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Preparation of a MoS₂/carbon nanotube composite as an electrode material for high-performance supercapacitors†

Xiaobo Chen,^a Jingguo Ding,^a Jing Jiang,^a Guoce Zhuang,^a Zhihai Zhang^a and Peizhi Yang^{*b}

MoS₂ and MoS₂/carbon allotrope (MoS₂/C) composites for use as anodes in supercapacitors were prepared via a facile hydrothermal method. In this study, we report the effects of various carbon-based materials (2D graphene nanosheet (GNS), 1D carbon nanotube (CNT), and 0D nano carbon (NC)) on the electrochemical performances. Among all nanocomposites studied, MoS₂/CNT exhibited the best electrochemical performance. Specifically, the MoS₂/CNT composite exhibits remarkable performances with a high specific capacitance of 402 F g⁻¹ at a current density of 1 A g⁻¹ and an outstanding cycling stability with 81.9% capacitance retention after 10 000 continuous charge–discharge cycles at a high current density of 1 A g⁻¹, making it adaptive for high-performance supercapacitors. The superiority of MoS₂/CNT was investigated by field emission scanning electron microscopy and transmission electron microscopy, which showed that MoS₂ nanosheets were uniformly loaded into the three-dimensional interconnected network of nanotubes, providing an excellent three dimensional charge transfer network and electrolyte diffusion channels while effectively buffering the collapse and aggregation of active materials during charge–discharge processes. Overall, the MoS₂/CNT nanocomposite synthesized by a simple hydrothermal process presents a new and promising candidate for high-performance anodes for supercapacitors.

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1. Introduction

There has been an ever-increasing and urgent demand for developing renewable energy resources and electrical energy storage technologies in response to the increasing issues of environmental pollution and energy consumption. Among various energy storage devices, supercapacitors (SCs) are considered as one of the most promising devices for energy storage due to high power density, high energy density, long lifespan, quick charging/discharging.¹ There are three factors that are most commonly considered in performance assessment of SCs: electrode material, electrolyte and SCs device architecture.² Among which, the electrode material plays an important role in determining the performance of SCs, so it is still very attractive to develop and design a novel electrode material with prominent electrochemical properties.

Two-dimensional (2D) layered MoS₂ nanosheets may perform best for applications in electrochemical energy storage

because of their large surface area and exposed active sites.^{3,4} Several recent reports have been focused on the supercapacitor performance of pure MoS₂ with diverse morphologies. For examples, Huang *et al.* reported that MoS₂ nanosheets as electrode materials exhibited a specific capacitance up to 129.2 F g⁻¹ at 1 A g⁻¹ with capacitance retention of 85.1% after 500 cycles;⁵ Wang *et al.* prepared flower-like MoS₂ which showed a specific capacitance of 168 F g⁻¹ at a current density of 1 A g⁻¹ when applied as electrode materials for electrochemical capacitors.⁶ These results indicate that MoS₂ is a promising electrode material for supercapacitors owing to its two-dimensional sheets-like morphology, which can provide large surface area for double-layer charge storage. In addition, the easy diffusion of electrolyte into the inner region (interlayers) of the MoS₂ electrode at lower scan rates can provide faradaic capacitance, which acts a good part in improved charge storage capability.⁷ These studies on the supercapacitive properties of pure MoS₂ showed only moderate performance as an electrode material due to the relatively poor electronic conductivity and stability. Therefore, the electrochemical properties of MoS₂ can be improved *via* making composites with other electroactive materials, such as carbon allotrope materials and conductive polymer to overcome the above problem.^{4,7–10} Various kinds of carbon materials have been suggested as conductivity supporters (or additives) for MoS₂, such as nano carbon black,^{10–12} graphene,^{3,4,13} and carbon nanotube.^{8,14,15} However,

^aSchool of New Energy and Electronic Engineering, Yancheng Teachers University, Yancheng, 224051, P. R. China. E-mail: chenxbok@126.com

^bKey Laboratory of Education Ministry for Advance Technique and Preparation of Renewable Energy Materials, Institute of Solar Energy, Yunnan Normal University, Kunming, 650500, P. R. China. E-mail: pzhyang@hotmail.com

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up to present, the comparative investigation about the effect of carbon supporters on the electrochemical performance for supercapacitor application is still rarely reported.

Herein, to combine all these merits and enhance the electrochemical performance of MoS_2 , a straightforward hydrothermal method has been developed in this work to synthesize MoS_2/NC , MoS_2/G and MoS_2/CNT composites through hydrothermal method. The morphology, surface areas, porous structures and electrochemical properties of as-prepared MoS_2/NC , MoS_2/G and MoS_2/CNT composites had been studied comparatively.

