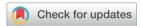
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Solid state p-type dye sensitized NiO-dye-TiO₂ core-shell solar cells†

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Solid state p-type dye sensitized NiO-dye-TiO2 core-shell solar cells with an organic dye PB6 were successfully fabricated for the first time. With Al₂O₃ as an inner barrier layer, the recombination process between injected holes in NiO and injected electrons in TiO2 was significantly suppressed and the charge transport time was also improved.

p-Type dye sensitized solar cells (p-DSCs) have attracted intense interest due to their different charge transfer kinetics with respect to more common studied n-DSCs, 1-3 and the potential application in tandem solar cells⁴⁻⁶ and solar fuel devices.⁷⁻⁹ The conventional p-DSCs are based on liquid redox electrolytes. To avoid having a liquid phase in the p-DSCs, we have recently proposed and proven the concept of solid state p-DSCs, in which a solid state phenyl-C61-butyric acid methyl ester (PCBM) was used as an electron transport material (ETM) between the dve sensitized photocathode and the back contact. 10 Optimization of photosensitizer represents one strategy to improve the performance of this kind of solar cells. 11,12 Inspired by the conventional dye sensitized TiO2 solar cells, we proposed that TiO2 should be an alternative ETM to PCBM due to the fast electron injection from dyes into TiO₂^{13,14} and good electron transport property.¹⁵ Thus, we recently fabricated a dye sensitized NiOdye-TiO₂ core-shell film and, in contrast to previous work, ¹⁶⁻¹⁸ the nanoporous NiO film was first sensitized with the dye and a TiO₂ coating was applied afterwards. This dye sensitized coreshell film showed ultrafast hole ($t_{1/2}$ < 120 fs) and electron injection into NiO and TiO2 respectively, resulting in ultrafast dye regeneration upon electron injection, $t_{1/2} \leq 500$ fs.¹⁹ In the present work, we proved that the dye sensitized NiO-dye-TiO2 core-shell mesoporous film can be used for fabrication of solid

Fig. 1 shows the configuration and working principle of the proposed p-type dye sensitized NiO-dye-TiO₂ core-shell solar cells. The donor- π -acceptor dye PB6 was utilized as photosensitizer since its reduction/oxidation potentials in excited state match with the valence band of NiO and conduction band of TiO2. 19

The fabrication of the photoelectrode used in this study can be described briefly as follows. A compact NiO layer (60 nm) was sputtered onto a FTO substrate, then a mesoporous NiO layer (1.3 µm) was prepared by doctor-blading NiCl₂ gel on FTO glass and sintered at 450 °C for 0.5 h.20 Subsequently, the NiO electrode was immersed into 0.2 mM PB6 dichloromethane (DCM) solution overnight. The dye loading in the film was determined to be 31.6 nmol cm⁻² by desorption experiment (see ESI†). After rinsing with methanol, the dye sensitized NiO film was dried and coated with metal oxides by atomic layer deposition (ALD). For NiO-PB6-TiO2 photoelectrode, ca. 10 nm TiO₂ layer was coated directly on a dye sensitized NiO film. Because an insulating Al₂O₃ layer has been proved to be an effective barrier to suppress charge recombination process in DSCs, ^{21–24} an 1 nm inner Al₂O₃ layer was introduced by ALD on the dye sensitized NiO film, followed by deposition of ca. 10 nm TiO₂ coating to form NiO-PB6-Al₂O₃-TiO₂ photoelectrode.

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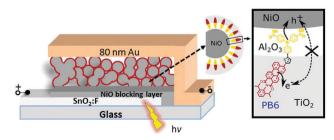


Fig. 1 The configuration and working principle of the p-type dye sensitized NiO-dye-TiO₂ core-shell solar cell, NiO-PB6-Al₂O₃-TiO₂, with Al₂O₃ as an inner barrier layer and PB6 dye as photosensitizer.

state p-type dye sensitized solar cells. We also show the effect of an Al₂O₃ inner barrier layer between NiO and TiO₂ on the performance of solar cell.

