



Cite this: RSC Adv., 2018, 8, 1477

Received 22nd October 2017
Accepted 24th November 2017

DOI: 10.1039/c7ra11654c

rsc.li/rsc-advances

1. Introduction

Graphene, an atomically thin film made of honeycomb carbon atoms, has received considerable attention due to its fabulous electrical, optical and mechanical performance.^{1–5} The 2D planar structure of graphene makes it compatible with traditional electronic device architectures and it has been proposed for applications in frequency transistors, transparent conducting electrodes, and sensors for optical, gas, and strain detection.^{6–10} To date, metal-catalyzed chemical vapor deposition (CVD) by decomposition of methane has been commonly utilized to fabricate large-scale and high-quality graphene.^{11–15} For industrial applications, large-area and high-quality graphene is demanded at low cost. However, CVD growth of graphene is usually performed at a high temperature (*i.e.*, 1050 °C),^{13–15} which requires precise and expensive equipment and makes the direct deposition of graphene in electronic device manufacturing processes infeasible due to the severe physical damage to substrates (*e.g.*, metals, semiconductors, and the junctions between them). In order to address this issue, low temperature synthesis of graphene urgently needs to be achieved. Plasma enhanced CVD (PECVD) can effectively reduce the growth temperature, but the etching and bombarding effects of high energy hydrogen ions ionized by the plasma may introduce a high density of defects into graphene.^{16,17} Besides, PECVD is not suitable for industrial applications due to its high

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Decreasing graphene synthesis temperature by catalytic metal engineering and thermal processing

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Chemical vapor deposition (CVD) from gaseous hydrocarbon sources has shown great promise for large-scale graphene growth, but the high growth temperature, typically 1050 °C, requires precise and expensive equipment and makes the direct deposition of graphene in electronic device manufacturing processes unfeasible due to the severe physical damage to substrates. Here we demonstrate a facile route to synthesize graphene by catalytic metal engineering and thermal processing. The engineered catalytic metal (copper) with carbon implantation could lower the synthetic temperature to 700 °C. And the resulting graphene shows few defects, uniform morphology and high carrier mobility, comparable to CVD graphene grown at 1050 °C. This technique could expand the applications of graphene in electronic and optoelectronic device manufacturing and is compatible with conventional microelectronics technology.

cost. Using less stable solid or liquid feedstock instead of methane is another approach for decreasing the CVD temperature of graphene growth. However, the flow of the pneumatolytic solid or liquid feedstock is difficult to control and most alternative feedstock such as benzene is hypertoxic. Therefore, low-temperature synthesis of graphene still remains a challenge.

In this work, we systematically analyze a facile approach to grow graphene by catalytic metal engineering prior thermal processing at a low temperature down to 700 °C. The catalytic metal (copper) was engineered by carbon implantation to form Cu/C alloys and the implanted carbon atoms acted as the feedstock for graphene synthesis. The as-deposited graphene shows few defects, uniform morphology and high carrier mobility, comparable to CVD graphene grown at 1050 °C. The resulting graphene by catalytic metal engineering and thermal processing could expand the applications of graphene in electronic and optoelectronic device manufacturing and was compatible with conventional microelectronics technology.

2. Experimental section

2.1. Low-temperature synthesis of graphene

As shown in Fig. 1, the catalytic metal (Cu) was first engineered by C implanting (30 keV) and the atomic ratio of Cu and C in the

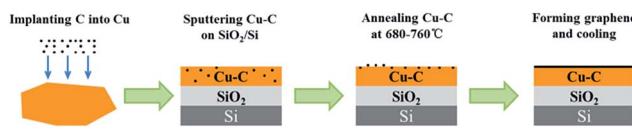


Fig. 1 The flow chart of low temperature graphene synthesis by catalytic metal engineering and thermal processing.

obtained Cu/C alloy was 99.06 : 0.04. Then Cu/C alloy was deposited on a SiO₂/Si substrate by sputtering prior annealing. The annealing process was maintained at the designed temperatures (680–760 °C) for 10 min followed by naturally cooling in an annealing furnace with H₂ shielding. The low content of C in the alloy was due to low solubility of C in Cu. In the alloy, C was the feedstock of graphene and Cu acted as the catalyst. Therefore, no external feedstock was needed for graphene synthesis. The thickness of Cu–C film was 1.14 μm. The design of annealing temperatures and Cu–C film thickness will be discussed in the theoretical calculation part.

