

Cite this: *RSC Adv.*, 2018, 8, 30530

The kinetics process of a Pb(II)/Pb(0) couple and selective fabrication of Li–Pb alloys in LiCl–KCl melts

Wei Han,[✉] Wenjuan Wang, Yongchang Dong, Mei Li,^{*} Xiaoguang Yang and Milin Zhang[✉]

The electrode reaction of Pb(II) and co-reduction of Li(I) and Pb(II) were investigated on a tungsten electrode in LiCl–KCl eutectic melts by a range of electrochemical techniques. From cyclic voltammetry and square wave voltammetry measurements, the reduction of Pb(II) was found to be a one-step diffusion-controlled reversible process with the exchange of 2 electrons. The diffusion coefficients of Pb(II) were computed, and they obey the Arrhenius law. Using the linear polarization technique, the kinetic parameters, such as exchange current intensity (j_0), standard rate constant (k^0) and charge transfer resistance (R_{ct}) for the Pb(II)/Pb(0) couple on a tungsten electrode were studied at different temperatures, and the activation energy is 27.32 kJ mol^{−1}, smaller than the one for diffusion of Pb(II), which further confirmed that the reduction of Pb(II) was controlled by diffusion. A depolarisation effect for Li(I) reduction was observed from the results of cyclic voltammetry, square wave voltammetry and chronopotentiometry due to the formation of Li–Pb alloys by co-reduction of Li(I) and Pb(II). Furthermore, five Li–Pb intermetallic compounds, LiPb, Li₈Pb₃, Li₃Pb, Li₁₀Pb₃ and Li₁₇Pb₄ characterized by scanning electronic microscopy and X-ray diffraction, were selectively prepared by potentiostatic electrolysis on a tungsten electrode and galvanostatic electrolysis on a liquid Pb electrode, respectively.

Received 26th July 2018
Accepted 22nd August 2018

DOI: 10.1039/c8ra06329j

rsc.li/rsc-advances

Introduction

A nuclear fusion reactor is considered to be a promising candidate for a future power source. As a tritium breeding blanket material for fusion reactors, Li and Pb alloy with Li 15.7–16.7 at%, namely, Li₁₇Pb₈₃ has attracted considerable interest,¹ because of its advantages,^{2–4} such as good tritium breeding, low Li vapor pressure, moderate melting temperature and high blanket energy multiplication. Therefore, Li–Pb alloy has been widely studied, for example, in studies on the solubility of hydrogen in Li–Pb eutectic alloy,^{5–11} the density and surface tension of Li–Pb melts,¹² the compatibility of structural materials with the liquid Li–Pb alloy^{13,14} the thermodynamic properties of Li–Pb alloys,^{15–19} etc.

At present, the preparation of Li–Pb alloy was performed using a rapid cooling of the melt of Li–Pb alloy with an atomic ratio of 17 : 83. However, during the manufacturing process of the Li–Pb alloy, a loss of Li through oxidation results in the formation of eutectic structures being inhibited. In order to ensure the efficient mixing of metallic Li and Pb and avoid the reaction of high reactivity Li with O₂, Pascual *et al.*²⁰ studied the

optimum process conditions for the preparation of Li–Pb eutectic alloys.

As a new preparation method, electrochemical formation of metal or alloy has been investigated in molten salts because of some promising advantages.²¹ One of the most promising advantages is that the alloy composition could be controlled by selecting electrochemical parameters, which was confirmed by some investigators.^{22–27}

So far, the electrochemical behavior of Pb(II) has been explored in molten chlorides by some investigators.^{28–35} They reported that the reduction reaction of Pb(II) is limited by mass transport of Pb(II) in molten AlCl₃–NaCl–KCl,²⁸ LiCl–KCl,^{29–33} NaCl–KCl³⁴ and ZnCl₂–2NaCl³⁵ melts. Meanwhile, the diffusion coefficients for Pb(II) ions were calculated at different temperatures by cyclic voltammetry (CV) and chronopotentiometry (CP). Haarberg *et al.*³³ determined the activation energy for the diffusion process of Pb(II) and found it to be 35.8 kJ mol^{−1}. However, there is sparse information about the electrochemical formation of Pb based alloys in molten salts. Only the preparation of Al–Pb alloys²⁸ and Mg–Li–Pb alloys³¹ has been reported in molten AlCl₃–NaCl–KCl and LiCl–KCl–PbCl₂–MgCl₂ melts, respectively.

According to the review mentioned above, it was found that the kinetics of electrode reaction for Pb(II)/Pb(0) couple on W electrode and electrochemical formation of Li–Pb alloy have never been investigated in molten salts. So, the electrode

Key Laboratory of Superlight Materials and Surface Technology, Ministry of Education, College of Material Science and Chemical Engineering, Harbin Engineering University, Harbin 150001, China. E-mail: weihan@hrbeu.edu.cn; Fax: +86 451 8253 3026; Tel: +86 451 8256 9890



reaction of Pb(II) was firstly investigated on tungsten electrode, and the kinetic properties, such as exchange current densities, standard rate constants and charge transfer resistances for Pb(II)/Pb(0) couple were estimated in the temperature range from 663 K to 753 K. Then, the co-reduction mechanism of Li(I) and Pb(II) was studied on tungsten electrode by various electrochemical methods. Furthermore, Li-Pb alloys were selective formed by potentiostatic electrolysis on tungsten electrode and galvanostatic electrolysis on liquid Pb electrode, and characterized by X-ray diffraction (XRD) and scanning electronic microscopy (SEM).

Experimental

Salt preparation

All electrochemical tests and samples preparation were performed in an argon-atmosphere glove box (less than 5 ppm O₂ and H₂O). Anhydrous LiCl (99.9%) was purchased from Shanghai Baishi Chemical Engineering Limited Company, KCl (99.9%) from Tianjin Yongda Chemical Reagent Limited Company and PbCl₂ (99.9%) from Shanghai Aladdin Reagent Limited Company. The mixed salts of LiCl-KCl with eutectic composition (45.8 : 54.2 mass) and PbCl₂ were dehydrated under vacuum at 473 K for 72 h to remove the residual water. Then, pre-electrolysis was carried out at -2.1 V (vs. Ag/AgCl) for 5 h to remove metal ion impurities in LiCl-KCl eutectic. The melts temperature was measured using a chromel-alumel thermocouple sheathed by an alumina tube.