2. Experimental

2.1. Material synthesis

In this work, graphene nanosheets (GNSs), carbon nanotubes (CNT) and nano carbon black (NCB) are purchased from Tanfeng Tech. Inc (Suzhou, China). Other reagents are analytical grade without further purification and purchased from Zhanyun Chemical Co., Ltd., Shanghai, China.

MoS_2/C composites (MoS_2/NC , MoS_2/G and MoS_2/CNT) were prepared through hydrothermal method. Firstly, 30 mg carbon based nanomaterials (NC, G, or CNT) were firstly blended with 20 mL ethanol and ultrasonicated for 1 h to obtain a solution A (carbon based nanomaterials dispersion). Secondly, 1 mmol $\text{Na}_2\text{MoO}_4 \cdot 2\text{H}_2\text{O}$ and 5 mmol thiourea were dissolved in 60 mL of DI water and added to the above dispersion and stirred for 1 h. The reaction mixture was then transferred into a Teflon-lined autoclave, heated up to 210 °C, and maintained at this temperature for 24 h. The resultant black precipitate was centrifuged, washed three times with DI water, followed by ethanol washing, and dried under vacuum. For comparison, the pure MoS_2 was also prepared *via* the same method without carbon.

2.2. Materials characterizations

The crystallographic structure was characterized by X-ray diffraction (XRD) on an X-ray powder diffractometer (Smartlab-9 X-ray diffractometer, Rigaku, Japan) using Cu K α radiation ($\lambda = 1.5418 \text{ \AA}$). The specific surface areas (BET method) and porous structures measurements were performed in a model ASAP 2020 Micromeritics apparatus at 77 K. Field emission scanning electron microscopic (FESEM) images were carried out by Zeiss Supra 35VP, transmission electron microscopic (TEM) and high-resolution transmission electron microscopic (HRTEM) images were observed by FEI Tecnai F20.

The electrodes were obtained by mixing active material (85 wt%), acetylene black (10 wt%) and polytetrafluoroethylene (PTFE, 60 wt% dispersion in water) binder (5 wt%). Then, it was dried in hot air oven at 80 °C for 0.5 h, and pressed under the pressure of 10 MPa and finally dried at 100 °C for 12 h. Each electrode contained about 5 mg of active material.

2.3. Electrochemical measurements

The electrochemical properties were tested using three-electrode system with the working electrode, Pt counter electrode and

Hg/HgO electrode as the reference electrode in the 1 M Na_2SO_4 aqueous solution at room temperature. Cyclic voltammetry (CV) measurements were performed on a CHI660E electrochemical workstation (CHI, Shanghai China) and galvanostatic charge-discharge (GCD) curves were measurements by a Neware battery testing workstation (Neware, Shengzhen China). Electrochemical impedance spectroscopy (EIS) was performed on the electrochemical work-station (CHI660D) by applying an ac amplitude of 0.5 mV in the frequency from 10^6 to 10^{-2} Hz.

3. Result and discussion

3.1. Structure characterizations

Fig. 1 shows XRD patterns of pure MoS_2 and MoS_2/NC , MoS_2/G and MoS_2/CNT composites. From the XRD pattern of as-prepared MoS_2 nanosheets, the peaks of 14.4°, 32.7°, 39.5° and 58.3° are in good accordance with the standard values of MoS_2 (JCPDS card no. 37-1492). In terms of MoS_2/NC , MoS_2/G and MoS_2/CNT composites, a broad diffraction hump between 21° and 27° could be attributed to the (002) reflection of a graphitic structure, suggesting the existence of carbon-based materials. In addition, no characteristic peaks of other impurities have been detected, implying that the MoS_2 and MoS_2/NC , MoS_2/G and MoS_2/CNT composites are of high purity.

3.2. Morphology characterizations

The shape and morphology of the hydrothermally synthesized samples were investigated by SEM and TEM images. The distinctive FE-SEM morphologies of MoS_2 , MoS_2/NC , MoS_2/G and MoS_2/CNT are displayed in Fig. 2 and S2.† The FE-SEM micrograph of pure MoS_2 (Fig. 2(a)) suggested that few layered 2D MoS_2 hierarchical nanosheets are aggregated to form microspheres with the size of 1 μm . The MoS_2/NCs show a nanoflower structure (Fig. 2(b)). There is no agglomeration of free nano carbon particles in the composite, implying nano carbon particles are intercalated in the interlayers of the MoS_2 nanosheets. Fig. 2(c) shows the SEM image of the MoS_2/G . It can be observed that the graphene nanosheets are decorated

