




Cite this: *RSC Adv.*, 2018, 8, 20394

# Photodegradation of spent wash, a sugar industry waste, using vanadium-doped TiO<sub>2</sub> nanoparticles†

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Waste from the sugar cane industry and alcohol distilleries is one of the sources of water pollution, and the degradation of this effluent is a very challenging task. Photocatalytic degradation can be an attractive alternative to conventional degradation processes. A vanadium-doped TiO<sub>2</sub> (V-TiO<sub>2</sub>) photocatalyst for the degradation of spent wash and industrial dyes has been studied and reported here. V-doped TiO<sub>2</sub> nanoparticles were prepared using a sol-gel method based on aqueous titanium peroxide with titanium isopropoxide as the Ti precursor and V<sub>2</sub>O<sub>5</sub> as the V precursor. In order to observe the effect of the dopant on sol-gel behaviour, physicochemical and structural properties, the concentration of V was varied between 1.0% and 5% by weight. The crystallization temperature and time were optimized to obtain the required phase of V-TiO<sub>2</sub>. The physicochemical and structural characteristics of the V-doped TiO<sub>2</sub> catalyst were determined using Brunauer-Emmett-Teller (BET), X-ray diffraction, FESEM, TEM, TG, FT-IR, Raman, PL and UV-visible spectroscopic techniques. UV-visible analysis showed a red shift in the absorption edge of TiO<sub>2</sub> upon doping with V metal, which suggested an increase in the absorption of visible light due to a decrease in the effective band gap. The application potential of the V-TiO<sub>2</sub> catalyst was studied via the degradation of sugar industry waste (spent wash) and Jakofix red dye (HE 8BN) under natural sunlight, as well as a standard artificial solar energy source (Xe lamp). The highest activity was observed for a 1% V-TiO<sub>2</sub> photocatalyst for the degradation of spent wash and Jakofix red dye under natural sunlight. The degradation of coloured compounds in spent wash was monitored by gel permeation chromatography (GPC), which showed the degradation of high-molecular-weight compounds into low-molecular-weight fractions. The catalyst decomposed 90% of Jakofix red dye (HE 8BN) in 3.5 h and 65% of spent wash in 5 h under irradiation with natural sunlight, whereas Degussa P-25 TiO<sub>2</sub> was only able to decompose 35% of the dye and 20% of spent wash under identical reaction conditions. A cycling stability test showed the high stability and reusability of the photocatalyst for degradation reactions, with a recovery of around 94–96%.

Received 3rd April 2018

Accepted 15th May 2018

DOI: 10.1039/c8ra02869a

rsc.li/rsc-advances

## 1. Introduction

Waste from the sugar cane distillery industry is one of the sources of aquatic pollution, and this industry generates large volumes of high-strength waste water containing molasses, which has unmanageable properties. The intense colour of the spent wash effluent leads to acute ecological problems when it is released in untreated form into the environment. Currently, anaerobic and aerobic as well as physicochemical methods have been employed to treat this effluent. The decolourisation of effluent generated by sugar cane distilleries involves large

amounts of dark brown waste water, which is known as molasses spent wash (MSW). MSW is an undesirable residual liquid waste, and many distilleries dispose of it without treatment into nearby uncultivated land, rivers and lakes. The highly acidic pH, intense dark brown colour, high ash content, unpleasant odour and high percentage of organic and inorganic matter make it difficult to convert MSW into less toxic waste at an affordable cost, and hence industries prefer to dispose of it in untreated form. As a result, the disposal of this effluent is a critical environmental issue. The dark brown colour is due to the presence of melanoidin, caramel, alkaline degradation products and polyphenols, *etc.* These colorants are actually concentrated in molasses after the crystallization of sugar and are further transferred to the spent slurry during the fermentation of molasses. Melanoidin is a natural condensation polymer of sugars and amino acids and is produced by non-enzymatic browning reactions known as Maillard reactions. It

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† Electronic supplementary information (ESI) available. See DOI: 10.1039/c8ra02869a



decreases the penetration of sunlight into rivers and lakes, which in turn decreases both photosynthetic activity and dissolved oxygen concentrations, which affects aquatic life. Melanoidin remains at levels of about 2% in molasses-based distillery effluent as a major recalcitrant coloured pollutant together with metals and phenolics. It is hardly degraded by microbes and behaves as an anionic hydrophilic polymer that has a high tendency to bind with metal cations, which results in an effluent with a more complex nature. Moreover, phenolics present in effluent form more complex compounds with heavy metals. This effluent is widely discharged in huge amounts as an environmental pollutant by various agriculturally based industries, especially from distilleries and fermentation-based industries. Hence, its degradation prior to discharge is essential for environmental safety. Considering the intensity of the problem, many researchers have paid attention to addressing the issue of environmental pollution due to molasses. Removal of the colour of spent wash by adsorption on activated charcoal and electrocoagulation was studied by Bhise *et al.* and Satyawali *et al.*<sup>1,2</sup> Photocatalytic removal of colour and COD in distillery effluent by solar radiation was studied by Vineetha *et al.*<sup>3</sup> Decolourization of spent wash using a nano- $\text{Al}_2\text{O}_3$ /kaolin photocatalyst was studied by David *et al.*<sup>4</sup> However, the researchers only reported a change in the colour of spent wash after irradiation with artificial solar light. There might be a possibility of the adsorption of colorants on the surface of the photocatalyst, which resulted in a decrease in colour rather than actual degradation, as details of the degradation products and mechanism were not reported in this study. The degradation of molasses by a nanocrystalline composite material was studied by Navgire *et al.*<sup>5</sup> Various effective physical and chemical methods such as ozonation, flocculation and adsorption on activated carbon, *etc.*, have been investigated for the removal of colour from biomethanated distillery spent wash. Although these techniques are widely used for the degradation of spent wash and other harmful dyes, researchers are working on new and environment-friendly techniques.

The degradation of organic pollutants using a  $\text{TiO}_2$  photocatalyst has been extensively studied as a relatively new technique for pollution abatement owing to the desirable properties of the catalyst, such as non-toxicity, a wide band gap, and stability in acidic as well as basic media.<sup>6–8</sup> However, the wide band gap of  $\text{TiO}_2$  means that it only absorbs light with a wavelength of less than 400 nm in the UV region, which restricts its application to uses in the presence of UV irradiation. For extensive application, a  $\text{TiO}_2$ -based catalyst that is effective in visible radiation or solar light needs to be developed as a future-generation photocatalytic material.  $\text{TiO}_2$  absorbs only 5% of the energy of the solar spectrum, and hence numerous studies have been performed to extend its photoresponse and photocatalytic activity by modifying its surface structure, surface properties and composition to shift its absorption into the visible region so as to improve its photocatalytic activity in visible/solar light.<sup>9–11</sup> Surface modification by doping with metal ions and organic polymers has been proven to be an efficient route to improve the photocatalytic activity of  $\text{TiO}_2$ .<sup>12–15</sup> Vanadium oxide ( $\text{VO}_x$ ) supported on various supports has been widely used for the

reduction of nitrogen oxides ( $\text{NO}_x$ )<sup>16–18</sup> and selective oxidation of volatile organic compounds (VOC).<sup>19–26</sup> Anpo *et al.* have studied the doping of  $\text{TiO}_2$  with transition metals such as V, Cr and Fe by three different methods: sol-gel, co-precipitation and ion implantation techniques.<sup>27–30</sup> The higher photocatalytic activity of vanadia-doped  $\text{TiO}_2$  prepared by ion implantation was related to the deep incorporation of vanadia into the titanium oxide lattice due to the bombardment of  $\text{TiO}_2$  targets with highly energetic vanadium ions. Vanadium as a dopant extends the absorption edge of  $\text{TiO}_2$  to the visible-light region ( $>400$  nm).<sup>12,31,32</sup> This extension is achieved by forming intermediate energy levels. Vanadium as a dopant is also known to promote the phase transformation of  $\text{TiO}_2$  from anatase to rutile.<sup>33</sup> Pure  $\text{TiO}_2$  undergoes an anatase-to-rutile phase transformation in the range of 400–1200 °C, depending on the preparation method.<sup>34</sup> Doping with vanadium can decrease the transition temperature by as much as 450 °C (from 1100 to 650 °C).<sup>35</sup> In this study, we report the synthesis of vanadium-doped mesoporous anatase  $\text{TiO}_2$  materials. In an attempt to alter the optical properties of  $\text{TiO}_2$ , doping of thin titania films with Fe and Au was successfully demonstrated in our earlier work to shift the absorption of  $\text{TiO}_2$  into the visible region and improve the photocatalytic activity of  $\text{TiO}_2$  under sunlight.<sup>36–38</sup> Thin films of vanadia-doped titania were deposited by simple dip-coating techniques using vanadium and titanium peroxide gel on various glass substrates and used for the photocatalytic degradation of MB and formaldehyde. In continuation of our efforts to study the application potential of V- $\text{TiO}_2$  for the degradation of other harmful substances, we used V- $\text{TiO}_2$  for the degradation of spent wash, *i.e.*, waste from the sugar distillery industry, and Jakofix red (HE 8BN) dye. The electrochemical decolourization and biodegradation of an electrochemically decolourised dye were studied by Sathishkumar *et al.*<sup>39</sup> and Yadav *et al.*<sup>40</sup> Treatment of distillery spent wash by ozonation and biodegradation and the significance of pH reduction and the removal of inorganic carbon were reported by Kumar *et al.*<sup>41</sup> The use of an AFBBR and an adsorption technique for the treatment of spent wash colour was studied by Lakshmikanth *et al.*<sup>42</sup> Although many processes have been employed for the decolourization of spent wash, each process has its own advantages and disadvantages. Many processes are costly and economically unviable. The search for new and effective methods is under way, and many researchers are working in this direction. Considering the above factors, we attempted to use V- $\text{TiO}_2$  for the degradation of spent wash under natural sunlight reaching the earth's surface.