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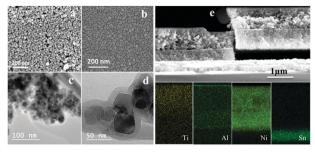


Fig. 2 SEM and TEM figures of the NiO-PB6-Al₂O₃-TiO₂ sample. (a and b) SEM surface images before and after ALD; (c and d) TEM images in different magnification; (e) SEM image in the cross section and the corresponding EDX element mapping analysis.

In addition, we prepared two reference samples without dye sensitization, denoted NiO-TiO₂ and NiO-Al₂O₃-TiO₂, respectively, to exclude the potential photovoltaic performance from the excitation of the p-n junction formed at the interface of NiO and TiO₂.

In previous work, we have shown that the TiO₂ layer penetrated into the dye sensitized mesoporous NiO electrode. In order to monitor the penetration of Al₂O₃ into the dye sensitized mesoporous NiO electrode, as well as TiO2 into the Al2O3 coated electrode afterwards, the morphology of NiO-PB6-Al₂O₃-TiO₂ photoelectrode with 1 nm Al₂O₃ (calculated from the average ALD deposition rate from multi-cycles) was characterized by SEM and TEM (Fig. 2). A more compact surface (Fig. 2b) was observed after subsequent ALD of Al2O3 and TiO2 compared to the mesoporous dye sensitized NiO film (Fig. 2a). From TEM images in Fig. 2c and d, one can note that NiO nanoparticles are coated by a conformal layer, which is assigned to TiO₂ and Al₂O₃ layers. From EDX elements analysis images of the cross section of the electrode, the presence of Al and Ti inside the NiO mesoporous layer was confirmed, signifying that Al₂O₃ can penetrate into the mesoporous NiO film and that the thin Al₂O₃ layer does not block the TiO₂ penetration.

In order to confirm the potential application of these coreshell photocathodes in solar cells, an 80 nm Au layer was thermally evaporated on the top of these electrodes as back contact to complete the solar cell. The performance of such solar cells was characterized under AM 1.5 G illumination with light intensity of 100 mW cm⁻² and the *J-V* curves are shown in Fig. 3a. A short-circuit current density (J_{sc}) of 12 μ A cm⁻² and an open-circuit voltage (V_{oc}) of 0.40 V were obtained in NiO-PB6-TiO₂ solar cell. The device based on NiO-TiO₂ film without PB6 dye was also fabricated and showed photovoltaic effect, resulting from the p-n junction of NiO-TiO₂ due to the direct contact between NiO and TiO2. From Incident Photon-to-Current Efficiency (IPCE) spectra (Fig. 3b), we can note that the PB6 dye in NiO-PB6-TiO₂ solar cell has light response in wavelength from 480 nm to 700 nm, which is in agreement with the dye absorption spectrum (Fig. S2, ESI†), proving that the photocurrent is generated by dye. IPCE curve of the NiO-TiO2 solar cell abruptly decreases to zero at wavelengths higher than 500 nm (2.48 eV), and only small part of the visible light can be converted into photocurrent in these solar cells. By using 1 nm Al₂O₃ to coat NiO, the NiO-TiO₂ p-n junction structure in NiO-Al2O3-TiO2 was inhibited and the solar cell did

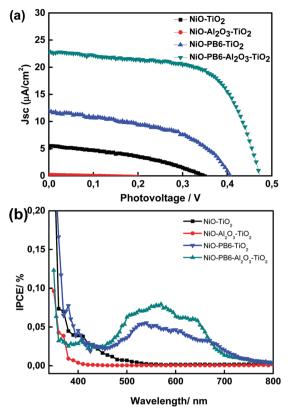


Fig. 3 J-V curves and IPCE of different dye sensitized p-type core—shell solar cells and reference solar cells.