2.2. Graphene characterization

The obtained graphene films were characterized by Raman spectroscopy, atomic force microscope (AFM) and high resolution transmission electron microscopy (HRTEM), respectively. The Raman measurements were carried out using a Raman spectroscopy of HORIBA Jobin Yvon HR800. All samples were excited by Ar⁺ laser with wavelength of 514 nm with the spot size of 1 μm in diameter. The obtained graphene film was transferred from the Cu/SiO₂/Si substrates onto glass slides by a PMMA-assisted wet-transfer method in order to directly evaluate the thickness of the graphene film. The SiO₂/Si substrates were etched by 10% potassium hydroxide solution while Cu was etched by 30% iron trichloride solution. Then the cross-section microstructure of graphene on the glass was characterized by the HRTEM (JEOL-2100F). Graphene field effect transistors (GFETs) were fabricated by lithography and the device structure was detected by the scanning electron microscope (SEM) using a JEOL 7800F with an accelerating voltage of 5 kV beam. The GFETs were measured at room temperature under ambient conditions by an analytical probe station.

3. Results and discussion

3.1. Theoretical calculation

Cu–C alloys were selected as the sputtering target in which the atomic ratio of Cu and C was 99.06 : 0.04. The thickness of deposited Cu–C alloy could be calculated by the following equation

$$\eta\rho\tau = \gamma \quad (1)$$

where η is the copper concentration and carbon solubility (99.96/0.04), ρ is the copper atomic density (8.36×10^{22} atoms per cm³), τ is the Cu–C film thickness, and γ is the amount of carbon atoms in graphene (3.8×10^{15} atoms per cm²). Therefore, the thickness of Cu–C film could be calculated as 1.14 μm. The annealing temperature could be estimated by the following Arrhenius equation,

$$D = D_0 \exp\left(-\frac{\Delta G_m}{kT}\right) \quad (2)$$

where D_0 is the diffusion constant of C in Cu, ΔG_m is the activation energy (0.990 eV) for the migration of the interstitial atoms, k is the Boltzmann constant, and T is the temperature. D_0 can be extracted from the following equation,

$$D_0 = \frac{1}{6} \alpha^2 z v \exp\left(\frac{\Delta S_m}{R}\right) \quad (3)$$

where α is the lattice constant (0.377 nm for Cu), z is the number of sites available (0.6), v is the vibration frequency (10^{15} Hz), ΔS_m is the entropy of migration (2.554×10^4 eV K⁻¹), and R is the gas constant (5.189×10^{19} eV K⁻¹). As a result, D_0 was estimated to be $0.14 \text{ cm}^2 \text{ s}^{-1}$. The thickness (τ) of Cu–C film has been calculated as 1.14 μm and the distance (τ') of carbon atoms need to travel is typically an order of magnitude greater than τ , which can be also estimated by the following equation,

$$\tau' \approx \sqrt{Dt} \quad (4)$$

where t is the diffusion time. Assuming the diffusion time is 1 s and combining eqn (2) and (4), the annealing temperature could be estimated as 720 °C.

3.2. Synthetic temperature verification

The theoretical estimation of the synthetic temperature was confirmed by experiments. Fig. 2a shows the Raman spectra of graphene by annealing Cu/C alloys at different temperatures (680–760 °C). The main features in the Raman spectrum of graphene are the G and 2D peaks, lying at around 1580 cm⁻¹ and 2700 cm⁻¹, respectively. The G peak corresponds to optical E_{2g} phonons at the Brillouin zone center (T), whereas the 2D peak originates from a process where momentum conservation is satisfied by two phonons with opposite wave vectors.^{18–21} Therefore, no defects are required for the activation of 2D peak, and it thus always presents. The D peak lying at around 1300 cm⁻¹ is due to the breathing modes of six-member rings and requires a defect for its activation.^{21–23} Defects, which include structural disorder such as grain boundaries or vacancies, edges, and functionalized groups, have the effect of increasing D peak intensity (I_D), decreasing 2D band intensity (I_{2D}) and broadening the 2D full width at half-maximum (FWHM). As shown in Fig. 2a and b, the intensity ratio of D peak and G peak (I_D/I_G ratio) rises

