


Cite this: *RSC Adv.*, 2018, 8, 4314

Received 1st November 2017
Accepted 22nd December 2017

DOI: 10.1039/c7ra11260b

rsc.li/rsc-advances

Structural phase transition and bonding properties of high-pressure polymeric CaN_3 [†]

Pugeng Hou,^{*a} Lili Lian,^b Yongmao Cai,^a Bao Liu,^a Bo Wang,^a Shuli Wei^c and Da Li^{ID}^{*c}

Alkaline-earth metal polynitrides can be used as a type of starting materials in the synthesis of polymeric nitrogen, which is used as a potential high-energy-density material. The structural evolutionary behaviors of nitrogen in CaN_3 were studied at the pressure up to 100 GPa using a particle-swarm optimization structure search combining with density functional theory. Two new stable phases with $\bar{P}1$ and $C2/m$ space groups at the pressures of 26 and 60 GPa were identified for the first time. Throughout the stable pressure range, these two structures are semiconductors and consist of the N atoms in sp^2 -hybrid states. To the best of our knowledge, this is the first time an N_6 chain is reported in case of alkaline-earth metal polynitrides. The stable polynitrogen compounds and polymeric nitrogen as high-energy-density materials have potential applications. The present results open a new possible avenue to synthesize high-energy-density polynitrogen.

1 Introduction

To discover and synthesize environmentally harmless high energy-density materials (HEDMs) is becoming increasingly important at present. Among the myriad HEDMs, the polynitrogen compounds are of significant interest due to their particularly high energy and high enthalpy of formation among other features.^{1–9} Nitrogen exists in form of N_2 molecules in nitrogen gas with $\text{N}\equiv\text{N}$ bonds. It is significantly interesting if the abundant N_2 molecules can be transformed into polynitrogen with $\text{N}-\text{N}$ bonds and $\text{N}=\text{N}$ bonds. The $\text{N}-\text{N}$ and $\text{N}=\text{N}$ or $\text{N}\equiv\text{N}$ bonds have large energy difference between them, which can be used in industrial applications. Although numerous theoretical studies predicting that some high-pressure nitrogen phases might be stable,^{7,10} it was not until 2004 that the cubic gauche phase of nitrogen (cg-N) was experimentally synthesized using high pressure and high temperature treatment (110 GPa, 2000 K).¹¹ It has been predicted that the single-bond phase of solid nitrogen has more than three times higher energy storage capacity than that of the most powerful energetic materials.¹²

Recently, metal azides, as a type of precursors, have been proposed to be used in the formation of polymeric nitrogen. One feasible approach to obtain such polymeric nitrogen phases is to form nitrogen-rich alloys with other elements.

Such “chemical precompression” may reduce the pressure of polymerization significantly. Theoretical and experimental studies are inspired by elusive polymeric nitrogen and considerable progress has been achieved in the exploration of stable nitrogen-rich compounds. Recently, metal azides can be used as the starting materials to synthesize numerous types of potential high-energy-density materials because the synthesis pressure of metal azides is potentially lower than that of pure nitrogen gas. Over the years, various metal azides have been proposed in numerous experimental and theoretical studies, for example, LiN_3 ,^{13–16} LiN_5 ,¹³ NaN_3 ,^{17–21} KN_3 ,^{20,22–27} CsN_3 ,^{23,28} and AlN_3 .²⁹ Moreover, N_2H ,^{30,31} C–N system,^{32,33} S–N system,³⁴ and the P–N system³⁵ are also studied as high-energy materials. These successes precipitate an increasing interest in searching for other forms of stable polynitrogen in metal polynitrogen compounds in the recent years.

The chemistry of calcium is relatively underdeveloped compared to that of its analogues (Be, Mg, Sr, and Ba). Recently, we theoretically explored the phase diagram of the Mg–N system at the pressures ranging from 0 to 100 GPa³⁶ and discovered a series of new compounds in this family, including Mg_2N_3 , MgN , MgN_3 , MgN_4 , and MgN_5 . The predicted compounds contain a rich variety of polynitrogen forms ranging from small molecules (N_2 , N_3 , N_4 , and N_6) to the extended nitrogen chains. Alkaline-earth metal calcium and alkali metal potassium are adjacent in the periodic elements table. In addition, calcium can exhibits a similar chemical activity as compared to that of alkali metal element potassium. Thus, we can expect that mixing reactive calcium element with nitrogen will realize more diverse structures that obtain higher energy densities of the products. Therefore, a study of calcium polynitride would help

^aCollege of Science, Northeast Electric Power University, No. 169 Changchun Road, Jilin City, 132012, P. R. China. E-mail: houpugeng@126.com

^bThe First Hospital of Jilin University, Changchun, 130021, P. R. China

^cState Key Laboratory of Superhard Materials, College of Physics, Jilin University, Changchun, 130012, P. R. China. E-mail: dali@jlu.edu.cn

[†] Electronic supplementary information (ESI) available. See DOI: 10.1039/c7ra11260b



to theoretically investigate more promising high-energy materials among alkaline-earth metal polynitrides. High pressure-treatment has emerged as a powerful tool to investigate the physical and chemical behaviours of materials and it can be a crucial thermodynamic parameter. This study has led to the synthesis of new and unexpected products. A new orthorhombic phase was predicted as the thermodynamic ground state of CaN_2 , which has considerably lower energy than that of the synthesized phase as well as the theoretical models considered earlier.³⁷ Zhang *et al.* identified a tetragonal Ca_2N structure under high pressures, giving an excellent agreement between the theoretical and experimental XRD data.³⁸ Recently, Zhu *et al.* theoretically explored the phase diagram of the Ca–N system, including Ca_2N_3 , CaN , CaN_3 , CaN_4 , and CaN_5 .³⁹ The predicted CaN_x compounds contained a rich variety of polynitrogen forms ranging from small molecules (N_2 , N_4 , N_5 , and N_6) to the extended chains (N_∞). The CaN_x crystals with polynitrogens were expected to be highly exothermic, making them as the potential high-energy-density materials. The newly predicted CaN_3 phase was calculated to become thermodynamically stable under low pressure ($P = 8$ GPa). At 8 GPa, CaN_3 has an orthorhombic *Pmma* structure.

