





Cite this: *RSC Adv.*, 2018, 8, 2708

Synthesis of manganese phosphate hybrid nanoflowers by collagen-templated biomineralization†

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Construction of protein–inorganic hybrid materials with hierarchical nanostructures is critical for the creation of advanced multi-functional materials. We herein for the first time report the synthesis of protein–manganese phosphate hybrid nanomaterials by environmentally amiable biomineralization approach. We have demonstrated that collagen provides an excellent biotemplate to modulate the morphology of the hybrid materials, leading to exquisite nanoflowers with branched petals. In this time-dependent biomineralization process, collagen played an essential role in the production of protein–manganese phosphate hybrid materials by inducing the nucleation of manganese phosphates to form a scaffold as well as serving as a glue to hold the petals together. The as-prepared CL–Mn₃(PO₄)₂ nanoflowers exhibited good catalytic activity towards water oxidation. The unique (Gly–X–Y)_n amino acid sequences and triple helix structure may provide extraordinary capability for collagen to create hybrid nanomaterials *via* collagen-templated biomineralization. The single-size and high purity may endow recombinant collagen as a powerful strategy to establish superior biotemplates. This facile and green approach to produce collagen–manganese phosphate hybrid nanoflowers greatly advances our capability to construct manganese phosphates-based functional materials.

Received 20th November 2017
 Accepted 5th January 2018

DOI: 10.1039/c7ra12628j

rsc.li/rsc-advances

1. Introduction

Construction of protein–inorganic hybrid materials with hierarchical nanostructures plays a fundamental role in the development of improved functional materials.^{1–3} The mutual interaction and complementarity of biological molecules and inorganic materials synergetically reinforces the functionality of these hybrid materials.⁴ It remains a major challenge to create hybrid materials with exquisitely mediated hierarchical structures. Biomineralization is a classical process widely used by nature to produce protein–inorganic materials with exceptional functionality and delicate morphology, including bones and teeth as typical examples.^{5–8} Proteins serve as excellent bio-templates to control the nucleation and growth of inorganic materials in living organisms. Inspired by nature, protein-based biomineralization has been extensively explored to generate a variety of inorganic nanomaterials such as calcium carbonate and calcium phosphate.^{9–15}

Transition metal phosphates have been widely investigated due to their superior redox properties with promising applications in heterogeneous catalysis. They have been explored as cathode electrode materials in lithium batteries considering their low cost, environmentally friendly nature and large theoretical specific capacity.¹⁶ Owing to the relatively high specific area, transition metal phosphates have been applied as adsorbents for radionuclide materials and heavy metals.¹⁸ Numerous phosphate materials with outstanding catalytic features have been generated till date, including zirconium, titanium, vanadium, nickel, iron, chromium, zinc, yttrium, tantalum and niobium phosphates.^{17,19–26}

Manganese phosphates are present as switzerite in nature, and have attracted special attention due to their outstanding electrochemical properties.^{27–30} In nature, the water oxidation catalyst of photosynthesis was composed of earth-abundant elements such as Mn and Ca in the form of cubical CaMn₄O₅ clusters, which functioned under neutral conditions with very low overpotential value.³¹ The unique capability of natural system has inspired the design of manganese phosphate as water oxidation catalyst with superior catalytic performance at neutral pH.³² Manganese phosphates have also been applied as oxidation catalysts for methyl mandelate and adonitol, and they played a major role in corrosion protection in mild steels.^{30,33} Various template-free and organically-templated approaches have been developed to create a number of manganese

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† Electronic supplementary information (ESI) available: Fig. S1 and S2. See DOI: 10.1039/c7ra12628j



phosphates.^{34,35} For example, piperazine and ethylenediamine have been established as organic templates to produce manganese phosphates using hydrothermal methods.^{34,35}

Collagen is the most abundant protein in human body, and it is the major component of bone, skin and tendon.³⁶ Unlike globular proteins, collagen possesses a distinct triple helical conformation and repetitive (Gly-X-Y)_n amino acid sequences.^{37,38} The rod-like structure of collagen endows it as an excellent building block for nanodevices, and its superior biological features such as multiple interactions with cell receptors and minimal immunogenicity lead to broad biomedical applications.^{39,40} Compared with animal-derived collagen, recombinant collagen can be generated of uniform size, high purity, superior solubility and zero risk of infectious diseases.⁴¹ Therefore, recombinant collagen has obtained increasing attention in the construction of various biomaterials.

