



Cite this: *Nanoscale*, 2018, **10**, 6270

Received 1st December 2017,

Accepted 6th March 2018

DOI: 10.1039/c7nr08979a

rsc.li/nanoscale

Large-scale dewetting assembly of gold nanoparticles for plasmonic enhanced upconversion nanoparticles†

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Plasmonic nanostructures have been broadly investigated for enhancing many photophysical properties of luminescent nanomaterials. Precisely controlling the distance between the plasmonic nanostructure and the luminescent material is challenging particularly for the large-scale production of individual nanoparticles. Here we report an easy and reliable method for the large-scale dewetting of plasmonic gold nanoparticles onto core-shell (CS) upconversion nanoparticles (UCNPs). A commensurate NaYF₄ shell with a thickness between 5 nm and 15 nm is used as a tunable spacer to control the distance between the UCNP and the plasmonic gold nanoparticles. The upconversion emission intensity of single gold decorated core-inert shell (Au-CS) UCNPs is quantitatively characterized using a scanning confocal microscope. The results demonstrate the highest feasible enhancement of upconversion emission and a record reduction in lifetime for UCNPs fabricated in this manner. The Au-CS UCNPs are further investigated by simulation and synchrotron near edge X-ray absorption fine structure (NEXAFS) analysis.

bioimaging,^{3,4} biosensing,⁵ nanoscopy, theranostics, anti-counterfeiting, displays and photovoltaics.⁶ One of the most efficient UCNPs materials, lanthanide ion (Ln³⁺) doped β -NaYF₄ host nanocrystals, still suffers a low quantum yield ($\phi < 5\%$),⁷ which is far lower than the theoretical value for the two-photon process of upconversion ($\phi = 50\%$),⁸ limiting the practical applications of UCNPs.

To address this issue, a number of methods have been applied to enhance upconversion emission including host lattice manipulation, surface passivation, energy transfer modulation, broadband sensitization, surface plasmon coupling and photonic crystal engineering.⁹ Plasmonic nanoparticles of noble metals such as gold (Au) and silver (Ag) with tunable visible light absorption and scattering are frequently used to couple with UCNPs to enhance their upconversion emission. This plasmonic enhancement is typically dependent on the size, shape and composition of the noble metal nanoparticles used as well as the distance between the UCNPs and noble metal nanoparticles.^{10,11} These factors have led to both Au and Ag being used for plasmonic coupling in the forms of spherical nanoparticles,¹² nanowires,¹³ nanoislands¹⁴ and lithographically patterned nanoarrays.¹⁵

In this work, Au is easily produced into uniform spherical nanoparticles by sputtering, thermal annealing and dewetting. Dewetting is the disaggregation of a thin film into discrete particles, mediated by a minimisation of surface energy.¹⁶ This dewetting method allows the shape and size of the Au nanoparticles to be precisely controlled on a large scale so that their plasmon resonance can be tuned to match the emission wavelength of Er³⁺ emitters in NaYF₄ UCNPs¹⁷ by simply varying the film thickness, annealing temperature and annealing time.^{18,19} Secondly, the annealing during the dewetting of the Au onto the surface of the CS UCNPs promotes the diffusion of Ln³⁺ dopants within the UCNPs, reducing internal crystal defects which can lead to an increased quantum efficiency.²⁰

The design, consisting of an optically active UCNP core, an inert shell spacer and satellite plasmonic Au nanoparticles decorated on the shell, is chosen to suit the dewetting method

1. Introduction

The sequential absorption of two or more low energy photons to emit one higher energy photon is the process referred to as photon upconversion.¹ The current upconversion research is focused on developing upconversion nanoparticles (UCNPs),² with sizes in the range of 5 to 100 nm for applications such as

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†Electronic supplementary information (ESI) available: General procedures, simulation details, additional TEM/SEM images of the synthesized materials, simulation results, additional NEXAFS/XRD spectra, additional scanning confocal images and an optical path diagram of the confocal scanning microscope. See DOI: 10.1039/c7nr08979a



for a large-scale assembly of Au nanoparticles onto single and mono-dispersed hybrid CS UCNP. Fundamental to this design is the spacer which is vital for minimising the non-radiative energy transfer which causes luminescence quenching due to the direct contact of the UCNP with the Au nanoparticle surface.¹²

From this design, similar structures have been previously fabricated including both metal core-spacer-UC shell nanoparticles^{21–23} and UC core-spacer-metal shell nanoparticles^{24,25} as summarised in Table S1.† However, these previous studies relied exclusively on a spacer of metal, polymer or SiO₂ for attaching the Au nanoparticles by solution processing, which inhibits the plasmonic enhancement of luminescence due to the apparent parasitic absorption by these materials at the thicknesses required for optimum enhancement.^{26,27}

This limitation is overcome in this work by a third enhancement method using an undoped inert crystal shell. The thickness of inert shells is highly controllable and can be varied, allowing nanometre control by adjusting the concentration and amount of the shell precursor added during the epitaxial growth.^{28–30} Inert shells of the host crystal (NaYF₄ devoid of Ln³⁺) are employed to achieve both a reduction in surface defects to enhance the luminescence³¹ and functionality as the spacer in the design. To the authors' knowledge, this integrated method constitutes the first combination of core-shell UCNP, annealing and plasmonic enhancement in a single process to produce a singularly enhanced UCNP.