In the present study, the possible formation and the stability of the polymeric nitrogen compounds were investigated in the binary CaN_3 under high pressure. The structure searches for the stable CaN_3 compound were performed using an unbiased structure prediction method based on particle-swarm optimization algorithm in conjunction with density-functional calculations. Then, the electronic properties and chemical bonding of the identified Ca–N compounds were investigated systematically in order to study the properties and the potential applications of the binary CaN_3 compound as high energy-density materials. The lattice dynamics, structural stability, electronic structure, and bonding nature were studied, providing more insights into the mechanism of the pressure-induced physical properties. The understanding of metal–nitrogen interactions and the bonding nature in these polynitrogens is of great importance to the research on nitrogen chemistry.

2 Computation details

The crystal structure prediction for the CaN_3 system *via* global minimization of free energy surfaces was performed through CALYPSO code, which has demonstrated recent success in predicting the high-pressure structures of numerous substances from the elemental to binary and ternary compounds and validated with various known compounds.^{40–42} The lowest-energy structures for CaN_3 were determined at 0, 20, 50, 80, and 100 GPa with the systems containing one to six formula units (f.u.) in the simulation cell. The underlying *ab initio* structural relaxations and electronic band structural calculations for CaN_3 were performed in the framework of the density functional theory (DFT) within generalized gradient approximation Perdew–Burke–Ernzerhof (GGA-PBE)⁴³ as implemented in the VASP (Vienna *ab initio* simulation package) code.⁴⁴ The projector-augmented wave (PAW)⁴⁵ pseudopotentials are used from the VASP potential library. The $3s^23p^64s^2$ for

Ca and $2s^22p^3$ for N were treated as the valence states. The computational parameters were tested such that the energy calculations were well converged to better than 1 meV per atom. A plane-wave energy cutoff of 620 eV and appropriate Monkhorst–Pack⁴⁶ *k*-meshes (*k*-points grid of 0.03 \AA^{-1}) were used in the Brillouin zone. We calculated the net charge based on Bader analysis.^{47,48} The supercell method with PHONOPY code was used to calculate the phonon frequencies for all structures.⁴⁹ In addition, we used the DFT + D2 approach,^{50–52} which is a simple and efficient way to approximately account for the long-range van der Waals interactions. In this method, the sum of the induced dipole–dipole interactions (E_{disp}) was added to the self-consistent total energy calculated within Kohn–Sham DFT ($E_{\text{KS-DFT}}$), $E_{\text{DFT+D2}} = E_{\text{KS-DFT}} + E_{\text{disp}}$. We choose a cutoff radius of 30 Å for the van der Waals interactions and the global scaling parameter s_6 was set to 0.75.

3 Results and discussions

3.1 Phase stability

The most favorable Ca–N compounds obtained in the structure search at various pressures are presented in Fig. 1a.³⁹ For the energetically most favorable CaN_3 structures, the calculated formation enthalpies ΔH_f in the pressure range from 0 to 100 GPa are shown in Fig. 1b. The ΔH_f of each CaN_3 structure was calculated by $\Delta H_f(\text{CaN}_3) = H(\text{CaN}_3) - H(\text{Ca}) - 3H(\text{N})$ at $T = 0$ K. The known solid calcium (α , β , and γ phase) and solid nitrogen (α -, *Pbcn*-, *P2/c*-, and *cg*-phases)⁵³ were used as the reference structures. These structures are energetically most favorable in their corresponding stable pressure ranges. In order to determine the phase transition pressure for the predicted CaN_3 , we have plotted out the enthalpy curves relative to the ambient pressure $P1\text{-CaN}_3$. The enthalpies of the most energetically competitive structures are compared over the pressure range from 0 GPa to 100 GPa as shown in Fig. 1. The most stable structure has a monoclinic phase with $P\bar{1}$ symmetry from 26 GPa up to 40 GPa, which is then replaced by a lower-enthalpy *C2/c* structure that has been proposed by Zhu *et al.*³⁹ With further compression above 60 GPa, another monoclinic phase, *C2/m*- CaN_3 , becomes the most thermodynamically stable phase and remains the lowest-enthalpy phase up to 100 GPa. Since the two low-pressure $P\bar{1}$ - CaN_3 and high-pressure *C2/m*- CaN_3 phases are novel for alkali-earth metal polynitrides and metal azides, the investigation about the properties of the two high-pressure phases in detail is essential. The ambient-pressure decomposition of $P\bar{1}$ - CaN_3 and *C2/m*- CaN_3 to CaN_2 and N_2 are estimated to release 1.04 eV and 1.34 eV energy per formula unit (f.u.), which corresponds to an energy density of approximately 1.14 kJ g^{-1} and 1.47 kJ g^{-1} , respectively. Such high energy content can establish that CaN_3 is indeed a high energy density material. It can replace modern high explosives, such as TATB, RDX, and HMX, which typically have energy densities ranging from 1 to 3 kJ g^{-1} .⁵⁴ Thus, we may find their applications as a high energy materials. This study, therefore, represents an exciting prediction and can hopefully encourage the experimental efforts in their synthesis.



