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The production of an efficient visible light photocatalyst for CO oxidation through the surface plasmonic effect of Ag nanoparticles on SiO₂@α-Fe₂O₃ nanocomposites

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A process for the photo deposition of noble Ag nanoparticles on a core-shell structure of SiO₂@α-Fe₂O₃ nanocomposite spheres was performed to produce a CO photo oxidation catalyst. The structural analyses were carried out for samples produced using different Ag metal nanoparticle weight percentages on SiO₂@α-Fe₂O₃ nanocomposite spheres by X-ray diffraction (XRD), field emission-scanning electron microscopy (FE-SEM), UV-vis spectroscopy, Raman spectroscopy and Fourier transform infrared spectroscopy (FTIR). A computational study was also performed to confirm the existence of the synergic effect of surface plasmon resonance (SPR) for different weight percentages of Ag on the SiO₂@α-Fe₂O₃ nanocomposites. The mechanism for CO oxidation on the catalyst was explored using diffuse reflectance infrared Fourier transform spectroscopy (DRFIT). The CO oxidation results for the Ag (2 wt%)-SiO₂@α-Fe₂O₃ nanocomposite spheres showed 48% higher photocatalytic activity than α-Fe₂O₃ and SiO₂@α-Fe₂O₃ at stable temperature.

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1. Introduction

Nowadays interest in the use of photocatalytic processes for the conversion of solar energy to chemical energy has been growing rapidly, being seen as a means to save our environment by reducing the production of greenhouse gases. Light absorption and the utilization of photoinduced electron and hole charge carriers are necessary for solar energy conversion. This conversion process includes plasmonic effects, which are used for applications such as solar water splitting, photo oxidation and degradation of organic pollutants.^{1,2} Carbon monoxide (CO) oxidation is one of the most effective reactions for the remedying of everyday environmental problems and as a consequence has received much scientific interest.³⁻⁷ For efficient conversion for CO oxidation, a metal oxide is needed, and many conditions need to be satisfied related to the loading of the metal, co-catalysts and surface adsorption.^{5,8-11} It has been found that the Fe and Co based catalysts are more suitable for CO₂ conversion and hydrocarbon removal.¹² Metals such as Ag, Au and Cu are the most easily adoptable materials for CO oxidation applications and for effective adsorption on the surface of semiconductor materials. These metals are all

important photocatalyst for the oxidation of CO and can be applied for improving organic photo oxidation.^{13,14} Recently, researchers have begun to focus on Ag-based catalysts in particular. This material has attracted intense attention because it is inexpensive and abundant in nature. Different weight percentages of Ag and different sizes of semiconductor nanostructures have been studied to examine the reduction in the recombination rate of photo-generated charges and the performance of different types of photocatalytic activity.¹⁵ In comparison with Au, Pt and Cu, Ag produces stronger plasmonic effects which enhance CO oxidation along with the advantages of low cost and higher productivity. For example, Ag or Au/TiO₂ nanorods have been used for CO oxidation and CO₂ reduction under visible light, which enhances photocatalytic activity due to the surface plasmonic resonance effect (SPR).¹⁶⁻²² The Ag based catalysts have attracted more study because of the greater surface area and better catalytic activity which are needed to achieve reasonable conversion rates of CO₂ from CO.

Recently, the metal doped n-type semiconductor catalysts have played the dominant role in photocatalytic oxidation applications, due to their nontoxicity and enormous variety of applications in industry. Further, the metal oxides, such as TiO₂, Fe₂O₃ and ZnO, which have been extensively studied as stable compounds for the prevention of dye degradation and are relatively cheaper than non-oxide semiconductors, could potentially be adapted for photodegradation and CO oxidation.²³⁻²⁸ Among the metal oxides, Fe₂O₃ has been shown to have

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the properties of being a visible light photocatalyst, earth abundant, non-toxic, and photochemically stable.²⁹ This visible light catalyst creates more excited charges and reduces the electron-hole recombination rate which is useful for improving photodegradation of inorganic pollutants and CO oxidation.^{30–32} Although the photocatalytic oxidation of carbon monoxide has been reported in many studies, there have been only a few papers about α -Fe₂O₃ core-shell structure.^{29,33} Many types of core-shell materials can be used for the development of visible light response in nanostructures due to SPR effect. Many have used SiO₂ as a shell material because of its superior chemical and photochemical stability.³⁴ Increasingly, Ag modified SiO₂ core shell nanoparticles have been shown to have upgraded optical properties.^{35,36} The core and thin shells of the nanoparticle structure could be expected to show excellent reactivity and stability for chemical looping and increased adsorption properties. This type of SiO₂ layer has the advantage of being a template free structure, with a higher adsorption capacity, so has more often been used as a core layer in recent years. Over the past decades, Fe₂O₃ has been successfully prepared by various methods to form different structures, such as cubes, rods, needles, wires, tubes, belts, disks, flakes, hollow spheres.^{37–41} Recently the preparation of self-assembled hematite hierarchical structures with promising applications in many areas using a simple template free sol-gel method has been reported.^{33,37,40–42} The SPR intensity and wavelength can be enhanced using hematite nano structure and coupling effect of nano particles such as Ag or Au.^{18,43} However, a clear understanding and evidence of the mechanism of SPR on the photolytic surface of Ag-SiO₂@ α -Fe₂O₃ nanocomposites are still needed.

In this work, the objective is to study the efficiency of CO oxidation and formation of hydrocarbons on SiO₂@ α -Fe₂O₃ nanocomposite spheres with different weight percentages of Ag nanoparticles. The SPR effects were observed on SiO₂@ α -Fe₂O₃ nanocomposites spheres. This phenomenon is proved with the computational method. In addition, the photocatalytic abilities of these products were investigated at various temperatures and almost with constant pressure using DRIFT.

2. Experimental method

2.1 Materials

The TEOS (tetraethylortho silicate), anhydrous ethanol, ammonia, iron(III) chloride hexahydrate, urea and AgNO₃ were purchased from the Aldrich. All the chemicals were analytical grade and were used as received.

2.2 Fabrication of Ag-SiO₂@ α -Fe₂O₃ nanocomposites spheres

For the preparation of the SiO₂ nanospheres, we followed the procedure detailed in our previous work.⁴⁴ Briefly, 5 mL of TEOS were mixed with 25 mL anhydrous ethanol and 5 mL water. After 30 min of stirring, 1 mL of ammonia solution was added and stirring extended for another 10 min. Then, 15 mL of anhydrous ethanol were added and the solution stirred

continuously for 12 h. Finally, the resulting solid SiO₂ spheres were washed copiously with ethanol and water using centrifugation. They were then dried in a vacuum oven at 60 °C for 10 h to remove trace amount of organic solvents. Subsequently, 100 mg of SiO₂ spheres were mixed with 50 mg of iron(III) chloride hexahydrate and 30 mg of urea then stirred continuously for 8 h at 95 °C. After stirring, the collected products were washed with water and ethanol and dried at 60 °C for 10 h. To produce crystalline SiO₂@ α -Fe₂O₃ the samples were calcined at 450 °C for 2 h.

Ag nanoparticles were photo-deposited by dispersing SiO₂@ α -Fe₂O₃ nanocomposite spheres in a 50 mL beaker with the addition of 10 mL distilled water and 10 mL ethanol. After 30 minutes of vigorous stirring, 1 wt% of AgNO₃ were added and stirring extended for another 30 min. The above solution was stirred continuously for 60 min under UV light and the final product was obtained by centrifugation and washed with water and ethanol. The Ag photo-deposited SiO₂@ α -Fe₂O₃ solid products were dried in a vacuum oven at 60 °C for 6 h. For comparison of the effect of the loading of Ag on the SiO₂@ α -Fe₂O₃ nanocomposites, nanoparticles with different weight percentages of Ag nanoparticles (2 and 3) were also prepared by the above method.