Experimental set-up

A three electrode system was established to conduct electrochemical measurement. Spectral pure graphite rod with a diameter of 6.0 mm and tungsten wire (99.9%) with a diameter of 1.0 mm were served as counter electrode and working electrode, respectively. As for the reference electrode, a silver wire with a diameter of 1.0 mm was dipped into eutectic LiCl-KCl containing 1.0 wt% of AgCl (99.99%), which was placed into a Pyrex tube. The measured potentials were relative to this Ag/AgCl reference couple. After each electrochemical test, the active electrode area was measured by measuring the immersed depth of the electrode in the molten salts. However, liquid Pb electrode was used as working electrode to investigate the electrochemical formation of Li-Pb alloys. Liquid Pb working electrode consisted of a pool of liquid Pb (99.99%) in an alumina crucible. Tungsten wire with a diameter of 1.0 mm was served for electrical connections. Metrohm Electrochemical Workstation (Autolab PGSTAT302N) with electrochemical software (NOVA 1.10) was used to perform the electrochemical tests.

Instrumentation

The Li-Pb alloys were fabricated by potentiostatic and galvanostatic electrolysis, respectively. After electrolysis, the cathodic deposits were washed with ethylene oxide (99.7%) to remove solidified salt attached to their surface. These deposits were analyzed by X-ray diffraction (XRD, Rigaku D/max-TTR-III diffractometer) by Cu-K α radiation at 40 kV, 150 mA and

scanning electronic microscopy (SEM, JSM-6480A; JEOL Co., Ltd) to characterize the composition and morphology of Li-Pb alloys.

Results and discussion

Electrode reaction of Pb(II) on tungsten electrode

Number of exchanged electrons. Fig. 1 depicts the cyclic voltammograms acquired in LiCl-KCl (black dotted curve) and LiCl-KCl-PbCl₂ (red solid curve) melts on tungsten electrode, respectively. A couple of redox signals A/A', detected at about -2.35/-2.20 V in the dotted curve, corresponds to the deposition and re-dissolution of metallic Li. After PbCl₂ (0.60 wt%) was added into the melts, a number of reduction/oxidation signals can be recorded in the red curve. The reduction/oxidation peaks B/B', detected at -0.40 V/-0.20 V, are related to the deposition and oxidation of metallic Pb.^{28,31} It is obvious that a series of redox peaks between the peaks A/A' and B/B' correspond to the formation and re-dissolution of different Li-Pb intermetallics. The details are discussed in part of electrochemical co-reduction of Li(I) and Pb(II) in LiCl-KCl melts.

Square wave voltammetry (SWV) has a higher sensitivity and resolution than CV,³⁶⁻⁴⁰ it is usually applied to estimate the number of exchanged electrons in electrochemical reaction. For a simple reversible electrode reaction, the width of the half peak $W_{1/2}$ is related to the number of exchanged electrons by the following equation:^{41,42}

$$W_{1/2} = 3.52RT/nF \quad (1)$$

where R denotes the ideal gas constant (J K⁻¹ mol⁻¹), T is the absolute temperature (K), n represents the number of electrons transferred and F is the Faraday constant.

To further research the electrochemical reaction of Pb(II) ion on tungsten electrode, SWV was conducted and a series of square wave voltammograms of Pb(II) acquired on tungsten electrode at different frequencies are presented in Fig. 2(a).

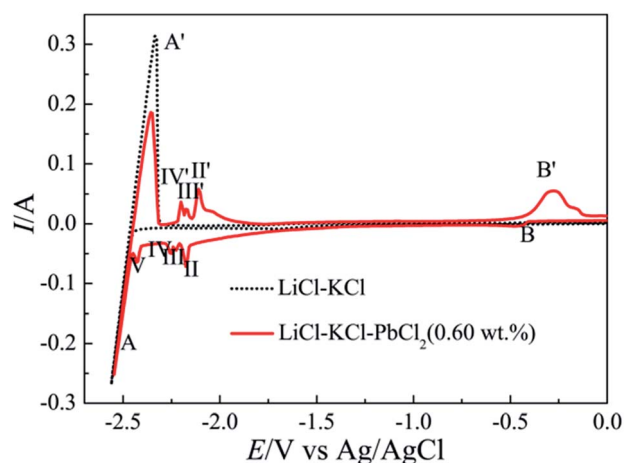


Fig. 1 Cyclic voltammograms of LiCl-KCl melts before (dotted curve) and after the addition of PbCl₂ (0.60 wt%) (red curve) at 723 K on tungsten electrode (0.322 cm²) at scan rate of 0.1 V s⁻¹.



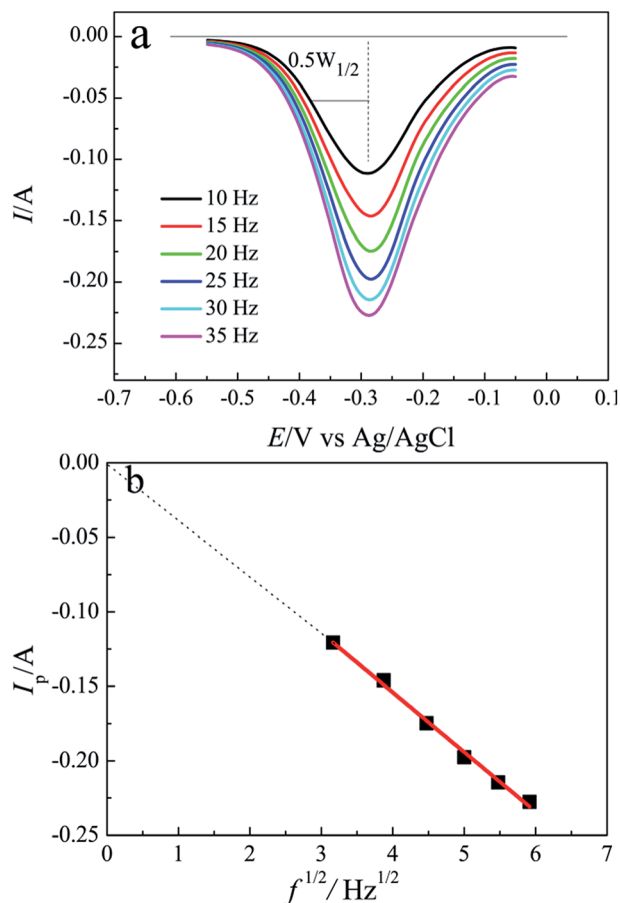
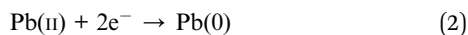


Fig. 2 (a) Square wave voltammograms of LiCl–KCl–PbCl₂ (0.60 wt%) melts on tungsten electrode (0.322 cm²) at different frequencies at 723 K; (b) peak current vs. square root of frequency. Pulse amplitude: 0.1 V; potential step: 0.005 V.