Collagen has been utilized by human body as the main biological template to produce hydroxyapatite with well-controlled hierarchical structures, offering the collagen-hydroxyapatite hybrid materials unique structural, mechanical and biochemical properties.⁴² Protein-inorganic nanoflowers have drawn huge interests due to their remarkable features such as large surface areas, high porosity, and active surface functional groups with enhanced functionality.^{4,43-45} We herein for the first time report the synthesis of protein-manganese phosphate hybrid nanomaterials. We have demonstrated that collagen provides an excellent biotemplate to modulate the morphology of the hybrid materials, leading to well-defined nanoflowers with branched petals.

2. Experimental section

Preparation of recombinant collagen CL

Protein V-CL was expressed in *E. coli* BL21 strain as previously reported.^{37,46} Briefly, 50 mL overnight culture was transferred into 1000 mL LB medium containing ampicillin (50 µg mL⁻¹), and was grown at 37 °C until optical density (OD) at 600 nm reaching 0.8. Protein expression was induced by 1 mM isopropyl beta-D-thiogalactopyranoside (IPTG) at 25 °C. Cells were collected after overnight incubation and re-suspended in 50 mL binding buffer (20 mM NaH₂PO₄, 500 mM NaCl, 50 mM imidazole, pH 7.4). Cells were disrupted by sonication, and cell debris was removed. The collected supernatants were loaded into a Ni-NTA-Sepharose column, and were eluted using the binding buffer supplemented with 500 mM imidazole. The collected V-CL protein were immediately dialyzed against 50 mM glycine buffer, pH 8.6. The protein purity was confirmed by the presence of a single band on SDS-PAGE. The V domain was removed from the purified V-CL by the treatment with trypsin at 28 °C for 24 h. The reaction was stopped by adding 1 mM phenylmethanesulfonyl fluoride (PMSF). After dialysis against 50 mM Tris buffer (pH 7.4), recombinant collagen CL was collected, and its purity was confirmed by SDS-PAGE. The purified CL was then lyophilized and stored at 4 °C for future use.

Synthesis of collagen-manganese phosphate hybrid nanoflowers

Different concentrations of CL protein were prepared in PBS buffer (20 mM, pH 7.4). 1 mL of CL solution was added to 1 mL MnCl₂ solution (3.6 mM). The mixture was stirred for at least 10 min and then incubated at 25 °C for 24 h. The precipitates were collected by centrifugation, and washed by ethanol 3 times. The products were air dried at 25 °C. To study the effect of collagen on the morphology, the concentration of CL was varied from 0 to 0.2 wt%, while [Mn(II)] was set as 1.8 mM.

Characterization of collagen-manganese phosphate hybrid nanoflowers

SEM measurements were performed on a Hitachi S-4800 field emission scanning electron microscope (Hitachi limited, Japan) with an operating voltage of 5.0 kV. X-ray diffraction (XRD) patterns were recorded on a Rigaku D/ma-2500 diffractometer with Cu Kα radiation (40 kV, 200 mA) at a scanning rate of 0.02° s⁻¹ in the 2θ range from 20 to 80°. X-ray photoelectron spectroscopy (XPS) measurements were carried out on a Kratos Axis Ultra^{DLD} X-ray photoelectron spectrometer (England) with a monochrome X-ray source using Al Kα (1486.6 eV) radiation. The binding energies measured by XPS were corrected by referencing the C 1s line to 284.5 eV. TEM and electron diffraction (EDX) measurements were recorded using a JEM-2100 transmission electron microscope (JEOL, Japan) at 200 kV. Fourier-transform infrared (FT-IR) spectra were obtained using a Nicolet NEXUS 670 infrared spectrometer. Thermogravimetric analysis (TGA) was performed on a TGA/NETZSCH STA449F3 instrument under a nitrogen atmosphere, employing a heating rate of 10 °C min⁻¹ from 25 °C to 500 °C.

Electrochemical measurements

The catalytic activity of CL-Mn₃(PO₄)₂ hybrid nanomaterials was investigated by cyclic voltammetry (CV). 5 mg of synthesized hybrid nanomaterials were dispersed in 1 mL water with 100 µL neutralized Nafion solution. The mixture was then sonicated for at least 20 min to make a homogeneous solution. 50 µL mixture was dropped onto the FTO substrate, which was then dried in 80 °C oven overnight as working electrode. Before recording electrochemical data, the working electrode was cycled at least 3 times (potential: from 0.7 to 1.5 V; scan rate: 10 mV s⁻¹) for getting the smooth curves. The CV measurements were performed under a three-electrode electrochemical cell system at ambient temperature. A BASi Ag/AgCl/3 M NaCl reference electrode and a Pt foil were used as a reference electrode and a counter electrode, respectively. The electrolyte was phosphate buffer with 500 mM buffer strength under pH 7.0.