3. Characterization

The crystallographic structure of the Ag-SiO₂@ α -Fe₂O₃ nanocomposite spheres was analyzed by X-ray diffraction (XRD, with an Analytical X'Pert PRO) using filtered Cu-K α radiation ($\lambda = 1.5418 \text{ \AA}$) operated at a voltage of 45 kV and current of 40 mA. The morphology of as-prepared samples were studied and structural analysis were carried out using field emission scanning electron microscopy (JEOL JSM-7610) with energy dispersive X-ray analysis (EDX). XRF (X-ray fluorescence S₂ PICOFOX) is used to determine the chemical composition of Ag-SiO₂@ α -Fe₂O₃ nanocomposite spheres. The UV-vis diffuse reflectance spectra were captured by a UV-vis spectrometer (Agilent Cary 5000) using a reference of wavelength of 200–800 nm and Raman Spectroscopy was carried out using DONGWOODM500i, Gyeonggibo, Korea. Electrochemical impedance spectroscopy (EIS) was observed using CHI614c electrochemical analyzer. In this measurement a glassy carbon electrode (GCE) was used as a working electrode and platinum wire and an Ag/AgCl electrode (Sat. KCl) were used as counter and reference electrodes, respectively. The temperature programmed reduction of hydrogen (H₂-TPD) was measured in H₂ at the heating rate of 20 °C min⁻¹.

3.1 Reduction of CO₂ using *in situ* FTIR

In this study, the *in situ* FTIR spectrometer was fitted with a DRIFT (PerkinElmer FT-IR Spectrometer LX 10-8873, Frontier). The DRIFT cell was overlapped by a 3-window dome with two of the windows allowing for IR transparency and one for UV light radiation. The catalytic activity of the Ag-SiO₂@ α -Fe₂O₃ nanocomposite spheres was monitored by analyzing the gas composition during CO oxidation. The DRIFT experiments were



carried out from room temperature to 350 °C to find compositional changes on the catalyst surface. The Ag-SiO₂@ α -Fe₂O₃ nanocomposite spheres were examined with initial concentration of 2% in air and a flow rate of 50 standard cubic centimeter per minute (SCCM).

4. Results and discussion

4.1 X-ray diffraction

The crystal phase structures of SiO₂, α -Fe₂O₃, SiO₂@ α -Fe₂O₃ and 1, 2 and 3 wt% Ag deposited SiO₂@ α -Fe₂O₃ nanocomposite were investigated by XRD measurement, as shown in Fig. 1. The broad peaks observed at 24° correspond to the SiO₂ nanocomposite spheres assigned to the partial crystallized material. Two diffraction peaks at 32° and 35.7° indicate the hematite phase (α -Fe₂O₃) of the reference profile: JCPDS number 33-0664.⁴² No other extra peaks were observed and the intensity of the SiO₂ and α -Fe₂O₃ ascribed to the standard planes of the SiO₂ and α -Fe₂O₃ showed that maintenance of the structure of the SiO₂ after the deposition of α -Fe₂O₃. The Ag-SiO₂@ α -Fe₂O₃ spectrum of the Ag nanoparticles shows the appearance of two diffraction peaks for the nanocomposites at 38.2° and 44.3° which can be ascribed to the (111) and (200) planes. This confirms the formation of the active phase in the Ag nanoparticles on the SiO₂@ α -Fe₂O₃ surface.⁴⁵ The weak diffraction peak for the 1 wt% Ag indicates the low loading of Ag on the composite spheres. After increasing the Ag content to 2 and 3 wt%, the intensity became higher as the Ag nanoparticle content increased.

4.2 FE-SEM analysis

FE-SEM analysis was performed to show the surface morphology of the α -Fe₂O₃ and different weight percentage of Ag nanoparticles on SiO₂@ α -Fe₂O₃ nanocomposites. In Fig. 2(a) and (b) shows the SiO₂ nanospheres and the α -Fe₂O₃ nanoparticles that are nearly uniform in size and highly ordered.

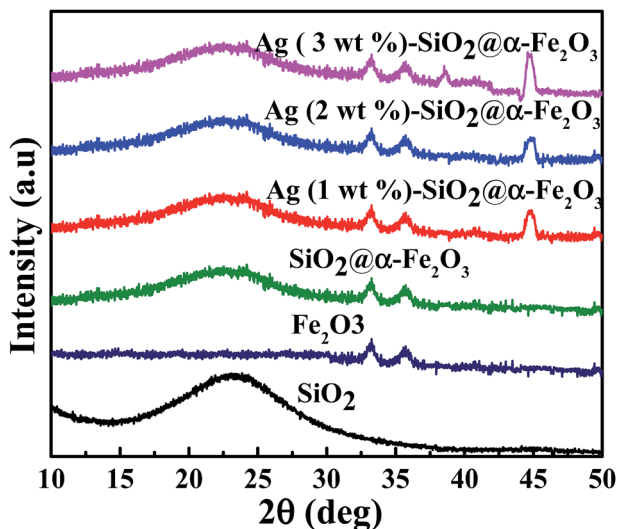


Fig. 1 XRD spectra of SiO₂, α -Fe₂O₃, SiO₂@ α -Fe₂O₃ and 1, 2 and 3 wt% Ag deposited SiO₂@ α -Fe₂O₃ nanocomposites.

Fig. 2b shows the distribution of the α -Fe₂O₃ nanoparticles on SiO₂ sphere which had an average size of 20–25 nm. After deposition of the α -Fe₂O₃ nanoparticles over SiO₂, there is not change in shape of the spheres but the surface did change from smooth to rough, suggesting the formation of a thin layer of α -Fe₂O₃ on the surface of the SiO₂. Fig. 2d–f show the 1, 2 and 3 wt% Ag deposited SiO₂@ α -Fe₂O₃ nanocomposite spheres, respectively. Ag-nanoparticle sizes of 10 to 30 nm can be detected. The Ag nanoparticles within the range of 2 to 50 nm in size are highly interactive producing the plasmonic effect.⁴⁶ For 1 and 2 wt% Ag deposited on SiO₂@ α -Fe₂O₃, the Ag nanoparticles are homogeneously deposited on the surface of the SiO₂@ α -Fe₂O₃ whereas for the 3 wt% Ag nanoparticles they are aggregated into bigger particles almost covering the SiO₂ and α -Fe₂O₃ particles. The energy dispersive X-ray analysis (EDX) results for the Ag-SiO₂@ α -Fe₂O₃ nanocomposites are shown in Fig. 2f. The mapping confirms the distribution of the α -Fe₂O₃ on the SiO₂ spheres and Ag nanoparticles dispersed on the surface of the α -Fe₂O₃ nanocomposites. XRF analysis was carried out to observe the elemental composition of Ag-SiO₂@ α -Fe₂O₃ nanocomposites. It shows the presence of Si, Fe, and Ag metals confirming the existence of these elements in the nanocomposites. From Table 1 it can be revealed that the 1, 2 and 3 wt% of Ag with respect to α -Fe₂O₃ was present and there is a no metals impurities are observed in the nanocomposites.

