

Cite this: *RSC Adv.*, 2018, 8, 39115

# Solution-processed amorphous ZrO<sub>2</sub> gate dielectric films synthesized by a non-hydrolytic sol–gel route†

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Solution-processed zirconium oxide (ZrO<sub>2</sub>) dielectrics were formed *via* a non-hydrolytic sol–gel route at low-temperature, and are suitable for flexible thin film transistor (TFT) devices. Precursor solutions with equimolar zirconium halide and zirconium alkoxide were prepared, and amorphous ZrO<sub>2</sub> films were obtained by spin-coating and annealing at 300 °C through the direct condensation reaction between them. The ZrO<sub>2</sub> films exhibited a high dielectric constant near 10, and a low leakage current density of  $5 \times 10^{-8}$  A cm<sup>-2</sup> at a field of 1 MV cm<sup>-1</sup>. High mobility p-type pentacene TFTs were fabricated using the ZrO<sub>2</sub> dielectrics, with a saturation field-effect mobility of 3.7 cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup>, a threshold voltage of -2.7 V, an on/off ratio of  $1.1 \times 10^6$  and a subthreshold swing of 0.65 V dec<sup>-1</sup>.

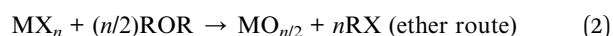
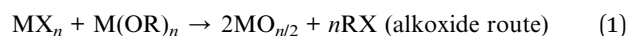
Received 18th August 2018  
Accepted 7th November 2018

DOI: 10.1039/c8ra06911e

rsc.li/rsc-advances

High dielectric constant (high-*k*) materials have received great interest as alternatives to conventional dielectric materials such as silicon dioxide (SiO<sub>2</sub>). These high-*k* dielectrics can be used as key components of the semiconductor and display industry in the form of capacitor dielectrics and transistor gate insulators.<sup>1</sup> In the semiconductor industry, as the traditional gate dielectric becomes extremely thin (~a few atomic layer thick), leakage currents are generated due to direct tunneling of electrons through the dielectric itself, resulting in power dissipation and heat emission, which is a serious problem to be solved. High-*k* dielectrics are expected to reduce the leakage currents since thicker dielectric layers can be used without altering the required dielectric properties.<sup>1,2</sup> In the display industry, high-*k* dielectrics are considered core materials that allow high on-currents in low voltage operation resulting in low-power consumption devices. Among many high-*k* materials, zirconium oxide (ZrO<sub>2</sub>) was chosen because of its outstanding physical and chemical properties such as high intrinsic dielectric constants, high melting points, and great chemical stability.<sup>3</sup> ZrO<sub>2</sub> is an important metal oxide and has been applied to several technologies including gate dielectrics.<sup>2–4</sup>

Compared with vacuum-based processes such as chemical vapor deposition (CVD) and sputtering, solution-based processes (especially sol–gel methods) have distinguished advantages: low-cost and ease of large area deposition. However, solution-processes based on conventional hydrolytic sol–gel methods involve hydrolysis and condensation reactions, which require high thermal energy to form three-dimensional solid networks of pure components.<sup>5</sup> In the past, several groups have reported on the synthesis of inorganic oxides *via* non-hydrolytic sol–gel routes.<sup>6–12</sup> The basis of these methods is the reaction between a metal halide and an oxygen donor in the absence of water. The oxygen donor may be an alkoxide, an ether or an alcohol. The reaction formulae using an alkoxide or an ether as an oxygen donor are as follows.<sup>8–12</sup>



In the above reaction scheme, the ether route is finally followed by the alkoxide route, in which the alkoxide generation is preceded by a substitution reaction between an alkoxide and an ether.



Finally, the metal oxide network formation is completed through the condensation reaction between the metal halide and the alkoxide as described in eqn (1).

By direct condensation without any hydroxylation of the conventional hydrolytic sol–gel route, the above reactions complete at lower temperatures than the hydrolytic ones.<sup>12</sup> The

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† Electronic supplementary information (ESI) available. See DOI: 10.1039/c8ra06911e



advantages of low temperature process are variously known, including flexible and low-cost processes. The most important characteristic we thought of is the ability to form amorphous films. Amorphous gate insulators are generally preferred because of their smooth surfaces, which eliminates some of problems associated with grain boundaries in crystalline films. The crystalline rough surfaces induce charge trapping and leakage currents, deteriorating the channel mobility and the reliability of thin film devices.<sup>13–16</sup> Even though  $\text{ZrO}_2$  crystallizes at much lower temperatures than  $\text{SiO}_2$ ,<sup>2</sup> the crystallinity of thin  $\text{ZrO}_2$  films can be controlled by adjusting the process temperature of the non-hydrolytic sol-gel method.

