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Photocatalytic NO removal over calcium-bridged siloxenes under ultraviolet and visible light irradiation†

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Ca-Bridged siloxenes (Ca-siloxenes) composed of two-dimensional siloxene planes with Ca bridging were prepared and their photocatalytic properties for nitrogen oxide (NO) removal were investigated. Ca-Siloxenes were synthesized *via* a solid-state metathesis reaction using TaCl₅ to extract Ca from CaSi₂ with different Cl₂/Ca molar ratios of 0.25, 1.25 and 2.5 (CS0.25, CS1.25 and CS2.5, respectively) in an attempt to control the extent of Ca extraction. Ca-Siloxenes have a wide optical absorption band from the visible to ultraviolet region with absorption edges of 1.5, 2.9, and 3.1 eV for CS0.25, CS1.25, and CS2.5, respectively. Ca-Siloxenes exhibited photocatalytic activity for NO removal under irradiation with visible ($\lambda > 400$ nm (<3.10 eV)) and ultraviolet light ($\lambda > 290$ nm (<4.28 eV)). The photocatalytic activity was particularly improved by mixing the Ca-siloxene with acetylene black as a conductive material, which might have inhibited the recombination of photogenerated electrons and holes. The mixture of Ca-siloxene and acetylene black exhibited improved photocatalytic activity in the presence of ¹O₂ as one of the active oxygen species formed under ultraviolet light irradiation.

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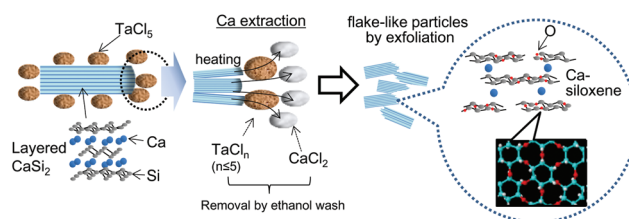
1. Introduction

Nitrogen oxide (NOx) is known as an air pollutant that is mainly generated from combustion at industrial power plants and in automobiles. Several negative impacts of NOx on the environment have been recognized, such as the formation of acid rain, photochemical air pollution and ozone layer depletion. Therefore, trials to minimize the effects of NOx have been performed, and catalysts have often been applied for NOx removal at the outlet of combustion systems. For example, selective catalytic reduction and lean NOx traps are useful devices for the removal of NOx in automobiles.^{1–3} However, once released to the atmosphere, the removal of NOx is difficult under ambient conditions due to low NOx concentration and energy sources required for the purification and regeneration of conventional catalysts.

The use of photocatalysts is one possible method to reduce NOx under ambient conditions with the use of sunlight irradiation as a natural resource. TiO₂ is the most common

photocatalyst⁴ and effective for photocatalytic NOx removal,⁵ and other materials such as perovskite-type oxides have also been investigated.^{6,7} Silicon-based materials have been investigated for hydrogen production by water splitting;^{8–11} however, the application of Si-based materials to NOx removal is rare.¹²

In our previous study, we reported a novel calcium-bridged siloxene (Ca-siloxene) produced *via* a convenient solid-state metathesis reaction with an ethanol wash (Scheme 1).^{13,14} In this synthesis, Ca is extracted from layered CaSi₂ with the help of TaCl₅ as a Cl source to form CaCl₂, while maintaining partial Ca-bridging between Kautsky-type siloxene planes^{15,16} with Si₆ rings connected through Si–O–Si bridges. The presence of Kautsky-type siloxene planes was determined using Raman and Fourier transform-infrared spectroscopy measurements, which showed Si₆ rings in Si-based planes and those connected through Si–O–Si bridges, respectively.¹³ Ca-Bridging



Scheme 1 The solid state reaction employed for Ca-siloxene synthesis.