4.3 UV-visible spectrophotometer

Fig. 3a shows the UV-vis spectroscopic results for the SiO₂ spheres, α -Fe₂O₃, SiO₂@ α -Fe₂O₃, and Ag-SiO₂@ α -Fe₂O₃ composite spheres with different Ag percentages for investigation of the plasmonic properties as a nanocomposite catalyst. The peaks at 450 and 500 nm are confirming the absorbance of α -Fe₂O₃ (ref. 44) and there is no observable peak for SiO₂ sphere which is due to the insulating material. For the Ag (1 wt%)-SiO₂@ α -Fe₂O₃ sample, the absorbance peak at 525 nm can be attributed to the SPR effect of Ag nanoparticles, which is confirmed by D. Philip *et al.*^{47,48} An intense band centered at 525 nm related to metal charge transfer in the SiO₂@ α -Fe₂O₃ structure given the existence of Ag nanoparticles. Strong absorption of visible light was observed for the core-shell nanocomposites of Ag-SiO₂@ α -Fe₂O₃ due to localized surface plasmon resonance (SPR) with excitation of the electrons from the conduction band to the valance band and suppression of the charge recombination rate. The plasmon resonance results showed the samples with 1 and 2 wt% Ag nanoparticles coated over SiO₂@ α -Fe₂O₃ to be more active in the visible region, giving broader absorption. The active plasmonic effect was observed for the lower amounts and smaller sized Ag nanoparticles. For the Ag 3 wt% samples, the sizes of the particles were larger and with greater aggregation with one another, which decrease the SPR effect. Furthermore, the higher loading of Ag caused the accumulation and creation of new recombination centers leading to suspension of electron and charge transfer.

It is generally observed that the transfer of electron density by chemisorbed ligands results in an effect of changes in plasmon band position for the Ag nano particles with the sizes of



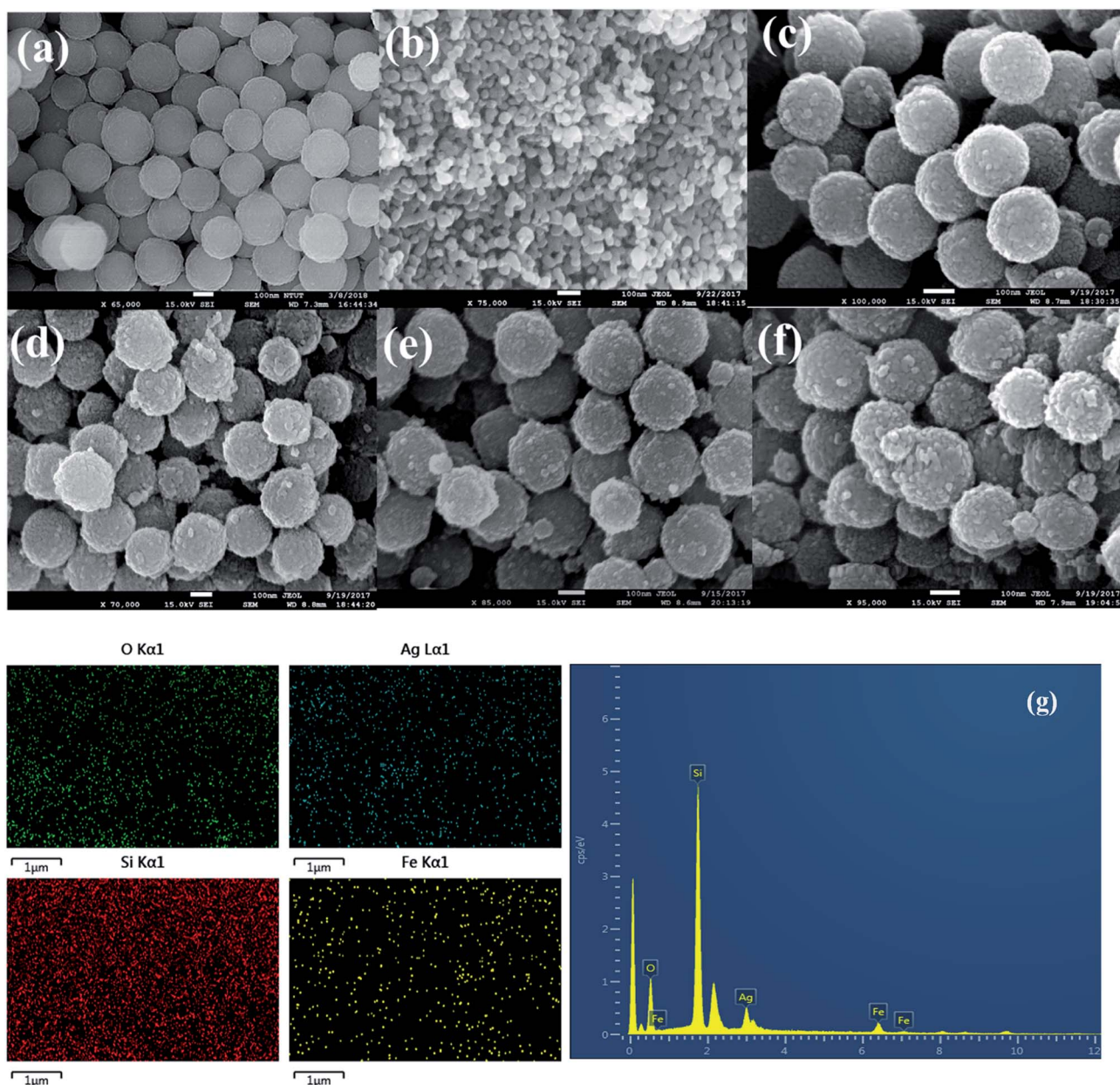


Fig. 2 FE-SEM picture of (a) SiO_2 (b) $\alpha\text{-Fe}_2\text{O}_3$, (c) $\text{SiO}_2@\alpha\text{-Fe}_2\text{O}_3$ (d) Ag (1 wt%)- $\text{SiO}_2@\alpha\text{-Fe}_2\text{O}_3$, (e) Ag (2 wt%)- $\text{SiO}_2@\alpha\text{-Fe}_2\text{O}_3$, (f) Ag (3 wt%)- $\text{SiO}_2@\alpha\text{-Fe}_2\text{O}_3$ nanocomposite and (g) EDX spectra of Ag 3 wt % - $\text{SiO}_2@\alpha\text{-Fe}_2\text{O}_3$ nanocomposite.

about 5–50 nm.^{18,49,50} In Fig. 3a, the absorbance at 525 nm observed by the effect of Surface Plasmon Resonance (SPR) for different concentration of Ag nanoparticle on $\text{SiO}_2@\alpha\text{-Fe}_2\text{O}_3$ nanocomposite with the Ag nanoparticles sizes of 10, 20 and 40 nm. The Ag nanoparticles on the $\alpha\text{-Fe}_2\text{O}_3$ photocatalyst

enhances the visible light absorption and it is useful for increasing the utilization efficiency of solar light.^{19–21} When the concentration of Ag nanoparticles increased from 1 to 2 wt%, the Ag nanoparticles sizes are increased as 10 and 20 nm, respectively. This increase leads to the optical scattering and the

Table 1 XRF analysis of the Ag- $\text{SiO}_2@\alpha\text{-Fe}_2\text{O}_3$, nanocomposites with different Ag weight percentage

Samples	Si concentration	Fe concentration	Ag concentration
$\text{SiO}_2@\alpha\text{-Fe}_2\text{O}_3$	93.25	6.753	0
Ag (1 wt%)- $\text{SiO}_2@\alpha\text{-Fe}_2\text{O}_3$	92.18	6.756	0.071
Ag (2 wt%)- $\text{SiO}_2@\alpha\text{-Fe}_2\text{O}_3$	92.25	6.751	0.147
Ag (3 wt%)- $\text{SiO}_2@\alpha\text{-Fe}_2\text{O}_3$	91.65	6.753	0.210