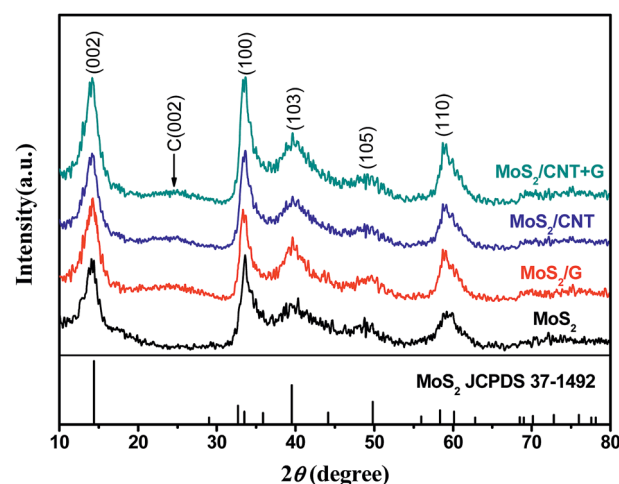


Fig. 1 XRD patterns of MoS_2 , MoS_2/NC , MoS_2/G and MoS_2/CNT .



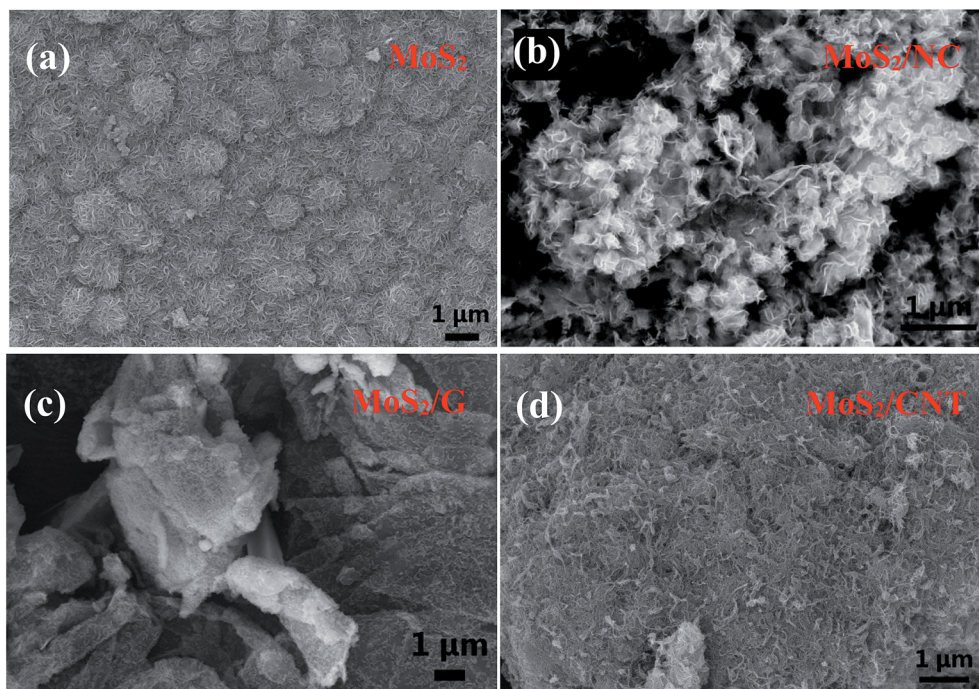


Fig. 2 FESEM images of (a) MoS_2 , (b) MoS_2/NC , (c) MoS_2/G and (d) MoS_2/CNT composites.

