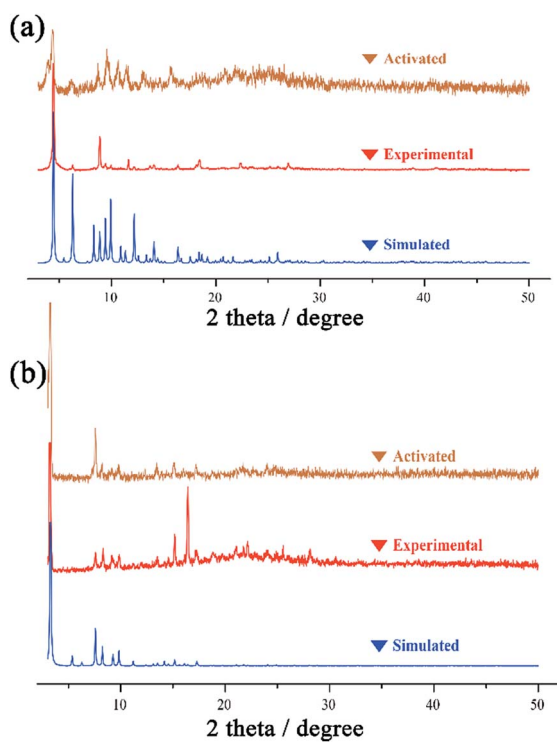


Fig. 2 The PXRD patterns of simulated, experimental and activated samples of NKU-112 (a) and NKU-113 (b).

analyses showed that NKU-112 can retain its original structure up to a temperature of 370 °C, whereas NKU-113 can retain its structure up to 400 °C (Fig. S3†).

Single crystal X-ray diffraction reveals that NKU-112 crystallises in the cubic space group $Ia\bar{3}$. Apart from the guest molecules, the asymmetric unit contains two Ni^{2+} cations, one L^{4-} anion, three water and two DMF molecules. The Ni1 is six

coordinated in a distorted octahedron geometry with four carboxylate oxygen atoms (O1, O15, O22 and O24) from four different organic ligands, one oxygen atom (O3) from a terminal water molecule and one oxygen atom (O13) from a μ_2 -water molecule. In contrast, the coordination sphere of Ni2 includes two carboxylate oxygen atoms (O23 and O25), two oxygen atoms from terminal DMF molecules (O9 and O29), one oxygen atom from a terminal water molecule (O8) and one oxygen atom from a μ_2 -water molecule (O13), which can also be described as a distorted octahedral geometry (Fig. S4†). Ni1 and Ni2 are bridged by two carboxylate groups and one μ_2 -H₂O molecule to form a discrete $[\text{Ni}_2(\text{COO})_4(\mu_2\text{-H}_2\text{O})(\text{H}_2\text{O})_2(\text{DMF})_2]$ cluster in which a water molecule is inserted between the Ni^{2+} ions, and the Ni1–O13–Ni2 angle is close to 120° (117.463° to be precise), a completely different value to that of a classical $[\text{M}_2(\text{COO})_4\text{O}_2]$ SBU (Fig. S5†). The SBUs were further expanded by the organic linker to form a three-dimensional (3D) network. Careful examination of the network structure reveals that the framework is composed of three different types of cage. As shown in Fig. 1, the octahedral cage **A**, with a diameter of ca. 6.8 Å, is defined by six Ni clusters, each being at a vertex of the octahedron, and twelve isophthalate moieties along the edges. The other octahedral cage **B**, with a diameter of ca. 21.6 Å, is composed of six Ni clusters and twelve organic ligands, whilst the third type of cage, the distorted cuboctahedral cage **C** with a diameter of ca. 15.8 Å, consists of twelve Ni clusters and six organic ligands (Fig. 1b). As for the packing of NKU-112, each distorted cuboctahedral **C** cage is surrounded by four **A** cages and four **B** cages. Cage **C** shares three Ni clusters and one isophthalic moiety on the truncated surface with cage **A**, and three Ni clusters and L^{4-} with cage **B** (Fig. 3a). Topologically, since cage **A** occupies vertices of cage **B** and **C**, cage **A** can be regarded as nodes for clarity and the framework features a uninodal 12-connected net, which is classified as *fcu* as determined using TOPOS software (Fig. S6a†).⁷ Due to the large void in the network, two-fold interpenetration occurs in NKU-112, and the solvent-accessible volume of the interpenetrated NKU-112 is estimated to be 45.7% per unit cell by PLATON.⁸