In this paper, the preparation of a V- $\text{TiO}_2$  catalyst by a non-polluting sol-gel method based on an aqueous peroxide is reported. Samples were characterized by XRD, FE-SEM, TEM, TG, FT-IR, Raman, PL and UV-vis absorption spectroscopy. The photocatalytic activity of sol-gel vanadium-doped  $\text{TiO}_2$  for the degradation of spent wash and Jakofix red dye (HE 8BN) was studied, and the results are reported here. Degradation trials conducted on the laboratory scale showed very promising results and suggested that this photocatalyst is a potential candidate for this application. Although the results that are reported are valid on the laboratory scale, the experimental data that were generated can be used for actual field trials. In field



trials the conditions are different, and the reaction parameters therefore need to be optimized again by considering the local conditions. In laboratory-scale experiments, diluted spent wash was used for degradation reactions under natural sunlight rather than an artificial energy source. This method is only effective for diluted samples of spent wash, as the penetration of sunlight may be restricted by the intense colour of concentrated spent wash, which ultimately affects the rate of the degradation reaction. This is the only limitation of this photocatalytic method and can be overcome by recycling the water discharged from a waste water treatment plant and other sources. Water from different sources within a distillery, such as water discharged from the effluent treatment plant, waste water remaining after the cleaning of sugar cane, and water used for cooling, can be reused for the dilution of spent wash to make the process more professional and feasible. The main advantages of this method over other methods are its negligible or no requirement for external energy, low cost and high stability of the photocatalyst, use of abundantly available sunlight and generation of less harmful products with no environmental pollution. In addition, a much smaller amount of the photocatalyst is required for this process, and it can be reused for many degradation cycles. The main advantage of our method is that it is very cheap and more cost-effective than other methods that have been reported and are currently in use. These qualities make the process more attractive and economically viable than other methods already employed for the degradation of spent wash.

## 2. Experimental methods

### 2.1. Materials and reagents

The chemicals and raw materials were used as received without further treatment/purification. Titanium(IV) tetraisopropoxide ( $\text{Ti}(\text{OCH}(\text{CH}_3)_2)_4$ ) was from Acros Organics, and 30% (w/w) hydrogen peroxide ( $\text{H}_2\text{O}_2$ ) and vanadium pentoxide ( $\text{V}_2\text{O}_5$ ) were from Merck India Ltd. Milli-Q water was used (conductivity of  $<10 \Omega^{-1} \text{ cm}^{-1}$ ). P-25  $\text{TiO}_2$  (Degussa) was imported and supplied by a local supplier, namely, M. M. Suppliers, Pune, India.

### 2.2. Preparation of catalysts

A series of V-doped  $\text{TiO}_2$  catalysts with V contents ranging from 1 to 2, 3, 4 and 5 wt% were prepared by an aqueous sol-gel technique using vanadium pentoxide as the vanadium(V) precursor and titanium isopropoxide as the titanium precursor. In a typical synthesis, 4.02 g of titanium isopropoxide (Aldrich) was hydrolysed using 30 mL of Milli-Q water (conductivity of  $<10 \Omega^{-1} \text{ cm}^{-1}$ ). Aqueous hydrogen peroxide (30%, Merck) was added to produce a transparent orange sol of titanium peroxide. Vanadium pentoxide (71.90 mg for 1.0% V- $\text{TiO}_2$ , and so on) ( $\text{V}_2\text{O}_5$ ) (Merck) was suspended in 20 mL of Milli-Q water, and 3 mL of 30% aqueous hydrogen peroxide was added to give a clear red solution of peroxovanadic acid. This solution was added to the titanium peroxide solution with stirring for 30 min. The mixture of both sols was slowly converted into

a transparent orange-yellow viscous gel. The vanadia-titanium peroxide gel was dried at ambient temperature to give a powder. The dry powder was then heated in a hot air oven at  $110^\circ\text{C}$ , followed by calcination at  $400^\circ\text{C}$  under a static air flow in a muffle furnace. The heating/cooling rate was  $5^\circ\text{C min}^{-1}$ , with a dwell time of 4 h at the selected temperature. Pure  $\text{TiO}_2$  was also prepared similarly by the sol-gel technique using a peroxide precursor for comparison.

### 2.3. Characterization techniques

An X-ray diffraction (XRD) study was performed using a Bruker D8 Advance AXS diffractometer ( $\text{Cu K}\alpha$  radiation,  $\lambda = 1.5406 \text{ \AA}$ ) with an operating voltage of 40 kV and a current of 44 mA. Data were collected in the  $2\theta$  range of  $20\text{--}80^\circ$  with a step size of  $0.028^\circ$  and a counting time of 15 s for each step. Diffuse-reflectance UV-vis spectra (DRS) were recorded in the range of 200–800 nm with a spectral resolution of 0.5 nm in air at ambient temperature using a Shimadzu UV-3600 spectrophotometer. Powder samples were mixed with barium sulphate powder (NIST traceable) and pressed into the sample holder for UV-visible measurements. Raman spectra were recorded using a Raman microscope (Renishaw 2000) using a wavelength of 532 nm, an exposure time of 16 seconds, and an accumulation number of 4. FT-IR spectra of samples were recorded using a PerkinElmer Spectrum Two U-ATR instrument. Thirty-two scans were performed with a spectral resolution of  $4 \text{ cm}^{-1}$  in the range of  $4000\text{--}450 \text{ cm}^{-1}$ . The thermal stability and behaviour of samples were analysed using a TG/DTA instrument (Mettler Toledo) with a heating rate of  $50^\circ\text{C min}^{-1}$  in an air atmosphere. The surface areas of samples were determined by the Brunauer–Emmett–Teller (BET) method using a Micromeritics Gemini VII analyzer. The pore size distribution was calculated *via* the Barrett–Joyner–Halenda (BJH) method from the desorption isotherm. Prior to the measurements, the powder samples were degassed using UHP grade nitrogen gas at  $150^\circ\text{C}$  for 2 h. The surface morphology and microstructure were analysed using an FESEM instrument. For these measurements, powder samples were initially sonicated in ethanol using an ultrasonic bath and then drop-cast onto carbon tape. HRTEM measurements were performed with a JEOL instrument operated at an accelerating voltage of 300 kV. Samples for HRTEM analysis were prepared by vacuum-drying drop-cast solutions of  $\text{TiO}_2$  nanoparticles on carbon-coated copper grids. Image processing and calculations of interplanar distances ( $d$ ) were performed with the help of Gatan Micrograph software. TOC measurements were performed with a Shimadzu TOC-VSCH analyser. Powder samples of equal weights were used for PL measurements (Horiba Fluorolog-3 spectrofluorometer). The widths of the entrance and exit slits were kept the same.

## 3. Photocatalytic study

### 3.1. Photocatalytic decomposition of spent wash (SW) and Jakofix HE 8BN dye

The photocatalytic activity of the prepared photocatalyst was tested in the degradation of industrial waste, namely, spent



wash (alcohol-based sugar industry) and Jakofix red (HE 8BN) dye (textile industry) using a 100 mL quartz reactor with a water circulation arrangement to maintain a constant temperature. Solutions of spent wash ( $10 \text{ g L}^{-1}$  in water) and Jakofix dye (100 ppm) were used as the contaminants. A suspension was prepared by mixing 0.025 g V-TiO<sub>2</sub> photocatalyst with 100 mL of spent wash (10 000 ppm) or Jakofix dye (100 ppm). Before exposure to light, the suspension was kept in the dark for 1 h to attain adsorption-desorption equilibrium. For the photocatalytic degradation of spent wash and Jakofix red dye, powder samples of V-TiO<sub>2</sub> were used as a photocatalyst. The suspension was then irradiated in ambient conditions with natural solar light during sunny days (from March to May) between 10 am and 4 pm at Pune, which is located in Maharashtra state, India, for stipulated times. At regular intervals, samples were collected and allowed to settle for 20 min to separate the photocatalyst particles, and the supernatant solution was analyzed using a UV-visible spectrophotometer (Cary 100). Prior to the photocatalytic measurements, the UV-visible instrument was calibrated with solutions of spent wash and Jakofix dye of known concentrations.

### 3.2 Adsorption and recovery test

A 25 mg sample of a solution of Jakofix red dye (HE 8BN) was used to determine the adsorption ratio in dark conditions. The adsorption ratio was calculated from the ratio of the concentration of the dye after adsorption ( $C$ ) to the original concentration of the dye ( $C_0$ ). For the recovery test, 100 mg 1% V-TiO<sub>2</sub> catalyst was mixed with 25 mL of the 100 ppm dye solution and kept in the dark for 1 h with continuous stirring. The suspension was then irradiated for 3.5 h and centrifuged, and the supernatant liquid was removed by decantation. The photocatalyst was then washed with 25 mL ethanol and separated by centrifugation for 10 min at 5000 rpm. The separated photocatalyst was then dried at 100 °C in an oven and weighed using an electronic balance. The same process was repeated for second and third cycles. The amount of catalyst (mg) recovered after each cycle was plotted *versus* the number of use cycles to get an idea of the loss and recovery of the photocatalyst after each cycle.

