1. Introduction

Due to the growing population, the global energy demand is increasing year by year. Moreover, the global energy demand is predicted to double by 2050.1,2 Thus, the development of renewable energy becomes an imminent requirement, such as water energy, wind energy, and solar energy. The photovoltaic power generation capacity is installed to be 303 GW and increased 75 GW in 2016. In 2016, photovoltaic power generation accounted for only 1.5% of the world’s total electricity generation. So high performance, long-term stability, low cost and environmental friendly solar cells become the focus of current energy research.

PSCs have attracted great attention in photovoltaic research in recent years, because of their high-efficiency (certified 22.1%)3 and low-cost. Meanwhile, organic–inorganic perovskites have a high optical absorption coefficient4 and the diffusion lengths exceed 1 μm for electrons and holes.5 So, organic–inorganic perovskite is an ideal absorber material for solar cells,6–19 photodetectors,20–22 light-emitting diodes,23–29 etc.

In recent years, hybrid metal halide perovskite materials have revolutionized the field of photovoltaics materials research, due to the power conversion efficiency (PCE) of PSC devices having been rapidly improved, from the point 3.8% in 2009,4 up to 22.6% in 2017 (ref. 3) (certified 22.1%).3 It attracted attention of researchers working on various photovoltaic technologies, especially dye solar cells (DSCs) and organic photovoltaic (OPV) with emphasis on better efficiency. In 2009, T. Miyasaka et al.3 has creatively made CH3NH3PbBr3/TiO2-based and CH3NH3PbI3/TiO2-based DSCs, the PCE of the cells is 3.13% and 3.81%, respectively. The PSCs attracted researchers’ attention then happened in 2012, when M. Grätzel and N. G. Park et al.27 made PSCs device using perovskite films as the photoactive absorber layer, the mp-TiO2 and spiro-MeOTAD were used as the electron transport layer (ETL) and hole transport layer (HTL), respectively (Fig. 1), achieving the PCE of 9.7%. In 2013, M. Z. Liu, M. B. Johnston and H. J. Snaith30 fabricated planar heterojunction PSCs via vapor deposition, and the efficiency of the PSCs device is up to 15.4%. The yttrium (Y) doping the TiO2 (ETL) improves the electron transport channel in the PSCs device, and increase its carrier concentration and modify the ITO electrode to reduce its work function. These changes achieved a PCE of 19.3%31. In 2015, S. I. Seok et al.28 attained an efficiency of PSCs up to 20.1%. In 2016, A. Zettl et al.29 made an architecture of GaN/CH3NH3SnI3/monolayer h-BN/CH3NH3-PbI2.3Brx/HTL and graphene aerogel/Au (Fig. 2). The graded bandgap PSCs demonstrated with PCE averaging 18.4%, with a best of 21.7%. Other researchers, E. H. Sargent et al.31 (2017) achieved the certified efficiencies of 20.1% via contact-passivation strategy, retaining 90% (97% after dark recovery) of their initial PCE after 500 hours of continuous room-temperature. Meanwhile, E. K. Kim, J. H. Noh, and S. I. Seok et al.3 reported that the introduction of additional iodide ions into the organic cation solution, that was used to form the perovskite layers through an intramolecular exchanging process and decrease the concentration of deep-level defects. The certified PCE of PSCs attained 22.1%.3

In addition, high efficiency PSCs devices include not only small devices, but also larger cells. A PSCs device with area of large-area (≥1 cm2) and maximum PCE of 20.5% (certified
19.7%) has been reported. Table 1 shows some results for large-area PSCs have been reported in the literatures.

But, for large-area PSCs device, it still has some issues need to be solved, namely fabrication, stability, hysteresis, fabrication cost and environmental concerns. Such as, the continuous fabrication of cracks-free and pinholes-free the perovskite and the selective carrier extraction layers films is difficult with large-area PSCs devices. The dilemma with optimizing such charge carrier extraction layers in solar cells is that the film should be thin to minimize resistive losses, while at the same time, it should cover the entire collector area in a contiguous and uniform manner.

In the large-area PSCs device, surfaces, bulk defects and interfaces introduce recombination centers that lead to fast nonradiative losses, and interface losses, which lead to the \( V_{oc} \), \( J_{sc} \) and fill factor (FF) decrease. Meanwhile, the perovskite material is easily thermal decomposition and hydrodecomposition, that leads to the lack of stability for PSCs device. The poor stability of the perovskite materials and devices is a big challenge, which hinder the PSCs device could be transferred from the laboratory to industry and outdoor applications. Thus, for large-area PSCs device, the major challenges relate to the improving efficiency and keeping the stability of the device. In this review paper, giving an update of the PSCs field, briefly, introducing the history of PSCs and then focus on the key progress of the fabrication, improving the efficiency and the stability of the large-area PSCs device.

2. Perovskite structure and typical PSCs structure

2.1 Perovskite structure and characteristics

Perovskite was discovered in 1839, which originally referred to a kind of ceramic oxides with the general molecular formula \( ABX_3 \). Recently, PSCs absorber layer is mainly organic-inorganic perovskite layer, the general molecular formula is also \( ABX_3 \) (Fig. 3), where A is an organic cation (i.e. \( \text{CH}_3\text{NH}_3^+ \), \( \text{NH}_2\text{CH} = \text{NH}_3^+ \), \( \text{CH}_3\text{CH}_2\text{NH}_3^+ \)), B is metal cation (i.e. \( \text{Pb}^{2+} \), \( \text{Sn}^{2+} \), \( \text{Ge}^{2+} \)) and X is halogen anion (i.e. \( \text{F}^- \), \( \text{Cl}^- \), \( \text{Br}^- \), \( \text{I}^- \)), are the most relevant ones for PSCs.