The transformation between interpenetrated and self-penetrated MOFs has been studied in the past.⁹ After specific connecting components are added, two or more folds of interpenetrated frameworks are covalently linked to each other to form a single framework. If these components are removed, the framework can be restored to its original topology. The self-penetrated structure can also be constructed using elaborate ligand design, and the key is finding a suitable ligand. Typically, as two major coordination groups, carboxylate groups and N-containing heterocyclic groups are employed in the majority of MOF constructions.¹⁰ Among the various groups, a chelating bipyridine moiety could form a relatively stable coordination structure with a metal center to stabilize the SBU. Accordingly, a bipyridine group was selected to be inserted in the backbone of the organic ligand $\text{H}_4\text{L1}$, and the resulting $\text{H}_4\text{L2}$ ligand was used to construct NKU-113.

Single-crystal X-ray diffraction reveals that NKU-113 crystallises in the cubic space group $Fd\bar{3}m$. The asymmetric unit comprises two Co^{2+} cations, one L^{4-} anion, and three water



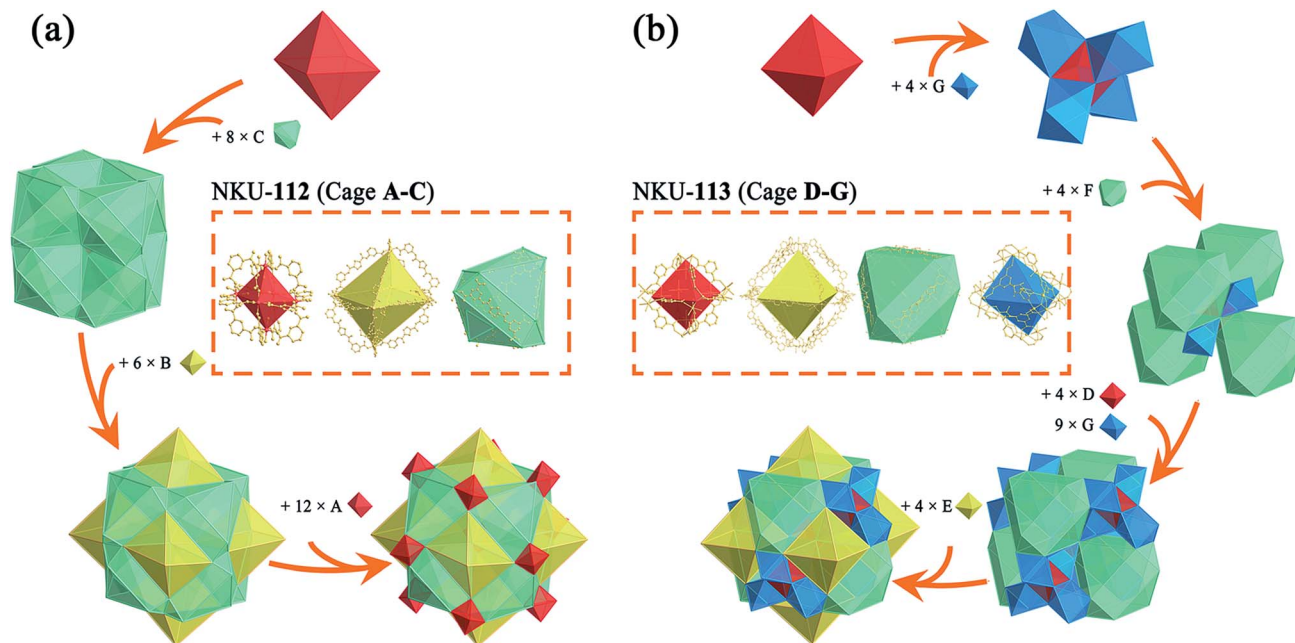


Fig. 3 Structure combination diagrams of the cages presented in NKU-112 (a) and NKU-113 (b). Inset: coordination structures of cages A-G.

