Cite this: *RSC Adv.*, 2018, 8, 35306

Core-shell structured NaYF₄:Yb,Tm@CdS composite for enhanced photocatalytic properties

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NaYF₄:Yb,Tm upconversion nanocrystals with hexagonal structure possess excellent photoluminescence emission characteristics. Under near infrared (NIR) light irradiation, the Yb³⁺ ions act as sensitizers to absorb the NIR light and transform NIR light into ultraviolet (UV) and visible (Vis) light continuously. Hybrid NIR-activated photocatalysts can be fabricated by combining upconversion nanocrystals with various semiconductor nanocrystals. In this paper, NaYF₄:Yb,Tm micro-rods were hydrothermally synthesized with oleic acid as capping ligand. The NaYF₄:Yb,Tm@CdS composite was fabricated by *in situ* generation of CdS nanoclusters on the surface of NaYF₄:Yb,Tm micro-rods. The morphologies and structures of NaYF₄:Yb,Tm and NaYF₄:Yb,Tm@CdS were characterized by XRD, SEM, TEM, XPS, UV-Vis and PL spectroscopy. The results of photocatalytic experiments indicated that the NaYF₄:Yb,Tm@CdS composite displayed photocatalytic activity under NIR irradiation. In comparison with pure CdS, the photocatalytic ability of NaYF₄:Yb,Tm@CdS composite under Vis-NIR irradiation was obviously enhanced. 82% of RhB was degraded by NaYF₄:Yb,Tm@CdS catalyst within 75 min under Vis-NIR irradiation, which was more effective than pure CdS (65% degradation of RhB).

Received 13th August 2018
Accepted 8th October 2018

DOI: 10.1039/c8ra06800c

rsc.li/rsc-advances

1 Introduction

Recently, environmental pollution has become an urgent problem to be solved worldwide. It is well known that photocatalytic techniques are widely applied to solve the issue of water pollution because they are environmentally friendly.^{1–4} Among various photo-catalysts, nano-structured cadmium sulfide (CdS) is one of the most widely studied and used photocatalysts because of its narrow band gap (~2.42 eV) at room temperature.^{5,6} In order to improve photocatalytic efficiency, many methods have been applied including the adjustment of particle size, morphology and surface area of the photocatalyst.^{7–9} Furthermore, many technologies, such as deposition of noble metals or non-metals, doping of noble metals or non-metals and coupling with other semiconductors, have been used to improve the utilization of visible light by CdS.^{10–13} However, near-infrared light that accounts for about 45% of solar energy was not fully utilized by photocatalysts in those efforts.¹⁴

To address the above problems, upconversion (UC) nanocrystals can be used as light transmitter. It is recognized as one of the promising methods to improve the utilization of sunlight by combining them with matching semiconductors.^{15–17} Recently, the β-NaYF₄:25 wt% Yb³⁺, 0.6 wt% Tm³⁺@TiO₂ composites were synthesized by the hydrolysis of TTIP, which decomposes about 69% of MO (76% of RhB and 68% of MB)

within 24 h under simulated 980 nm laser irradiation.¹⁸ Shi *et al.* synthesized the β-NaYF₄:Yb³⁺,Tm³⁺@SiO₂@TiO₂ core-shell nanocrystals by the micro-emulsion method. The results of photocatalytic experiment showed that more than 76% MO were degraded within 60 min under the simulated solar light irradiation with the existence of NaYF₄:Yb³⁺,Tm³⁺@SiO₂@TiO₂ nano-composites. In contrast, only 7% MO were decomposed in the presence of the same amount of TiO₂.¹⁹ Guo *et al.* reported that β-NaYF₄:Yb³⁺,Tm³⁺@ZnO nano-composites were synthesized *via* a two-step high temperature thermolysis method. The photocatalyst of β-NaYF₄:Yb³⁺,Tm³⁺@ZnO resulted in 65% degradation of RhB within 30 h under simulated 980 nm laser irradiation.²⁰ Tou *et al.* prepared β-NaYF₄:Yb,Tm@amorphous SiO₂@wurtzite ZnO core-shell-shell nanoparticles. The nanoparticles showed good photocatalytic activity under the 980 nm laser irradiation and decomposed about 87% RhB within 100 min under the simulated sunlight irradiation.²¹ However, NaYF₄:Yb³⁺,Tm³⁺ nanoparticles show low UV emissions and relatively strong visible light emissions. Unfortunately, the visible light in those emissions cannot be harnessed by the ZnO and TiO₂ shells. In order to improve utilization of visible light emissions, Yu *et al.* reported the combination of NaYF₄:Yb³⁺,Tm³⁺ with a narrower band-gap CdS. However, the process of NaYF₄:Yb³⁺,Tm³⁺/CdS composites reported by Yu *et al.* is complex, which includes the individual preparation of functional NaYF₄:Yb,Tm and CdS particles, and then linking them through SHCH₂COOH and SHCH₂CH₂OH chains. Moreover, the synthetic NaYF₄:Yb³⁺,Tm³⁺ showed low upconversion fluorescence properties.²²