2. Results and discussion

Fig. 1(A) shows the TEM images of the core UCNP with an average diameter of 25 nm. The core UCNP is well-dispersed nanoparticles with a highly morphological uniformity. Fig. 1(B) and Fig. S1(A & C)† show the CS UCNP with an inert NaYF₄ shell coating of thicknesses 5 nm, 10 nm and 15 nm respectively; thus, the formed uniform CS UCNP are of sizes 35 nm, 45 nm and 55 nm. The ellipsoid shape of the 35 nm CS UCNP is due to the longitudinal preference of the epitaxial growth, which is especially pronounced when thin shell coatings are grown. This variation in the UCNP diameter of the near-spherical UCNP is within 2 nm, significantly smaller than the incremental step (5 nm) in the varied shell thickness. The dewetted Au nanoparticles shown in Fig. S1(B)† are uniformly scattered on the substrate having an average size of 15 nm. The Au-CS UCNP in Fig. 1(C) show the typical morphology of the dewetted Au onto the surface of CS UCNP. The CS UCNP have not undergone any morphology change while the Au nanoparticles have directly dewetted onto the surface of the UCNP and the desired structure of Au-CS UCNP is produced. Fig. S1(D)† shows an additional image of the same sample as shown in Fig. 1(C) of the Au-CS UCNP, and it highlights that the method can produce well-dispersed single Au-CS UCNP. The far separated CS UCNP coated with Au can be used directly for single nanoparticle luminescence measure-

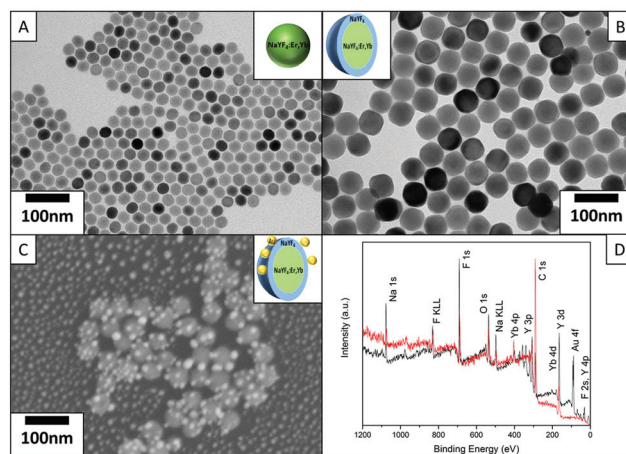


Fig. 1 TEM image of 25 nm diameter core UCNP of β -NaYF₄:20% Yb,2% Er (A), 55 nm diameter CS UCNP of β -NaYF₄:20% Yb,2% Er@NaYF₄ (B), SEM image of assembly of Au-CS UCNP of 55 nm diameter β -NaYF₄:20% Yb,2% Er@NaYF₄ after annealing at 400 °C for 1 hour (C) and XPS survey spectra of 55 nm diameter CS UCNP (red) and the assembled Au-CS UCNP of 55 nm diameter after annealing (black) (D).

ment. XPS survey spectra of both 55 nm CS UCNP and the as-obtained Au-CS UCNP after dewetting in Fig. 1(D) confirm the composition of the samples. The presence of Na, Y, F and Yb in both spectra can be attributed to the CS UCNP and the C and O peaks result from adventitious carbon.³² The additional Au peak is only present in the Au-CS UCNP after dewetting. High surface density samples were prepared for XPS on a Si substrate. The absence of Si signals in both spectra indicates that XPS signals arise from the UCNP films with no contributions from the silicon substrate. As a result, the entire Au signal is detected from the dewetted Au nanoparticles on the CS UCNP rather than Au dewetted on the surface of the substrate.

The dewetting treatment at 400 °C for 1 hour is not only crucial to convert the Au film into discrete nanoparticles of specific size¹⁸ on the CS UCNP surface, but it is also the optimum condition to prevent the CS UCNP from aggregating and changing the phase. The XRD characterization in Fig. S2 (C)† on the CS UCNP before and after annealing at 400 °C for 1 hour shows that the CS UCNP retain their β phase.^{33,34}

To investigate the effect of annealing on the diffusion of Er and Yb dopants within the CS UCNP, Near Edge X-ray Absorption Fine Structure (NEXAFS) analysis was conducted at the Australia Synchrotron. NEXAFS analysis in partial electron yield (PEY) and total fluorescent yield (TFY) using a modified procedure³⁵ was performed to measure the re-distribution of the Er and Yb dopants in the CS UCNP due to the annealing process. Fig. 2(A & C) shows NEXAFS spectra in TFY mode which probes the entire UCNP volume. The TFY intensity for both Er (Fig. 2A) and Yb (Fig. 2C) before and after annealing is the same, and this indicates that for both samples, the surface coverage of UCNP on the substrate is approximately equal. In PEY mode, the NEXAFS spectra are being primarily generated