The perovskite arrangement is approximated on its geometric tolerance factor (\( t \)),

\[
t = \frac{r_A + r_X}{\sqrt{2(r_B + r_X)}}
\]

where \( r_A \), \( r_B \) and \( r_X \) are the efficient ionic radius for A, B and X ions, respectively. When the \( t = 1.0 \), the perovskite is a perfect cubic perovskite. However, octahedral distortion is assessed when \( t < 1 \), which influences electronic characteristics. For alkali metal halide perovskite, formability is anticipated for 0.813 < \( t < 1.107 \). In Table 2, the \( r_A \) in \( \text{APbX}_3 \) (X = Cl, Br, I) perovskite has been calculated for \( t = 0.8 \) and \( t = 1 \) based on effective ionic radii. As the tolerance of \( \text{CH}_3\text{NH}_3\text{PbI}_3 \) (MAPbI_3)
Table 1  Summary of large-area perovskite solar cells

<table>
<thead>
<tr>
<th>PSCs configuration</th>
<th>Cells area (cm²)</th>
<th>Active area (cm²)</th>
<th>PCE (%)</th>
<th>Ref.</th>
</tr>
</thead>
<tbody>
<tr>
<td>ITO/PEDOT:PPS/MAPbI₃/PCBM/Cos/BCP/Al</td>
<td>1</td>
<td>1</td>
<td>12.2</td>
<td>32</td>
</tr>
<tr>
<td>FTO/TiO₂/MAPbI₃/spiro-OMeTAD/Au</td>
<td>1</td>
<td>1</td>
<td>11.7</td>
<td>33</td>
</tr>
<tr>
<td>ITO/HTL/PFN/CH₃NH₂PbI₃/PCBM/Al</td>
<td>1</td>
<td>1</td>
<td>17.04</td>
<td>34</td>
</tr>
<tr>
<td>FTO/TiO₂/[FAPbI₃]₀.₈₅(MAPbBr₃)₀.₁₅/spiro-OMeTAD/Au</td>
<td>1</td>
<td>1</td>
<td>18.32</td>
<td>35</td>
</tr>
<tr>
<td>FTO/c-TiO₂/me-TiO₂/MAPbI₃/spiro-OMeTAD/Au</td>
<td>1</td>
<td>1</td>
<td>19.3</td>
<td>36</td>
</tr>
<tr>
<td>FTO/bl-TiO₂/mp-TiO₂/perovskite/spiro-OMeTAD/Au</td>
<td>1</td>
<td>1</td>
<td>19.6 (certified)</td>
<td>37</td>
</tr>
<tr>
<td>FTO/TiO₂/me-TiO₂/perovskite/PTAA/Au</td>
<td>1</td>
<td>1</td>
<td>19.7 (certified)</td>
<td>3</td>
</tr>
<tr>
<td>ITO/SnO₂/[FAPbI₃]₀.₈₅(MAPbBr₃)₀.₁₅/spiro-OMeTAD/Au</td>
<td>1</td>
<td>1</td>
<td>20.1 (certified)</td>
<td>38</td>
</tr>
<tr>
<td>FTO/TiO₂/CH₃NH₂PbI₃₋ₓClₓ/spiro-OMeTAD/Au</td>
<td>4</td>
<td>1</td>
<td>13.6</td>
<td>39</td>
</tr>
<tr>
<td>FTO/c-TiO₂/me-TiO₂/perovskite/spiro-OMeTAD/Au</td>
<td>1.01</td>
<td>1.01</td>
<td>16.61</td>
<td>40</td>
</tr>
<tr>
<td>FTO/NiMgLiO/MAPbI₃/PCBM/Ti(Nb)O₂/Ag</td>
<td>1.02</td>
<td>1.02</td>
<td>16.2 (15 certified)</td>
<td>41</td>
</tr>
<tr>
<td>Anode/HEL/perovskite/gradient interlayer/ETL/cathode</td>
<td>1.022</td>
<td>1.022</td>
<td>18.21 (certified)</td>
<td>42</td>
</tr>
<tr>
<td>FTO/c-TiO₂/me-TiO₂/perovskite/spiro-OMeTAD/Au</td>
<td>1.05</td>
<td>1.05</td>
<td>15.89</td>
<td>43</td>
</tr>
<tr>
<td>FTO/ZnO/MAPbI₃/spiro-OMeTAD/Au</td>
<td>1.10</td>
<td>1.10</td>
<td>15.98</td>
<td>44</td>
</tr>
<tr>
<td>FTO/TiO₂/Cl/FA₀.₈₅MA₀.₁₅PbI₂.₅₅Br₀.₄₅/spiro-OMeTAD/Au</td>
<td>1.10</td>
<td>1.10</td>
<td>19.5 (certified)</td>
<td>45</td>
</tr>
<tr>
<td>SAM/PC₆/BB/MAPbI₃/PTAA/Au</td>
<td>1.20</td>
<td>1.20</td>
<td>18.32</td>
<td>46</td>
</tr>
<tr>
<td>SAM/PC₆/MM/MAPbI₃/spiro-OMeTAD/Au</td>
<td>5.04</td>
<td>5.04</td>
<td>12.79</td>
<td>47</td>
</tr>
<tr>
<td>FTO/c-TiO₂/TiO₂, or MAPbI₃/spiro-OMeTAD/Au</td>
<td>10.1</td>
<td>10.1</td>
<td>6.0</td>
<td>48</td>
</tr>
<tr>
<td>FTO/c-TiO₂/me-TiO₂/MAPbI₃/spiro-OMeTAD/Au</td>
<td>36.0</td>
<td>36.0</td>
<td>15.7 (12.1 certified)</td>
<td>49</td>
</tr>
<tr>
<td>FTO/c-TiO₂/TiO₂ or MAPbI₃/spiro-OMeTAD/Au</td>
<td>100</td>
<td>100</td>
<td>4.3</td>
<td>50</td>
</tr>
</tbody>
</table>

Fig. 3  ABX₃ perovskite structure.