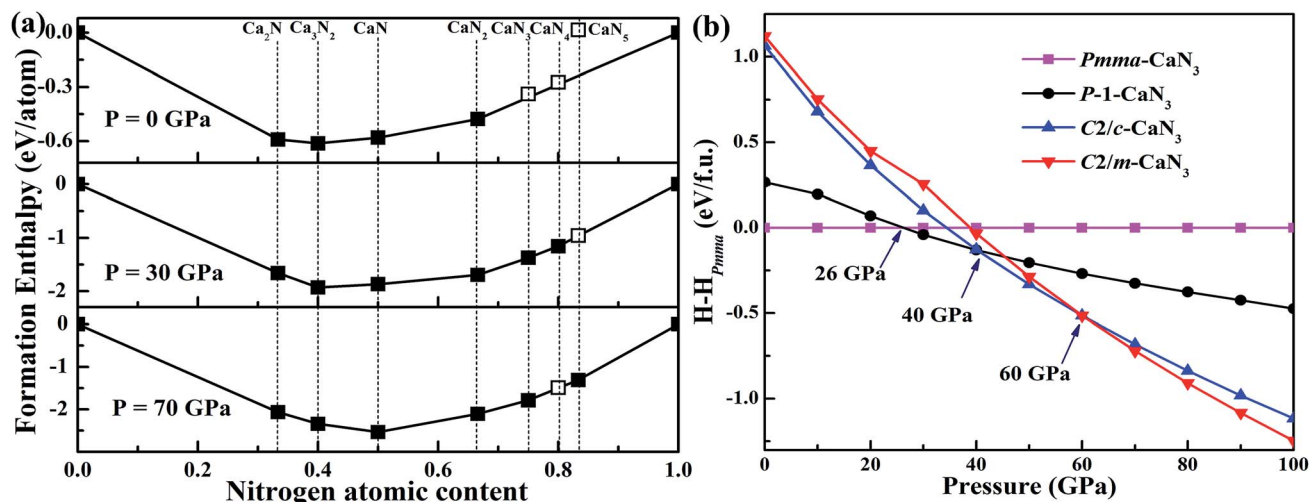


Fig. 1 (a) Relative enthalpies of the formation of Ca–N phases with respect to solid calcium and nitrogen. Reference to the results of Zhu *et al.*³⁷ (b) Relative enthalpies of the CaN_3 relative to the Pmma-CaN_3 calculated by the DFT + D2 approach as a function of pressure.

The calculated relations of phonon dispersion of low-pressure $\text{P}\bar{1}\text{-CaN}_3$ and high-pressure C2/m-CaN_3 also confirmed their mechanical and dynamical stability as shown in Fig. 2. No imaginary phonon mode can be observed, which

proves clearly their lattice dynamical stability. Under some kinetic regimes at ambient conditions, the calcium polynitride CaN_3 also has the possibility of stabilizing. Advantageously, these two structures $\text{P}\bar{1}\text{-CaN}_3$ and C2/m-CaN_3 are mechanically stable at ambient conditions and they may possibly quench under an ambient pressure.

Another important indicator of the existence of these structures is their mechanical stability. According to the Born–Huang criterion,⁵⁵ it was found that $\text{P}\bar{1}\text{-CaN}_3$ and C2/m-CaN_3 were mechanically stable at an ambient pressure and 60 GPa. In order to be used in potential technological and industrial applications, the understanding of the mechanical properties (elastic constant and anisotropy) of the predicted CaN_3 structures is important. The elastic constants (C_{ij}), bulk modulus (B), shear modulus (G), Young's modulus (Y), and Poisson's ratio (ν) of the $\text{P}\bar{1}\text{-CaN}_3$ and C2/m-CaN_3 structures are summarized in Table 1. The elastic constants for both $\text{P}\bar{1}\text{-CaN}_3$ and C2/m-CaN_3 were calculated using the strain–stress method. For the monoclinic crystals, if they are mechanically stable, the elastic constants need to satisfy the following mechanical criteria:

$$C_{11} > 0, C_{22} > 0, C_{33} > 0, C_{44} > 0, C_{55} > 0, C_{66} > 0$$

$$[C_{11} + C_{22} + C_{33} + 2(C_{12} + C_{13} + C_{23})] > 0$$

$$(C_{33}C_{55} - C_{35}^2) > 0, (C_{44}C_{66} - C_{46}^2) > 0, (C_{22} + C_{33} - 2C_{23}) > 0$$

$$[C_{22}(C_{33}C_{55} - C_{35}^2) + 2C_{23}C_{25}C_{35} - C_{23}^2C_{55} - C_{23}^2C_{33}] > 0.$$

As illustrated in Table 1, we can observe that the elastic constants for the monoclinic $\text{P}\bar{1}\text{-CaN}_3$ and C2/m-CaN_3 structures satisfy the above conditions, which implies that the two phases, $\text{P}\bar{1}\text{-CaN}_3$ and C2/m-CaN_3 , are elastically stable at ambient pressure and 60 GPa, respectively. Hence, the calculations indicate that $\text{P}\bar{1}\text{-CaN}_3$ and C2/m-CaN_3 are elastically stable. It is valuable to be mentioned that the bulk modulus for the C2/m-CaN_3 phase, which was calculated from the elastic