not render any distinguishable photocurrent from J-V and IPCE characterization (see Fig. 3a and b). The same strategy was adopted in a NiO-PB6-Al₂O₃-TiO₂ solar cell, in which Al₂O₃ could also act as a blocking layer to suppress the charge recombination between injected holes from dye in NiO and injected electrons from dye in TiO₂. As the molecular extension of the PB6 dye was estimated to be ca. 2.5 nm, a 1 nm Al₂O₃ layer should not completely cover the electron donor unit of the PB6 dye, assuming the standing dye configuration on the metal oxide surface. It should provide an opportunity for the TiO₂ layer to fully cover the dye acceptor unit and collect the electrons. Noticeably, the photovoltaic performance of the NiO-PB6-Al₂O₃-TiO₂ solar cell was significantly improved compared to the device without Al_2O_3 , showing a J_{sc} of 23 μ A cm⁻² and a $V_{\rm oc}$ of 0.48 V, and higher IPCE values in visible region. Meanwhile, a fill factor (FF) of 66% was achieved in the NiO-PB6-Al₂O₃-TiO₂ solar cell, which is the highest FF reported³ in p-type DSCs to our best knowledge (Table S1, ESI†).

To further elucidate the electron and hole injection in ${\bf NiO-PB6-Al_2O_3-TiO_2}$ electrode, femtosecond transient absorption spectroscopy was carried out (Fig. S3, ESI†). A pronounced Stark effect was observed upon light excitation, resulting from dye regeneration after hole and electron injection from dye into NiO and ${\bf TiO_2}$ respectively. The dye regeneration time in ${\bf NiO-PB6-Al_2O_3-TiO_2}$ is $t_{1/2} \leq 500$ fs. These are consistent with our previous study of ${\bf NiO-PB6-TiO_2}$, ¹⁹ signifying that the ${\bf Al_2O_3}$ layer essentially did not affect the dye regeneration rate. Meanwhile, the same kinetic behaviour as ${\bf NiO-PB6}$ reported

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(a) 10³

SE 10²

---NiO-PB6-TiO₂

---NiO-PB6-Al₂O₃-TiO₂

10⁰
0.1 0.2 0.3 0.4 0.5 0.6 0.

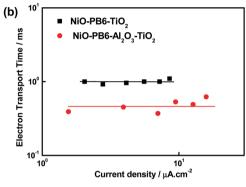


Fig. 4 Electron lifetime as a function of $V_{\rm oc}$ and transport time as a function of photocurrent density of NiO-PB6-Al₂O₃-TiO₂ and NiO-PB6-TiO₂.

in our previous study¹⁹ was found in NiO-PB6-Al₂O₃ (Fig. S4, ESI \dagger), further confirming that the Al₂O₃ layer did not significantly influence hole injection.

In order to gain insight into the effect of the Al_2O_3 layer on the charge recombination process in solar cells, charge lifetimes as a function of $V_{\rm oc}$ of the solar cells were measured (Fig. 4a). It shows that NiO-PB6-Al $_2$ O $_3$ -TiO $_2$ solar cell has longer charge lifetime than NiO-PB6-TiO $_2$ solar cell at a given $V_{\rm oc}$ value. From charge extraction experiments (Fig. S5, ESI †), two types of solar cells show almost identical extracted charges at a specific $V_{\rm oc}$, implying that the Al $_2$ O $_3$ layer does not influence the energy band position in either NiO or ALD TiO $_2$. These results suggest that the Al $_2$ O $_3$ layer indeed plays the role of an inner barrier layer in suppressing charge recombination between NiO and TiO $_2$, thus contributing to the enhanced $V_{\rm oc}$ and increased $J_{\rm sc}$.

Interestingly, the charge transport time (Fig. 4b) of the ${\bf NiO-PB6-Al_2O_3-TiO_2}$ solar cell is shorter than that of the ${\bf NiO-PB6-TiO_2}$ solar cell. We suggest that the ${\bf TiO_2}$ formed in the presence of ${\bf Al_2O_3}$ is more compact due to the more hydrophilic properties after ALD ${\bf Al_2O_3}$ (the existence of –OH units on the ALD ${\bf Al_2O_3}$ surface). ${\bf TiO_2}$ of better quality is expected to be grown on the more hydrophilic surface during ALD in contrast to the NiO surface sensitized with an organic dye, ²⁵ although the amorphous structure in both samples is confirmed by TEM. The shorter charge transport time in ${\bf NiO-PB6-Al_2O_3-TiO_2}$ solar cell should be responsible for higher $J_{\rm sc}$ and higher fill factor as compared to the ${\bf NiO-PB6-TiO_2}$ solar cell. Notably, the charge transport times of both types of solar cells are almost independent of the photocurrent (light intensity), which is unusual for the conventional n-type and p-type DSCs.