Table 2  Estimation of A cation radii in APbX₃

<table>
<thead>
<tr>
<th>Pb⁺⁺ 1.19 Å</th>
<th>X⁻⁻</th>
<th>rₐ</th>
<th>rₛ for t = 0.8</th>
<th>rₛ for t = 1.0</th>
</tr>
</thead>
<tbody>
<tr>
<td>(Cl⁻⁻ rₑ = 1.81 Å)</td>
<td>1.58 Å</td>
<td>2.43 Å</td>
<td></td>
<td></td>
</tr>
<tr>
<td>(Br⁻⁻ rₑ = 1.96 Å)</td>
<td>1.60 Å</td>
<td>2.50 Å</td>
<td></td>
<td></td>
</tr>
<tr>
<td>(I⁻⁻ rₑ = 2.20 Å)</td>
<td>1.64 Å</td>
<td>2.59 Å</td>
<td></td>
<td></td>
</tr>
</tbody>
</table>

* Effective ionic radii for coordination number of 6.

b rₛ = t × 2(rₑ + rₓ) − rₓ.¹⁰,¹¹

is 0.83, in this manner, the deviation from a perfect cubic structure is likely to happen.⁵⁰,⁵¹

In the visible range, for the MAPbI₃, the effective absorption coefficient is around 1.0 × 10⁵ (mol L⁻¹ cm⁻¹) at 550 nm,⁵² when the thickness of perovskite films range is 500–600 nm, it can absorb complete light in films. Meanwhile, organic–inorganic perovskite exhibits better charge transfer characteristics. H. J. Snaith et al.⁴ reported the diffusion lengths (L_D) of the electrons and holes in MAPbI₃ and MAPbI₃₋ₓClₓ, the L_D of MAPbI₃ is 130 nm (electrons) and 100 nm (holes) and this of MAPbI₃₋ₓClₓ is 110 nm (electrons) and 1200 nm (holes), respectively.² So, the organic–inorganic perovskite is an ideal absorber material for solar cells.

2.2 Typical PSCs structure

Some of the typical structures of PSCs are shown in Fig. 4. The typical PSCs structures include the mesoporous structure (Fig. 4(a)), the planar heterojunction structure (Fig. 4(b)) and the inverted planar heterojunction structure (Fig. 4(c)). PSCs with regular configuration is transparent conductive oxide (TCO)/blocking layer (electron transport layer (ETL))/perovskite absorber layer/hole transport layer (HTL)/metal/gold (Au). The widely accepted a simplified operation principle of PSCs is presented as: perovskite absorber layer absorbs light and generates charge while the light on the PSCS. The electrons and holes pairs are created by the thermal energy, which diffuse and get separate through electron and hole selective contacts, respectively (Fig. 4(d)).¹³ Once electrons and holes are present at the cathode and anode, respectively, external load can be powered by connecting a circuit through it.

TiO₂ is the most common ETL material,⁵⁶,⁵₇,⁵₄ meanwhile, other ETL materials have been used to attain over 10% efficiencies (e.g. ZnO,⁵₃,⁵₆ SnO₂,⁵₇–⁵₉ PCBM,⁶₀–⁶₅ LBSO,⁶₆ etc.). Spiro-OMeTAD is the widely used HTL material,⁵₅,⁵₇,⁶₇ the certified PCE of 22.1% in small cells.⁴ Meanwhile, other HTL materials have been used to achieve over 10% efficiencies (e.g. PTAA,¹⁰,⁵₆,⁵₉ P3HT,¹⁰,⁷₁ PEDOT:PSS,⁶₀,¹₁,⁷₂ CuSCN,⁷₃,⁷₄ triazine-Th-OMeTPA,²¹ PVCz-OMeDAD,⁷₆ OMeTPA-BDT,⁷₇ NiO₂,⁵₆,⁵₇,⁷₉ CuGaO₂,⁵₄ X26, ⁸⁰ X₃₆,⁸⁰ NiMgLiO⁴ etc.). Carbon materials,⁸¹–⁸₅ aluminum,⁸₆,⁸₇,⁸₉ silver,⁸₆,⁸₇,⁹₀ and gold⁹₀ have been used as electrode.³¹ Fig. 5 shows the energy levels for some commonly used ETL materials, HTL materials and absorbers materials.
3. Large-area ($\geq 1$ cm$^2$) perovskite films fabricating technologies

The continuous fabrication of cracks- and pinholes-free the perovskite films and the selective carrier extraction layers films is difficulty for the large-area PSCs devices. So, some researchers have reported many fabrication methods to improve the quality of the large-area perovskite films.

3.1 Spin-coating and vacuum flash-assisted solution process (VASP)

Spin-coating has been widely used to fabricate the large-area perovskite films. \cite{16,33,39-41,45,90} The main advantage of the spin-coating method is to deposit thin films with well-defined the composition of chemical elements and the film thicknesses. Spin-coating includes one step spin-coating and two step spin-coating. One step spin-coating, briefly, methyl ammonium iodide (MAI) and lead iodide (PbI$_2$) powders are mixed and dissolved in N,N-dimethylformamide (DMF) or dimethyl sulfoxide (DMSO), the mixed solution is spun on a TCO substrate and then annealed, attaining the perovskite films (Fig. 6). In 2015, M. Grätzel and L. Y. Han \textit{et al.} \cite{41} prepared perovskite absorber films via one step spin-coating, they achieved large-area PSCs with an active area 1.02 cm$^2$ that had a PCE > 15\% (certified 15\%). In 2016, W. Qiu and P. Heremans \textit{et al.} \cite{39} achieved large-area PSCs with 4 cm$^2$ aperture area and an active area of 1 cm$^2$, that had a PCE of 13.6\%.