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in Ca-siloxenes was characterized using the X-ray absorption fine structure, which showed the presence of Si–Ca bonding.¹³ The extent of Ca-extraction and the amount of the resulting Ca-bridging are controlled by variation in the TaCl₅ concentration, and both primary and secondary particles become smaller with increased pore volumes as the Cl₂/Ca molar ratio is increased by fragmentation.^{13,14}

Ca-Siloxenes had advantages over conventional Kautsky-type siloxenes. For example, the structural stability of Ca-siloxene based on its Ca-bridging was confirmed to be responsible for its stable charge and discharge performance as a Li-ion battery anode.¹⁴ A similar phenomenon for stability by ion intercalation has also been reported in 2D materials such as graphene and g-C₃N₄ for enhancing the interlayer van der Waals force^{17,18} for photocatalysis.^{19,20} In addition, it has been suggested that Ca-siloxene had a tunable direct bandgap energy that is dependent on the amount of Ca-bridging.¹³

A bare Ca-siloxene was reported to exhibit photocatalytic NO removal;¹² however, its compositional dependence was not examined. It was also implied that the addition of metallic particles may enhance the photocatalytic activity of the bare Ca-siloxene.¹² Herein, we report a systematic investigation of these points; the Ca-siloxenes are characterized with respect to optical and electrical properties by variation in the extent of Ca extraction in the synthesis step and the amount of conductive additives. The photocatalytic NO removal by Ca-siloxenes and the effect of conductive additives on the photocatalytic performance under ultraviolet and visible light irradiation are also discussed as a potential application. Chemiluminescence emission measurements for phosphorescence decay are also conducted to deduce a possible reactive species for photocatalytic NO removal.

2. Experimental

2.1. Synthesis of Ca-siloxenes¹³

Ca-Siloxenes were synthesized using Cl₂/Ca molar ratios of 0.25, 0.5 and 2.5 (CS0.25, CS1.25 and CS2.5, respectively). In the synthesis, CaSi₂ (3.2, 1.6 and 0.32 mmol for CS0.25, CS1.25 and CS2.5, Rare Metallic Co.) and TaCl₅ (0.32 mmol, Wako Pure Chemical) were used as starting materials. A mixture of CaSi₂ and TaCl₅ was placed in boron nitride (BN) crucibles. These mixtures were calcined at 215 °C for 5 h at a heating rate of 100 °C h⁻¹. The mixtures packed in the BN crucibles were subsequently placed in sealed stainless steel cells (inner volume of 10 cm³). The products were then washed with anhydrous ethanol to remove the by-product CaCl₂ and excess TaCl₅ or their derivatives, before drying under vacuum at 80 °C. Each step in the synthesis was conducted under an Ar atmosphere. Kautsky-type siloxene as a reference sample was synthesized by the oxidation of CaSi₂ with HCl aqueous solution, as previously described in the literature,^{15,16} and CaSi₂ was also used as a reference sample. To examine the effects of mixing conductive materials in the samples, Ca-siloxene/acety-

lene black (AB) mixtures with mass ratios of 99/1, 95/5, 90/10, and 70/30 were prepared.

2.2. Characterization of Ca-siloxenes

The crystalline structures of Ca-siloxenes were investigated using X-ray diffraction (XRD; Rigaku RINT-TTR). Ultraviolet (UV)/visible absorption spectra were acquired with a spectrometer (JASCO V-670) over a 2 mm diameter of each sample in the diffuse reflectance mode from 200 to 2500 nm at room temperature under air. Optical bandgap energy values were derived from Tauc plots²¹ of the associated absorption data processed under the Kubelka–Munk function,²² based on the following eqn (1)–(3):

$$F(R_{\infty}) = (1 - R_{\infty})^2 / 2R_{\infty} = K/S, \quad (1)$$

$$R_{\infty} = R_{\infty}(\text{sample}) / R_{\infty}(\text{standard sample}), \quad (2)$$

$$(F(R_{\infty})h\nu)^{1/n} = A(h\nu - E_g), \quad (3)$$

where R_{∞} is the absolute reflectance, K is the absorption coefficient, S is the scattering coefficient, $h\nu$ is the photon energy, n is determined according to the transition mode, A is a coefficient, and E_g is the optical bandgap or the absorption edge. BaSO₄ was used as a standard sample, and $n = 1/2$ was applied for direct allowed transitions, as suggested for Ca-siloxenes.^{12,13}

The electrical conductivity of the samples was evaluated using a two-probe method with pelletized specimens at room temperature under air. These samples were uniaxially pressed at 8 MPa under DC current application.

The scanning electron microscopy (SEM) image of CS0.25 with 10 wt% of acetylene black was obtained for morphology characterization, using a Hitachi S-3600N scanning electron microscope.