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In the present work, a new composite consisting of NaYF₄:Yb³⁺,Tm³⁺ and narrower band-gap CdS nanoparticle was fabricated through a two-step hydrothermal method. The photocatalytic properties of the core-shell structured NaYF₄:Yb,Tm@CdS composite were investigated by photo-degradation of RhB under the irradiation of NIR and Vis-NIR light. The developed composites may have prospective applications in the environmental remedy.

2 Experimental

2.1 Materials

Rare earth oxide (Y₂O₃, Yb₂O₃, Tm₂O₃, 99.99 wt%) were all purchased from Shang Macklin Biochemical Co. Ltd., China. Sodium hydroxide (NaOH, 98 wt%), ammonium fluoride (NH₄F, 98 wt%), oleic acid (OA, 90%), trisodium citrate dihydrate (C₆H₅Na₃O₇·2H₂O, 98 wt%), ammonia aqueous solution (NH₃·H₂O, 25–28% wt%), cadmium chloride (CdCl₂, 99 wt%), and ethanol (99.7 wt%) were purchased from Aladdin Industrial Corporation, China. Hydrochloric acid (HCl, 36–38 wt%) was supplied by Wu Xi City Yasheng Chemical Co. Ltd., China. Thioacetamide (TAA, 99.0 wt%) was purchased from Sinopharm Chemical Reagent Co. Ltd., China. All raw materials were used as starting materials without further purification.

2.2 Synthesis of water-dispersible NaYF₄:Yb,Tm microrods

The NaYF₄:30%Yb, 0.5%Tm microrods was synthesized according to the previous method.²³ The detailed synthesis process was described as follows. Firstly, Y₂O₃ (0.7 mmol), Yb₂O₃ (0.3 mmol), Tm₂O₃ (0.005 mmol) were fully dissolved in the hydrochloric acid, and then the solution was heated to evaporate water completely to obtain rare earth chlorides. Secondly, a 7.5 ml deionized water solution of 1.5 g NaOH was mixed with 25 ml ethanol and 25 ml oleic acid under stirring. Thirdly, the as-prepared rare earth chlorides were dissolved in 10 ml deionized water to obtain mixture solution of rare earth chlorides. Then, the 10 ml as-prepared rare earth chlorides mixture solution was added to the above mixture solution and stirred continuously for 30 min, followed by adding 5 ml of NH₄F (2 M) into the flask and continuous stirring for another 60 min. Finally, the liquid mixture was poured into the Teflon-lined autoclave, which was sealed and maintained at 220 °C for 12 h. After the autoclave was cooled to room temperature, the obtained product was collected by centrifugation, and washed with distilled water and absolute ethanol three times.

The water-dispersible NaYF₄:Yb,Tm microrods were prepared by protonation method.²⁴ Typically, 1 mmol NaYF₄:Yb,Tm was added into 20 ml deionized water, the pH value of the NaYF₄:Yb,Tm solution was adjusted to 4 by slow dropping diluted HCl (0.1 M) under magnetic stirring. This process was maintained for 2 h at room temperature. Then 10 ml of hexane was mixed with the solution and the water-dispersible NaYF₄:Yb,Tm was collected by centrifugation, washed with ethanol and deionized water three times and finally added to 10 ml aqueous solution.