molecules. Co1 and Co2 are connected to each other through a bridge comprising two carboxylate groups from $L2^{4-}$ and a μ_2 -water molecule. Co1 is coordinated by four $L2^{4-}$ anions and two

water molecules in a distorted trigonal bipyramid defined by O4, O5 and four O1 atoms from four $L2^{4-}$ anions. Co2 is coordinated by two carboxyls from $L2^{4-}$, two nitrogen atoms from

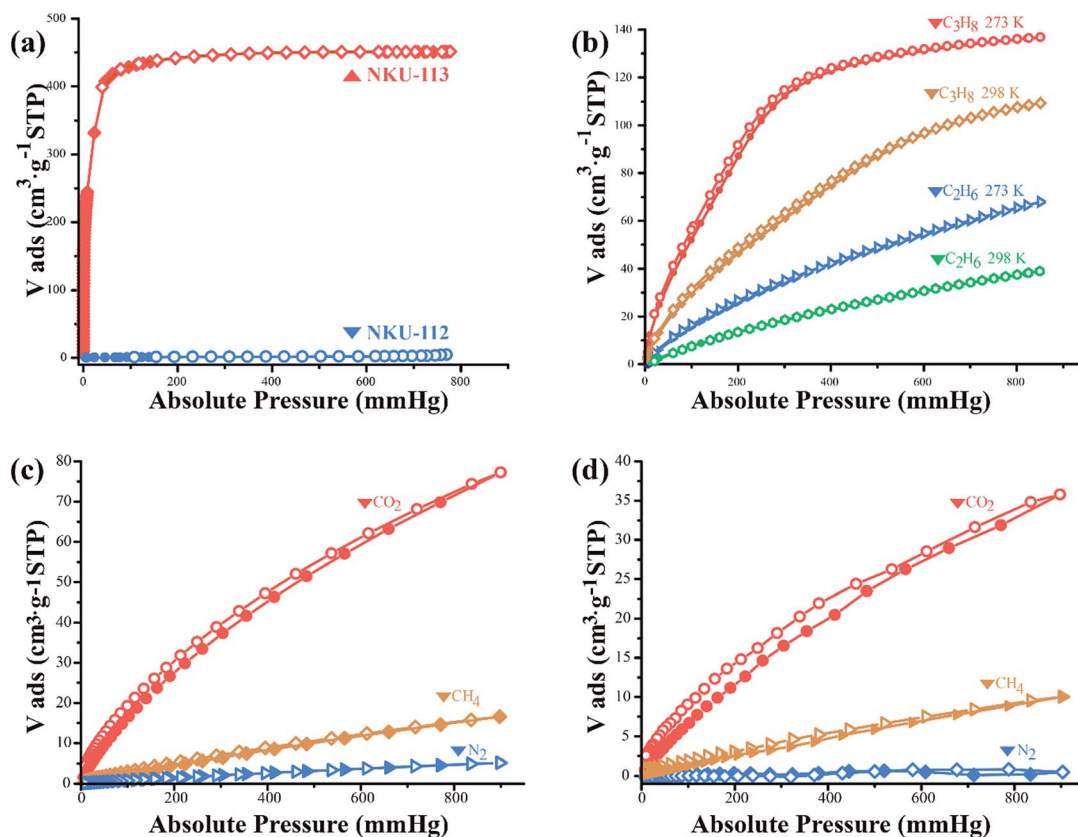


Fig. 4 Gas sorption isotherms. (a) N_2 isotherms of NKU-112 and NKU-113 at 77 K. (b) C_3H_8 and C_2H_6 isotherms of NKU-113 at 273 K and 298 K. (c and d) CO_2 , CH_4 , and N_2 isotherms of NKU-113 at 273 K (c) and at 298 K (d). The filled and open symbols represent the adsorption and desorption data, respectively.