3. Results and discussion

Synthesis and characterization of collagen-manganese phosphate hybrid nanoflowers

Collagen protein CL (MW: 22.8 kDa) was prepared using the high expression cold-shock vector system as previously described, and its purity was confirmed by SDS-PAGE.³⁷ The



collagen–manganese phosphate hybrid nanomaterials CL-Mn₃(PO₄)₂ were prepared by incubating the mixture of 1.8 mM aqueous solution of Mn(NO₃)₂ and 0.2 wt% of CL protein in 20 mM PBS buffer at 25 °C for 24 h. X-ray diffraction (XRD) and X-ray photoelectron spectroscopy (XPS) techniques were used to characterize the as-prepared products (Fig. 1). The XRD pattern of the synthesized nanomaterials fitted well with that of hydrated manganese phosphate Mn₃(PO₄)₂·3H₂O (JCPDS no. 3-426) (Fig. 1a). The XPS spectra of the nanomaterials displayed the composition of Mn, P, O, N and C elements, confirming the presence of collagen and Mn₃(PO₄)₂ (Fig. 1b).

FT-IR spectra of the as-prepared hybrid nanomaterials were also acquired (Fig. S1†). The observed vibration peaks corresponding to the N–H, C–H, C=O and NH₂ bonds further confirmed the presence of collagen within the synthesized collagen–manganese phosphate hybrid nanomaterials CL-Mn₃(PO₄)₂. The content of collagen CL in the hybrid nanomaterials was further evaluated by thermogravimetric analysis (TGA) (Fig. S2†). TGA plots showed two weight-loss stages, which corresponded to the loss of absorbed water (below 200 °C) and the thermal decomposition of collagen (200–500 °C), respectively. All the results indicated that a significant amount of collagen was embedded within the hybrid nanomaterials.

The morphology of the CL-Mn₃(PO₄)₂ hybrid nanomaterials was examined by field-emission scanning electron microscopy (FESEM) and transmission electron microscopy (TEM) techniques (Fig. 2). The SEM images showed flower-shaped nanostructures consisting of well-organized petals (Fig. 2a). A closer

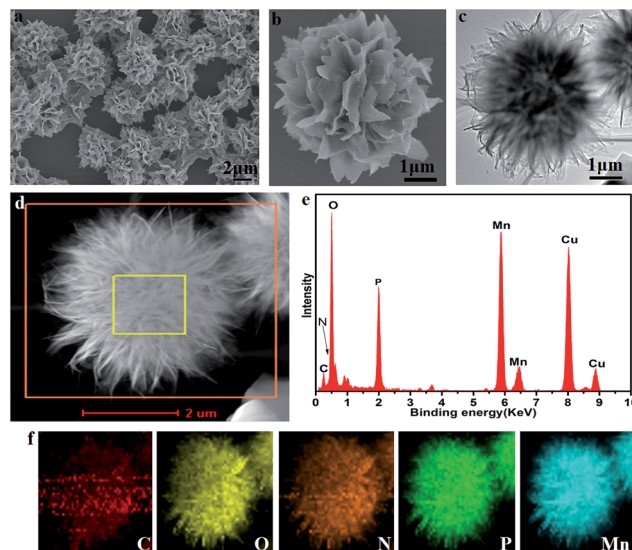


Fig. 2 FESEM images of the obtained CL-Mn₃(PO₄)₂ nanomaterials (a and b); TEM image (c) and HAADF-STEM image (d) of the same hybrid nanomaterials; EDX spectrum of the synthesized nanomaterials (e); EDS mapping images of C, O, N, P, and Mn for the square region of the particles in (d and f).

observation of a single nanoflower showed clear hierarchical arrangements of the petals (Fig. 2b). TEM images indicated that the CL-Mn₃(PO₄)₂ hybrid nanoflowers were self-assembled through an oriented growth (Fig. 2c). The high-angle annular dark-field scanning TEM (HAADF-STEM) image suggested that the primary manganese phosphate nanoparticles gradually aggregate to form nanoflowers with hierarchical structures (Fig. 2d). Energy dispersive X-ray (EDX) analysis further confirmed the presence of C, N, O, P and Mn elements in the hybrid nanomaterials and the corresponding elemental mapping results by energy dispersive X-ray spectroscopy (EDS) showed that all the elements were distributed homogeneously on the particles, demonstrating the encapsulation of collagen in the synthesized nanoflowers (Fig. 2e and f). All the results demonstrated that the CL-Mn₃(PO₄)₂ hybrid particles formed well-defined flower-like supramolecular structures. This is the first report of protein-templated biomineralization that has been successfully utilized to generate Mn₃(PO₄)₂ nanomaterials.