2.3 Synthesis of NaYF₄:Yb,Tm@CdS core-shell composites

Typically, 0.06 g NaYF₄:Yb,Tm was dispersed in 140 ml deionized water by sonication treatment for 30 min. Trisodium citrate dihydrate (7 mL, 0.1 M) and CdCl₂ (7 mL, 0.08 M) were sequentially added to the above solution under continuous stirring for 20 min and 40 min, respectively. After then, ammonia aqueous solution was slowly dropped in the mixture solution until the pH value reached 10.5. Subsequently, the mixture solution was slowly heated to 65 °C in a bath. Then, 10 ml TAA (0.063 M) was slowly injected into the above solution with 0.1 ml min⁻¹ by the peristaltic pump. The reaction was kept for another 1 h at 65 °C. The final product was washed with ethanol and deionized water for several times, and dried at 60 °C for 12 h in an oven.

2.4 Characterizations

Powder X-ray diffraction (XRD) was carried out on a Bruker D8 ADVANCE X-ray diffractometer equipped with Cu K_α radiation (λ = 0.15418 nm). The SEM images was obtained using FEI QUANTA 400. High Resolution Transmission electron microscopy (HRTEM) was performed on a FEI Tecnai G2 F20 electron microscope at an accelerating voltage of 200 kV. The HRTEM samples were prepared by dropping a suspension of NaYF₄:Yb,Tm@CdS composites on a carbon-film coated copper grid. UV-Vis absorption spectra were obtained under the diffuse reflection mode using a METASH UV-9000S spectrometer with BaSO₄ as the reference sample. The surface chemical environments were analyzed by X-ray photoelectron spectra (XPS) using an ESCALAB 250 Xi spectrometer with monochromatic Al K_α X-rays at 150 W. The photoluminescence spectra of photocatalysts were tested on the Edinburgh Instruments FS5 fluorescence spectrometer with a 980 nm laser as the excitation source.

2.5 Photocatalytic experiments

The photocatalytic performances of the CdS and NaYF₄:Yb,Tm@CdS composites were evaluated by the degradation of RhB solution under irradiation of a 300 W xenon lamp (PLS-SXE300). Visible band-pass (400–800 nm) and NIR band-pass (800–2500 nm) were obtained through choosing the suitable light filters. For each experiment, 10 mg photocatalyst was mixed with 50 ml RhB (2 × 10⁻⁵ M). Before illumination, the suspensions were magnetically stirred in dark for 1 h to reach adsorption-desorption equilibrium. During the process of light irradiation, aliquots of about 4 ml were withdrawn at given time intervals and centrifuged to remove the photocatalyst particles and then was tested the concentration of RhB by UV-Vis spectroscopy at 554 nm.

3 Results and discussion

3.1 SEM and TEM images

Fig. 1 shows the typical SEM images of the as-synthesized NaYF₄:Yb,Tm micro-rods. These micro-rods are uniform in size and exhibit a hexagonal rod-like shape with a diameter of ca. 240–270 nm (diagonal length) and a length of ca. 1.4–1.7 μm. Careful observation from Fig. 1, the single micro-rod of

Fig. 1 SEM images of as-prepared $\text{NaYF}_4\text{:Yb,Tm}$ microrods at different magnifications (a) $\times 40\text{k}$ and (b) $\times 160\text{k}$.

$\text{NaYF}_4\text{:Yb,Tm}$ presents a clear and smooth surface. In order to efficiently deposit CdS particles on the surface of $\text{NaYF}_4\text{:Yb,Tm}$ micro-rod, the hydrophobic ligand of oleic acid on the surface of $\text{NaYF}_4\text{:Yb,Tm}$ micro-rod was removed by washing with diluted hydrochloric acid. After removing of hydrophobic ligand, the wettability of $\text{NaYF}_4\text{:Yb,Tm}$ changed from hydrophobic to hydrophilic.