Because the peak currents exhibit a linear dependence with the square root of frequency and the intercept is nearly zero shown in Fig. 2(b), eqn (1) can be used to compute the number of transferred electrons. The average width of half peak is about 0.13 V and the number of electron exchanged is found to be 1.7, closed to 2.0. The result illustrates that the reduction of Pb(II) ion to Pb metal on tungsten electrode proceeds in one step with the exchange of 2-electron.



Diffusion coefficients of Pb(II) in LiCl–KCl melts. CV was conducted in LiCl–KCl–PbCl₂ (0.60 wt%) melts at scan rates of 0.02 to 0.06 V s^{−1} shown in Fig. 3(a). It is clear that cathodic and anodic peak currents increase with the increase of scan rates. The linear relationship between peak currents (I_{pc} and I_{pa}) and the square root of the scan rates can be seen in Fig. 3(b), which indicates that the reduction of Pb(II) is a simple diffusion controlled process. While the peak potentials (E_{pc} and E_{pa}) are almost independent with the logarithm of the scan rates (Fig. 3(c)). According to the

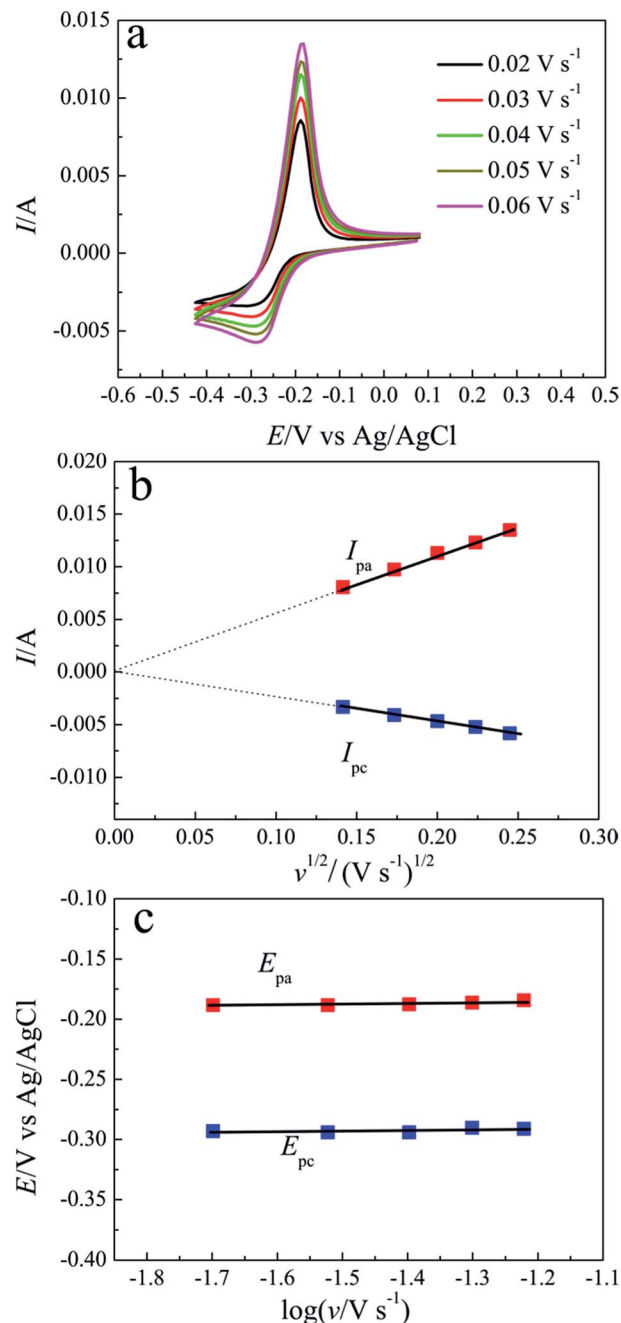


Fig. 3 (a) Cyclic voltammograms recorded in LiCl–KCl–PbCl₂ melts at 723 K on tungsten electrode (0.322 cm²) at different scan rates, (b) anodic and cathodic peak currents as a function of square root of scan rates and (c) anodic and cathodic peak potentials vs. logarithm of scan rates.

results, we infer that the reduction of Pb(II) ion is a reversible and diffusion controlled process in the scan rates of 0.02–0.06 V s^{−1} in LiCl–KCl melts.

For a reversible system, the Berzins and Delahay equation⁴³ can be applied to calculate diffusion coefficient D .

$$I_p = -0.611SC_0(nF)^{3/2}(Dv/RT)^{1/2} \quad (3)$$



Table 1 Comparison of diffusion coefficient of Pb(II) in LiCl–KCl melts

| T/K | $D_{\text{Pb(II)}} \times 10^5 / \text{cm}^2 \text{ s}^{-1}$ | | | | 753 |
|----------------------------|--|------|---------------|----------------|-----------------------|
| | 663 | 693 | 723 | | |
| CV | 0.77 | 0.96 | 1.12 | | 0.78 ± 0.03 (ref. 35) |
| CP | | | 1.7 (ref. 29) | 2.18 (ref. 32) | 0.83 ± 0.02 (ref. 35) |
| $E_a / \text{kJ mol}^{-1}$ | 30.48 ± 0.43 (35.8 (ref. 33)) | | | | 1.58 (ref. 33) |

where I_p denotes the cathodic peak current (A), S represents the electrode surface area (cm^2), C_0 denotes the bulk concentration of Pb(II) ions (mol cm^{-3}), v is the potential scan rate (V s^{-1}), R is the ideal gas constant ($\text{J K}^{-1} \text{ mol}^{-1}$), T is the absolute temperature (K), n represents the number of electrons transferred and F the Faraday constant.

According to the eqn (3), the diffusion coefficients of Pb(II) in LiCl–KCl–PbCl₂ melts were calculated in the temperature range of 663–753 K. The comparison of the calculated results and the ones from ref. 29, 32, 33 and 35 are listed in Table 1. It can be seen from Table 1, our results are agreement with the data acquired by Haarberg *et al.*³³ using CV and CP. However, they are slightly smaller than those achieved by Castrillejo *et al.*³⁵ using cyclic voltammetry and chronopotentiometry. The difference may be caused by different chloride melts and concentration of Pb(II) ions. A linear relationship between the logarithm of diffusion coefficients and the reciprocal of temperatures are acquired from Fig. 4, in line with Arrhenius formula. Therefore, eqn (4) can be applied to estimate the activation energy for diffusion of Pb(II), and value is found to be 30.48 kJ mol^{-1} , slightly smaller than that calculated by Haarberg *et al.*³³ The reason caused the difference may be that the measured temperature range is different.

$$D = D_0 \exp(-E_a/RT) \quad (4)$$

where D_0 denotes the pre-exponential factor, E_a is the activation energy for diffusion.