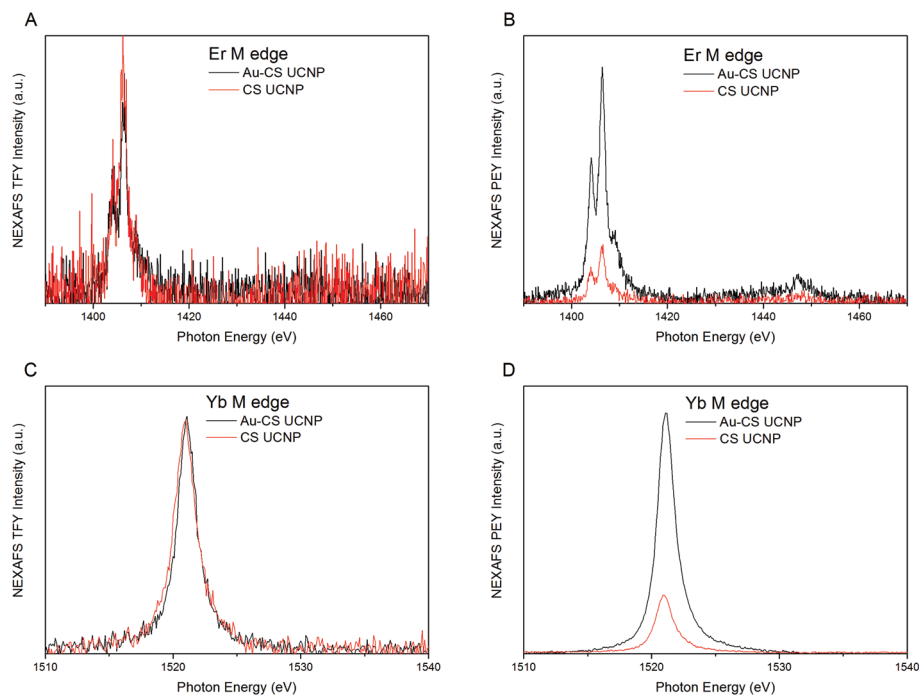


Fig. 2 NEXAFS spectra for Er M edge and Yb M edge of the CS UCNP with a 5 nm shell before (CS UCNP) and after annealing (Au-CS UCNP) in TFY (A & C) and PEY (B & D) modes.

from only the shell volume allowing for any change in the distribution of the dopants to be identified. For both Er (Fig. 2B) and Yb (Fig. 2D) after annealing, there is an increase in the intensity in the PEY spectrum. This suggests a denser Yb and Er dopant distribution towards the surface of the CS UCNP after annealing, indicating that both dopants have diffused further from the core towards the inert shell during the annealing process. The driving force for the diffusion would be the concentration gradient of the dopant from the core nanocrystal to the inert shell accelerated under the annealing conditions. As the surface coverage was approximately equal, the relative increase of dopants diffused towards the shell after annealing can be estimated to be a factor of 3.8 and 4.4 for Er and Yb, respectively. As expected, the increase in Yb is higher than Er due to the higher concentration gradient of Yb to Er (20% to 2%) across the CS UCNP. The same result was demonstrated by the CS UCNP and Au-CS UCNP with a 15 nm shell. The results are shown in Fig. S2,† in which a relative increase in Yb after annealing is estimated to be a factor of 1.9.

The diffusion of the dopant ions from the core UCNP to the inert shell firstly produces a reduced effective distance between the Au nanoparticles on the inert shell and the active core UCNP. Concurrently, this diffusion also confirms that the annealing has facilitated an increase in the diffusion coefficient of the Ln^{3+} dopants producing an improved and refined crystal from internal defect removal during diffusion.^{20,34}

To study the surface plasmon resonance of the Au-CS UCNP system, the luminescent emission of CS UCNP and the optical transmission of the dewetted Au nanoparticles

were characterized. As shown in Fig. 3(A), the Au nanoparticles of size 15 nm produced from dewetting show a broad absorption peaked at 540 nm. The luminescent emission of the CS UCNP shows the green emission at 525 nm and 540 nm and the red emission at 650 nm and 660 nm. The surface plasmon resonance of the Au nanoparticles overlaps precisely with the green emissions which are the radiative transitions of $(^2\text{H}_{11/2}, ^4\text{S}_{3/2}) \rightarrow ^4\text{I}_{15/2}$ of Er^{3+} from the UCNP as shown in Fig. 3(B).