### 3.3 Stability test

The cycling stability of the photocatalyst was tested by repeating the photocatalytic degradation of Jakofix red dye (HE 8BN) several times under identical conditions. In the first cycling experiment, 25 mg V-TiO<sub>2</sub> photocatalyst was suspended in the 100 ppm dye solution, and the reaction mixture was irradiated for 3 h with natural sunlight. The degradation products were analysed by monitoring the reaction using a UV-visible spectrophotometer. The photocatalyst was separated by centrifugation and dried at 100 °C before reuse for the second cycling experiment. Before the second cycling experiment, an airtight photoreactor was kept in the dark and de-aerated by purging with high-purity N<sub>2</sub> gas. The dye solution together with the photocatalyst was added to the photoreactor and again irradiated with solar light. The same procedure was also repeated for

the third cycling experiment. After the third cycle, the photocatalyst was recovered and dried and its crystal phase was analysed using an XRD technique, as well as its pore size and surface area using the BET technique.

## 4. Results

### 4.1 Structural analysis using X-ray diffraction (XRD)

The gel was allowed to dry in an open atmosphere at ambient temperature and used for XRD analysis. The dried gel was further heated at 100 °C in an oven overnight and finally calcined at various temperatures in the range of 200–400 °C. The results showed good agreement with the calculated values based on the weights of vanadium and titanium during the preparation. The XRD patterns of the as-prepared dry 1% V-TiO<sub>2</sub> gel and gel samples calcined at 200, 300 and 400 °C for 4 h are shown in Fig. 1 (curves a–d). As expected, the XRD pattern of the as-prepared V-TiO<sub>2</sub> sample is a horizontal line with no sign of any peaks (curve a), which may be due to the amorphous nature of the air-dried V-TiO<sub>2</sub> gel.

Curve b in the patterns corresponds to the V-TiO<sub>2</sub> gel calcined at 200 °C. This pattern displays some very weak peaks at the peak positions of anatase TiO<sub>2</sub>, which indicates the beginning of crystallization of V-TiO<sub>2</sub> at this temperature and a slight rearrangement of atoms into a crystalline structure. On the further increase in the calcination temperature to 300 °C, curve c displays an increase in the intensity of the characteristic peaks of the anatase phase, which suggests the further growth of the anatase phase. The sample calcined at 400 °C (curve d) exhibited peaks of a fully grown anatase phase. From the XRD patterns, it is observed that the crystallization temperature of the anatase phase of TiO<sub>2</sub> in V-TiO<sub>2</sub> was lowered in comparison with that for pure TiO<sub>2</sub>.

The effect of the V content on the crystallization behaviour of doped samples was studied by varying the concentration of V added to TiO<sub>2</sub>. The XRD patterns of pure TiO<sub>2</sub> and V-TiO<sub>2</sub> samples containing different concentrations of vanadium are shown in Fig. S1 of the ESI.† In Fig. S1,† curves a to c of the XRD patterns closely match that of the anatase phase; however,

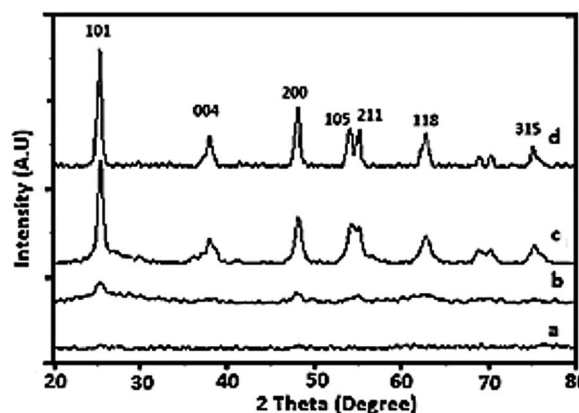


Fig. 1 XRD patterns of (a) as-prepared 1% V-TiO<sub>2</sub> gel and gels calcined at (b) 200, (c) 300 and (d) 400 °C.





curves d and e display some very weak peaks of the rutile phase together with those of the anatase phase. This suggests that samples containing less than 3% vanadium did not experience any interference from V with the crystal phase. However, the samples containing more than 3% vanadium exhibited a few weak peaks of rutile together with those of the anatase phase, which suggests that an increase in the V content catalyses the formation of the rutile phase of  $\text{TiO}_2$  at a much lower temperature than in pure titanium peroxide. In fact, in pure titanium peroxide the anatase-to-rutile transformation starts at a temperature of 700 °C or higher, whereas V- $\text{TiO}_2$  samples containing more than 3% vanadium displayed a similar transition at a much lower temperature. In addition, one interesting phenomenon was observed during the crystallization of samples containing more than 1% vanadium. Careful observation of curves a to f shows that there was an increase in the intensity of the (101) peak for the 1% V- $\text{TiO}_2$  sample, but the samples containing more than 1% vanadium displayed a decreasing trend in the intensity of the (101) anatase peak, which suggests that a higher vanadium concentration slows down the crystallization of the anatase phase and promotes the direct conversion of titanium peroxide into the rutile phase. In general, in photocatalytic activity the crystal phase as well as the crystallinity is a very important parameter for higher activity. This may be the reason why the photocatalytic activity of the 1% V- $\text{TiO}_2$  sample was higher than those of the other V- $\text{TiO}_2$  samples.

The effects of the vanadium content on the surface area and porosity were studied by measuring the surface area and porosity of V- $\text{TiO}_2$  samples containing 1–5% vanadium. All these samples were calcined at 400 °C prior to the surface area measurements. The porosity of the samples was measured according to the ASTM method ASTM-D4284-12(2017)e1, *i.e.*, by a mercury intrusion porosimetry technique.<sup>43</sup> The results for the BET surface area and porosity are summarized in Table 1.

The results of the measurements of surface area and porosity in Table 1 are quite interesting, as bare  $\text{TiO}_2$  had a surface area of 72.59  $\text{m}^2 \text{g}^{-1}$ , whereas vanadium-doped  $\text{TiO}_2$  displayed a steady increase in surface area with the vanadium content until this reached 4% (4% V- $\text{TiO}_2$ ). Interestingly, the surface area of 5% V- $\text{TiO}_2$  decreased in comparison with 4% V- $\text{TiO}_2$ . The increase in surface area may be attributed to the formation of a homogeneous gel until the dopant concentration reached 4%. In this V- $\text{TiO}_2$  gel system, when a vanadium peroxide sol is added to a titanium peroxide solution at a  $\text{V}_2\text{O}_5$  content of less

than 4% it forms a homogeneous greenish gel, but as the vanadium content is increased beyond 4% the characteristics of the gel change and an inhomogeneous gel with agglomerated flocks is formed. In general, in sol-gel synthesis the nature of the gel is very important, as the pores in the gel are filled with solvent molecules and, upon careful removal of these solvent molecules, the porous structure remains undisturbed, which ultimately helps to increase the surface area. In the case of a vanadium dopant content of up to 4% the gel network is perfect, and hence there is an increase in surface area, but as the vanadium percentage is increased beyond this the gel network becomes disturbed with the formation of V- $\text{TiO}_2$  flocks. These flocks are denser than the gel and tend to separate from the solvent *via* an agglomeration process. The agglomerated flocks are less porous in comparison with the undisturbed gel, which ultimately results in a decrease in surface area in comparison with the  $\text{TiO}_2$  samples with lower vanadium contents.

Nitrogen adsorption-desorption isotherms for (a) pure  $\text{TiO}_2$ , (b) as-prepared 1% V- $\text{TiO}_2$  and (c) a sample of 1% V- $\text{TiO}_2$  recovered after the third cycle of the photodegradation reaction are shown in Fig. 2. The isotherms were recorded at 77 K with a Micromeritics Gemini VII 2390 instrument. The isotherm (a–c) is classified as a type IV adsorption isotherm with a hysteresis loop. As seen in Fig. 2C, the sample displayed a late adsorption edge ( $P/P_0 > 0.8$ ), which suggests an increase in pore size. In general, the surface area depends on the particle size, pore size, number and type of pores, and distribution of pores, together with voids. A smaller pore size and smaller particles implies a higher surface area. The surface area also increases if the particles contain pores. In our case, the surface area of the sample of 1% V- $\text{TiO}_2$  recovered after the third cycle decreased to 89.21  $\text{m}^2 \text{g}^{-1}$  from the value of 95.04  $\text{m}^2 \text{g}^{-1}$  for as-prepared 1% V- $\text{TiO}_2$ . The results are in agreement with reported observations.<sup>44</sup>

These samples were further characterized using FESEM and TEM to elucidate their surface characteristics and porosity.

## 4.2. Optical properties of V- $\text{TiO}_2$ powder

In order to perceive application potential and discuss the results of photocatalysis, it is necessary to study the optical properties of undoped and V- $\text{TiO}_2$  nanoparticles, given that this is a reasonably clear way to reveal the effect of dopant-induced electronic properties. The activity of a semiconductor photocatalyst is strongly associated with its optical absorption ability in the visible/solar region. The results of UV-vis characterization are shown in Fig. 3.