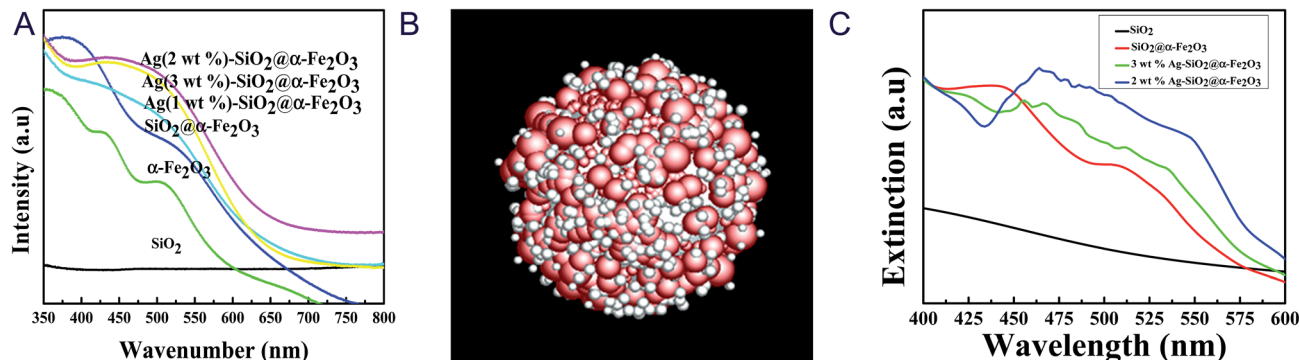


Fig. 3 (a) UV-vis spectra of SiO_2 , $\alpha\text{-Fe}_2\text{O}_3$, $\text{SiO}_2@ \alpha\text{-Fe}_2\text{O}_3$ and 1, 2 and 3 wt% Ag deposited $\text{SiO}_2@ \alpha\text{-Fe}_2\text{O}_3$ nanocomposites. (b) Simulation geometry with Ag nanoparticles. The rough thickness of the red $\alpha\text{-Fe}_2\text{O}_3$ shell is about 25 nm and the radius of the white SiO_2 core is 125 nm. (c) UV-vis simulation spectra of SiO_2 , $\text{SiO}_2@ \alpha\text{-Fe}_2\text{O}_3$ and Ag with radius of 15 nm and 40 nm deposited $\text{SiO}_2@ \alpha\text{-Fe}_2\text{O}_3$ nanocomposites.

electron fields are spatially non-homogeneous. Further, it was found that the possibility of energy transfer between Ag and semiconductor can be maximally enhanced when the size of the metal particles are around 20 nm.⁵¹ In addition, when plasma induced photoexcited from SRP effect, the electrons transfer to the conduction band of $\alpha\text{-Fe}_2\text{O}_3$ that enhances the photocatalytic activity when irradiates the visible light. When particles sizes are above 40 nm with accumulation, the 3 wt% Ag nanoparticles are not strong enough to excite the electron to the semiconductor and the electron-hole pair recombination is occurred, thus leads to lower the plasmonic effect.

We simulated the UV-visible spectra to observe the real conditions of the plasmonic effects ascribed to the different weight percentages of Ag nanoparticles as shown in Fig. 3b. In order to understand the effect of the silver nanoparticles on the extinction spectra of the Ag- $\text{SiO}_2@ \alpha\text{-Fe}_2\text{O}_3$ -shell nanoparticle, finite-difference time-domain (FDTD) method is adopted which solves the Maxwell's equations. Fig. 3b is the core shell nanoparticle without Ag, where the thickness of $\alpha\text{-Fe}_2\text{O}_3$ rough shell (red spheres) is around 40 nm and the SiO_2 core (white core) has a radius of 125 nm. Fig. 3b is the core shell nanoparticle with Ag, where the radii of the random silver nanoparticles are 10 nm and 15 nm. The simulation domain is $1.5 \mu\text{m} \times 1.5 \mu\text{m} \times 1.5 \mu\text{m}$ with uniform mesh, $\Delta x = \Delta y = \Delta z = 3 \text{ nm}$. The refractive index of $\alpha\text{-Fe}_2\text{O}_3$ is set according to M. Gartner *et al.*⁵² The refractive indexes of SiO_2 and Silver are fitted by a FDTD package Lumerical Inc. according to Palik and Johnson and Christy's experiment results. The total field scattered field is selected and the wavelength is set from 300 nm to 800 nm. There are six monitors inside the source to collect the absorption power and six monitors outside the source to collect the scattering power. The extinction power is the sum of the absorption power and the scattering power. Fig. 3c shows the extinction power of SiO_2 , $\text{SiO}_2@ \alpha\text{-Fe}_2\text{O}_3$, Ag- $\text{SiO}_2@ \alpha\text{-Fe}_2\text{O}_3$ with $r_{\text{ag}} = 20 \text{ nm}$, and Ag- $\text{SiO}_2@ \alpha\text{-Fe}_2\text{O}_3$ with $r_{\text{ag}} = 7 \text{ nm}$. Adding silver nanoparticles reduces the scattering power in short wavelength around 300 nm but increases the absorption power in a broad range from 400 nm to 600 nm. Comparing to the UV-vis Spectra in Fig. 3a, the plasmonic effect is confirmed with a similar trend.

4.4 Raman Spectroscopy

The Raman spectra of the SiO_2 , $\alpha\text{-Fe}_2\text{O}_3$, $\text{SiO}_2@ \alpha\text{-Fe}_2\text{O}_3$, Ag- $\text{SiO}_2@ \alpha\text{-Fe}_2\text{O}_3$ nanocomposite with different Ag weight percentages are shown in Fig. 4. The strong peaks observed around 225, 247, 299, 412, 497 and 613 cm^{-1} confirm the presence of $\alpha\text{-Fe}_2\text{O}_3$. The figure shows the clear Raman effects of $\alpha\text{-Fe}_2\text{O}_3$. The peaks at 305, 513 and 660 cm^{-1} belonging to pure Fe_2O_3 were not observed.⁵³ However, the Ag-related signal was not observed for 1 and 2 wt% Ag because of the weak Raman scattering except for the 3 wt% Ag nanoparticles. The Raman peak indicates that the presence of $\alpha\text{-Fe}_2\text{O}_3$ on the surface of the $\alpha\text{-Fe}_2\text{O}_3$ without any phase change.

4.5 FTIR spectroscopy results

Fig. 5 shows the chemical structure of the SiO_2 , $\alpha\text{-Fe}_2\text{O}_3$, $\text{SiO}_2@ \alpha\text{-Fe}_2\text{O}_3$, Ag- $\text{SiO}_2@ \alpha\text{-Fe}_2\text{O}_3$ nanocomposites with different Ag weight percentages. The peak at 1080–1105 cm^{-1} demonstrates the symmetric vibration of Si–O–Si and 940–960 cm^{-1} which can be assigned to the Si–OH vibration. The absorption band at 558 cm^{-1} in the curve is attributed to the