2.3. Photocatalytic properties

The photocatalytic performance of the samples was measured based on the NO concentration at the outlet of the reactor using a NOx analyzer (Yanaco, ELC-88A) in the same manner as published in the literature.^{7,12,23,24} The reactor volume was 373 cm³ and the size of the sample glass holder was 20 × 15 × 0.5 mm. A constant flow rate of 200 mL min⁻¹ of 1 ppm NO–50 vol% air with N₂ balance was applied for the reaction. Before light irradiation, the reactor with samples was purged under a constant flow of 1 ppm NO–50 vol% air with N₂ balance for 10 min to achieve diffusion and adsorption balance. Light was irradiated using a 450 W high-pressure mercury lamp using wavelengths of $\lambda > 290$ nm (<4.28 eV) or >400 nm (<3.10 eV) with filtering out the UV emission. NO conversion was calculated based on the following eqn (4):

$$\text{NO conversion (\%)} = 100 \times C_{\text{NO}} / C_{\text{NO-feed}} \quad (4)$$

where C_{NO} is the NO concentration measured using a NOx analyzer at the outlet of the reactor and $C_{\text{NO-feed}}$ is the feed NO concentration measured before light irradiation. Durability



tests were conducted by alternately switching UV light ($\lambda > 290$ nm (<4.28 eV)) on and off for 1 h each.

The apparent quantum efficiency was calculated based on the following eqn (5):²⁴

$$\Phi(\%) = 100 \times F_{\text{NO}}\alpha_{\lambda}/P_{\lambda}SA_{\lambda}, \quad (5)$$

where F_{NO} ($\mu\text{mol s}^{-1}$) is the NO amount rate in the feed gas, α_{λ} is the NO conversion of samples (%), P_{λ} ($\mu\text{mol s}^{-1}$) is the light intensity on the surface of the sample holder ($\lambda > 290$ nm: 1139.8 ($\mu\text{mol s}^{-1}$), >400 nm: 846.7 ($\mu\text{mol s}^{-1}$)), S is the surface area of the sample holder (1.28×10^{-3} m²), and A_{λ} is the average light absorption ratio of samples at different light wavelengths based on the UV/visible absorption spectra.

The phosphorescence decay profile of singlet oxygen ($^1\text{O}_2$) at a wavelength of 634 nm was measured after UV light ($\lambda = 375$ nm) irradiation for 5 s using a multi-luminescence spectrometer (Tohoku Electric Ind., MLA-GOLDS). The data were obtained by subtracting the luminous intensity of $\lambda > 640$ nm from that of $\lambda > 620$ nm using two different cutoff filters.

3. Results and discussion

3.1. Characterization of Ca-siloxenes for optical and conductive properties

After the reaction of CaSi_2 with TaCl_5 , the synthesized samples exhibited broad peaks in the region from 25 to 40° due to the formation of an amorphous phase (Fig. S1†). Although CS0.25 contained a crystalline phase derived from unreacted CaSi_2 , broad peaks became dominant in CS1.25 and CS2.5, which suggests the fragmentation of layered CaSi_2 by Ca extraction and simultaneous exfoliation during synthesis.¹³ Thus, CS0.25 is composed of a mixture of Ca-siloxene particles and remaining crystalline CaSi_2 particles, of which the average molar ratio of Ca to Si is $0.61 : 2$.¹⁴ On the other hand, CS1.25 and CS2.5 are mainly composed of Ca-siloxene particles with average Ca : Si molar ratios of $0.27 : 2$ and $0.09 : 2$, respectively.¹⁴

The UV/visible absorption spectra of Ca-siloxenes and Kautsky-type siloxene measured in the diffuse reflectance mode are shown in Fig. 1(A). Absorption spectra were measured from 200 nm (6.20 eV) to provide more accurate spectra in the UV region in contrast to those measured from 300 nm (4.13 eV) in our previous work.¹³ All Ca-siloxenes had a characteristic broad absorption band ranging from the UV to visible light region with a clear peak at less than 250 nm and a shoulder and/or a peak at around 400 nm, which may be preferable for photocatalytic reactions from visible to ultraviolet light irradiation. All the Ca-siloxene samples exhibited semi-conducting characteristics due to the Ca-extraction synthesis, in contrast to metallic CaSi_2 . According to the Tauc plots produced with the assumption of a direct transition (Fig. 1(B)), the absorption edges were increased with the Cl_2/Ca molar ratio during the synthesis step, and were determined to be 1.5, 2.9, and 3.1 eV for CS0.25, CS1.25, and CS2.5, respectively. Siloxene has layered 2D Si-based planes composed of $\text{Si}_6\text{O}_3\text{H}_6$ with Si_6 rings connected through Si–O–Si bridges.^{15,16} It has