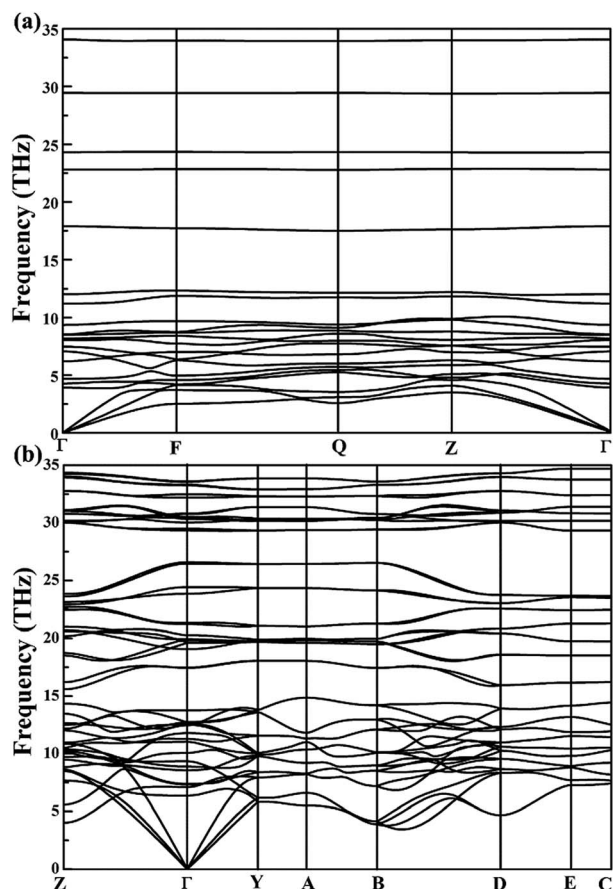


Fig. 2 Phonon dispersion curves for (a) $\text{P}\bar{1}\text{-CaN}_3$ at ambient pressure and (b) C2/m-CaN_3 at 60 GPa.



Table 1 Calculated elastic constants C_{ij} 's (GPa), bulk modulus (B_0), shear modulus (G), Young's modulus (Y), and Poisson's ratio (ν) for the $P\bar{1}$ - CaN_3 and $C2/m$ - CaN_3 structures under $P = 0$ GPa and 60 GPa

	C_{11}	C_{22}	C_{33}	C_{44}	C_{55}	C_{66}	C_{12}	C_{13}	C_{23}	C_{35}	C_{46}	B_0	G	Y	ν
$P\bar{1}$ - CaN_3	216	208	122	64	50	86	77	46	82	−1.4	−5.5	101	59	149	0.25
$C2/m$ - CaN_3	546	546	596	246	193	190	235	194	233	−0.3	18.0	333	191	481	0.26

constants, is 333 GPa (Table 1). Hence, the $C2/m$ - CaN_3 phase is also a high-incompressibility material.

3.2 Structural features

CaN_3 is predicted to take the monoclinic $P\bar{1}$ space group at 0 GPa as shown in Fig. 3a. The lattice constants of the predicted $P\bar{1}$ - CaN_3 are $a = 4.744$ Å, $b = 5.307$ Å, $c = 3.182$ Å, $\alpha = 76.70^\circ$, $\beta = 97.94^\circ$, and $\gamma = 80.80^\circ$ at $P = 30$ GPa. Four inequivalent atoms occupy the crystallographic structure: Ca at $2i$ (0.696, 0.198, 0.077), N1 at $2i$ (0.307, 0.321, 0.399), N2 at $2i$ (0.140, 0.517, 0.507) and N3 at $2i$ (0.204, 0.136, 0.262) positions. The $C2/m$ - CaN_3 crystal is predicted to become energetically stable near 70 GPa. This phase has the optimized lattice parameters: $a = 8.125$ Å, $b = 5.175$ Å, $c = 3.206$ Å, $\alpha = 90.0^\circ$, $\beta = 67.2^\circ$, and $\gamma = 90.0^\circ$. The atomic positions are as follows: Ca atoms at $4i$ (0.673, 0.00, 0.185) and N1 atom at $8j$ (0.422, 0.275, 0.204), N2 atom at $4i$ (0.085, 0.0, 0.542).

The potential energy storage capabilities for the two predicted new $P\bar{1}$ - CaN_3 and $C2/m$ - CaN_3 have a close relationship with their structure motifs and the patterns of the N atoms. Fig. 3a shows the crystal structure of $P\bar{1}$ - CaN_3 , in which the N atoms adopt the N_6 chain in the (001) plane. The N atoms in this structure are sp^2 -hybridized containing a high content of N–N single bonds in $P\bar{1}$ - CaN_3 . To the best of our knowledge, this is the first time the N_6 chain is reported in case of alkaline-earth metal polynitrides. In addition, the bond lengths of N–N are from 1.323 Å to 1.352 Å, which are greater than that of an $\text{N}=\text{N}$ bond (1.20 Å) and shorter than that of an N–N single bond (1.45 Å). This intermediate distance has a connection with the

electron delocalization, in which the π -electrons are distributed equally among six the N atoms. Each N atom of the N_6 chain is coordinated with two nearest Ca atoms and each Ca atom is six-coordinated with nearby N atoms ($d_{\text{Ca-N}} = 2.51$ Å at 0 GPa). With an increase in the pressures to 60 GPa, a promising monoclinic $C2/m$ phase including partial single N–N bonds was identified for CaN_3 . The high-pressure $C2/m$ - CaN_3 , interestingly, has a puckered layer-like structure with six N atoms forming a nonplanar N_6 ring as shown in Fig. 3b. The hexagonal N_6 ring has been researched in numerous theoretical studies, but they are not satisfactory.^{56,57} By adding the coordinate-covalent bonds from oxygen, the isoelectronic with benzene N_6 ring can be formed.⁵⁸ Moreover, the planar N_6 ring can be remarkably enhanced using the incorporation of a metal atom in these metal- N_6 molecule compounds.⁵⁹ We can obtain the N_6 rings by theoretical simulation calculation in the bulk materials. This shows that additional metal atoms can play a very important role in stabilizing the N_6 ring. More recently, a $C2/m$ - CaN_3 structure containing the N_6 ring has been reported, which has been also predicted in LiN_3 ,^{16,60} NaN_3 ,²¹ MgN_3 ,³⁶ and CaN_3 .³⁹ Moreover, the pressure effect on calcium azide might have a great impact in the rational design of potential polymeric nitrogen.