In this letter, we present the solution-processed amorphous  $\text{ZrO}_2$  thin films by a non-hydrolytic sol-gel route that can be applied as gate insulators for TFTs. To demonstrate advantages of the high dielectric quality of the  $\text{ZrO}_2$  films, we selected a pentacene as channel material for TFT devices. Pentacene was thermally evaporated, which minimizes secondary effects that may influence the properties of the gate insulator, either chemically or thermally. To grow the  $\text{ZrO}_2$  thin films, the precursor solution was spin-coated onto Si wafers and glass substrates and then annealed at low temperature ( $\sim 300^\circ\text{C}$ ). Also, to investigate the physical properties of the thin  $\text{ZrO}_2$  films including morphological and electrical characteristics, we carried out several analyses, such as X-ray diffraction (XRD), scanning electron microscopy (SEM), current-voltage ( $I_V$ ) and TFT characterization. The  $\text{ZrO}_2$  films exhibited a high dielectric constant near 10 and p-type pentacene TFTs with high mobility were fabricated using the  $\text{ZrO}_2$  dielectrics, with a saturation field-effect mobility of  $3.7\text{ cm}^2\text{ V}^{-1}\text{ s}^{-1}$ .

Materials such as zirconium(iv) isopropoxide isopropanol complex ( $\text{Zr}[\text{OCH}(\text{CH}_3)_2]_4(\text{CH}_3)_2\text{CHOH}$ ), zirconium(iv) chloride ( $\text{ZrCl}_4$ ) and 2-methoxy ethanol were used as received without further purification. We prepared equimolar amounts of  $\text{ZrCl}_4$  and  $\text{Zr}[\text{OCH}(\text{CH}_3)_2]_4(\text{CH}_3)_2\text{CHOH}$  precursor solutions with concentrations of 5 wt% and 16 wt% in 2-methoxyethanol respectively. The prepared solutions were then spin-coated onto silicon and glass substrates at 500 rpm. Finally, the coated substrates were baked at  $100^\circ\text{C}$  on a hot plate for 1 min for solvent evaporation and then post-annealed at  $300^\circ\text{C}$  and  $400^\circ\text{C}$  on a hot plate and  $600^\circ\text{C}$  in a furnace for 1 h for the curing of the thin films to confirm their morphology at each temperature.

To study the electrical properties of the  $\text{ZrO}_2$  thin films, we fabricated metal-insulator-metal (MIM) structures (Fig. 1(a)). The dielectric constants, the leakage currents, and the breakdown voltages of the  $\text{ZrO}_2$  thin films were measured. TFT devices were also fabricated as shown in Fig. 1(b) to investigate the high- $k$  effect of  $\text{ZrO}_2$  gate dielectrics with pentacene channel layers. Pentacene channel layers were deposited by thermal evaporation. In both devices, we used shadow masks for patterning aluminum (Al) top electrodes, pentacene channel layer, and gold (Au) source/drain electrodes. In the TFT device structure, molybdenum/tungsten alloy (MoW) gate electrodes were patterned by photolithographic processes.

The thickness of  $\text{ZrO}_2$  thin films were measured with a spectroscopic ellipsometer (M-2000V, J. A. Woollam. Co. Inc.).

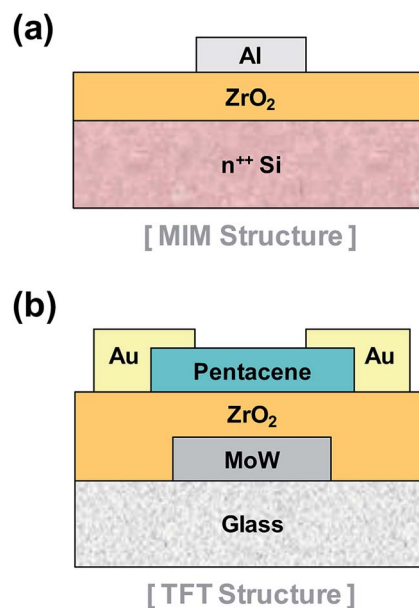


Fig. 1 (a) Metal-insulator-metal (MIM) structure used to measure the dielectric constants, the leakage currents, and the breakdown voltages of the  $\text{ZrO}_2$  films (Al diameter = 1 mm). (b) TFT structure with a 700 Å pentacene channel and a 1500 Å  $\text{ZrO}_2$  gate dielectric layer ( $L$  (channel length) = 160  $\mu\text{m}$ ;  $W$  (channel width) = 1 mm).