$L2^{4-}$ and two water molecules that together form another distorted trigonal bipyramid defined by two O2 atoms, two N1 atoms, O5 and O6 (Fig. S2†). These two metal ions are coordinated by the nitrogen atoms of $L2^{4-}$, featuring a different scenario to that of the Ni cluster in NKU-112 (Fig. S5†). The ligand in NKU-113 has two-fold disorder with an occupancy of 0.5 each, and Co2 has four-fold disorder with an occupancy of 0.25 each. There are four kinds of cages in NKU-113: the octahedron cage **D**, with a diameter of *ca.* 10.6 Å, is defined by six Co clusters on the vertices and twelve isophthalic moieties in $L2^{4-}$; octahedral cage **E**, with a diameter of *ca.* 28.6 Å, is defined by six Co clusters and twelve $L2^{4-}$ ligands; distorted cuboctahedral cage **F** features four triangles combined with three Co clusters on the cross-section and six $L2^{4-}$ on the edge, and distorted octahedron cage **G** with a diameter of *ca.* 22.0 Å, which has no corresponding structure in NKU-112, defined by six Co clusters and six $L2^{4-}$ ligands (Fig. 1a and 3b). As for the packing mode of NKU-113, each distorted cuboctahedral cage **F** is surrounded by four octahedral **D** cages by sharing three Co clusters and three isophthalic moieties on the surface; it is additionally surrounded by twelve distorted octahedral **G** cages by sharing two Co clusters and half of an $L2^{4-}$ ligand on the edge of the cross-section. Half of the **F** cages wrap one octahedral **E** cage, with the other half of the **F** cages wrapping four **G** cages and a **D** cage (Fig. 1d and S7†). Cages **D** and **E** & **F** possess two kinds of pores: a smaller pore in the range of 0.5–0.9 nm in cage **D**, and a larger pore in the range of 1.9–4.5 nm in cage **E** & **F**. Benefiting from the presence of the coordinating bipyridine moiety in the backbone of the $L2^{4-}$ ligand, the pyridine group bonds to a different cluster in the framework of NKU-113, resulting in the formation of a self-penetrated structure. Accordingly, the tiling structure of the framework of NKU-113 also changed with respect to that of NKU-112 because of its self-support property (Fig. S6b†). Topologically, by regarding cage **D** as nodes, the framework of NKU-113 could be simplified as a 3D binodal (3,18)-connected net with a point symbol of $(4^2 \cdot 6)_6(4^{60} \cdot 6^{93})$, as determined by the TOPOS software. The solvent-accessible volume of NKU-113 is estimated by PLATON to be 67.2% per unit cell.

Compared with NKU-112, the structure deployment in NKU-113 causes several changes. The SBU in NKU-112, with a formula of $[Ni_2(COO)_4(\mu_2-H_2O)(H_2O)_2DMF_2]$, is similar to that of $[Co_2(COO)_4(\mu_2-H_2O)(H_2O)_2]$ in NKU-113 except for the fact that in the latter cluster the chelating bipyridine moiety in the $L2^{4-}$ ligand replaces the coordinating DMF molecules (Fig. S5†). Furthermore, because of the additional coordination of the bipyridine moiety to the metal cluster, NKU-113 features an additional octahedral cage composed of six $L2^{4-}$ ligands and six metal clusters (Fig. 1a) with respect to NKU-112. It is known that the coordination of solvent molecules will diminish the rigidity and stability of SBUs, and therefore replacing these coordinated solvent molecules with coordinate bonds between the SBU and the framework can strengthen the overall structure.

In detail, if Co–N bonds in NKU-113 are ignored, the packing mode of the two frameworks is different from that in NKU-112. By regarding cage **A** and cage **D** as nodes and other ligands as linkers, one structure can be simplified into an fcc-like

framework (Fig. S8†). Cages **A** & **D** construct tetrahedral and octahedral voids. In NKU-112, the second framework occupies the centres of larger octahedral voids. In contrast, the second framework occupies the centers of smaller tetrahedral voids in NKU-113 (Fig. S9 and S10†). This structure change is induced by the presence of new coordination bonds since the smaller tetrahedral voids are suitable for the linking of the penetrated framework through Co–N bonds. According to the changes in packing mode, the nestification mode in NKU-113 has also varied (Fig. S11 and S12†). The presence of additional coordinate bonds causes a reduction of the distance between the two structures, which contributes to the increase in porosity and stability observed in NKU-113 with respect to NKU-112. In summary, although NKU-112 and NKU-113 have similar building blocks, the insertion of linkers in the latter is associated with obvious differences that lead to an enhancement in the structure stability and porosity.