3.3 Electronic structures

To better understand the nature of the chemical bonding between two atoms and the formation mechanism of low-pressure $P\bar{1}$ - CaN_3 and high-pressure $C2/m$ - CaN_3 , we have calculated their projected density of states (PDOS) at ambient pressure and high pressure $P = 60$ GPa as shown in Fig. 4. We

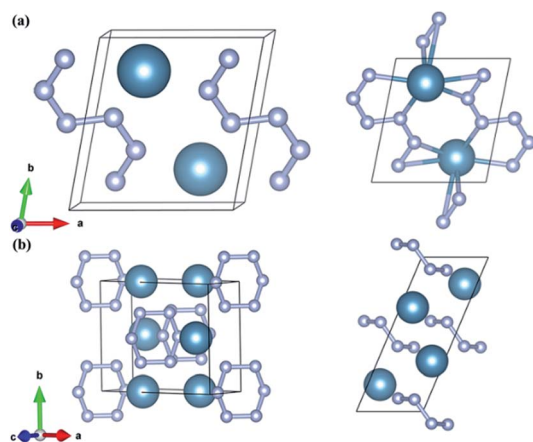


Fig. 3 Crystal structures of the predicted stable (a) $P\bar{1}$ - CaN_3 and (b) $C2/m$ - CaN_3 . The large and small spheres denote calcium and nitrogen atoms, respectively.

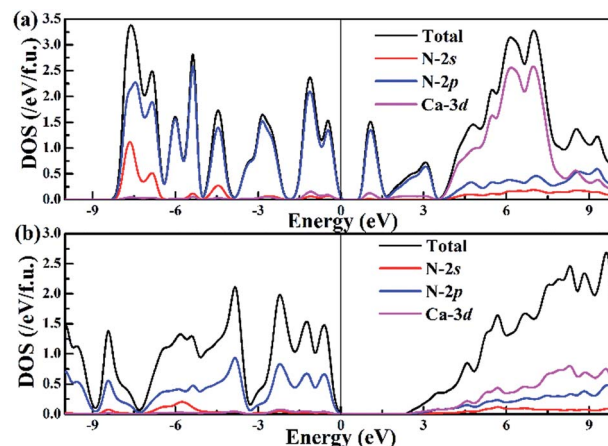


Fig. 4 Band structure and projected density of states of (a) $P\bar{1}$ - CaN_3 at ambient pressure and (b) $C2/m$ - CaN_3 at 60 GPa.



can observe that $P\bar{1}$ - CaN_3 and $C2/m$ - CaN_3 are semiconductors characterized by an indirect band gap of ~ 0.76 eV and ~ 2.38 eV, respectively. The atom-resolved DOS of the low-pressure $P\bar{1}$ - CaN_3 and high-pressure $C2/m$ - CaN_3 structures reveal that the densities of states near Fermi levels are primarily originated from the N orbital electrons. In the “ N_6 ” chain of $P\bar{1}$ - CaN_3 (Fig. 3a), all N atoms are sp^2 -hybridized, forming five N–N σ bonds with neighboring N atoms and six lone pairs. The sp^2 -hybridized orbitals of the four central N atoms in the “ N_6 ” chain form two σ bonds with the sp^2 -hybridized orbital located at both N atoms at each side. The remaining sp^2 -hybridized orbitals at four central N atoms as well as the two sp^2 -hybridized orbitals at both terminal N atoms are filled and form lone pairs. Therefore, all of the bonding states and lone pair states are filled and all of the anti-bonding states are unoccupied in $P\bar{1}$ - CaN_3 , leading to a semiconductive state. The strong covalent bondings between different inequivalent N atoms and the lone pairs on the side N atoms are also revealed by the ELF (Fig. 5a). This 6-membered ring with 10 π -electrons in high-pressure $C2/m$ - CaN_3 follows the aromatic $4n + 2$ Hückel rule and it is similar to the well-known isoelectronic inorganic cyclic S_3N_3^- and P_6^{4-} species.⁶¹ In case of MgN_3 , Yu *et al.* reported that three bonding π -molecular orbitals (MO) as well as two degenerated anti-bonding π^* -molecular orbitals in the N_6 rings were fully occupied over the six π -levels, leading to only one π -bond delocalized over the 6 nitrogen atoms.⁶² Their electronic situation explains well the structural feature encountered in N_6^{4-} , in which the N–N bonds are slightly stronger than the known N–N single bonds. Thus, in $C2/m$ - CaN_3 , an energy gap exists in between π_4^* and π_5^* MO—top of the valence band and the empty π_6^* MO—bottom of the conduction band; hence, it behaves as a semiconductor.

In addition, to study the impact of the partial occupation of N-2p orbital on the electronic properties, we calculated the charge of Ca and N atoms of $P\bar{1}$ - CaN_3 and $C2/m$ - CaN_3 based on Bader analysis. The Bader method is used to analyze the charge transfer by implementing an algorithm developed by Henkelman *et al.*^{47,48} The result reveals that the Ca atoms contribute almost 1.3e to the N atoms forming the charged “ N_6 ” chain, which suggests that $P\bar{1}$ - CaN_3 has ionic characteristics in the chemical bonds. In the predicted $P\bar{1}$ - CaN_3 and $C2/m$ - CaN_3 crystal, the Ca atoms behave as electron donors, whose concentration strongly influences the N–N bonding. This finding contributes to better understanding the chemical bonding behaviors.