Fig. 2 shows the morphologies of the prepared $\text{NaYF}_4\text{:Yb,Tm@CdS}$ samples were characterized by HRTEM. HRTEM images of $\text{NaYF}_4\text{:Yb,Tm@CdS}$ composite give a better insight into their structural information. As shown in Fig. 2a and b, one can observe the obvious deposition of CdS particles on the surface of $\text{NaYF}_4\text{:Yb,Tm}$ micro-rod and formation of core-shell structure. The black cores are the $\text{NaYF}_4\text{:Yb,Tm}$ micro-rods. The grayish white spots around the surface of micro-rods should be CdS particles and formation the shell of the composite. The thickness of the CdS shell is about 22–25 nm. The lattice distance of 0.1753 nm matches well with the d spacing of (311) planes of the cubic CdS crystal (See Fig. 2c). Furthermore, it can be clearly observed that the CdS particles exist in the form of nano-clusters. One single nano-cluster possesses microporous-like structure which formed by many tiny CdS particles. Comparison with compact CdS shell, the nano-clusters of CdS shell with microporous-like structure may provide two advantages. On the one hand, the microporous-like structure of nano-clusters is beneficial for the catalyst adsorption of the reactant. Sufficient adsorption of reactants is a crucial step for the whole

photocatalytic process and it is also a key parameter to determine the overall efficiency of the photocatalyst. On the other hand, the microporous-like structure of CdS shell may allow more NIR light to reach the $\text{NaYF}_4\text{:Yb,Tm}$ core due to the reduced shielding effect of the microporous CdS shell.

Currently, upconversion materials preparation technologies mainly include hydrothermal and solvothermal methods. In solvothermal method, the mixture of oleic acid/octadecene or oleic acid/oleylamine is usually used as a solvent. The function of oleic acid not only plays the role of capping ligand but also controls the crystal growth. $\text{NaYF}_4\text{:Yb,Tm}$ upconversion materials prepared by solvothermal method have the characteristics of nanometer particle in size and regular morphology. In hydrothermal method, the deionized water is used as a solvent, and organic polymers such as PVP, CTAB and SDS are used as templates.^{25,26} Hydrothermal method is conducive to the synthesis of upconversion materials with larger particle size, but the morphology is generally difficult to be controlled. It should be noted that the light conversion capability of upconversion materials is closely to the size of the materials. The larger the particle size, the stronger the light conversion capability of the material.²⁷

In our work, we successfully hydrothermally synthesized the $\text{NaYF}_4\text{:Yb,Tm}$ micro-rod with regular hexagonal structure using oleic acid as capping ligand. The size of $\text{NaYF}_4\text{:Yb,Tm}$ micro-rod with hexagonal structure is on the micro-scale in the length direction and on the nano-scale in the diameter direction. Such

Fig. 2 TEM images of prepared $\text{NaYF}_4\text{:Yb,Tm@CdS}$ at different magnifications (a) scale bar, 100 nm, (b) scale bar, 50 nm and (c) scale bar, 20 nm.

micro- and nano-scale structures of $\text{NaYF}_4\text{:Yb,Tm}$ hexagonal micro-rod can effectively improve the specific surface area and light conversion capability, and ultimately enhancing the photocatalytic degradation of organic reactant.

3.2 XRD and XPS analysis

Fig. 3 shows the XRD patterns of $\text{NaYF}_4\text{:Yb,Tm}$, CdS, and $\text{NaYF}_4\text{:Yb,Tm@CdS}$, respectively. As shown in Fig. 3a, all characteristic peaks from the sample can be matched well with that of pure $\beta\text{-NaYF}_4\text{:Yb,Tm}$ crystal (JCPDS no. 16-0334). No peaks from cubic-phase crystals or impurities are found, suggesting the sample is hexagonal phase with highly crystallization. After deposition of CdS on the surface of $\text{NaYF}_4\text{:Yb,Tm}$ to form core-shell structure, two sets of diffraction peaks were found in the XRD patterns of $\text{NaYF}_4\text{:Yb,Tm@CdS}$ composite (see Fig. 3b). One set of peaks is consistent with the $\beta\text{-NaYF}_4\text{:Yb,Tm}$ crystals (hexagonal structure), while the other set of peaks matches finely with cubic CdS crystals (JCPDS no. 10-0454) (see Fig. 3c). However, it can be found from Fig. 3b that the diffraction peaks of CdS obviously widened, suggesting the crystal grain of CdS particles refining. In order to estimate the size of CdS crystal grain, the CdS particles were prepared with same experimental parameters in the absence of $\text{NaYF}_4\text{:Yb,Tm}$. The XRD pattern of pure CdS is shown in Fig. 3c. The diffraction peaks in Fig. 3c is accordance with the cubic CdS crystal (JCPDS no. 10-0454). It is interesting found that all the diffraction peaks of CdS crystal grain widened, suggesting the size of crystal grain of CdS is on the nano-scale. The average size of CdS crystal grain is 3–5 nm calculated by Scherrer equation. This result is consistent with the observation result of HRTEM image.