Upon doping with a foreign element, the absorption band will be altered because of the substitution of an extrinsic dopant element into the lattice. Diffuse-reflectance UV-vis spectra were used to study this phenomenon. Fig. 3 shows the UV-vis DRS results for undoped and V- $\text{TiO}_2$  samples. Undoped  $\text{TiO}_2$  (curve a) exhibited a cutoff wavelength of around 400 nm, as expected, in the UV range. Curve b in the figure represents the UV-visible spectrum of 1% V- $\text{TiO}_2$ , which displays absorption above 400 nm, and, with an increase in the percentage of V from 2% to

Table 1 Effect of V doping on surface area, porosity and pore size

Sr. no.	Catalyst	Surface area, $\text{m}^2 \text{g}^{-1}$	Porosity, %	Pore size, nm
1	$\text{TiO}_2$	72.59	42.74	8.1
2	1% V- $\text{TiO}_2$	95.04	44.47	49.8
3	2% V- $\text{TiO}_2$	99.84	56.36	49.2
4	3% V- $\text{TiO}_2$	109.07	52.50	47.4
5	4% V- $\text{TiO}_2$	129.85	59.87	44.2
6	5% V- $\text{TiO}_2$	125.57	62.16	53.8



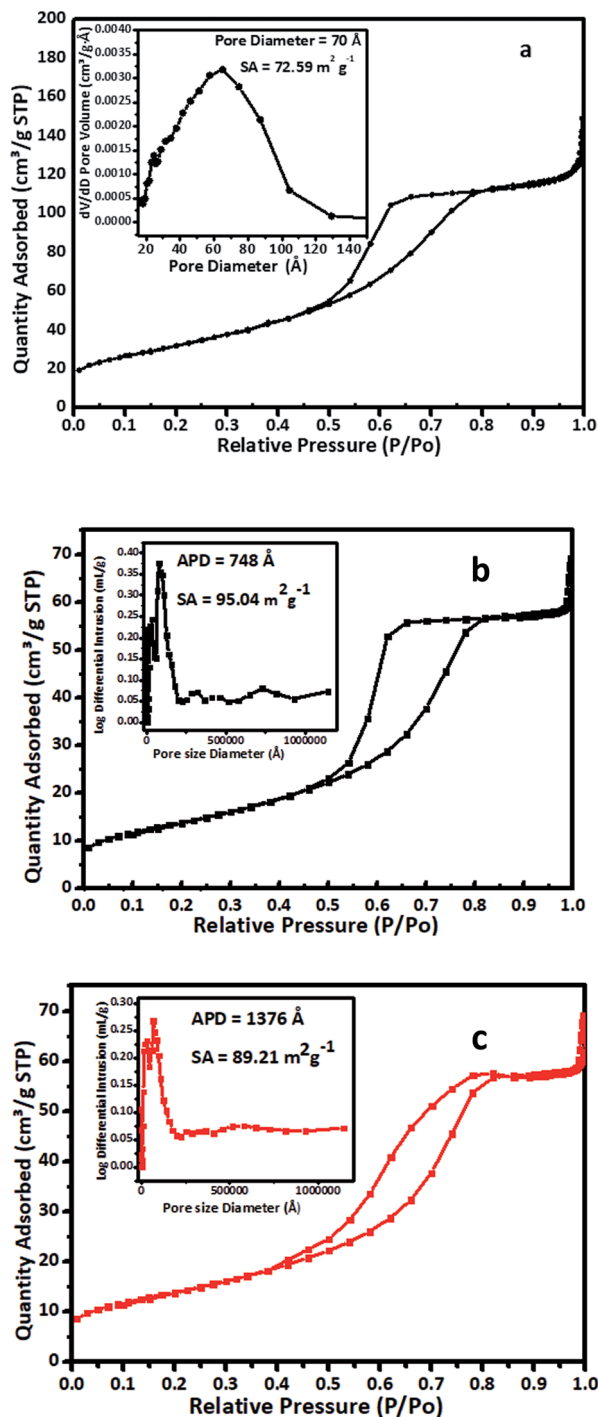


Fig. 2  $N_2$  sorption isotherms for (a)  $\text{TiO}_2$ , (b) as-prepared 1% V- $\text{TiO}_2$  photocatalyst and (c) 1% V- $\text{TiO}_2$  catalyst recovered after the third cycle. Insets: BJH desorption pore size distribution curves for (a)  $\text{TiO}_2$ , (b) as-prepared 1% V- $\text{TiO}_2$  photocatalyst and (c) 1% V- $\text{TiO}_2$  catalyst recovered after the third cycle.

5% (curves c–f), the absorption was shifted further towards the visible side. All the doped  $\text{TiO}_2$  samples exhibited a red shift in comparison with undoped  $\text{TiO}_2$ . The tailing of the absorption curves shows that all the doped samples were able to absorb photons with wavelengths of greater than 400 nm, which are in

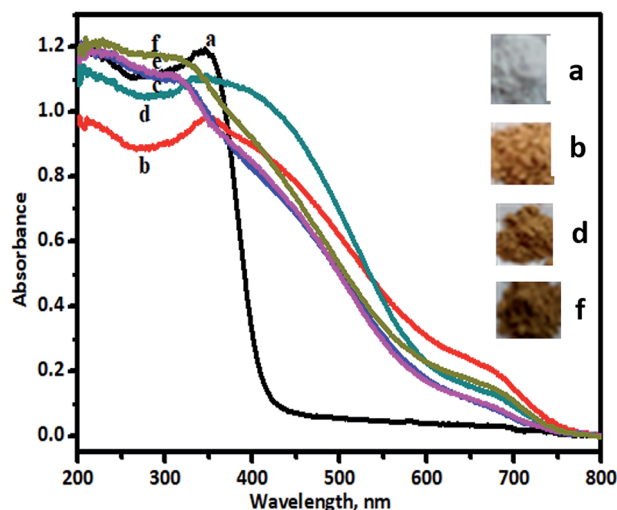


Fig. 3 UV-visible DRS spectra of powder samples of (a)  $\text{TiO}_2$ , (b) 1% V- $\text{TiO}_2$ , (c) 2% V- $\text{TiO}_2$ , (d) 3% V- $\text{TiO}_2$ , (e) 4% V- $\text{TiO}_2$ , and (f) 5% V- $\text{TiO}_2$  (Inset: change in colour of V- $\text{TiO}_2$  with increase in V content).

the visible range. No separate peaks due to  $\text{V}_2\text{O}_4$  or  $\text{V}_2\text{O}_5$  were observed in the recorded UV-vis spectra, which display major peaks at 476 and 338 nm and at 355 and 240 nm, respectively.<sup>31,32</sup> Hence, the absorption band in the visible region may be due to the presence of vanadium in the  $\text{V}^{4+}$  and  $\text{V}^{5+}$  states in the  $\text{TiO}_2$  lattice.<sup>12</sup>

The absorption of visible light by the V- $\text{TiO}_2$  powder samples is an indication of the excitation of a 3d electron from a vanadium centre into the  $\text{TiO}_2$  conduction band. The band gap of the titania catalyst is 2.90 eV, whereas the vanadia-doped catalysts exhibited a band gap of around 1.79 to 1.69 eV, which was determined from Tauc plots, and the band gap energy levels that were found are shown in Fig. 4.

The Tauc plots (Fig. 4) show that there was a substantial decrease in band gap energy for the V-doped samples. The plot of  $(\alpha h\nu)^{1/2}$  versus the energy of the band gap ( $h\nu$ ) between the VB

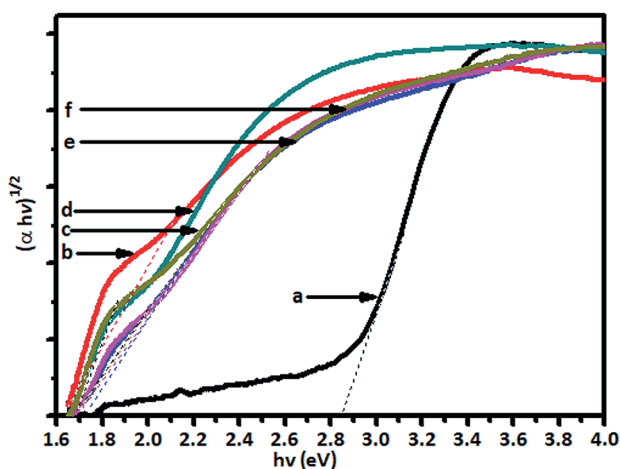


Fig. 4 Tauc plots for (a)  $\text{TiO}_2$  and (b–f) V- $\text{TiO}_2$  with various concentrations.



and CB in eV shows that pristine  $\text{TiO}_2$  displayed an energy threshold in the UV region that corresponded to an estimated band gap energy ( $E_{\text{bg}}$ ) of 3.05 eV. It was found that the band gap energy of  $\text{TiO}_2$  is 2.90 eV, whereas the band gap of doped V- $\text{TiO}_2$  is narrower (1.79 to 1.69 eV) than that of  $\text{TiO}_2$ .

#### 4.3. FT-IR and Raman spectra of V- $\text{TiO}_2$ catalyst

The FT-IR spectra of pure  $\text{TiO}_2$  and the V- $\text{TiO}_2$  catalyst calcined at 400 °C are shown in Fig. 5.

There are characteristic wide peaks in the region of 1000–400  $\text{cm}^{-1}$ , which are related to the bending vibrations of Ti–O bonds. New absorption peaks at 950 and 817  $\text{cm}^{-1}$  were observed for the V- $\text{TiO}_2$  sample. It can also be seen that the intensity of the peak at 476  $\text{cm}^{-1}$  of V-doped  $\text{TiO}_2$  is higher than that of  $\text{TiO}_2$ . Raman spectra that give confirmation of the doping of vanadia in titanium dioxide nanoparticles are shown in Fig. 6.