The critical role of collagen in hybrid nanoflowers

In order to evaluate the critical role of collagen in the production of hybrid nanoflowers, the morphologies of CL-Mn₃(PO₄)₂ nanomaterials were characterized under a range of collagen concentrations from 0 to 0.2 wt% while keeping a constant Mn(II) concentration (1.8 mM) (Fig. 3). In the absence of any collagen, manganese phosphate aggregates did not display well-ordered structure (Fig. 3a). When the collagen concentration was increased, petals became evident, and they tended to get together to form a flower bouquet. The addition of 0.2 wt% of collagen resulted in well-defined branched nanoflowers (Fig. 3b–d). The clear dependence of the morphologies of CL-Mn₃(PO₄)₂ on the concentration of CL indicated that collagen

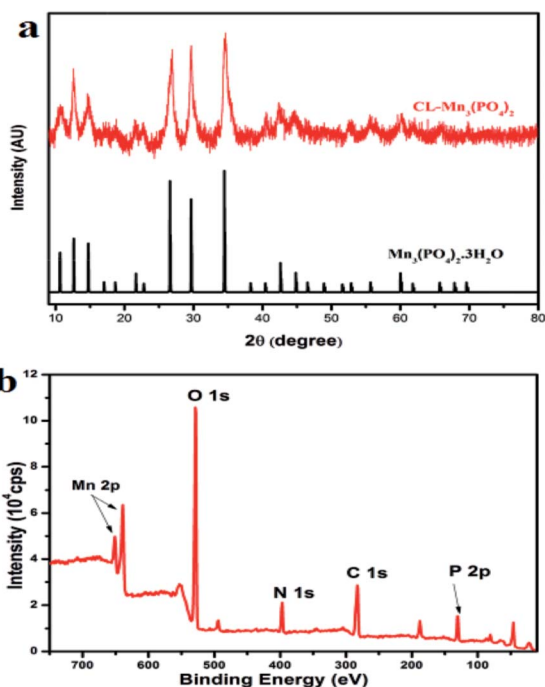


Fig. 1 XRD and XPS characterization of CL-Mn₃(PO₄)₂ hybrid nanomaterials. (a) XRD patterns of CL-Mn₃(PO₄)₂ hybrid nanomaterials (red) and a standard manganese phosphate hydrate Mn₃(PO₄)₂·3H₂O (JCPDS no. 3-426) (black). (b) XPS spectra of the CL-Mn₃(PO₄)₂ nanomaterials.



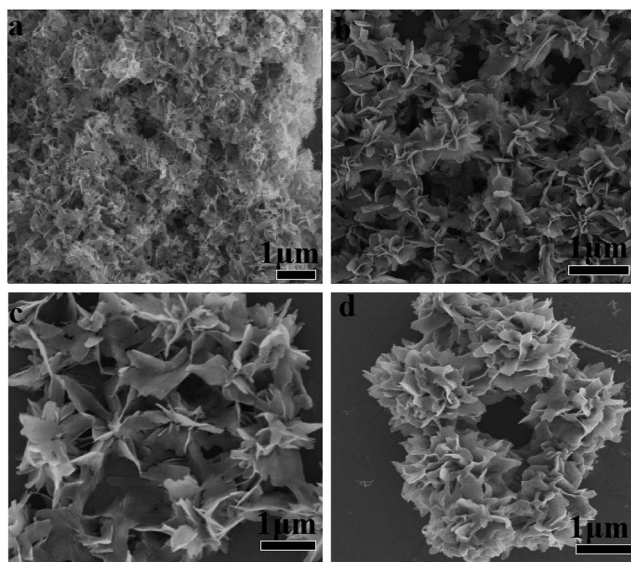


Fig. 3 SEM images of the CL-Mn₃(PO₄)₂ hybrid nanomaterials obtained after 24 h incubation at 25 °C using a constant concentration of MnCl₂ (1.8 mM) and various concentrations of collagen: 0 wt% (a), 0.01 wt% (b), 0.05 wt% (c) and 0.2 wt% (d).

could conveniently mediate the nanostructures of hybrid materials.