Kinetics properties of Pb(II)/Pb(0) couple on tungsten electrode. The kinetic parameters, such as exchange current intensity (j_0), standard rate constant (k^0) and charge transfer resistance (R_{ct}), were studied on tungsten electrode. For the electrode reaction of Pb(II)/Pb(0) couple, the current density corresponding to the its reaction rate can be given using the Butler–Volmer equation.⁴⁴

$$j = j_0 \left[e^{-\frac{\alpha n F}{RT} \eta} - e^{\frac{(1-\alpha) n F}{RT} \eta} \right] \quad (5)$$

where j denotes the current density, n is the number of exchanged electrons, F is the Faraday constant, α is charge transfer coefficient, η ($\eta = E - E_{eq}$) is overpotential, E and E_{eq} are the electrode potential and equilibrium potential, respectively.

For a small value of x , the exponential e^x can be approximated as $1 + x$. Thus, for sufficiently small η , eqn (5) can be expressed as follows:

$$j = -j_0(nF/RT)\eta \quad (6)$$

In analogy to Ohm's law, the charge transfer resistance R_{ct} can be computed using eqn (7).⁴⁴

$$R_{ct} = (d\eta/di)_{\eta \rightarrow 0} = RT/nFSj_0 \quad (7)$$

Meanwhile, the standard reaction rate constant (k^0) is also calculated according to eqn (8).⁴⁴

$$j_0 = nFSk^0C_0 \quad (8)$$

Fig. 5(a) shows the linear polarization curve recorded at the equilibrium potential of Pb(II)/Pb(0) on tungsten electrode at scan rate of 5 mV s^{-1} and overpotentials of ± 15 mV in the temperature of 663–753 K. The slopes of fitted line are employed to evaluate the exchange current density j_0 . Meanwhile, according to eqn (7) and (8), the charge transfer resistance R_{ct} and standard reaction rate constant k^0 are also estimated and listed in Table 2. It is obvious that the charge transfer resistance significantly decreased, the exchange current densities and standard reaction rates increase with the increasing of temperature. The values of standard reaction rates are in the order of 10^{-3} . The relationship

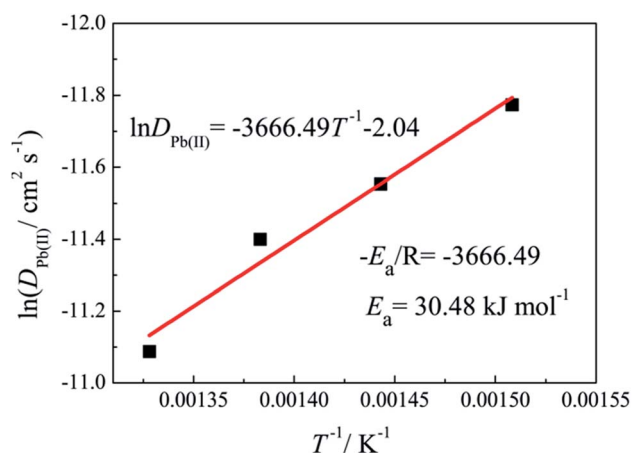


Fig. 4 The change of diffusion coefficients with temperatures.



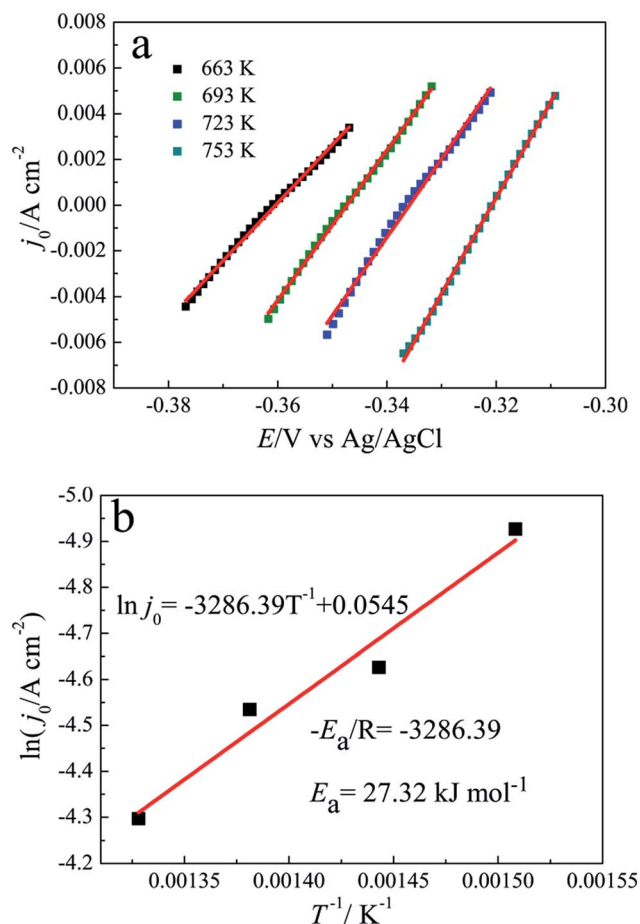


Fig. 5 (a) The linear polarization of LiCl–KCl–PbCl₂ (0.60 wt%) melts on tungsten electrode (0.322 cm²) at different temperature at scan rate of 0.005 V s^{−1} and (b) the relationship between the reciprocal of the temperatures and the logarithm of exchange current densities.

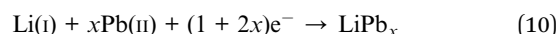
between the reciprocal of the temperature and the logarithm of exchange current density is presented in Fig. 5(b), which obeys to Arrhenius law. Thus, the activation energy for the electrode reaction of Pb(II)/Pb(0) couple can be acquired.

$$\ln j_0 = -E_a/RT + \ln A \quad (9)$$

where E_a denotes reaction activation energy, A is pre-exponential factor.

Using eqn (9), the activation energy for the electrode reaction of Pb(II)/Pb(0) couple is estimated to be 27.32 kJ mol^{−1}, which is smaller than the one for diffusion of Pb(II) in LiCl–KCl melts. The result can further identify that the electroreduction of Pb(II) on tungsten electrode is diffusion-controlled process.