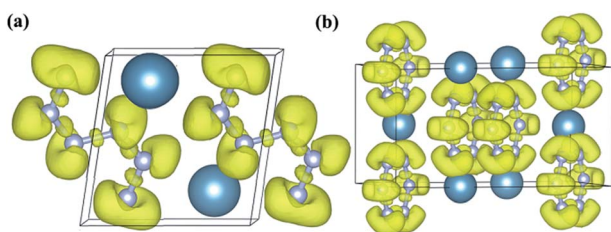


Fig. 5 The ELF distributions of (a) $P\bar{1}$ - CaN_3 at ambient pressure and (b) $C2/m$ - CaN_3 at 60 GPa. The value of isosurface is 0.75.

3.4 Chemical bonding

Subsequently, in order to understand the bonding character of two CaN_3 structures, the electronic localization function (ELF)⁶³ was calculated as shown in Fig. 5. For two CaN_3 structures, the strong covalent bonding between nitrogen atoms and the lone pair electrons is revealed clearly by the ELF (Fig. 5). The large areas with the large ELF values between N and N atoms are shown in Fig. 5, which is typical of a strong covalent bonding. In the “ N_6 ” chain of $P\bar{1}$ - CaN_3 shown in Fig. 1b, all N atoms are sp^2 -hybridized. For intermediate four N atoms, two sp^2 -hybridized orbitals form two N–N σ bonds with neighboring two atoms. The remaining sp^2 -hybridized orbital of this N atom is filled and forms one lone pair. For bicephalous two N atoms, one sp^2 -hybridized orbital forms one σ bond with the sp^2 -hybridized orbital of one neighboring N atom. The remaining two sp^2 -hybridized orbitals of the N atoms are filled and form lone pairs. The covalent bondings between different inequivalent N atoms and the lone pairs on the side N atoms are also revealed by the ELF (Fig. 5). Therefore, all of the bonding states and lone pair states are filled and all of the anti-bonding states are unoccupied in $P\bar{1}$ - CaN_3 , leading to a semiconducting state.

To get further insight into the N–N bonding information in the $P\bar{1}$ - CaN_3 compound, we performed crystal orbital Hamilton population (COHP) analysis. COHP can partition the band-structure energy into the orbital-pair interactions and it is useful to indicate bonding, nonbonding, and antibonding contributions for the band-structure energy. Herein, the familiar COHP approach that stems from a plane-wave calculation was adopted and it was dubbed “projected COHP” (pCOHP).^{64,65} Then, we analyzed the COHP and integral COHP (ICOHP) to characterize the bonding nature of the N–N pairs in $P\bar{1}$ - CaN_3 . Fig. 6 shows that the nitrogen atom pairs with different distances exist in the form of the covalent bonds. The COHP plot of the N1–N3 pairs and N1–N5 pairs are shown in

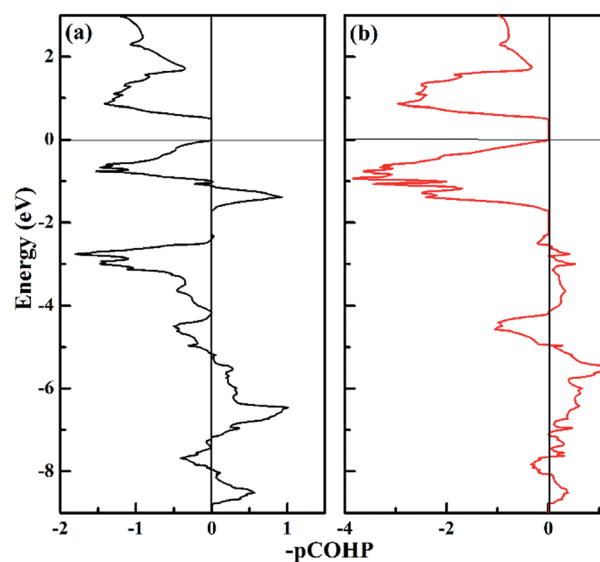


Fig. 6 Plot of COHP for $P\bar{1}$ - CaN_3 . The positive and negative COHP values denote bonding and antibonding interactions, respectively. (a) N1–N3 pairs separated by 1.34 Å. (b) N1–N5 pairs separated by 1.30 Å.



Fig. 6a and b, respectively. It is clear to see that the bonding states in $P\bar{1}$ - CaN_3 are fully occupied and the antibonding states are unoccupied, which can lead to the covalent bonding between two adjacent nitrogen atoms. The bonding strength based on counting the energy-weighted population of wave functions between two atomic orbitals in $P\bar{1}$ - CaN_3 can be expressed by the theoretical values of ICOHP. The values of ICOHP corresponding to those shown in Fig. 6 are listed in the order -4.243 and -3.044 . It is shown that the interaction between N1–N5 nitrogen atoms is stronger than that between N1–N3 atoms. For high-pressure $C2/m$ - CaN_3 , all N atoms in the “ N_6 ” ring are sp^2 -hybridized, two sp^2 -hybrid orbitals form two N–N σ bonds with neighboring two atoms, and the remaining sp^2 -hybrid orbital of this N atom is filled and forms one lone pair. The single-bond feature can be inferred from the similar bond lengths and strengths of these N–N bonds. As a potential high-energy-density material, the dissociation of $P\bar{1}$ - CaN_3 and $C2/m$ - CaN_3 is highly exothermic because of the existence of the single-bond feature.