The morphologies of CL-Mn₃(PO₄)₂ hybrid nanoflowers were further examined after treatments with calcination and glutaraldehyde/EDTA (Fig. 4). Calcination of the synthesized hybrid nanomaterials at 350 °C for 6 h led to the loss of flower-like structures and scattered petals, indicating that collagen played an essential role to maintain the supramolecular structure of (Fig. 4a). When the synthesized nanoflowers were first cross-linked employing glutaraldehyde, and then treated with ethylenediaminetetraacetic acid (EDTA) to remove the Mn²⁺ ions, they also lost the well-ordered flower-shape, indicating the Mn²⁺ ions were also the determinant elements for the formation of nanoflowers (Fig. 4b). These results suggested that both collagen and Mn²⁺ ions were uniformly distributed in the hybrid nanomaterials, and their co-participation finely modulate the flower-like morphology. Collagen likely triggered the nucleation of the manganese phosphate crystals to form a framework for the petals and played as a glue to hold the petals together.

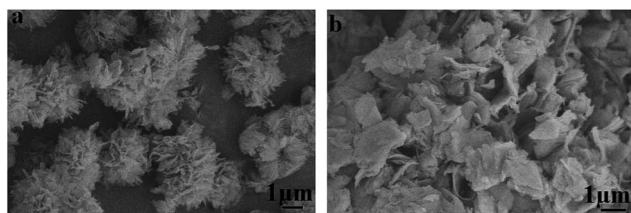


Fig. 4 SEM images of the CL-Mn₃(PO₄)₂ hybrid nanoflowers treated by calcination (a), and glutaraldehyde/EDTA (b). Calcination was performed at 350 °C for 6 h. The CL-Mn₃(PO₄)₂ hybrid nanoflowers were first treated with glutaraldehyde (0.8 wt%) and then EDTA (1 wt%).

Time-dependent growth of collagen-manganese phosphate hybrid nanoflowers

SEM was used to monitor the time dependent evolution process of CL-Mn₃(PO₄)₂ nanoflowers (Fig. 5). Nanoparticles with small petals started to appear after 1 h, and the petals became better organized after 6 h (Fig. 5a-d). When the incubation time extended to 24 h, the hybrid materials formed well-defined delicate flower-shaped nanostructures with multilayered petals (Fig. 5e and f). These results indicated that it was a time-dependent gradual growing process for the hybrid materials to form multi-layered nanoflowers.

The collagen-manganese phosphate hybrid nanoflowers as water oxidation catalysts

The catalytic capacity of water oxidation of the synthesized CL-Mn₃(PO₄)₂ hybrid nanoflowers was investigated by cyclic voltammetry (CV) in 0.5 M sodium phosphate electrolyte at pH 7.0 (Fig. 6). The current values were normalized to the catalyst' effective surface area on the working electrode. The current density of CL-Mn₃(PO₄)₂ normalized to the surface area was 16 mA cm⁻² at the overpotential of 680 mV. The Tafel slope obtained from the polarization-corrected CV curves was

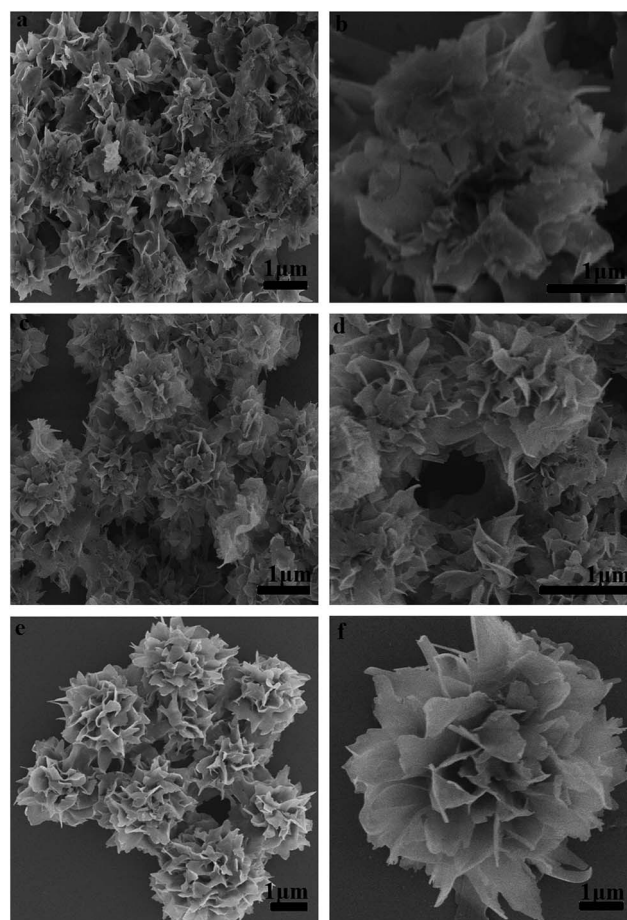


Fig. 5 SEM images of CL-Mn₃(PO₄)₂ hybrid nanomaterials obtained by different incubation time: 1 h (a, b), 6 h (c, d), and 24 h (e, f). The mixture of CL (0.2 wt%) and MnCl₂ (1.8 mM) were incubated at 25 °C.