To investigate the luminescence performance of the Au-CS UCNP, the emission intensity of single CS UCNP, annealed CS UCNP and assembled Au-CS UCNP were recorded using a purpose-built scanning confocal microscopy system for single nanoparticles.^{36,37} The scanned images of Fig. 4 show the single nanoparticle emission images of the core UCNP (A), CS UCNP with a 5 nm, 10 nm and 15 nm inert shell (B–D), core Au-UCNP (E), and Au-CS UCNP with a 5 nm, 10 nm and 15 nm inert shell (F–H), respectively. Fig. S3† additionally shows the annealed CS UCNP scanned images of the core UCNP (A), and CS UCNP with a 5 nm, 10 nm and 15 nm inert shell (B–D) after annealing under identical conditions but without an Au film. Single UCNP can be reliably identified using this method due to their near identical emission intensity. The summary of these emission intensity measurements in Fig. 4(I) has the CS UCNP's emission enhancement saturated at a shell thickness of 5 nm and above as reported previously,³¹ while the luminescence intensity of the annealed CS UCNP without Au displays an insignificant variation to the CS UCNP with increasing shell thickness. This result is expected from the XRD results in Fig. S2(C)† that show the



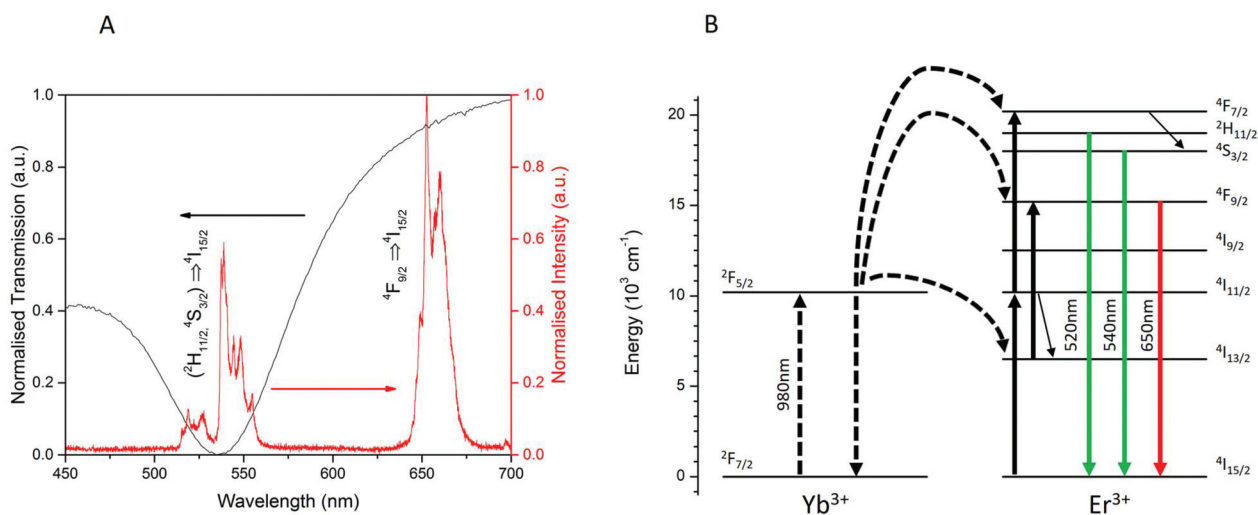


Fig. 3 Luminescence spectrum of β -NaYF₄:20% Yb, 2% Er UCNPs (red) overlaid with the UV-Vis transmission of pure dewetted Au nanoparticles (black) (A) and the energy level diagram of the Yb–Er upconversion system showing the upconversion excitation and the green and red emissions with their wavelengths (B). The surface plasmon resonance (~ 540 nm) of the dewetted Au nanoparticles matches the $(^2\text{H}_{11/2}, ^4\text{S}_{3/2}) \rightarrow ^4\text{I}_{15/2}$ emission of Er³⁺.