Raman spectroscopy is recognized to be extremely sensitive and is extensively used for the detection of phase changes and nano-oxide clusters in samples. Fig. 6 shows the Raman spectrum of a pure  $\text{TiO}_2$  sample with typical vibrational modes of anatase at 137, 391, 508 and 631  $\text{cm}^{-1}$ . Fig. 6b shows the Raman spectrum of a 1% V- $\text{TiO}_2$  sample, in which all the peaks are indexed to the anatase phase with no appearance of any peaks corresponding to individual vanadium oxides. The absence of peaks due to individual vanadium oxides suggests that the doped metal occupied substitution positions in V- $\text{TiO}_2$  rather than acting as an impurity. However, a small shift in the main peak at 139  $\text{cm}^{-1}$  was observed, and marginal broadening of peaks can be seen in Fig. 6b in comparison with the undoped  $\text{TiO}_2$  sample (Fig. 6a). This observation is in accordance with the XRD results and confirms the lattice substitution. V- $\text{TiO}_2$  samples that were calcined at 400 °C displayed fundamental vibrational modes of the anatase phase with small amounts of rutile. The Raman spectra serve as further proof that an increase in the concentration of vanadium induces a transformation from anatase to rutile.<sup>35</sup>

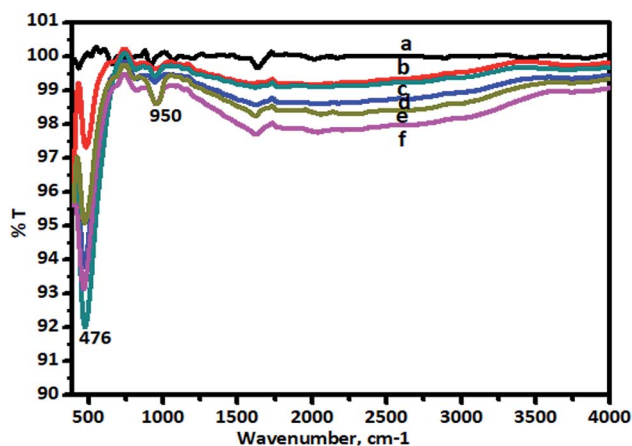


Fig. 5 FT-IR spectra of (a)  $\text{TiO}_2$ , (b) 1.0% V- $\text{TiO}_2$ , (c) 2% V- $\text{TiO}_2$ , (d) 3.0% V- $\text{TiO}_2$ , (e) 4.0% V- $\text{TiO}_2$ , and (f) 5.0% V- $\text{TiO}_2$  calcined at 400 °C.

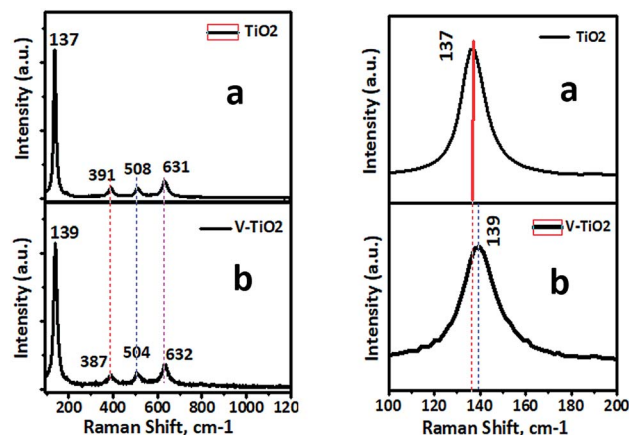


Fig. 6 Raman spectra of (a) undoped  $\text{TiO}_2$  and (b) a 1% V- $\text{TiO}_2$  sample.

#### 4.4. Thermal and crystallization behaviour

A thermogravimetric analysis (TG) curve for an air-dried 1% V- $\text{TiO}_2$  gel is shown in Fig. S2 of the ESI.† The sample that was subjected to TG analysis exhibited a total weight loss of about 14% in two stages in the temperature range of 30–450 °C. The initial sharp weight loss at around 100–120 °C may be ascribed to the loss of absorbed water in the V- $\text{TiO}_2$  gel. The weight loss in the second stage may be due to the decomposition of the titanium peroxide complex into crystalline titanium dioxide and hydrous titanium oxide. Most of the titanium peroxide complex is converted into crystalline  $\text{TiO}_2$  in this temperature range alone, as XRD results show the crystallization of  $\text{TiO}_2$  in this temperature range alone. A very slight weight loss between 450 and 900 °C was observed, which could be attributed to the conversion of the very small portion of hydrous oxide left undecomposed into  $\text{TiO}_2$ .

#### 4.5. Surface morphology using FESEM and HRTEM

The surface morphology/microstructure of powder samples was analysed by FESEM, which showed that the powder had rough surfaces. A granular texture with spheroidal particles and particle agglomerates was observed on the surface. Pure  $\text{TiO}_2$  exhibits a granular texture (Fig. 7a and b) with compactly bound particles, and the inter-particle distance is much less, which suggests that the sample has lower porosity. This could be the reason for the smaller surface area of pure  $\text{TiO}_2$ . In the FESEM image of  $\text{TiO}_2$ , although the particles have a granular texture the morphology of the particles is not uniform in size and shape. However, V- $\text{TiO}_2$  (Fig. 7c and d) exhibits granules with some empty space between the particles, and as a result the texture of the sample appears porous. The porosity of the particles is high in V- $\text{TiO}_2$  samples, which could be the reason why the surface areas of these samples are larger in comparison with the undoped  $\text{TiO}_2$  sample. The FESEM results are in accordance with the surface area results in a previous section.

The surface morphology was further confirmed using TEM analysis. TEM images of vanadium-doped  $\text{TiO}_2$  are shown in Fig. 8. The V- $\text{TiO}_2$  sample that was calcined at 400 °C was used





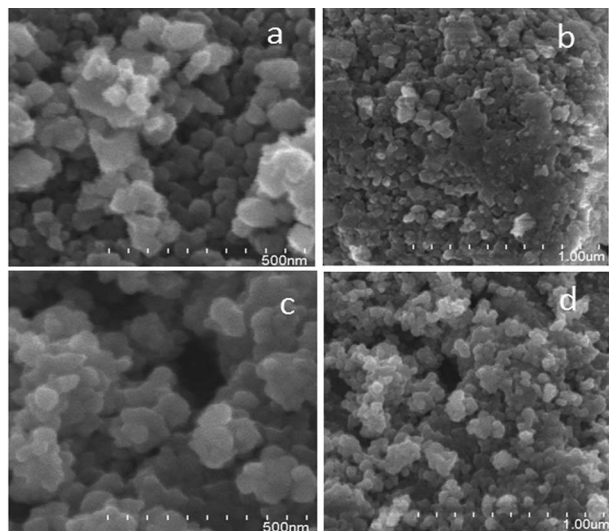


Fig. 7 FESEM (a and b) high- and low-magnification images of  $\text{TiO}_2$  and (c and d) high- and low-magnification images of 1% V- $\text{TiO}_2$ .

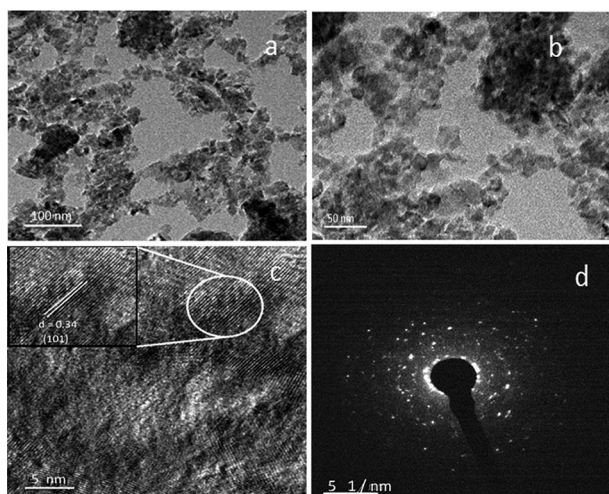


Fig. 8 TEM images: (a and b) low- and high-magnification images and (c) lattice image and (d) SAED pattern of 1% V- $\text{TiO}_2$  sample.

for this analysis as a representative sample. The TEM image indicates clearly that the particle size is homogeneous and fairly small. Vanadium oxide species were not detected in powder XRD and Raman spectroscopic analyses, and hence structural characterization was also carried out using HRTEM (Fig. 8). All the results were obtained from a 1% V- $\text{TiO}_2$  sample as a representative sample, and scanning was performed in different regions of the sample. No individual vanadium oxide species were observed during the analysis, which further confirms the incorporation of vanadium into the  $\text{TiO}_2$  lattice. In Fig. 8b, aggregated nanoparticles with an average size of  $\sim 10\text{--}15$  nm are seen with high uniformity in size and shape. The lattice image in Fig. 8c suggests that the measured  $d$ -spacing (0.34 nm) corresponds to the (101) planes of the anatase phase. This domain exhibits comparatively smaller particles with mesopores evenly dispersed between the particles. The SAED pattern

in Fig. 8d clearly shows a dot-type diffraction pattern, which was indexed to the (101) planes of anatase  $\text{TiO}_2$  and suggests that the nanoparticles were single-crystalline in nature.