4 Conclusions

In summary, the high-pressure polymeric nitrogen phase of CaN_3 was explored using first-principles calculations up to a high-pressure of 100 GPa. One new monoclinic $P\bar{1}$ - CaN_3 was predicted to become energetically stable under low pressure. For the first time, we identified a novel phase featuring the charged “ N_6 ” chain in the $P\bar{1}$ - CaN_3 structure. On further compression above 60 GPa, another monoclinic phase $C2/m$ - CaN_3 becomes the most thermodynamically stable phase and remains the lowest-enthalpy phase up to 100 GPa. Moreover, this new predicted high-pressure structure can be mechanically stable at ambient conditions, which may make an ambient-pressure recovery possible. The nitrogen atoms are sp^2 -hybridized in the “ N_6 ” chain and the “ N_6 ” ring, which indicates that each N atom forms one or two σ bonds with its neighboring N atoms, suggesting an exceptionally high content of the single N–N bonds. The present study provides new insights into the understanding of polynitrogens and encourages the experimental exploration of these promising materials in the future.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work was supported by the National Natural Science Foundation of China (No. 11747003, 11404134), Jilin Provincial Science and Technology Development Project of China (20160520016JH). Project 2017005 supported by Graduate Innovation Fund of Jilin University.

References

- 1 Y. Li, J. Hao, H. Liu, S. Lu and J. S. Tse, *Phys. Rev. Lett.*, 2015, **115**, 105502.
- 2 Z. Raza, C. J. Pickard, C. Pinilla and A. M. Saitta, *Phys. Rev. Lett.*, 2013, **111**, 235501.
- 3 L. Andrews, M. Zhou, G. V. Chertihin, W. D. Bare and Y. Hannachi, *J. Phys. Chem. A*, 2000, **104**, 1656–1661.
- 4 M. M. G. Alemany and J. L. Martins, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2003, **68**, 024110.
- 5 B. Hirshberg, R. B. Gerber and A. I. Krylov, *Nat. Chem.*, 2014, **6**, 52–56.
- 6 F. Zahariev, S. V. Dudiy, J. Hooper, F. Zhang and T. K. Woo, *Phys. Rev. Lett.*, 2006, **97**, 155503.
- 7 X. Wang, Y. Wang, M. Miao, X. Zhong, J. Lv, T. Cui, J. Li, L. Chen, C. J. Pickard and Y. Ma, *Phys. Rev. Lett.*, 2012, **109**, 175502.
- 8 F. Zahariev, A. Hu, J. Hooper, F. Zhang and T. Woo, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2005, **72**, 214108–6.
- 9 C. Mailhot, L. H. Yang and A. K. McMahan, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1992, **46**, 14419–14435.
- 10 Y. Ma, A. R. Oganov, Z. Li, Y. Xie and J. Kotakoski, *Phys. Rev. Lett.*, 2009, **102**, 065501.
- 11 M. I. Eremets, A. G. Gavriliuk, I. A. Trojan, D. A. Dzivenko and R. Boehler, *Nat. Mater.*, 2004, **3**, 558–563.
- 12 J. Uddin, V. Barone and G. E. Scuseria, *Mol. Phys.*, 2006, **104**, 745–749.
- 13 F. Peng, Y. Yao, H. Liu and Y. Ma, *J. Phys. Chem. Lett.*, 2015, **6**, 2363–2366.
- 14 X. Huang, D. Li, F. Li, X. Jin, S. Jiang, W. Li, X. Yang, Q. Zhou, B. Zou, Q. Cui, B. Liu and T. Cui, *J. Phys. Chem. C*, 2012, **116**, 9744–9749.
- 15 S. A. Medvedev, I. A. Trojan, M. I. Eremets, T. Palasyuk, T. M. Klapotke and J. Evers, *J. Phys.: Condens. Matter*, 2009, **21**, 195404.
- 16 X. Wang, J. Li, J. Botana, M. Zhang, H. Zhu, L. Chen, H. Liu, T. Cui and M. Miao, *J. Chem. Phys.*, 2013, **139**, 164710.
- 17 M. I. Eremets, M. Y. Popov, I. A. Trojan, V. N. Denisov, R. Boehler and R. J. Hemley, *J. Chem. Phys.*, 2004, **120**, 10618–10623.
- 18 K. Ramesh Babu and G. Vaitheeswaran, *Chem. Phys. Lett.*, 2013, **586**, 44–50.
- 19 B. A. Steele and I. I. Oleynik, *Chem. Phys. Lett.*, 2016, **643**, 21–26.
- 20 H. Zhu, F. Zhang, C. Ji, D. Hou, J. Wu, T. Hannon and Y. Ma, *J. Appl. Phys.*, 2013, **113**, 033511.
- 21 M. Zhang, K. Yin, X. Zhang, H. Wang, Q. Li and Z. Wu, *Solid State Commun.*, 2013, **161**, 13–18.
- 22 J. Hooper and E. Zurek, *J. Phys. Chem. C*, 2012, **116**, 13322–13328.
- 23 D. Hou, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2011, **84**, 064127.
- 24 D. Hou, F. Zhang, C. Ji, T. Hannon, H. Zhu, J. Wu, V. I. Levitas and Y. Ma, *J. Appl. Phys.*, 2011, **110**, 023524.
- 25 C. Ji, F. Zhang, D. Hou, H. Zhu, J. Wu, M.-C. Chyu, V. I. Levitas and Y. Ma, *J. Phys. Chem. Solids*, 2011, **72**, 736–739.
- 26 J. Li, X. Wang, N. Xu, D. Li, D. Wang and L. Chen, *Europhys. Lett.*, 2013, **104**, 16005.
- 27 J. Zhang, Z. Zeng, H.-Q. Lin and Y.-L. Li, *Sci. Rep.*, 2014, **4**, 4358.