Aiming at uncovered pinhole areas derive from large perovskite grains, M. J. Kim and G. H. Kim \textit{et al.} \cite{35} also developed one step spin-coating, and using high-temperature short-time annealing (HTSA) process (Fig. 7(a)), achieving the perovskite grains with sizes more than 1 μm without pinhole (Fig. 7(d, e, h, i)). In addition, the VASP was used to fabricate perovskite film (Fig. 8(a)), the sizes of perovskite grains were between 400 and 1000 nm (Fig. 8(c)), which covered the TiO$_2$ layer. \cite{37}

Two-step spin coating, briefly, MAI and PbI$_2$ powders are dissolved in DMF or DMSO, respectively. \cite{38} First, the PbI$_2$ solution is spun coating on a TCO substrate and then annealing,
achieving the PbI₂ films. Second, the MAI solution is spun coating on PbI₂ films and then annealing, achieving the perovskite films (Fig. 9). In 2016, C. Chang et al.²⁵ prepared perovskite absorber films with two step spin-coating, they achieved large-area PSCs with an active area 1.2 cm² that had a PCE of 16.2%. In 2017, E. K. Kim, J. H. Noh and S. I. Seok et al.³ achieved large-area PSCs with an active area 1 cm² that had a certified PCE of 19.7%. In 2017, X. W. Zhang and J. B. You et al.²⁸ have adopted two-step spin-coating method to fabricate the (FAPbI₃)₁₋ₓ(MAPbBr₃)ₓ films and configure n–i–p planar structure PSCs with an active area 1 cm² that has a PCE of 20.1%.

3.2 Vapor deposition
Comparing to the fabrication of the PSCs device with the spin-coating technology, vapor deposition technology offers a very superior device and superior performance (Fig. 10(a)). The vapor deposition includes dual-source evaporation technology, vapor–solid reaction, and vapor-assisted method, etc. For dual-source co-evaporation technology, it is that PbI₂ powders and MAI powders are made as target source, and pre-heated to 116 °C and 325 °C, respectively, which has achieved the PSCs yield an PCE of 15.4%. This method fabricates high quality and uniformity of the perovskite films, subsequently resulting in good performance. But this method is very dependent on high
temperature and high vacuum conditions. Alternate methods research in the literature is vapor–solid reaction (VSR), depositing the perovskite film with low temperature (Fig. 10(b)). First, the PbI$_2$ film was spin-coated onto the ETL, and then baking on a 70°C hot plate in air for 10 min. Second, MAI powders were dissolved in ethanol. Then the solution was homogeneously sprayed onto the bottom surface of the top plate that had been keeping at 80°C. Finally, inside vacuum desiccator, two parallel hot plates (PHP) were putted together to synthesize perovskite thin films. H. Zhou and S. Yin et al. used this method to achieve the 8 x 8 cm$^2$ PSCs module, the average PCE was 6.0% with the active area of 1.5 cm$^2$.

### 3.3 Gas-induced method

For the organic–inorganic halide perovskites (OIHPs) materials, gas-induce formation/transformation (GIFT) reveal surprising properties, such as gas-induced phase/morphology transformation. Z. Zhou, S. Pang, G. Cui et al. reported that the discovery of CH$_3$NH$_2$ (MA) induced phase/morphology transformation of the MAPbI$_3$. As show in Fig. 11, MA gas is introduced at room temperature (RT), after 120 min, two MAPbI$_3$ single-crystals become liquefied (MAPbI$_3$ xCH$_3$NH$_2$), eventually, merge into one liquid sphere. Then MA gas is removed, after 120 min, perovskite back-conversion completed. Fig. 12(b) shows a poor quality of MAPbI$_3$ thin film (incomplete coverage, rough), then the MA gas treatment has been introduced to create smooth, uniform and full coverage MAPbI$_3$ thin films (Fig. 12(c)).

In 2017, M. Grätzel and L. Han et al. achieved 8 x 8 cm$^2$ perovskite films via GIFT, briefly, at atmospheric environment, dried CH$_3$NH$_2$ gas (0.5 l min$^{-1}$) was passed into a bottle that contained 2 mmol CH$_3$NH$_3$I or PbI$_2$ powders (Fig. 13). After 30 min, the CH$_3$NH$_3$I powders changed into transparent colorless liquid (CH$_3$NH$_2$I:3CH$_3$NH$_3$I), and the PbI$_2$ powders...
changed into a pale-yellow paste (PbI$_2$·CH$_3$NH$_2$, Fig. 13). For the synthesis of perovskite precursor, CH$_3$NH$_2$I·3CH$_3$NH$_2$ and PbI$_2$·CH$_3$NH$_2$ were blended stoichiometrically and ultrasonicated for 15 min (Fig. 13). The perovskite precursor (200 µl) was dropped on a 8 × 8 cm$^2$ substrate and then the precursor was covered by the polyimide (PI) film. A pressure of 120 bar was loaded via a pneumatically driven squeezing board which spread the liquid precursor under the PI film. The pressure was held for 60 s and then unloaded. The thin liquid film covered with the PI film was heated at 50 °C for 2 min before peeling off the PI film. After peeling the PI film (50 mm s$^{-1}$), a dense and uniform perovskite film was formed (Fig. 14(b)). They achieved the PSCs with the device area 36 cm$^2$ (Fig. 14(c)) that had a certified PCE of 12.1%.  

### 3.4 Other approaches

In addition, a blade coating technology is also frequently used, the schematic shows in Fig. 15. The advantage of the blade
coating technology can control the distance between blade and the substrate, and the in situ thermal-treatment temperature. In 2015, S. Razza and A. D. Carlo et al. used the blade coating technology, which achieved a module PSCs with a 10.1 cm² active area that had the efficiency of 10.4%. Meanwhile, an efficiency of 4.3% had been measured for a module area of 100 cm². In 2016, J. L. Yang et al. reported an approach to fabricate ultra-long nanowires array and highly oriented CH₃NH₃PbI₃ thin films in ambient environments, briefly, this approach included large-scale roll-to-roll micro-gravure printing and doctor blading (Fig. 16), which produced perovskite nanowires lengths as long as 15 mm.