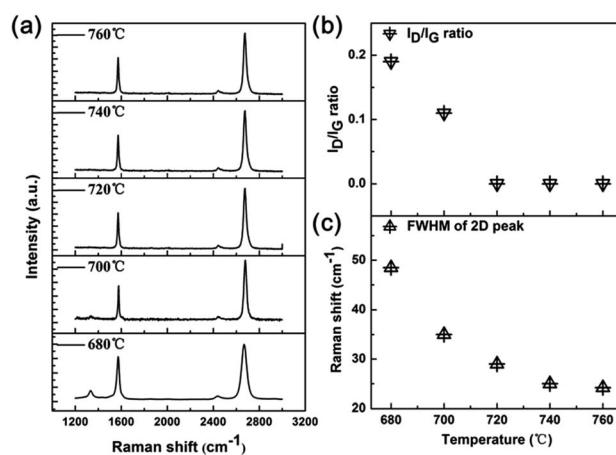


Fig. 2 (a) Raman spectra of graphene by annealing Cu/C alloys at 680–720 °C. (b) The I_D/I_G ratio of graphene by annealing Cu/C alloys at 680–720 °C. (c) The FWHM of 2D peak of graphene by annealing Cu/C alloys at 680–720 °C.

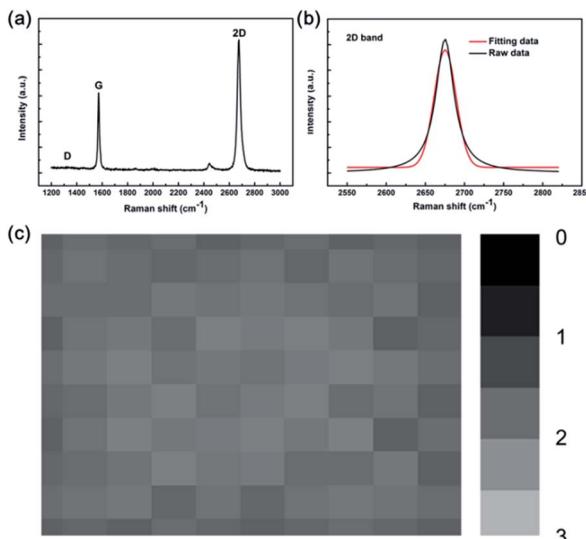


Fig. 3 (a) Raman spectrum of graphene by catalytic metal (Cu) engineering and thermal processing at 700 °C. (b) 2D band can be split into single Lorentzian curve (red curve). (c) Two-dimensional Raman mapping of the graphene film. The gradient bar to the right of each map represents the I_{2D}/I_G band ratio.

with annealing temperature increases. When the annealing temperature increases to 700 °C, the D peak is not distinct, and when the annealing temperature increases to 720 °C, the D peak is undetectable, indicating the synthesized graphene films is in high quality comparable to that of CVD graphene grown at 1050 °C.^{11–15} In addition, the 2D band is symmetric and its full width at half maximum (FWHM) is down to $\sim 35\text{ cm}^{-1}$ when the temperature is 700 °C, which is in consistent with monolayer graphene.²¹ The FWHM of 2D peak can be further decreased by rising the annealing temperature (Fig. 2c).

Fig. 3a depicts the representative Raman spectrum of the as-deposited graphene by annealing Cu/C alloys at 700 °C. The defect-related D band is negligible, implying that the graphene film has few defects or disorder. Furthermore, as shown in Fig. 3b, the symmetric 2D band with a FWHM of 35 cm^{-1} can be well fitted by a single Lorentzian curve providing evidence of monolayer graphene.²¹ To determine the quality, uniformity, and thickness of the graphene films deposited on a large-scale substrate, Raman mapping of the 2D to G peak intensity ratio over a $10\text{ }\mu\text{m} \times 10\text{ }\mu\text{m}$ area with a spot size of $1\text{ }\mu\text{m}$ and a step size of $1\text{ }\mu\text{m}$ is performed, as shown in Fig. 3c. The I_{2D}/I_G ratio is quite uniform over the region investigated and the I_{2D}/I_G ratio is in the range of 1.5–2, indicating complete monolayer graphene coverage in the scanned area. Those measurements give evidence of the high crystal quality in graphene samples and indicate the graphene resultant temperature can be down to 700 °C by catalytic metal (Cu) engineering and thermal processing.