- 28 X. Wang, J. Li, H. Zhu, L. Chen and H. Lin, *J. Chem. Phys.*, 2014, **141**, 044717.
- 29 Z. Liu, D. Li, S. Wei, W. Wang, F. Tian, K. Bao, D. Duan, H. Yu, B. Liu and T. Cui, *Inorg. Chem.*, 2017, **56**, 7494–7500.
- 30 A. F. Goncharov, N. Holtgrewe, G. R. Qian, C. H. Hu, A. R. Oganov, M. Somayazulu, E. Stavrou, C. J. Pickard, A. Berlie and Y. Fei, *J. Chem. Phys.*, 2015, **142**, 648–651.
- 31 K. Yin, Y. Wang, H. Liu, F. Peng and L. Zhang, *J. Mater. Chem. A*, 2015, **3**, 4188–4194.
- 32 A. Hu and F. Zhang, *J. Phys.: Condens. Matter*, 2010, **22**, 505402.
- 33 J. Wang, L. Zhang, F. Long, W. Wang, Y. Gu, S. Mo, Z. Zou and Z. Fu, *RSC Adv.*, 2016, **6**, 23272–23278.
- 34 D. Li, F. Tian, Y. Lv, S. Wei, D. Duan, B. Liu and T. Cui, *J. Phys. Chem. C*, 2017, **121**, 1515–1520.
- 35 Z. Raza, I. Errea, A. R. Oganov and A. M. Saitta, *Sci. Rep.*, 2013, **4**, 5889.
- 36 S. Wei, D. Li, Z. Liu, X. Li, F. Tian, D. Duan, B. Liu and T. Cui, *Phys. Chem. Chem. Phys.*, 2017, **19**, 9246–9252.
- 37 H. Wang, Y. Yao, Y. Si, Z. Wu and G. Vaitheeswaran, *J. Phys. Chem. C*, 2014, **118**, 650–656.
- 38 Y. Zhang, W. Wu, Y. Wang, S. A. Yang and Y. Ma, *J. Am. Chem. Soc.*, 2017, **139**, 13798–13803.
- 39 S. Zhu, F. Peng, H. Liu, A. Majumdar, T. Gao and Y. Yao, *Inorg. Chem.*, 2016, **55**, 7550–7555.
- 40 J. Lv, Y. Wang, L. Zhu and Y. Ma, *Phys. Rev. Lett.*, 2011, **106**, 015503.
- 41 Y. Wang, J. Lv, L. Zhu and Y. Ma, *Comput. Phys. Commun.*, 2012, **183**, 2063–2070.
- 42 Y. Wang, J. Lv, L. Zhu and Y. Ma, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2010, **82**, 094116.
- 43 J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1998, **80**, 891.
- 44 G. Kresse and J. Furthmüller, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1996, **54**, 11169–11186.
- 45 P. E. Blöchl, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1994, **50**, 17953–17979.
- 46 H. J. Monkhorst and J. D. Pack, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1976, **13**, 5188–5192.
- 47 G. Henkelman, A. Arnaldsson and H. Jónsson, *Comput. Mater. Sci.*, 2006, **36**, 354–360.
- 48 W. Tang, E. Sanville and G. Henkelman, *J. Phys.: Condens. Matter*, 2009, **21**, 084204.
- 49 A. Togo, F. Oba and I. Tanaka, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2008, **78**, 134106.
- 50 S. Grimme, *J. Comput. Chem.*, 2006, **27**, 1787–1799.
- 51 T. Bücko, J. Hafner, S. Lebégue and J. G. Ángyán, *J. Phys. Chem. A*, 2010, **114**, 11814.
- 52 J. P. Prates Ramalho, J. R. B. Gomes and F. Illas, *RSC Adv.*, 2013, **3**, 13085.
- 53 C. J. Pickard and R. J. Needs, *Phys. Rev. Lett.*, 2009, **102**, 125702.
- 54 W. J. Evans, J. Lipp, C.-S. Yoo, H. Cynn, J. L. Herberg, R. S. Maxwell and M. F. Nicol, *Chem. Mater.*, 2006, **18**, 2520.
- 55 Z.-J. Wu, E.-J. Zhao, H.-P. Xiang, X.-F. Hao, X.-J. Liu and J. Meng, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2007, **76**, 054115.
- 56 E. D. Raczynska, *Comput. Theor. Chem.*, 2011, **971**, 38–41.
- 57 M. N. Glukhovtsev and P. von Ragué Schleyer, *Chem. Phys. Lett.*, 1992, **198**, 547–554.
- 58 K. J. Wilson, S. A. Perera, R. J. Bartlett and J. D. Watts, *J. Phys. Chem. A*, 2001, **105**, 7693–7699.
- 59 H.-X. Duan and Q.-S. Li, *Chem. Phys. Lett.*, 2006, **432**, 331–335.
- 60 M. Zhang, H. Yan, Q. Wei, H. Wang and Z. Wu, *Europhys. Lett.*, 2013, **101**, 26004.
- 61 C. W. Allen, *RSC Pub.*, 2009, 17031–17032.
- 62 S. Yu, B. Huang, Q. Zeng, A. R. Oganov, L. Zhang and G. Frapper, *J. Phys. Chem. C*, 2017, **121**, 11037–11046.
- 63 A. D. Becke and K. E. Edgecombe, *J. Chem. Phys.*, 1990, **92**, 5397.
- 64 V. L. Deringer, A. L. Tchougréeff and R. Dronskowski, *J. Phys. Chem. A*, 2011, **115**, 5461–5466.
- 65 S. Maintz, V. L. Deringer, A. L. Tchougréeff and R. Dronskowski, *J. Comput. Chem.*, 2013, **34**, 2557–2567.