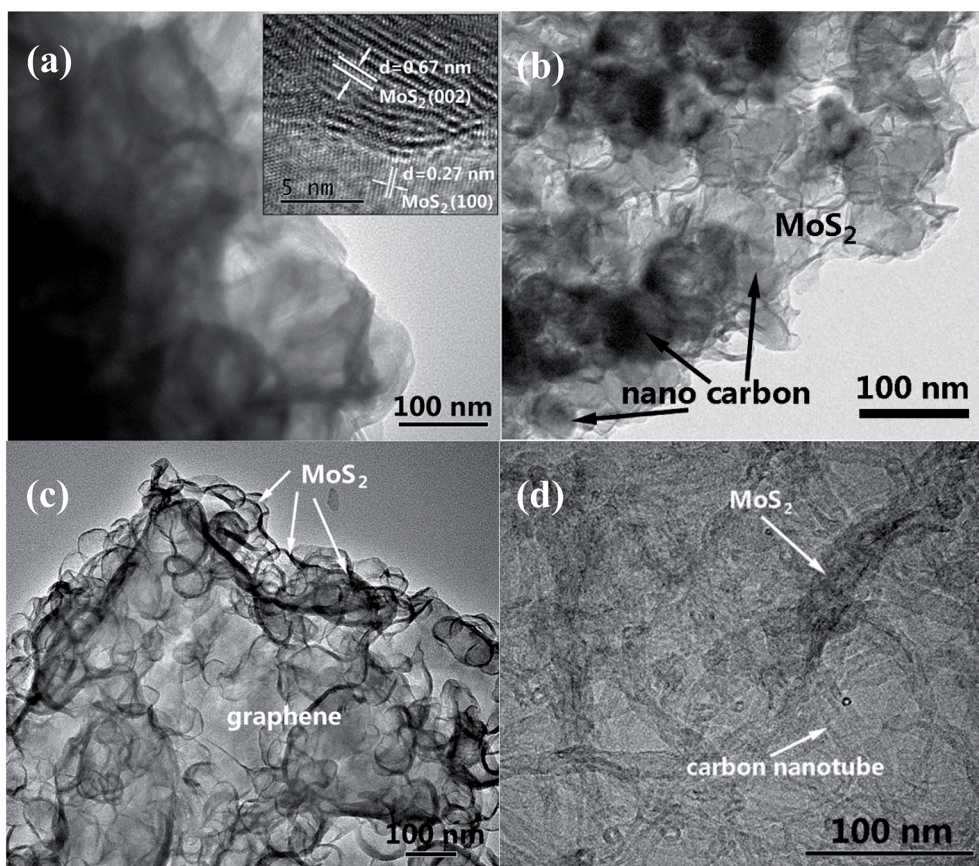


Fig. 3 TEM images of (a) MoS_2 , (b) MoS_2/NC , (c) MoS_2/G and (d) MoS_2/CNT composites.



randomly with MoS₂ nanosheets. Incorporation of CNTs can effectively reduce the stacking and yield highly separated ultrathin MoS₂ nanosheets (Fig. 2(d)), which are later determined by TEM. More importantly, the relatively long CNTs form interconnected network to facilitate the fast electron transport. It is clearly observed that CNTs interweaved with MoS₂ nanosheets. This unique architecture is anticipated to dramatically increase the electronic conductivity and maintain structural integrity thus to boost the rate performance and cycling stability in supercapacitor applications.

As shown in Fig. 3(a), the TEM image of MoS₂ forms two-dimensional nanosheets. The high-resolution TEM (HRTEM) image in the inset of Fig. 3(a) indicates the lattice figure with interlayer spacing of 0.67 nm which corresponds to the (002) plane of MoS₂.¹⁶ Also, the lattice spacing of 0.27 nm which belongs to the (100) plane of MoS₂ can be observed in the same image.¹⁷ SEM image of MoS₂/NC (Fig. 3(b)) displays that the MoS₂ shell coats nano carbon tightly. It can be also seen that GS are decorated randomly by MoS₂ nanosheets to generate MoS₂/G composite in Fig. 3(c). Fig. 4(d) shows more evidence of the extreme porous nature of the nanohybrid composed of ultrathin MoS₂ nanosheets supported/separated by CNTs.

Furthermore, the synthesized MoS₂/CNT was also characterized by EDS. From Fig. S1(a),† the signals of C, O, S and Mo elements can be seen. From the observation of the four element mappings of Mo, S, C and O (Fig. S1(b)†), the elements are uniformly distributed throughout the MoS₂/CNT composite. Therefore, the HRTEM and EDS results indicated that the MoS₂/carbon hybrid has been successfully synthesized.

Fig. 4 shows the N₂ adsorption–desorption isotherms and corresponding pore size distribution curves of samples. The isotherms of as-prepared materials all exhibit the type IV hysteresis loop as defined by IUPAC. The BET surface areas of the MoS₂, MoS₂/NC, MoS₂/G and MoS₂/CNT are 116, 219, 236 and 275 m² g^{−1}, respectively. Obviously, the surface area of MoS₂/CNT is 2.4 times as much as pure MoS₂, attributing to the compact interconnected network structure, which not only provides large surface area for charge storage but also facilitates electrolyte penetration through the mesopores. The pore size distributions of these samples are shown in the insets, confirming that the samples have mesoporous characteristics. The well-developed 3D interconnected porous network with multi-level pores can provide not only favorable transport channels for electrolyte but also strong mechanical strength. Based on the above mentioned merits, the MoS₂/CNT composite can be employed as an excellent electrode material for SCs and the detailed electrochemical measurements are performed as follows.