Stabilities and adsorption properties of NKU-112 and NKU-113

To evaluate the porosity and gas storage/separation potential of NKU-112 and NKU-113, gas adsorption experiments have been performed. Before the measurements, the samples of NKU-112 and NKU-113 were soaked in ethanol for solvent exchange and supercritically dried using carbon dioxide. However, the framework of NKU-112 partially collapsed after the activation procedures (Fig. 2a), and NKU-113 showed a well retained framework structure (Fig. 2b). It should be noted that the collapse of NKU-112 can be ascribed to the loss of coordinated solvents during activation in the SBU as discussed above.

N_2 sorption tests were first conducted at 77 K to characterize the porosity of the materials (Fig. 4a). The N_2 adsorption isotherm of NKU-113 has the typical characteristics of a type I profile, and the Brunauer–Emmett–Teller (BET) surface area and Langmuir surface area are $1486 \text{ m}^2 \text{ g}^{-1}$ and $1966 \text{ m}^2 \text{ g}^{-1}$, respectively. The pore distribution analysis performed using the Horvath–Kawazoe (H–K) method shows a main distribution of 0.7–1.2 nm and 1.9–4.2 nm (Fig. S13†), indicating that two kinds of pore existed in the framework, which is consistent with the crystal structures.

Considering the highly porous framework of NKU-113 and the presence of amide groups which may benefit its gas sorption performance, a series of sorption tests were performed with C_2H_6 , C_3H_8 , CO_2 , and CH_4 (Fig. 4b–d). The gas uptake of NKU-113 at 273 K is $16 \text{ cm}^3 \text{ g}^{-1}$ (STP) for CH_4 , $63 \text{ cm}^3 \text{ g}^{-1}$ (STP) for C_2H_6 , $135 \text{ cm}^3 \text{ g}^{-1}$ (STP) for C_3H_8 and $77 \text{ cm}^3 \text{ g}^{-1}$ (STP) for CO_2 . At 298 K, the framework shows gas uptakes of $10 \text{ cm}^3 \text{ g}^{-1}$ (STP) for CH_4 , $36 \text{ cm}^3 \text{ g}^{-1}$ (STP) for C_2H_6 , $105 \text{ cm}^3 \text{ g}^{-1}$ (STP) for C_3H_8 , and $36 \text{ cm}^3 \text{ g}^{-1}$ (STP) for CO_2 . On the basis of the investigation of capacities, the heat of sorption of different gases was also investigated (Fig. S14†). The initial heat of sorption values are 15.4 kJ mol^{-1} for CH_4 , 28.2 kJ mol^{-1} for C_2H_6 , 27.2 kJ mol^{-1} for C_3H_8 and 30.4 kJ mol^{-1} for CO_2 . The considerable uptakes and heat of adsorption values of NKU-113 towards alkanes and CO_2 suggest its potential in gas storage and separation applications.



Conclusions

In summary, through the tuning of coordination sites in organic ligands, interpenetrated MOF NKU-112 and self-penetrated MOF NKU-113 have been constructed. Owing to the additional chelating bipyridine coordination sites introduced through the ligand, the NKU-113, featuring a self-penetrated framework, reveals enhanced stability and porosity compared with those of the interpenetrated framework of NKU-112. The approach reported herein may provide a valuable method for the stability and gas sorption performance enhancement of penetrated MOFs.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