The quantification of vanadium species was further performed using energy-dispersive X-ray analysis (EDAX). For this analysis, one sample with a low vanadium content (1% V- $\text{TiO}_2$ ) and another sample with a high V content (4% V- $\text{TiO}_2$ ) were chosen. EDAX spectra showing a compositional analysis of both samples together with the content of each element are shown in Fig. S3 of the ESI.†

#### 4.6. Photoluminescence spectroscopy (PL)

PL analysis was used to investigate the separation of photo-generated electron-hole pairs and the radiative recombination efficiency of the photocatalysts. The intensity of the emission spectrum depends on the charge carrier density and the probability of transitions from a higher-energy state to a vacant lower-energy state.<sup>45</sup> A lower intensity suggests enhanced separation and transfer of charge carriers. The PL spectra of samples of pure  $\text{TiO}_2$ , 1% V- $\text{TiO}_2$  and 5% V- $\text{TiO}_2$  that were calcined at  $400^\circ\text{C}$  were recorded with an excitation wavelength of 300 nm and are shown in Fig. 9. The emission peak at around 400 nm originates from a near-band-edge (NBE) free-excitation emission, *i.e.*, a direct transition from the VB to the CB. The latter additional emission signals at 437, 452, 469, 484, 493 and 532 nm originate from V- $\text{TiO}_2$ . Careful observation of the PL spectra in Fig. 9 shows that 1% V- $\text{TiO}_2$  displayed the lowest PL emission intensity, which indicated the lowest electron-hole recombination efficiency in comparison with pure  $\text{TiO}_2$  and 5% V- $\text{TiO}_2$ .

## 5. Photocatalytic study

### 5.1. Photocatalytic degradation of spent wash

The problem of spent wash effluent is on a large scale and, owing to the high cost of effluent treatment, many industries

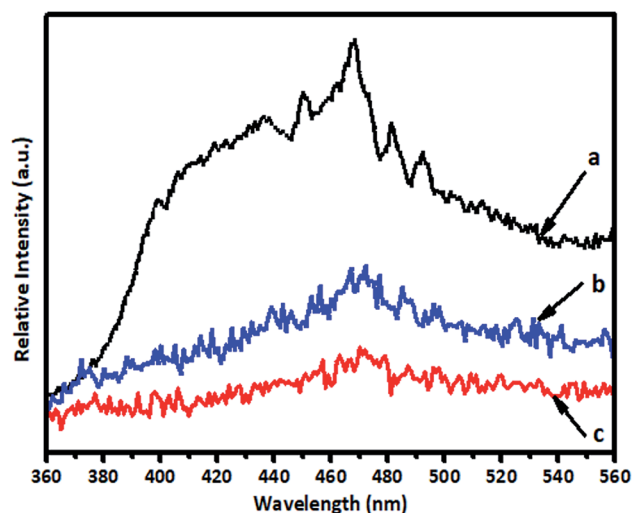


Fig. 9 PL spectra of samples calcined at  $400^\circ\text{C}$ : (a)  $\text{TiO}_2$ , (b) 5% V- $\text{TiO}_2$  and (c) 1% V- $\text{TiO}_2$ .





dispose of the effluent into nearby rivers/land without any treatment to save the cost. Therefore, the use of a natural energy source (sunlight) was selected in place of an artificial energy source (Xe lamp). Photocatalytic activity was tested in the degradation of a spent wash solution from a sugar cane distillery (1 g/100 mL solution) under solar radiation using V-TiO<sub>2</sub> powder. The UV-visible absorption spectra of standard samples of melanoidin and caramel were recorded for reference (Fig. S4 in the ESI†). The changes in concentration and colour in samples irradiated for different time intervals with solar radiation were monitored using a UV-visible spectrometer (absorbance at 270 nm) and compared with a blank kept in sunlight under identical experimental conditions. Typical UV spectra of the spent wash solution in the presence of V-TiO<sub>2</sub> catalysts before and after irradiation with solar light for different time intervals are shown in Fig. S5 in the ESI.†

The 1% V-TiO<sub>2</sub> catalyst achieved the maximum degradation of spent wash within 5 h. This is attributed to the decomposition of colorants present in spent wash such as melanoidin and caramel. These colorants exhibit very strong absorption in the UV region with absorption maxima at a wavelength of around 275 nm. Melanoidin<sup>46</sup> and caramel<sup>47</sup> are homogeneous compounds with dissimilar molecular weights. Caramel has a molecular weight in the region of 248 Da, whereas melanoidin has an MW in the range from 251 396 to 213 Da. The results of the degradation of colorants in spent wash in daltons are given in Table 2.

The concentrations of colorants such as melanoidin and caramel polymers in spent wash samples were determined using a PerkinElmer series 200 instrument by gel permeation chromatography (GPC) using Aqua OH-30 PL gel (300 × 7.5 mm, 10 Å) and Aqua OH-40 PL gel (300 × 7.5 mm, 100 Å). The column oven had a temperature of 30 °C with a refractive index detector, and the flow rate of the mobile phase (pH 7.0) was 1 mL min<sup>-1</sup>. Total Chrome (version 6.3.1) and GPC Turbo SEC software (version 6.3.1.0504) were used for data processing. Traceable American Polymer Standards for polyethylene glycol were used for calibration. The GPC chromatograms of standard caramel and melanoidin are shown in Fig. S6 (a and b) in the ESI.†

A sample solution of spent wash was monitored by gel permeation chromatography before and after irradiation with sunlight for colour degradation, and the chromatograms are shown in Fig. 10a and b. From the results, it is seen that high-

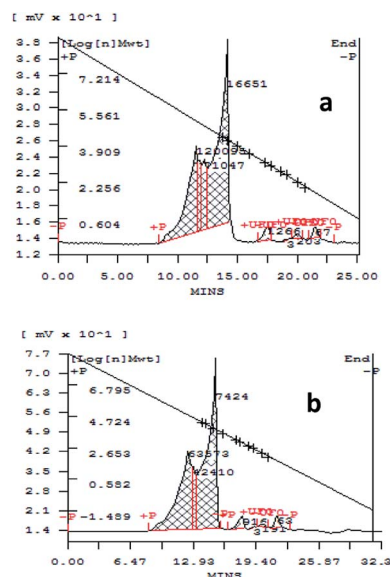


Fig. 10 GPC chromatograms (a) before and (b) after irradiation with sunlight of a sample solution of spent wash.

and low-molecular-weight (MW) polymeric compounds in the spent wash sample were degraded under sunlight in the presence of a V-TiO<sub>2</sub> photocatalyst. High-molecular-weight compounds were degraded from 120 095 to 63 572 Da and from 100 060 to 42 410 Da in 5 h at a slow degradation rate, whereas the degradation rate of low-MW compounds (71 047 Da) was fast. This may be attributed to the molecular arrangement in high- and low-molecular-weight compounds.

TOC measurements of the original solutions of the dye and spent wash and samples obtained after photocatalytic degradation using 1% V-TiO<sub>2</sub> were carried out using a TOC analyzer. TOC analysis of Jakofix dye (100 ppm) and the spent wash solution (10 000 ppm) gave TOC values of 3.33 ppm and 428 ppm, respectively. The TOC values measured after photodegradation were 2.10 ppm and 275 ppm, respectively. The samples after photodegradation using 1% V-TiO<sub>2</sub> exhibited a decrease in TOC values, which indicated the degradation of these molecules into lower-molecular-weight compounds and mineralization of some low-molecular-weight compounds to CO<sub>2</sub>, H<sub>2</sub>O and other gases (Fig. 11). Degradation of the colour of spent wash by almost 65% was achieved after irradiation with

Table 2 Changes in molecular weight with irradiation time

Sr. no	RT, min	MW in daltons, before irradiation with sunlight (0 h)	MW in daltons, after irradiation with sunlight (5 h)
1	11.519	120 095	63 572
2	11.760	100 060	42 410
3	12.212	71 047	—
4	14.129	16 651	7424
5	17.533	1266	916
6	19.142	374	360
7	19.950	203	191
8	21.417	67	63



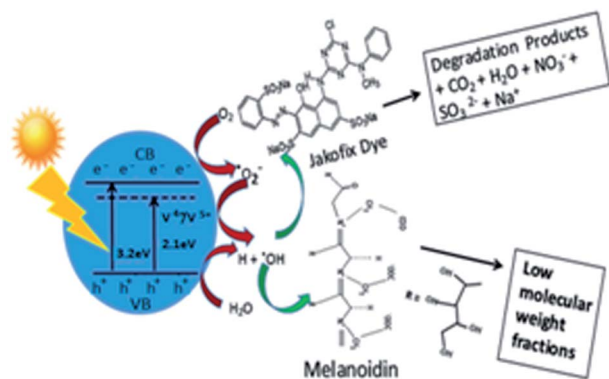


Fig. 11 Mechanism of the photocatalytic degradation of spent wash and Jakofix red dye.