3.3. Electrochemical performances

Both CV and GCD measurements are carried out to survey the electrochemical properties. As shown in Fig. 5(a), the CV curves of MoS₂ and MoS₂/NC, MoS₂/G and MoS₂/CNT composites are performed within a potential window from 0.9 to 0.2 V at a scanning rate of 10 mV s^{−1} in 1 M Na₂SO₄. The measured voltammograms of MoS₂ and MoS₂/NC, MoS₂/G and MoS₂/CNT show quasi-rectangular shapes with a pair of small humps at *ca.* −0.6 V and *ca.* −0.4 V, which represent intercalation and

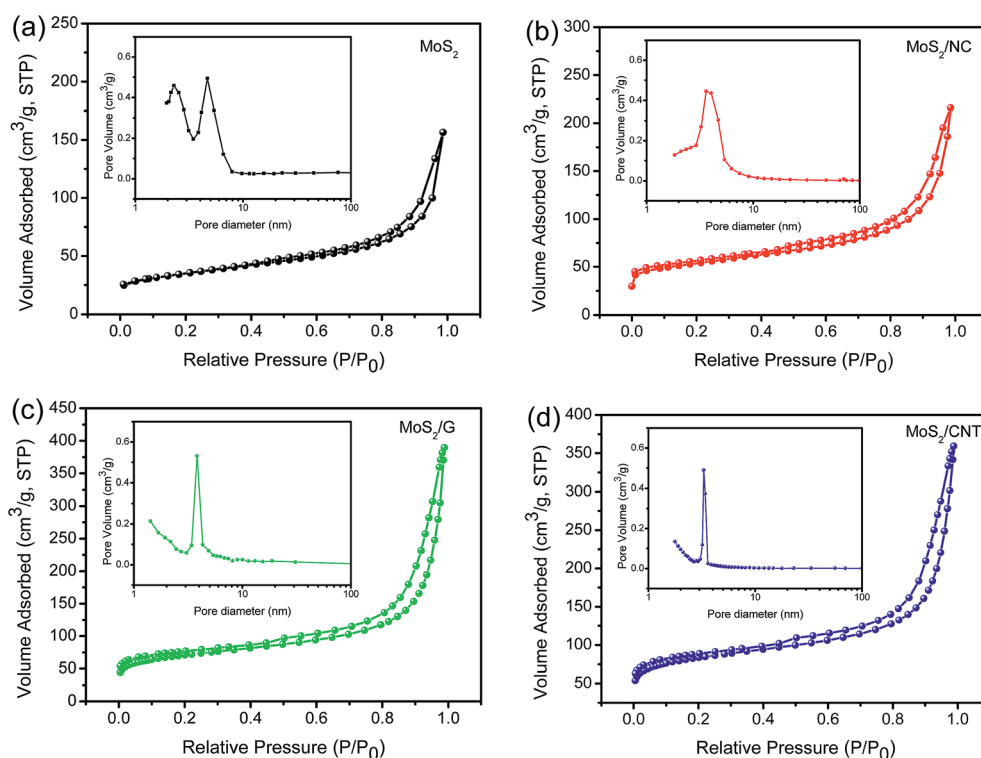


Fig. 4 The N₂ adsorption–desorption isotherms of MoS₂ (a), MoS₂/NC (b), MoS₂/G (c) and MoS₂/CNT composites (d). The insets show corresponding pore size distributions.



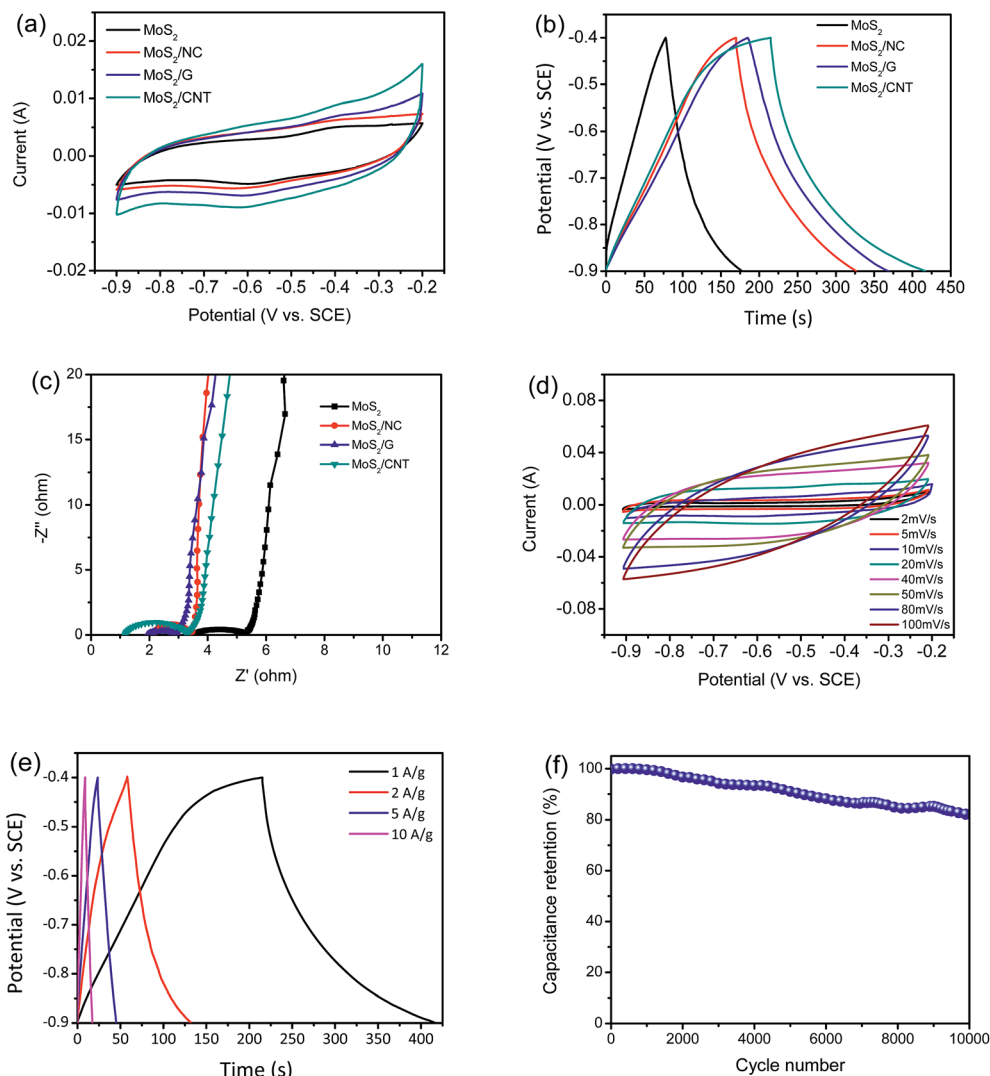
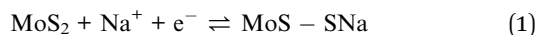