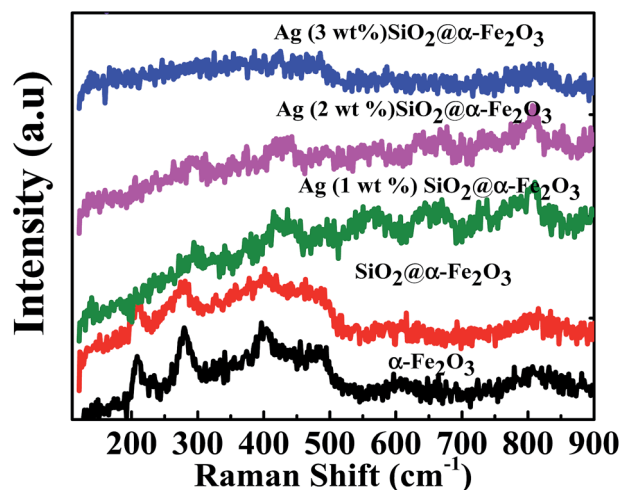


Fig. 4 Raman spectra of the SiO_2 , $\alpha\text{-Fe}_2\text{O}_3$, $\text{SiO}_2@ \alpha\text{-Fe}_2\text{O}_3$, Ag- $\text{SiO}_2@ \alpha\text{-Fe}_2\text{O}_3$ nanocomposite with different Ag weight percentages.



bending vibrations of the Fe–O in α -Fe₂O₃ spectra. In addition, the absorption band at about 685 cm⁻¹, observed in α -Fe₂O₃ that can be assigned to the bending modes of Fe–O–H.⁵⁴ Moreover, there is not much difference in chemical structure of the SiO₂@ α -Fe₂O₃ composites spheres with various Ag compositions and indicates that the different weight percentages of Ag did not interfere with the structure of the SiO₂@ α -Fe₂O₃. This means that the catalytic active force between Ag and SiO₂@ α -Fe₂O₃ nanocomposites is a physical process, rather than chemical reaction.

4.6 PL spectroscopy

The PL spectra of SiO₂ sphere, SiO₂@ α -Fe₂O₃, SiO₂@ α -Fe₂O₃, Ag (1 wt%)-SiO₂@ α -Fe₂O₃, Ag (2 wt%)-SiO₂@ α -Fe₂O₃ and Ag (3 wt%)-SiO₂@ α -Fe₂O₃ samples are shown in Fig. 6. The SiO₂ sphere shows the broad-band emission around 450 nm. In SiO₂@ α -Fe₂O₃ nanocomposites, the emission at 540 nm can be assigned to the recombination of photoexcited holes with electrons inhabiting the singly ionized oxygen vacancies in α -Fe₂O₃.⁵⁵ After depositing Ag nanoparticles, the recombination peak at 540 nm is largely decreased, indicating that the recombination of electron–hole pairs has been reduced. Ag (2 wt%) SiO₂@ α -Fe₂O₃ nanocomposites shows minimum intense PL spectra. This low intensity suggests that the surface modified Ag (3 wt%) SiO₂@ α -Fe₂O₃ nanocomposites have significant recombination of photogenerated electrons–hole pairs when compared with other two Ag percentages.

4.7 Electrochemical impedance study

Electrochemical impedance measurement was also performed for the SiO₂ sphere, α -Fe₂O₃, SiO₂@ α -Fe₂O₃, SiO₂@ α -Fe₂O₃, Ag (1 wt%)-SiO₂@ α -Fe₂O₃, Ag (2 wt%)-SiO₂@ α -Fe₂O₃ and Ag (3 wt%)-SiO₂@ α -Fe₂O₃ samples as shown in Fig. 7. The small semicircles observed for the 2 wt% of Ag showing the electron transfer impedance of the electrode surface specifying the better charge transfer resistance on the Ag (2 wt%)-SiO₂@ α -

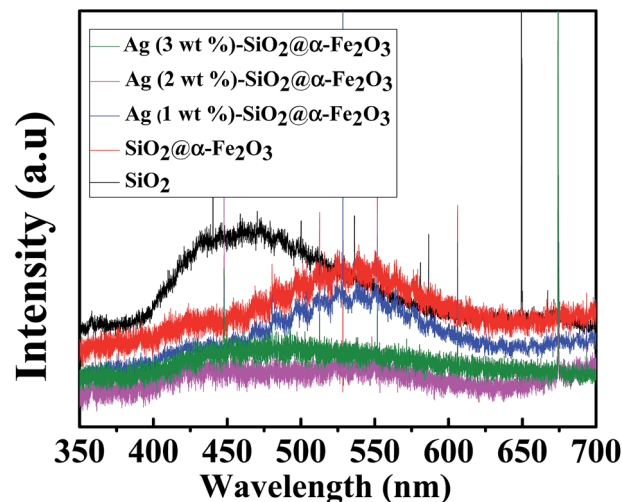


Fig. 6 PL spectra for SiO₂ sphere, α -Fe₂O₃, SiO₂@ α -Fe₂O₃, SiO₂@ α -Fe₂O₃, Ag (1 wt%)-SiO₂@ α -Fe₂O₃, Ag (2 wt%)-SiO₂@ α -Fe₂O₃ and Ag (3 wt%)-SiO₂@ α -Fe₂O₃ samples.

Fe₂O₃ interface. The diameter of the arc radius on the EIS Nyquist plot is very low due to a more effective separation of photogenerated electron–hole pairs and the faster interface charge transfer rate which appeared for the 2 wt% Ag nanoparticles SiO₂@ α -Fe₂O₃ composites spheres. It is also worth mentioning that the higher impedance values produced for the 2 wt% Ag on SiO₂@ α -Fe₂O₃ composites spheres when compared to 1 and 3 wt% Ag on SiO₂@ α -Fe₂O₃ composites spheres. The improvement in the impedance of the charge transfer of electrons from the SiO₂@ α -Fe₂O₃ to Ag nanoparticles is confirmed by the UV-vis spectra as shown in Fig. 3

4.8 Temperature programmed desorption (H₂-TPD)

Fig. 8 shows the H₂-TPD profiles of the α -Fe₂O₃, SiO₂@ α -Fe₂O₃ and Ag (2 wt%)-SiO₂@ α -Fe₂O₃ nanocomposite spheres. For α -Fe₂O₃ and SiO₂@ α -Fe₂O₃, we can realize that the peak

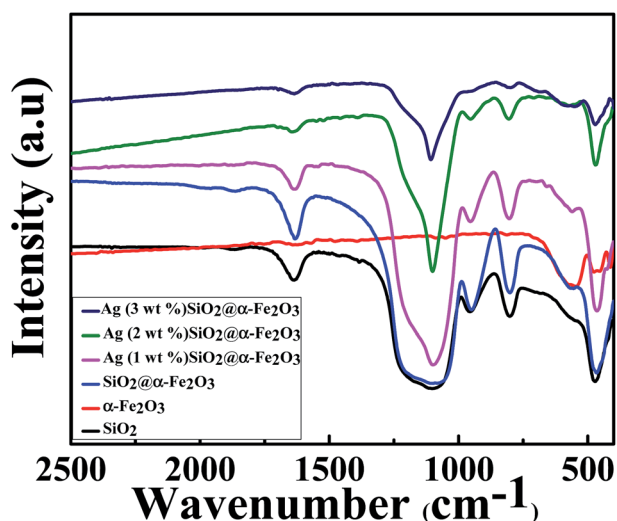


Fig. 5 FT-IR spectra of the SiO₂, α -Fe₂O₃, SiO₂@ α -Fe₂O₃, Ag-SiO₂@ α -Fe₂O₃ nanocomposites with different Ag weight percentages.