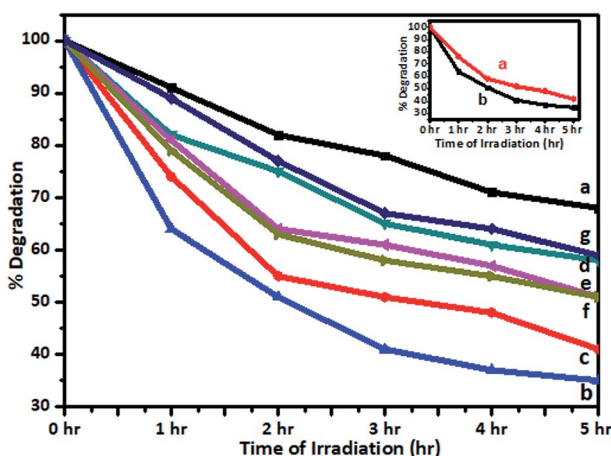


Fig. 12 Degradation of colour of spent wash after irradiation with sunlight for (a) undoped TiO<sub>2</sub>, (b) 1%, (c) 2%, (d) 3%, (e) 4%, and (f) 5% V-TiO<sub>2</sub> and (g) P-25 catalyst. Inset: degradation of colour of spent wash with 1% V-TiO<sub>2</sub> (a) under xenon and (b) under solar light.

sunlight for 5 h in the presence of 1% V-TiO<sub>2</sub>. The percentage degradation of the colour of spent wash for the 1–5% V-TiO<sub>2</sub> catalysts is graphically represented in Fig. 12.

The total solar energy that reaches the earth's surface is around  $\sim 1120 \text{ W m}^{-2}$ , which includes 42–43% visible light (400–800 nm) and 3–5% UV light (<400 nm). Direct sunlight has a luminous efficacy of  $\sim 93$  lumens (total quantity of visible light) per watt radiant flux. Luminous efficacy is a measure of how efficiently any light source produces visible energy. Of course, this is a much higher efficacy than that of artificial lighting. Many researchers have used a xenon lamp (visible light source) as a source of artificial sunlight for photocatalytic reactions in the presence of glass filters to eliminate UV light. Despite the fact that a xenon lamp produces continuous light in the visible range that is very similar to daylight, it is very costly as an energy supply is essential to illuminate the lamp. A xenon arc lamp has a luminous efficacy of about  $30\text{--}50 \text{ lm W}^{-1}$  (lumen per watt) at a bulb power of 1000 W, and the overall luminous efficacy is around 4.4–7.3% (expressed in dimensionless form), because all lamps use electric ballast. Considering all these

factors, instead of using high-wattage incandescent lamps, photocatalytic degradation in natural sunlight is potentially scalable in terms of system cost and is eco-friendly as well.<sup>45</sup>

Therefore, the degradation of colour in spent wash was carried out under a xenon lamp and natural sunlight using a 1% V-TiO<sub>2</sub> catalyst.

A graph of the change in the concentration of spent wash with the irradiation time is shown in the inset of Fig. 12. From the graph, it is seen that degradation of spent wash of nearly 54% and 65% took place under the Xe lamp and natural sunlight, respectively. The colour degradation rate was faster under natural sunlight than under a standard source of energy, *i.e.*, a xenon lamp. This may be because the luminous efficacy of sunlight is higher than that of an Xe lamp.

## 5.2. Photocatalytic degradation of Jakofix red dye (HE 8BN)

Recently, several visible-light-active photocatalysts have been reported for the degradation of toxins and dyes found in industrial effluent.<sup>45,48</sup> The photocatalytic activity of a V-TiO<sub>2</sub> catalyst was also tested in the degradation of Jakofix red dye (100 ppm solution) under solar radiation using V-TiO<sub>2</sub> powder. The change in the concentration of the dye in samples irradiated for different time intervals with solar radiation was monitored using a UV-visible spectrometer (200–800 nm) and compared with undoped TiO<sub>2</sub> and P-25 TiO<sub>2</sub> (Degussa) kept in sunlight under identical experimental conditions. Typical UV spectra of the Jakofix red dye solution in the presence of 1% V-TiO<sub>2</sub> catalysts before and after irradiation with solar light for different time intervals are shown in Fig. S7 in the ESI.†

The results for the photocatalytic efficiency of TiO<sub>2</sub> and vanadium-doped TiO<sub>2</sub> in the degradation of Jakofix red dye under irradiation with visible light are shown in Fig. 13. Under sunlight, degradation of 90% of Jakofix red dye (HE 8BN) was observed after 3.5 h using a 1% V-TiO<sub>2</sub> catalyst. Degradation of 50% was observed within a time interval of 2 h for a dye solution. V-TiO<sub>2</sub> samples exhibited higher activity than pure TiO<sub>2</sub>

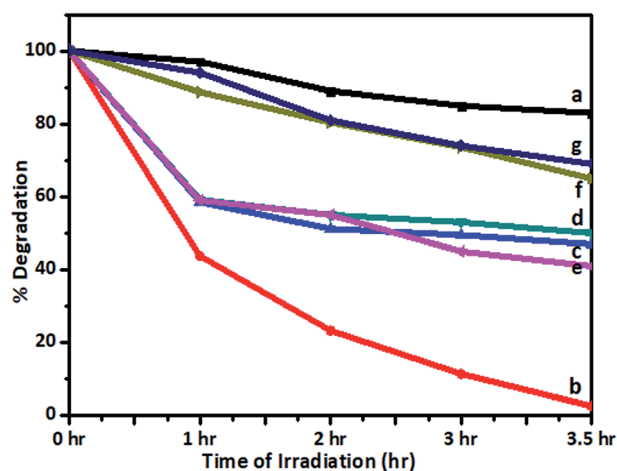


Fig. 13 Degradation of dye (%) remaining after irradiation with sunlight using (a) undoped TiO<sub>2</sub>, (b) 1%, (c) 2%, (d) 3%, (e) 4%, and (f) 5% V-TiO<sub>2</sub> and (g) P-25 TiO<sub>2</sub>.



and Degussa P-25 TiO<sub>2</sub>. This shows that V helped to increase the activity in photodegradation reactions. The increase in photocatalytic activity may be attributed to the combined effect of the doping of V into TiO<sub>2</sub> and the formation of hierarchical nanostructures in V-doped samples. Fang *et al.*<sup>49,50</sup> have reported that the development of a hierarchical nanostructure in a TiO<sub>2</sub> microsphere not only favors fast mass transport, which results in enhanced photocatalytic activity, but may also contribute to an improvement in photocatalytic activity due to enhanced multi-light scattering. The results are in accordance with the observations reported by Fang *et al.*

The efficiency of photocatalysts is based on the recombination rate of electron/hole pairs. In this study, the introduction of vanadium into the TiO<sub>2</sub> matrix is believed to contribute to the high electron/hole separation rate. The introduction of vanadium would alter the electronic properties of TiO<sub>2</sub>. The V<sup>4+</sup>/V<sup>5+</sup> pair could form an impurity energy band around 2.1 eV, which can act as an e<sup>-</sup>/h<sup>+</sup> trap that improves the separation efficiency.<sup>51</sup> The substitution of a Ti<sup>4+</sup> ion by a V<sup>5+</sup> ion leaves a free electron, which can easily be excited to the TiO<sub>2</sub> conduction band by irradiation with light.<sup>52</sup> Moreover, V<sup>4+</sup> ions trap photogenerated holes to form V<sup>5+</sup> ions.<sup>49</sup> Therefore, the V<sup>4+</sup>/V<sup>5+</sup> pair acts as an electron/hole trap center, which substantially decreases the recombination rate. A higher loading (2–5%) of V in TiO<sub>2</sub> powders results in relatively lower photocatalytic activity, although the band gap energy is lower than those of all the other V-TiO<sub>2</sub> samples. This may be because the additional vanadium ions may act as electron–hole recombination centres (as also seen in the photoluminescence spectra), which ultimately results in low photocatalytic activity for samples that contain larger amounts of vanadium. The repeatability and reproducibility of the degradation reaction were assessed by conducting the degradation reaction for 3 consecutive cycles under identical conditions. Table 3 shows the results of degradation experiments carried out to confirm the repeatability and reproducibility of results using a 1% V-TiO<sub>2</sub> photocatalyst. From the results it is seen that there was no considerable decrease in the activity of the photocatalyst in the second and third cycles, which suggests that the repeatability and reproducibility of the results were high.

The stability of the catalyst is a very important parameter in a catalytic reaction, as it has a direct effect on the cost of the process and the final product. A stability test was performed to study the degradation of the photocatalyst itself, as many photocatalysts such as CdS and ZnS have problems with photocorrosion in the presence of light. The XRD patterns of 1% V-TiO<sub>2</sub> before photocatalytic use (curve a) and recovered after 3 cycles of the photodegradation reaction (curve b) are shown in Fig. S8 in the ESI.† The XRD patterns show no change in

structural behaviour, as all the peaks and their positions are identical to those of the 1% V-TiO<sub>2</sub> photocatalyst before use. The XRD pattern of V-TiO<sub>2</sub> recovered after 3 consecutive cycles of photodegradation experiments suggests that the photocatalyst has high stability under the given experimental conditions. This indicates that the photocatalyst can be used for several experiments without much decrease in activity.

The result of a recyclability test of a 1% V-TiO<sub>2</sub> catalyst for the degradation of colour in spent wash is shown in Fig. S9 in the ESI.† From the figure, it seems that not much decrease in the performance of the photocatalyst was observed after the second and third cycles, which suggests that the recyclability and stability of the photocatalyst in this reaction were high. A recovery test was performed by recovering the photocatalyst after each cycle of the experiment. A plot of the quantity of photocatalyst recovered *versus* the number of cycles is shown in Fig. S10 in the ESI.† The loss of weight of the catalyst was only 6% after three cycles of the degradation reaction, which confirms that large quantities of the catalyst can be recovered easily.