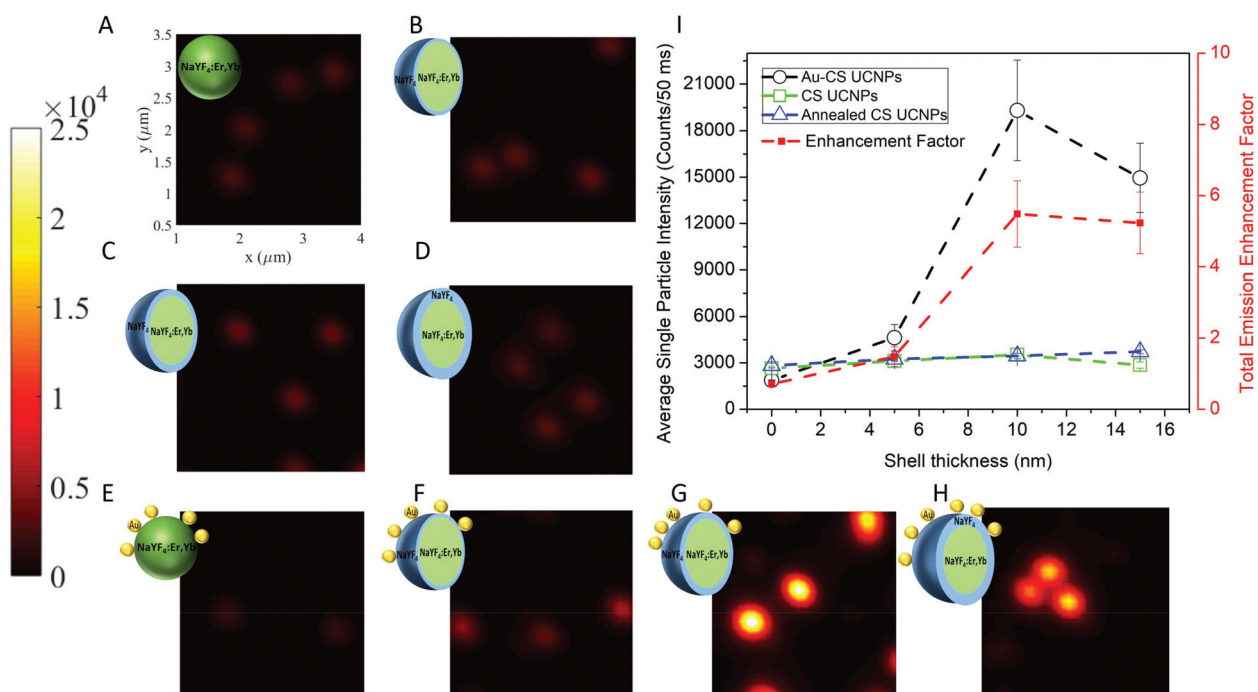


Fig. 4 The scanning confocal images of the core UCNPs (A), CS UCNPs with a 5 nm, 10 nm and 15 nm inert shell (B–D), and core Au-UCNPs (E), Au-CS UCNPs with a 5 nm, 10 nm and 15 nm inert shell (F–H), respectively. Each scanned image is $3 \mu\text{m} \times 3 \mu\text{m}$ as indicated in (A) and the colour bar indicates the emission intensity range of the single UCNP. Single UCNP total emission intensity and enhancement factor (I) for CS UCNPs (A–D), Au-CS UCNPs (E–H) and annealed CS UCNPs (Fig. S3A–D†). The laser power was fixed at 20 mW, which was measured at the objective back aperture.

annealed CS UCNPs retained their β phase. It also suggests that any luminescence enhancement resulting from the improved crystal quality obtained by the reduction of internal defects during the diffusion of dopants as demonstrated by the NEXAFS results in Fig. 2 has been effectively cancelled out

by the increase in the distance between the dopant Yb and Er ions this diffusion would cause. This increase in distance between the dopant ions would reduce the probability of energy transfer between Yb–Yb and Yb–Er reducing upconversion emission.

On the other hand, the luminescence intensity of the CS UCNPs decorated with dewetted Au nanoparticles (Au-CS UCNPs) demonstrates a clear dependence on the shell thickness. A maximum enhancement factor of 5.5 for total emission is achieved by the 10 nm shell Au-CS UCNPs. This maximum enhancement factor at a separation of 10 nm is consistent with the simulation results in Liu's work.³⁸ These simulations ascribe the total enhancement (f_{tot}) from the Au nanoparticle to be a product of the excitation enhancement from the local E -field intensity (E) and the emission enhancement from the increased Purcell factor (F) plus antenna efficiency (η_a). As η_a is related to the initial quantum efficiency (η_0) of the UCNPs, f_{tot} from the Au nanoparticle is highly dependent on the η_0 of the UCNPs, with a higher η_0 resulting in substantially lower total enhancement factors, *i.e.* $\eta_0 = 0.1\%$ and $f_{\text{tot}} = 30$ *vs.* $\eta_0 = 10.0\%$ and $f_{\text{tot}} = 5$. The coating of an inert shell onto a UCNP increases the quantum efficiency by reducing surface

defects,^{6,8} so does the annealing of certain UCNPs by reducing internal defects.²⁰ In this work, both processes are applied cooperatively to the UCNPs. Simulations of the initial quantum efficiency attenuation to the internal defect density in Fig. S4(B)† show that the η_0 of UCNPs is highly dependent on the density of internal defects. This explains the modest total enhancement factor of 5.5 to be a value approaching the highest feasible factor achievable for the already highly efficient annealed CS UCNPs.

To further verify the surface plasmonic effect of Au nanoparticles on the CS UCNPs, we compared the lifetime of the single CS UCNPs and Au-CS UCNPs. The CS UCNPs display in Fig. 5(A) a common trend of increased lifetime with increasing shell thickness due to the nature of the lifetime of the $\text{Yb}^{3+} {}^2\text{F}_{5/2}$ energy level increasing continuously with the inert shell thickness as non-radiative surface-quenching processes become passivated more effectively. This increased $\text{Yb}^{3+} {}^2\text{F}_{5/2}$