Fig. 5 Electrochemical characterizations of MoS₂, MoS₂/NC, MoS₂/G and MoS₂/CNT composites. (a) CV curves of samples at a scan rate of 10 mV s⁻¹; (b) GCD curves of samples at a current density of 1 A g⁻¹; (c) Nyquist plots; (d) CV and (e) GCD curves of MoS₂/CNT in 1 M Na₂SO₄ electrolyte at various current densities and voltage scan rates, respectively. (f) Specific capacitance of MoS₂/CNT composite versus cycle number at the current density of 1 A g⁻¹.

expulsion of Na⁺ ions between the MoS₂ layers and can be represented as follows:³



The output current of MoS₂/CNT is larger than those of MoS₂ and MoS₂/NC and MoS₂/G. One major weakness of pure MoS₂ based SCs is its inherent resistivity.^{3,4} The results indicate that the conductivity of MoS₂ can be improved by combining MoS₂ and CNT together to form composites, meanwhile, the specific capacitance also increases. Fig. 5(b) shows the GCD curves of MoS₂ and MoS₂/NC, MoS₂/G and MoS₂/CNT at a current density of 1 A g⁻¹ with voltage between -0.9 and -0.4 V. It is clear that the discharge time of MoS₂/CNT is much longer compared to MoS₂, MoS₂/NC and MoS₂/G. The specific capacitances can be calculated using the eqn (2):

$$C_s = I \times \Delta t / (m \times \Delta V) \quad (2)$$

where I is the constant discharge current (A), m is the mass of active materials in electrode (g) and Δt is the discharge time (s) in the potential window ΔV (V).¹⁸ The specific capacitances of MoS₂, MoS₂/NC, MoS₂/G and MoS₂/CNT are 194, 316, 360 and 402 F g⁻¹ at 1 A g⁻¹, respectively, which is in agreement with the result of the CV curves. This exhibits the excellently electrochemical activity of MoS₂/CNT composite. To further understand the excellent electrochemical performances of MoS₂/CNT, the EIS measurements for samples of MoS₂, MoS₂/NC, MoS₂/G and MoS₂/CNT are carried out (Fig. 5(c)). Obviously, the equivalent series resistance (R_{ESR} , the total resistance of ionic resistance of electrolyte, intrinsic resistance of substrate, and contact resistance at the interface of the active material/current collector) of the MoS₂/CNT (1.19 Ω) is smaller than those of the pure MoS₂ (3.28 Ω), MoS₂/NC (1.99 Ω), and MoS₂/G (1.37 Ω). The



Table 1 Comparison on electrochemical properties of MoS₂/carbon composites with electrolyte of 1 M Na₂SO₄^a