The film crystallinity was measured at 40 kV and 30 mA using a Phillips X'pert Pro X-ray diffractometer equipped with a Cu  $K\alpha$  source. The surface morphology were measured using a Hitachi S4800 scanning electron microscopy. The dielectric constants of  $\text{ZrO}_2$  thin films were measured using an Agilent 4284A precision LCR meter in the frequency range between 20 Hz and 100 kHz. Leakage currents, breakdown voltages and TFT characteristics were measured by a Keithley 4200, semiconductor characterization system.

Through the spin-coating and annealing process, we have obtained thin  $\text{ZrO}_2$  films by the non-hydrolytic sol-gel route. The film thicknesses were 420 Å and 1500 Å with concentration of 5 wt% and 16 wt%, respectively. We chose the alkoxide route mentioned in reaction (1), where zirconium chloride and zirconium isopropoxide act as a metal halide and a metal alkoxide (an oxygen donor), respectively. XRD and SEM analyses were performed to confirm the possibility of controlling the film morphology by varying the annealing temperature. Fig. 2 shows the X-ray diffraction patterns of the thin  $\text{ZrO}_2$  films with post-annealing temperatures of  $300^\circ\text{C}$ ,  $400^\circ\text{C}$ , and  $600^\circ\text{C}$ . An amorphous phase is observed for the samples with  $300^\circ\text{C}$  and  $400^\circ\text{C}$ . Only the sample annealed at  $600^\circ\text{C}$  exhibits a crystalline structure.  $\text{ZrO}_2$  exists in three major crystal phases: cubic, tetragonal, and monoclinic polymorphs.<sup>17</sup> In our experiment, the thin films annealed at  $600^\circ\text{C}$  showed the tetragonal phase. From the full width at half maximum height of the peak, the crystalline domain size can be estimated by the Scherrer relation ( $L_s = n\lambda/\beta \cos \theta$ , where  $n$  is unity,  $\lambda$  the wavelength of incident X-rays and  $\theta$  the diffracted angle). The crystalline domain size was calculated to be about 160 Å, which means that grain boundaries are caused by the nanocrystals in the



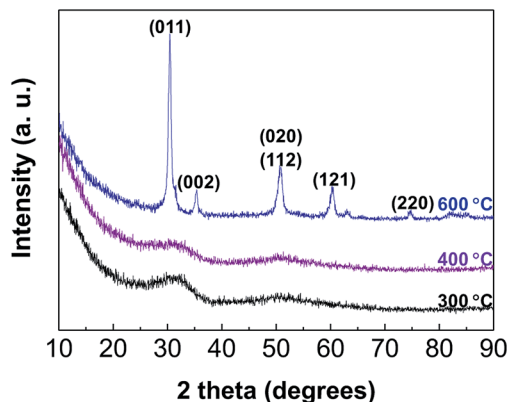


Fig. 2 XRD patterns of the spin-coated thin  $\text{ZrO}_2$  films post-annealed at 300 °C, 400 °C, and 600 °C, respectively.

amorphous matrix. Through SEM analysis (Fig. S1†), the morphology of the thin film surface could be confirmed intuitively, and a comprehensive conclusion could be made comparing with the crystallographic analysis of XRD. As expected, the conditions of 300 °C and 400 °C show microscopically flat surfaces, and at 600 °C, it is a rough surface due to the crystal growth as confirmed by XRD. To avoid deleterious effects of the grain boundaries and take advantage of the various features of the low temperature processes, we adopted the low temperature processed ( $\sim 300$  °C) films as gate insulators for TFTs fabrication. In addition, because of the tendency of binary oxides to crystallize at low thermal energy, multi-metal oxides are commonly used to induce amorphous structures.<sup>13</sup> One of the important facts to consider in this work is that reliable amorphous gate insulators can be obtained using only binary oxides at relatively low temperatures by virtue of the non-hydrolytic sol-gel route.