3.3. Characterization analysis

The surface morphology of transferred graphene on a SiO_2/Si substrate has been investigated by AFM (Fig. 4a). The surface root-mean-square (RMS) roughness is down to 0.4 nm , manifesting the high uniformity and flatness of graphene deposited by catalytic

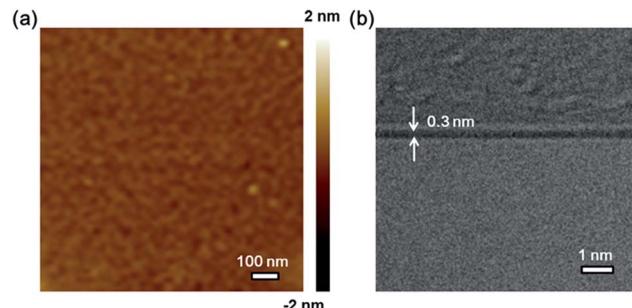


Fig. 4 (a) The AFM image of transferred graphene on a SiO_2/Si substrate. (b) The HRTEM image of graphene shows its monolayer property.

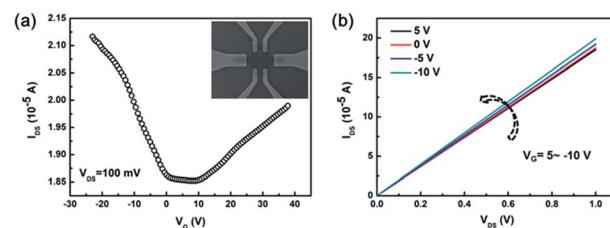


Fig. 5 (a) The $I_{DS}-V_G$ curve of the graphene transistor at $V_{DS} = 100\text{ mV}$. The inset shows the SEM image of a back-gated GFETs. (b) The $I_{DS}-V_{DS}$ curves of the graphene transistor with different V_G .

metal (Cu) engineering and thermal processing. The as-deposited graphene film has also been transferred onto TEM grids by a polymethyl methacrylate (PMMA)-assisted wet-transfer method in order to directly evaluate the thickness of the graphene film as shown in Fig. 4b. The cross-section microstructure of graphene characterized by HRTEM shows its thickness is 0.3 nm , further demonstrating the as-deposited graphene is monolayer.

To determine the transport properties of the synthesized graphene film, the back-gated GFET has been fabricated on the SiO_2 (300 nm)/Si substrates. The SEM image of GFET is shown in the inset of Fig. 5a. The highly reproducible transfer characteristics ($I_{DS}-V_G$) of the GFET measured at room temperature under ambient conditions is shown in Fig. 5a. The typical $I_{DS}-V_G$ curve measured at a V_{DS} of 100 mV shows that the gate can cause either hole or electron conduction. The V-shaped ambipolar transfer characteristic is typical of monolayer graphene with a zero bandgap.^{24–33} The Dirac point of the GFET shifts slightly to a positive gate at $V_G = 5\text{ V}$, demonstrating slight p-type hole doping performance. The slight p-type doping of graphene is due to the charge transfer doping by the adsorption of atmospheric hole dopants (e.g. O_2) in air.^{34–37} In addition, the residue of PMMA may also lead to p-type doping of graphene.^{37,38} According to the two slopes of the linear regions on both sides of the V-shaped curve, the hole mobility (μ_h) is $\sim 1200\text{ cm}^2\text{ V}^{-1}\text{ s}^{-1}$ and the electron mobility (μ_e) is $\sim 1100\text{ cm}^2\text{ V}^{-1}\text{ s}^{-1}$, both of which are comparable to values reported recently from transferred CVD graphene.^{24–33} The presence of defects, wrinkles, and overlaps generated from the transfer process may degrade the performance of GFETs, thus underestimating the carrier mobility of the synthesized graphene film. In the output characteristics of the GFET, as shown in Fig. 5b,