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References

- (a) L. B. Sun, J. R. Li, W. Lu, Z. Y. Gu, Z. Luo and H. C. Zhou, *J. Am. Chem. Soc.*, 2012, **134**, 15923–15928; (b) H. Li, M. Eddaoudi, M. O'Keeffe and O. M. Yaghi, *Nature*, 1999, **402**, 276–279; (c) D. S. Zhang, Z. Chang, Y. F. Li, Z. Y. Jiang, Z. H. Xuan, Y. H. Zhang, J. R. Li, Q. Chen, T. L. Hu and X.-H. Bu, *Sci. Rep.*, 2013, **3**, 3312; (d) J. R. Li, D. J. Timmons and H. C. Zhou, *J. Am. Chem. Soc.*, 2009, **131**, 6368–6369; (e) Q. Gao, J. Xu, D. Cao, Z. Chang and X.-H. Bu, *Angew. Chem., Int. Ed.*, 2016, **55**, 15027–15030; (f) F. Bu, Q. Lin, Q. Zhai, L. Wang, T. Wu, S. T. Zheng, X. Bu and P. Feng, *Angew. Chem., Int. Ed.*, 2012, **51**, 8538–8541; (g) L. Li, R.-B. Lin, R. Krishna, X. Wang, B. Li, H. Wu, J. Li, W. Zhou and B. Chen, *J. Am. Chem. Soc.*, 2017, **139**, 7733–7736; (h) X. Cui, K. Chen, H. Xing, Q. Yang, R. Krishna, Z. Bao, H. Wu, W. Zhou, X. Dong, Y. Han, B. Li, Q. Ren, M. J. Zaworotko and B. Chen, *Science*, 2016, **353**, 141–144; (i) M. Bosch, S. Yuan, W. Rutledge and H.-C. Zhou, *Acc. Chem. Res.*, 2017, **50**, 857–865.
- (a) G. Barin, G. W. Peterson, V. Crocellà, J. Xu, K. A. Colwell, A. Nandy, J. A. Reimer, S. Bordiga and J. R. Long, *Chem. Sci.*, 2017, **8**, 4399–4409; (b) P. Z. Moghadam, J. F. Ivy, R. K. Arvapally, A. M. dos Santos, J. C. Pearson, L. Zhang, E. Tylanakis, P. Ghosh, I. W. H. Oswald, U. Kaipa, X. Wang, A. K. Wilson, R. Q. Snurr and M. A. Omary, *Chem. Sci.*, 2017, **8**, 3989–4000; (c) K. Manna, T. Zhang, F. X. Greene and W. B. Lin, *J. Am. Chem. Soc.*, 2015, **137**, 2665–2673; (d) Y. Wang, H. Cui, Z.-W. Wei, H.-P. Wang, L. Zhang and C.-Y. Su, *Chem. Sci.*, 2017, **8**, 775–780; (e) J. Liu, P. K. Thallapally, B. P. McGrail, D. R. Brown and J. Liu, *Chem. Soc. Rev.*, 2012, **41**, 2308–2322; (f) B. Chen, S. Xiang and G. Qian, *Acc. Chem. Res.*, 2010, **43**, 1115–1124; (g) B. Aguila, Q. Sun, J. A. Perman, L. D. Earl, C. W. Abney, R. Elzein, R. Schlaf and S. Ma, *Adv. Mater.*, 2017, **29**, 1700665; (h) J. Park, D. Feng, S. Yuan and H.-C. Zhou, *Angew. Chem., Int. Ed.*, 2015, **54**, 430–435; (i) W.-Y. Pei, G. Xu, J. Yang, H. Wu, B. Chen, W. Zhou and J.-F. Ma, *J. Am. Chem. Soc.*, 2017, **139**, 7648–7656.
- (a) Z. Chang, D. H. Yang, J. Xu, T. L. Hu and X.-H. Bu, *Adv. Mater.*, 2015, **27**, 5432–5441; (b) D. Ma, Y. Li and Z. Li, *Chem. Commun.*, 2011, **47**, 7377–7379; (c) H. Babaei, A. J. H. McGaughey and C. E. Wilmer, *Chem. Sci.*, 2017, **8**, 583–589.