In order to investigate the potential of thin  $\text{ZrO}_2$  films as gate insulators of TFTs, the dielectric constants, the leakage currents, and the breakdown voltages of the  $\text{ZrO}_2$  thin films were measured. Fig. 3(a) shows the measured dielectric constants of a  $\text{ZrO}_2$  film over the predetermined frequency range, indicating a high dielectric constant of about 10 at 100 kHz. This value is generally less than the known value, over 20.<sup>2–4</sup> Our solution-processed thin  $\text{ZrO}_2$  films are amorphous and inevitably contain pores as a consequence of the sol-gel process.<sup>5</sup> This, in turn, produces a film of lower density than a vacuum-deposited crystalline film, and as a result the dielectric constant of the film must be small. Even though the non-hydrolytic sol-gel  $\text{ZrO}_2$  films have relatively small dielectric constants compared to their vacuum deposited counterparts, the amorphous nature could offset this negative effect by minimizing the grain boundary-induced device instability, and they still have a much higher dielectric constant value than silicon dioxide ( $\sim 3.9$ ). Fig. 3(b) shows the leakage current characteristics of the  $\text{ZrO}_2$  films. The leakage current at an electric field of 1  $\text{MV cm}^{-1}$  is observed at about  $5 \times 10^{-8} \text{ A cm}^{-2}$  and the breakdown voltage is greater than 4  $\text{MV cm}^{-1}$ , which demonstrates the applicability of gate dielectrics. Besides, the

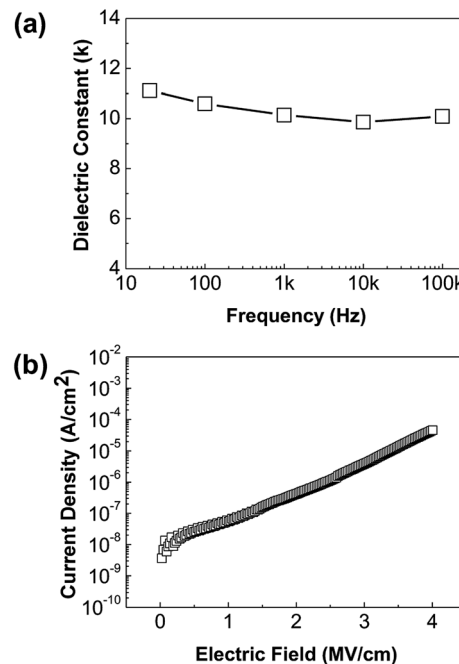


Fig. 3 (a) Dielectric constants of a thin  $\text{ZrO}_2$  films as a function of frequency, (b) leakage current across a thin  $\text{ZrO}_2$  film.

completion of the  $\text{ZrO}_2$  non-hydrolytic sol-gel reaction at a temperature of 300 °C or higher can also be confirmed by the electrical characteristics at 300 °C. Similar dielectric constants were measured according to frequencies. Generally, when the reaction is insufficient, abnormal dielectric constant values appear at high frequencies due to the mobile charges in the films. In addition, low leakage currents also represent a stable  $\text{ZrO}_2$  thin film state.

For the pentacene TFT, a representative output curve of the drain current ( $I_{\text{DS}}$ ) versus the drain voltage ( $V_{\text{DS}}$ ) at various gate voltages ( $V_{\text{GS}}$ ) is shown in Fig. 4(a). The TFT demonstrates a typical transistor behaviour. Current saturation is observed at high  $V_{\text{DS}}$  as the accumulation layer of the pentacene channel is pinched off near the drain electrode. A transfer curve of  $I_{\text{DS}}$  versus  $V_{\text{GS}}$  with a  $V_{\text{DS}}$  of  $-10$  V is plotted in Fig. 4(b). By taking the square root of both sides in the following equation, TFT characteristics including the field effect mobility ( $\mu$ ) were obtained in the saturation regime.<sup>18</sup>

$$I_{\text{DS}} = \frac{WC_i\mu}{2L}(V_{\text{GS}} - V_{\text{T}})^2 \quad (4)$$

where,  $C_i$  is the areal capacitance of the  $\text{ZrO}_2$  insulator,  $W$  is the channel width,  $L$  is the channel length,  $V_{\text{T}}$  is the threshold voltage. The TFTs exhibit excellent device characteristics: a field-effect mobility ( $\mu_{\text{sat}}$ ) of  $3.7 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ , a threshold voltage of  $-2.7$  V, a on/off ratio of  $1.1 \times 10^6$ , and a subthreshold swing of  $0.65 \text{ V dec}^{-1}$ . The field-effect mobility of  $3.7 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  at low  $V_{\text{GS}}$  is one of the highest results reported so far. Compared with pentacene TFTs with  $\text{SiO}_2$  gate insulators, the superiority of  $\text{ZrO}_2$  high- $k$  dielectrics of this study can be demonstrated. In general, for pentacene TFTs with  $\text{SiO}_2$  gate insulators, the field effect mobility is less than  $1 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ ,