To further prove the chemical composition of the as-synthesized photocatalyst, the XPS of $\text{NaYF}_4\text{:Yb,Tm@CdS}$ in which the elements of Cd, S, Na, Y, Yb, Tm, and F were given in Fig. 4. Two binding energy peaks representing at 405.1 eV and 411.7 eV in Fig. 4b corresponded to the core levels of Cd $3d_{3/2}$

and Cd $3d_{5/2}$, respectively. The core levels of S $2p_{1/2}$ and S $2p_{3/2}$ are located at 161.6 eV and 163.6 eV in Fig. 4c. These results were in accordance with the reported core levels of the CdS nanoparticles.^{6,28} XPS peaks of Na (1s, 1072.2 eV), Y ($3d_{3/2}$, 159.3 eV), Y ($3d_{5/2}$, 161.6 eV), F (1s, 185.52 eV) were also found in Fig. 4d–f.^{29–31}

3.3 Optical properties

The UV-Vis diffuse reflectance spectra of the CdS, $\text{NaYF}_4\text{:Yb,Tm@CdS}$ and $\text{NaYF}_4\text{:Yb,Tm}$ are displayed in Fig. 5a, respectively. UV-Vis diffuse reflectance spectra shows that the pure $\text{NaYF}_4\text{:Yb,Tm}$ do not display evident absorption in the UV-Vis region and the pure CdS shows obvious absorption in the UV-Vis region. However, the as-synthesized $\text{NaYF}_4\text{:Yb,Tm@CdS}$ shows a strong absorption in the UV-Vis region. Furthermore, it can be concluded from Fig. 5a that the CdS has a narrow band gap of 1.97 eV (absorption edge at 628 nm), while the band gap of $\text{NaYF}_4\text{:Yb,Tm@CdS}$ is about 2.03 eV (absorption edge at 610 nm).

It is well known that NaYF_4 particle is an excellent host-matrix for upconversion luminescence when co-doped Yb^{3+} and Tm^{3+} in the NaYF_4 . The $\text{NaYF}_4\text{:Yb,Tm}$ would produce strong upconversion emissions under 980 nm NIR excitation. Yb^{3+} plays the role of sensitizer ion for continuous absorption of 980 nm NIR light and Tm^{3+} acts as an activator ion to emit of a series of spectra with wavelengths less than 980 nm. Fig. 5b shows the upconversion photoluminescence spectra of $\text{NaYF}_4\text{:Yb,Tm}$ and $\text{NaYF}_4\text{:Yb,Tm@CdS}$ under 980 nm wavelength excitation, respectively. A series of spectral peaks appears for $\text{NaYF}_4\text{:Yb,Tm}$ under 980 nm wavelength excitation. Three strong peaks located at 290, 347 and 362 nm, which derived from the radioactive transitions of $^1\text{I}_6 \rightarrow ^3\text{H}_6$, $^1\text{I}_6 \rightarrow ^3\text{F}_4$ and $^1\text{D}_2 \rightarrow ^3\text{H}_6$. The other two peaks at 452 and 476 nm can be assigned to the transitions of the $^1\text{D}_2 \rightarrow ^3\text{F}_4$ and $^1\text{G}_4 \rightarrow ^3\text{H}_6$ of Tm^{3+} ion. However, it is found that all these upconversion peaks weaken evidently after deposited with CdS. The remarkable decrease in upconversion emissions is mainly ascribed to two factors. On the one hand, the existence of CdS shell has a scattering and shielding effects on the incident NIR light. On the other hand, the decrease in upconversion emissions would be attributed to the absorption of the CdS nanocluster shell. Just as shown in Fig. 5a, CdS can effectively absorb light with wavelength lower than 628 nm, but hardly absorbs light with wavelength higher than 628 nm. Thus, the peak intensities in the lower light wavelengths (250 nm to 620 nm) of $\text{NaYF}_4\text{:Yb,Tm@CdS}$ are lower to that of $\text{NaYF}_4\text{:Yb,Tm}$ due to the absorption of the CdS shell. The peak intensities in the higher light wavelengths (>620 nm) are quite similar for the two samples since CdS shell can hardly absorbs light waves with wavelength higher than 620 nm.