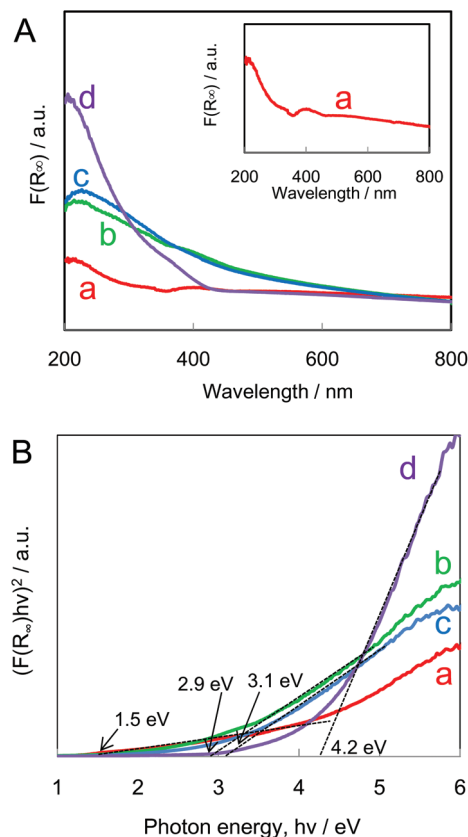


Fig. 1 (A) UV/visible spectra of Ca-siloxenes and (B) Tauc plots based on UV/visible spectral data: (a) CS0.25, (b) CS1.25, (c) CS2.5, and (d) Kautsky-type siloxene. Inset in (A): enlarged UV/visible spectra of CS0.25.

been reported that the siloxene structure exhibits a direct optical transition due to the presence of oxygen (O) atoms in Si planes with the σ - n mixing of Si σ electrons and O lone-pair electrons, unlike amorphous Si (a-Si) or polysilane structures.²⁵ The lowest unoccupied conduction band composed of Si planes is reduced by the electron withdrawing properties of O atoms, and the highest occupied valence band is formed by O lone-pair electrons, which leads to the broadening of the bandgap. In the case of Ca-siloxenes, the amount of O atoms increased with the extent of Ca extraction according to the Cl_2/Ca ratio during the synthesis step: $\text{Ca}_{1.8}\text{Si}_6\text{O}_{1.5}$ for CS0.25 and $\text{Ca}_{0.3}\text{Si}_6\text{O}_{2.8}$ for CS2.5.¹³ Therefore, it would be possible to control the bandgap energy of the Ca-siloxene according to the amount of incorporated O atoms. The optical band gap energy of the Kautsky-type siloxene was determined to be 4.2 eV, which was larger than those previously reported (2.5–2.9 eV) in the literature.^{16,25} The measurements in the previous studies were typically conducted in light of less than 4.13 eV (300 nm) because these Si-based materials were considered to primarily absorb light below *ca.* 4 eV.¹³ However, the previous study showed that similar Si-based materials could absorb light with higher photon energies of up to *ca.* 6 eV.¹² The density of states generally show a sharp increase at photon energies



higher than that of the absorption edge; however, density of states with slight tailing at lower energy are often observed in the case of doped materials or impurity fluctuations. The adsorption edge is derived by the extrapolation of the linear region of the Tauc plots observed at higher photon energy. Thus, photon energies of up to 6.20 eV (200 nm) were used for clarification in the present study, which corresponds to a widening of the measured energy by *ca.* 2 eV, and the impact at lower energy was reduced. The optical band gap energy observed in this study for Ca-siloxene and the Kautsky-type siloxene would be more accurate than those of previous studies. The UV/visible spectra of the Ca-siloxenes also contain broad tailing at less than *ca.* 4 eV, especially in CS1.25 and CS2.5, in contrast to the conventional Kautsky-type siloxenes. One of the reasons for this Urbach tail feature may be due to the amorphous structure of Ca-siloxenes with inconsistent particle size and composition.²⁶