Electrode materials	Preparation method	Maximum specific capacitance	Electrolyte	Cyclic ability	Ref.
MoS ₂ /graphene	Solution phase exfoliation	11 mF cm ⁻² at 5 mV s ⁻¹	1 M Na ₂ SO ₄	250% at 1 mA cm ⁻² after 10 000 cycles	3
MoS ₂ /graphene aerogel	Lithium intercalation and hydrothermal assembly	268 F g ⁻¹ at 0.5 A g ⁻¹	1 M Na ₂ SO ₄	93% at 1 A g ⁻¹ after 1000 cycles	20
MoS ₂ /graphene hybrid films	Layer-by-layer assembly	282 F g ⁻¹ at 20 mV s ⁻¹	1 M Na ₂ SO ₄	93% at 2 A g ⁻¹ after 1000 cycles	21
MoS ₂ /MWCNT	Hydrothermal method	452.7 F g ⁻¹ at 1 A g ⁻¹	1 M Na ₂ SO ₄	95.8% at 1 A g ⁻¹ after 1000 cycles	8
MoS ₂ /C	Hydrothermal method	201.4 F g ⁻¹ at 0.2 A g ⁻¹	1 M Na ₂ SO ₄	94.1% at 1.6 A g ⁻¹ after 1000 cycles	10
MoS ₂ /carbon aerogel	Hydrothermal method	260 F g ⁻¹ at 1 A g ⁻¹	1 M Na ₂ SO ₄	92.4% at 1 A g ⁻¹ after 1500 cycles	22
MoS ₂ nanoflowers/3DG	Hydrothermal method	410 F g ⁻¹ at 1 A g ⁻¹	1 M Na ₂ SO ₄	113.6% at 2 A g ⁻¹ after 1000 cycles	4
MoS ₂ /CNT	Hydrothermal method	402 F g ⁻¹ at 1 A g ⁻¹	1 M Na ₂ SO ₄	99.6% at 1 A g ⁻¹ after 1000 cycles, 81.9% at 1 A g ⁻¹ after 10 000 cycles	This work

^a Abbreviations: MWCNT – multi-walled carbon nanotube; C – carbon; 3DG – three-dimensional grapheme; GS – graphene sheets; CNT – carbon nanotube.

lowest R_{ESR} value for MoS₂/CNT indicates the improved electrical conductivity of the MoS₂/CNT composite material, attributing to the short transport path of ions and charges of porous structure as well as the effective contact between MoS₂ and CNT.

CV measurement is carried out to further investigate the charge storage mechanism of MoS₂/CNT in the three electrode system. Fig. 5(d) shows CV curve area of MoS₂/CNT increases with increasing scan rates (from 2 to 100 mV s⁻¹). The CV loop of MoS₂/CNT electrode at 100 mV s⁻¹ is still close to rectangular, suggesting that the MoS₂/CNT composite can be employed as an excellent electrode material. It should be emphasized that the curves for the initial cycles at lower scan rates exhibit faradaic capacitance in addition to electric double layer capacitance, due to diffusion of the ions into the MoS₂ interlayers at low scan rates,¹⁹ improving charge storage capabilities. The specific capacitances of the MoS₂/CNT calculated from the GCD curves (Fig. 5(e)) are 402, 292, 215 and 172 F g⁻¹ at current densities of 1, 2, 5 and 10 A g⁻¹, respectively. Since such a decrease of specific capacitances is common with increasing scan rates, and caused by the difficulty of ion diffusion and the insufficient faradaic redox reaction at high scan rates.⁴

The cycle stability of MoS₂/CNT composite is important for its practical applications. The cycling performance of MoS₂/CNT at a current density of 1 A g⁻¹ was shown in Fig. 5(f). Importantly, the specific capacitance still maintains 81.9% level after 10 000 consecutive cycles. Particularly, there is no clear capacitance drop occurred over 1000 cycles (99.6% retention) due to the support between CNT and MoS₂ nanosheets, presenting excellent cycling stability of MoS₂/CNT. More interestingly, compared with other MoS₂/carbon composites with electrolyte of 1 M Na₂SO₄, MoS₂/CNT showed higher specific capacitance and more excellent electrochemical stability than those of most reported MoS₂/carbon composite electrode (Table 1). The electrochemical reversibility can be further explained by EIS measurements (Fig. S3†). Noticeably, the EIS spectra are similar in terms of the curve shape except that a moderate change of equivalent series resistance. Before and after 10 000 cycles, the R_{ESR} values of MoS₂/CNT electrode are 1.19 Ω and 1.62 Ω,

respectively, indicating that the MoS₂/CNT electrode has a relatively good stability. We further capture the SEM image of the hierarchical MoS₂/CNT composite after 10 000 cycles in Fig. S4.† The possible reason can be found in Fig. S4,† the composite after 10 000 cycles with somewhat aggregation (highlighted by red circles), thus leading to decrease of the capacity.