Electrode reaction of Pb(II) on tungsten electrode. Cyclic voltammograms recorded at different reversion potentials with compartmentalized the redox peaks are demonstrated in Fig. 6(a). Six pairs of redox peaks, I/I', II/II', III/III', IV/IV', V/V' and B/B' are clearly seen. Except for the signals of B/B' mentioned above, five couple of signals, I/I', II/II', III/III', IV/IV' and V/V', are recorded at about −2.03/−1.96 V, −2.14/−2.01 V, −2.21/−2.14 V, −2.25/−2.16 V, −2.43/−2.31 V respectively. It is obvious that a depolarisation effect for Li(I) reduction occurs due to the formation of Li–Pb intermetallics when Li(I) is reduced on pre-deposited Pb metal. According to the XRD results in part of preparation and characterization of Li–Pb alloys on different electrodes, the five Li–Pb intermetallic compounds are LiPb, Li₈Pb₃, Li₃Pb, Li₁₀Pb₃ and Li₁₇Pb₄, respectively. Thus, the attribution of five couples of redox peaks can be ascertained, and their corresponding reduction potentials are listed in Table 3. Since metallic Li is not found when the formation of five Li–Pb alloys, the co-reduction process of Li(I) and Pb(II) is suggested as the following two-step reactions: (1) $x\text{Pb(II)} + 2x\text{e}^- \rightarrow x\text{Pb}$; (2) $\text{Li(I)} + \text{e}^- + x\text{Pb} \rightarrow \text{LiPb}_x$. Thus, the electrochemical formation of Li–Pb intermetallic compounds can be expressed as follows:



To investigate the co-reduction of Li(I) and Pb(II) on tungsten electrode in LiCl–KCl–PbCl₂ (0.60 wt%) melts, SWV was conducted at a frequency of 20 Hz and step potential of 5 mV shown in Fig. 6(b). Seven current peaks are recorded around at −2.01 V, −2.13 V, −2.22 V, −2.25 V and −2.42 V, respectively. Cathodic peaks A and B are associated with the deposition of metallic Li and Pb, respectively. The others between peaks A and B should pertain to the formation of five Li–Pb compounds. These reduction potentials are also displayed in Table 3. It is obvious that these potentials are consistent with those from cyclic voltammogram shown in Fig. 6(a).

Fig. 6(c) depicts a range of chronopotentiograms recorded in LiCl–KCl–PbCl₂ (0.60 wt%) melts on tungsten electrode at different current intensities. Six potential plateaus are detected in chronopotentiograms. The first potential plateau (I) recorded at about −2.04 V is ascribed to the formation of LiPb intermetallic compound when the cathodic current reaches −30 mA. While the cathodic current is more positive than −70 mA, the second potential plateau II occurs at around −2.14 V due to the deposition of Li₈Pb₃ intermetallic compound. At the cathodic current intensity of −95 mA, three potential plateaus (III, IV and V) appear at about −2.23 V, −2.28 V, −2.43 V, corresponding to

Table 2 The reaction kinetic parameter of Pb(II)/Pb(0) couple on tungsten electrode at different temperature

| T/K | 663 | 693 | 723 | 753 |
|------------------------------------|-------------------|-------------------|-----------------|------------------|
| $j_0 \times 10^2/\text{A cm}^{-2}$ | 0.73 ± 0.0075 | 0.98 ± 0.0073 | 1.1 ± 0.015 | 1.4 ± 0.0078 |
| R_{ct}/Ω | 12.15 ± 0.12 | 9.46 ± 0.07 | 8.79 ± 0.12 | 7.19 ± 0.04 |
| $k^0 \times 10^3/\text{cm s}^{-1}$ | 1.03 ± 0.01 | 1.41 ± 0.01 | 1.59 ± 0.02 | 2.05 ± 0.01 |
| $E_a/\text{kJ mol}^{-1}$ | 27.32 ± 0.44 | | | |



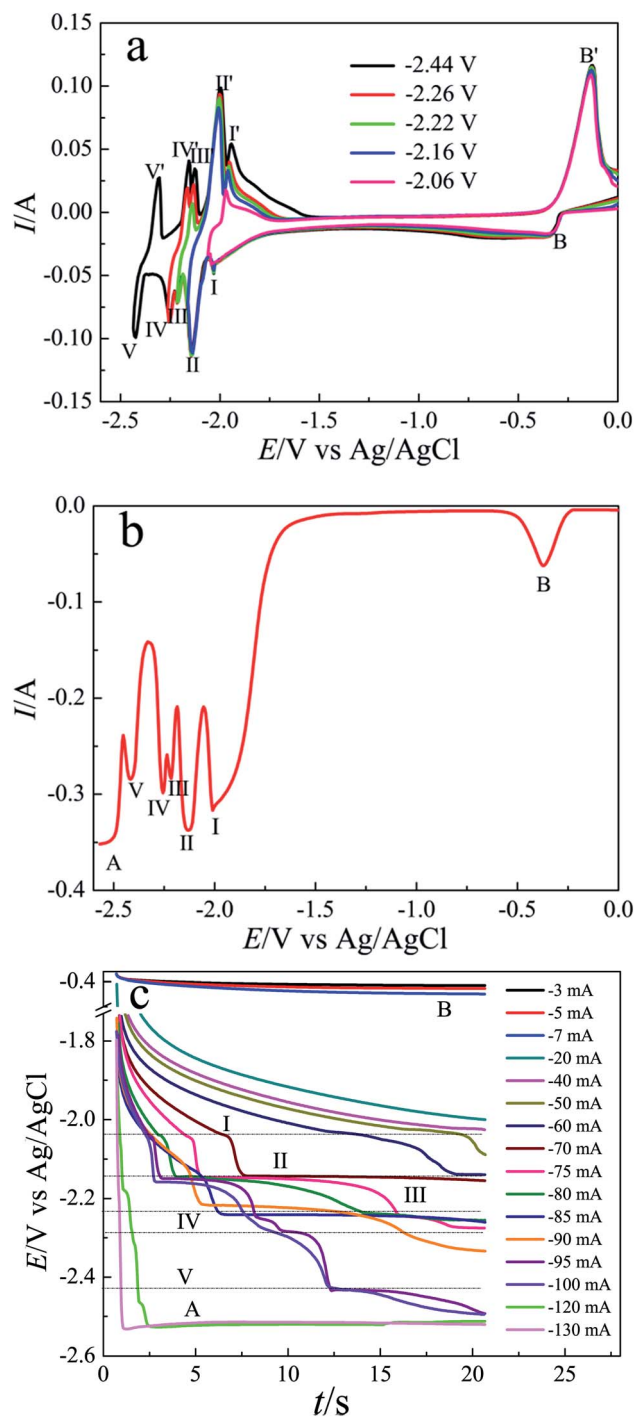


Fig. 6 (a) Cyclic voltammograms recorded at different switching potentials at scan rate of 0.1 V s^{-1} , (b) square wave voltammogram recorded at pulse amplitude of 0.1 V , potential step of 0.005 V and frequency of 20 Hz , and (c) chronopotentiograms recorded at different current intensities on tungsten electrode (0.322 cm^2) in LiCl-KCl-PbCl_2 ($0.60 \text{ wt}\%$) melts at 723 K .