3.4 Photocatalytic study

Photodegradation of the $\text{NaYF}_4\text{:Yb,Tm@CdS}$ composite was evaluated by using RhB as model pollutant. For the purpose of comparison, both naked $\text{NaYF}_4\text{:Yb,Tm}$ and CdS are employed as references. Fig. 6a and c displays the UV-Vis absorption spectra of the RhB solution photo-catalyzed by

Fig. 3 XRD patterns of (a) $\text{NaYF}_4\text{:Yb,Tm}$, $\text{NaYF}_4\text{:Yb}$, (b) Tm@CdS , and (c) CdS. Standard XRD patterns of $\text{NaYF}_4\text{:Yb,Tm}$ and CdS are given as references.

Fig. 4 (a) XPS spectra survey of $\text{NaYF}_4:\text{Yb}^{3+}, \text{Tm}^{3+}@\text{CdS}$ composites; (b) XPS spectra of Cd element; (c) XPS spectra of S element; (d) XPS spectra of Na element; (e) XPS spectra of Y element; (f) XPS spectra of F element.

$\text{NaYF}_4:\text{Yb}, \text{Tm}@\text{CdS}$ composite for different irradiation time under NIR (800–2500 nm) and Vis-NIR (400–2500 nm) irradiation, respectively. It can be found that the concentration of RhB solution gradually decreases under irradiation of the NIR light (see Fig. 6a). This result verified that the $\text{NaYF}_4:\text{Yb}, \text{Tm}@\text{CdS}$ has perceivable photocatalytic activity even under NIR irradiation. Considering that both the naked $\text{NaYF}_4:\text{Yb}, \text{Tm}$ and CdS have almost no photocatalytic activity under NIR irradiation (see Fig. 6b), the photocatalytic effect of the $\text{NaYF}_4:\text{Yb}, \text{Tm}@\text{CdS}$ under NIR light should be ascribed to the synergistic influences between $\text{NaYF}_4:\text{Yb}, \text{Tm}$ and CdS particles.

Fig. 6b shows the relative concentration C/C_0 versus the degradation time of different photocatalysts under Vis-NIR and NIR irradiations, respectively. It can be found that the naked CdS shows good catalytic activity under Vis-NIR irradiation. The

degradation ratio of RhB attained to 65% under Vis-NIR irradiation for 75 min. However, it can be also seen from Fig. 6b that the naked CdS has almost no photocatalytic effect under NIR irradiation. This result indicated that the naked CdS possesses good photocatalytic degradation under visible light irradiation. Furthermore, it can be interesting found that the degradation rate of RhB catalyzed by $\text{NaYF}_4:\text{Yb}, \text{Tm}@\text{CdS}$ composite is obviously faster than that catalyzed by naked CdS under Vis-NIR irradiation. After Vis-NIR light irradiation for 75 min, 82% of RhB is degraded by $\text{NaYF}_4:\text{Yb}, \text{Tm}@\text{CdS}$ catalyst. In comparison, only 65% of RhB is degraded by naked CdS catalyst. Obviously, these results indicate that the photocatalytic activity of $\text{NaYF}_4:\text{Yb}, \text{Tm}@\text{CdS}$ is superior to that of naked CdS under Vis-NIR light irradiation. To investigate the stability of the as-prepared samples, we carried out a cyclic photocatalytic

Fig. 5 (a) UV-Vis light absorption spectra of different samples; (b) upconversion photoluminescence (PL) spectra of the $\text{NaYF}_4:\text{Yb}, \text{Tm}$ and $\text{NaYF}_4:\text{Yb}, \text{Tm}@\text{CdS}$ under 980 nm laser excitation at room temperature.