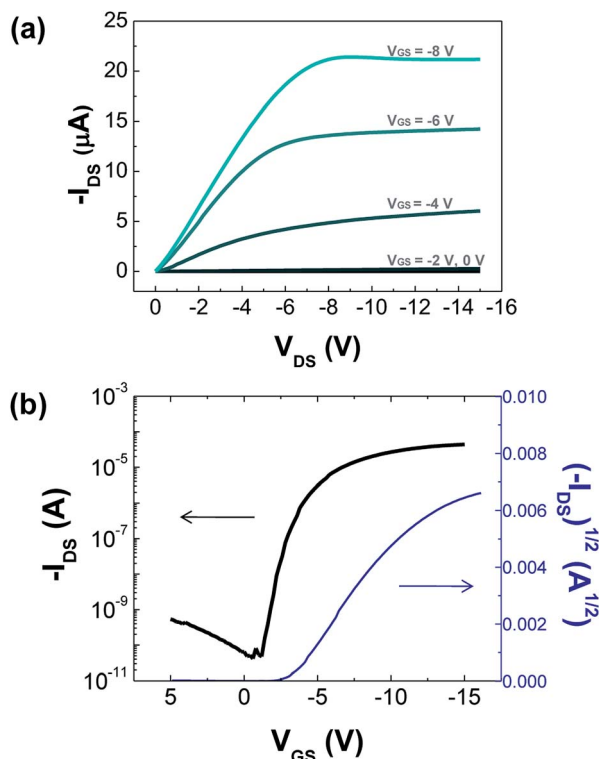


Fig. 4 Device characteristics of spin-coated  $\text{ZrO}_2$ /pentacene TFT devices. (a) Output characteristics:  $V_{\text{GS}}$  varied from 0 V to  $-8$  V, (b) Transfer characteristics:  $V_{\text{DS}} = -10$  V.

which is compared with the mobility of  $3.7 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  of this study.<sup>19</sup> This excellent result is attributed to the high- $k$  effect and smooth amorphous morphology of the  $\text{ZrO}_2$  film. High- $k$  dielectrics can induce more charge carriers in the semiconductor channel region. Unlike the traditional metal-oxide-semiconductor field-effect transistor (MOSFET) theory,<sup>20</sup> several studies have shown that mobility depends on  $N$  (accumulated carriers in the channel) in organic TFTs and support high mobility effect of our experiments.<sup>21–24</sup> This claim is based on the multiple trapping and release (MTR) model, which is widely used for the amorphous silicon (a-Si:H) TFT modeling.<sup>22–24</sup> Lee *et al.* have demonstrated the same relevance for oxide semiconductor TFTs using the MTR model and noted generalization for disordered semiconductor TFTs including the organic TFTs.<sup>25</sup> They modelled the relationship between mobility and capacitance and presented a generalized equation.<sup>25</sup> Another remarkable feature of our results is a very small subthreshold swing. It determines the voltage swing required to switch the device from the “off” to “on” state. The small subthreshold swing allows TFTs with low operating voltage to be implemented.

## Conclusions

In summary, we have demonstrated the potential of solution-processed amorphous  $\text{ZrO}_2$  high- $k$  dielectric films fabricated with a non-hydrolytic sol-gel route at low temperature to realize

high-performance devices at low voltage. We have succeeded in fabricating spin-coated  $\text{ZrO}_2$  thin films using a non-hydrolytic sol-gel route and demonstrated through chemical reactions that this is possible and have opened up a new way to develop high- $k$  dielectrics at low process temperatures ( $\sim 300^\circ\text{C}$ ). In spite of amorphous  $\text{ZrO}_2$ , a dielectric constant near 10 and a low leakage current density of  $5 \times 10^{-8} \text{ A cm}^{-2}$  at a field of  $1 \text{ MV cm}^{-1}$  were obtained. Besides, TFTs with these  $\text{ZrO}_2$  gate dielectrics showed excellent transistor characteristics: the field-effect mobility at the saturation region of  $3.7 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  and the subthreshold swing of  $0.65 \text{ V dec}^{-1}$ , which is critical in low voltage operation. In these results, the potential of non-hydrolytic sol-gel  $\text{ZrO}_2$  high- $k$  dielectrics for the enhancement of TFT device performance was successfully demonstrated.

## Conflicts of interest

There are no conflicts of interest to declare.

## Acknowledgements

This work was supported by Samsung Advanced Institute of Technology.

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