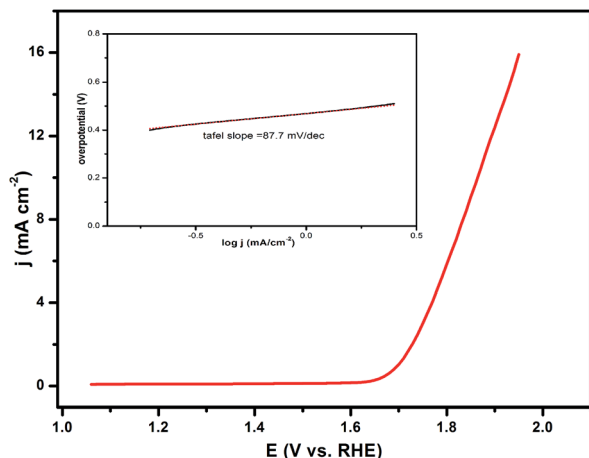


Fig. 6 Electrochemical characterization of CL-Mn₃(PO₄)₂ nanospheres. Polarization-corrected cyclic voltammogram (CV) curves and Tafel plots for CL-Mn₃(PO₄)₂ spin-coated on the FTO substrate. The current value was normalized to the surface area of the catalyst.

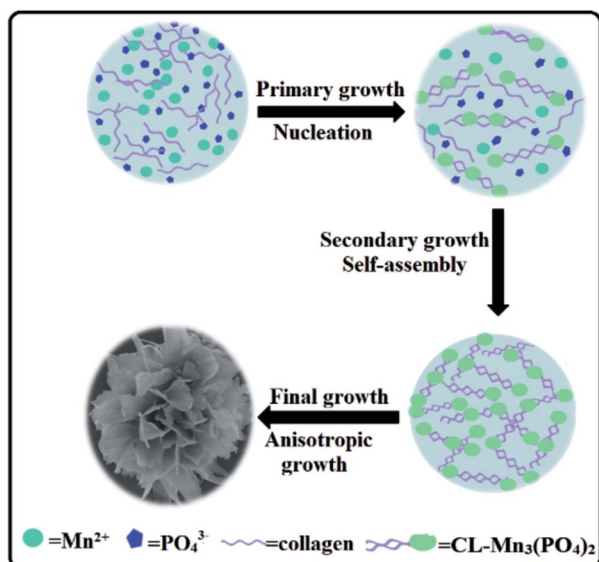


Fig. 7 Schematic illustration of the formation mechanism of CL-Mn₃(PO₄)₂ nanospheres.

87.8 mV dec⁻¹. All the results indicated that CL-Mn₃(PO₄)₂ nanospheres presented a good catalyst for water oxidation, which may be due to the synergistic effect between manganese phosphate and collagen.

4. Conclusions

Construction of protein-inorganic hybrid materials with hierarchical nanostructures is critical for the creation of advanced multi-functional materials.² Protein-inorganic nanoflowers have gained growing attention due to their superior features including large surface areas and active surface functional groups. Collagen-templated biomineralization has been selected by nature to generate hierarchical protein-inorganic

hybrid nanomaterials for the construction of bone and teeth for human being. We herein for the first time have demonstrated that recombinant collagen provides an excellent biotemplate to generate protein-manganese phosphate hybrid materials with exquisite flower-like nanostructures.

Recombinant collagen contains a large number of charged amino acids, which could promote the interaction of manganese ions and collagen. At the initial primary growth stage, Mn(III) ions first form a complex with collagen by interaction with charged amino acids distributed along the protein chain, resulting in the formation of primary crystals (Fig. 7). In the secondary growth step, further self-assembly of the manganese phosphate primary crystals leads to the formation of nanoflowers with small petals. Collagen probably covers the surface, and stabilizes the primary nanoparticles. At the final anisotropic growth stage, the nanoparticles continued to assemble into the final well-ordered multi-layered nanoflowers with branched petals. In this time-dependent biomineralization process, collagen played an essential role in the production of protein-manganese phosphate hybrid materials by inducing the nucleation of manganese phosphates to form a scaffold as well as serving as a glue to hold the petals together.