Fig. 2(A) shows the electrical conductivity of Ca-siloxenes with Ca-extraction to various extents. The conductivity is almost inversely proportional to the extent of Ca-extraction in the Ca-siloxenes, and the conductivity of CS0.25 was 0.64 mS cm⁻¹, which was the highest electrical conductivity of all the Ca-siloxenes examined. The conductivities of CS1.25 and CS2.5 were *ca.* 0.5 and 0.05 μS cm⁻¹, respectively. Therefore, some conductive materials, such as metal particles¹² and carbon nanotubes,²⁷ may assist the construction of an electron pathway in the photocatalytic reaction that inhibits the recombination of photogenerated electrons and holes, and that improves catalytic activity.^{12,27} The electrical conductivity of the mixture of Ca-siloxenes and AB as a conductive material is shown in Fig. 2(B). The conductivity was increased with the amount of the AB additive and reached 0.1 S cm⁻¹ at 10 wt% or more. Fig. 2(C) shows the SEM image of CS0.25 with 10 wt% of AB. Almost consistent mixture of Ca-siloxene secondary particles with a relatively larger

size and AB secondary particles with less than 1 μm diameter was observed, which reflect the increased conductivity shown above. Thus, 10 wt% of AB seems to be sufficient to cover almost all surfaces of the Ca-siloxene particles. The effect of the addition of conductive materials on the increased activity for NO removal is investigated in the next section.

3.2. Photocatalytic NO removal by Ca-siloxenes

Fig. 3(A) shows the NO conversion over Ca-siloxenes under UV light irradiation ($\lambda > 290$ nm (<4.28 eV)). Photocatalytic NO

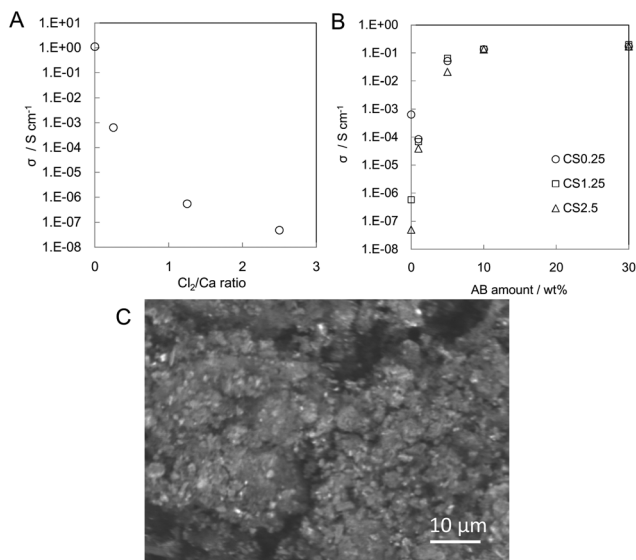


Fig. 2 Electrical conductivity of (A) Ca-siloxenes as a function of Cl₂/Ca ratio, (B) the mixture of Ca-siloxene and acetylene black (AB) as a function of AB amount, and (C) SEM image of CS0.25 with 10 wt% of AB.