---

**Fig. 13** Diagram of the CH₃NH₂ introduced CH₃NH₃I and Pbl₂ powers and the mixture of CH₃NH₃·3CH₃NH₂ and Pbl₂·CH₃NH₂.

**Fig. 14** Diagram of the pressure processing method for the deposition of perovskite films. (a) The steps of the pressure processing method. (b) SEM images. (c) Photograph of a perovskite module.
4. Methods of improving PCE for large-area ($\geq 1 \text{ cm}^2$) perovskite solar cells

For the large-area PSCs device, improving the PCE, the first method is to change the chemical composition of perovskite, adjusting its band gap and increasing the charge generation.\textsuperscript{3,29,31,35,38,46} The second approach is to increase the grain size of perovskite, decreasing the cracks and pinholes, that reduces the bulk defect recombination and electric leakage, and increase $V_{oc}$.\textsuperscript{35–38} The third approach is interface modification, which reduces interface contact resistance, and reduce interface and surface recombination, and increase $J_{sc}$.\textsuperscript{31,41,45,95}

For the large-area PSCs device, with the increasing of cell size, the series resistance ($R_s$) increase among the charge transfer layers, the absorber layer and the electrode layers. At the same time, the number of the crack and the pinholes increase, that from the shunt resistance ($R_{sh}$) and the value of $R_{sh}$ decrease. Incorporating these resistances into the circuit
model of the solar cells device shows in Fig. 17. The increasing of \( R_s \) and the decreasing of \( R_{sh} \) increase the interface losses of the large-area PSCs device, that is the major reason of the lower efficiency for the large-area PSCs device.\(^\text{95}\)

The current expression in the circuit can be written as eqn (2).\(^\text{96}\)

\[
I = I_{SC0} - I_{D1}\left[\frac{e^{(V+IR_s)/kT}-1}{kT}\right] - I_{D2}\left[\frac{e^{(V+IR_s)/kT}-1}{kT}\right] - \frac{V + IR_s}{R_{sh}}
\]

where \( I_{SC0} \) is the short-circuit current when there are no parasitic resistances \( (R_s \) and \( R_{sh} \). The effect of these parasitic resistances on the \( I-V \) characteristic is shown in Fig. 18. Form the eqn (2), the series resistance, \( R_s \) increase, has no effect on the open-circuit voltage, but reduces the short-circuit current \( (I_{sc}) \) and fill factor \( (FF) \) (Fig. 18(a)). Conversely, the shunt resistance, \( R_{sh} \) decrease, has no effect on the short circuit current, but reduces the open-circuit voltage \( (V_{oc}) \) and FF (Fig. 18(b)).

4.1 Chemical molecular engineering

For the perovskite material, its band gap can regulate via exchange the chemical molecular or element, achieving an ideal band gap of the perovskite material. Such as, through intramolecular exchange, formamidinium (FA) molecular is frequently used to replace methylamine (MA) in MAPbI\(_{3}\), forming FAPbI\(_{3}\) and adjusting the band gap. In 2015, W. S. Yang and J. H. Noh \(^\text{35}\) have fabricated FAPbI\(_{3}\) films, its band gap is 1.47 eV smaller than MAPbI\(_{3}\) (1.50 eV). Meanwhile, the PCE of FAPbI\(_{3}\)-based PSCs is up to 20.1%. In 2017, J. Y. Kim and D. S. Kim \(^\text{35}\) fabricated \((\text{FAPbI})_{0.85} \times (\text{MAPbBr})_{0.15}\) (1.55 eV) as absorber layer of the PSCS device, which had a maximum PCE exceeding 18% over a 1 cm\(^2\) active area. In 2017, E. K. Kim, J. H. Noh and S. I. Seok \(^\text{3}\) introduced additional iodide ions into the organic cation solution, that decreased the concentration of deep-level defects. They fabricated of the \((\text{FAPbI})_{x} \times (\text{MAPbBr})_{1-x}\)-based PSCS with a certified PCE of 19.7% in 1 cm\(^2\) cells.\(^\text{3}\) Adding inorganic cesium to triple-cation perovskite compositions, E. H. Sargent \(\text{et al.}\)\(^\text{31}\) have reported the best-performance large-area (1.1 cm\(^2\)) PSCs \((\text{Cs}\times \text{FA})\times \text{MA}_{14} \times \text{PbI}_{25.5} \times \text{Br}_{45.5}, 1.60 \text{ eV}\), that has a PCE of 20.3%. In 2017, X. W. Zhang and J. B. You \(\text{et al.}\)\(^\text{38}\) used the \((\text{FAPbI})_{1-x} \times (\text{MAPbBr})_{x}\) (1.55 eV) as absorber layer for PSCS with the certified efficiency of 20.1% in large-area (1 cm\(^2\)).\(^\text{38}\)

4.2 Improving preparation technology

The high quality (cracks- and pinholes-free) large-area perovskite film is precondition for the achieving high PCE of PSCS. Because the cracks and pinholes can form electric leakage (forming the \( R_{sh} \)), which lead to the decreasing the \( V_{oc} \) and FF, and reduce the PCE of PSCS. So, M. J. Kim and G. H. Kim \(\text{et al.}\)\(^\text{93}\) developed one step spin-coating, and using high-temperature short-time annealing process (Fig. 7(a)), achieving the perovskite grains with sizes more than 1 \( \mu \)m without pinhole (HTSA-400, Fig. 7(d, e, h and i)). They fabricated PSCS device with 1 \( \text{cm}^2 \), which achieved the PCE of 18.32% with HTSA-400 (Fig. 19(d)), but the PCE is only 13.82% with HTSA-100 (Fig. 19(c)).\(^\text{35}\) X. Li and M. Grätzel \(\text{et al.}\)\(^\text{37}\) used the vacuum flash-assisted solution processing (VASP) to fabricate perovskite film (Fig. 8(a)), the sizes of perovskite grains were between 400 and 1000 nm (Fig. 8(c)). They fabricated the PSCS device with an aperture area exceeding 1 \( \text{cm}^2 \), the certified PCE of 19.6%.\(^\text{77}\) In 2015, Z. Zhou, S. Pang, G. Cui \(\text{et al.}\)\(^\text{93}\) reported that the MA gas treatment has been introduced to create smooth, uniform and full coverage MAPbI\(_{3}\) thin films (Fig. 12(c)).\(^\text{38}\) This MAPbI\(_{3}\) was used to fabricate the PSCS device, the PCE increased from 5.7% to 15.1%, was observed, which was clearly the result of the improving film morphology.\(^\text{93}\)