The unique (Gly-X-Y)_n amino acid sequences and triple helix structure may provide extraordinary capability for collagen to create hybrid nanomaterials *via* collagen-templated biomineralization. The single-size and high purity may endow recombinant collagen as a powerful strategy to establish superior biotemplates. The application of these newly constructed CL-Mn₃(PO₄)₂ nanomaterials as water oxidation catalysts has demonstrated their excellent functional features. This facile and environmentally amiable biomineralization approach to produce collagen-manganese phosphate hybrid nanoflowers greatly advances our capability to construct manganese phosphates-based functional materials.

Conflicts of interest

The authors declare no competing conflict of interests.

Acknowledgements

This work was supported by grants from the National Natural Science Foundation of China (Grant no. 21775059) and the Fundamental Research Funds for the Central Universities (Grant no. lzujbky-2016-k10).

References

- 1 S. Lagziel-Simis, N. Cohen-Hadar, H. Moscovich-Dagan, Y. Wine and A. Freeman, *Curr. Opin. Biotechnol.*, 2006, **17**, 569–573.
- 2 M. B. Dickerson, K. H. Sandhage and R. R. Naik, *Chem. Rev.*, 2008, **108**, 4935–4978.
- 3 X. Wang, H. C. Schroder and W. E. Muller, *Trends Biotechnol.*, 2014, **32**, 441–447.
- 4 J. Ge, J. Lei and R. N. Zare, *Nat. Nanotechnol.*, 2012, **7**, 428–432.