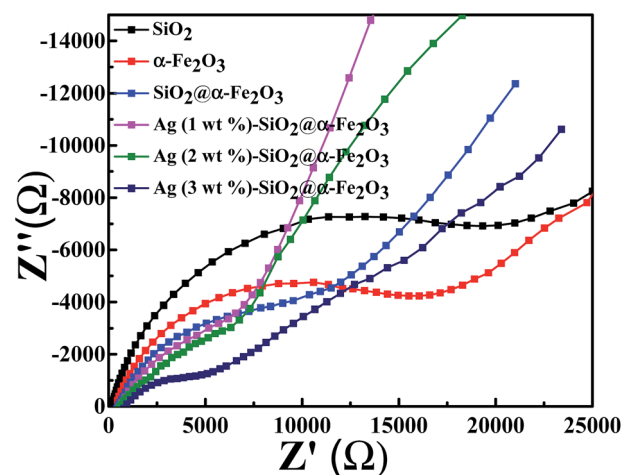


Fig. 7 Electrochemical impedance spectra for SiO₂ sphere, α -Fe₂O₃, SiO₂@ α -Fe₂O₃, SiO₂@ α -Fe₂O₃, SiO₂@ α -Fe₂O₃, Ag (1 wt%)-SiO₂@ α -Fe₂O₃, Ag (2 wt%)-SiO₂@ α -Fe₂O₃ and Ag (3 wt%)-SiO₂@ α -Fe₂O₃ samples.



maximized at 500 °C is assigned to the reduction of Fe_2O_3 to Fe_3O_4 as confirmed in previous work.⁵⁶ The other peak at 600 °C attributed to FeO to Fe^0 . In addition, the weak peak observed around 200 °C is probably arises from the movement of Ag nanoparticles to the adjacent $\text{Ag-SiO}_2@ \alpha\text{-Fe}_2\text{O}_3$ nanocomposites. It indicates that after adding Ag nanoparticles, the catalytic activity increases that leads to the CO oxidation due to high plasmonic effect.

4.9 Photocatalytic oxidation of CO examined with DRIFT

The DRIFT was monitored to collect information about the temperature dependent gaseous phase on $\text{Ag (2 wt\%)} \text{SiO}_2@ \alpha\text{-Fe}_2\text{O}_3$ nanocomposites during CO oxidation reaction as shown in Fig. 9. With the introduction of CO to the $\text{Ag (2 wt\%)} \text{SiO}_2@ \alpha\text{-Fe}_2\text{O}_3$ nanocomposites at room temperature, peaks appear at 2176 and 2142 cm^{-1} which are due to the gaseous carbon monoxide doublet. In Fig. 9, as the temperature increases to 100 °C, new peaks appear at 2353 cm^{-1} and 2362 cm^{-1} , implying CO_2 absorption and adsorbed doublets of carbon dioxide which are visible in the spectrum. The CO_2 peak reflects the reaction between CO and adsorbed oxygen, which leads to the formation of reactive formats which can be easily oxidized by oxygen absorbed by the silver nanoparticles.⁵⁷ The intensity of carbon dioxide doublet peaks increases with increasing the temperature to 300 °C. Furthermore, new bands also appear at 1621, 1414 and 1222 cm^{-1} indicating the formation of a bicarbonate after the adsorption of CO_2 on $\alpha\text{-Fe}_2\text{O}_3$. This confirms that the main reason for the oxidation process is based on the presence of Ag nanoparticles with high CO oxidation activity. These results agree well with the H_2 -TPD results. For the $\text{Ag-SiO}_2@ \alpha\text{-Fe}_2\text{O}_3$ nanocomposite spheres, a high intensity efficient peak at 2353 cm^{-1} and 2362 cm^{-1} can be observed at 250 and 300 °C, as shown in the Fig. 8. The bicarbonates remained stable at temperatures up to 350 °C, after which the band at 2134 cm^{-1} was decreased, which can be attributed to the low

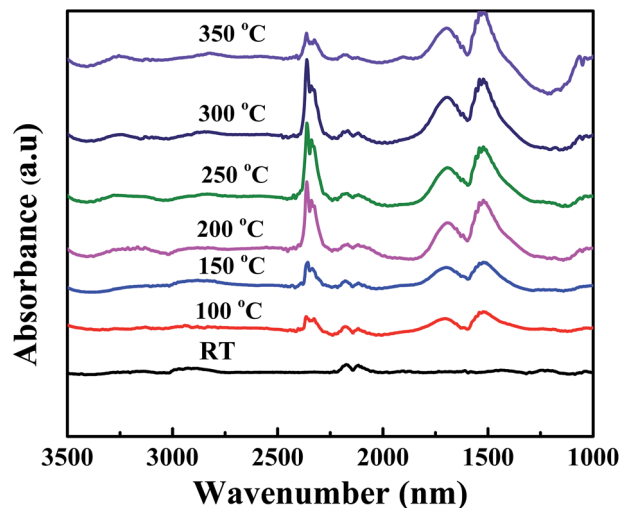


Fig. 9 DRIFT spectra of $\text{Ag (2 wt\%)} \text{SiO}_2@ \alpha\text{-Fe}_2\text{O}_3$ nanocomposites measured at different temperature during CO oxidation.

reactivity between the CO and the adsorbed oxygen. Furthermore, when the temperature was increased to 350 °C there was a very less effective reaction occurring between the CO and adsorbed oxygen due to the lack of the $-\text{OH}$ group.

One should note that CO oxidation is based on the exothermic reaction and it generates more $-\text{OH}$ groups on the surface of the adsorbent. When CO molecules act as an electron donor that adsorbed on the surface of Ag and oxygen in the Ag are responsible for the oxidation of CO. During the photo oxidation of CO using different concentration of Ag on $\text{SiO}_2@ \alpha\text{-Fe}_2\text{O}_3$ nanocomposites catalyst, the SiO_2 spheres act as a substrate as well as a good absorbent, while the $\alpha\text{-Fe}_2\text{O}_3$ acts as a semiconductor for the generation of electrons and holes. Moreover, the Ag nanoparticles on the $\text{SiO}_2@ \alpha\text{-Fe}_2\text{O}_3$ catalyst assist the trapping of electrons through the conduction band of $\alpha\text{-Fe}_2\text{O}_3$ which diminishes the electron-hole recombination rate. However, the presence of Ag nanoparticles on the surface of $\alpha\text{-Fe}_2\text{O}_3$ generally increases the contact area between Ag nanoparticles and $\alpha\text{-Fe}_2\text{O}_3$. Further, an Ag atom and CO molecules give a strong interactions between the bridging oxygen and the surface of $\alpha\text{-Fe}_2\text{O}_3$ due to the plasmonic effect, thus leads to photo oxidation. When the temperature increased from 100 to 350 °C the catalytic activity of Ag nanoparticles generating more and more superoxide radicals which leads to CO oxidation. Interestingly, when temperature increases from 250 to 300 °C the more amount of CO oxidation occurred and high SPR effect from Ag nanoparticles also helped to improve the CO oxidation activity. The plasmonic electron injection from SPR from Ag nano particles leads to generate more resonant photons for photon harvesting, thereby facilitating the photocatalytic process.^{47,58} This work confirmed that the SPR and thermal energy are responsible for the transfer of energetic electrons from the Ag metal to the semiconductor or a consequence of the direct interaction of surface plasmon with adsorbents.^{59,60} The SPR can utilize thermal energy and a low-intensity photon flux to improve the rate of catalytic oxidation reactions at significant temperatures on the surface of the catalysts.