4.3 Interface engineering

Interface engineering can optimize interface contact, mitigate carrier recombination and increase carrier collection, which is extremely important to achieve high-performance and high-
stability PSCs. Interface engineering includes doped, plasma etching, self-assembled monolayers and interface buffer layer etc.

Doping for the charge transport layers, that can improve their electrical performance, such as improving carrier concentration and mobility. For Li–Mg co-doped NiO films, the conductivity is $2.32 \times 10^{-3}$ S cm$^{-1}$, $\sim$12 times greater than that of the pure Mg$_{x}$Ni$_{1-x}$O.41 The conductivity of Nb$^{5+}$ doped TiO$_{2}$ films is $\sim$$10^{4}$ S cm$^{-1}$, $\sim$100 to 1000 times greater than that of the pure TiO$_{2}$.41 In 2015, M. Grätzel and L. Y. Han et al.41 have used Mg–Li co-doped NiO as HTL and Nb doped TiO$_{2}$ as ETL material in inverted planar PSCs to achieve very rapid carrier extraction, increasing the cell FF from 0.64 to 0.827. Meanwhile, they fabricated a large-area (>1 cm$^{2}$) PSC with a certified efficiency of 15%.41 The contact-passivation can mitigate interfacial recombination and improve interface binding in low-temperature planar PSCs. H. R. Tan and E. H. Sargent et al.31 reported a contact-passivation strategy using chlorine-capped TiO$_{2}$ (Cl–TiO$_{2}$) colloidal nanocrystal film as ETL, the charge-recombination lifetime increased from 64 $\mu$s to 145 $\mu$s compare with pure TiO$_{2}$ film.31 They fabricated the planar PSCs for active areas of 1.1 cm$^{2}$, that achieved a certified efficiency of 19.5% without hysteresis.31 Interlayers are thin layers or monolayers of organic molecules that modify a specific interface in the solar cell.97 In 2016, C.Y. Chang and Y. C. Chang et al.45 reported an approach for the modification of interface layer via introducing thiol-functionalized self-assembled monolayers (SAMs, Fig. 21(b)), which decreased interface charge recombination and increased the value of $J_{sc}$ (19.43 mA cm$^{-2}$ to 21.68 mA cm$^{-2}$) and FF (0.67 to 0.72). They fabricated a large-area (1.2 cm$^{2}$) PSCs with the PCE up to 15.98%.45 Y. Wu and X. Yang et al.42 reported a perovskite–fullerene graded heterojunction structure, which improved the photoelectron collection and reduced recombination loss. They fabricated the PSCs of 1.022 cm$^{2}$, that had a certified PCE of 18.21%.42

5. Stability of large-area ($\geq$1 cm$^{2}$) perovskite solar cells

In recent years, the certified PCE of the large-area (1 cm$^{2}$) PSCs has achieved 20.1%.38 However, the major issue of large-area PSCs for commercial applications is the poor long-term device stability. For the stability of the perovskite materials and devices, it is necessary to consider the effects of temperature, illumination and ambient (oxygen, moisture) exposure. Many papers have reported about this important issue.1,51,53,97–107

5.1 Degradation mechanisms

The degradation of the PSCs device includes the degradation of the active layer, the degradation of charge transport layers, and the degradation of electrodes.104 The MAPbI$_{3}$ films are frequently used as absorber layer film. But the major problem with MAPbI$_{3}$ is that has thermal decomposition (exceeding 85 °C)$^{108,109}$ and water decomposition.118,119 Some researchers have reported the decomposition process of MAPbI$_{3}$. B. Philippe and H. Rensmo et al.109 exposed the MAPbI$_{3}$-Cl$_{2}$ to various environments. From the photoelectron spectroscopy results with
In the different environments, the perovskite has decomposed into PbI$_2$, but this degradation seems to occur already at 100 °C and is not only related to large humidity (Fig. 22(a)). Meanwhile, they observed a slow degradation occurs even when stored in an inert atmosphere such as argon. L. D. Wang et al. verified that oxygen, together with moisture, could lead to the irreversible degradation of MAPbI$_3$. They exposed TiO$_2$/CH$_3$NH$_3$PbI$_3$ film to air with a humidity of 60% at 35 °C for 18 h, and then, the absorption between 530 and 800 nm greatly decreased (Fig. 22(b)), the MAPbI$_3$ decomposed into PbI$_2$ and I$_2$ (Fig. 21(c)).

The degradation mechanism of MAPbI$_3$ upon exposure to moisture in absence of illumination involves the formation of hydrate form, which can be reversible. However, continuing exposure to moisture and/or exposure to illumination leads to the irreversible degradation to PbI$_2$. For ETL material, TiO$_2$ is especially sensitive to ultraviolet light, in the ultraviolet light, Ti$_4^{++}$ adsorb O$_2$ and convert into Ti$_3^{+}$, increasing the charge recombination. Meanwhile, the lithium salt in spiro-MeOTAD is easy to absorb moisture and decrease the PSCs device stability.