- 5 L. Addadi, S. Raz and S. Weiner, *Adv. Mater.*, 2003, **15**, 959–970.
- 6 A.-W. Xu, Y. Ma and H. Cölfen, *J. Mater. Chem.*, 2007, **17**, 415–449.
- 7 S. Weiner and L. Addadi, *J. Mater. Chem.*, 1997, **7**, 689–702.
- 8 A. Arakaki, K. Shimizu, M. Oda, T. Sakamoto, T. Nishimura and T. Kato, *Org. Biomol. Chem.*, 2015, **13**, 974–989.
- 9 T. Wang, D. Porter and Z. Shao, *Adv. Funct. Mater.*, 2012, **22**, 435–441.
- 10 Y. Cai and J. Yao, *Nanoscale*, 2010, **2**, 1842.
- 11 R. Kurapati and A. M. Raichur, *J. Mater. Chem. B*, 2013, **1**, 3175.
- 12 X. Q. Li, Z. Feng, Y. Xia and H. C. Zeng, *Chemistry*, 2012, **18**, 1945–1952.
- 13 C. Cheng, Z. Shao and F. Vollrath, *Adv. Funct. Mater.*, 2008, **18**, 2172–2179.
- 14 Z. Ding, H. Han, Z. Fan, H. Lu, Y. Sang, Y. Yao, Q. Cheng, Q. Lu and D. L. Kaplan, *ACS Appl. Mater. Interfaces*, 2017, **9**, 16913–16921.
- 15 T. Kato, A. Sugawara and N. Hosoda, *Adv. Mater.*, 2002, **14**, 869.
- 16 A. K. Padhi, K. S. Nanjundaswamy, C. Masquelier, S. Okada and J. B. Goodenough, *J. Electrochem. Soc.*, 1997, **144**, 1609.
- 17 X. Tian, W. He, J. Cui, X. Zhang, W. Zhou, S. Yan, X. Sun, X. Han, S. Han and Y. Yue, *J. Colloid Interface Sci.*, 2010, **343**, 344–349.
- 18 A. Dutta, A. K. Patra and A. Bhaumik, *Microporous Mesoporous Mater.*, 2012, **155**, 208–214.
- 19 J. Jiménez-Jiménez, P. Maireles-Torres, P. Olivera-Pastor, E. Rodríguez-Castellón, A. Jiménez-López, D. J. Jones and J. Rozière, *Adv. Mater.*, 1998, **10**, 812–815.
- 20 A. Bhaumik and S. Inagaki, *J. Am. Chem. Soc.*, 2001, **123**, 691–696.
- 21 J. Yu, A. Wang, J. Tan, X. Li, J. A. van Bokhoven and Y. Hu, *J. Mater. Chem.*, 2008, **18**, 3601.
- 22 U. R. Pillai and E. Sahle-Demessie, *Chem. Commun.*, 2004, 826–827, DOI: 10.1039/b313747c.
- 23 A. Tarafdar, S. Biswas, N. K. Pramanik and P. Pramanik, *Microporous Mesoporous Mater.*, 2006, **89**, 204–208.
- 24 W. He, S. Yan, Y. Wang, X. Zhang, W. Zhou, X. Tian, X. Sun and X. Han, *J. Alloys Compd.*, 2009, **477**, 657–660.
- 25 Q. Luo, S. Shen, G. Lu, X. Xiao, D. Mao and Y. Wang, *J. Mater. Chem.*, 2009, **19**, 8079.
- 26 Y. Zhang, J. Wang, J. Ren, X. Liu, X. Li, Y. Xia, G. Lu and Y. Wang, *Catal. Sci. Technol.*, 2012, **2**, 2485.
- 27 S. Fernández, J. L. Pizarro, J. L. Mesa, L. Lezama, M. I. Arriortua, R. Olazcuaga and T. Rojo, *Inorg. Chem.*, 2001, **40**, 3476–3483.
- 28 K. Jin, J. Park, J. Lee, K. D. Yang, G. K. Pradhan, U. Sim, D. Jeong, H. L. Jang, S. Park, D. Kim, N.-E. Sung, S. H. Kim, S. Han and K. T. Nam, *J. Am. Chem. Soc.*, 2014, **136**, 7435–7443.
- 29 S. Ferdov, A. M. L. Lopes, Z. Lin and R. A. S. Ferreira, *Chem. Mater.*, 2007, **19**, 6025–6029.
- 30 F. Ben Brahim and H. Boughzala, *J. Mol. Struct.*, 2013, **1034**, 336–345.
- 31 J. Barber, *Chem. Soc. Rev.*, 2009, **38**, 185–196.
- 32 X. Shen, Q. Wang, Y. Liu, W. Xue, L. Ma, S. Feng, M. Wan, F. Wang and C. Mao, *Sci. Rep.*, 2016, **6**, 28989.
- 33 A. G. Fadnis and S. K. Kulshrestha, *React. Kinet. Catal. Lett.*, 1982, **19**, 267–269.
- 34 K. O. Kongshaug, H. Fjellvåg and K. P. Lillerud, *J. Solid State Chem.*, 2001, **156**, 32–36.
- 35 J. Escobal, J. L. Pizarro, J. L. Mesa, L. Lezama, R. Olazcuaga, M. I. Arriortua and T. Rojo, *Chem. Mater.*, 2000, **12**, 376–382.
- 36 M. D. Shoulders and R. T. Raines, *Annu. Rev. Biochem.*, 2009, **78**, 929–958.
- 37 A. Yoshizumi, Z. Yu, T. Silva, G. Thiagarajan, J. A. Ramshaw, M. Inouye and B. Brodsky, *Protein Sci.*, 2009, **18**, 1241–1251.
- 38 Y. Xu, D. R. Keene, J. M. Bujnicki, M. Hook and S. Lukomski, *J. Biol. Chem.*, 2002, **277**, 27312–27318.
- 39 Y. Li and S. M. Yu, *Curr. Opin. Chem. Biol.*, 2013, **17**, 968–975.
- 40 M. He, L. Wang, J. Wu and J. Xiao, *Chem.–Eur. J.*, 2016, **22**, 1914–1917.
- 41 Y. Y. Peng, L. Howell, V. Stoichevska, J. A. Werkmeister, G. J. Dumsday and J. A. M. Ramshaw, *Microb. Cell Fact.*, 2012, **11**, 146.
- 42 E. A. Zimmermann and R. O. Ritchie, *Adv. Healthcare Mater.*, 2015, **4**, 1287–1304.
- 43 M. Li, M. Luo, F. Li, W. Wang, K. Liu, Q. Liu, Y. Wang, Z. Lu and D. Wang, *J. Phys. Chem. C*, 2016, **120**, 17348–17356.
- 44 P. Koley, M. Sakurai and M. Aono, *ACS Appl. Mater. Interfaces*, 2016, **8**, 2380–2392.
- 45 B. Zhang, P. Li, H. Zhang, L. Fan, H. Wang, X. Li, L. Tian, N. Ali, Z. Ali and Q. Zhang, *RSC Adv.*, 2016, **6**, 46702–46710.
- 46 B. An, V. Abbonante, S. Yigit, A. Balduini, D. L. Kaplan and B. Brodsky, *J. Biol. Chem.*, 2013, **289**, 4941–4951.