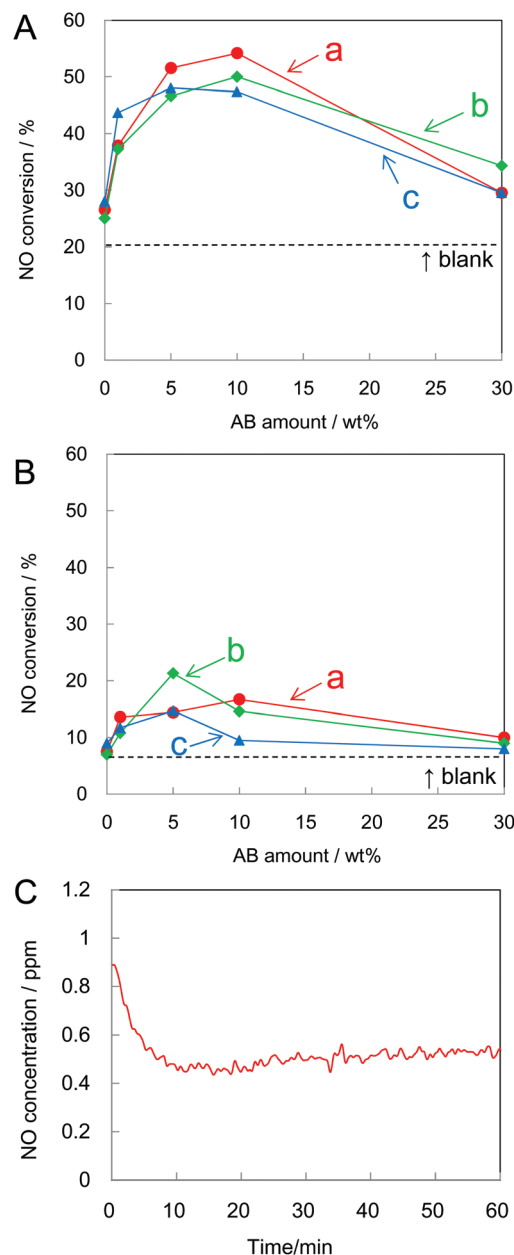


Fig. 3 NO conversions of Ca-siloxenes irradiated at (A) $\lambda > 290$ nm and (B) $\lambda > 400$ nm as a function of acetylene black amount: (a) CS0.25, (b) CS1.25 and (c) CS2.5, and (C) NO conversion of CS0.25 with 10 wt% of acetylene black as a function of irradiation time at $\lambda > 290$ nm.



removal was observed with all the Ca-siloxene samples, and NO conversion was affected by the mixing of AB with Ca-siloxenes in all the samples. The removal of NO is expected to form N_2 and HNO_3 , as often reported with TiO_2 catalysts, with the aid of a small amount of water in air.^{24,28,29}

NO conversion was increased with the amount of AB up to 10 wt%, and the maximum NO conversion was observed for the photocatalysts containing 10 wt% AB. This tendency corresponds to the electrical conductivity of the Ca-siloxene and AB mixtures. The low electrical conductivity of Ca-siloxene may cause the recombination of photogenerated electrons and holes, which results in low photocatalytic performance. As discussed in section 3.1, all Ca-siloxene particles were brought into contact with the addition of 10 wt% AB, and therefore the highest photocatalytic performance was achieved. On the other hand, the NO conversion of all samples mixed with 30 wt% AB showed a decrease in photocatalytic performance. AB itself has no photocatalytic activity and thus the decrease in the amount of Ca-siloxene in the samples corresponds to the decrease in NO conversion over the samples with 30 wt% AB. The best photocatalytic performance was observed in CS0.25 with 10 wt% AB addition. One reason for this performance was caused by the largest electric conductivity of CS0.25 in all Ca-siloxenes.

Fig. 3(B) shows the NO conversion of Ca-siloxenes under visible light irradiation ($\lambda > 400$ nm (<3.10 eV)). Samples mixed with 2.5–10 wt% AB exhibited clear photocatalytic activity for NO removal, although NO conversion under visible light irradiation was less than that under UV light irradiation. The absorption in the visible light region is present in all Ca-siloxenes, as shown in Fig. 1; however, the intensity was not as high as that observed in the UV light region, which resulted in the relatively lower photocatalytic activity.

To confirm the activity change, Fig. 3(C) shows the NO conversion of CS0.25 with 10 wt% AB under UV light irradiation for 60 min. The photocatalytic activity was almost constant, and an obvious decrease in activity was not observed, which supports the stability of Ca-siloxenes during the photocatalytic reaction.

The apparent quantum efficiency of samples mixed with 10 wt% AB was roughly estimated in Fig. S2.† The apparent quantum efficiency under UV light irradiation ($\lambda > 290$ nm) was around 0.0065%, which is almost 1/4 of that found in TiO_2 species under similar experimental conditions using a mercury lamp.²⁴ The apparent quantum efficiency under visible light irradiation ($\lambda > 400$ nm) was around 0.0017% for CS2.5 and 0.0028% for CS0.25.

For photocatalytic NO removal using TiO_2 , oxygen radicals ($\cdot O_2$) are formed from O_2 at the conduction band and hydroxyl radicals ($\cdot OH$) are formed from H_2O at the valence band.²³ 1O_2 has also been known as one of the active oxygen species formed by the oxidation of $\cdot O_2$ through electron transfer at photogenerated holes during photocatalytic oxidation,³⁰ and 1O_2 is also known to have higher energy than ground-state triplet oxygen for the enhancement of photocatalytic NO removal.²⁸ 1O_2 is detectable from the typical dimol emissions

at around 634 and 703 nm ($^1O_2 + ^1O_2 \rightarrow 2^3O_2 + h\nu$), and from the monomol emission at 1270 nm ($^1O_2 \rightarrow ^3O_2 + h\nu$).²⁸ The phosphorescence decay profile at 634 nm using a cut-off filter after UV light irradiation for 5 s was measured and the results are shown in Fig. 4. The Ca-siloxenes, especially CS0.25 and CS 2.5, exhibited phosphorescence decay derived from 1O_2 after UV light irradiation. The mixture of Ca-siloxenes and AB as a conductive additive may facilitate the oxidation of photo-generated $\cdot O_2$ to 1O_2 at available holes due to the inhibition of the recombination of photogenerated electrons and holes, as reported for nanocomposite powders of Si nanoflakes and metallic particles.¹² Therefore, it is suggested that the same active oxygen species as proposed for photocatalysis with TiO_2 is also formed with Ca-siloxenes, which leads to N_2 and HNO_3 formation in the photocatalytic removal of NO.