Fig. 6 (a) UV-Vis absorption of the RhB solution catalyzed by $\text{NaYF}_4\text{:Yb,Tm@CdS}$ composite under NIR irradiation; (b) photocatalytic activities of the photocatalysts for the degradation of RhB under different light wavelength band; (c) UV-Vis absorption of the RhB solution catalyzed by $\text{NaYF}_4\text{:Yb,Tm@CdS}$ composite under Vis-NIR irradiation; (d) cycle runs for photocatalytic degradation test using $\text{NaYF}_4\text{:Yb,Tm@CdS}$ as the photocatalyst under NIR irradiation; (e) photocatalytic rate constant of the photocatalysts for the degradation of RhB under different light wavelength band.

degradation test using $\text{NaYF}_4\text{:Yb,Tm@CdS}$ as the photocatalyst. As shown in Fig. 6d, no obvious decrease in the photo-activity of $\text{NaYF}_4\text{:Yb,Tm@CdS}$ can be observed after three consecutive runs, indicating that the $\text{NaYF}_4\text{:Yb,Tm@CdS}$ has good photocatalytic stability.

Most of the photocatalytic reactions follow the Langmuir-Hinshelwood adsorption model, and the L-H model can be simplified to a pseudo-first-order expression: $\ln(C_0/C) = kt$ (where C_0 and C are the equilibrium concentration of adsorption and the concentration of RhB at the irradiation time, t , respectively, and k is apparent rate constant). Using regression-

fitting techniques, the linear plots of $\ln(C_0/C)$ versus irradiation time t are shown in Fig. 6e. It can be found that the apparent rate constant k for $\text{NaYF}_4\text{:Yb,Tm@CdS}$ ($2.37 \times 10^{-2} \text{ min}^{-1}$) is higher than that for naked CdS ($1.14 \times 10^{-2} \text{ min}^{-1}$) under Vis-NIR light irradiation, suggesting that the photocatalytic activity of $\text{NaYF}_4\text{:Yb,Tm@CdS}$ is superior to that of naked CdS under Vis-NIR light.

$\text{NaYF}_4\text{:Yb,Tm}$ is an excellent upconversion material and shows typical photoluminescence emission characteristics. Under 980 nm light irradiation, the emission peaks of Tm^{3+} both in UV light region at 290, 347, 362 nm and in visible light region at 452, 476 nm are observed (see Fig. 5b). Fig. 7 shows the enhanced photocatalytic mechanism of $\text{NaYF}_4\text{:Yb,Tm@CdS}$ under Vis-NIR light. Under NIR light irradiation, the Yb^{3+} ions act as sensitizer to absorb the NIR light and transforms NIR light into UV and visible light continuously, then the UV and visible light are absorbed immediately by CdS catalyst on the surface of the $\text{NaYF}_4\text{:Yb,Tm}$ micro-rods to produce electrons and holes in the CB and VB bands, respectively. The electrons and holes can not only directly decompose the RhB in water solution, but also can degrade them indirectly through oxidation of H_2O molecules to $\cdot\text{OH}$ with high oxidative ability.^{32,33}

4 Conclusions

In summary, hybrid NIR-activated photocatalyst $\text{NaYF}_4\text{:Yb,Tm@CdS}$ core-shell structured composite has been synthesized by hydrothermal method and surface deposition technology. Observation by HRTEM showed that the CdS shell

Fig. 7 Illustration of enhanced photocatalytic mechanism of the core-shell $\text{NaYF}_4\text{:Yb,Tm@CdS}$ composite.

existed in the form of porous nano-clusters. This microporous-like structure of nano-clusters is beneficial for the catalyst adsorption of the reactant. The photocatalytic experiment verified that the NaYF₄:Yb,Tm@CdS micro-rods displayed photocatalytic activity and good photocatalytic stability under NIR light irradiation due to efficient transformation NIR light to UV and visible light by NaYF₄:Yb,Tm. Furthermore, the photocatalytic activity of NaYF₄:Yb,Tm@CdS is obvious superior to that of naked CdS under Vis-NIR light irradiation.

Conflicts of interest

There are no conflicts to declare.

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