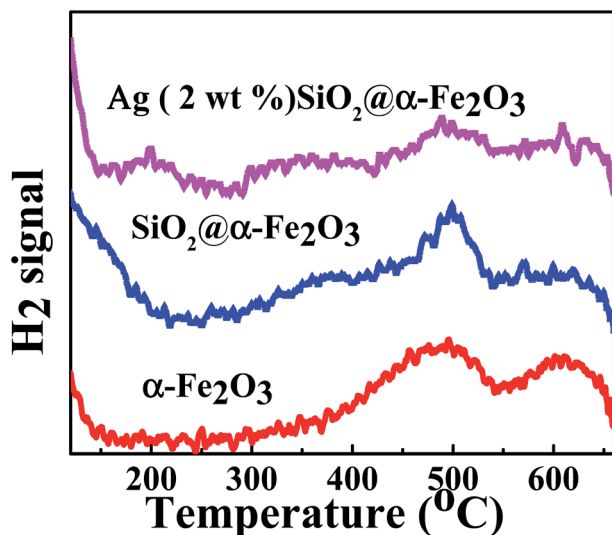


Fig. 8 H_2 -TPD profiles of the $\alpha\text{-Fe}_2\text{O}_3$, $\text{SiO}_2@ \alpha\text{-Fe}_2\text{O}_3$ and $\text{Ag (2 wt\%)} \text{SiO}_2@ \alpha\text{-Fe}_2\text{O}_3$ nanocomposite spheres.



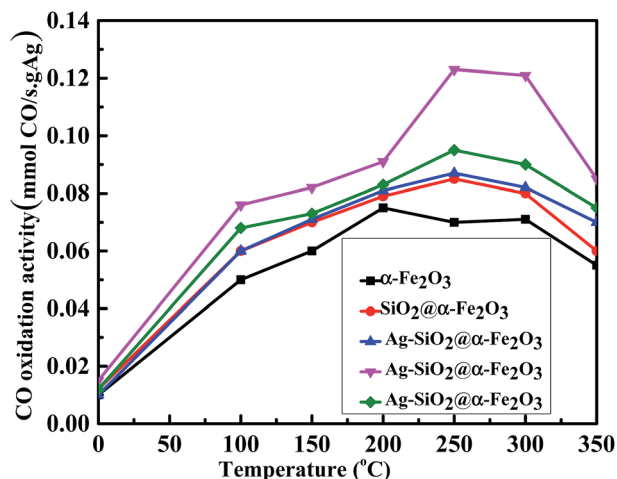


Fig. 10 Effect of CO oxidation activity in $\alpha\text{-Fe}_2\text{O}_3$, $\text{SiO}_2@\alpha\text{-Fe}_2\text{O}_3$, $\text{Ag-SiO}_2@\alpha\text{-Fe}_2\text{O}_3$, $\text{Ag-SiO}_2@\alpha\text{-Fe}_2\text{O}_3$ and $\text{Ag-SiO}_2@\alpha\text{-Fe}_2\text{O}_3$ samples.

Fig. 10 shows the rate of conversion of CO to CO_2 using 1, 2 and 3 wt% Ag nanoparticles on $\text{SiO}_2@\alpha\text{-Fe}_2\text{O}_3$ catalysts from room temperature to 350 °C. It is clear that 2 wt% $\text{Ag-SiO}_2@\alpha\text{-Fe}_2\text{O}_3$ catalyst activity is more than 48% higher than that of pure Fe_2O_3 and $\text{SiO}_2@\alpha\text{-Fe}_2\text{O}_3$. The rate of charge-carrier transfer increases during the intermediate plasmon phase between Ag (2 wt%) and the $\text{SiO}_2@\alpha\text{-Fe}_2\text{O}_3$ nanocomposites. This proves that the CO oxidation readily takes place on the 2 wt% $\text{Ag-SiO}_2@\alpha\text{-Fe}_2\text{O}_3$ catalyst surface when oxygen is available in atomic form.^{61,62} The CO oxidation rate of 3 wt% Ag is lower when compared to others that are higher recombination rates due to the aggregation of particles with recession in terms of the photo absorption and resonant energy transfer. Further, the photo excitations are dependent on the Ag concentration which reduces the oxidation performance in 3 wt% $\text{Ag-SiO}_2@\alpha\text{-Fe}_2\text{O}_3$. This indicates that there is low inter-reaction between the CO and adsorbed oxygen and excess Ag may cover the nanocomposites and hinder direct contact with the $\text{SiO}_2@\alpha\text{-Fe}_2\text{O}_3$ catalyst.

5. Conclusions

A facile method for the photo deposition of different weight percentage Ag nanoparticles on $\text{SiO}_2@\alpha\text{-Fe}_2\text{O}_3$ nanocomposites was investigated for proper CO oxidation at different temperatures. The most effective CO oxidation was observed to occur at 250 °C with Ag nanoparticles at a weight percentage of 2. The improved photocatalytic activity is attributed to the plasmonic effect of the Ag and the charge transfer property. The computation method proved that the high plasmonic effect was observed on the $\text{Ag-SiO}_2@\alpha\text{-Fe}_2\text{O}_3$ nano composites materials. The Ag loaded $\text{SiO}_2@\alpha\text{-Fe}_2\text{O}_3$ catalyst is shown to be a promising material for conversion of solar energy into CO oxidation.

Conflicts of interest

There are no conflicts to declare.

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References

- R. Li, H. Han, F. Zhang, D. Wang and C. Li, *Energy Environ. Sci.*, 2014, 7, 1369–1376.
- J. Yang, D. Wang, H. Han and C. Li, *Acc. Chem. Res.*, 2013, 46, 1900–1909.
- I. H. Kim, H. O. Seo, E. J. Park, S. W. Han and Y. D. Kim, *Sci. Rep.*, 2017, 7, 40497.
- S. Gatla, D. Aubert, G. Agostini, O. Mathon, S. Pascarelli, T. Lunkenbein, M. G. Willinger and H. Kaper, *ACS Catal.*, 2016, 6, 6151–6155.
- C. Wu, R. Brescia, M. Prato, S. Marras, L. Manna and M. Colombo, *J. Mater. Chem. A*, 2016, 4, 18075–18083.
- Y. Yi, P. Zhang, Z. Qin, C. Yu, W. Li, Q. Qin, B. Li, M. Fan, X. Liang and L. Dong, *RSC Adv.*, 2018, 8, 7110–7122.
- F. Du, G. Wu, D. Mao and G. Lu, *RSC Adv.*, 2016, 6, 111070–111078.
- W. Leitner, *Coord. Chem. Rev.*, 1996, 153, 257–284.
- M. Aresta, in *Carbon Dioxide as Chemical Feedstock*, Wiley-VCH Verlag GmbH & Co. KGaA, 2010, pp. 1–13, DOI: 10.1002/9783527629916.ch1.
- Y. Tang, J. Zhou, Z. Shen, W. Chen, C. Li and X. Dai, *RSC Adv.*, 2016, 6, 93985–93996.
- Y. Guo, C. Li, S. Lu and C. Zhao, *RSC Adv.*, 2016, 6, 7181–7188.
- D. Schanke, A. M. Hilmen, E. Bergene, K. Kinnari, E. Rytter, E. Ådnanes and A. Holmen, *Catal. Lett.*, 1995, 34, 269–284.
- M. Murdoch, G. I. N. Waterhouse, M. A. Nadeem, J. B. Metson, M. A. Keane, R. F. Howe, J. Llorca and H. Idriss, *Nat. Chem.*, 2011, 3, 489.
- M. M. Trandafir, L. Pop, N. D. Hadade, M. Florea, F. Neațu, C. M. Teodorescu, B. Duraki, J. A. van Bokhoven, I. Grosu, V. I. Parvulescu and H. Garcia, *Catal. Sci. Technol.*, 2016, 6, 8571–8573.
- D. P. Dutta and A. K. Tyagi, *Mater. Res. Bull.*, 2016, 74, 397–407.
- X. Ren, E. Cao, W. Lin, Y. Song, W. Liang and J. Wang, *RSC Adv.*, 2017, 7, 31189–31203.
- T. S. Rodrigues, A. G. M. da Silva, A. B. L. de Moura, I. G. Freitas and P. H. C. Camargo, *RSC Adv.*, 2016, 6, 62286–62290.
- K. Uma, N. Arjun, G.-T. Pan and T. C. K. Yang, *Appl. Surf. Sci.*, 2017, 425, 377–383.
- C. Peng, W. Wang, W. Zhang, Y. Liang and L. Zhuo, *Appl. Surf. Sci.*, 2017, 420, 286–295.