## 6. Conclusions

A vanadium-doped titanium dioxide (V-TiO<sub>2</sub>) powder photocatalyst was prepared using a simple sol-gel method based on aqueous titanium peroxide and tested for the degradation of spent wash and Jakofix dye. The V-TiO<sub>2</sub> powder photocatalyst exhibited very high activity in the photocatalytic degradation of spent wash and Jakofix red dye under natural sunlight. Among the photocatalysts that were investigated, 1% V-TiO<sub>2</sub> was found to be the most active for the degradation of colour in spent wash (65%) and Jakofix red dye (90%). The photocatalyst was tested for its activity and exhibited higher activity under natural sunlight than under an artificial source of energy (Xe lamp). The activity under visible light is attributed to the incorporation of V species into the TiO<sub>2</sub> matrix and the subsequent shift in the absorption of the doped catalyst in the visible region. Gel permeation chromatography shows a breakdown of high-molecular-weight colorant compounds into lower-molecular-weight compounds. The photocatalyst is quite stable, and hence it can be used for many degradation cycles with a recovery of the photocatalyst of 94–96% after 3 cycles.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

The authors would like to thank the Director General, C-MET, Pune and Department of Chemistry, A.M. College, Pune for the encouragement. Mr Shrikant P. Takle is grateful to the Centre for Materials for Electronics Technology (C-MET) for providing the research facilities and the Nanocrystalline Materials Group, C-MET, Pune for their support.

Table 3 Results of reproducibility test of 1% V-TiO<sub>2</sub>

Degradation	Time in h	1 <sup>st</sup> cycle	2 <sup>nd</sup> cycle	3 <sup>rd</sup> cycle
Spent wash	5	65%	64%	62%
Jakofix dye	3.5	90%	87%	86%





## References

- 1 R. M. Bhise, A. A. Patil, A. R. Raskar, P. J. Patil and D. P. Deshpande, *Res. J. Recent Sci.*, 2012, **1**, 66–69.
- 2 Y. Satyawali and M. Balakrishnan, *Bioresour. Technol.*, 2007, **98**, 2629–2635.
- 3 M. N. Vineetha, M. Matheswaran and K. N. Sheeba, *Sol. Energy*, 2013, **91**, 368–373.
- 4 C. David, M. Arivazhagan and M. Ibrahim, *J. Saudi Chem. Soc.*, 2015, **19**, 537–548.
- 5 M. Navgire, A. Yelwande, D. Tayde, B. Arbad and M. Lande, *Chin. J. Catal.*, 2012, **33**, 261–266.
- 6 M. R. Hoffmann, S. T. Martin, W. Choi and D. W. Bahnemann, *Chem. Rev.*, 1995, **95**, 69–96.
- 7 M. A. Fox and M. T. Dulay, *Chem. Rev.*, 1993, **93**, 341–357.
- 8 A. L. Linsebigler, G. Lu and J. T. Yates Jr, *Chem. Rev.*, 1995, **95**, 735–758.
- 9 M. Anpo, *Res. Chem. Intermed.*, 1989, **11**, 67–106.
- 10 S. T. Martin, C. L. Morrison and M. R. Hoffmann, *J. Phys. Chem.*, 1994, **98**, 13695–13704.
- 11 W. Choi, A. Termin and M. R. Hoffmann, *J. Phys. Chem.*, 1994, **98**, 13669–13679.
- 12 S. Klosek and D. Raftery, *J. Phys. Chem. B*, 2001, **105**, 2815–2819.
- 13 X. Hong, Z. Wang, W. Cai, F. Lu, J. Zhang, Y. Yang, N. Ma and Y. Liu, *Chem. Mater.*, 2005, **17**, 1548–1552.
- 14 G. Zhao, H. Kozuka, H. Lin and T. Yoko, *Thin Solid Films*, 1999, **339**, 123–128.
- 15 G. C. Bond and S. F. Tahir, *Appl. Catal.*, 1991, **71**, 1–31.
- 16 D. J. Stacchiola, S. D. Senanayake, P. Liu and J. A. Rodriguez, *Chem. Rev.*, 2013, **113**(6), 4373.
- 17 Z. Zhang and V. E. Henrich, *Surf. Sci.*, 1992, **277**, 263.
- 18 G. Centi, D. Pinelli and F. J. Trifiro, *Mol. Catal.*, 1990, **59**, 221.
- 19 I. Muylaert and P. Van Der Voort, *Phys. Chem. Chem. Phys.*, 2009, **11**, 2826.
- 20 E. L. Lee and I. E. Wachs, *J. Phys. Chem. C*, 2008, **112**, 20418.
- 21 M. Baron, H. Abbot, O. Bondarchuk, D. Stacchiola, A. Uhl, S. Shaikhutdinov, H.-J. Freund, C. Popa, M. V. Ganduglia-Pirovano and S. Joachim, *Angew. Chem., Int. Ed.*, 2009, **48**, 8006.
- 22 I. E. Wachs, *Catal. Today*, 2005, **100**, 79.
- 23 H. H. Kung, *Transition Metal Oxides: Surface Chemistry and Catalysis*, Elsevier, Amsterdam, 1989.
- 24 G. C. Bond and S. F. Tahir, *Appl. Catal.*, 1991, **71**, 1.
- 25 V. Shapovalov and H. Metiu, *J. Phys. Chem. C*, 2007, **111**, 14179.
- 26 V. I. Avdeev and V. M. Tapilin, *J. Phys. Chem. C*, 2010, **114**, 3609.
- 27 M. Anpo, Y. Ichihashi, M. Takeuchi and H. Yamashita, *Res. Chem. Intermed.*, 1998, **24**, 143–149.
- 28 M. Anpo, H. Yamashita, S. Kanai, K. Sato and T. Fujimoto, *US Pat.* 6077492, 2000.
- 29 M. Anpo, N. Aikawa, Y. Kubokawa, M. Che, C. Louis and E. Giamello, *J. Phys. Chem.*, 1985, **89**, 5017–5021.
- 30 M. Anpo, I. Tanahashi and Y. Kubokawa, *J. Phys. Chem.*, 1980, **84**, 3440–3443.
- 31 K. Bhattacharyya, S. Varma, A. K. Tripathi, S. R. Bharadwaj and A. K. Tyagi, *J. Phys. Chem. C*, 2008, **112**, 19102–19112.
- 32 R. Bulánek, L. Čiček, M. Setnicíčka and P. Čiček, *J. Phys. Chem. C*, 2011, **115**, 12430–12438.
- 33 R. Akbarzadeh, S. B. Umbarkar, R. S. Sonawane, S. Takle and M. K. Dongare, *Appl. Catal., A*, 2010, **374**, 103–109.
- 34 D. J. Stacchiola, S. D. Senanayake, P. Liu and J. A. Rodriguez, *Chem. Rev.*, 2012, **113**, 4373–4390.
- 35 Z. Luo, A. S. Poyraz, C.-H. Kuo, R. Miao, Y. Meng, S.-Y. Chen, T. Jiang, C. Wenos and S. L. Suib, *Chem. Mater.*, 2015, **27**, 6–17.
- 36 R. S. Sonawane, B. B. Kale and M. K. Dongare, *Mater. Chem. Phys.*, 2004, **85**, 52–57.
- 37 R. S. Sonawane and M. K. Dongare, *J. Mol. Catal. A: Chem.*, 2006, **243**, 68–76.
- 38 R. S. Sonawane, S. G. Hegde and M. K. Dongare, *Mater. Chem. Phys.*, 2003, **77**, 744–750.
- 39 K. Sathishkumar, S. Sathiyaraj, P. Parthipan, A. Akhil, K. Murugan and A. Rajasekar, *Chemosphere*, 2017, **183**, 204–211.
- 40 S. Yadav and R. Chandra, *Biodegradation*, 2012, **23**, 609–620.
- 41 A. Kumar, D. P. Saroj, V. Tare and P. Bose, *Water Environ. Res.*, 2006, **78**, 994–1004.
- 42 R. Lakshmikanth and A. S. Virupakshi, *Int. J. Sci. Res.*, 2012, **3**, 1–12.
- 43 M. Boualleg and B. Guichard, *US Pat.* App. 15/318151, 2017.
- 44 L. Lucarelli and M. P. Line, *Microstructure Product Line*, Thermo Electron Corporation, Milan-Italy, 2010.
- 45 S. K. Khore, N. V. Tellabati, S. K. Apte, S. D. Naik, P. Ojha, B. B. Kale and R. S. Sonawane, *RSC Adv.*, 2017, **17**, 33029.
- 46 S. Ohmomo, I. Aoshima, Y. Tozawa, N. Sakurada and K. Ueda, *Agric. Biol. Chem.*, 1985, **49**, 2047–2053.
- 47 S. Agnihotri, *Int. J. Curr. Microbiol. Appl. Sci.*, 2015, **4**, 12–17.
- 48 W. W. Zhong, Y. F. Lou, S. F. Jin, W. J. Wang and G. Liw, *Sci. Rep.*, 2016, **6**, 23235.
- 49 B. Fang, A. Bonakdarpour, K. Reilly, Y. Xing, F. Taghipour and D. P. Wilkinson, *ACS Appl. Mater. Interfaces*, 2014, **6**, 15488–15498.
- 50 B. Fang, Y. Xing, A. Bonakdarpour, S. Zhang and D. P. Wilkinson, *ACS Sustainable Chem. Eng.*, 2015, **3**, 2381–2388.
- 51 X. Yang, C. Cao, K. Hohn, L. Erickson, R. Maghirang, D. Hamal and K. Klabunde, *J. Catal.*, 2007, **252**, 296–302.
- 52 I. E. Wachs, *J. Chem. Soc., Faraday Trans.*, 1997, **93**, 1655–1660.