- (a) X. Wang, W.-Y. Gao, J. Luan, L. Wojtas and S. Ma, *Chem. Commun.*, 2016, **52**, 1971–1974; (b) T. Wu, L. Shen, M. Luebbbers, C. Hu, Q. Chen, Z. Ni and R. I. Masel, *Chem. Commun.*, 2010, **46**, 6120–6122; (c) V. Colombo, S. Galli, H. J. Choi, G. D. Han, A. Maspero, G. Palmisano, N. Masciocchi and J. R. Long, *Chem. Sci.*, 2011, **2**, 1311–1319; (d) D. Tian, Q. Chen, Y. Li, Y. H. Zhang, Z. Chang and X. H. Bu, *Angew. Chem., Int. Ed.*, 2014, **53**, 837–841; (e) G. Chang, B. Li, H. Wang, T. Hu, Z. Bao and B. Chen, *Chem. Commun.*, 2016, **52**, 3494–3496; (f) A. Kumar, D. G. Madden, M. Lusi, K.-J. Chen, E. A. Daniels, T. Curtin, J. J. Perry and M. J. Zaworotko, *Angew. Chem., Int. Ed.*, 2015, **54**, 14372–14377; (g) K. Tan, S. Zuluaga, E. Fuentes, E. C. Mattson, J.-F. Veyan, H. Wang, J. Li, T. Thonhauser and Y. J. Chabal, *Nat. Commun.*, 2016, **7**, 13871.
- (a) D. Banerjee, H. Wang, Q. Gong, A. M. Plonka, J. Jagiello, H. Wu, W. R. Woerner, T. J. Emge, D. H. Olson, J. B. Parise and J. Li, *Chem. Sci.*, 2016, **7**, 759–765; (b) S. Yuan, Y.-P. Chen, J.-S. Qin, W. Lu, L. Zou, Q. Zhang, X. Wang, X. Sun and H.-C. Zhou, *J. Am. Chem. Soc.*, 2016, **138**, 8912–8919; (c) Y. Chen, L. Wojtas, S. Ma, M. J. Zaworotko and Z. Zhang, *Chem. Commun.*, 2017, **53**, 8866–8869; (d) K.-J. Chen, D. G. Madden, T. Pham, K. A. Forrest, A. Kumar, Q.-Y. Yang, W. Xue, B. Space, J. J. Perry, J.-P. Zhang, X.-M. Chen and M. J. Zaworotko, *Angew. Chem., Int. Ed.*, 2016, **55**, 10268–10272; (e) X.-Z. Song, C. Qin, W. Guan, S.-Y. Song and H.-J. Zhang, *New J. Chem.*, 2012, **36**, 877–882.
- S. Ma, X.-S. Wang, D. Yuan and H.-C. Zhou, *Angew. Chem., Int. Ed.*, 2008, **47**, 4130–4133.
- V. A. Blatov, A. P. Shevchenko and D. M. Proserpio, *Cryst. Growth Des.*, 2014, **14**, 3576–3586.
- L. J. Farrugia, *J. Appl. Crystallogr.*, 2012, **45**, 849–854.
- (a) Q. Chen, Z. Chang, W. C. Song, H. Song, H. B. Song, T. L. Hu and X.-H. Bu, *Angew. Chem., Int. Ed.*, 2013, **52**, 11550–11553; (b) P. Shen, W.-W. He, D.-Y. Du, H.-L. Jiang, S.-L. Li, Z.-L. Lang, Z.-M. Su, Q. Fu and Y.-Q. Lan, *Chem. Sci.*, 2014, **5**, 1368–1374.
- (a) J. R. Li, A. A. Yakovenko, W. Lu, D. J. Timmons, W. Zhuang, D. Yuan and H. C. Zhou, *J. Am. Chem. Soc.*, 2010, **132**, 17599–17610; (b) S. T. Zheng, T. Wu, B. Irfanoglu, F. Zuo, P. Feng and X. Bu, *Angew. Chem., Int. Ed.*, 2011, **50**, 8034–8037; (c) T. F. Liu, L. Zou, D. Feng, Y. P. Chen, S. Fordham, X. Wang, Y. Liu and H. C. Zhou, *J. Am. Chem. Soc.*, 2014, **136**, 7813–7816.