the formation of Li_3Pb , $\text{Li}_{10}\text{Pb}_3$ and Li_4Pb intermetallic compounds, respectively. The last potential plateau of A is detected at around -2.51 V , associated with the deposition of metallic Li. The deposition potentials of the formation of Li-Pb

intermetallic compounds are also presented in Table 3. It is clear that the potential ranges for the formation of Li-Pb intermetallic compounds are in accordance with those observed in other electrochemical tests.

Preparation and characterization of Li-Pb alloys on different electrodes

In order to confirm the attribution of five reduction peaks/plateaus, potentiostatic electrolysis was carried out at different applied potentials in term of the deposition potentials of Li-Pb intermetallic compounds acquired by CV, SWV and CP (listed in Table 3) at 723 K . Fig. 7 displays the XRD patterns of deposits prepared by co-reduction of Li(I) and Pb(II) using potentiostatic electrolysis at -2.04 V , -2.14 V , -2.28 V and -2.43 V for 6 h on tungsten electrode (0.322 cm^2) in LiCl-KCl-PbCl_2 ($6 \text{ wt}\%$) melts. The results of XRD indicate that five Li-Pb intermetallic compounds, including LiPb , Li_8Pb_3 , Li_3Pb , $\text{Li}_{10}\text{Pb}_3$ and $\text{Li}_{17}\text{Pb}_4$, were produced. It can be seen from Fig. 7(a) that only one intermetallic compound of LiPb was generated by potentiostatic electrolysis at -2.04 V , which can ascertain the peak/plateau I detected at about -2.04 V in Fig. 6 is related to the formation of LiPb compound, which reveal that eutectic Li-Pb alloy can be produced by potentiostatic electrolysis at -2.04 V . We can find that metallic Pb is dominant. The reason may be that the pre-deposited Pb did not react completely.

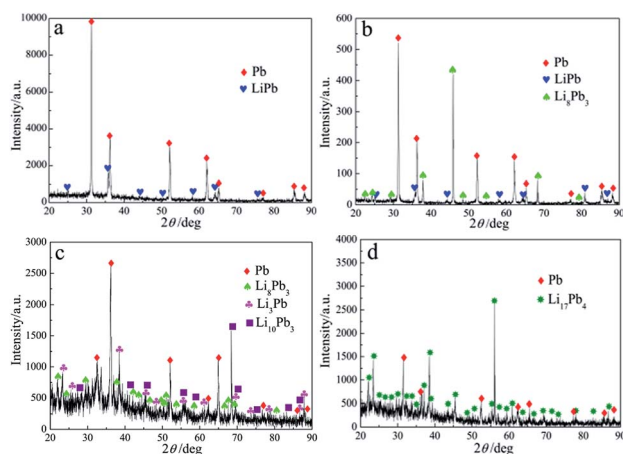
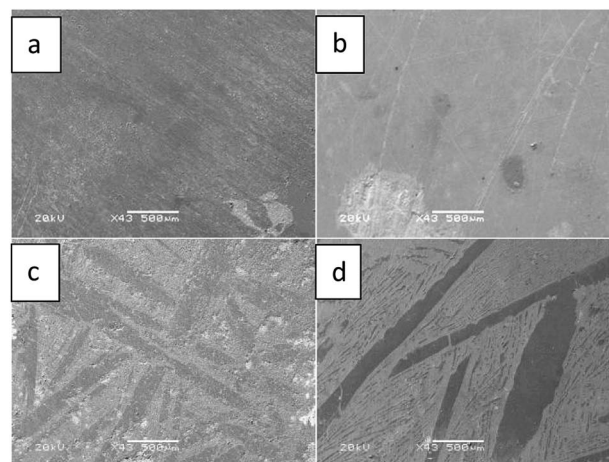
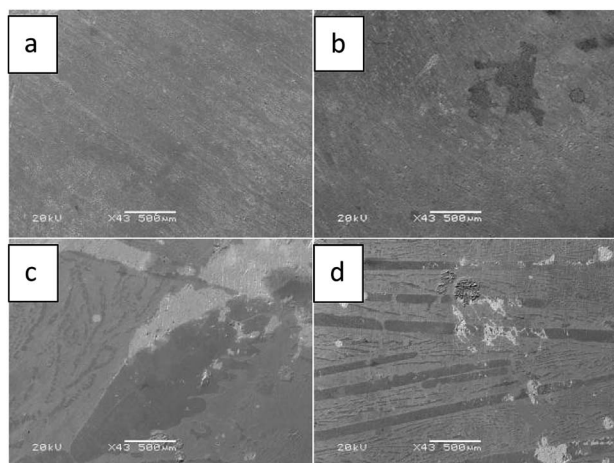
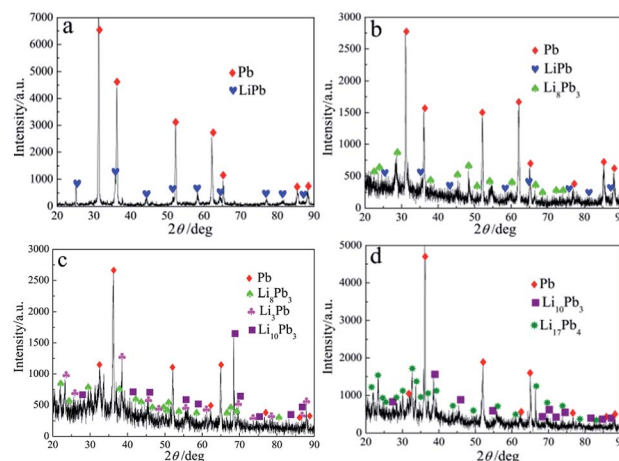
While the applied potential increased to -2.14 V , two Li-Pb compounds, LiPb and Li_8Pb_3 , were formed shown in Fig. 7(b). Thus, the peak/plateau II in Table 3 is related to the deposition of Li_8Pb_3 . Since the reduction potentials of peaks/plateaus III and IV shown in Fig. 6 are very close, the potentiostatic electrolysis was conducted at -2.28 V for 6 h . Two new Li-Pb compounds, Li_3Pb and $\text{Li}_{10}\text{Pb}_3$, are characterized by XRD (see Fig. 7(c)). Thus, the two peaks/plateaus III and IV shown in Fig. 6 should pertain to the formation of Li_3Pb and $\text{Li}_{10}\text{Pb}_3$, respectively. When applied potential at -2.43 V , $\text{Li}_{17}\text{Pb}_4$ was prepared shown in Fig. 7(d), which reveal the peak/plateau of IV shown in Fig. 6 pertaining to the deposition of $\text{Li}_{17}\text{Pb}_4$ compound. The results of potentiostatic electrolysis show that different Li-Pb compounds could be formed by controlling deposition potential. Because Li atom is too light to be determined by EDS (energy dispersive spectrometry) mapping analysis the microstructure of the Li-Pb alloys attained by potentiostatic electrolysis at different applied potential was detected by SEM shown in Fig. 8. It is clear that distinct regions are observed from the surface of the sample, which indicates the formation of different Li-Pb phases.