One possible explanation is that the TiO₂ phase is not continuous inside the mesoporous NiO film (1.3 µm), resulting in that only a very thin NiO film coated by ALD TiO₂ works for the charge collection. Another explanation would be that holes and/or electrons transferred *via* the semiconductor surfaces.²⁸ The former one can also be applied to explain why the photocurrent is much lower than the conventional p-DSCs, although the dye regeneration in the core–shell solar cell is much faster¹⁹ and the charge lifetime is comparable to those p-DSCs.^{26,29} Therefore, making continuous TiO₂ layer or increasing the pore-filling of TiO₂ inside mesoporous NiO could be a reasonable strategy to significantly improve the efficiency of such solar cells.

As mentioned above, the length of PB6 dye is ca. 2.5 nm, which provides us an opportunity to study the effect of the thickness of Al₂O₃ on the performance of the solar cells. In principle, a too thin Al₂O₃ layer cannot effectively suppress charge recombination loss and render unsatisfactory performance. Conversely, a too thick Al₂O₃ layer could completely bury the dye and the TiO2 cannot sufficiently contact with the acceptor of the dye, which will induce unsatisfactory performance as well due to low regeneration efficiencies. There should be an optimal thickness of Al2O3 giving an optimized performance. Different thicknesses of Al₂O₃ (0.08, 0.5, 1, 2 and 10 nm) were therefore investigated in NiO-PB6-Al₂O₃-TiO₂ solar cells. From Fig. 5, ca. 1 nm of Al₂O₃ was proven to be the optimal thickness, because it only covers the entire electron donating unit, triphenylamine (TPA) of PB6, and allows the whole electron acceptor part (pervlene monoimide (PMI)) buried inside TiO2, which is expected to be the optimal configuration for charge separation between dye and semiconductors.

With the Al_2O_3 layer thinner than 1 nm, less photocurrent and photovoltage was observed, implying more extensive recombination in these devices. With 2 nm Al_2O_3 film, the photocurrent from the corresponding device also significantly decreased, suggesting that the charge separation probably is inefficient. The hypothesis can be further supported from the results utilizing a 10 nm Al_2O_3 layer, resulting in solar cells without any obvious photocurrent likely to be caused by the absence of contact between dye and TiO_2 .

In summary, solid state p-type dye-sensitized NiO-dye-TiO₂ core-shell solar cells have been proposed and fabricated for the first time, in which the NiO is sensitized by the organic PB6 dye, after which a TiO2 layer is deposited by ALD to form the dye sensitized NiO-dye-TiO2 core-shell structure. By inserting a thin insulating Al₂O₃ layer, the charge recombination between NiO and TiO₂ after charge injection from dye was significantly retarded. On the basis of the molecular extension of dye, the effect of Al₂O₃ layer thickness on the solar cell performance was further investigated, showing an optimal thickness of ca. 1 nm for the PB6-based devices. The optimal Al₂O₃ thickness should be different from different dyes, which encourages us to apply the strategy on other dye systems in future study. Furthermore, other n-type metal oxides can be considered as electron transport materials instead of TiO₂ in the core-shell solar cells. This type of solar cell holds the potential to rival conventional liquid p-type dye sensitized solar cells after systematic studies and

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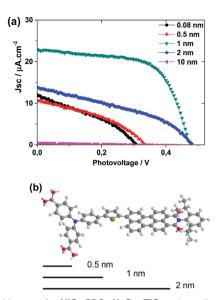


Fig. 5 (a) J-V curves for NiO-PB6-Al₂O₃-TiO₂ solar cells with different Al₂O₃ thickness layers; (b) the 3-D molecular structure of PB6 with 0.5 nm, 1 nm and 2 nm length scales as comparison.

optimization of different components, and may show potential for application in solar fuel devices as well.

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Conflicts of interest

The authors declare no competing financial interests.

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