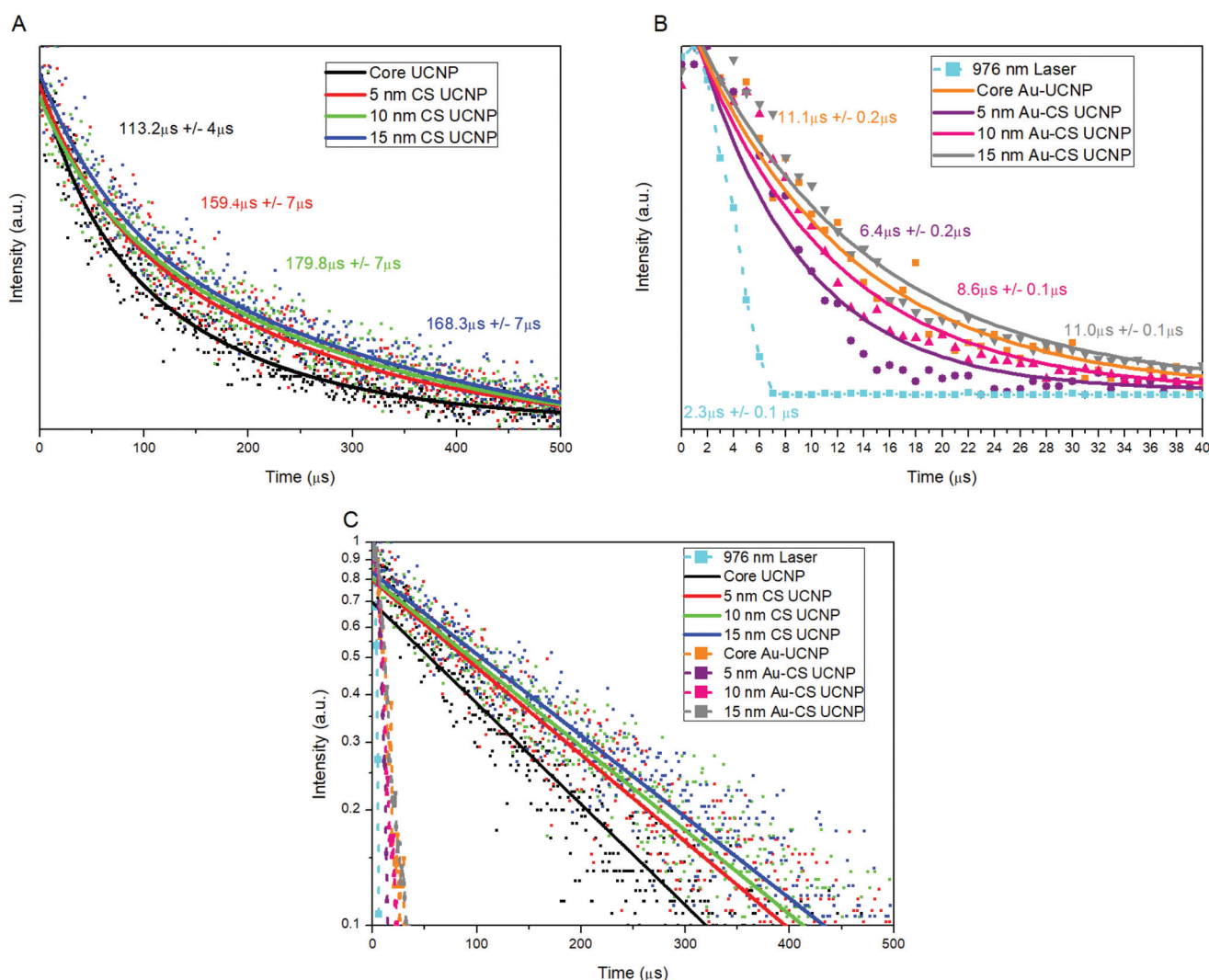


Fig. 5 Single nanoparticle lifetime measurement recorded for CS UCNP (A), Au-CS UCNP (B) and both single CS UCNP and single Au-CS UCNP on a semi-log plot (C). The laser power was fixed at 20 mW, which was measured at the objective back aperture. The luminescence lifetime of the CS UCNPs decreases by more than an order of magnitude after decorating with dewetted Au nanoparticles.



lifetime allows more time for the excitation energy to migrate through the nanocrystal, increasing the probability of energy transfer to an Er^{3+} .³¹ For all Au-CS UCNPs, a significant reduction in total lifetime was observed as shown in Fig. 5(B). The total lifetime of the Au-CS UCNPs was reduced at a maximum by a factor of 25, and this magnitude is clearly shown in Fig. 5(C). This approached the same magnitude as the lifetime of the laser used in the system measured at 2.3 μs to verify the sensitivity of the system.

The decrease in emission lifetime for all Au-CS UCNPs is significant being among the highest recorded for plasmonic coupling^{39,40} demonstrating the competition between two processes occurring, the Purcell effect and quenching from Förster-type energy transfer (FRET).⁴¹ The first process is the Purcell effect, which can increase the radiative recombination rate of an emitter by increasing the local density of transition states, thereby enhancing the emission intensity and reducing the emission lifetime. For upconversion emitters, e.g. Er^{3+} , the intrinsic non-radiative recombination rate is much faster than the radiative decay rate,⁴² thus the change in the total lifetime of the emitter is dominated by the change in its radiative rate. Therefore, the total lifetime of Au-CS UCNPs is dramatically reduced, as its emission wavelength matches with the plasmonic resonance of the Au nanoparticles as shown in Fig. 3(A), which is consistent with previous research.⁴³

Förster-type energy transfer (FRET) is the other process where the carriers on the emitter's energy level directly interact with the Au nanoparticles, leading to a reduction in both the emission intensity and lifetime through increasing the non-radiative rate. FRET is still significant when the effective separation distance between the dewetted Au nanoparticles and Er^{3+} emitters is below 15 nm (ref. 38), especially for the Au-CS UCNPs due to the direct contact of Au nanoparticles with the surface of the CS UCNP NaYF_4 host material.