In order to investigate the preparation of Li-Pb intermetallic compounds, galvanostatic electrolysis was also carried out on liquid Pb electrode at 1.0 A . On the basis of the result of chronopotentiogram, 1.0 A is far larger than the deposited current of metallic Li, thus, the Li-rich Li-Pb compounds could be prepared with prolonging the run time. Fig. 9 illustrates the SEM image of Li-Pb alloys gained by galvanostatic electrolysis at 1.0 A for 4 h , 6 h , 7 h and 9 h , respectively. Two different zones,



Table 3 Summary of the peak or plateau potentials obtained by CV, SWV and CP in the LiCl–KCl–PbCl₂ melts at 723 K

| Peak or plateau | I (LiPb) | II (Li ₈ Pb ₃) | III (Li ₃ Pb) | IV (Li ₁₀ Pb ₃) | V (Li ₁₇ Pb ₄) |
|-----------------|----------|---------------------------------------|--------------------------|--|---------------------------------------|
| CV/V | −2.03 | −2.14 | −2.21 | −2.25 | −2.43 |
| SWV/V | −2.01 | −2.13 | −2.22 | −2.25 | −2.42 |
| CP/V | −2.14 | −2.14 | −2.23 | −2.28 | −2.43 |

**Fig. 7** XRD patterns of deposits attained on tungsten electrode (0.322 cm²) in LiCl–KCl–PbCl₂ melts by potentiostatic electrolysis at (a) −2.04 V, (b) −2.14 V, (c) −2.28 V and (d) −2.43 V for 6 h.**Fig. 9** SEM images of the surface of Li–Pb alloys prepared on liquid Pb electrode in LiCl–KCl–PbCl₂ melts by galvanostatic electrolysis at 1.0 A for (a) 4 h, (b) 6 h, (c) 7 h, and (d) 9 h.**Fig. 8** SEM images of the surface of Li–Pb alloys prepared by potentiostatic electrolysis at (a) −2.04 V, (b) −2.14 V, (c) −2.28 V and (d) −2.43 V for 6 h on tungsten electrode (0.322 cm²) in LiCl–KCl–PbCl₂ melts.**Fig. 10** XRD patterns of the samples obtained on a liquid Pb electrode in LiCl–KCl–PbCl₂ melts by galvanostatic electrolysis at 1.0 A for (a) 4 h, (b) 6 h, (c) 7 h and (d) 9 h at 723 K.

a light zone and dark one, are observed, which correspond to the formation of different phases.

The results of XRD patterns illustrate that galvanostatic electrolysis was operated at 1.0 A for 4 h, only one Li–Pb intermetallic compound LiPb is produced shown in Fig. 10(a). While prolonging the electrolysis time to 6 h, two compounds of LiPb

and Li₈Pb₃ can be prepared (shown in Fig. 10(b)). With prolonging the deposition time, Li₃Pb, Li₁₀Pb₃ and Li₁₇Pb₄ phases are formed and shown in XRD patterns of Li–Pb samples (Fig. 10(c) and (d)). The results indicate that five Li–Pb compounds and the eutectic Li–Pb alloy can be produced by galvanostatic electrolysis at 1.0 A by controlling electrolysis time.



Conclusions

The electrode reaction of Pb(II) was investigated on an inert tungsten electrode in LiCl–KCl eutectic melts using CV and SWV. The results displayed that the reduction of Pb(II) is a one-step reversible process with the exchange of two electrons. The diffusion coefficients were computed using the Berzins–Delahay equation in the temperature range from 663 K to 753 K, and the activation energy for diffusion of Pb(II) was found to be 30.48 kJ mol^{−1}. Using the linear polarization method, the kinetic parameters, such as exchange current intensity (j_0), standard rate constant (k^0) and charge transfer resistance (R_{ct}) for the Pb(II)/Pb(0) couple were evaluated, and the temperature dependence of j_0 was investigated to evaluate the activation energy for the electrode of the Pb(II)/Pb(0) couple and it was found to be 27.32 kJ mol^{−1}. Since the activation energy for diffusion of Pb(II) is larger than the one for the electrode reaction of the Pb(II)/Pb(0) couple, the electrochemical reaction of Pb(II) is controlled by diffusion.

A depolarisation effect was observed when Li(I) was reduced on pre-deposited Pb metal proving that five Li–Pb intermetallics are formed by co-reduction of Li(I) and Pb(II) using various electrochemical techniques. The Li–Pb alloys were generated by potentiostatic electrolysis at different deposition potentials on tungsten electrode and galvanostatic electrolysis on liquid Pb electrode. The five intermetallics, LiPb, Li₈Pb₃, Li₃Pb, Li₁₀Pb₃ and Li₁₇Pb₄ characterized by XRD and SEM, could be produced, which proves the feasibility to selectively prepare Li–Pb alloys by controlling deposition potentials and time.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

The work was financially supported by the National Natural Science Foundation of China (11675044, 11575047 and 21790373), the Major Research plan of the National Natural Science Foundation of China (91326113 and 91226201) and the Fundamental Research funds for the Central Universities (HEUCFP201849).