the linear I_{DS} – V_{DS} behavior indicates a good ohmic contact between the Au/Ti contact and graphene channels. In addition, I_{DS} increases with V_G negative shift from 5 to -10 V and it is indicative of p-type behavior as well. The electrical transport data also reveal that the graphene deposited by catalytic metal (Cu) engineering and thermal processing is of good quality which can be further improved by refining the transfer process.

4. Conclusions

In summary, catalytic metal engineering by implanting C into Cu and thermal processing were established to deposit graphene at a low temperature down to 700 °C. Parametric studies show that the superior quality and homogeneous monolayer graphene in large scale can be directly achieved on the Cu/C substrates. Raman, AFM and HRTEM are utilized to determine both the quality and thickness of the graphene film. GFETs were designed and characterized to measure the electrical properties of the synthesized graphene film. The obtained graphene by catalytic metal engineering and thermal processing is scalable and compatible with the mainstream microelectronics technology, thus paving the way to the application of graphene in microelectronic field at a low temperature.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work was supported by the Shanghai Sailing Program (Grant No. 17YF1422700) and the National Scientific Foundation of Shanghai (Grant No. 16ZR1442300). We would like to thank Prof. Xiaoming Xie, Prof. Ya-Hong Xie, Prof. Zengfeng Di, A. Prof. Yan Cheng, Dr Tianbo Wang, Dr Zirui Liu, Dr Jimmy Ng, Dr Owen Liang, Dr Ming Xia, Dr Xinke Yu and Dr Zhongbo Yan for their generous help.