The competition between the Purcell effect and FRET has been explored in the Au-CS UCNPs through the lifetime measurements. As a result of the competition, the lifetime of Au-UCNPs (11 μs) is longer than that of the Au-CS UCNPs with a shell thickness of 5, 10 and 15 nm, as shown in Fig. 5 (B). As the shell thickness is increased to 15 nm, the influence of FRET starts to diminish; however, the lifetime of the Au CS UCNPs is still an order of magnitude lower than the CS UCNPs lifetime, indicating that the Purcell effect dominates over the FRET. Further evidence of the dominant role of the Purcell effect is the decreased intensity in Fig. 4(I) and the increased lifetime in Fig. 5(B) and (C), as the shell thickness is increased beyond 10 nm on the Au-CS UCNPs. The thicker shell will reduce the Purcell effect, thereby reducing the radiative recombination rate, which leads to a decrease in intensity and an increase in lifetime.

3. Conclusion

The novel combination of CS UCNP synthesis, thermal annealing and dewetting of Au explored in this work produces Au-CS

UCNPs that exhibit the optimized plasmonic enhanced single particle upconversion emission. This enhancement is strongly dependent upon the thickness of the inert shell with a maximum enhancement factor of 5.5 in total emission intensity achieved by the 10 nm shell Au-CS UCNPs. The design and integrated method of fabrication of these Au-CS UCNPs represents a new direction in the research on the plasmonic enhancement of upconversion emission.

Conflicts of interest

The authors declare no competing financial interest.

Acknowledgements

XPS and NEXAFS measurements were undertaken on the Soft X-ray Spectroscopy beamline at the Australian Synchrotron. We wish to acknowledge the technical assistance of Drs Bruce Cowie and Anton Tadich. This project was primarily supported by the Australian Government Research Training Program Scholarship (Christian Clarke) and China Scholarship Council CSC scholarships (Chaohao Chen: No. 201607950009 and Yongtao Liu: No. 201607950010).

References

- 1 F. Auzel, *Chem. Rev.*, 2004, **104**, 139–174.
- 2 S. Gai, C. Li, P. Yang and J. Lin, *Chem. Rev.*, 2013, **114**, 2343–2389.
- 3 M. Nyk, R. Kumar, T. Y. Ohulchanskyy, E. J. Bergey and P. N. Prasad, *Nano Lett.*, 2008, **8**, 3834–3838.
- 4 L. Cheng, K. Yang, Y. Li, X. Zeng, M. Shao, S.-T. Lee and Z. Liu, *Biomaterials*, 2012, **33**, 2215–2222.
- 5 L. Wang, R. Yan, Z. Huo, L. Wang, J. Zeng, J. Bao, X. Wang, Q. Peng and Y. Li, *Angew. Chem., Int. Ed.*, 2005, **44**, 6054–6057.
- 6 J. C. Goldschmidt and S. Fischer, *Adv. Opt. Mater.*, 2015, **3**, 510–535.
- 7 M. Pokhrel, A. Kumar Gangadharan and D. K. Sardar, *Mater. Lett.*, 2013, **99**, 86–89.
- 8 J.-C. Boyer and F. C. Van Veggel, *Nanoscale*, 2010, **2**, 1417–1419.
- 9 S. Han, R. Deng, X. Xie and X. Liu, *Angew. Chem., Int. Ed.*, 2014, **53**, 11702–11715.
- 10 S. Fischer, F. Hallermann, T. Eichelkraut, G. von Plessen, K. W. Krämer, D. Biner, H. Steinkemper, M. Hermle and J. C. Goldschmidt, *Opt. Express*, 2012, **20**, 271–282.
- 11 S. Fischer, D. Kumar, F. Hallermann, G. von Plessen and J. C. Goldschmidt, *Opt. Express*, 2016, **24**, A460–A475.
- 12 S. Schietinger, T. Aichele, H.-Q. Wang, T. Nann and O. Benson, *Nano Lett.*, 2009, **10**, 134–138.
- 13 W. Feng, L.-D. Sun and C.-H. Yan, *Chem. Commun.*, 2009, 4393–4395.