References

- 1 E. Mas de les Valls, L. A. Sedano, L. Batet, I. Rikapito, A. Aiello, O. Gastaldi and F. Gabriel, *J. Nucl. Mater.*, 2008, **376**, 353–357.
- 2 C. H. Wu, *J. Nucl. Mater.*, 1983, **114**, 30–33.
- 3 P. Hubberstey, T. Sample and M. G. Barker, *J. Nucl. Mater.*, 1992, **191–194**, 283–287.
- 4 S. Fukada, M. Okada, Y. Edao, H. Okitsu and S. Yoshimura, *Fusion Sci. Technol.*, 2013, **64**, 636–640.
- 5 F. Reiter, *Fusion Eng. Des.*, 1991, **14**, 207–211.
- 6 P. Fauvet and J. Sannier, *J. Nucl. Mater.*, 1988, **155–157**, 516–519.
- 7 Y. C. Chan and E. Veleckis, *J. Nucl. Mater.*, 1984, **122–123**, 935–940.
- 8 R. Schumacher and A. Weiss, *Ber. Bunsenges. Phys. Chem.*, 1990, **94**, 684–691.
- 9 A. Aiello, A. Ciampichetti and G. Benamati, *Fusion Eng. Des.*, 2006, **81**, 639–644.
- 10 H. Katsuta, H. Iwamoto and H. Ohno, *J. Nucl. Mater.*, 1985, **133–134**, 167–170.
- 11 A. Pozio, M. Carewska, A. Santucci and S. Tosti, *Int. J. Hydrogen Energy*, 2017, **42**, 1053–1062.
- 12 V. N. Lesev and V. A. Sozayev, *Bull. Russ. Acad. Sci.: Phys.*, 2016, **80**, 1367–1371.
- 13 B. A. Pint and K. A. Unocic, *J. Nucl. Mater.*, 2013, **442**, S572–S575.
- 14 U. Jain, A. Mukherjee, S. Sonak, S. Kumar, R. Mishra and N. Krishnamurthy, *Fusion Eng. Des.*, 2014, **89**, 2554–2558.
- 15 S. Terlicka, A. Dębski and W. Gąsior, *J. Mol. Liq.*, 2018, **249**, 66–72.
- 16 J. Saar and H. Ruppertsberg, *J. Phys. F: Met. Phys.*, 1987, **17**, 305–314.
- 17 M. L. Saboungi, J. Marr and M. Blander, *J. Chem. Phys.*, 1978, **68**, 1375–1384.
- 18 W. Gąsior and Z. Moser, *J. Nucl. Mater.*, 2001, **29**, 77–83.
- 19 J. Wang, P. King and R. A. Huggins, *Solid State Ionics*, 1986, **20**, 185–189.
- 20 L. Pascual, M. I. Barrena, J. M. Gomez de Salazar, A. Soria, M. Fernández, E. Conde and J. Quinones, *Fusion Eng. Des.*, 2014, **89**, 1269–1273.
- 21 T. Iida, T. Nohira and Y. Ito, *Electrochim. Acta*, 2003, **48**, 1531–1536.
- 22 M. Gibilaro, L. Massot, P. Chamelot and P. Taxil, *Electrochim. Acta*, 2009, **55**, 281–287.
- 23 Y. L. Liu, L. Y. Yuan, G. A. Ye, K. Liu, L. Zhu, M. L. Zhang, Z. F. Chai and W. Q. Shi, *Electrochim. Acta*, 2014, **147**, 104–113.
- 24 M. Li, Q. Q. Gu, W. Han, Y. D. Yan, M. L. Zhang, Y. Sun and W. Q. Shi, *Electrochim. Acta*, 2015, **167**, 139–146.
- 25 L. X. Luo, Y. L. Liu, N. Liu, L. Wang, L. Y. Yuan, Z. F. Chai and W. Q. Shi, *Electrochim. Acta*, 2016, **191**, 1026–1036.
- 26 M. Li, B. Liu, N. Ji, Y. Sun, W. Han, T. Jiang, S. Peng, Y. D. Yan and M. L. Zhang, *Electrochim. Acta*, 2016, **193**, 54–62.
- 27 Y. C. Wang, M. Li, W. Han, M. L. Zhang, T. Jiang, S. M. Peng and Y. D. Yan, *J. Alloys Compd.*, 2017, **695**, 3484–3494.
- 28 M. Jafarian, I. Danaee, A. Maleki, F. Gobal and M. G. Mahjani, *J. Alloys Compd.*, 2009, **478**, 83–88.
- 29 R. J. Heus and J. J. Egan, *J. Electrochem. Soc.*, 1960, **107**, 824–828.
- 30 G. M. Haarberg, L. E. Owe, B. Qin, J. Wang and R. Tunold, *ECS Trans.*, 2012, **50**, 215–219.
- 31 M. L. Zhang, L. J. Chen, W. Han, Y. D. Yan and P. Cao, *Trans. Nonferrous Met. Soc. China*, 2012, **22**, 711–716.
- 32 H. A. Laitinen and H. C. Gaur, *Anal. Chim. Acta*, 1958, **18**, 1–13.
- 33 G. M. Haarberg, T. Støre and R. Tunold, *Electrochim. Acta*, 2012, **76**, 256–261.
- 34 R. B. Stein, *J. Electrochem. Soc.*, 1959, **106**, 528.



- 35 Y. Castrillejo, S. Palmero, M. A. Garcia, L. Deban and P. Sanchez Batanero, *Electrochim. Acta*, 1996, **41**, 2461–2468.
- 36 L. Ramaley and M. S. Krause, *Anal. Chem.*, 1969, **41**, 1362–1365.
- 37 J. J. O'Dea, J. Osteryoung and R. A. Osteryoung, *J. Chem. Phys.*, 1983, **87**, 3911–3918.
- 38 J. J. O'Dea, K. Wikiel and J. Osteryoung, *J. Chem. Phys.*, 1990, **94**, 3628–3636.
- 39 S. P. Kounaves, J. J. O'Dea and P. Chandrasekhar, *Anal. Chem.*, 1986, **58**, 3199–3202.
- 40 J. H. Christle, J. A. Turner and R. A. Osteryoung, *Anal. Chem.*, 1977, **49**, 1899–1903.
- 41 J. G. Osteryoung and R. A. Osteryoung, *Anal. Chem.*, 1985, **57**, 101A–110A.
- 42 L. Ramaley and M. S. Krasue, *Anal. Chem.*, 1969, **41**, 1362–1365.
- 43 T. Berzins and P. Delahay, *J. Am. Chem. Soc.*, 1953, **75**, 555–559.
- 44 A. J. Bard and L. R. Faulkner, *Electrochemical Methods: Fundamentals and Applications*, John Wiley & Sons, New York, 2nd edn, 2001, pp. 61–80.