- 20 X. Zhang, Y. Wang, F. Hou, H. Li, Y. Yang, X. Zhang, Y. Yang and Y. Wang, *Appl. Surf. Sci.*, 2017, **391**, 476–483.
- 21 X. Yang, Y. Wang, X. Xu, Y. Qu, X. Ding and H. Chen, *Chin. J. Catal.*, 2017, **38**, 260–269.
- 22 J. Low, J. Yu, M. Jaroniec, S. Wageh and A. A. Al-Ghamdi, *Adv. Mater.*, 2017, **29**, 1601694.
- 23 E. Liu, Y. Hu, H. Li, C. Tang, X. Hu, J. Fan, Y. Chen and J. Bian, *Ceram. Int.*, 2015, **41**, 1049–1057.
- 24 T. C. K. Yang, S.-F. Wang, S. H. Y. Tsai and S.-Y. Lin, *Appl. Catal., B*, 2001, **30**, 293–301.
- 25 M. Ni, M. K. H. Leung, D. Y. C. Leung and K. Sumathy, *Renewable Sustainable Energy Rev.*, 2007, **11**, 401–425.
- 26 A. Sarkar, A. K. Singh, G. G. Khan, D. Sarkar and K. Mandal, *RSC Adv.*, 2014, **4**, 55629–55634.
- 27 G. Liu, L. Wang, H. G. Yang, H.-M. Cheng and G. Q. Lu, *J. Mater. Chem.*, 2010, **20**, 831–843.
- 28 M. A. Henderson, *Surf. Sci. Rep.*, 2011, **66**, 185–297.
- 29 H. Shan, C. Liu, L. Liu, L. Wang, S. Li, X. Zhang, X. Bo and X. Chi, *Sci. China: Chem.*, 2013, **56**, 1722–1726.
- 30 J.-X. Sun, Y.-P. Yuan, L.-G. Qiu, X. Jiang, A.-J. Xie, Y.-H. Shen and J.-F. Zhu, *Dalton Trans.*, 2012, **41**, 6756–6763.
- 31 R. Asahi, T. Morikawa, H. Irie and T. Ohwaki, *Chem. Rev.*, 2014, **114**, 9824–9852.
- 32 L. Wang and T. Sasaki, *Chem. Rev.*, 2014, **114**, 9455–9486.
- 33 M. Cao, T. Liu, S. Gao, G. Sun, X. Wu, C. Hu and Z. L. Wang, *Angew. Chem., Int. Ed.*, 2005, **44**, 4197–4201.
- 34 H.-J. Zhang, H.-M. Xiong, Q.-G. Ren, Y.-Y. Xia and J.-L. Kong, *J. Mater. Chem.*, 2012, **22**, 13159–13165.
- 35 J. C. Flores, V. Torres, M. Popa, D. Crespo and J. M. Calderón-Moreno, *J. Non-Cryst. Solids*, 2008, **354**, 5435–5439.
- 36 M. Shanthil, R. Thomas, R. S. Swathi and K. George Thomas, *J. Phys. Chem. Lett.*, 2012, **3**, 1459–1464.
- 37 L. S. Zhong, J. S. Hu, H. P. Liang, A. M. Cao, W. G. Song and L. J. Wan, *Adv. Mater.*, 2006, **18**, 2426–2431.
- 38 X. Su, C. Yu and C. Qiang, *Appl. Surf. Sci.*, 2011, **257**, 9014–9018.
- 39 X. Hu, J. C. Yu and J. Gong, *J. Phys. Chem. C*, 2007, **111**, 11180–11185.
- 40 L.-P. Zhu, H.-M. Xiao, X.-M. Liu and S.-Y. Fu, *J. Mater. Chem.*, 2006, **16**, 1794–1797.
- 41 S.-W. Cao and Y.-J. Zhu, *J. Phys. Chem. C*, 2008, **112**, 6253–6257.
- 42 H. Cui, Y. Liu and W. Ren, *Adv. Powder Technol.*, 2013, **24**, 93–97.
- 43 A. Moores and F. Goettmann, *New J. Chem.*, 2006, **30**, 1121–1132.
- 44 K. Uma, N. Arjun, G.-T. Pan and T. C. K. Yang, *Appl. Surf. Sci.*, 2017, **425**, 377–383.
- 45 Y. Tang, V. P. Subramaniam, T. H. Lau, Y. Lai, D. Gong, P. D. Kanhere, Y. H. Cheng, Z. Chen and Z. Dong, *Appl. Catal., B*, 2011, **106**, 577–585.
- 46 D. D. Evanoff and G. Chumanov, *ChemPhysChem*, 2005, **6**, 1221–1231.
- 47 D. Philip and G. Aruldas, *J. Solid State Chem.*, 1995, **114**, 129–137.
- 48 D. Philip, A. John, C. Y. Panicker and H. T. Varghese, *Spectrochim. Acta, Part A*, 2001, **57**, 1561–1566.
- 49 K. B. Mogensen and K. Kneipp, *J. Phys. Chem. C*, 2014, **118**, 28075–28083.
- 50 Y. Lin, G. Yuan, R. Liu, S. Zhou, S. W. Sheehan and D. Wang, *Chem. Phys. Lett.*, 2011, **507**, 209–215.
- 51 C. Sönnichsen, T. Franzl, T. Wilk, G. v. Plessen and J. Feldmann, *New J. Phys.*, 2002, **4**, 93.
- 52 M. Gartner, M. Crisan, A. Jitianu, R. Scurtu, R. Gavrilă, I. Oprea and M. Zaharescu, *J. Sol-Gel Sci. Technol.*, 2003, **26**, 745–748.
- 53 X. Zhang, Y. Niu, X. Meng, Y. Li and J. Zhao, *CrystEngComm*, 2013, **15**, 8166–8172.
- 54 V. A. Bondzie, S. C. Parker and C. T. Campbell, *J. Vac. Sci. Technol., A*, 1999, **17**, 1717–1720.
- 55 L. Yuan, R. Cai, J. I. Jang, W. Zhu, C. Wang, Y. Wang and G. Zhou, *Nanoscale*, 2013, **5**, 7581–7588.
- 56 G. Munteanu, L. Ilieva and D. Andreeva, *Thermochim. Acta*, 1997, **291**, 171–177.
- 57 M. Khoudiakov, M. C. Gupta and S. Deevi, *Appl. Catal., A*, 2005, **291**, 151–161.
- 58 V. Amendola, O. M. Bakr and F. Stellacci, *Plasmonics*, 2010, **5**, 85–97.
- 59 D. Mulugeta, K. H. Kim, K. Watanabe, D. Menzel and H.-J. Freund, *Phys. Rev. Lett.*, 2008, **101**, 146103.
- 60 K. Watanabe, D. Menzel, N. Nilius and H.-J. Freund, *Chem. Rev.*, 2006, **106**, 4301–4320.
- 61 K. Narasimharao, A. Al-Shehri and S. Al-Thabaiti, *Appl. Catal., A*, 2015, **505**, 431–440.
- 62 A. Biabani-Ravandi, M. Rezaei and Z. Fattah, *Chem. Eng. J.*, 2013, **219**, 124–130.