References

- 1 A. K. Geim and K. S. Novoselov, *Nat. Mater.*, 2007, **6**, 183.
- 2 K. S. Novoselov, A. K. Geim, S. V. Morozov, D. Jiang, Y. Zhang, S. V. Dubonos, I. V. Grigorieva and A. A. Firsov, *Science*, 2004, **306**, 666.
- 3 K. S. Novoselov, A. K. Geim, S. V. Morozov, D. Jiang, M. I. Katsnelson, I. V. Grigorieva, S. V. Dubonos and A. A. Firsov, *Nature*, 2005, **438**, 197.
- 4 M. S. Fuhrer, *Science*, 2013, **340**, 1413.
- 5 J. H. Chen, C. Jang, S. Xiao, M. Ishigami and M. S. Fuhrer, *Nat. Nanotechnol.*, 2008, **3**, 206.
- 6 Y.-M. Lin, C. Dimitrakopoulos, K. A. Jenkins, D. B. Farmer, H.-Y. Chiu, A. Grilli, *et al.*, *Science*, 2010, **323**, 662.
- 7 X. Li, Y. Zhu, W. Cai, M. Borysiak, B. Han, D. Chen, *et al.*, *Nano Lett.*, 2009, **9**, 4359.
- 8 B. Y. Zhang, T. Liu, B. Meng, X. Li, G. Liang, X. Hu, *et al.*, *Nat. Commun.*, 1811, 2013, **4**.
- 9 S. Basu and P. Bhattacharyya, *Sens. Actuators, B*, 2012, **173**, 1.
- 10 S.-H. Pan, H. Medina, S.-B. Wang, L.-J. Chou, Z. M. Wang, K.-H. Chen, *et al.*, *Nanoscale*, 2014, **6**, 8635.
- 11 Y. Wu, Y. Hao, H. Y. Jeong, Z. Lee, S. Chen, W. Jiang, *et al.*, *Adv. Mater.*, 2013, **25**, 6744.
- 12 Q. Li, H. Chou, J. H. Zhong, J. Y. Liu, A. Dolocan, J. Zhang, *et al.*, *Nano Lett.*, 2013, **13**, 486.
- 13 X. Li, W. Cai, L. Colombo and R. S. Ruoff, *Nano Lett.*, 2009, **9**, 4268.
- 14 S. Bhaviripudi, X. Jia, M. S. Dresselhaus and J. Kong, *Nano Lett.*, 2010, **10**, 4128.
- 15 Y. Wu, Y. Hao, H. Y. Jeong, Z. Lee, S. Chen, W. Jiang, *et al.*, *Adv. Mater.*, 2013, **25**, 6744.
- 16 R. John, A. Ashokreddy, C. Vijayan and T. Pradeep, *Nanotechnology*, 2011, **22**, 165701.
- 17 L. Xie, L. Jiao and H. Dai, *J. Am. Chem. Soc.*, 2010, **132**, 14751.
- 18 L. Zheng, X. Cheng, D. Cao, G. Wang, Z. Wang, D. Xu, *et al.*, *ACS Appl. Mater. Interfaces*, 2014, **6**, 7014.
- 19 L. Zheng, X. Cheng, D. Cao, Z. Wang, C. Xia, Y. Yu, *et al.*, *Appl. Phys. Lett.*, 2014, **104**, 023112.
- 20 D. C. Elias, R. R. Nair, T. M. G. Mohiuddin, S. V. Morozov, P. Blake, M. P. Halsall, *et al.*, *Science*, 2009, **323**, 610.
- 21 A. C. Ferrari and D. M. Basko, *Nat. Nanotechnol.*, 2013, **8**, 235.
- 22 A. C. Ferrari and J. Robertson, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2000, **61**, 14095.
- 23 K. K. Mani and R. Ramani, *Phys. Status Solidi B*, 1974, **61**, 659.
- 24 V. Singh, D. Joung, L. Zhai, S. Das, S. I. Khondaker and S. Seal, *Prog. Mater. Sci.*, 2011, **56**, 1178.
- 25 S. Bhaviripudi, X. Jia, M. S. Dresselhaus and J. Kong, *Nano Lett.*, 2010, **10**, 4128.
- 26 R. John, A. Ashokreddy, C. Vijayan and T. Pradeep, *Nanotechnology*, 2011, **22**, 165701.
- 27 M. Zhu, Z. Du, Z. Yin, W. Zhou, Z. Liu, S. H. Tsang, *et al.*, *ACS Appl. Mater. Interfaces*, 2016, **8**, 502.
- 28 R. Munoz and C. Gomez-Aleixandre, *Chem. Vap. Deposition*, 2013, **19**, 297.
- 29 I. A. Kostogrud, K. V. Trusov and D. V. Smovzh, *Adv. Mater. Interfaces*, 2016, **3**, 1500823.
- 30 S. J. Chae, F. Güneş, K. K. Kim, E. S. Kim, G. H. Han, S. M. Kim, *et al.*, *Adv. Mater.*, 2009, **21**, 2328.
- 31 V. K. Sangwan, D. Jariwala, S. A. Filippone, H. J. Karmel, J. E. Johns, J. M. Alaboson, *et al.*, *Nano Lett.*, 2013, **13**, 1162.
- 32 S. K. Lee, H. Y. Jang, S. Jang, E. Choi, B. H. Hong, J. Lee, *et al.*, *Nano Lett.*, 2012, **12**, 3472.
- 33 L. Wang, X. Chen, Y. Wang, Z. Wu, W. Li, Y. Han, *et al.*, *Nanoscale*, 2013, **5**, 1116.
- 34 S. Ryu, L. Liu, S. Berciaud, Y. J. Yu, H. Liu, P. Kim, *et al.*, *Nano Lett.*, 2010, **10**, 4944.
- 35 L. Zheng, X. Cheng, Z. Wang, C. Xia, D. Cao, L. Shen, *et al.*, *J. Phys. Chem. C*, 2015, **119**, 5995.
- 36 D. W. Boukhvalov and M. I. Katsnelson, *J. Phys. Chem. C*, 2009, **113**, 14716.
- 37 U. N. Maiti, W. J. Lee, J. M. Lee, Y. Oh, J. Y. Kim, J. E. Kim, *et al.*, *Adv. Mater.*, 2014, **26**, 40.
- 38 W. J. Yu, L. Liao, S. H. Chae, Y. H. Lee and X. Duan, *Nano Lett.*, 2011, **11**, 4759.