3.3. Durability test

The durability tests of CS0.25 with 10 wt% of AB were conducted, and the results are shown in Fig. 5. Photocatalytic activity was almost maintained through the cycles, except for the initial deterioration during the first cycle. A rapid decrease in photocatalytic activity with degradation and deactivation through hydrogen release by hydrolysis has been reported for photocatalysis with siloxene in water.³¹ In the present experiment, such deactivation and fading performance was not observed, although the Ca-siloxene was exposed to air with a small amount of water during this photocatalytic reaction. Furthermore, siloxene has been reported to gradually form dangling bonds under ultraviolet light irradiation associated with structural degradation;²⁶ however, photocatalytic activity was almost maintained in the Ca-siloxenes, which implies that structural stability is due to the Ca-bridging of siloxenes. This is required for maintained photocatalytic activity in contrast to the siloxene structure with no bridges connecting the layered Si planes except for van der Waals forces, which allows unexpected cleavage between planes. The initial deterioration at the first cycle is explained as the adsorption of HNO_3 on the sample surface, which is formed as a product during the NO

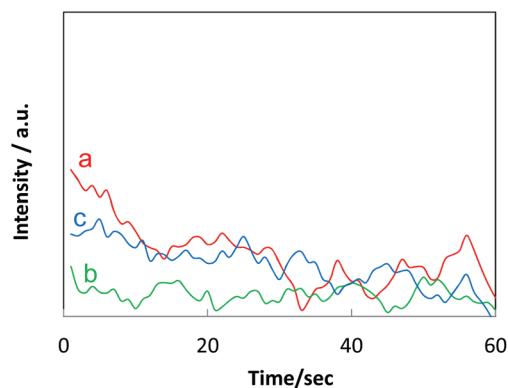


Fig. 4 Phosphorescence decay profile at 634 nm of 1O_2 generated at 298 K in air after 5 s of UV light irradiation: (a) CS0.25, (b) CS1.25, and (c) CS2.5.



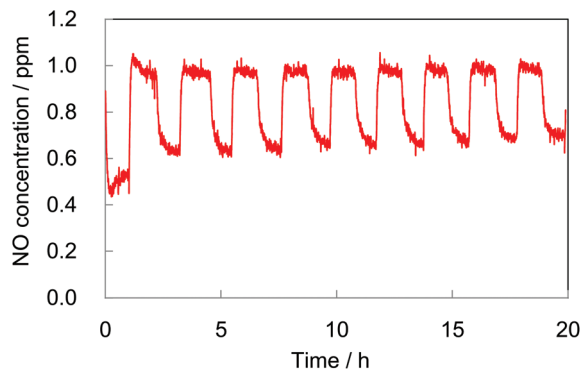


Fig. 5 NO concentration profiles of CS0.25 during the durability test under the irradiation of the ultraviolet light region ($\lambda > 290$ nm).

removal reaction, which is known to lead to catalyst poisoning in TiO_2 .^{32,33} Therefore, the observed initial deterioration is considered to be similar poisoning, although the Ca-siloxenes are not influenced further by HNO_3 after the initial poisoning.

4. Conclusions

Ca-Siloxenes with two-dimensional siloxene planes formed by Ca-bridging were synthesized, and their optical and electrical properties, and photocatalytic performance for NO removal were investigated. The Ca-siloxenes exhibited a wide and tunable optical absorption band from the visible to ultraviolet region with bandgap energies from 1.5 to 3.1 eV. All the Ca-siloxenes exhibited photocatalytic activity for NO removal under visible to ultraviolet light irradiation, which was enhanced by mixing the Ca-siloxene with AB as a conductive material, probably due to the inhibition of the recombination of photogenerated electrons and holes. The phosphorescence decay profile of the Ca-siloxenes suggested the presence of $^1\text{O}_2$ as an active oxygen species that enhances photocatalytic NO removal.

Conflicts of interest

There are no conflicts to declare.

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