5.2 Methods of improving stability

In recent years, many methods have been researched to improve the PSCs device stability. Due to the poor stability of MAPbI$_3$, the first method is to modify the chemical constituents or structure of the perovskite. For example, 2D perovskites, compared with 3D perovskites, 2D perovskites have the higher carrier mobility while maintaining good ambient stability. The 2D Ruddlesden–Popper layered perovskites ((BA)$_2$(MA)$_2$Pb$_3$I$_{10}$ and (BA)$_2$(MA)$_3$Pb$_4$I$_{13}$) have been studied. H. Tsai and W. Nie et al. have achieved a PCE of 12.51% with 2D (BA)$_2$(MA)$_3$Pb$_4$I$_{13}$ PSCs device. Under the constant light illumination, after 2500 h, the 2D perovskite devices is retaining 70% of its original PCE without encapsulated and 98% with encapsulated. The 3D perovskite devices have degraded < 10% of its original PCE after 2500 h (Fig. 24(a and c)). Fig. 24(b) shows the PCE of the unencapsulated 2D and 3D devices, that shows degradation after 60 h, under 65% relative humidity. With simple encapsulation, after 2500 h, the 2D devices retained 80% of its original PCE under 65% relative humidity, but the 3D devices had...
been degraded (Fig. 24(d)).\textsuperscript{115} K. Yao \textit{et al.}\textsuperscript{46} used the poly-
ethylenimine (PEI) cations to fabricate the 2D perovskite compounds (PEI)\textsubscript{2}(MA)\textsubscript{n}–\textsubscript{1}Pb\textsubscript{n+1}I\textsubscript{3n+1} (\(n = 3, 5, 7\)), which was used as absorber layer to fabricate PSCs with an aperture area of 2.32 cm\(^2\) under ambient humidity that have a PCE up to 8.77%. After 500 h, the PCE of the 2D large-area PSCs device only decreased by \(\sim\)5%.\textsuperscript{46}

Furthermore, the alkali metal cation is introduced into the perovskite material, which can improve the stability of the PSCs device.\textsuperscript{31,68,116} E. H. Sargent \textit{et al.}\textsuperscript{31} added cesium cation to fabricate a triple-cation perovskite compositions films (Cs\textsubscript{0.05}FA\textsubscript{0.81}MA\textsubscript{0.14}Pb\textsubscript{2.55}Br\textsubscript{0.45}), that was made the large-area (1.1 cm\(^2\)) PSCs with a PCE up to 20.3% (Fig. 25(b and c)). After 90 days, the PSCs devices retained 96% of its initial PCE (Fig. 25(a)).\textsuperscript{31} Rubidium (Rb) cations can stabilize the black
phase of FA perovskite and be integrated into PSCs. M. Saliba and M. Grätzel et al.\textsuperscript{116} have used RbCsMAFAPbI\textsubscript{3} as absorber layer of the PSCs device. After 500 h at 85 °C under continuous illumination, the device has retained 95% of its initial PCE (Fig. 26(d)).\textsuperscript{116}

Fig. 24 (a and c) Photostability tests under constant AM1.5G illumination for 2D ([(BA)\textsubscript{2}(MA)\textsubscript{3}Pb\textsubscript{4}I\textsubscript{13}; red) and 3D (MAPbI\textsubscript{3}; blue) perovskite devices. (b and d) Humidity stability tests under 65% relative humidity at in a humidity chamber for 2D ([(BA)\textsubscript{2}(MA)\textsubscript{3}Pb4I\textsubscript{13}; red) and 3D (MAPbI\textsubscript{3}; blue) perovskite devices.\textsuperscript{115}

Fig. 25 Long-term device stability of PSCs with TiO\textsubscript{2}−Cl and TiO\textsubscript{2}. (a) Dark storage stability of non-encapsulated PSCs. (b) Continuous maximum power point tracking for 500 hours of a high performance unsealed CsMAFA cell with TiO\textsubscript{2}−Cl in nitrogen atmosphere under constant simulated solar illumination. (c) J−V curves of the PSCs (CsMAFA) from (b) at various stages.\textsuperscript{11}
The second method for improving the PSCs device stability is to modify the charge transport layer (ETL and HTL), or use the new type charge transport material. Because TiO₂ is especially sensitive to ultraviolet light, some new ETL materials have been reported. A. D. Carlo et al. reported an additional lithium-neutralized graphene oxide (GO-Li) layer as interface layer was inserted between TiO₂ ETL and perovskite layer, that improved the stability of PSCs devices. A. Hagfeldt et al. has used ZnO nanorod arrays as ETL replace the TiO₂, achieving the PSCs device, it has been exposed in atmospheric environment without encapsulation, and maintaining 90% of the original efficiency. X. W. Zhang and J. B. You et al. have used SnO₂ as ETL for planar-structure PSCs, it is found that the devices can maintain almost their original efficiency when store in dry air conditions for 40 days. J. H. Noh and S. I. Seok et al. used La-doped BaSnO₃ as ETL, the PSCs retained 93.3% of its initial PCE.

![Fig. 26](a) J–V curve of RbCsMAFA solar cell. (b) J–V curve of the highest- Voc device. (c) EQE electroluminescence (EL) as a function of voltage. (d) Thermal stability test of a perovskite solar cell.

![Fig. 27](a) Photograph of a module. (b) Diagram of the module structure. (c) Evolution of the photovoltaic stability of an encapsulated perovskite solar module.)
after 1000 hours, whereas the TiO₂ cells had completely degraded within 500 hours.

For the HTL materials, spiro-OMeTAD is the most commonly used HTL material, with the certified PCE of 22.1% in small cell. But the lithium salt in spiro-MeOTAD is easy to absorb moisture and reduce the PSCs device stability. So inorganic and hydrophobic hole transport material are used to improve the PSCs device stability. M. Grätzel and L. Y. Han et al. used Li₃[Os₂Cl₂(O₂CR)₂] as HTL material and Ti(Nb)O₃ as ETL to fabricate inverted planar heterojunction structure device (p-i-n), under simulated solar light, the PSCs device maintained 90% of the original efficiency after 1000 h. S. H. Yang et al. has fabricated inverted planar heterojunction structure for NiO-based PSCs device (p-i-n), achieving more than 85% of its original PCE has been kept after 150 days. Z. B. He et al. used NiO nanocrystal as HTL in planar PSCs device. After 1000 h, the PCE of PSCs device maintained 87% of its initial value. N. Arora and M. Grätzel et al. used one new HTL material CuSCN. They achieved the PSCs with PCE > 20%, after 1000 hours at 60 °C, the PSCs devices retained >95% of their initial efficiency. CuGaO₂ as HTL in n-i-p configuration PSCS, exposing it directly to the ambient environment without encapsulation. After 30 days, it maintains 87% of its initial PCE.