- 14 H. Zhang, D. Xu, Y. Huang and X. Duan, *Chem. Commun.*, 2011, **47**, 979–981.
- 15 H. P. Paudel, L. Zhong, K. Bayat, M. F. Baroughi, S. Smith, C. Lin, C. Jiang, M. T. Berry and P. S. May, *J. Phys. Chem. C*, 2011, **115**, 19028–19036.
- 16 C. M. Müller and R. Spolenak, *Acta Mater.*, 2010, **58**, 6035–6045.
- 17 R. Gupta, M. Dyer and W. Weimer, *J. Appl. Phys.*, 2002, **92**, 5264–5271.
- 18 C. Worsch, M. Kracker, W. Wisniewski and C. Rüssel, *Thin Solid Films*, 2012, **520**, 4941–4946.
- 19 M. Kracker, C. Worsch, C. Bocker and C. Rüssel, *Thin Solid Films*, 2013, **539**, 47–54.
- 20 N. C. Dyck, F. C. van Veggel and G. P. Demopoulos, *ACS Appl. Mater. Interfaces*, 2013, **5**, 11661–11667.
- 21 F. Zhang, G. B. Braun, Y. Shi, Y. Zhang, X. Sun, N. O. Reich, D. Zhao and G. Stucky, *J. Am. Chem. Soc.*, 2010, **132**, 2850–2851.
- 22 W. Ge, X. Zhang, M. Liu, Z. Lei, R. Knize and Y. Lu, *Theranostics*, 2013, **3**, 282–288.
- 23 C. Zhang and J. Y. Lee, *J. Phys. Chem. C*, 2013, **117**, 15253–15259.
- 24 L. L. Li, R. Zhang, L. Yin, K. Zheng, W. Qin, P. R. Selvin and Y. Lu, *Angew. Chem., Int. Ed.*, 2012, **51**, 6121–6125.
- 25 P. Kannan, F. Abdul Rahim, R. Chen, X. Teng, L. Huang, H. Sun and D.-H. Kim, *ACS Appl. Mater. Interfaces*, 2013, **5**, 3508–3513.
- 26 Z. Wang, W. Gao, R. Wang, J. Shao, Q. Han, C. Wang, J. Zhang, T. Zhang, J. Dong and H. Zheng, *Mater. Res. Bull.*, 2016, **83**, 515–521.
- 27 J. Shen, Z. Li, Y. Chen, X. Chen, Y. Chen, Z. Sun and S. Huang, *Appl. Surf. Sci.*, 2013, **270**, 712–717.
- 28 X. Li, D. Shen, J. Yang, C. Yao, R. Che, F. Zhang and D. Zhao, *Chem. Mater.*, 2012, **25**, 106–112.
- 29 N. J. Johnson, A. Korinek, C. Dong and F. C. van Veggel, *J. Am. Chem. Soc.*, 2012, **134**, 11068–11071.
- 30 D. Liu, X. Xu, Y. Du, X. Qin, Y. Zhang, C. Ma, S. Wen, W. Ren, E. M. Goldys and J. A. Piper, *Nat. Commun.*, 2016, **7**, 10254.
- 31 S. Fischer, N. D. Bronstein, J. K. Swabeck, E. M. Chan and A. P. Alivisatos, *Nano Lett.*, 2016, **16**, 7241–7247.
- 32 T. L. Barr and S. Seal, *J. Vac. Sci. Technol.*, 1995, **13**, 1239–1246.
- 33 G. Yi, H. Lu, S. Zhao, Y. Ge, W. Yang, D. Chen and L.-H. Guo, *Nano Lett.*, 2004, **4**, 2191–2196.
- 34 H. Assaaoudi, G.-B. Shan, N. Dyck and G. P. Demopoulos, *CrystEngComm*, 2013, **15**, 4739–4746.
- 35 X. Xu, C. Clarke, C. Ma, G. Casillas, M. Das, M. Guan, D. Liu, L. Wang, A. Tadich and Y. Du, *Nanoscale*, 2017, **9**, 7719–7726.
- 36 C. Ma, X. Xu, F. Wang, Z. Zhou, D. Liu, J. Zhao, M. Guan, C. I. Lang and D. Jin, *Nano Lett.*, 2017, **17**, 2858–2864.
- 37 C. Ma, X. Xu, F. Wang, Z. Zhou, S. Wen, D. Liu, J. Fang, C. I. Lang and D. Jin, *J. Phys. Chem. Lett.*, 2016, **7**, 3252–3258.
- 38 X. Liu and D. Y. Lei, *Sci. Rep.*, 2015, **5**, 15235.
- 39 D. M. Wu, A. García-Etxarri, A. Salteo and J. A. Dionne, *J. Phys. Chem. Lett.*, 2014, **5**, 4020–4031.
- 40 W. Zhang, F. Ding and S. Y. Chou, *Adv. Mater.*, 2012, **24**, OP236–OP241.
- 41 W. Park, D. Lu and S. Ahn, *Chem. Soc. Rev.*, 2015, **44**, 2940–2962.
- 42 M. Saboktakin, X. Ye, S. J. Oh, S.-H. Hong, A. T. Fafarman, U. K. Chettiar, N. Engheta, C. B. Murray and C. R. Kagan, *ACS Nano*, 2012, **6**, 8758–8766.
- 43 D. Lu, S. K. Cho, S. Ahn, L. Brun, C. J. Summers and W. Park, *ACS Nano*, 2014, **8**, 7780–7792.