4. Conclusions

In summary, the MoS₂/C composites were successfully synthesized *via* a facile hydro-thermal method with carbonaceous materials (GNSs, CNTs, and CB). The as-prepared MoS₂/C composites showed superior electrochemical performances than that of bare MoS₂ counterpart when used as an anode in supercapacitors. Especially, the MoS₂/CNT exhibited more excellent than those of most reported MoS₂/carbon composite electrode. The MoS₂/CNT composite exhibited remarkable performances with a high specific capacitance of 402 F g⁻¹ at a current density of 1 A g⁻¹ and an outstanding cycling stability with 81.9% capacitance retention after 10 000 continuous charge-discharge cycles at a high current density of 1 A g⁻¹, making it desirable for high-performance supercapacitors. FESEM and TEM analysis confirmed that MoS₂ nanosheets have been loaded into the 3-D network of CNT. The 3-D network structure could buffer the volume changes of MoS₂ nanosheets at high current densities and maintain high electrical conductivity. Hence, the MoS₂/CNT composite material is a promising candidate for electrode of SCs.

Conflicts of interest

The authors declare no conflict of interest.

Acknowledgements

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References

- 1 A. Burke, *J. Power Sources*, 2000, **91**, 37–50.
- 2 J. Yan, Q. Wang, T. Wei and Z. Fan, *Adv. Energy Mater.*, 2014, **4**, 1300816.
- 3 M. A. Bissett, I. A. Kinloch and R. A. W. Dryfe, *ACS Appl. Mater. Interfaces*, 2015, **7**, 17388.
- 4 T. Sun, Z. Li, X. Liu, L. Ma, J. Wang and S. Yang, *J. Power Sources*, 2016, **331**, 180–188.
- 5 K. J. Huang, J. Z. Zhang, G. W. Shi and Y. M. Liu, *Electrochim. Acta*, 2014, **132**, 397–403.
- 6 X. Wang, J. Ding, S. Yao, X. Wu, Q. Feng, Z. Wang and B. Geng, *J. Mater. Chem. A*, 2014, **2**, 15958–15963.
- 7 G. Ma, H. Peng, J. Mu, H. Huang, X. Zhou and Z. Lei, *J. Power Sources*, 2013, **229**, 72–78.
- 8 K. J. Huang, L. Wang, J. Z. Zhang, L. L. Wang and Y. P. Mo, *Energy*, 2014, **67**, 234–240.
- 9 H. Tang, J. Wang, H. Yin, H. Zhao, D. Wang and Z. Tang, *Adv. Mater.*, 2015, **27**, 1117–1123.
- 10 L. Q. Fan, G. J. Liu, C. Y. Zhang, J. H. Wu and Y. L. Wei, *Int. J. Hydrogen Energy*, 2015, **40**, 10150–10157.
- 11 Q. Weng, X. Wang, X. Wang, C. Zhang, X. Jiang, Y. Bando and D. Golberg, *J. Mater. Chem. A*, 2015, **3**, 3097–3102.
- 12 S. Zhang, R. Hu, P. Dai, X. Yu, Z. Ding, M. Wu, G. Li, Y. Ma and C. Tu, *Appl. Surf. Sci.*, 2016, **396**, 994–999.
- 13 E. G. D. S. Firmiano, A. C. Rabelo, C. J. Dalmaschio, A. N. Pinheiro, E. C. Pereira, W. H. Schreiner and E. R. Leite, *Adv. Energy Mater.*, 2014, **4**, 1301380.
- 14 M. Chen, Y. Dai, J. Wang, Q. Wang, Y. Wang, X. Cheng and X. Yan, *J. Alloys Compd.*, 2017, **696**, 900–906.
- 15 S. Wang, J. Zhu, Y. Shao, W. Li, Y. Wu, L. Zhang and X. Hao, *Chem.–Eur. J.*, 2017, **23**, 3438–3446.
- 16 Z. Huang, L. Zhang, M. Li, W. Ran, Y. Lu, B. Yang and Z. Long, *Nanosci. Nanotechnol. Lett.*, 2017, **9**, 56–60.
- 17 X. Liu, Z. Xing, H. Zhang, W. Wang, Y. Zhang, Z. Li, X. Wu, X. Yu and W. Zhou, *ChemSusChem*, 2016, **9**, 1118–1124.
- 18 W. Qian, Z. Chen, S. Cottingham, W. A. Merrill, N. A. Swartz, A. M. Goforth, T. L. Clare and J. Jiao, *Green Chem.*, 2012, **14**, 371–377.
- 19 T. Khawula, *J. Mater. Chem. A*, 2016, **4**, 6411–6425.
- 20 M. Yang, J. M. Jeong, Y. S. Huh and B. G. Choi, *Compos. Sci. Technol.*, 2015, **121**, 123–128.
- 21 S. Patil, A. Harle, S. Sathaye and K. Patil, *CrystEngComm*, 2014, **16**, 10845–10855.
- 22 K. J. Huang, L. Wang, J. Z. Zhang and K. Xing, *J. Electroanal. Chem.*, 2015, **752**, 33–40.