Other methods for improving the PSCs device stability include the PSCs structure optimization, interface optimization, encapsulation, etc. A hole-conductor-free structure of the PSCs can achieve long-term stability. Exposing the PSCs device (c-TiO₂/m-TiO₂/ZrO₂/carbon) under full AM 1.5 simulated sunlight for more than 1000 hours, the PCE maintains 100% of its initial value. To improve the stability of the device, the insulation material encapsulate the PSCs device is frequently used. M. Grätzel and L. Y. Han et al. encapsulated the large-area PSCs device (36.1 cm², TiO₂ ETL, Fig. 27(a and b)) by the insulation material, the module retained 90% of its initial performance after 500 h (Fig. 27(c)).

6. Other issues

6.1 Cost analysis

For conventional solar PV technology, it need high energy and vacuum to process solar cells. Thus, these PSCs can turn-out to be a promising solution in replacing the conventional PV technology. In this section, we briefly analyze the cost for various raw materials of a 1 m² PSCs module. Conventional PSCs device architecture is shown in Fig. 4(a), that include glass substrate, TCO (FTO), ETL (TiO₂), perovskite absorber layer (MAPbI₃), HTL (spiro-OMeTAD) and metal electrode (Au). For 1 m² conventional PSCs module, raw material utilization for cleaning, deposition of various layers and encapsulation of the module were extracted from various available literature sources and their corresponding data are included in Table 3. From the data (Table 3), it is clear that about 43% of the total raw material cost is from FTO substrate, about 34% from the HTL material (spiro-OMeTAD), and 18% from metal electrode (Au).

These data suggest the need for replacement of conventional FTO substrate, HTL material and Au electrode. The efficiency of the PSCs device on ITO-free analogues achieved 11%. Some new and cheap HTL materials have been reported, such as NiO (PCE ~ 18.47%), triazine-Th-OMeTPA (PCE ~ 12.51%), CuGaO₂ (PCE ~ 18.51%), CuSCN (PCE ~ 20.4%), NiMgLiO (PCE ~ 16.2%), etc. Meanwhile, the efficiency for carbon based HTM-free PSCs devices achieved 15.9%. Although, the PCE of spiro-OMeTAD-free PSCs device is little lower than the conventional PSCs, with small sacrifice in efficiency, low-cost and highly stable carbon based HTM-free PSCs can be fabricated.

6.2 Environmental issues – the presence of lead

Environmental issues are a well-recognized issue for PSCs. Like CdTe, a toxic heavy metal exists in the PSCs devices. But, the CdTe is very chemically stable, organolead halide perovskites are not stable and upon ambient exposure they can degrade into products that are readily leached into the environment. In the life cycle assessments (LCA), the hazards of Pb for environmental impacts exist in all stages, which include raw material extraction, synthesis of starting products, fabrication, use and decommissioning. Thus, ideally PSCs should be subject to even more stringent safety standards and
any commercial products should have clear plans for end-of-life disposal and/or recycling.124,125

To address the concerns about lead, lead-free perovskite materials have attracted the attention of many researchers, which include tin-based perovskite materials and other perovskites materials (lead-free and tin-free perovskites, such as MA₂CuCl₆Br₆₋ₓ,126 CsGeI₃, MAGeI₃, and FAGEI₃,125 A₂SbI₆ (A = Cs, Rb),126 Cs₂BiAgCl₆,127 (N-methylpyrrolidinium)₃Sb₂Br₉,128 etc.). But, compare with lead-based perovskites, the efficiencies of tin-based PSCS commonly well below 10%,129,130 the PCE values for other perovskites have been below 1%,124,126

Thus, improving encapsulation technologies, it could limit the Pb leakage during the cell operation. Researching the lead-free perovskite materials, achieving high performance lead-free PSCS device, which could to replace the lead-based PSCS device.

7. Conclusions

In this article, we briefly summarized the studies on large-area PSCS in recent years. Progress has been made in manufacturing larger area cells as well as modules, which is the interesting for commercialization of the technology. Approaches for fabricating the lager-area perovskite film layer are described such as spin-coating, vapor deposition, gas-induced and blade coating etc. It is demonstrated that these processes are useful to realize more uniform perovskite layer with larger grain size and better surface coverage, which strongly affect consequent photovoltaic performance of devices.

Going forward, PSCS will have to reduce non-radiative recombination and improve charge transport in order to achieve the highest possible $V_{oc}$ values and fill factors. For the large-area PSCS device, improving the PCE, the first method is to change the chemical composition of perovskite, adjusting its band gap and increasing the charge generation. The second approach is to increase the grain size of perovskite, decreasing the cracks and pinholes, that reduces the bulk defect recombination and electric leakage, and increase $V_{oc}$. The third approach is interface modification, which reduces interface contact resistance, and reduce interface and surface recombination, and increase $J_{sc}$. Meanwhile, one key issue of the large-area PSCS is the long-term poor stability. To the improving of the stability of PSCS, which requires interdisciplinary research to find new stable materials, the choice of electrodes, barrier layers, charge transport layers and encapsulation strategies. Undoubtedly, in the near future, halide perovskite materials have emerged as an attractive alternative to conventional silicon solar cells.

Conflicts of interest

The authors declare that there is no conflict of interests regarding the publication of this paper.

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